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Abstract

In this work, a hybrid polymer electrolyte membrane based on poly(ethylene oxide) (PEO)/poly(vinylidene fluoride-co-hexafluoropropylene) (PVDF-HFP) blend containing chemically exfoliated nanosilicate platelets was prepared to replace the conventionally used liquid electrolyte for dye sensitized solar cells (DSSC). The layered montmorillonite nanosilicates were chemically exfoliated using aminopropyltrimethoxy silane (APS). The modifications on nanosilicate by APS molecules were characterized by Fourier infrared spectroscopy (FTIR) and wide angle X-ray diffraction (WAXD). The polymer composite membranes were subjected to FTIR, WAXD and Differential scanning calorimetry (DSC) studies in order to understand the intercalation/exfoliation and crystallinity. The surface and porous structure of composite membranes were examined by scanning electron microscope (SEM) analysis and mercury porosimetry. The porosity and weight uptake enhanced with the addition of surface modified nanoplatelets. The UV-VIS spectra indicated an increase in free ion concentration with the addition of exfoliated nanoplatelets, which led to an increase in the ionic conductivity up to 2.52×10^{-3} S/cm for 6 wt% and it decreased afterwards. On the other hand, same amount of unmodified counterpart achieved an ionic conductivity of about 5.41×10^{-4} S/cm. In addition to this, an increase in ion concentration and ionic diffusion coefficient was found with the addition of surface modified MMT platelets. The linear steady state voltammetry indicated that incorporation of surface modified nanoplatelets reduces the ionic diffusion length and increases the diffusion coefficient to about 4.8×10^{-9} cm²/s. The photovoltaic performance exhibits an enhanced open circuit voltage (V_{oc}) 0.73 V and short circuit current (J_{sc}) 7.7 mA/cm² under the illuminations of 100mW/cm².

47

48 1. Introduction

49 In 1991, the first original Dye Sensitized Solar Cell (DSSC) was proposed by O'rgan
50 and Gratzel.¹ After the invention of DSSC, it has attracted significant attention owing to its
51 simple structure, low material cost, and low energy consumption for production.² The typical
52 DSSC consists of dye adsorbed TiO₂ as photo anode, I-/I₃⁻ redox electrolyte and thin Pt layer as a
53 counter electrode.³ Till date, the best solar conversion efficiency achieved for a liquid electrolyte
54 based DSSC is about 11-13%.⁴ However, low durability, solvent leakage, dye desorption and
55 electrode corrosion in DSSC due to the liquid electrolytes restricts the commercial viability. In
56 order to replace the conventional liquid electrolyte system, many solid state alternatives such as
57 P-type semiconductor, organic hole conductors, room temperature molten salts, polymeric or gel
58 materials comprising redox active electrolyte solution have been studied for DSSC applications.⁵⁻
59 ⁷ Among the solid alternatives, polymer gel electrolytes have several advantages like limited
60 internal shorting, low vapor pressure, excellent contact and filling properties between nano
61 structured photo anode and counter electrode, high ionic conductivity, thermal stability than
62 other solid alternatives.⁷ Variety of polymers have been employed as an electrolyte host like
63 PEO, PVDF, PAN, PU, PVDF-HFP, PMMA etc. ⁹⁻¹³ Among them, PVDF-HFP and PEO are
64 selected as the host polymer electrolyte, because PVDF-HFP is photo chemically stable and has
65 high ionic conductivity. Semi-crystalline PVDF gives it the mechanical strength and HFP
66 provides flexibility.¹⁴ Poly (ethylene oxide) (PEO) offers better ion transport by a consolidation
67 of its cation-complexation capability and fast diffusion of chains in the amorphous phase. But the
68 main drawback is its crystallization temperature, which is around 50-60°C. Blending with
69 PVDF-HFP reduces its crystallinity significantly and obtains proper compatibility through the
70 interaction of CF₂ group present in PVDF-HFP and C-O-C in PEO. ¹⁵

71 Hybridization of inorganic fillers with polymer matrix is a well established technique. It
72 can give combined properties of organic and inorganic materials, like thermal, mechanical and
73 electrical properties. These properties mainly depend upon the size, shape and crystallinity of
74 fillers as well as the bonding between surface functional group of polymer and nanofillers.¹⁶ Till
75 date, TiO₂, Al₂O₃, SiO₂ etc, have been effectively utilized for polymer nanocomposite electrolyte
76 applications. The maximum solar conversion efficiency reported is about 7.2 % employing TiO₂
77 as fillers.¹⁷ Montmorillonite (MMT) is a type of clay with a high aspect ratio. Its crystal structure
78 consists of layers made up of two tetrahedrally-coordinated silicon atoms fused to an edge-
79 shared octahedral sheet of either aluminium or magnesium hydroxide. The layer thickness is
80 around 1 nm, and the lateral dimensions of these layers are around 100 nm. Stacking of the
81 layers leads to a regular van der Waals gap between the layers called the interlayer or gallery.

82 The incorporations of MMT nano platelets into the host matrix notably reduce the
83 crystallinity and increase the conductivity through efficient intercalation of polymers. It is also
84 observed that the incorporated nanoclays effectively minimize the ion pair formations. When the
85 polymer chains were confined and entangled by MMT nano platelets, the absorbed electrolytes
86 can migrate diagonally and enhance the conductivity, ion transfer and interfacial properties.
87 Recently, Lee *et al* discovered a novel method to exfoliate montmorillonite (MMT) prepared by
88 using soap-free emulsion polymerization in the presence of MMT to fabricate the nanocomposite
89 latices. The exfoliated MMT sheets were used in ionic liquid based gel electrolyte for dye
90 sensitized solar cells and a conversion efficiency of about 7.8% was achieved. The addition of
91 MMT platelets into gel/polymer electrolytes significantly reduced the charge transfer resistance
92 of various interfaces in DSSC.¹⁸⁻²²

93 There are numerous articles on MMT incorporated polymer nanocomposites for cation
94 exchange membrane and mechanical applications using solution intercalations and melt
95 intercalations techniques, respectively²³⁻²⁵. To the best of our knowledge, there are no
96 comprehensive articles on chemically exfoliated MMT incorporated PEO/PVDF-HFP composite
97 electrolytes for DSSC applications.

98 Apart from electrolytes, photo electrode plays the major role in highly efficient dye
99 sensitized solar cells. Various metal oxide semiconductors such as ZnO, ZrO₂, SiO₂, Al₂O₃ and
100 TiO₂ were employed as photo anode materials. Among them TiO₂ has been extensively used as a
101 photo anode material for dye sensitized solar cell applications, but its fast recombination of
102 electron/hole pairs and poor photo-response hinders the photo conversion efficiency. In order to
103 improve the performance of DSSC with TiO₂, several attempts are being carried out by the
104 research community. Another well known material for this application is ZnO, which has higher
105 electron mobility ($>150 \text{ cm}^2\text{V}^{-1}\text{S}^{-1}$) and is easily synthesized with various structures than TiO₂.
106 However, the formation of Zn²⁺/dye complexation on the ZnO nanoparticles surface cause harms
107 the stability and performance of the device²⁶⁻²⁹. When the small amount of ZnO nanoparticles (<
108 30%) are blended with TiO₂ or covered by TiO₂, it provides better surface morphology, surface
109 area, and transport along with photo electrochemical properties. The photo conversion efficiency
110 of ZnO/TiO₂ based nanocomposites is several orders higher than that of individual
111 components.³⁰

112 In this work, the aforesaid key issue of DSSCs has been taken into account for the
113 development of a solid state dye sensitized solar cell. In this regard, the MMT layered silicate
114 nano clays were chemically exfoliated / delaminated using simple chemical method with APS
115 coupling agent. Further, the effect of exfoliated nanosilicate platelets (modified MMT) on

116 structural, thermal, morphological, optical and electrochemical properties of PEO/PVDF-HFP
117 was studied. The performance of ZnO/TiO₂ nano composite as a photo electrode material was
118 also explored. The photovoltaic performances of solid state dye sensitized solar cells were
119 studied.

120 2. Experimental procedure

121 2.1. Materials

122 Poly(vinylidene fluoride-co-hexafluoropropylene) (PVdF-HFP, Kynar flex 2801) was
123 supplied by Arkema (India). Poly(ethylene oxide) (PEO) (Mw 5×10^6), aminopropyltrimethoxy
124 silane (APS), triton X-100, chloroplatinic acid hexahydrate and titanium IV- n-butoxide were
125 purchased from Sigma India. Zinc nitrate hexahydrate (Zn(NO₃)₂·6H₂O), sodium hydroxide
126 (NaOH), trisodium citrate dihydrate (HOC(COONa)(CH₂COONa)₂·2H₂O), anhydrous lithium
127 iodide (LiI) and iodine (I₂) were purchased from Thermo fisher scientific India. The dye, cis-
128 dithio-cyanate-N,N2-bis-(4-carboxylate-4-tetrabutylammoniumcarboxylate-2,2-bipyridine)
129 (N719) was purchased from Solaronix, Switzerland. Anhydrous dimethyl formamide (DMF),
130 ethanol, acetylacetone, nitric acid, propylene carbonate (PC) and ethylene carbonate (EC) were
131 purchased from Himedia, India. 4-*tert*-Butylpyridine (purity > 96%) and 1-Methyl-3-
132 propylimidazolium Iodide (purity > 95%) were purchased from TCI chemicals, India. The
133 conducting FTO glass plates ($R_{sh} < 10 \text{ ohm/cm}^2$) were supplied by M/s. Shilpa Enterprises, India.
134 Montmorillonite (Na⁺MMT) nanoclay was purchased from M/s Southern Clay products, USA.

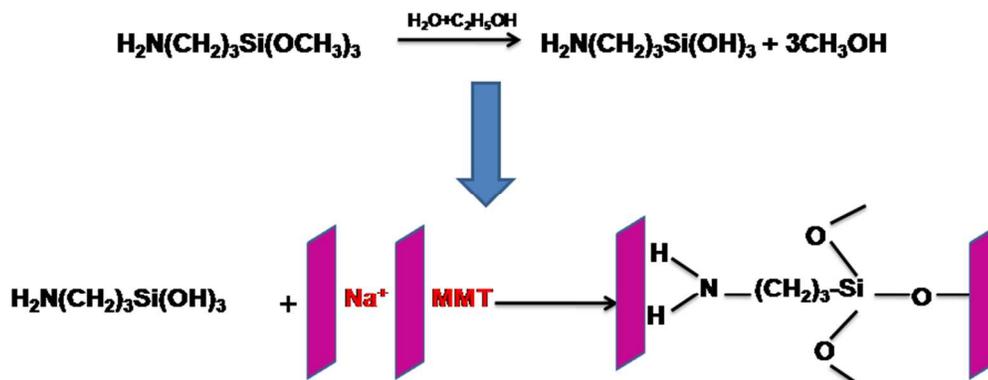
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137

138 2.2. Preparation of exfoliated MMT platelets (silane modified MMT)

139 The nanoclay was dried at 60°C for 10 h in a vacuum oven prior to use. The chemically
 140 exfoliated nanosilicate platelets (MMT) were prepared as mentioned the procedure with slight
 141 modification from our earlier report.³¹. For silane modification, 1 g of Na⁺MMT was dispersed
 142 in 75% ethanol solution (250 ml) and vigorously stirred at room temperature for 5 h. Then 2 ml
 143 of (3-aminopropyl) trimethoxysilane (APS) was slowly added into the Na⁺MMT nanoplatelets
 144 suspension and refluxed at 80°C for 24 h. Then the whole Na⁺MMT suspension was filtered and
 145 washed with distilled water to obtain the silane modified Na⁺MMT. In order to achieve the
 146 proper exfoliation the silane modified MMT dispersion was introduced into ultra sound
 147 sonication for 3 h. The silane modified MMT was named as e-MMT (exfoliated nanosilicate
 148 platelets). The chemical exfoliation reaction mechanism of Na⁺MMT is depicted in **Scheme 1**



149
150 **Scheme 1 schematic diagram of Silane modification**

151 2.3. Preparation of polymer blend nanocomposite electrolyte

152 An optimized composition of (6:4 wt %) of poly (vinylidene fluoride-co-
 153 hexafluoropropylene) (PVDF-HFP) and poly(ethylene oxide) (PEO) is dissolved in DMF (99%):
 154 glycerol (1%), (solvent and non-solvent, respectively) at 80 °C. Then, predried surface modified
 155 (e-MMT) nano platelets were slowly introduced into the polymer solution with different ratios

156 (1-10 wt %). The mixture was thoroughly mixed under continuous stirring for about 7-8 hours.
157 Subsequently, the mixtures were poured on the glass plate and the solvent was allowed to
158 evaporate at 100°C for 36 h.

159 **2.4. Porosity and electrolyte permeation measurements**

160 The porosity of the membranes was determined by employing gravimetric method. The
161 membranes were soaked in 1- butanol for 2 h and the membrane surface was dried by filter paper
162 afterwards. The membrane weight of the PEO/PVDF-HFP/eMMT membranes was measured
163 before and after absorption immersion.

164 The porosity was calculated using the following equation (1):

$$165 \text{ Porosity } \varepsilon (\%) = 100 \times (W_t - W_o) / \rho V \text{ ----- (1)}$$

166
167 Where, W_t and W_o is the weight of the wet and dry membrane, respectively. V is the apparent
168 volume of the membrane. ρ is the density of 1-butanol

169 To determine the electrochemical properties of blend, nanocomposite membranes were
170 soaked in an electrolyte solution containing 0.6M 1-Methyl-3-propylimidazolium Iodide, 0.1 M
171 LiI and 0.05 MI₂ and 0.5 M 4-tert-butylpyridine in EC/PC(1:1 wt%). The weight of the dried
172 and wet composite membranes was measured. After that, the excess liquid electrolyte on the
173 surface of wetted membrane was removed by wiping softly with a tissue paper. The electrolyte
174 uptake was calculated using the following equation (2)

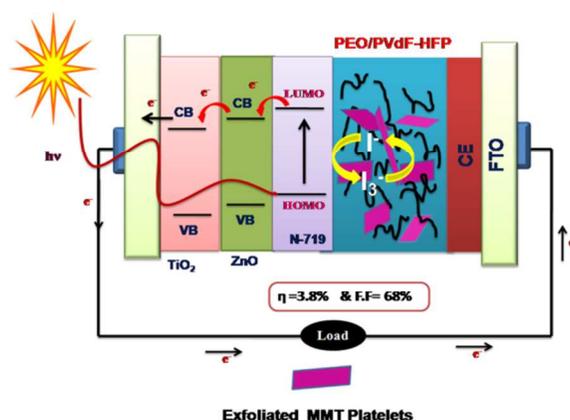
$$\text{Elcetrolyte uptake} = \frac{M_{wet} - M_{dry}}{M_{dry}} \times 100 \text{ ----- (2)}$$

175 Where, M_{wet} and M_{dry} are the weight of the membranes after and before soaking in the liquid
176 electrolyte.

177 **2.5. Construction and characterization of solid state dye sensitized solar cells (SSDSSC)**

178 The chemically synthesized ZnO nano sheets and TiO₂ nanoparticles were used as
 179 photoanode materials.^{29, 30, 32} The dye sensitized photoanode (N 719- ZnO/TiO₂) and counter
 180 electrode were prepared as mentioned in our earlier article.³² The optimized polymer composite
 181 electrolyte membrane was sandwiched between N-719 dye adsorbed TiO₂ /ZnO photo anode
 182 and Pt coated FTO glass plate with insulating spacer to avoid any short circuit. The schematic
 183 diagram of the prepared DSSC is shown in **Scheme 2**.

184



185

186

Scheme.2. Solid State sensitized solar cells

187

188 2.6. Characterization

189 FTIR spectra of Nanosilicate (NaMMT and modified MMT) PEO/PVDF-HFP composites
 190 were performed on a NICOLET 6700, USA FTIR spectrometer. X-ray analysis of samples was
 191 conducted in Shimadzu X-ray diffractometer, Japan using CuK_α (1.514Å) at a scanning rate of
 192 5° /min over a 2θ interval from 2° to 80°. SEM characterization of the samples was carried out
 193 using a scanning electron microscope (EVO MA 15, M/s Carl Zeiss SMT Ltd, UK). TEM
 194 analysis of TiO₂/ZnO nanoparticles was studied by M/S JEOL 1200EX, Japan. Images were
 195 taken at an accelerating voltage 100 kV. The mercury porosimetry (Carlo-Erba Porosimeter

2000) was carried out to measure the porosity of the membranes. The mercury diffused under an increasing pressure in the range of 0.01-100 MPa. DSC analysis of the polymer membranes was carried out using DSC, Q20, M/s TA Instruments (USA) equipment. Samples of ≤ 7 mg was heated from -60°C to 200°C , at the heating rate $5^{\circ}\text{C}/\text{min}$ under nitrogen atmosphere. The ionic conductivity measurements were carried out from electrochemical impedance spectroscopy technique by using WonATech-ZiveSP2 (Korea) electrochemical workstation. A perturbation of sinusoidal voltage signal of 10 mV was applied over a frequency range of 1 mHz-1MHz at room temperature. The impedance plots were fitted using Zview2-smart manager software. The samples were sandwiched between circular metal blocking electrodes with a diameter of 10 mm. The thickness of polymer nanocomposite electrolyte membranes was varied from 50 to 300 microns measured by thickness gauge. To characterize the ion ($\text{I}^{\ominus}/\text{I}_3^{\ominus}$) diffusion, steady state voltammetry was carried by CHI660 electrochemical work station and the scan rate of 5mV/s was applied to obtain the steady state current-voltage curves. The photovoltaic performance of the DSSCs is studied using Oriel Class-A Simulator (M-91900 A, Newport) with Xenon lamp as a light source having intensity of $100\text{mW}/\text{cm}^2$. A computer controlled Auto lab PGSTAT302N electrochemical workstation was used for J-V measurements. The active area of the cell was 0.5 cm^2 . To analyze the internal resistance of DSSC, EIS measurements were carried out by Auto lab PGSTAT302N electrochemical workstation with a signal frequency range from 1 mHz-1MHz under the illumination of 100 mWcm^{-2} .

215 3. Results and discussion

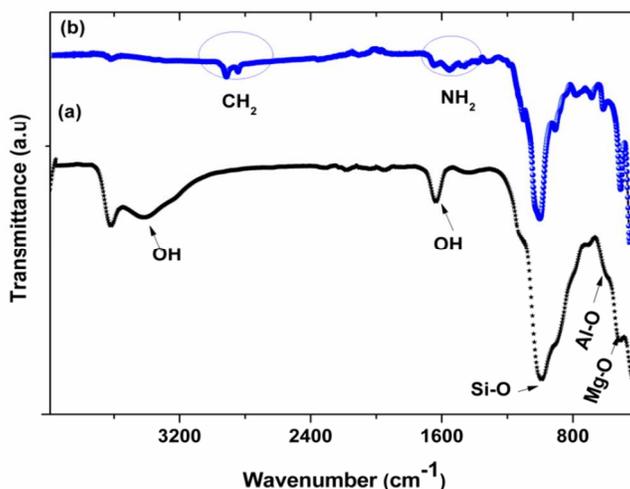
216 In the following discussion, first the chemical modification of nanosilicate platelets were
217 confirmed and their amount of addition within PEO/PVDF-HFP was optimized at 6 wt% based
218 on ionic conductivity measurements. In order to compare the optimum amount of exfoliated

219 MMT nanoplatelets (surface modified MMT) with pristine MMT counter parts, the same amount
220 of pristine MMT (6 wt %) was also separately added into the PEO/PVDF-HFP matrix. The
221 optimized compositions such as PEO/PVDF-HFP, PEO/PVDF-HFP-MMT (6 wt% MMT) and
222 PEO/PVDF-HFP-eMMT (6 wt% e-MMT) were further characterized and discussed.

223 3.1. Conformation of chemical modification nanosilicate platelets by FTIR analysis

224 **Fig.1** shows FTIR spectra of pristine clay (NaMMT) and chemically exfoliated
225 nanosilicate platelets (modified MMT). In the FTIR spectra of the Na-MMT (Fig. 1a), the bands
226 in the range of 3500 and 3700 cm^{-1} and near 3400 cm^{-1} are indicating montmorillonite. The
227 broad peak in the region of 3200-3510 cm^{-1} and 1640 cm^{-1} refers to -OH stretching and bending
228 signifying the existence of entrapped water molecules in the interlayer galleries of NaMMT. The
229 bands around 3620 and 3690 cm^{-1} are due to the presence of -OH stretching of Al-OH and Si-
230 OH of montmorillonite structure.³³

231



232

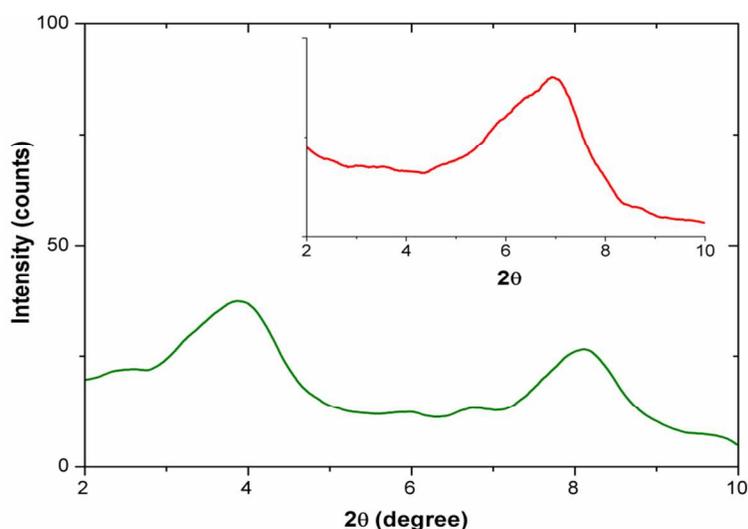
233 **Fig. 1 FTIR spectra of the (a) NaMMT and (b) silane-modified MMT (e-MMT)**

234 The Si-O and Al-O bonds are observed at 1034 and 620 cm^{-1} , respectively and the Mg-
235 O is assigned between 470 and 530 cm^{-1} .³⁴⁻³⁵ After treating with APS, new peaks were observed

236 at $2900\text{--}2780\text{ cm}^{-1}$, which is due to CH or CH_2 structure in the coupling agent. The asymmetric
237 stretching of Si-O-Si (1040 cm^{-1}) was enhanced after the silane modification. Moreover, after
238 APS modification, the intensity of hydroxyl group ($-\text{OH}$) peak at $3520\text{--}3200\text{ cm}^{-1}$ and 1640 cm^{-1}
239 was reduced, which indicates that silane moieties are attached on the surface of MMT nano
240 platelets via condensation reaction. The peak observed at 1556 cm^{-1} corresponds to $-\text{NH}_2$
241 bending of functional amino group present in the coupling agent.³⁵

242 3.2. Confirmation of exfoliation of nanosilicate platelets using WAXD

243 The WAXD analysis is used to examine the intercalation and exfoliation phenomena of
244 layered silicates. The XRD pattern of e-MMT (pristine Na^+MMT , inset Fig. 3) is presented in
245 **Fig. 3.** The WAXD of pristine Na^+MMT exhibits a diffraction peak (d_{001}) at 7.1° with d-space
246 about 13.2 \AA . After the introduction of silane molecules into Na^+MMT galleries (e-MMT), their
247 diffraction peak shifted to lower angle $2\theta = 3.94^\circ$ ($d = 20.96\text{ \AA}$). The increased basal spacing of
248 Na-MMT indicates that the introduced aminosilane groups were exfoliated into the silicate
249 layers.³¹



250

251 **Fig.2. WAXD patterns of exfoliated nanosilicate platelets (e-MMT) (inset Fig**
252 **pristine Na⁺MMT)**

253

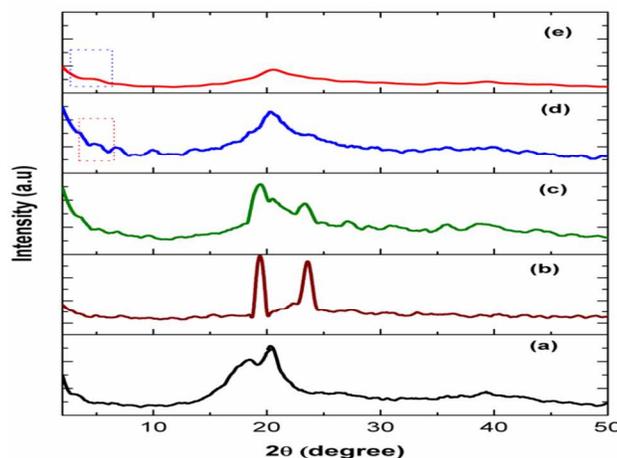
254 3.3. FTIR analysis of polymer nanocomposite membranes

255 FTIR Transmittance spectra of the PEO/PVdF-HFP and its composite membranes are
256 shown in **Fig. s1[†]**. The IR spectra of the electrolyte membranes were characterized in the range
257 of 4000-400 cm⁻¹ at room temperature. The main characteristics bands like CF₃ group, CH₂
258 rocking, CH₂ wagging of vinylidene, C-H bending (out of plane), C-F symmetric stretching, CF₂
259 stretching, CF₃ symmetrical stretching and CH₂ scissoring vibration of vinylidene group of
260 PVDF-HFP were observed at 759, 853, 870, 971, 1050, 1142, 1280, 1379 and 1407 cm⁻¹,
261 respectively. Similarly, the peaks observed at 1465 and 1354 cm⁻¹ is mainly due to the stretching
262 and bending modes of CH₂ in PEO, respectively.³⁶ The deformed vibration of CH₂ groups
263 appears at 1403 cm⁻¹.^{37,38} The strong absorption band around 1095 cm⁻¹ is assigned to the main
264 characteristic peak of non resolved anti-symmetric stretching and symmetric stretching of C-O-C
265 in polymer back bone of pure polycrystalline PEO.³⁹ The intensity of crystalline characteristics
266 peaks of PEO and PVDF-HFP decreased with the incorporation of pristine MMT nano clay and
267 exfoliated MMT nanoplatelets.

268 The overall results confirmed that the crystallinity of PEO/PVdF-HFP blend has been
269 completely distorted through the strong interactions with MMT layers and intercalation into the
270 galleries. The effect of nanoplatelets on its structural and thermal properties will be explained in
271 the forthcoming sections. The broad band absorptions of -OH stretching was observed in the
272 range of 3600-3200 cm⁻¹, which is attributed to the large number of intermolecular and
273 intramolecular hydrogen bonds formed by the interactions of solvent and PEO/PVDF-HFP. In

274 Na-MMT incorporated membrane, the surface hydroxyl group (-OH) within the MMT galleries
275 might be a reason of absorption in these regions. The intensity of -OH stretching was decreased
276 in surface modified MMT incorporated electrolyte membrane, which indicates the hydrolysis
277 condensation reaction of APS molecules present on the MMT nanoplatelets.

278 3.4. Structural studies of PEO/PVDF-HFP nanocomposites



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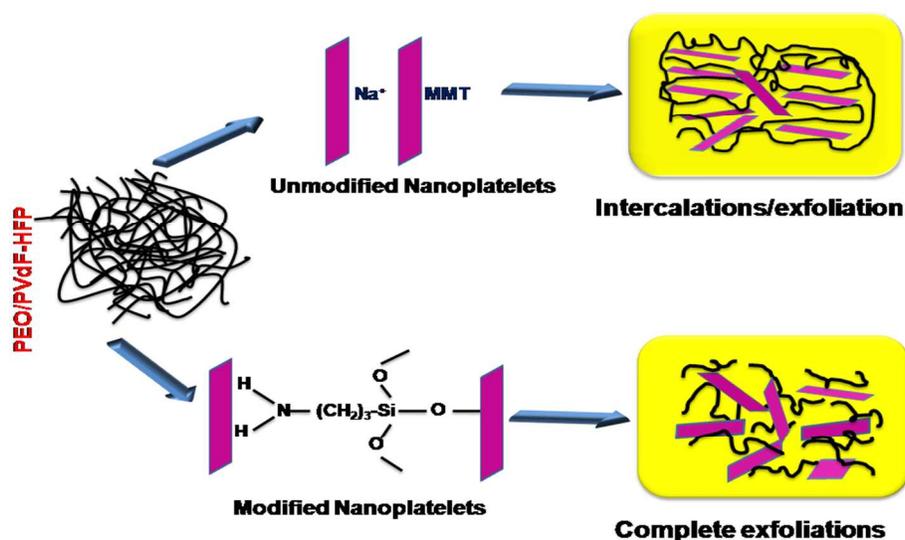
280 **Fig. 3 WAXD pattern of (a) PVDF-HFP, (b) PEO, (c) PEO/PVdF-HFP (d) PEO/PVdF-**
281 **HFP-MMT and (e) PEO/PVDF-HFP-eMMT nanocomposite**

282 **Fig. 3 (a-e)** shows the WAXD patterns of PVDF-HFP, PEO, PEO/PVdF-HFP blend and
283 its nanocomposite electrolyte membranes. The characteristics diffraction peaks of PVdF-HFP
284 and PEO were noticed at 18.4° , 20.3° , 26.9° and 19.36° , 23.52° for PVdF-HFP and PEO
285 respectively in Fig (3a & 3b).^{40, 41} The crystalline peaks of PVdF-HFP were marginally reduced
286 in blend system, which indicates that the PEO is properly blended with PVDF-HFP and the re-
287 crystallization was inhibited. These diffraction peaks were broadened and intensity was
288 decreased with the addition of MMT nano platelets. The decreased intensity may be due to more
289 intercalation of PEO/PVdF-HFP long range chain into the galleries of pristine NaMMT. The
290 absence of diffraction peak in e-MMT incorporated membrane confirms that more reduction in

291 crystallinity and this behavior might be originated by APS moieties attached in the galleries of
 292 nano platelets. The possible intercalation/exfoliations mechanisms are depicted in **Scheme 3**.
 293 Additionally, the shifting of characteristic (d_{001}) diffraction peak of pristine MMT ($2\theta=6.9^\circ$) and
 294 e-MMT ($2\theta = 3.94^\circ$) to lower angle side (**Fig. 3(d and e)**) confirmed that MMT nano platelets are
 295 properly dispersed in PEO/PVDF-HFP and might be completely exfoliated.³² The calculated
 296 structural (micro strain and d_{001}) parameters are given in **Table.1**. The increased amorphous
 297 nature of polymer matrix might lead to higher ionic conductivity through more amount of
 298 electrolyte uptake, which is discussed in next section .⁴²

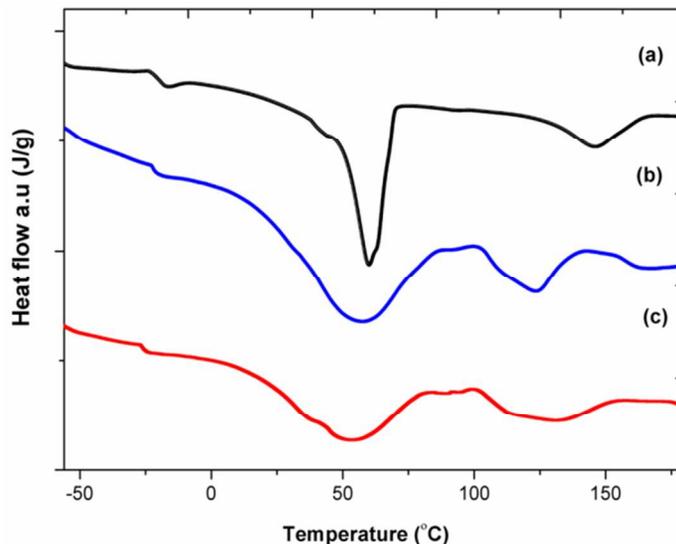
299 **Table 1: Structural parameters of PEO/PVDF-HFP composite electrolytes**

Sample	d-space (d_{001}) Å	Micro strain
6% NaMMT	14.54	0.2066
6% e-MMT	22.52	0.2736



Scheme 3 Intercalation/ exfoliation of nanoplatelets in PEO/PVdF-HFP blends

307 3.5. Thermal analysis



308

309 **Fig. 4 DSC thermograms of (a) PEO/PVDF-HFP (b) with pristine MMT and (c) e- MMT**
 310 **(Si-MMT) membranes**

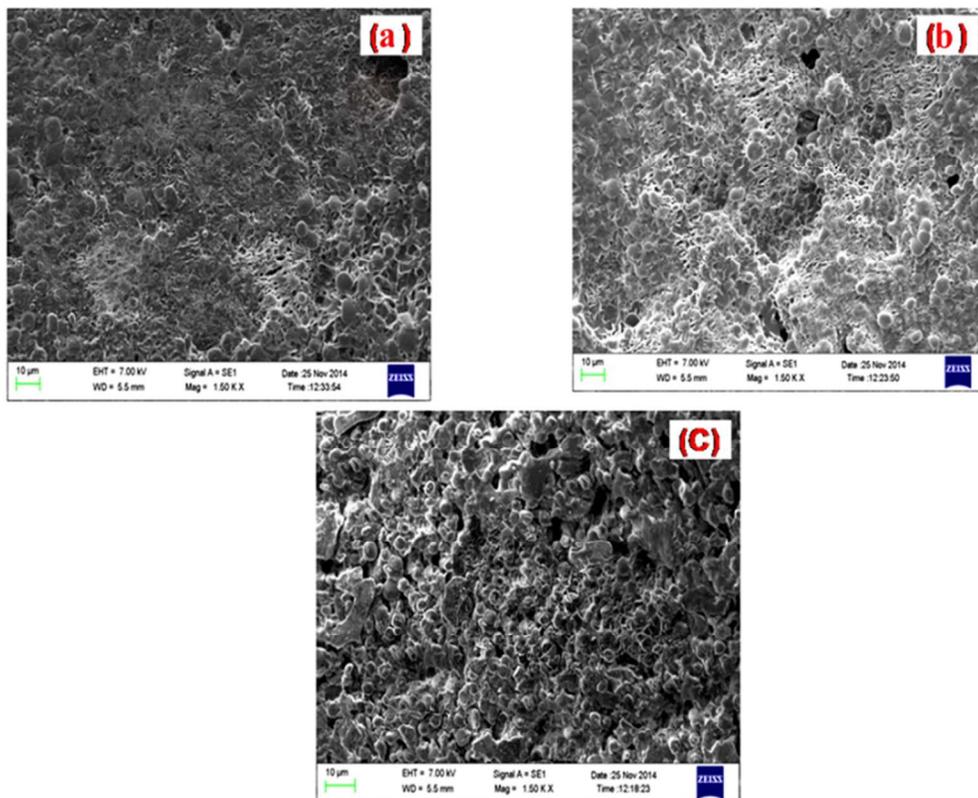
311 **Fig. 4** shows the DSC thermograms of PEO/PVDF-HFP with pristine NaMMT and chemically
 312 exfoliated MMT (silane modified MMT) membranes in range of temperature -60°C to 180°C . As
 313 shown in the figure, the melting peaks around -60°C and 150°C of PEO/PVDF-HFP blend are
 314 attributed to PEO and PVDF-HFP, respectively. The glass transition temperature of PEO/PVDF-
 315 HFP decreased with the addition of MMT nanoclays (see **Table. s1[†]**), which is attributed to the
 316 plasticization effect of nanoplatelets. The addition of nanoplatelets broadened the melting peak
 317 of PEO and the melting enthalpy decreased from 45.91 to 35.26 and 33.48 J/g for NaMMT and
 318 e-MMT respectively. This suggests that there is reduction in the crystallization of PEO and
 319 thereby enhancement in the fraction of free volume. The relative percentage of crystallinity ($\% \chi$)
 320 of PEO was calculated from the following **equation (3)** and summarized in **Table 2**.

$$321 \quad \% \chi = \frac{\Delta H_m}{\Delta H_m^0} \times 100\% \quad \text{----- (3)}$$

322 where, ΔH_m^0 is the heat of fusion of pure PEO, which is equal to 213.7 Jg^{-1} when, assumed as a
323 100% crystalline material, $^{43} \Delta H_m$ is heat of fusion of polymer blend electrolyte.
324 Similarly, melting enthalpy of PVDF-HFP also decreased from 27.12 to 23.72 J/g. From the
325 **Table.s1**[†], it is observed that reduction in crystallinity for silane modified MMT (chemically
326 exfoliated) is in agreement with XRD results. In e-MMT (silane modified) nanoplatelets
327 incorporated membrane, the MMT nano platelets were completely exfoliated and new silane
328 molecules such as Si-O-Si, NH₂ and CH₂ were anchored on the their surface, which might help
329 to increase the intercalation of polymer chains into the layered structure of nanoclay. The more
330 free volume of composite membranes can help for more electrolyte uptake and easy ions
331 transportation.

332 3.6. Surface characterization

333 The surface images of PEO/PVDF-HFP and its composite membranes are shown in **Fig.**
334 **5**. It is clearly visible that all membranes have porous structure. It is interesting to note that
335 number of pores are increased with the addition of nano platelets (pristine MMT and exfoliated
336 MMT). **Fig 5.c** shows the sponge-like structure and increased pores size (diameter ~ 3-5
337 microns) with the incorporation of surface modified nano platelets.

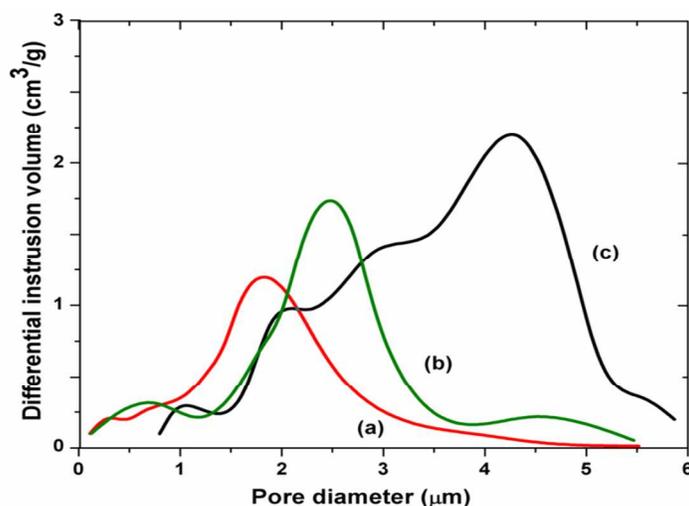


338

339 **Fig. 5: Surface SEM Images of (a) PEO/PVdF-HFP, (b) PEO/PVdF-HFP/MMT and (c)**
340 **PEO/PVdF-HFP/eMMT**

341 The porosity and electrolyte uptake measurements of PEO/PVDF-HFP are shown in **Fig.**
342 **s2[†]**. It is seen that porosity of PEO/PVDF-HFP membranes increased with the addition of nano
343 platelets and reaches maximum about 68 % for 6% e-MMT. Beyond this limit there is no
344 significant change in porosity with the further addition of clay. A similar behavior has been
345 observed by R. Prasanth et al, in PVDF-MMT clay composite electrolyte membranes for Li-ion
346 battery applications⁴⁴. Similarly, the amount of liquid electrolyte uptake increased up to 60% for
347 6 wt% of nanoplatelets and thereafter it decreased. The enhancement in electrolyte uptake with
348 the addition of fillers mainly attributed to the higher affinity of the MMT nanoplatelets toward
349 the solvents of the liquid electrolyte and ionic liquid⁴⁵. The electrolyte uptake is not only related

350 with the porosity of the membranes, but also to the crystallinity of PEO/PVDF-HFP blend. From
 351 the XRD and DSC, it was confirmed that modified MMT (exfoliated MMT) significantly
 352 reduced the crystallinity of PEO/PVdF-HFP.



353

354 **Fig. 6: Pore size distribution curve of (a) PEO/PVDF-HFP, (b) PEO/PVDF-HFP-MMT**
 355 **and (c) PEO/PVDF-HFP-eMMT.**

356 Fig. 6 shows the differential intrusion volume as a function of pore diameter. In Fig.6 (a),
 357 it is seen that the pore size of the pure PEO/PVDF-HFP was varying in the range of 0.2 - 4
 358 microns and approximate pore diameter of 1.9 micron. With the addition of pristine MMT nano
 359 silicate multiple peaks at 0.6, 2.7 and 4.65 micron were observed and majority of the pores are in
 360 the range of 2.7 microns. However in the case of chemically exfoliated nanosilicate plates
 361 incorporated PEO/PVDF-HFP membrane, multiple peaks with large pore size in the range of 4-5
 362 microns is seen. The observed results well agreed with the pore size measured from SEM.

363 The porosity of the membranes was calculated from following relation equation (4)

$$\text{Porosity } (P_M)\% = V_p W \times 100/V \quad \text{---(4)}$$

364 where, V_p is total pore volume (cm^3/g) which deduced from mercury intrusion curve, W is the
 365 weight of the samples, V is volume of the samples.⁴⁶ The porosity of PEO/PVDF-HFP increased
 366 with the addition of pristine MMT and chemically exfoliated nanosilicate platelets (e-MMT).
 367 The calculated porosity of PEO/PVDF-HFP composites is higher than porosity measured from
 368 porosimetry (P_M). In general, wet type liquid was used in gravimetric method to measure the
 369 porosity, hence the calculated porosity showed higher value. The average pore size, pore volume
 370 and porosity were measured using mercury porosimetry and is summarized in **Table 2**.

371 **Table 2 Porosimetry data of PEO/PVDF-HFP membranes**

Sample	Average Pore size (μm)	Total pore volume (cm^3/g)	Porosity measured from porosimetry (%)	Calculated porosity (P_M) %
PEO/PVDF-HFP	1.5	1.48	30	33
PEO/PVDF-HFP-MMT	2.3	1.96	42	48
PEO/PVDF-HFP-eMMT	4.6	2.9	56	60

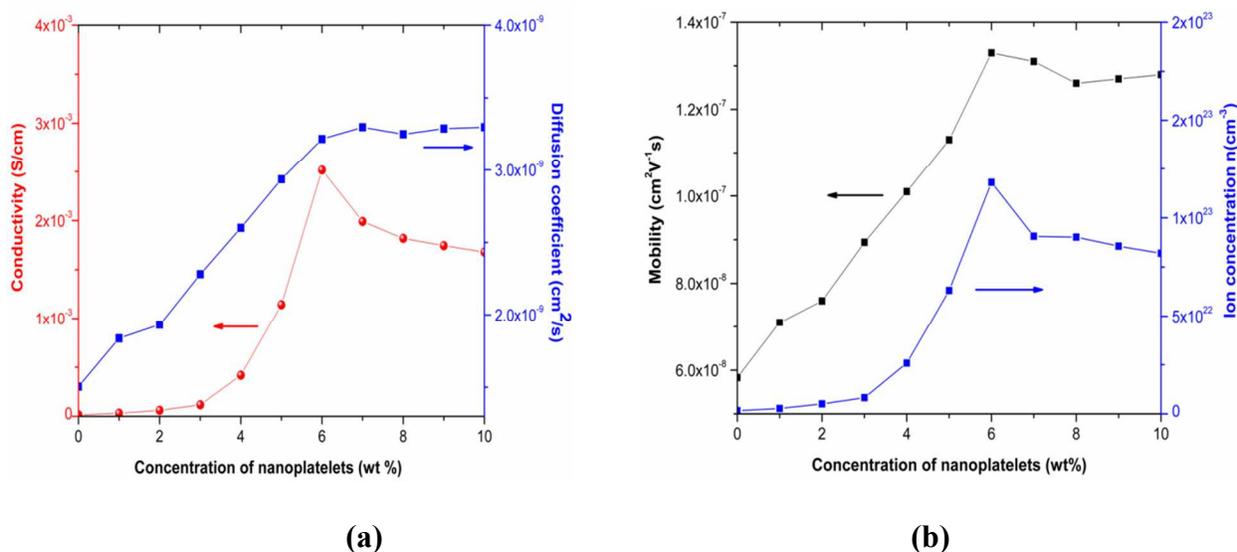
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373 3.7. Optical studies

374 The UV-Vis absorption spectra of MMT and silane treated PEO/PVDF-HFP composite
 375 electrolytes membranes are illustrated in **Fig. s3[†]**. The peaks of Γ^- and I_3^- observed about 230 nm
 376 and 290 nm for the samples, which is attributed to the absorption of Γ^- and I_3^- , respectively⁴⁰. The
 377 intensity of these two peaks was increased with the incorporation of silane modified
 378 nanoplatelets (e-MMT). This indicates that PEO/PVDF-HFP blend absorbed more amount of
 379 electrolyte due to their high free volume space. And also the concentration of Γ^- and I_3^- increased

380 due to the dissociation of salt by surface hydroxyl group and silane functional group in surface
 381 modified MMT nanoplatelets. The porosity and electrolyte uptake of modified MMT
 382 incorporated PEO/PVDF-HFP is higher than the PEO/PVDF-HFP/NaMMT membrane. From the
 383 above experimental findings it can be concluded that more free anions existed in e-MMT
 384 incorporated PEO/PVDF-HFP electrolytes and it is expected to improve the ion transport
 385 properties in DSSC.

386 3.8. Electrochemical characterizations of PEO/PVDF-HFP composite electrolytes



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 388
 389 **Fig. 7 (a) Ionic conductivity, diffusion Coefficient and (b) Ionic mobility, No. of charge**
 390 **carriers of PEO/PVDF-HFP electrolyte membrane with different amount of modified**
 391 **MMT nanoplatelets**

392 The conductivity, ionic mobility, ionic diffusion co-efficient and ionic concentrations of
 393 the PEO/PVDF-HFP gel electrolytes were characterized using EIS technique and shown in **Fig.**
 394 **7.** The bulk resistance of the electrolytes R_b was calculated from the interception of inclined
 395 straight line with the real axis in Nyquist plot (Z' vs Z''). The conductivity (σ) of the electrolytes
 396 was obtained using the following **Equation (5)**

$$\sigma = t/R_b A \quad \text{----- (5)}$$

397 Where, t is the thickness of the membrane, R_b is bulk resistance; A is the cross section
 398 area of electrode in contact with electrolyte membrane. Other ionic transport parameters such as
 399 diffusion coefficient (D), ionic mobility (μ) and charge carrier density were calculated by using
 400 the following equations (6, 7 and 8) with the help of fitting parameters and plotted as a function
 401 of different amount of nanoplatelets.^{47,48}

$$D = \frac{d^2}{\tau_2 \delta^2} \quad \text{----- (6)}$$

$$n = \frac{\sigma_{dc}}{e\mu} \quad \text{----- (7)}$$

$$\mu = \frac{eD}{kT} \quad \text{----- (8)}$$

402 Where d = thickness of sample/2, $\delta=d/\lambda$, $\lambda= \epsilon_0\epsilon A/k^{-1}$ is the electrical double layer thickness, A is
 403 the area of electrolyte membrane, ϵ_0 is vacuum permittivity, ϵ dielectric constant of polymer
 404 blend electrolytes $\tau_2=1/\omega_2$, ω_2 is the angular frequency at which the spike interest at the real
 405 impedance axis. The changes in ionic conductivity and diffusion coefficient of the PEO/PVdF-
 406 HFP blend electrolyte with the additions of exfoliated nanoplatelets/silane modified (e-MMT)
 407 were shown in **Fig. 7(a)**. The ionic conductivity of membrane is linearly increased with the
 408 addition of exfoliated nanoplates (Silane modified MMT) and attained optimal value of 2.52
 409 mS/cm at 6 wt%. This increasing trend of ionic conductivity is due to the incorporated
 410 nanoplatelets, which might influence the kinetics of high molecular weight polymer chain which
 411 promotes localized amorphous regions and porosity. The more volume of the pores and high
 412 amorphous nature of the blend help to uptake more electrolytes. The mobile charge carriers
 413 (iodide/tri-iodide) in the electrolyte can easily move into the porous and liquid like amorphous

414 structures. Hence, the more electrolytes cause an increase the ionic conductivity and diffusion
415 coefficient.

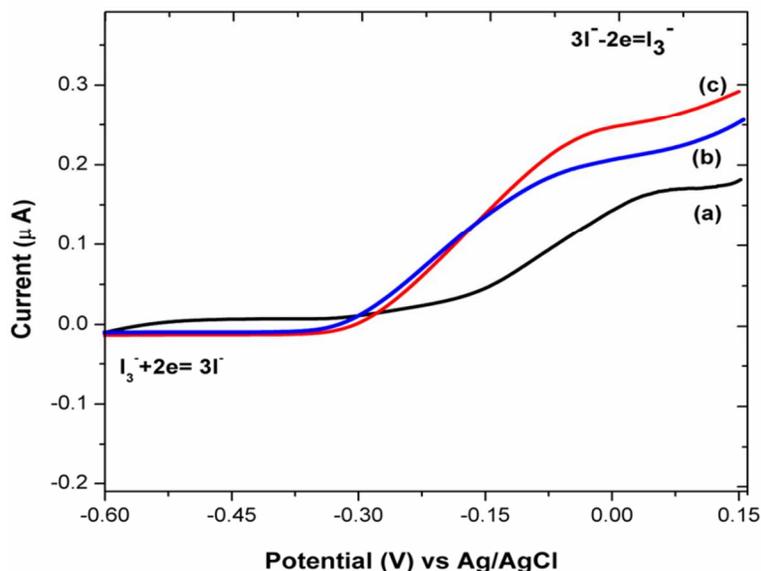
416 When the incorporation of nanoplatelets exceeds the optimum level (> 6 wt.%), it reduces
417 the ionic conductivity of PEO/PVDF-HFP electrolytes due to the negative effect of MMT
418 nanoplatelets: as more amount of nanoplatelets induces the aggregation of polymer chains; hence
419 the rate of recrystallization has increased.⁴⁹ This crystalline phase of polymer electrolytes block
420 the ions movement, hence ionic mobility and diffusion coefficient also decreased beyond the
421 optimum level of nano platelets as shown in **Fig 7 (a&b)**. The number of ions was found to be
422 same with further additions of MMT, which indicates the 6% wt can dissociate/ coordinate the
423 maximum number of Li^+ and gives the maximum ionic conductivity (**Fig. 7(b)**). For comparison,
424 ionic conductivity of same amount (6% wt) of untreated MMT nanoplatelets incorporated
425 PEO/PVdF-HFP was found to be about 5.41×10^{-4} S/cm and it is lower than the same amount of
426 silane modified MMT. In addition, the enhancement in ionic transport properties of silane treated
427 MMT is higher than its unmodified counterpart, which hints the possibility of additional
428 interaction between Li^+ in salt and newly attached surface functional groups within PEO/PVdF-
429 HFP blend electrolyte.

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 435 **Fig. 8. Steady state voltammograms of (a) PEO/PVdF-HFP (b) PEO/PVdF-HFP /MMT**
 436 **and (c) PEO/PVdF-HFP /eMMT electrolytes**

437 The transport properties of anion (I^-/I_3^-) redox couples in electrolytes are playing a vital role in
 438 photovoltaic characteristics of DSSCs. Linear sweep voltammetry technique was used to
 439 determine the diffusion coefficient of redox couples (I^-/I_3^-) in the polymer electrolytes. **Fig. 8**
 440 shows that steady state voltammograms for PEO/PVDF-HFP composite electrolytes at room
 441 temperature. From the cathodic and anodic steady state limiting current (I_{ss}), the apparent
 442 diffusion coefficient (D_{app}) of I_3^- and I^- , respectively can be derived from the following
 443 equation⁵⁰.

$$444 \quad I_{ss} = 4nRFCD_{app} \quad \text{-----} \quad (9)$$

445 where, n is the number of electrons involved in each electrochemical reaction, F is the Faraday
 446 constant, R is the radius of ultra microelectrode, and C is the concentration of electro active
 447 species.

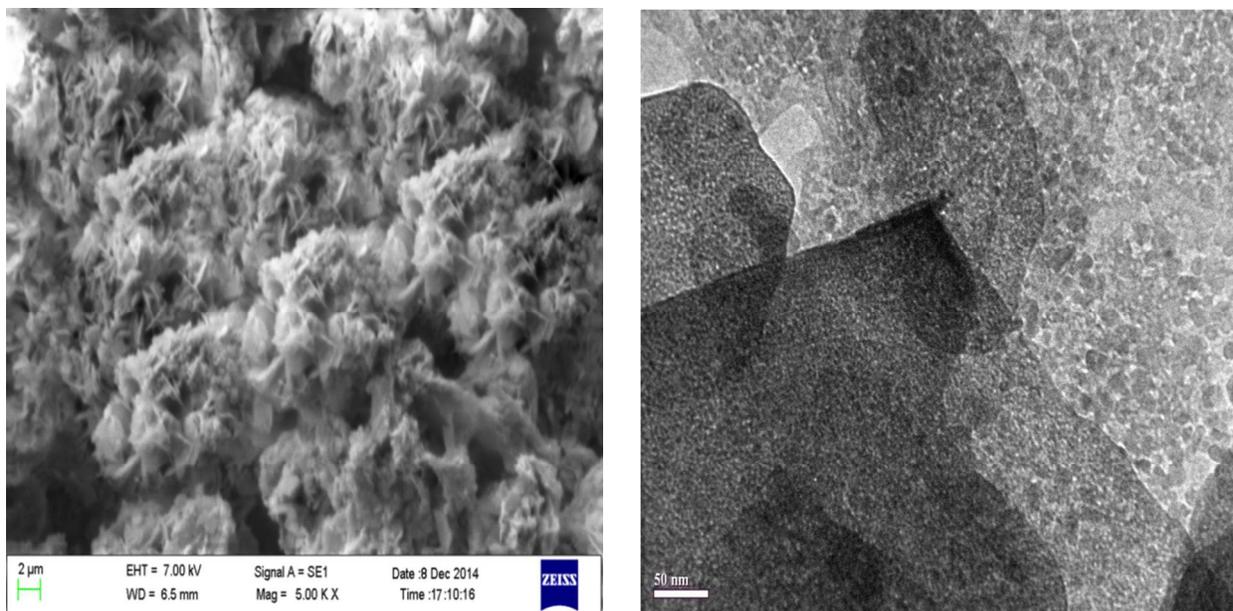
448 The addition of pristine MMT and e-MMT has marginally increased the diffusion
 449 coefficient I_3^- and I^- of the PEO/PVdF-HFP blend. This enhancement in diffusion coefficient

450 indicates that incorporated NaMMT and exfoliated MMT nanoplatelets hinders the long range
451 chain of high molecular weight polymers and facilitates the path way for ionic movements. It is
452 well known that dominant characteristics of I^- diffusion in I^-/I_3^- redox couples within the
453 PEO/PVdF-HFP nanocomposites is mainly due to the smaller ionic radius of I^- compared to I_3^- .⁵¹
454 These phenomena can be understood that the transport behavior of redox couples I_3^-/I^- in the
455 electrolyte was mainly through the physical diffusion in the liquid like phase called as free
456 volume space in amorphous nature of polymers. The DSC and WAXD results also support the
457 effect of rapid reduction in crystallinity of PEO/PVdF-HFP with the additions of modified MMT
458 nano platelets. The determined ionic diffusion coefficient (I_3^-/I^-) from LSV is also corroborated
459 with EIS results as shown in **Fig. 7** and mentioned in **Table s2**[†]. The increased ionic diffusion
460 coefficient (D_{app}) of PEO/PVdF-HFP blends is due to the uniformly distributed modified MMT
461 nanoplatelets. It is understood that the negative charges in the MMT nanoplatelets reduce the
462 attractive force between cation (Li^+) and anion (I^-) present in the electrolyte salt. Hence, apparent
463 diffusion coefficient has been increased.

464 **3.9. Morphological studies of TiO_2/ZnO (SEM and TEM)**

465 The morphology of TiO_2/ZnO composite was examined using SEM and TEM
466 micrographs. **Fig. 9** shows that agglomeration of TiO_2 spherical shaped nanoparticles or grains
467 covered on the flower like structure of nano-ZnO (Fig. 9 (a)). TEM image of TiO_2/ZnO reveals
468 hexagonal shaped thin sheets in the range of 0.5-1 microns (Fig. 9 (b)). These nano/micro sheets
469 combined each other and formed a flower like morphology. The spherical shaped TiO_2
470 nanoparticles were uniformly distributed on the nano/micro ZnO sheets. The selected area
471 diffraction patterns of TiO_2/ZnO revealed that polycrystalline nature of the nanocomposite.⁵²

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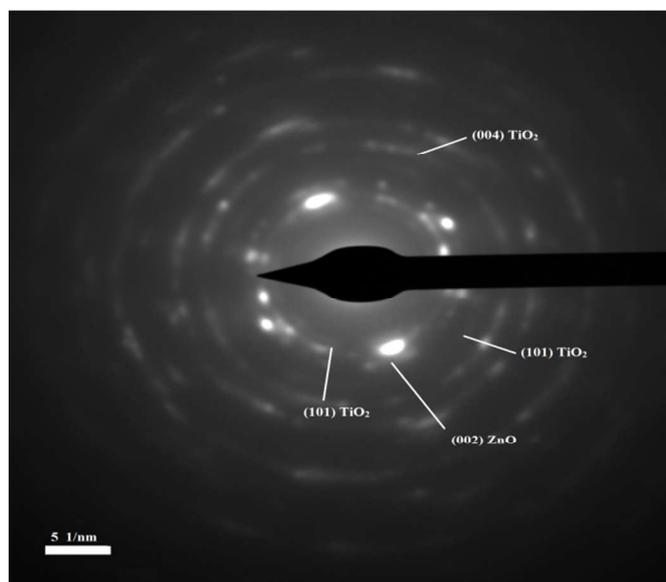


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(a)

(b)



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(c)

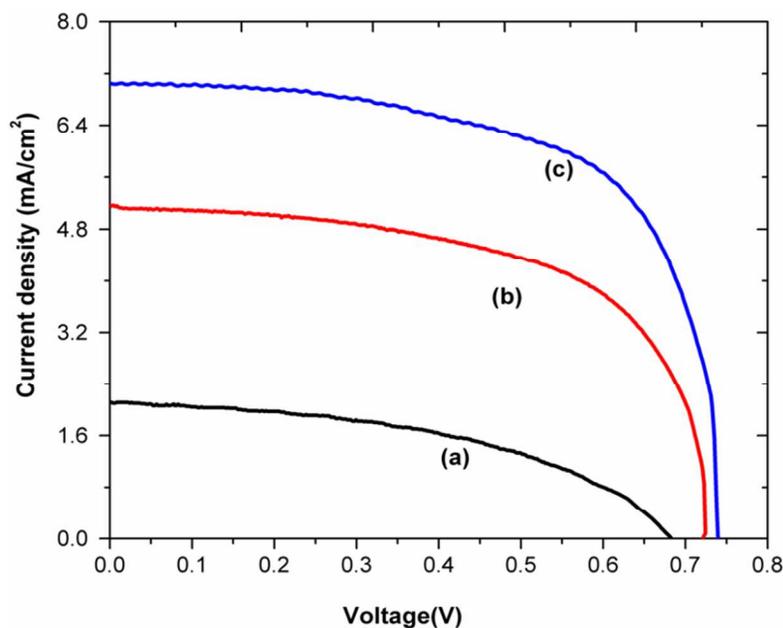
477 **Fig. 9 (a) SEM (b) TEM images of TiO₂/ZnO and (c) SAED pattern of TiO₂/ZnO**

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481 3.10. Photovoltaic performance study



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483 **Fig. 10** J-V characteristics curves DSSC based on (a) PEO/PVDF-HFP (b) PEO/PVDF-

484 HFP/MMT and (c) PEO/PVDF-HFP/eMMT

485 Photocurrent-Voltage (J-V) curves of dye sensitized solar cells (DSSC) were recorded

486 under the light illumination of 100 mW/cm^2 and are depicted in the **Fig. 10**. The measured487 photovoltaic parameters of all DSSC are summarized in **Table 3**. From the table, it observed that488 the addition of exfoliated nanoplatelets do not significantly change the open circuit voltage (V_{oc}),489 but photo-current density (J_{sc}) has been increased from 5.21 mA/cm^2 to 7.7 mA/cm^2 compared490 with pristine MMT. The enhancement in J_{sc} may be related to the reduction in charge transfer491 resistance with the additions of exfoliated MMT nanoplatelets. However, the gap between V_{max} 492 and V_{oc} showed a higher difference in all the samples, the values of $\Delta V/V_{oc}$ for PEO/PVDF-

493 HFP, PEO/PVDF-HFP/MMT and PEO/PVDF-HFP/e-MMT electrolytes were 33%, 19% and

494 17%, respectively, which indicated that the serial resistance of DSSC decreased. It is also

495 beneficial for higher fill factor (F.F). Among the electrolytes, chemically exfoliated nano

496 platelets (silane modified MMT) incorporated electrolyte shows the highest photovoltaic
 497 performance. The more amorphous phase of PEO/PVdF-HFP/e-MMT membrane provided easy
 498 path for ion migration and this can help rapid regeneration of dye from its oxidized states (D^+),
 499 hence performance enhanced. And also the high concentration of I^- and I_3^- ions (**Fig. s3[†]**) may
 500 be the reason to increase the transport properties and favors to achieve the maximum photo-
 501 current conversion efficiency ($\eta=3.8\%$).

502 **Table 3 Photovoltaic parameters for PVDF-HFP and its blend nanocomposite electrolytes**

Polymer Electrolytes	V_{oc}	J_{sc}	V_{max}	I_{max}	$\Delta V=V_{oc}-V_{max}$	F.F	Efficiency
	V	mA	V	mA	V	(%)	η %
PEO/PVDF-HFP	0.67	2.1	0.45	1.49	0.22	48	0.7
PEO/PVDF-HFP/MMT	0.71	5.2	0.58	3.96	0.13	62	2.3
PEO/PVDF-HFP/eMMT	0.73	7.7	0.61	6.30	0.10	68	3.8

503 V_{oc} : open circuit voltage (V), J_{sc} : short circuit current density (mA/cm²), V_{max} : maximum voltage (V), J_{max} : maximum current density, F.F:
 504 fill factor

505 The solar conversion efficiency (3.8%) of the present device is higher than the efficiency
 506 of 3.3% for pure PVDF-PEO based blend nanocomposite electrolytes and 2.82 % of ZnO/TiO₂
 507 photoanode based devices^{53, 54}. The comparison of solar conversion efficiency is summarized in
 508 Table 4.

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514 **Table 4 Comparison on photovoltaic parameters of similar DSSC**

515

Polymer Electrolytes	V_{oc}	J_{sc}	F.F	Efficiency	Reference
	V	mA	(%)	η %	
PEO/PVDF-HFP/e-MMT	0.73	7.7	68	3.8	Present
PEO/PVDF-HFP/modified TiO ₂ electrolyte	0.71	6.37	62	2.8	[40]
PEO/PVDF-HFP/SiO ₂ electrolyte	0.63	13.3	58	4.9	[55]
ZnO/TiO ₂ :liquid electrolyte	0.8	6.14	56	2.8	[54]
PEO/PVDF	0.54	2.7	42	0.6	[53]

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517

518 **3.11. Interfacial resistance of DSSC**

519 The charge transfer resistance of various interfaces in DSSC was measured by
 520 electrochemical impedance spectroscopy (EIS). The Fig. s4[†] shows Nyquist plots of DSSC based
 521 on TiO₂/ZnO nano sheet photo anode and PEO/PVdF-HFP with pristine MMT and e-MMT
 522 electrolytes. From, the EIS spectra, it is observed that there are three distinct semi-circles in the
 523 measured frequency range of 1 mHz –100 kHz. These semicircles are due to Nernst diffusion
 524 within the electrolyte, the electron transfer at the TiO₂-ZnO/electrolyte interface and the redox
 525 reaction at the platinum counter electrode. The experimental EIS data fitted with equivalent
 526 circuit and charge transfer resistance of each interface has been derived and summarized in
 527 **Table s2[†]**.⁵⁶

528 R_s is denoted as the series resistance of the electrolytes and electric contacts in the
 529 DSSCs. R_{ct1} , R_{ct2} and R_{diff} corresponds to the charge transfer process occurring at the Pt counter

530 electrode/electrolyte (corresponding to the first arc), the FTO/TiO₂/ZnO/electrolyte interface
531 (corresponding to the second arc) and the Warburg element of the ionic diffusion for the redox-
532 couple (I/I₃⁻) ion diffusion in the electrolyte (third arc), respectively.^{57,58} The DSSC based on
533 silane modified or exfoliated nanoplatelets incorporated PEO/PVDF-HFP electrolyte shows
534 smaller R_{ct2} and R_{diff} than unmodified counterpart. It is mainly due to the more number of pores
535 and increased electrolyte uptake which results in high I/I₃⁻ ion diffusion as well as high ionic
536 conductivity. Effective improvement was achieved in ionic conductivity, mobility, concentration
537 of ions in (section 3.7) EIS measurements and high diffusion coefficient of anions (I/I₃⁻) in LSV
538 studies (section 3.9) of PEO/PVDF-HFP/6% e-MMT composites.

539

540 4. Conclusion

541 In summary, poly(ethylene oxide)/polyvinylidene-hexafluoropropylene (PEO/PVDF-
542 HFP) with chemically exfoliated montmorillonite (MMT) nano platelets nanocomposite
543 membranes were successfully prepared for dye sensitized solar cells. The exfoliation of MMT
544 nano clays was achieved using aminopropyltrimethoxy silane and it was confirmed by Fourier
545 infrared spectroscopic analysis and wide angle x-ray diffraction (WAXD). X-ray diffraction and
546 differential scanning calorimetry studies confirmed that the addition of exfoliated nanoplatelets
547 significantly reduced the crystallinity of PEO/PVDF-HFP. The porosity and electrolyte uptake
548 increased with the addition of nanoplatelets. The mercury porosimetry results also suggested that
549 the pore volume and pore diameter has increased for MMT incorporated PEO/PVDF-HFP
550 nanocomposites. The UV-visible spectra indicated that the free ion concentration increased for
551 surface modified MMT. The addition of exfoliated nanoplatelets increase the ionic conductivity
552 up to 2.52×10^{-3} S/cm for 6 wt% and it decreased further, whereas unmodified counterpart

553 achieved about 5.41×10^{-4} S/cm. Besides that, ionic mobility, ion concentration and ionic
554 diffusion coefficient also found to increase with the incorporation of exfoliated MMT platelets. It
555 was observed from linear steady state voltammetry that the incorporation of surface modified
556 nanoplatelets reduce the ionic diffusion length, thereby enhancing the ionic diffusion coefficient
557 (Γ) upto 4.8×10^{-9} cm²/s. The solid state dye sensitized solar cell has been fabricated by using
558 optimized composition of polymer nanocomposites electrolyte and TiO₂/ZnO composite as a
559 photo anode. The presence of exfoliated MMT nanoplatelets in the electrolyte significantly
560 improves the photovoltaic performance of DSSC. The optimal solar conversion efficiency of
561 DSSC with exfoliated MMT nanoplatelets/ PEO/PVDF-HFP electrolytes is determined as 3.8%,
562 which is about 65% higher than that of the pristine MMT incorporated electrolyte based DSSC.
563 The results suggest that reinforcement of exfoliated MMT nanoplatelets is an effective approach
564 to enhance the transport properties within polymer electrolytes.

565

566 **Acknowledgement**

567 One of the authors (K.Prabakaran) is thankful to Prof. P. Ramasamy and Dr. M. Senthil Pandian,
568 SSN Research Centre, Chennai, Tamilnadu for their support of photo anode preparation. Also
569 thankful to Ms. Sriparna De Mr.R.K. Swain and B.Nayak, LARPM, CIPET, Bhubaneswar for
570 their assistance in morphological and thermal characterizations.

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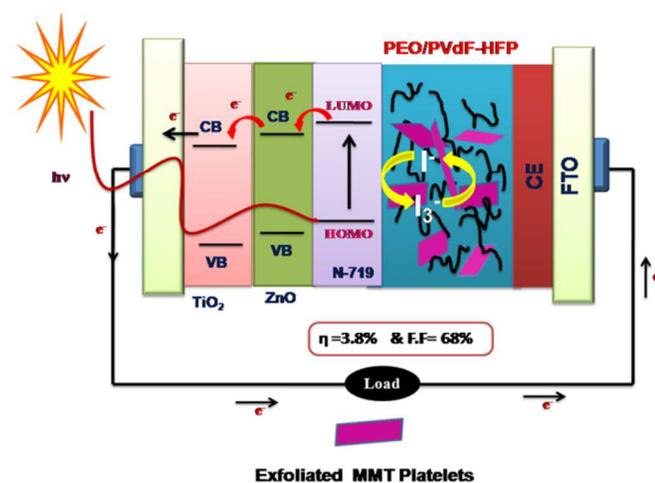
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Graphical Abstract



Exfoliated MMT nanoplatelets incorporated PEO/PVdF-HFP electrolyte and TiO₂/ZnO photoanode based DSSC showed improved solar energy conversion efficiency of about 3.8%.