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## CO oxidation mechanism on W(111) surface and W helical nanowire by the density functional theory calculation

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### Abstract

Two CO oxidation reactions ( $\text{CO} + \text{O}_2 \rightarrow \text{CO}_2 + \text{O}$  and  $\text{CO} + \text{O} \rightarrow \text{CO}_2$ ) were considered in the Eley-Rideal (ER) reaction mechanism. These oxidation processes on the W(111) surface and W helical nanowire were investigated by the density functional theory (DFT) calculation. The stable adsorption sites of  $\text{O}_2$  and O as well as their adsorption energies were obtained first. In order to understand the catalytic properties of W helical nanowire, the Fukui function and local density of state (LDOS) profiles were determined. The nudged elastic band (NEB) method was applied to locate transition states and minimum energy pathways (MEPs) of CO oxidation processes on the W helical nanowire and on the W(111) surface. In this study, we have demonstrated that the catalytic ability of W helical nanowire is superior to that of W(111) surface for the CO oxidation.

**Keywords:** W(111) surface, W helical nanowire, the density functional theory, local density of state, minimum energy pathway.

## Introduction

According to the past studies, it has been found that tungsten (W) and W-based metals possess good catalytic activities for some specific gas molecules, such as  $\text{CO}_x$  [1],  $\text{H}_2\text{S}$  [2], and  $\text{NO}_x$  [3]. For example, W carbides have been used as electrocatalysts, co-catalysts, catalyst supports, and electrolytes in different types of fuel cells. WC and  $\text{W}_2\text{C}$  are also demonstrated to possess a Pt-like characteristic in a variety of catalytic reactions [4]. Many previous studies have reported the interaction mechanisms of O [5-10],  $\text{O}_2$  [11], CO [1, 12], and  $\text{CO}_2$  [1] on the W single crystal surfaces. These studies can partially provide some useful information about the mechanism steps for the CO oxidation reaction including the adsorptions of reactants, dissociation of  $\text{O}_2$ , and desorption of  $\text{CO}_2$ .

The W(111) surface is more structurally open than the close-packed surfaces of W(100) and W(110). Therefore, the W(111) surface can accommodate more adsorbates per unit surface and has a much higher sticking probability. Chen *et al.* [12] explored the adsorption and dissociation mechanisms of the CO molecule on the W(111) surface by the DFT calculation. They found the activation energy of CO dissociation is about 0.8 eV, which is smaller than the barrier of a CO molecule desorbed from the surface. By the DFT calculation, the adsorption and dissociation mechanisms of CO and  $\text{CO}_2$  on W(111) surface were investigated in Chen's study [1]. The molecular structures, vibration frequencies, and binding energies of W(111)/ $\text{CO}_2$ , W(111)/CO, W(111)/C, and W(111)/O systems were reported. From the results of these studies,  $\text{O}_2$  adsorbed on a top (T) site (located at the top of a W atom) with the side-on configuration is the most stable configuration, whereas the O atom prefers to adsorb on the bridge site (located between two four-coordinated W atoms). In addition, the CO molecule also prefers to coordinate to a bridge site with the lying-down coordination where the C and O atoms are bound preferentially on the

bridge and top sites of metal surface, respectively.

In our previous study [13], we have investigated the CO oxidation processes in the Eley–Rideal (ER) and Langmuir–Hinshelwood (LH) mechanisms on the W(111) surface and W<sub>10</sub> nanoparticle. The two reaction barriers (0.16 and 0.62 eV) of pathway ( $\text{O}_{2(\text{ads})} + \text{CO}_{(\text{gas})} \rightarrow \text{O}_{(\text{ads})} + \text{O}_{(\text{ads})} + \text{CO}_{(\text{gas})} \rightarrow \text{CO}_{2(\text{gas})}$ ) for the W<sub>10</sub> surface are lower than the barrier (0.92 eV) for the W(111) surface, indicating that the formation of CO<sub>2</sub> on the W<sub>10</sub> surface occurs more easily than that on the W(111) surface. These results demonstrate that the shape and size effects of nano-materials have the strong influences on the catalytic activities and electronic properties of catalysts [14, 15]. For the one-dimension transition metal materials, An *et al.* [16] investigated the oxidation properties of CO molecule on the single-walled helical Au(5,3) nanowire by the DFT calculation. They found that the CO and O<sub>2</sub> were co-adsorbed on the Au(5,3) helical nanowire, and formed the peroxo-type (OOCO) in the exothermic process; it only needs to pass through a very small barrier. This result represents that the nanowire exactly possesses the high activity for the CO oxidation.

To understand the catalytic ability of one-dimension nanomaterial, the W nanowire was chosen to be the basic material for investigating CO oxidation mechanism and to compare with W(111) surface. The W nanowire structure with the lowest energy was constructed by the simulated annealing basin-hopping (SABH) method and the DFT calculation was further used to study the structural and adsorption properties of CO and O<sub>2</sub> on the W nanowire and on the W(111) surface. Furthermore, the reaction barriers and electronic properties in the CO oxidation processes were also investigated in this work. According to all reaction barriers for the CO oxidation mechanisms on the W<sub>10</sub> surface and on the W(111) surface in our previous study [13], it can be seen both the W<sub>10</sub> nanoparticle and W(111) surfaces prefer the ER mechanism. Consequently, only the ER mechanism was considered in the current study and the

co-adsorption of CO<sub>2</sub> and O<sub>2</sub> for LH mechanism was not discussed. We believe this study is vital for understanding the factors affecting the CO oxidation catalyzed by the nanometer-scale tungsten nanowire.

### Simulation model

All DFT calculations were employed by the DMol<sup>3</sup> package with the spin polarization [17]. To investigate the adsorption properties of molecules on the metal surface, the generalized gradient approximation (GGA) with the revised Perdew–Burke–Ernzerhof (rPBE) [18] density functional was used. The calculations were carried out by using the  $(1 \times 1 \times 6)$  and  $(3 \times 4 \times 1)$  Monkhorst–Pack mesh k-points for the W helical nanowire and W(111) surface, respectively. For all calculations, the convergences of total energies for the electronic step and the ionic step were set as  $1 \times 10^{-5}$  Ha and  $2 \times 10^{-5}$  Ha, respectively.

The structure of W helical nanowire is shown in Figure 1(a). This structure possesses the global minimal energy among all structures by the SABH method search [19] with the tight-binding (TB) potential [20]. The detailed global minimal search procedure for the W nanowire can be found in our previous study [21]. The structure from the SABH was then optimized by the DFT calculation, and the optimized structure has the length of 27.9 Å along the periodic axial direction. In the configuration of W helical nanowire, four one-atom chains entangle with a helical angle of 12.07° to form a helical structure, which is designated as the helical nanowire. The helical nanowire is quite similar to the structure found in the previous study by using a TB potential through the genetic algorithm search [22].

For the W(111) surface, the calculations for the CO adsorption on  $p(3 \times 2)$  and  $p(3 \times 3)$  lateral cells of the W(111) surface were conducted, which correspond to the

coverage of 1/6 ML and 1/9 ML. These studies show that the coverage effect on the calculated stability of this species is negligible (smaller than 0.1 eV). Therefore, the computationally less expensive  $p(3 \times 2)$  W(111) surface was used as the simulation model. The  $p(3 \times 2)$  lateral cell of W(111) surface was modeled as the periodically repeated slab with 6 layers, as shown in the right panel of Figure 2. The bottom three atomic layers were kept frozen and set to the experimentally estimated bulk parameters, and the remaining layers were fully relaxed during the calculations. The lateral cell has dimensions of  $a = 13.5 \text{ \AA}$ ,  $b = 7.80 \text{ \AA}$ , and  $c = 17.47 \text{ \AA}$  including a vacuum region of thickness 15  $\text{\AA}$  upon the W(111) substrate, which guarantees no interactions between the upper and lower slabs of the W(111) substrate.

In this study, the adsorption energies were calculated according to the following equation:

$$\Delta E_{\text{ads}} = E[\text{total}] - (E[\text{substrate}] + E[\text{CO}] + E[\text{O}_2(\text{or O})]) \quad (1)$$

In the aforementioned equation, the  $E[\text{total}]$ ,  $E[\text{substrate}]$ ,  $E[\text{CO}]$ , and  $E[\text{O}_2(\text{or O})]$  correspond to the electronic energies of adsorbed species on the W substrate, the bare W substrate, gaseous CO, and gaseous O<sub>2</sub> (or O), respectively. Here  $E$  is the electronic energy calculated at 0 K in vacuum. Therefore, the negative adsorption energy indicates a stable adsorption state.

The nudged elastic band (NEB) method [23, 24] was applied to locate the transition states and the corresponding minimum energy pathways (MEPs). The NEB method is an efficient method for finding the minimum energy pathway between the given initial and final states of a transition state. Sixteen images were used to locate each transition state (TS) for the NEB method.

## Results and discussion

To make certain that our DFT approach is appropriate for the tungsten material,

Table 1 lists the calculated lattice constant and the cohesive energy compared with experimental values of tungsten body center cubic (BCC), face centered cubic (FCC), and hexagonal close packed (HCP) lattices [25, 26]. The cohesive energies of different tungsten configurations are close to the bulk values in experiments. Moreover, the lattice constant is 3.16 Å in BCC, which is also in agreement with the experimental values of 3.16 Å and 3.25 Å [26, 27]. The functional, basis set, and core treatment of rPBE, DND, and DSPP were employed to calculate the bond length, vibration frequency, and dissociation energy of tungsten dimer and the calculation results are listed in Table 2. Comparing with previous experimental and theoretical studies, the bond length of dimer is 2.075 Å, and the frequency and dissociation energy are 334.2 cm<sup>-1</sup> and 4.74 eV, which are very close to the experimental values [28, 29]. These results represent the setups of rPBE/DND/DSPP are reasonable in this work. In the calculations of adsorbents, the bond distances of gas phase O<sub>2</sub>, CO and CO<sub>2</sub> were also predicted to be 1.232 Å (O–O for O<sub>2</sub>), 1.147 Å (C–O for CO) and 1.177 Å (C–O for CO<sub>2</sub>) in a 15×15×15 Å<sup>3</sup> cubic box, which are also in good agreement with available experimental data and theoretical results (1.21 Å for O–O of O<sub>2</sub> [30]; 1.15 Å for C–O of CO [31]; 1.16 Å for C–O of CO<sub>2</sub> [31]).

It is difficult to recognize and determine the regions of activation directly for the helical W nanowire, because the possible adsorption sites of helical W nanowire are not simply symmetrical as those of perfect crystal surfaces. In our previous study for finding the most reactive adsorption site of a W<sub>10</sub> nanocluster [13], three possible criterions including the orbital roughness [31], Fukui function [32], and the coordination number [33] were used. The Fukui function was found to be the best criterion for finding the most reactive adsorption sites for CO oxidation process. In this study, Fukui indices were computed for describing the electronic distribution.

Parr *et al.* stated the atom with the largest value of Fukui function is associated with the most reactive site [34]. They found that a site having a larger  $f_k^-$  value is a better electron donor; whereas one having a larger  $f_k^+$  value is a better electron acceptor. The Fukui function  $f(r)$  is defined either as the first derivative of the chemical potential with respect to the external potential  $V(r)$  at a constant number of electrons  $N$  or as the first derivative of the electronic density  $\rho(r)$  with respect to the number of electrons  $N$  at constant external potential  $V(r)$ :

$$f(r) = \left[ \frac{\delta u}{\delta V(r)} \right]_N = \left[ \frac{\partial \rho(r)}{\partial N} \right]_V \quad (2)$$

Because  $\rho(r)$  as a function of  $N$  has slope discontinuities, three reaction indices of governing electrophilic attack, governing nucleophilic attack, and governing radical attack are provided as follows:

$$f^-(r) = \left[ \frac{\partial \rho(r)}{\partial N} \right]_V \quad (3)$$

$$f^+(r) = \left[ \frac{\partial \rho(r)}{\partial N} \right]_V \quad (4)$$

$$f^0(r) = 1/2[f^+(r) + f^-(r)] \quad (5)$$

Yang and Mortier [35] defined  $f(r)$  in a condensed form; these condensed Fukui functions of an atom  $k$  in a molecule with  $N$  electrons are defined as Eqs. (6), (7), and (8):

$$f_k^+ = [q_k(N+1) - q_k(N)] \quad (6)$$

$$f_k^- = [q_k(N) - q_k(N-1)] \quad (7)$$

$$f_k^0 = 1/2[q_k(N+1) - q_k(N-1)] \quad (8)$$

in which  $f_k^+$ ,  $f_k^-$ , and  $f_k^0$  represent nucleophilic, electrophilic, and radical attack, respectively;  $q_k$  is the electronic population of atom  $k$  in a molecule. Gázquez *et al.* [36] stated that the largest value of the Fukui function is, in general, associated with the most reactive site. According to the Fukui function analysis, when  $O_2$  and  $CO$  are adsorbed on the W metal surface, both molecules are electronic acceptors, indicating



the distribution of  $f_k^-$  should be considered. In the current study, the  $f_k^-$  values were computed by using the Hirshfeld charge according to Eq. (7). Therefore, the distribution of Fukui function ( $f_k^-$ ) are shown in Figure 3(a) for determining different adsorption sites for  $O_2$  and O on the W nanowire. In Figure 3(a), the W atoms marked in red have the larger  $f_k^-$  values, indicating these sites have the higher activity than those marked in blue. It is clear that the helical structure consists of two W one-atom chains with higher reactivity and the other two with lower reactivity, helically entangling with one another. For convenience to present our DFT results, the top sites with higher activity (larger  $f_k^-$  values) are designated as  $T_H$ , while those with low activity (smaller  $f_k^-$  values) are designated as  $T_L$ . In Figure 3(b) and 3(c), the local structures around the  $T_H$  and  $T_L$  are displayed, and the bond between any two W atoms is shown if the distance between these two W atoms is shorter than 2 times Van der Waals radius of W atom (4.20 Å). In Figure 3(b), one can see the bond lengths between two nearest  $T_H$  atoms of the same one-atom chain range from 2.711 Å to 2.832 Å, which are slightly longer than those between  $T_H$  and  $T_L$  atoms ranging from 2.624 Å to 2.645 Å. In Figure 3(c), the  $T_L$  atom is bounded by its six nearest W atoms with similar bond lengths. According to the geometrical information about  $T_H$  and  $T_L$  atoms displayed in Figure 3(b) and 3(c), three types of adsorption sites were considered including top sites ( $T_H$ ,  $T_L$ ), bridge sites ( $B_{H-H}$ ,  $B_{H-L1}$ ,  $B_{H-L2}$ ,  $B_{L-L}$ ) and hollow sites ( $H_{HHL}$ ,  $H_{HLL}$ ).

In our previous study [13], three CO oxidation reactions of  $CO + O_2 \rightarrow CO_2 + O$ ,  $CO + O + O \rightarrow CO_2 + O$ , and  $CO + O \rightarrow CO_2$  were considered in ER reaction mechanisms by the  $W_{10}$  nanocluster. The intermediate states including the carbonate ( $CO_3$ ) and peroxo-type ( $OOCO$ ) are not found for the LH mechanism, because  $O_2$  is easily decomposed to two O atoms on  $W_{10}$ , and CO is easier to react with  $O_2$  rather

than O atom. Therefore, only the ER mechanism on the W(111) surface and W helical nanowire was concerned. The optimized geometries of O<sub>2</sub> and O adsorbed on the W helical nanowire are shown in Figure 4 and the corresponding adsorption energies are listed in Table 3.

For O<sub>2</sub>-T<sub>H</sub>-W<sub>wire</sub> and O<sub>2</sub>-T<sub>L</sub>-W<sub>wire</sub>, the O<sub>2</sub> molecule was adsorbed at two different top sites of W helical nanowire. The bond lengths of W-O and O-O for O<sub>2</sub>-T<sub>H</sub>-W<sub>wire</sub> are 1.852 Å and 1.326 Å (Fig. 4(a)). For O<sub>2</sub>-T<sub>L</sub>-W<sub>wire</sub>, the bond lengths of W-O and O-O are 1.832 Å and 1.322 Å (Fig. 4(b)), respectively. For O<sub>2</sub>-B<sub>H-H</sub>-W<sub>wire</sub>, O<sub>2</sub>-B<sub>H-L1</sub>-W<sub>wire</sub> and O<sub>2</sub>-B<sub>L-L</sub>-W<sub>wire</sub>, the bond lengths of two W-O bonds and the O-O bond for O<sub>2</sub>-B<sub>H-H</sub>-W<sub>wire</sub> are 2.030, 2.078 Å and 1.392 Å (Fig. 4(e)), respectively. For O<sub>2</sub>-B<sub>H-L1</sub>-W<sub>wire</sub>, the bond lengths of two W-O bonds and the O-O bond are 2.217, 2.048 and 1.367 Å (Fig. 4(f)), respectively. For O<sub>2</sub>-B<sub>L-L</sub>-W<sub>wire</sub>, the bond lengths of two W-O bonds and the O-O bond are 2.034, 2.103 Å and 1.392 Å (Fig. 4(i)), respectively. Furthermore, Because of the restriction of geometric configuration, the O<sub>2</sub> at B<sub>H-L2</sub> is decomposed to two O atoms. For the O<sub>2</sub> adsorption, the larger adsorption energies are represented on T<sub>H</sub> and B<sub>H-H</sub> sites in Table 3, the energies are -2.000 and -2.268 eV, respectively.

For the O adsorbed on the W helical wire surface, the bond lengths of W-O on O-T<sub>H</sub>-W<sub>wire</sub> and O-T<sub>L</sub>-W<sub>wire</sub> sites are 1.759 Å and 1.748 Å (Figs. 4(c) and 4(d)). On the bridge sites, the bond lengths of two W-O bonds are 1.957 and 1.992 Å for O-B<sub>H-H</sub>-W<sub>wire</sub> (Fig. 4(g)), 2.009 and 1.970 Å for O-B<sub>H-L1</sub>-W<sub>wire</sub> (Fig. 4(h)), and 1.969 and 1.891 Å for O-B<sub>L-L</sub>-W<sub>wire</sub> (Fig. 4(j)). In addition, the bond length of O<sub>2</sub> on T<sub>H</sub> site is longer than that of O<sub>2</sub> on T<sub>L</sub> site by 0.03 Å, and the weaker bonding leads more easily broken in the oxidation process. Because the O<sub>2</sub> molecule is easily dissociated to two O atoms on the hollow sites of T<sub>H</sub>-T<sub>H</sub>-T<sub>L</sub>, and T<sub>H</sub>-T<sub>L</sub>-T<sub>L</sub>, the hollow sites are

not considered for adsorption energy calculations. Besides, both bond lengths of O<sub>2</sub> molecule on T<sub>H</sub> and T<sub>L</sub> sites are longer than the gas-phase oxygen bond, because of the charge transfer from the W helical nanowire to the O<sub>2</sub> molecule. The charge transfer causes the increase of electronic density in the antibonding  $\pi_g^*$  orbit of oxygen for the W and O orbital overlapping, resulting in the increment of the O-O bond length. Moreover, although the O atoms adsorbed on the W helical nanowire have larger absorption energy at the bridge sites, it is not the most favorite site for finding barrier. In addition, it should be mentioned that previous studies [37-40] show that the CO<sub>3</sub> and OOCO may exist on gold (Au) and silver (Ag) catalysts, which would be the important precursor states for CO<sub>2</sub> formation. However, both stable adsorption configurations of CO<sub>3</sub> and OOCO species on the W<sub>10</sub> and W(111) surfaces are not found in our previous work.

### **Reaction profile for CO oxidation on the W helical nanowire and W(111) Surfaces**

On the basis of adsorption results shown in Table 3, the reaction mechanism for CO oxidation on the W helical nanowire and W(111) surfaces were investigated. To find MEP for CO oxidation on W metal surface, the ER reaction mechanism was considered. For the W(111) surface, the CO+O<sub>2</sub> and CO+O energy profiles of CO oxidation process are shown in Figure 5. The reaction of the pathway **I** can be described as O<sub>2(ads)</sub>+CO<sub>(gas)</sub> → O<sub>(ads)</sub>+CO<sub>2(gas)</sub> as shown in Figure 5(a). In pathway **I**, the O<sub>2</sub> adsorbs at Top (I) site. In Figure 5, the pathway I of the ER mechanism starts from the individual CO and O<sub>2</sub> molecules in the gaseous phase as the reactant W(111)+O<sub>2</sub>+CO. In the next step, a CO molecule adsorbs on the O<sub>2</sub> molecule. After the dissociation of the adsorbed O<sub>2</sub>, the CO directly reacts with O and the energy barrier is 0.70 eV during the reaction. As previously described, because the O<sub>2</sub>

dissociates easily on the W surface, the pathway **I** oxidation reaction is more likely to occur, and thus the small energy barrier was observed in this pathway. In pathway **II**, the other reaction  $\text{O}_{(\text{ads})} + \text{CO}_{(\text{gas})} \rightarrow \text{CO}_{2(\text{gas})}$  is shown in Figure 5 (b), the CO adsorbs directly on the O atom, and then  $\text{CO}_2$  forms by passing through the energy barrier of 1.314 eV. Comparing with pathway **I**, the barrier of pathway **II** is relatively larger, because the O atom possesses the stronger bonding on the W surface. This result indicates that CO is not easy to react with the single O atom, and needs to pass through the high barrier for forming  $\text{CO}_2$  molecule.

A similar ER pathway analysis as those on the W(111) surface was also repeated for the W nanowire surface. The potential energy diagrams of  $\text{CO} + \text{O}_2$  and  $\text{CO} + \text{O}$  reactions on the W nanowire are given in Figures 6 (a)-(c), respectively. The pathway **I** (Figure 6(a)) is considered via the reaction:  $\text{O}_{2(\text{ads})} + \text{CO}_{(\text{gas})} \rightarrow \text{O}_{(\text{ads})} + \text{CO}_{2(\text{gas})}$ . In this reaction, the end-on adsorption of  $\text{O}_2$  on the W nanowire surface is considered as the intermediate wire- $\text{O}_2 + \text{CO}_{\text{IM1}}$ . To form the final product,  $\text{CO}_{2(\text{gas})}$ , the  $\text{CO}_{(\text{gas})}$  directly extracts one O atom of adsorbed  $\text{O}_{2(\text{ads})}$  with the end-on configuration via the transition state with an energy barrier of 0.468 eV. In addition, the adsorbed  $\text{O}_2$  with the side-on configuration is not shown in Figure 6(a) because the  $\text{O}_2$  bond is always broken when the gas CO is close to the adsorbed  $\text{O}_2$  in our calculations. Another possible pathway for  $\text{CO}_2$  oxidation is considered by the attachment of CO to an atomic O on the surface directly. This reaction can be described as pathway **II** of the ER reaction ( $\text{O}_{(\text{ads})} + \text{CO}_{(\text{gas})} \rightarrow \text{CO}_{2(\text{gas})}$ ), as shown in Figure 6(b). In this reaction mechanism, a pathway where the CO molecule reacts with an O atom at the top site ( $\text{T}_\text{H}$ ) was concerned. In Figure 6(b), the calculated reaction barrier is 1.529 eV for the formation of  $\text{CO}_2$  because the bonding of O on the W nanowire is too strong to react with CO. This result indicates that a much higher energy is required for the formation of  $\text{CO}_2$  for an adsorbed O atom when compared to that of  $\text{O}_2$ .

According to the results of the former adsorption energy calculations as shown in Table 3, the stable configurations are represented on bridge sites. To confirm the lower barriers obtained on top sites, the  $O_{(ads)} + CO_{(gas)} \rightarrow CO_{2(gas)}$  reaction at the  $B_{H-H}$  site was described as pathway **III** in Figure 6(c). The calculated reaction barrier is about 2.176 eV for the  $CO_2$  formation. This result reveals the O atom is very difficult to desorb from the bridge site, indicating the bonding of O on the bridge site is too strong to react with CO. In Figure 7, the diffusion barrier of O from  $B_{H-H}$  site to the  $T_H$  site is 0.798 eV, which is significantly smaller than the desorption barrier of pathway **III**. This result shows that the O on the  $T_H$  site is more prone to react with CO directly.

According to the adsorption results of O and  $O_2$ , the bonding of O on the bridge site is too strong to react with CO. However, the bonding of O is demonstrated to become weaker if the O coverage increasing. Due to dissociation of  $O_2$  molecule to O-O atoms occurs easily on W metals, the gas  $CO_2$  will be easily produced by the ER mechanism at the high O coverage. Consequently, the calculated barriers of W nanowire in pathway **I** is smaller than that on the W(111) surface in the same reaction path. However, the W(111) surface represents the better catalytic ability, although the difference is very slight between two materials in pathway **II**.

### Analysis of Charge and Electronic State during CO Oxidation

The analyses of charge and electronic state during CO oxidation were conducted by the local density of state (LDOS) onto the orbitals for the adsorbed constructs of  $O_2$  and CO species, as well as for the  $d$  orbital of bound W atom of the W nanowire and W(111) surface, which are represented in Figures 8(a)-(d). According to the overlapped regions, the extent of orbital hybridization can be observed. Figure 8(a1) and (b1) show the LDOS before the interactions between  $O_2$ , CO, and the W nanowire,

respectively. Furthermore, panel (b1) and (b2) show the  $O_2$  and CO molecules interacting with the W nanowire; panel (c1) and (c2) correspond to the LDOS of transition state; panel (d1) and (d2) are the process during  $CO_2$  forms the adsorption configurations on the W helical nanowire. For Figure 8(b1) and (c1), the  $O_2$  on the W nanowire possesses a part of overlap between the  $2\pi^*$  orbital of  $O_2$  with the W  $d$  orbit, indicating the stronger bonding between  $O_2$  and W atom. That is the reason for the  $O_2$  bond length increase while the back-donation from antibonding orbitals  $2\pi^*$  of  $O_2$  to W and the electronic resonance between the antibonding orbitals  $2\pi^*$  of  $O_2$  and the  $d$  state of the W atoms. Moreover, it also can be observed the slightly overlaps between the  $1\pi$  and  $5\sigma$  orbits of  $O_2$  with the W  $d$  orbit, and this result explains some electronic transfer from the W atom to  $O_2$ . Due to the CO bonds with an adsorbed  $O_2$  by physisorption, there is almost no charge transfer from adsorbed  $O_2$  to CO species. Therefore, the LDOS results do not have remarkable differences in the Figure 8(a2) and (b2). In addition, the LDOS results of transition state show that CO reacts to  $O_2$  nearly in the Figure 8(c2). The orbital hybridization results in the charge transfer, leading to the decrease of the peak of  $5\sigma$  orbit. Finally, as shown in Figure 8(d2), the LDOS for the products is similar to the separated  $CO_2$  gas phase and adsorbate O on the W nanowire, indicating the  $CO_2$  molecule is only physisorbed on the W nanowire.

## Conclusions

The CO oxidation process on the W nanowire has been investigated by the DFT calculation. In the CO oxidation pathways,  $CO + O_2 \rightarrow CO_2 + O$ , and  $CO + O \rightarrow CO_2$  were considered as initial configurations. From the results of adsorption calculations, the  $O_2$  end-on adsorption on a top site is the most stable configuration, while the O atom prefers to adsorb on the bridge site. The corresponding adsorption energies are about -2.0 eV and -4.49 eV, respectively. For the W nanowire and the

W(111) surface, the barriers of pathway **I** about 0.70 eV and 0.468 eV are lower than those of pathway **II** about 1.314 eV and 1.529 eV, respectively. Moreover, the reaction barriers for pathway **I** of the ER mechanism on the W nanowire is smaller than that on the W(111) surface, implying the larger catalytic activity of W nanowire for CO oxidation. According to the back-donation theory from antibonding orbitals  $2\pi^*$  of  $O_2$  to W atoms, the bond length changes and adsorption situations have been understood in detail. We hope these DFT simulation results could supply the predictions of structural and electronic properties as well as the perspective viewpoints for the related experiments.

### Acknowledgment

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**Table 1** The results of binding energy using rPBE functional for bulk tungsten materials

		<i>Exp.</i>	<i>DFT</i>
<b>W<sub>FCC</sub></b>	Cohesive Energy (eV)	8.33~8.45 <sup>a</sup>	8.61
<b>W<sub>HCP</sub></b>		8.33 <sup>b</sup>	8.19
<b>W<sub>BCC</sub></b>		8.9 <sup>a</sup>	8.94
	Lattice constant (Å)	3.16 <sup>b</sup> 3.25 <sup>c</sup>	3.16

<sup>a</sup>Reference [25], <sup>b</sup>Reference [26], <sup>c</sup>Reference[27], <sup>d</sup>Reference [32]

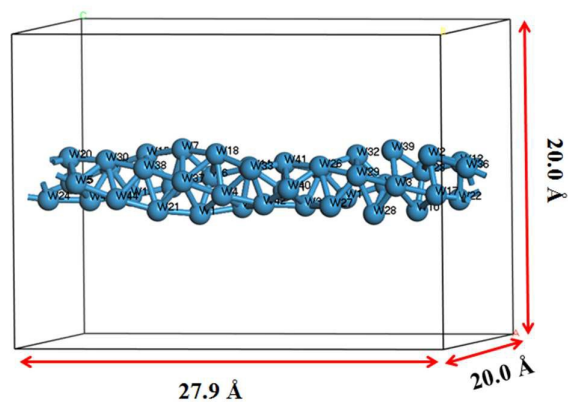
**Table 2** The results of bonding, frequency, and dissociative energy using rPBE functional for the tungsten dimer

	Method	R (Å)	$\omega(\text{cm}^{-1})$	$E_d(\text{eV})$
<b>W<sub>2</sub></b>	rPBE/DNP/DSPP	2.075 <sup>g</sup>	334.2 <sup>g</sup>	4.74 <sup>g</sup>
	Exp.		337 <sup>e</sup>	5 ± 1 <sup>f</sup>

<sup>e</sup>Reference [28], <sup>f</sup>Reference [29], <sup>g</sup>This work

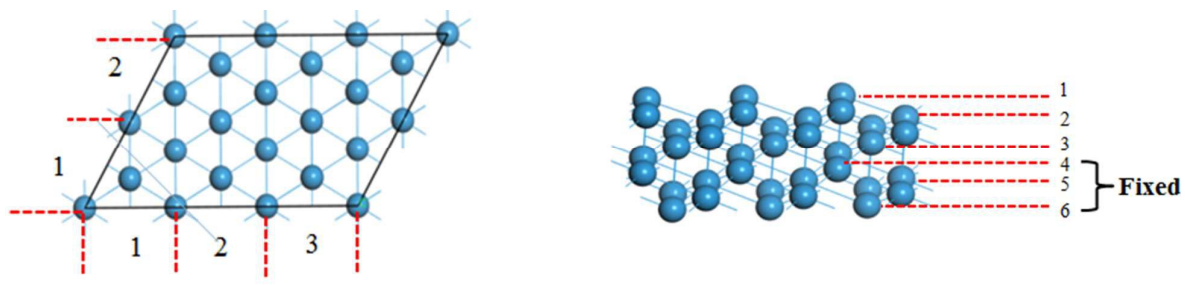
**Table 3** The adsorption energy and bond lengths of O atom and O<sub>2</sub> molecule on the W helical nanowire.

Species	Bond length(Å)	Bond length(Å)	Adsorption
	(O-O)	(W-O)	energy(eV)
O <sub>2</sub> molecule			
T <sub>H</sub>	1.326	1.852	-2.000
T <sub>L</sub>	1.322	1.832	-1.980
B <sub>H-H</sub>	1.372	2.034, 1.958	-2.268
B <sub>H-L1</sub>	1.370	2.197, 2.056	-1.219
B <sub>L-L</sub>	1.392	2.041, 2.103	-1.793
O atom			
T <sub>H</sub>		1.759	-3.853
T <sub>L</sub>		1.748	-3.888
B <sub>H-H</sub>		1.957, 1.992	-4.490
B <sub>H-L1</sub>		2.009, 1.970	-3.194
B <sub>L-L</sub>		1.969,1.891	-3.974



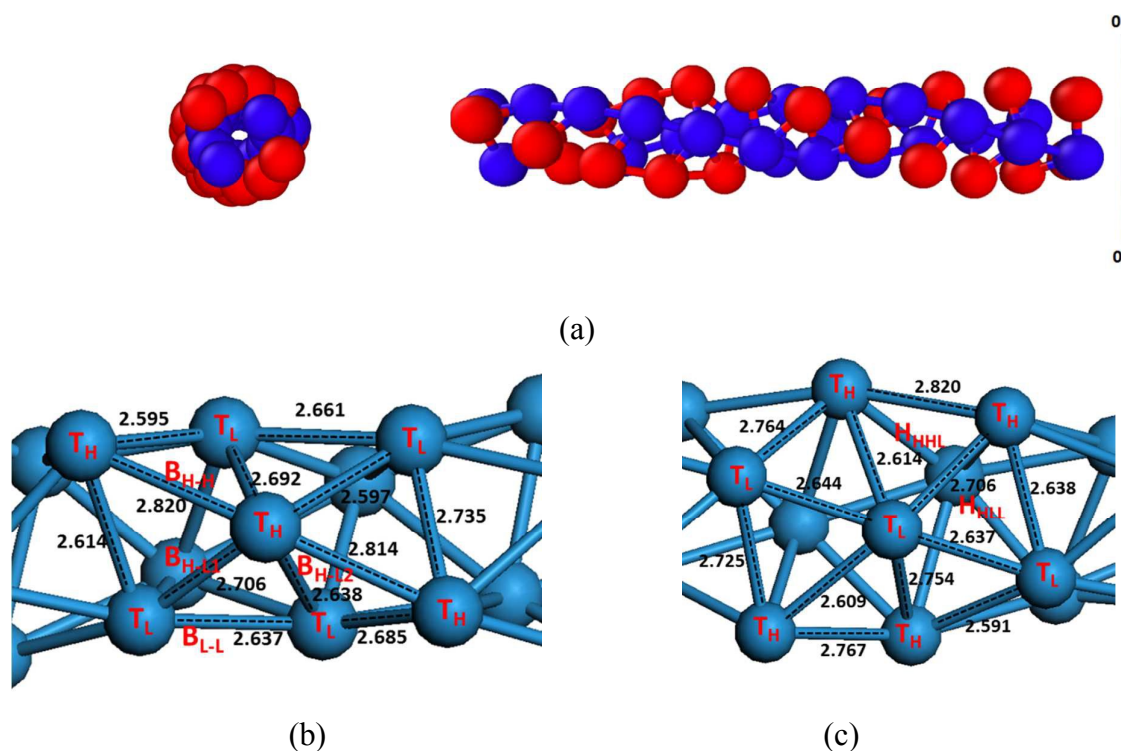
(a)

**Figure 1** (a) Optimized geometry of the W helical nanowire

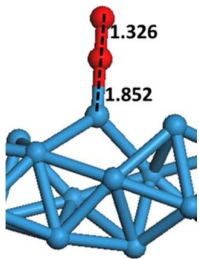


**Figure 2** schematic diagram of  $p(3 \times 2)$  W(111) surface: (left) top view; (right) side view.

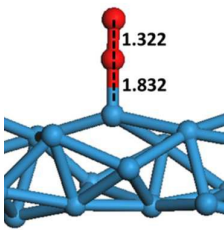




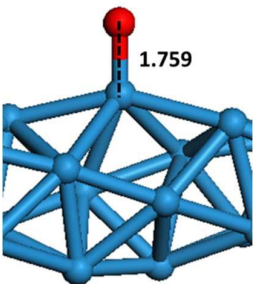
**Figure 3** Fukui function  $F^{k-}$  of the W helical nanowire: (left) cross-section view; (right) side view (b) The bond length of the high activity region. (c) The bond length of the low activity region. The labels T, B and H represent top, bridge, and hollow sites. The subscript of the labels H, L represent high and low activity region. Note that the bridge sites are between every two surface W atoms, and the hollow sites are in the middle sites of three W atoms.



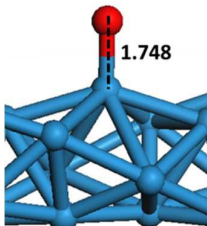
(a) O<sub>2</sub> On-Top (T<sub>H</sub>)



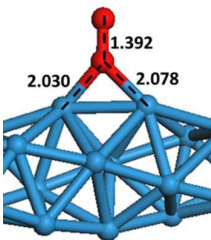
(b) O<sub>2</sub> On-Top (T<sub>L</sub>)



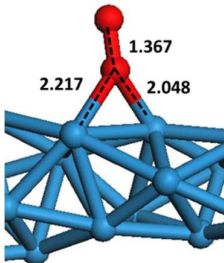
(c) O On-Top (T<sub>H</sub>)



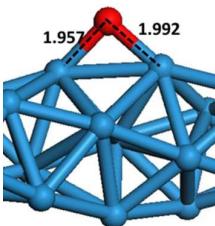
(d) O On-Top (T<sub>L</sub>)



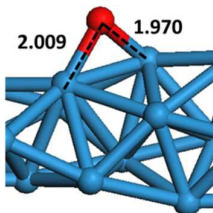
(e) O<sub>2</sub> Bridge (H-H)



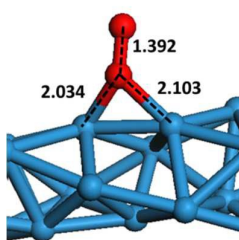
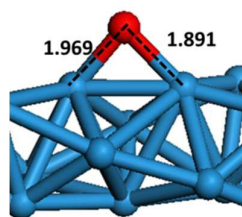
(f) O<sub>2</sub> Bridge (H-L1)



(g) O Bridge (H-H)

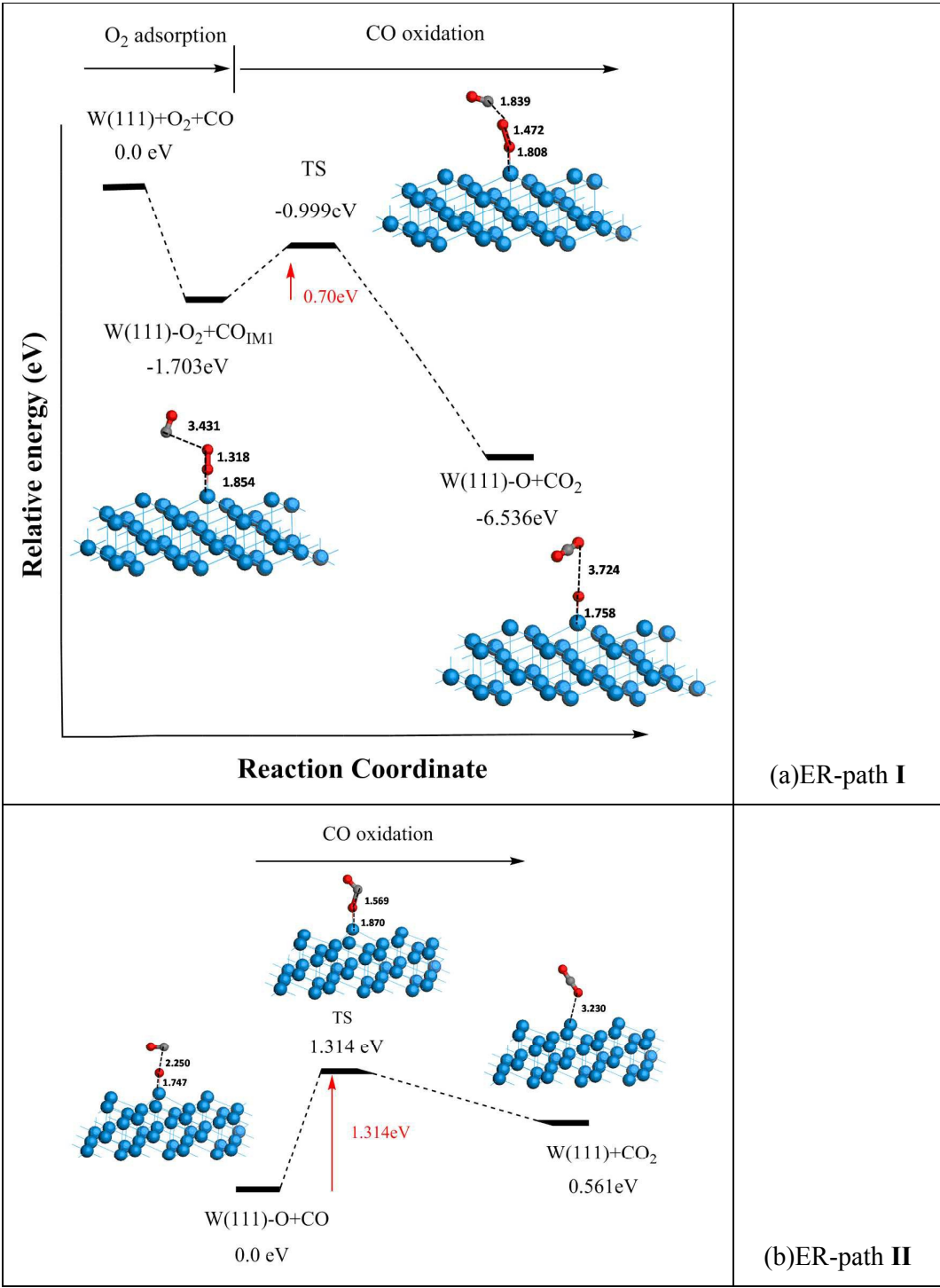


(h) O Bridge (H-L1)

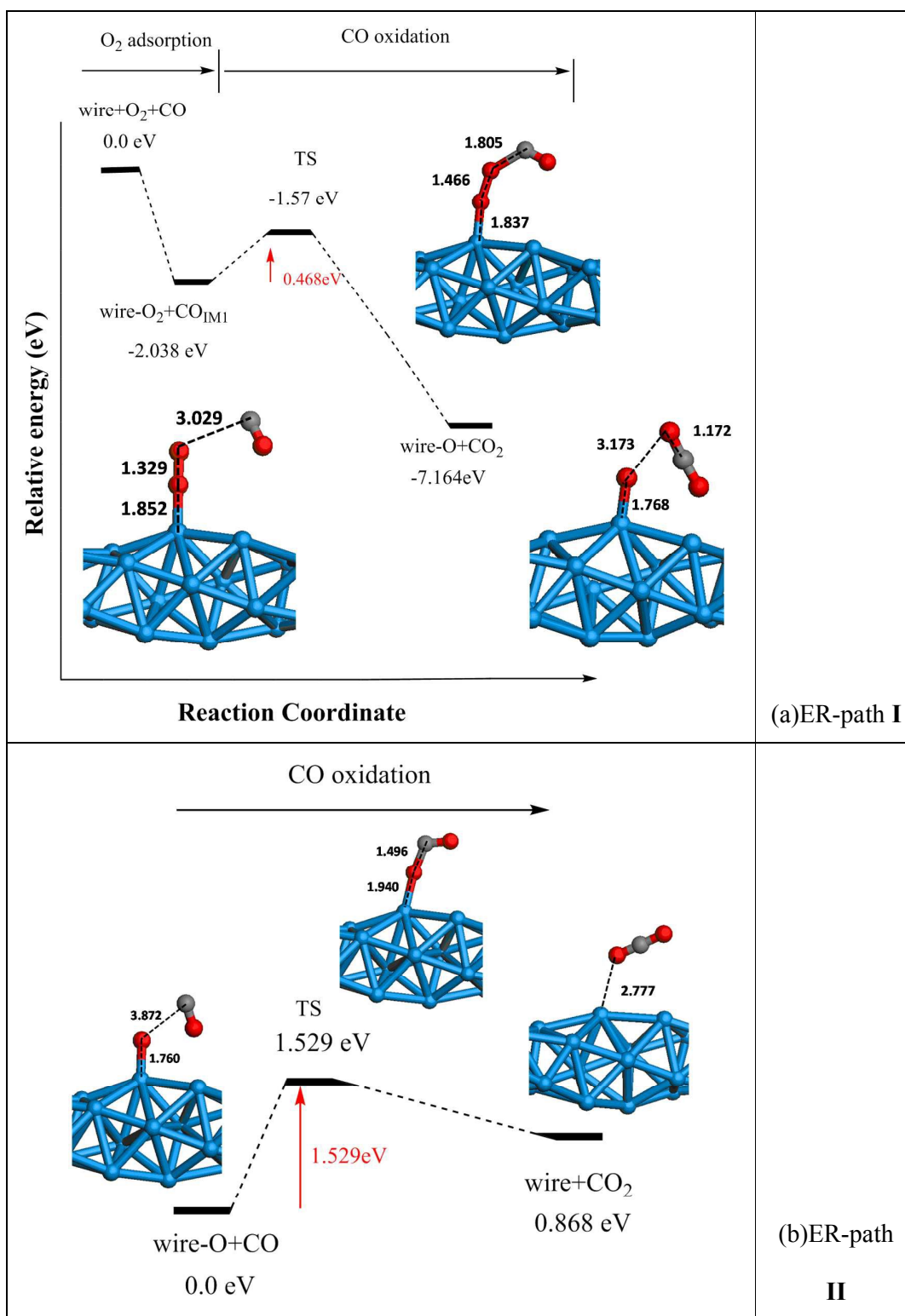
(i) O<sub>2</sub> Bridge (L-L)

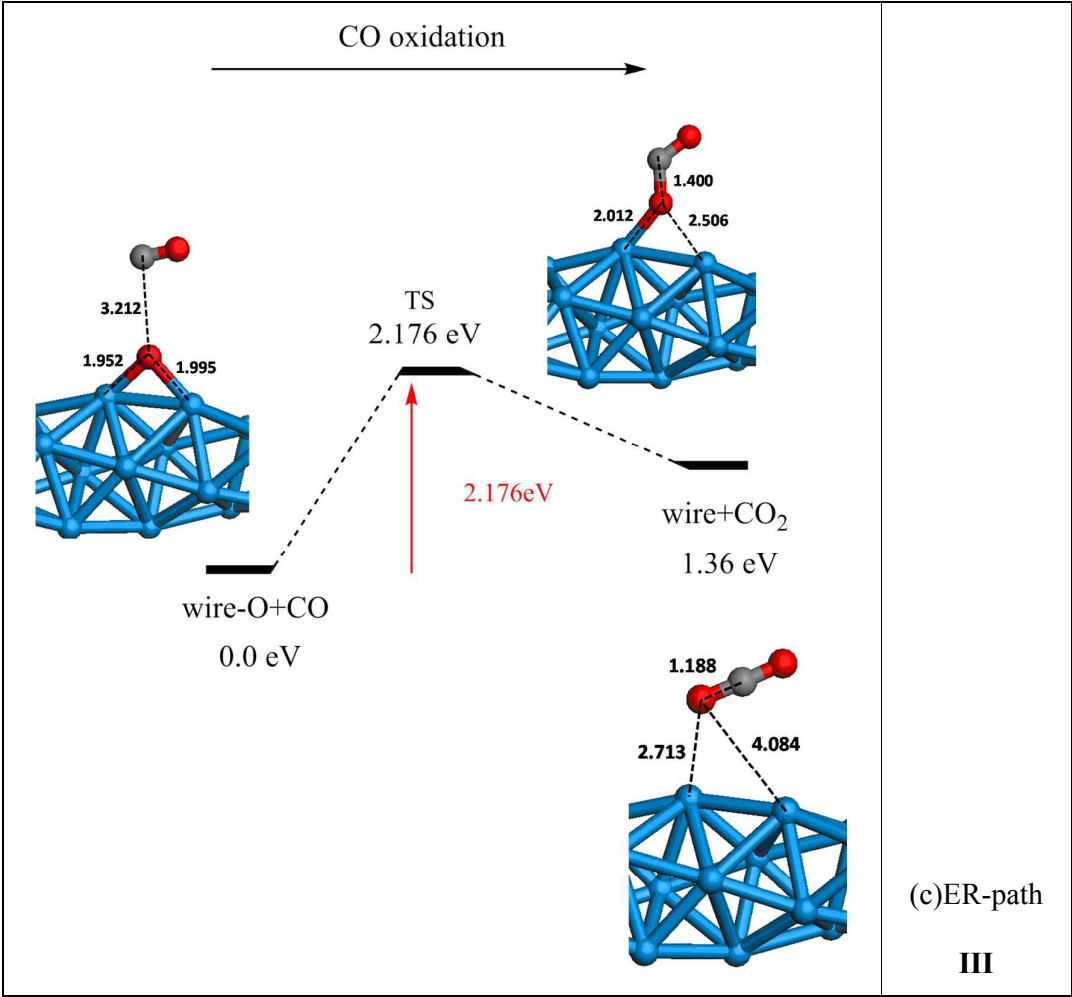
(j) O Bridge (L-L)

**Figure 4** Optimized geometries of O<sub>2</sub> and O adsorption on the W nanowire. The bond lengths are given in angstroms (Å).

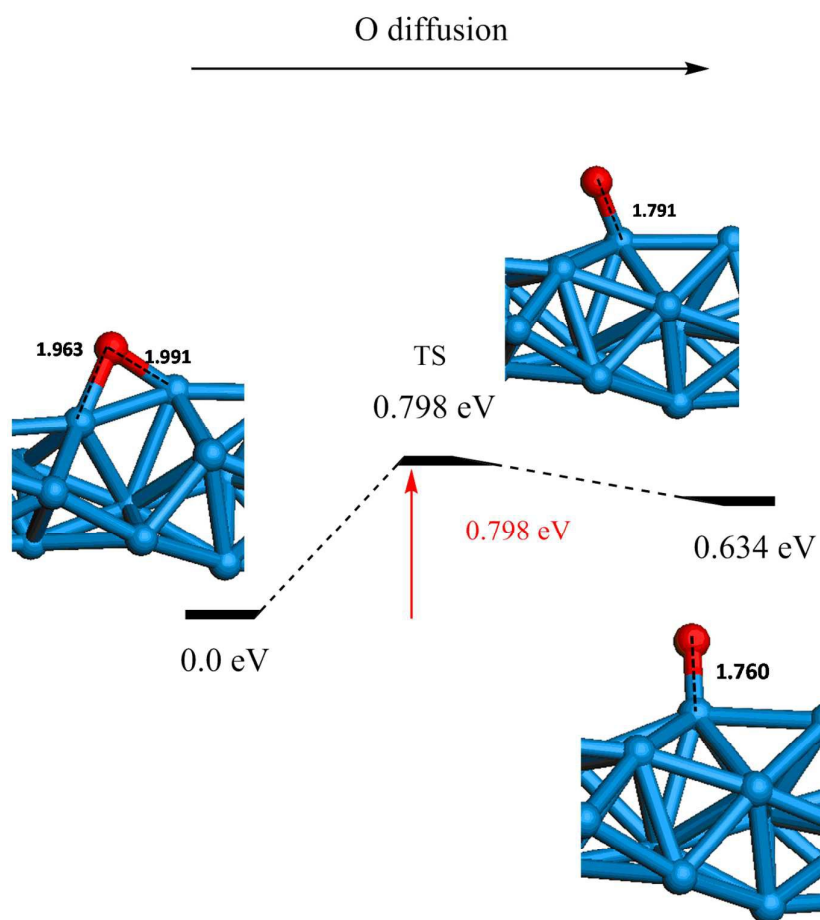


**Figure 5** Schematic potential energy profiles of the ER reaction paths for (a)  $\text{CO} + \text{O}_2$  and (b)  $\text{CO} + \text{O}$  reactions on the W(111) surface.

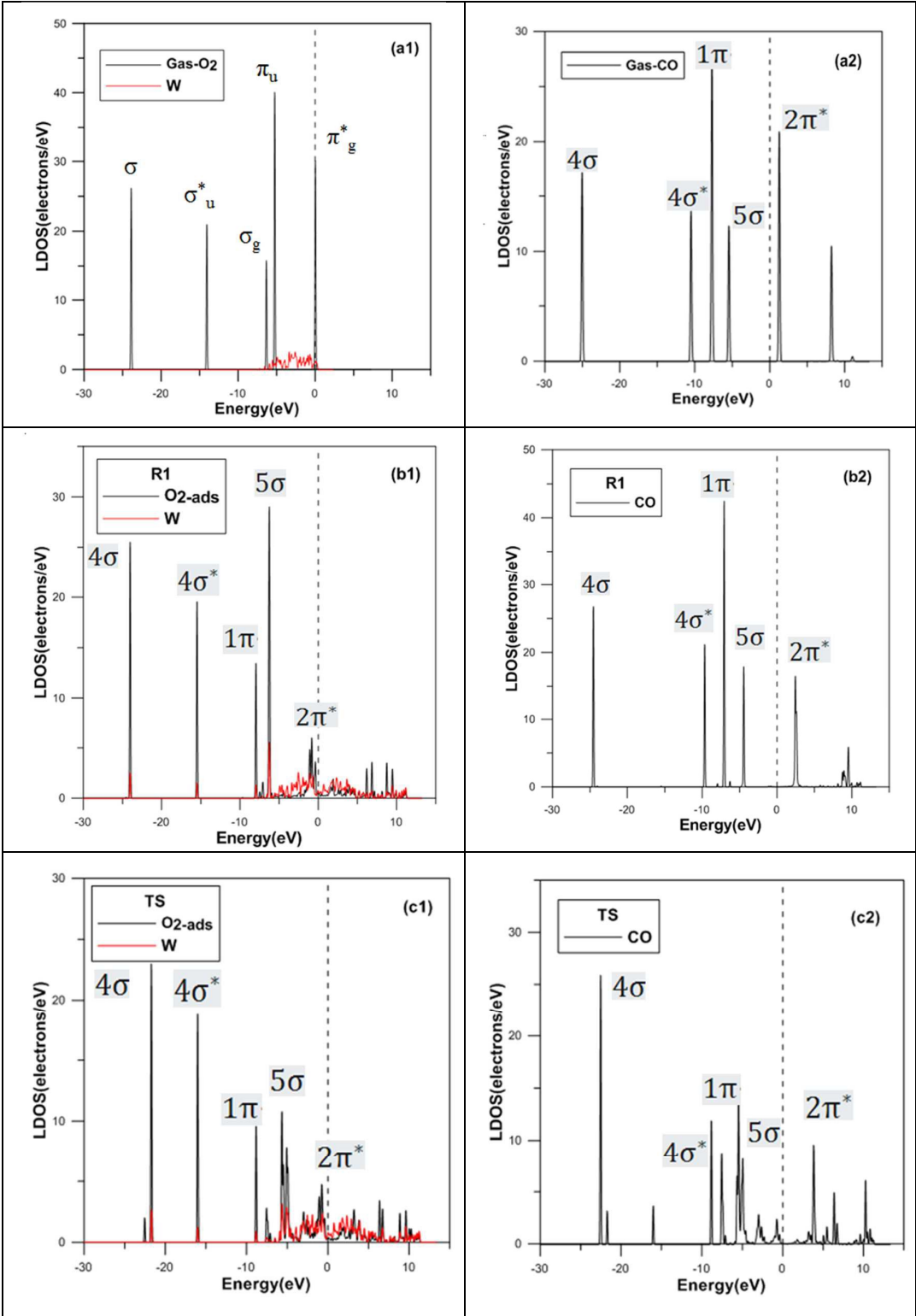




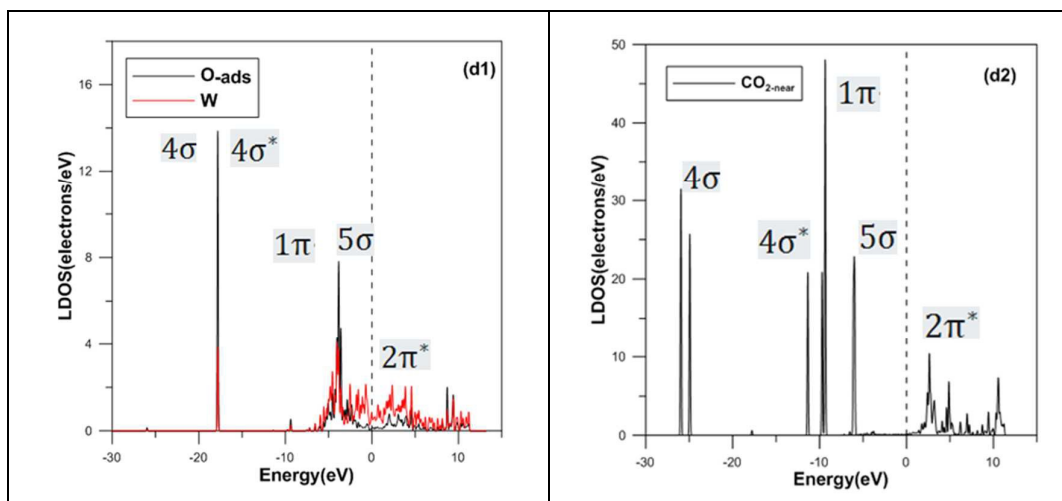
**Figure 6** Schematic potential energy profiles of the ER reaction path for (a) CO + O<sub>2</sub>, (b) CO + O(on top site), and (c) CO + O (on bridge site) reactions on the W nanowire.



**Figure 7** Schematic potential energy profiles of the diffusion reaction from B<sub>H-H</sub> site to the T<sub>H</sub> site on the W nanowire.







**Figure 8** Partial density of states (PDOS) projected onto O–O and C–O for CO oxidation on the W helical nanowire: (a) before interaction; (b)  $W_{\text{wire}}\text{--O}_2$  and CO molecule interacting with the  $W_{\text{wire}}$ ; (c) transition state; (d)  $W_{\text{wire}}\text{--O}$  and  $\text{CO}_2$  molecule interacting with the  $W_{\text{wire}}$ . The dashed line represents the Fermi level.