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PAPER

Controlled surface mineralization of metal oxides on nanofibers

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We report a versatile approach for the preparation of metal oxide/polymer hybrid nanofibers by in situ formation of metal oxide nanoparticles on surface-functionalized polymer fibers. Poly (styrene-*co*-vinylphosphonic acid) fibers were produced by electrospinning and used as supports for the in situ formation of ceria nanocrystals without further thermal treatment. The crystallization of ceria was induced by the addition of an alkaline solution to fibers loaded with the corresponding precursor. The formation of the inorganic material ¹⁰ at the fiber surface was investigated by transmission electron microscopy (TEM) and X-ray photoelectron spectroscopy (XPS). The extension of the approach to prepare polymer/titania hybrid nanofibers demonstrates its versatility.

1. Introduction

In recent years, materials based on transition metal oxides ¹⁵ have played an important role in the development of novel semiconductor materials for a wide range of applications. In this work, we have selected two representative metal oxides: cerium (IV) oxide (ceria) and titanium (IV) oxide (titania). Ceria is a rare-earth oxide material employed as ultraviolet absorbent, gas ²⁰ sensor, photocatalyst, and electrolyte for fuel cells.¹ Titania finds applications in photocatalysis, solar energy conversion, sensors, and mesoporous membranes.²

A large surface area of the metal oxide, which is needed for many applications, can be achieved by reducing the size of the ²⁵ materials to the nanometric range. However, the high surface energy of nanoparticles causes aggregation during synthesis or post-synthesis processes. To overcome this major issue, support materials are often used to hinder the aggregation of the nanoparticles. Polymer particles have been applied as supports, ³⁰ coating materials, and structure-directing agents to provide heterogeneous nucleation and to control the growth of inorganic materials. Several works have reported the crystallization of metal oxides on the surface of polystyrene particles with grafted anionic or cationic polyelectrolyte brushes. TiO₂ nanoparticles 35 were immobilized on polystyrene cores with long chains of poly(styrene sodium sulfonate).^{3, 4} Mixed valent manganese oxide 2- (MnO_x) nanoparticles were stabilized by (trimethylammonium)ethyl methacrylate chloride.⁵ Poly(styreneco-acetoxyethyl methacrylate) latex particles were also utilized $_{40}$ for the in situ crystallization of ZnO₅⁶ Ta₂O₅⁷ TiO₂⁸ and $In(OH)_3^9$ in alcoholic solvents. Recently, Fischer *et al.*¹⁰ synthesized specifically designed surface-active monomers (socalled surfmers) and applied the surfmer-functionalized polymer particles as nucleation surfaces for the controlled surface 45 precipitation of CeO₂, *α*-Fe₂O₃, Fe₃O₄, and ZnO nanoparticles from both alcoholic and aqueous media.

Polymer fibers can also be used as supports to place inorganic particles in a structural template in an isolated and nonaggregated fashion.^{11, 12} The advantage of fiber networks over ⁵⁰ nanoparticles is that they can be easily separated from the reaction media. The free-standing non-woven possesses large surface area and high porosity that enables surface modification or incorporation of particles in the fibers. In an analogous way to the achievement with polymer particles, tailor-made surface ⁵⁵ chemistry on the fibers can provide nucleation centers for controlled crystallization. The methods for fabricating functional metal oxide fibers can be classified in three groups: (i) electrospun metal oxide fibers by high temperature heat treatment (calcination);¹³⁻¹⁶ (ii) electrospinning polymer solution containing metal oxide nanoparticles;¹⁷⁻²⁰ and (iii) metal oxide coating of polymer fibers.²¹⁻²⁴ Although calcination of polymer/metal oxide

- ⁵ precursor fibers is the most common method, fibrous membranes after this process are typically brittle and lose their mechanical integrity.²⁵ In the second case, metal oxide nanoparticles are embedded in polymer fibers that reduce the performance of the available active sites due to surface coverage.^{26, 27} Thus, the
- ¹⁰ fabrication of organic/inorganic hybrids by the in situ formation of metal oxide nanoparticles on the surface of electrospun fibers can be suitable for fabricating hybrid materials, serving also at the same time as an interesting model to understand mineralization processes. This in situ crystallization method
- ¹⁵ provides on one hand the possibility of synthesis at lower temperature without any additional heat treatment, and on the other hand a controlled crystallization on the surface of the non-woven. The successful in situ synthesis of metal oxide nanoparticles on the surface of polymeric fibers, while
 ²⁰ maintaining the properties of the polymer template, remains very challenging, and only very few related works have been reported.^{21-24, 28, 29}

Surface treatment of electrospun fibers can prevent particle aggregation and ensure homogeneous distribution of particles on ²⁵ the fibers. Drew *et al.*²¹ reported the fabrication of electrospun polyacrylonitrile (PAN) nanofibers coated with metal oxides (TiO₂ and SnO₂) by liquid phase deposition. The formation of the metal oxide was driven by the immersion of the PAN fibrous membrane in an aqueous solution of the corresponding metal

- ³⁰ halide salts and halogen scavengers. The same group also reported that PAN nanofibers treated with hot sodium hydroxide could be functionalized with carboxylic groups, which induced fast nucleation of metal oxide growth on the fiber surface.²⁸ Following a different strategy, Nagamine and coworkers²²
- ³⁵ fabricated hollow TiO₂ fibers by electrospinning an aqueous PEO solution into a tetraisopropoxide/hexane solution. Employing the same procedure, PVA–TiO₂ composite nanofibers were fabricated by modifying the surface nanostructure by alkaline and subsequent acid treatment.²⁹ Zhang *et al.*²⁴ presented a method
- $_{40}$ for the preparation of TiO₂ nanoparticles in electrospun PMMA. The method involved the electrospinning of a TiO₂ precursor/PMMA solution followed by a mild hydrothermal treatment to retain the properties of the polymeric template. Su

and Li^{23} prepared electrospun cellulose acetate fibers, and the ⁴⁵ synthesis of TiO₂ nanoparticles were achieved on the cellulose fibers by the interfacial interaction between the hydroxyl groups and hydrated Ti(OBu)₄.

Here, we present a versatile approach to crystallize in situ inorganic metal oxides, exemplified with the cases of ceria and ⁵⁰ titania, on the surface of electrospun poly(styrene-*co*vinylphosphonic acid) fibers. Thereby, it is suggested that the polymer offers stable and absorption sites for metal oxides at high concentrations.

2. Experimental section

55 2.1 Materials

Styrene (Merck, 99%) was purified by passing through a column packed with aluminum oxide (Fluka, chromatography). Vinylphosphonic acid (VPA, Aldrich, 97%) was washed with diethyl ether (Aldrich, 99.5%) to remove the inhibitor and dried on under vacuum. Sodium dodecyl sulfate (SDS, Carl Roth, 99%), 2,2'-azobis-(2-methylbutyronitrile) (V59, Wako Chemicals, 98%), hexadecane (Aldrich, 99%), cerium(III) nitrate hexahydrate (Fluka, ≥99.0%), titanium isopropoxide (TIP, Aldrich, 99.9%), sodium hydroxide (Riedel de-Haën, 99%), isopropanol (Aldrich, 99.8%), and *N*,*N*-dimethylformamide (DMF, Aldrich, 99.8%) were all used as received without any further purification. Deionized water (18.2 MΩ·cm) was used throughout the experiments.

2.2 Synthesis of phosphonate-functionalized

70 polystyrene by miniemulsion

Polystyrene (PS) and poly(styrene-*co*-vinylphosphonic acid) (P(S-*co*-VPA)) aqueous dispersions were synthesized by miniemulsion polymerization. A 0.3 wt% aqueous solution of the surfactant was prepared by dissolving SDS (72 mg) in water (24 ⁷⁵ g) and used as a continuous phase. The surfactant solution was added to a solution containing styrene (6 g), hexadecane (250 mg), and the initiator V59 (100 mg). Hexadecane was used as a costabilizer to suppress the Ostwald ripening and stabilize the miniemulsions against molecular diffusion. The use of the oil-⁸⁰ phase initiator V59 minimizes the formation of water soluble oligomers in the continuous phase.³⁰ The amount of VPA was varied (0, 1, 5, and 10 wt% based on the total monomer amount) for the synthesis of P(S-*co*-VPA). The resulting polymers are named **P**₀, **P**_{0.01}, **P**_{0.04}, and **P**_{0.10} respectively, with respect to the similar ratio of VPA. After pre-emulsification of the initial mixture

for 1 h at 1000 rpm, the miniemulsion was prepared by ultrasonication for 2 min at 90% amplitude (Branson Sonifier W-450D, ¹/₂ inch tip) under cooling with an ice–water bath. The miniemulsion was flushed with argon for 10 min to remove ⁵ oxygen. The polymerization was carried out at 72 °C for 12 h and

the polymers were isolated by freeze-drying.

2.3 Fabrication of the P(S-*co*-VPA)/metal oxide hybrid nanofibers

- A sample of PS or P(S-*co*-VPA) was dissolved in DMF by ¹⁰ stirring until complete dissolution was achieved. The solid content of the solution was 8 wt %. The viscous solution was loaded in a plastic syringe connected with silicone rubber tubing. The electrospinning experiments were carried out with a commercial platform (ES1a, Electrospinz). The positive electrode
- ¹⁵ was applied to the spinneret and either an aluminum foil or a metal paper clip was used as a grounded counter electrode. Metal paper clips were attached on the foil to prevent folding of the non-woven for the subsequent crystallization process. A potential of 12 kV was applied and the flow rate of the polymer solution
- ²⁰ was kept at 1.5 mL/h by using a syringe pump (Bioblock, Kd Scientific). The tip-to-collector distance was 14 cm and the electrospinning time was fixed to 3 min. The as-prepared electrospun fibers ($\mathbf{F_0}$, $\mathbf{F_{0.01}}$, $\mathbf{F_{0.04}}$, and $\mathbf{F_{0.10}}$) were dried for 12 h at 50 °C under vacuum to remove residual solvent.
- For the crystallization of the ceria nanoparticles on the polymer fibers, the nanofiber coated substrate was immersed into 10 mL of a 0.5 mM Ce(NO₃)₃·6H₂O aqueous solution for 3 h under constant stirring (300 rpm) at room temperature. The Ce³⁺loaded polymer non-woven was then separated from the solution.
- ³⁰ Then, 10 mL of a 1.0 mM NaOH solution was added dropwise with a rate of 1 mL/h under constant stirring at 300 rpm. The resulting hybrid fiber was washed with deionized water at least three times and dried under vacuum at 40 °C for 14 h.
- To conduct the formation of titania, the electrospun ³⁵ nanofiber-coated substrate was immersed into an isopropanol/H₂O mixture (200:1, v/v) under constant shaking (300 rpm) at 40 °C. An isopropanol/titanium isopropoxide mixture (100:1, v/v) was prepared under argon atmosphere and 5 mL of the mixture was added dropwise with a rate of 5 mL/h to
- ⁴⁰ the solution containing non-woven. The resulting hybrid fiber was isolated from the solution and dried under vacuum at 40 °C for 14 h.

Characterization Methods. ³¹P-NMR spectra were recorded with a Bruker AVANCE II spectrometer. The ³¹P NMR (202 ⁴⁵ MHz) measurements were obtained with the inverse gated decoupling technique³¹ (30° degree flip angle was used), which had a 11 µs long 90° pulse for phosphorus with a relaxation delay of 5 s and 140 scans for the highest and 600 scans for the lowest concentration. All NMR spectra were measured in DMF-d₆ using ⁵⁰ triphenylphosphine (TPP) as an internal standard.

The fibers between the clips were used for morphological observations by scanning electron microscopy (SEM) in a Hitachi SU8000 microscope. The diameter of fibers was statistically estimated from SEM micrographs measuring over more than 50 fibers byusing the software Fiji/ImageJ. Thermogravimetric analysis (TGA) was carried out with a Mettler Toledo 851 thermobalance under nitrogen atmosphere and heating rate of 10 °C/min.

The cross-sectional transmission electron microscopy (TEM) ⁶⁰ images of the polymer/metal oxide hybrid nanofibers were captured in a FEI Tecnai F-20 microscope. The nanofibers were embedded into epoxy and cut into ultrathin slices using an ultramicrotome (Sorvall MT 5000).

Surface compositions were determined by X-ray ⁶⁵ photoelectron spectroscopy (XPS) (PHI LS 5600) with a standard Mg-K α X-ray source. The energy resolution of the spectrometer was set to 0.8 eV/step at pass energy of 187.85 eV for survey scans. The X-ray beam was operated at a current of 25mA and an acceleration voltage of 13kV. Charge effects were corrected using 70 Carbon C1s = 284.5 eV. The determination of the concentrations of surface elements was performed using CasaXP software.

3. Results and discussion

Miniemulsion polymerization is an efficient process to synthesize functionalized polymers either through homopolymerization of a ⁷⁵ functional monomer or its copolymerization with another comonomer. This technique allows the copolymerization of monomers with different polarities such as the hydrophobic styrene with the hydrophilic vinylphosphonic acid.³² The phosphonic groups can react with a metal precursor and provides ⁸⁰ a simple way for the surface adsorption.

The general procedure for the fabrication of P(S-*co*-VPA) by miniemulsion polymerization and the electrospun fibers covered by metal oxide nanoparticles is depicted in Figure 1.

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Fig. 1 Illustration of miniemulsion polymerization and mechanism of metal oxide formation at the surface of phosphonate-functionalized polystyrene

²⁰ Firstly, polymer nanoparticles with vinylphosphonic acid functionalities on their surfaces were synthesized by miniemulsion polymerization. P(S-*co*-VPA) nanoparticles with sizes of 140 ± 20 nm were successfully synthesized (Supporting ²⁵ Information, Figure S1). The amount of phosphorus in the

- copolymer was quantified by ³¹P-NMR spectroscopy (Supporting Information, Figure S2). As expected, the content of phosphorus in the copolymers increased (from 0.9 to 9.7 mol%) with the initial concentration of vinylphosphonic acid in the
- ³⁰ miniemulsions (from 0.5 to 9.8 mol%) (Table 1). On the contrary, the apparent molecular weights decreased from 315,000 to 220,000 g mol⁻¹ when vinylphosphonic acid was incorporated (Supporting Information, Table S1). The electron-withdrawing nature of vinylphosphonic acid reduced the electron density of
- ³⁵ the double bonds of the benzene ring, but also of the growing radical CH=CH₂. Consequently, there may be no further growth

of the polymer chain, leading to low molecular weights copolymers.

Table 1 Amount of vinylphosphonic acid (VPA) in poly(styrene-*co*vinylphosphonic acid) P(S-*co*-VPA) and the metal oxide contents after crystallization process.

Sample	VPA in	VPA in	Sample	$Ce_2O_3^{c}$	Sample	TiO ₂
Name	copolymer ^a	copolymer ^b	Name	[wt %]	Name	[wt ° 🤇
	[mol %]	[mol %]				
P ₀	-	-	F ₀ C	4.3	F0T	5.4
$P_{0.01}$	0.9	0.5	F _{0.01} C	5.7	F1T	8
$P_{0.04}$	4.8	4.4	F _{0.04} C	7.8	F5T	6.5
P _{0.10}	9.7	9.8	F _{0.10} C	8.9	F10T	9.2

^aTheoretical content, ^bdetermined by ³¹P-NMR, ^cMeasured with TGA.

Poly(styrene-*co*-vinylphosphonic acid) fibers were obtained ⁴⁵ by electrospinning of solutions of the copolymers in DMF (see corresponding SEM micrographs in Supporting Information, Figure S3). The obtained fibers were randomly distributed on the clips and form a porous membrane. The porosity probably originated from the humidity in the electrospinning environment ⁵⁰ (~50%). It is known that vapor and DMF undergo liquid–liquid phase separation and the rapid evaporation of water can leave pores on the fibers surface.^{33, 34} The average diameter of the fibers decreased from 320 ± 60 to 270 ± 50 nm with an increasing content of vinylphosphonic acid (Supporting Information, Table 5 S1).

Because the phosphonate groups should play the role of nucleating centers, it is important to quantify their amount on the surface of the fibers. Indeed, the concentration of phosphonate on the fibers surface may not reflect exactly the concentration of

¹⁰ phosphonate in the copolymer because preferential orientation of hydrophobic groups on the surface of the fibers is possible. The elemental composition of the fiber surfaces was characterized by XPS (see atomic concentrations in Supporting Information, Table S2,). Not surprisingly, the O1s atomic concentration of the ¹⁵ P(S_{0.96}-*co*-VPA_{0.04}) fibers was higher than that of the P(S_{0.99}-*co*-VPA_{0.01}) fibers, indicating the presence of higher amount of phosphonic groups. This result suggests the successful incorporation of the VPA comonomer into PS system.

The metal oxide formation occured on the surface of the ²⁰ electrospun fibers by immersing into the cerium precursor solution. After a sufficient complexation time of the precursor with vinylphosphonic acid, the crystallization was induced by the addition of a precipitating agent sodium hydroxide at a controlled rate. In previous work of our group, we had shown that metal ²⁵ oxide nanoparticles are directly formed on functinalized surfaces of nanoparticles, rather than formed in the solution and merely adsorbed or heterocoagulated on the polymer surface. We assume an analogous formation mechanism at the surface of the fibers.

The morphology and structure of the hybrid fibers obtained ³⁰ from aqueous solutions of Ce(III) was investigated by SEM and TEM. Figures 2a and b show SEM micrographs at different magnifications of P(S_{0.90}-*co*-VPA_{0.10}) fibers after ceria crystallization. The presence of vinylphosphonic acid on the surface of the fibers resulted in an efficient coverage with the ³⁵ ceria nanoparticles, which can be correlated with the concentration of phosphonate functionalities on the fibers surface. Ceria nanoparticles formed more homogeneously on surface of the fibers with increasing VPA amount (Supporting Information, Figure S4). The content of the inorganic component in the hybrid

⁴⁰ fibers, determined by thermogravimetric analysis (TGA), ranged from 4 to 9 wt% (Table 1). Oxidation of cerium in the presence of neat polystyrene fibers led to bulk crystallization consisting of large ceria particles not only on the fiber surface but also in the space between the fibers. Nucleation on the fiber surface, with ⁴⁵ almost no bulk crystallization, occured even when using the $P(S_{0.99}$ -*co*-VPA_{0.01}) fibers with the lowest VPA content (Supporting Information, Figure S5).



Fig. 2 Electron micrographs of the electrospun poly(styrene-*co*-vinylphosphonic acid) P(S_{0.90}-*co*-VPA_{0.10}) fibers after ceria crystallization: (a) and (b) SEM micrographs at different magnifiacion, (c) TEM dark-field micrograph of a microtomed section the hybrid fibers, and (d) high-resolution TEM image.

Elemental analysis by energy dispersive X-ray (EDX) spectroscopy proved the presence of cerium and oxygen in the 70 hybrid fibers, which is attributed to the formation of ceria nanoparticles (see Supporting Information, Figure S6). X-ray diffraction (XRD) patterns of the samples were dominated by a typical amorphous halo from the polymer. However, the darkfield TEM image in Figure 2c demonstrates that ceria nanocrystals are present on the surface of the fibers. Interplanar distances of 0.191 nm measured from high-resolution TEM micrographs (Figure 2d) correspond to the {220} planes of cubic CeO₂.

Surface elemental compositions of the hybrid fibers were also ⁸⁰ investigated by XPS. The spectra, presented in Figure 3, show characteristic signals for cerium (Ce 3d at 887.5 eV and 905 eV), oxygen (O 1s at 535.1 eV), carbon (C 1s at 286 eV), and phosphorus (P 2p 133.5 at eV). The XPS spectrum from cerium splits into Ce3d_{3/2} and Ce3d_{5/2} with multiple shake-up and shake-⁸⁵ down satellites. The signals between 875 and 895 correspond to the Ce3d_{5/2} while signals between 895 and 910 eV belong to the $Ce3d_{3/2}$ levels.³⁵ The signal at 918 eV is a characteristic for the presence of cerium(IV).³⁶



Fig. 3 XPS spectra of the electrospun poly(styrene-*co*-vinylphosphonic s acid) P(S_{0.96}-*co*-VPA_{0.04}) fibers (a) before and (b) after ceria crystallization.

The absence of this band indicates that the oxide on the surface is mostly Ce₂O₃, which would be rather unusual and different from the CeO₂ obtained under very similar conditions ¹⁰ with latex particles.¹⁰ For the origin of Ce(III) oxide, two scenarios are possible: (i) since Ce₂O₃ is an intermediate in the formation of CeO₂, its occurrence could be explained by an incomplete crystallization;³⁷ (ii) a reduction of Ce(IV) to Ce(III), as reported by previous works.³⁸⁻⁴⁰ We cannot rule out any of the ¹⁵ options, but the second appear more plausible, taking into account that the formation of CeO₂ is thermodynamically favorable under our synthesis conditions.

XPS measurements also indicate that phosphorus is present on the fibers on which cerium was crystallized in concentrations ²⁰ slightly higher than those expected for the bulk (cf. Table S2). The concentration of oxygen atoms also increased after crystallization/washing. This result suggests reorganization of vinyl phosphonate at the molecular level during the crystallization process. Indeed, the polymer of the fibers surface ²⁵ experienced first the air as interface and then an aqueous medium

during the crystallization.

To show the versatility of our approach, we extended the method to titania, as a representative case of a sol-gel system. The surface of the fibers was decorated with titania by adding ³⁰ isopropanol/titanium isopropoxide mixture to the isopropanol/water mixture containing VPA-functionalized polystyrene nanofiber. Figure 4 shows SEM micrographs of the $P(S_{0.99}$ -*co*-VPA_{0.01}) fibers after titania crystallization.



35 Fig. 4. SEM micrographs of the electrospun poly(styrene-covinylphosphonic acid) P(S_{0.99}-co-VPA_{0.01}) fibers after TiO₂ crystallization: (a) general view, (b) higher magnification of the fiber surface.

Homogeneously distributed titania nanoparticles were present on 40 the surface of the fibers, as also supported by EDX measurements (Supporting Information, Figure S7). It needs to be noted that the formation mechanisms for ceria and titania are very different: while ceria crystallizes easily at room temperature, titania follow a conventional sol–gel precess and the formed material is 45 amorphous. Nevertheless, the results showed that the method can be applied for crystallization not only from aqueous but also from alcoholic media.

4. Conclusions

A template-assisted approach for the fabrication of polymer/metal ⁵⁰ oxide hybrid fibers is presented. Poly(styrene-*co*-vinylphosphonic acid) P(S-*co*-VPA) nanofibers were coated with ceria or titania nanoparticles at low temperature. Metal oxide nanoparticle formation could be enhanced by the introduction of functional phosphonate groups in the copolymer, which is the complexation ⁵⁵ group on the surface. Increasing the VPA concentration on the fibers surface increased the concentration of metal oxide particles on the fibers. This observation supported the role of surface VPA as nucleation centers along with fiber. The proposed synthesis strategy is simple and versatile and can be extended for the fabrication of a wide range of polymer/metal oxide hybrid membranes that are expected to provide highly reactive surfaces for enhanced catalysis, sensing, and photoelectric applications.

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Notes and references

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Electronic Supplementary Information (ESI) available: SEM micrographs of the polymer nanoparticles, and nanofibers before and after crystallization, EDX spectrum of the nanofibers after crystallization, ³¹P-

- 20 NMR spectra of the polymers, Tables showing the diameter of the nanofibers and molecular weight of the polymer, and atomic concentrations of the nanofibers before and after ceria crystallization. See DOI: 10.1039/b00000x/
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Metal oxide/polymer hybrid nanofibers are prepared by in situ formation of metal oxide nanoparticles on surface-functionalized polymer fibers.

