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A One-step Approach to the Large-scale Synthesis of Functionalized MoS₂ Nanosheets by Ionic Liquid Assisted Grinding

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A prerequisite for exploiting most proposed applications for MoS₂ is the availability of water-dispersible functionalized MoS₂ nanosheets in large quantities. Here we report one-step synthesis and surface functionalization of MoS₂ nanosheets by a facile ionic liquid assisted grinding method with the presence of chitosan. The selected ionic liquid with suitable surface energy could efficiently overcome the van der Waals force between the MoS₂ layers. Meanwhile, chitosan molecules bind to the plane of MoS₂ sheets noncovalently, which prevents the reassembling of exfoliated MoS₂ sheets and facilitates the exfoliation progress. The obtained chitosan functionalized MoS₂ nanosheets possess favorable stability and biocompatibility, which renders them as promising and biocompatible near-infrared agent for photothermal ablation of cancer. This contribution provides a facile way for the green, one-step and large-scale synthesis of advanced functionalized MoS₂ materials.

Introduction

Transition-metal dichalcogenides materials have garnered increased attention and have been intensely studied recently due to their unique structural, mechanical, electronic and optical properties, which originate from low dimensionality. As the most prominent example of these emerging materials, molybdenum disulfide (MoS₂) has received a fair share of attention in areas ranging from energy and catalysis to sensing.¹⁻⁴ Particularly, with higher absorbance in the near-infrared (NIR) region than that of both graphene and gold nanorods, MoS₂ nanosheets have been widely applied as an efficient photothermal agent for the photothermal therapy (PTT) of cancers. Moreover, with high specific surface areas and hydrophobic plane, MoS₂ nanosheets can highly efficiently deliver therapeutic molecules for the combined therapy of cancer, such as combined photothermal and chemotherapy, combined photothermal and photodynamic therapy, and imaging guided photothermal therapy.⁵⁻⁷

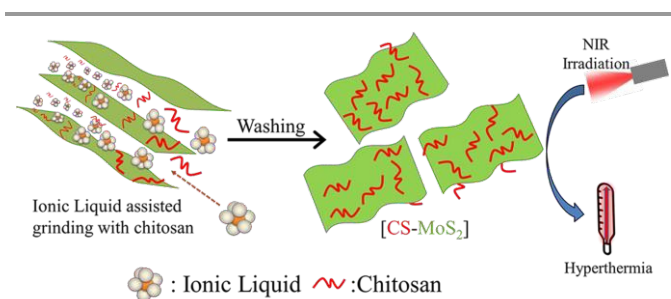
However, the exploitation of most proposed applications of MoS₂ has been hampered by the lack of a simple method for the availability of MoS₂ sheets in large quantities. In principle, the layered MoS₂ crystal is composed of hexagonal layers of Mo atoms sandwiched between two layers of S atoms covalently, arranged as three planes of atoms (S-Mo-S). Similar to graphite, the loosely stacking of adjacent sheets via van der Waals

interactions enables the formation of bulk crystal. Fortunately, the weak van der Waals forces between the layers are prone to be broken, dividing bulks into the layered compounds, and thus controllable exfoliation is an exclusive way to obtain mono- or few-layered MoS₂ through externally applied forces.^{8,9} To date, abundant efforts have been paid to the exfoliation of MoS₂ into individual layers, including micromechanical cleavage or the so-called “Scotch tape method”, intercalation-driven exfoliation, liquid-phase sonication exfoliation or grinding assisted liquid phase exfoliation, laser or plasma etching and electrochemical exfoliation. Of all these methods, the adhesive tape procedure, performed in most of the fundamental studies on the properties of the single and few layer MoS₂, however, is clearly incompatible with large-scale synthesis for practical applications.¹⁰ It has been known for many years that intercalation-driven exfoliation based on the intercalation of lithium ions (Li⁺) in the interlayer space of the bulk material can exfoliate layered MoS₂. However, due to structural deformation, this chemical exfoliation method results in the loss of the MoS₂ nanosheets' semiconducting properties and quenching of photoluminescence in 2D MoS₂.¹¹ Additionally, the lithium intercalation method is time-consuming, extremely sensitive to environment and not safe in laboratory.¹² A major breakthrough was made by the exfoliation of layered MoS₂ materials in various organic solvents via sonication to generate mono- or multilayer structures, which was initially proposed by Coleman co-workers in 2011 and then evolved by many scientists successively.^{13, 14} However, these procedures are incompatible with most solvents and require harsh solvents or expensive equipment with extremely time-consuming multiple steps, leading to environmentally unfriendly, unsustainable and defect-rich practices. Very recently, unconventional exfoliation

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methods with sophisticated equipment and extremely low yield by laser or plasma etching and electrochemistry also have been initiated.¹⁵⁻¹⁷ Consequently, the development of exfoliation method to obtain large amount of MoS₂ nanosheets remains to be solved before practical use can be realized.

In this contribution, we report a one-step, large-scale, facile, safe, low-cost, environmentally friendly method, namely ionic liquid (IL) assisted grinding, to obtain individual nanosheets from the bulk MoS₂. Ionic liquid is chosen owing to their unique properties, such as nonvolatile, non-flammability, low vapor pressure, and good electrical conductivity, which has emerged as a promising medium over the past decades in the area of synthesis, separation and electrochemistry.¹⁸⁻²³ Specially, researchers have demonstrated the debundling of single-walled carbon nanotubes (SWNTs) and functionalized ones with presence of IL.^{20, 22} Most importantly, the exfoliation of graphene nanosheets and nanodots based on IL assisted grinding has been firstly reported by Shang and co-workers.²³ Being inspired by the debundling of SWNTs and graphene with IL, exfoliation by IL assisted grinding method is initiated as a new approach to synthesize MoS₂ nanosheets, which has never been explored before. Our procedure occurs at room temperature equipping only a mortar and pestle to mix the reactants and provide mechanical shear forces for the exfoliation of the MoS₂ sheets from bulk MoS₂. Nevertheless, to fully harness the capabilities of MoS₂ nanosheets, principally limited by its poor dispersity and stability in aqueous solutions, chitosan (CS) is introduced during the grinding process. The result product was then characterized and employed as a photothermal agent for the photothermal ablation of cancer *in vitro*.



Scheme 1. Schematic to illustrate the facile exfoliation of MoS₂ by IL assisted grinding with the presence of chitosan, forming CS-MoS₂ nanosheets, which can serve as a photothermal agent.

Results and discussion

A one-step approach was employed to the large-scale synthesis of functionalized MoS₂ nanosheets by ionic liquid assisted grinding, as shown in Scheme 1. More details on the synthesis procedure can be found in the Experimental Section. In general for the solvent assisted exfoliation of the layered materials, the well matched surface energy of solvents and layered material could result in the minimized enthalpy of exfoliation and effective exfoliation of layered materials.^{13, 23-27} According to a previous study, the surface energy of a solvent (γ) can be

converted to surface tension (T) by equation $\gamma = T + TS_s$, where S_s is the surface entropy and the value of TS_s is ~ 29 mJ/m² for almost all liquids at room temperature.²⁷ Thus, the surface tension (~ 40 mN/m) of used IL could be converted to 69 mJ/m² the surface energy, which properly matches with literature value of the MoS₂ surface energy (that is ~ 75 mJ/m²).²⁷⁻²⁹ Consequently, the used IL can effectively overcome the inherent van der Waals forces between MoS₂ sheets, much in the same way as the debundling of graphene and SWNTs in IL, promoting the exfoliation of the individual MoS₂ sheets and preventing the detached MoS₂ layers from restacking. However, owing to their hydrophobic nature, the direct dispersion of native MoS₂ sheets in water has been generally considered unattainable.^{6, 30-32} Thus, during the grinding process, chitosan was introduced alternatively to facilitate the physiological stability and biocompatibility of MoS₂ nanosheets.^{5, 33} After removing organic residues and incompletely delaminated MoS₂ by centrifugation, a homogeneous and dark green dispersion of CS-MoS₂ nanosheets was obtained. To investigate the stability of the as-prepared CS-MoS₂ nanosheets, we monitored the UV-vis absorbance at 610 nm of the CS-MoS₂ dispersion for weeks (Fig. S1). The colloidal suspension showed feeble absorbance decay even stood for 14 days, implying excellent colloidal stability in water. More excitingly, our CS-MoS₂ nanosheets showed no sign of aggregation and precipitation in water, buffer solution and even in cell medium, indicating well physiological stability of CS-MoS₂ nanosheets (Fig. 1a). The favorable physiological and storage stability of CS-MoS₂ sheets occurs due to the existence of chitosan molecules bound to the exfoliated sheets, inducing the truly homogeneous codispersion of CS and MoS₂ sheets and avoiding the aggregation of sheets caused by van der Waals interactions and hydrophobic interactions.^{6, 34} The concentration of MoS₂ nanosheets was then determined to be 426.1 $\mu\text{g/mL}$ by atomic absorption spectrum (AAS, Fig. S2), indicating our procedure is large-scale and high-yield (~ 17 wt.%).^{24, 35}

To confirm the combination of CS and MoS₂ sheets, Fourier Transform Infrared spectroscopy (FT-IR) and Thermogravimetric analysis (TGA) experiments were conducted. FT-IR spectra of MoS₂, chitosan and our product were performed between 4000 and 400 cm⁻¹. As shown in Fig. 1b, CS-MoS₂ nanosheets produced very similar absorptions to native chitosan. The peaks at 1600 and 1410 cm⁻¹ are assigned to the -NH and -CH₂ bending, respectively. CH₃ and C-OH wagging are located at 1380 and 1340 cm⁻¹, respectively, while 1000 cm⁻¹ is resulted from skeletal vibrations of O-C-O stretching.^{33, 36} Moreover, the band at about 468 cm⁻¹ presented in both case of MoS₂ and CS-MoS₂ is corresponded to Mo-vibration.³⁷ The results indicate the coexistence of CS and MoS₂, meaning the formation of CS-MoS₂ complex. To better demonstrate the combination, TGA measurements under N₂ atmosphere at a heating rate of 10 $^{\circ}\text{C min}^{-1}$ were employed. As shown in Fig. 1c, negligible weight loss can be observed during the heating process from room temperature to 600 $^{\circ}\text{C}$ for native MoS₂. Two weight losses are observed in the TGA curve of CS. The weight loss before 150 $^{\circ}\text{C}$ is due to the moisture vaporization, while the weight loss over 220 $^{\circ}\text{C}$ is attributed

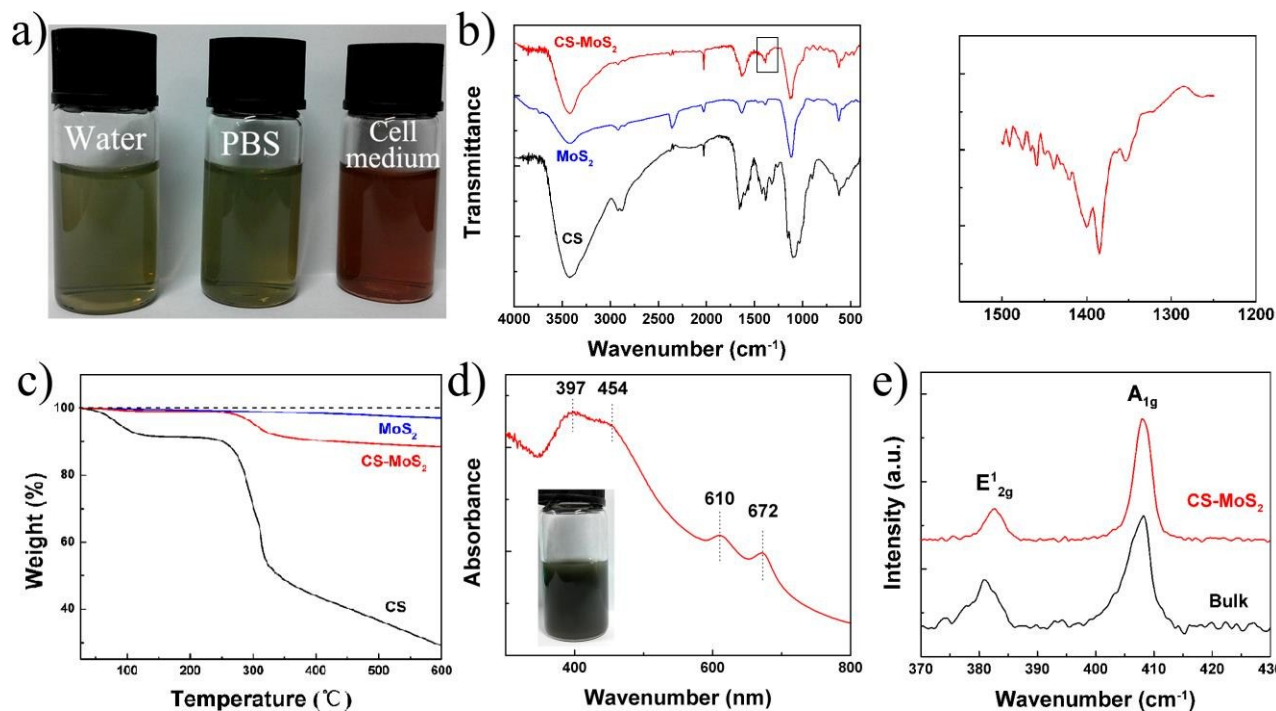


Fig. 1 Characterization of CS-MoS₂ nanosheets. (a) Photos of CS-MoS₂ nanosheets in water, PBS and cell medium, respectively. (b) FT-IR spectra for chitosan, native MoS₂ and CS-MoS₂ nanosheets, and enlarged FT-IR spectrum of CS-MoS₂ nanosheets in the range of 1250–1500 cm⁻¹. (c) TGA-mass loss curves of chitosan, native MoS₂ and CS-MoS₂ nanosheets. (d) UV-vis spectrum of CS-MoS₂ nanosheets, (Inset) a photo of CS-MoS₂ dispersion in water. (e) Raman spectra of natural MoS₂ and CS-MoS₂ nanosheets.

the degradation of CS molecules. Most prominently, once complex with CS, TGA of the intercalation compound reveals two decomposition onsets and a significant weight loss over 220 °C, which agrees well with that of pure CS. The results of FT-IR and TGA adequately manifest the interaction between CS and MoS₂. Moreover, from the result of TGA experiments, the content of CS in our CS-MoS₂ nanosheets was calculated to be *ca.* 12.5 wt.%, which is obviously higher than that of previous report, namely ~5 wt.%.⁵ The higher content of CS endows CS-MoS₂ nanosheets with better storage stability, superior biocompatibility and broader potential in many fields.^{33, 36, 38}

With the certitude of successful interaction between CS and MoS₂, further characterizations were then conducted. Firstly, the optical absorption spectrum for our CS-MoS₂ suspension was measured using a UV-vis spectrometer with 1 nm steps (Fig. 1d). Typical characteristic absorption bands of MoS₂ located at 672, 610, 454 and 397 nm are observed, which are in good agreement with few-layered 2H-MoS₂ obtained from a liquid-based exfoliation method.^{13, 25} The absorption peaks at 672 and 610 nm can be assigned to the direct excitonic transitions at the K point with the energy difference arising due to spin-orbital splitting of the valence band. Peaks at 454 and 397 nm correspond to the direct excitonic transitions of M point between higher density of state regions of the band structure.²⁵ According to the above results of UV-vis spectra and AAS, the extinction coefficient of the as-prepared MoS₂ was determined to be 62.6 L g⁻¹ cm⁻¹ (Fig. S3), which is higher than that of the

chemically exfoliated MoS₂.^{2, 6, 39} The result is reasonable for a chemically exfoliated MoS₂ with the intercalation of Li⁺ ion on the lamella of MoS₂, forming Li_xMoS₂, aggrandizes the relative molar mass of MoS₂, which in turn leads to the imprecise calculation of extinction coefficient.¹¹ Importantly, with mild grinding process only relied on shear forces to exfoliate the MoS₂ layers from the bulk materials, the formation of several defects on the crystalline plane is avoided. Thus, fewer defects generated during our synthesis process, compared to the chemical exfoliation, might also contributed to higher extinction coefficient.² The result indicates that our IL assisted grinding exfoliation process induces less defects, resulting in high quality MoS₂ sheets.

The structural changes in the MoS₂ before and after exfoliation were elucidated by Raman spectroscopy at room temperature and the spectra are shown in Fig. 1e. Raman spectra of both samples show two prominent peaks corresponding to the in-plane E_{2g}¹ and out-of-plane A_{1g} vibrations of MoS₂. Upon exfoliation of the bulk material to single and few layered MoS₂, the Raman peaks of E_{2g}¹ red shifts to 382.7 cm⁻¹ and A_{1g} shows a small blue shift, with the position difference (Δ) decreases from 27.3 of bulk to 25.3 cm⁻¹, further confirming the exfoliation of MoS₂. However, the gap between E_{2g}¹ and A_{1g} peaks of our exfoliated CS-MoS₂ is somewhat broadening when compared to those of the mechanically exfoliated MoS₂ layers.^{2, 25} This might be due to the larger thicknesses of MoS₂ nanosheets caused by the presence of CS on the surfaces. The typical morphology

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structures, and dispersivity of the CS-MoS₂ nanosheets were then analyzed by transmission electron microscopy (TEM) and scanning electron microscopy (SEM). In Fig. 2a, 2b and 2c, it is shown that CS-MoS₂ nanosheets have a well-defined laminar morphology with a uniform size of around 120 nm (Fig. 2d), wrinkled sheets and high dispersivity rather than large-size aggregated bulks, which are consistent with what is usually observed for exfoliated layered compounds.^{13, 25} As the foremost method allowing definitive measurement of the thickness of layer crystals currently, atomic force microscopy (AFM) was then conducted to unambiguously verify the thickness of exfoliated sheets. The AFM results shown in Fig. 2e and 2f clearly reveal that CS-MoS₂ nanosheets have uniform shapes with a typical thickness of *ca.* 1.4 nm and undergo about a 0.5 nm increase versus pure single-layer MoS₂ nanosheets (*ca.* 0.9 nm), mainly attributable to the attachment of CS on both planes of the MoS₂ sheets.^{5, 40}

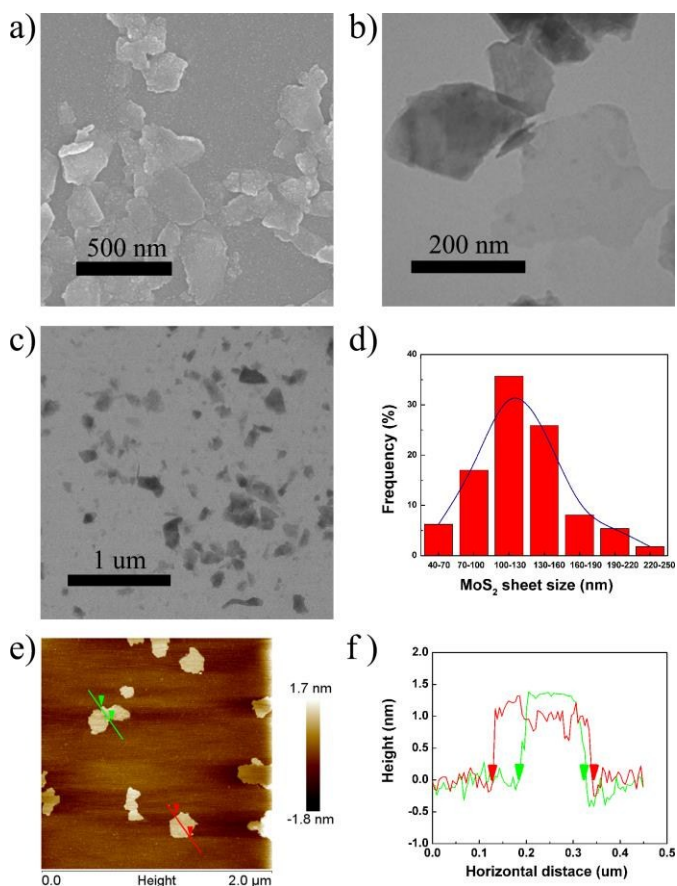


Fig. 2 Electron microscopes characterization of CS-MoS₂ nanosheets. Typical SEM (a), TEM (b and c) images of CS-MoS₂ nanosheets. (d) TEM measured size distribution of CS-MoS₂ nanosheets, over 100 sheets were counted. AFM image (e) and height profile across the CS-MoS₂ nanosheets in panel (f).

Additionally, the following three control experiments were also conducted: grinding with IL only, grinding in water or ethanol with chitosan. In all these cases, as expected, the obtained MoS₂ nanosheets dispersed in water aggregate significantly after standing for only 2 hours, shown in Fig. S4. Therefore, the co-grinding of chitosan and bulk MoS₂ in IL is

the crucial point of the successful functionalization of individual MoS₂.

In view of above results, we have successfully proposed a new strategy for the large-scale fabrication of stable and functionalized MoS₂ nanosheets by ionic liquid assisted grinding. Although Yin and coworkers have reported an oleic acid based method to high-throughput synthesize CS-MoS₂, this approach requires harsh solvents and extremely time-consuming multiple steps.⁵ By contrast, our facile strategy based on ionic liquid assisted grinding has several advantages: 1) with one-step approach to exfoliate and functionalize MoS₂ by grinding process, no expensive equipment with extremely time-consuming multiple steps was required, which seems to be more facile and convenient, 2) the reaction medium is ionic liquid, a recyclable and green organic solvent, thus with no environmental pollution, 3) the obtained CS-MoS₂ has high content of CS, which might endow our CS-MoS₂ nanosheets with better storage stability and superior biocompatibility. These advantages allow this proposed synthetic approach to pave the way for the synthesis of advanced functional MoS₂ materials and extensive applications in nanomedicines.

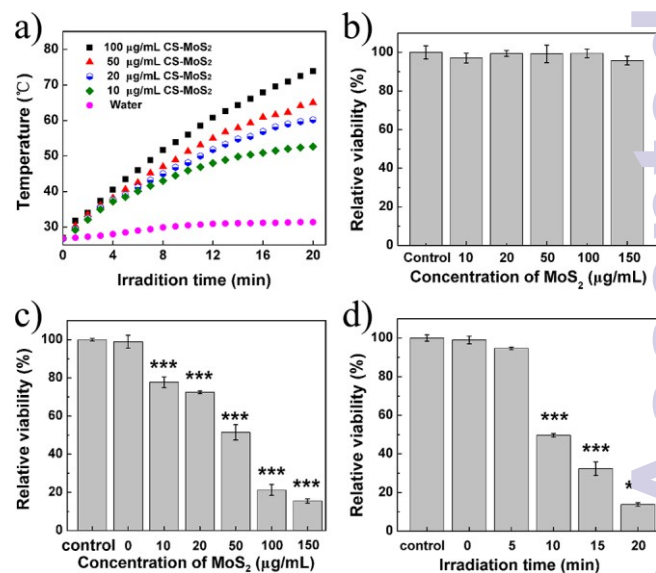


Fig. 3 Photothermal and biological activity of CS-MoS₂ nanosheets. (a) Photothermal heating curves of pure water and various concentrations of CS-MoS₂ nanosheets irradiated by 808 nm laser at power density of 2 W/cm² for 20 min. Relative cell viability data of HepG2 cells after incubation with CS-MoS₂ nanosheets at different concentrations for 24 h and then treated without (b) or with (c) 808 nm NIR irradiation for 10 min. (d) Relative cell viability after incubation with 50 μg/mL CS-MoS₂ nanosheets for 24 h and then treated with 808 nm NIR irradiation for various times. Results of cell viability are shown as the means ± SD of six separate experiments. *** < 0.001 versus control.

With superior performance, MoS₂ nanosheets and chitosan functionalized nanomaterials have been widely applied in nanomedicine.^{5, 33, 39, 41} Herein, to verify our CS-MoS₂ nanosheets processing inherent properties of MoS₂ and CS, the photothermal heating and *in vitro* cytotoxicity experiments were conducted. The high molar extinction coefficient of CS-MoS₂ nanosheets in the NIR region, calculated to be 62.6 L g⁻¹ cm⁻¹, indicated the potential of CS-MoS₂ nanosheets as

photothermal agent to efficiently convert NIR light into heat. To assess the light-to-heat conversion capability of aqueous dispersions containing different concentrations of the as-prepared CS-MoS₂ nanosheets (0–150 µg/mL), the solution temperature increased by NIR laser irradiation (808 nm, 2 W/cm²) was recorded. Fig. 3a shows the temperature of the dispersions as a function of irradiation time. The blank test without CS-MoS₂ nanosheets shows negligible increase of temperature by less than 5 °C. However, when irradiated with the presence of CS-MoS₂ nanosheets, the temperature of solution increased with the increasing concentration of CS-MoS₂ and irradiation time; the heating rate slower with extension of irradiation time, apparently as a result of faster heat loss at higher temperature.⁴² At a concentration of CS-MoS₂ nanosheets above 10 µg/mL, the temperature of CS-MoS₂ dispersion induced by NIR irradiation for 10 min is higher than 43 °C, which was considered to be high enough for PTT therapy of cancer.⁴¹ Excitingly, our CS-MoS₂ nanosheets exhibited great photothermal stability without any significant decrease in the UV-vis absorbance even after exposure under laser for a certain period of time (Fig. S5), in marked contrast to gold nanorods, which are currently the subject of great interest for PTT but melted after being irradiated by the laser as reported in the literature.⁴³ These results indicate the capacity of our CS-MoS₂ nanosheets as a stable photothermal agent to convert the 808 nm laser energy into thermal energy for PTT.

Moreover, the abundant of biocompatible chitosan on the plane of MoS₂ nanosheets was expected to reduce the toxicity of MoS₂. To prove this, before we move on to further PTT experiments, the intrinsic toxicity of CS-MoS₂ nanosheets was studied by MTT assay with HepG2 cells. From Fig. 3b, it is apparent that, no significant differences in the cell viability were observed in the absence or presence of (5–150 µg/mL) CS-MoS₂ nanosheets. More importantly, the cellular viability was estimated to be greater than 95% after incubation with CS-MoS₂ nanosheets for 24 h even at a concentration of 150 µg/mL. These data show that our chitosan coated MoS₂ nanosheets have lower cytotoxicity compared to native MoS₂ or Li_xMoS₂, which might attributes to the abundant interspersions of biocompatible chitosan on the plane of MoS₂ sheets.^{5, 44, 45}

With potent photothermal effect and excellent biocompatibility, the *in vitro* PTT therapy capacity of our CS-MoS₂ nanosheets was then evaluated. After incubation with 50 µg/mL CS-MoS₂ nanosheets for 24 hours in 24-well plates, HepG2 cells are exposed to an 808 nm laser irradiation for various time periods to evaluate the localized tumor photothermal effect of CS-MoS₂ nanosheets. To test their photothermal stability under an optical microscope, the irradiated cells are then stained with trypan blue. Fig. 4 shows images of samples irradiated for different time. As can be seen, cell death is shown as a blue spot and the proportion of dead cells increases with the extension of NIR laser irradiation time. By increasing the irradiation time to 15 min, the majority of the cancer cells were dead. The accordant results were then obtained by MTT test (Fig. 3d). After replanting the cells in a 96-well plate (n = 6) and incubation with or without 50 µg/mL CS-MoS₂ nanosheets, we monitored cell viability 24 h via MTT

method after the photothermal treatment. The samples without irradiation by NIR laser showed no considerable dead cells, which also indicates our CS-MoS₂ nanosheets has good biocompatibility, while the viability of cells decreased dramatically with the extension of irradiation time and only about 13.82% of HepG2 cells remained viable after irradiation for 20 min. To further understand the PTT activity of CS-MoS₂ nanosheets, cytotoxicity experiments induced by NIR laser with different concentrations of CS-MoS₂ nanosheets were carried out. The cells were incubated with increasing amounts of CS-MoS₂ nanosheets (10, 20, 50, 100, 150 µg/mL) for 24 h and then irradiated with an 808 nm NIR laser for 10 min under a power density of 2 W/cm². Upon laser irradiation, cell viability noticeably decreased with the increasing concentration, and less than 16% of the cells remained alive after irradiation for 10 min (Fig. 3c). In accordance with the excellent photothermal responsiveness, MoS₂ nanosheets in our experiments can efficiently kill cancer cells by hyperthermia. These preliminary *in vitro* results confirm our expectations that our CS-MoS₂ nanosheets possessing excellent photothermal property for nanomedicine are comparable to MoS₂ nanosheets reported elsewhere.

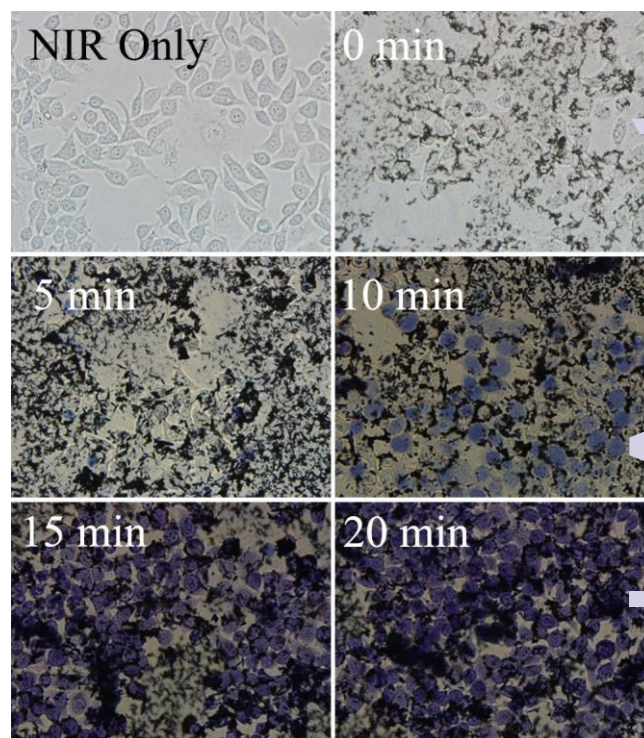


Fig. 4 Optical images of photothermal destruction of HepG2 cancer cells incubated with 50 µg/mL CS-MoS₂ nanosheets at various times of 808 nm NIR irradiation with power density of 2 W/cm². As can be stained by trypan blue, the dead cell emerges as a blue spot.

Experimental

Chemicals and Materials

Molybdenum (IV) sulfide (MoS₂) with a bulk particle size < 2 µm, 1-Butyl-3-methylimidazolium hexafluorophosphate

(BMIPF₆) ($\geq 97.0\%$), the IL used in this work, and CS were purchased from Sigma Aldrich. N,N-dimethylformamide (DMF), acetone and acetic acid, used during the washing process, were obtained from Sinopharm Chemical Reagent Co., Ltd. 3-(4, 5-dimethylthiazol-2-yl)-2, 5-diphenyltetrazolium bromide (MTT) and trypan blue were purchased from MP Biomedicals LLC. All chemicals were used as received without any further purification. Deionized water was used in all experiments.

Synthesis of CS-MoS₂ nanosheets

Typically, using an agate mortar with a pestle, 250 milligrams of bulk MoS₂ was ground with 100 milligrams of CS for a period of 10 minutes. 0.5 mL of IL was then added into the mortar, followed by grinding for another 50 min. The grinding mixture was collected from the mortar and pestle, and then washed with acetone, DMF and 0.5% acetic acid to remove the ionic liquid and the excess chitosan. This washing cycle was repeated three times. Finally, the sediment was dispersed in water and centrifuged at a speed of 1500 rpm for 20 min to remove the large/thick MoS₂. The obtained CS-MoS₂ nanosheets dispersion was diluted with water to 100 mL and stored in 4 °C for the following investigations. The concentration of MoS₂ sheets in suspension was determined by atomic absorption spectroscopy (AAS) (Z-2000, Hitachi).²

Characterizations

Fourier transform infrared (FT-IR) spectroscopy was recorded on a Vetex70 (BRUKER Corp., Germany). The weight loss curves were obtained with thermo gravimetric analyzer apparatus (STA449F3, Netzsch, Germany) from room temperature to 600 °C at the rate of 10 °C/min in N₂ calcinations. Raman measurements were done using HR LabRam Raman spectroscopy system. The UV-vis spectra were measured with a UV-2550 spectrophotometer (Shimadzu, Japan) at room temperature. Field emission scanning electron microscope (SEM) image was taken by an S-4800 (Hitachi, Japan). Transmission electron microscopy (TEM) measurements were performed with an H-600 (Hitachi, Japan). AFM measurement was performed on a Bruker NanoScope V instrument in tapping mode.

In vitro cytotoxicity of CS-MoS₂ nanosheets

The *in vitro* cytotoxicity experiments were carried out using a human hepatocyte carcinoma cell line (HepG2) derived from a well differentiated human hepatoblastoma. HepG2 were cultured in RPMI-1640 medium supplemented with 10% fetal bovine serum (FBS), benzylpenicillin (100 kU/L) and streptomycin (100 mg/L) at 37 °C and in atmosphere of 5% CO₂.⁴⁶ To determine the cell viability under dark condition (i.e. without laser treatment), HepG2 (2×10^5 cells/well) was seeded into 24 well cell-culture plates and incubated for 24 h at 37 °C. Then, cells were treated with different concentrations of CS-MoS₂ nanosheets for 24 h. Following treatment, cells were rinsed with DPBS and treated with 50 μ L 5 mg/mL MTT reagent in serum-free media. After incubation for further 3 h, formazan crystals in each well were solubilized in 0.5 mL of dimethyl sulfoxide (DMSO). The final solution in each well was centrifuged at 13,000 rpm to remove any solid residues.

The optical absorbance at 490 nm was then recorded by a Microplate reader (Thermo Multiskan MK3).

Photothermal activity of CS-MoS₂ nanosheets

As an efficient NIR absorber, the photothermal activity of the as-prepared CS-MoS₂ nanosheets was assessed. Firstly, CS-MoS₂ nanosheets dispersion was diluted to various concentrations and 5 mL of each solution was taken in a 10 mL glass beaker. This solution was then illuminated with an 808 nm NIR laser with a power density of 2 W/cm². Light induced heating in the solution was measured at 1 min intervals with a thermometer located inside the suspension for a total time of 20 min.

For NIR photothermal therapy, cells were seeded as previously. 24 h after cell seeding, the medium was replaced with the diluted CS-MoS₂ and incubated for further 24 h. After that, cells were washed thoroughly by fresh serum-free media and exposed to an 808 nm NIR laser source with the beam diameter of about 1 cm and power density of 2 W/cm². The cells were incubated for additional 24 h and cell viabilities were measured by MTT assay as previously described. Besides, to have some optical images from the photothermal destruction of the cells, cancer cells that cultured with 50 μ g/mL CS-MoS₂ nanosheets were exposed to the NIR laser irradiation for various periods of time. After further incubation for 24 h, cells were washed with PBS and stained with 0.04% trypan blue solution for 10 min. Microscopic images of cells were then taken using a microscope.

Statistical analysis

All data were expressed as mean values \pm standard deviation (SD). The intergroup variation was measured by one-way analysis of variance (ANOVA) followed by Duncan's multiple range tests. The level of statistical significance was established at $p < 0.05$.

Conclusions

In this contribution, we have demonstrated an easy way for the one-step exfoliation and functionalization of molybdenum disulfide in ionic liquid using an agate mortar with a pestle only. The concentration of the resulting product was as high as ~ 0.5 mg/mL accompanied with high-yield approaching 17%. The present exfoliation process establishes a high-throughput and soft method for the top-down fabrication of modified MoS₂ nanosheets, as compared to the conventional approach of using organolithium or various organic solvent with sophisticated equipment. The as-fabricated CS-MoS₂ nanosheets dispersions are endowed with well dispersity in aqueous solution, good biocompatibility for photothermal therapy. The ease of synthesis and functionalization of MoS₂ nanosheets make this inexpensive and rising nanostructure more attractive in the application of nanomedicine.

Acknowledgements

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