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LETTER

Cyano-Bridged Coordination Polymer Gel as a Precursor to Nanoporous $\text{In}_2\text{O}_3\text{--Co}_3\text{O}_4$ Hybrid Network for High-Capacity and Cycle-Stable Lithium Storage

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A novel type of nanoporous $\text{In}_2\text{O}_3\text{--Co}_3\text{O}_4$ hybrid network has been synthesized using a cyano-bridged coordination polymer gel as a precursor. When applied as an anode in lithium-ion batteries, the $\text{In}_2\text{O}_3\text{--Co}_3\text{O}_4$ network manifests remarkable capacity retention and high reversible capacities by virtue of its unique compositional and structural features.

As important categories of anodic materials in lithium-ion batteries (LIBs), metal oxides have been considered to be ideal anodic candidates to replace commercial graphite-based materials owing to their higher safety and specific capacities.^{1–15} Among them, indium oxide (In_2O_3) possesses similar alloying/de-alloying lithium-storage mechanism to other main-group metal oxides including SnO_2 ,^{7–9} GeO_2 ,^{10,11} and so forth. Generally, the lithium-storage processes of In_2O_3 can be described as follows: $\text{In}_2\text{O}_3 + 6\text{Li}^+ + 6\text{e}^- \rightarrow 2\text{In} + 3\text{Li}_2\text{O}$ (eqn 1); $\text{In} + 4.33\text{Li}^+ + 4.33\text{e}^- \leftrightarrow \text{Li}_{4.33}\text{In}$ (eqn 2).^{12–15} Despite the improved safety and higher theoretical capacities, the formation of inactive Li_2O (eqn 1) gives rise to large Li^+ -consumption and initial capacity loss, meanwhile the alloying/de-alloying processes described by eqn 2 lead to huge volume variations, progressive pulverization, and fast capacity decay.

Recently, it has proved that nanostructuring and hybridizing SnO_2 anodes with transition-metal oxides (M_xO_y , $\text{M} = \text{Mn}$,¹⁶ Fe ,^{17–20} Co ,²¹ Ni ,^{22–24} Cu ,²⁵ Mo ,²⁶ W ,²⁷ etc) could serve as an effective solution to reduce the initial irreversible capacity and improve the cycling stability of SnO_2 -based anodes. The *in situ* generated transition-metal nanocrystals from M_xO_y can activate/promote the decomposition of Li_2O and reversible formation of polymeric gel-like layer and thus increase their Coulombic efficiencies and reversible capacities, meanwhile the stepwise Li-storage reactions at different potentials of SnO_2 and M_xO_y together with rationally designed nanostructure can effectively accommodate the volume changes and improve the cycling performance of the $\text{SnO}_2\text{--M}_x\text{O}_y$ nanohybrids.^{16–27} Therefore, $\text{In}_2\text{O}_3\text{--M}_x\text{O}_y$ nanohybrids are anticipated to exhibit similar performance-enhancement mechanisms

and manifest ideal lithium-storage performances for advanced LIBs.^{15–27}

As inspired by this, we have rationally designed and synthesized a novel type of $\text{In}_2\text{O}_3\text{--M}_x\text{O}_y$ nanohybrids, i.e., nanoporous $\text{In}_2\text{O}_3\text{--Co}_3\text{O}_4$ hybrid network, *via* a facile freeze-drying and subsequent annealing approach using a cyano-bridged In(III)–Co(III) bimetallic coordination polymer gel (i.e., $\text{InCl}_3\text{--K}_3\text{Co(CN)}_6$ cyanogel) as a precursor (Fig. 1). When utilized as an anode in LIBs, the as-prepared $\text{In}_2\text{O}_3\text{--Co}_3\text{O}_4$ network manifests excellent capacity retention and high reversible capacities by virtue of its unique compositional and structural features.



Fig. 1 Schematic diagram for the synthesis of the nanoporous $\text{In}_2\text{O}_3\text{--Co}_3\text{O}_4$ hybrid network.

Cyano-bridged coordination polymer gels (abbreviated as cyanogels) are formed *via* the reaction of solutions of chlorometalates (K_2PtCl_4 ,²⁸ K_2PdCl_4 ,²⁹ SnCl_4 ,^{24,30} etc) and transition-metal cyanometalates including $\text{K}_3\text{Co(CN)}_6$, $\text{K}_2\text{Ni(CN)}_4$, $\text{K}_4\text{Fe(CN)}_6$, and so forth, which experience sol-gel transitions to form stable gels. Cyanogels possess unique structural and compositional features in terms of three-dimensional (3D) porous framework and double-metallic property, and thus could serve as ideal precursors for porous alloy^{28–30} and hybrid-oxide²⁴ networks with uniform distribution of metal species.

Herein, for the first time, a white-colored In(III)–Co(III) -based cyanogel (i.e., $\text{InCl}_3\text{--K}_3\text{Co(CN)}_6$ cyanogel) was conveniently obtained by simply mixing aqueous solutions of InCl_3 and $\text{K}_3\text{Co(CN)}_6$ at room temperature (Fig. 2). The molar ratio of In/Co in

the cyanogels can be easily adjusted by controlling the feeding ratios of InCl_3 and $\text{K}_3\text{Co}(\text{CN})_6$ reactants, and this feature of InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogels facilitates subsequent construction of alloy and hybrid-oxide products with controllable compositions and adjustable properties. As observed, stable InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogels can be obtained at a series of feeding ratios of InCl_3 and $\text{K}_3\text{Co}(\text{CN})_6$ from 2:1 (Fig. 2a), 1.5:1 (Fig. 2b), 1:1 (Fig. 2c), 1:1.5 (Fig. 2d), to 1:2 (Fig. 2e).

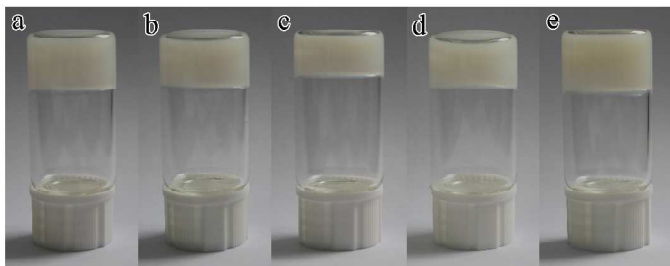


Fig. 2 Photographs of the InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogels prepared by simply mixing aqueous solutions of 0.6 M InCl_3 and 0.6 M $\text{K}_3\text{Co}(\text{CN})_6$ with different volume ratios: (a) 2:1, (b) 1.5:1, (c) 1:1, (d) 1:1.5, and (e) 1:2.

As a representative example, the physicochemical properties of the InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogel with a In/Co molar ratio of 1:1 were examined by Fourier transform infrared (FTIR) spectra (Fig. 3). As observed, the FTIR spectrum of $\text{K}_3\text{Co}(\text{CN})_6$ has a sharp cyano stretching vibration at 2130 cm^{-1} and a stretching vibration of $\nu(\text{Co}-\text{C})$ at 565 cm^{-1} (curve a). According to the previous report, $\nu(\text{C}\equiv\text{N})$ shifts to a higher frequency and $\nu(\text{M}-\text{C})$ shifts to a lower frequency if the $\text{M}-\text{C}\equiv\text{N}$ group forms a bridging cyano group of $\text{M}-\text{C}\equiv\text{N}-\text{M}'$.³¹ Herein, the $\nu(\text{C}\equiv\text{N})$ shifts to a higher frequency at 2199 cm^{-1} , meanwhile the $\nu(\text{Co}-\text{C})$ shifts to a lower frequency at 465 cm^{-1} with a shoulder at 425 cm^{-1} in the freeze-dried InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogel (curve b). This phenomenon can be attributed to the formation of bridging cyano groups of $\text{Co}-\text{C}\equiv\text{N}-\text{In}$ by analogy to similarly bridged species including $\text{Co}-\text{C}\equiv\text{N}-\text{Pt}$ ²⁸ and $\text{Co}-\text{C}\equiv\text{N}-\text{Pd}$.²⁹ The FTIR results confirm the presence of bridging cyano groups ($\text{Co}-\text{C}\equiv\text{N}-\text{In}$) and thus the formation of cyano-bridged $\text{In}(\text{III})$ – $\text{Co}(\text{III})$ bimetallic coordination polymer gel.^{28,29,31}

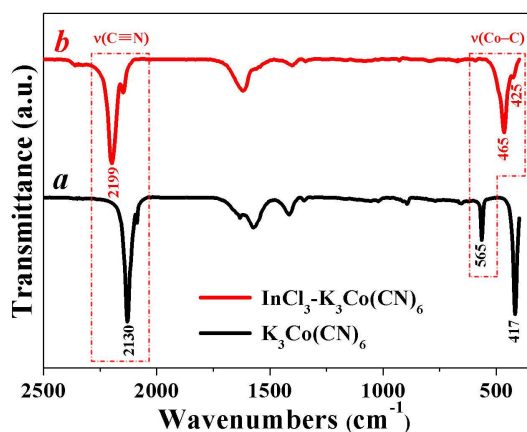


Fig. 3 FTIR spectra of the $\text{K}_3\text{Co}(\text{CN})_6$ (curve a) and freeze-dried InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogel (curve b).

The 3D porous structure feature and double-metallic composition characteristics of InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogel make it an ideal precursor for the construction of porous In_2O_3 – Co_3O_4 hybrid-oxide networks with uniform distribution and controllable ratios of In_2O_3 and Co_3O_4 components. As a proof-of-concept demonstration, nanoporous In_2O_3 – Co_3O_4 hybrid network was facily obtained through a freeze-drying and subsequent annealing route by using the InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogel with a In/Co molar ratio of 1:1 as a precursor. The crystalline state of the product was examined by X-ray powder diffraction (XRD) (Fig. 4). As observed, there are two set of diffraction peaks from the XRD pattern, which can be indexed to cubic In_2O_3 (JCPDS no. 65-3170) and cubic Co_3O_4 (JCPDS no. 43-1003), respectively. The XRD results indicate the successful formation of In_2O_3 – Co_3O_4 hybrid oxides.

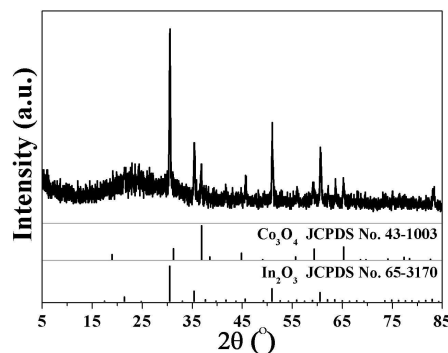


Fig. 4 XRD pattern of the nanoporous In_2O_3 – Co_3O_4 hybrid network.

The morphological, compositional, and structural features of the In_2O_3 – Co_3O_4 network has been examined by scanning electron microscopy (SEM) and transmission electron microscopy (TEM). As seen from SEM images, the product inherits the structural characteristics of the InCl_3 – $\text{K}_3\text{Co}(\text{CN})_6$ cyanogel precursor, and exists in the form of 3D porous network with particle size of about several microns to a dozen microns (Fig. S1). The TEM images further indicate that the In_2O_3 – Co_3O_4 network is consisted of numerous nanocrystal building units and there are abundant nanopores within the network (Fig. 5a and c). Moreover, the high-resolution TEM (HRTEM) image clearly reveals the lattice fringes and interface region of two interconnected nanocrystals (Fig. 5d). As observed, the lattice fringe with lattice spacing of about 0.24 nm corresponds to the (311) plane of Co_3O_4 nanocrystals, and the two lattice fringes appeared at the adjacent nanocrystal with angle of 80.4° and spacing of 0.41 nm can be indexed to the (211) and (1-21) planes of In_2O_3 nanocrystals, respectively. The HRTEM results further confirm that the porous hybrid network is assembled by interconnected In_2O_3 and Co_3O_4 nanocrystals. Fig. 5b reveals the energy-dispersive X-ray spectrometer (EDS) spectrum of the In_2O_3 – Co_3O_4 hybrid network. The observed strong peaks for In, Co, and O elements come from In_2O_3 and Co_3O_4 components, and the Cu elemental peak originate from the copper grid used in the TEM tests. Additionally, the molar ratio of indium and cobalt is determined to be 1.3:1 based on the EDS analysis, which is close to the feeding ratio of the InCl_3 and $\text{K}_3\text{Co}(\text{CN})_6$ reactants. Moreover, the TEM-EDS elemental mapping reveals the uniform distribution of In, Co, and O elemental signals within the selected area and thus confirms the homogeneous distribution of In_2O_3 and Co_3O_4 nano-units in the hybrid network (Fig. 5e and f).

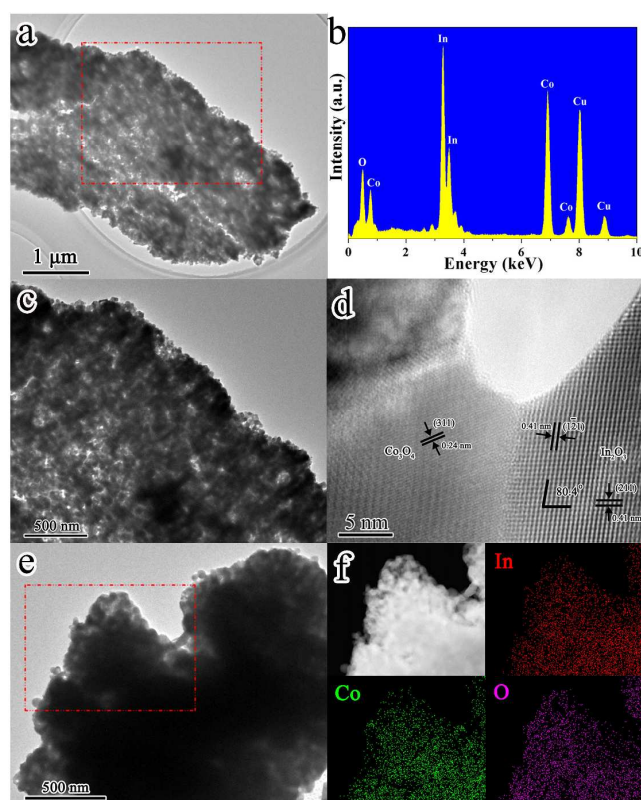


Fig. 5 Morphological, compositional, and structural features of the nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network: (a,c) TEM images, (b) EDS spectrum, (d) HRTEM image, and (e,f) TEM-EDS elemental mappings.

The surface area and pore volume of the hybrid network were further evaluated by nitrogen adsorption/desorption isotherms to demonstrate its porous characteristics (Fig. S2). As illustrated, the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network manifests a high Brunauer–Emmett–Teller (BET) surface area and a large Barrett–Joyner–Halenda (BJH) pore volume of $82.6 \text{ m}^2 \text{ g}^{-1}$ and $0.24 \text{ cm}^3 \text{ g}^{-1}$, respectively. The high surface area and large pore volume of the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network is believed to benefit from the porous $\text{InCl}_3\text{-K}_3\text{Co(CN)}_6$ cyanogel precursor, and more importantly, is beneficial for the strain accommodation and electrolyte contact and thus the enhanced lithium-storage performance in terms of cycling stability and rate capability.^{24,30}

Motivated by its unique compositional and structural features toward lithium-storage, the nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network was utilized as an anode material in LIBs. Fig. 6 reveals the cycling performance of the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network in the potential range of 0.01–3 V at a current density of 100 mA g^{-1} . As observed, the discharge capacities undergo a slowly fading process within the initial 20 cycles, which is mainly due to the formation of solid electrolyte interface (SEI) layer and locally slight pulverization within the network. Despite this, the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network manifests excellent capacity retention in subsequent cycles, and the reversible capacities keep steadily at about 630 mA h g^{-1} after 50 cycles. The *in situ* generated Co nanocrystals during lithium insertion in Co_3O_4 might act as an efficient catalyst to activate/promote the reversible formation of polymeric gel-like SEI layer^{32,33} and also the oxidation

of metallic indium and decomposition of Li_2O ,^{15–27,34} which can be largely responsible for the high capacity retention and reversible capacities. The voltage plateaus at about 2.5 V from the charge curves is more and more obvious upon cycling (Fig. 6a), which also suggests the probable existence of these catalytic processes by cobalt nanocrystals.^{24,25} Thus, the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network is able to deliver a high reversible capacity of $638.7 \text{ mA h g}^{-1}$ after 200 cycles, which is much higher than the theoretical capacity of graphite (372 mA h g^{-1}). The microscopic structural characterization after cycling confirms that the agglomeration and pulverization of hybrid oxide network can be effectively restrained during cycling (Fig. S3), and the enhanced structural stability of the hybrid network plays a key role in its improved Li-storage performance especially cycling stability. The high capacity retention and reversible capacities of the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network will pave the road for its practical utilization as a long-life and high-capacity anode material in advanced LIBs.

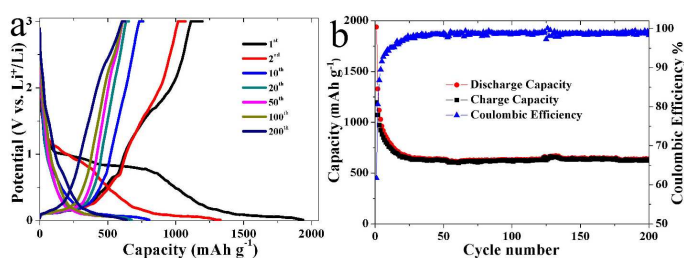


Fig. 6 Lithium storage performance of the nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network: (a) discharge and charge curves, and (b) cycling stability.

The excellent cycling performance and high reversible capacities of the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network can be attributed to its unique compositional and structural features. First, the 3D nanoporous network possesses the superiorities of both nano-building units and micro-assemblies toward lithium storage, and exhibits enhanced strain-accommodation and charge-transport capabilities during cycling.^{24,30} Second, the stepwise Li-storage reactions at different potentials of In_2O_3 and Co_3O_4 can effectively accommodate the volume variations and improve the cycling stability of the hybrid network.^{15–27} Finally, the *in situ* generated cobalt nanocrystals from Co_3O_4 might act as efficient catalyst to activate/promote the decomposition of Li_2O ^{15–27,34} and the reversible formation/decomposition of polymeric gel-like layer on anode surface,^{32,33} leading to high reversible capacities of the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network.

In summary, we have rationally designed and synthesized a novel type of nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network *via* a facile freeze-drying and subsequent annealing approach using $\text{InCl}_3\text{-K}_3\text{Co(CN)}_6$ cyanogel as a precursor. When utilized as an anode in LIBs, the nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network manifests remarkable lithium-storage performance in terms of cycling stability and reversible capacity. For example, a high reversible capacity of $638.7 \text{ mA h g}^{-1}$ can be delivered after 200 cycles at a current density of 100 mA g^{-1} . The synergistic effects between In_2O_3 and Co_3O_4 together with the nanoporous structure can be responsible for the remarkable capacity retention and high reversible capacities in the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network.

Experimental

Sample preparation

The $\text{InCl}_3\text{-K}_3\text{Co(CN)}_6$ cyanogel precursor was obtained by simply mixing freshly-made aqueous solutions of 0.6 M InCl_3 and 0.6 M $\text{K}_3\text{Co(CN)}_6$ with a volume ratio of 1:1 at room temperature. Subsequently, the $\text{InCl}_3\text{-K}_3\text{Co(CN)}_6$ cyanogel was freeze-dried and then the obtained xerogel was annealed at 600 °C in air for 3 h. The resulting black product was washed with distilled water and ethanol to remove KCl crystals and dried at 60 °C in air, yielding the nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network.

Characterization

The morphological, structural, and compositional features of the $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network were characterized by X-ray powder diffraction (XRD, Rigaku D/max-RC), scanning electron microscopy (SEM, JEOL JSM 5610LV), and high-resolution transmission electron microscopy (HRTEM, JEOL JEM-2010F, 200 kV) equipped with an energy-dispersive X-ray spectrometer (EDS, Thermo Fisher Scientific). The Fourier transform infrared (FTIR) spectrum was recorded on a Bruker Tensor 27 spectrometer. Nitrogen adsorption/desorption test was performed using a Quantachrome Nova Station B instrument at 77 K.

Electrochemical measurements

Electrochemical measurements were carried out by 2025-type coin cells, which were assembled in an Ar-filled glove-box (Innovative Technology, IL-2GB). To prepare the anode, 70 wt% active material ($\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ network), 15 wt% conductive carbon black (Super P), and 15 wt% binder (polyvinylidene fluoride, PVDF) were mixed in N-methyl-2-pyrrolidone (NMP) to form an uniform slurry, and then the slurry was coated on the surface of copper foams and dried under vacuum at 120 °C for 12 h. The counter electrode was lithium foil, and the electrolyte solution was 1 M LiPF_6 in ethylene carbonate (EC) and dimethyl carbonate (DMC) (1:1 v/v). Finally, the cells were aged for 12 h before tests. Charge-discharge characteristics were measured on a LANHE CT2001A battery tester (Wuhan LAND) in the potential range of 0.01-3.0 V at a current density of 100 mA g^{-1} .

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Notes and references

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Cyanogel-derived three-dimensional nanoporous $\text{In}_2\text{O}_3\text{-Co}_3\text{O}_4$ hybrid network as a high-capacity and long-life anode material for lithium-ion batteries

