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L-Aspartate Links for Stable Sodium Metal-Organic Frameworks

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Metal-organic frameworks (MOFs) based purely on sodium are rare, typically due to large numbers of coordinating solvent ligands. We designed a tetratopic aspartate-based linker with flexible carboxylate groups to enhance framework stability. We report two new air-stable sodium MOFs, MOF-705 and MOF-706, comprising 2D sodium oxide sheets.

Metal-organic frameworks (MOFs) are extended porous structures in which multi-metallic secondary building units (SBUs) are joined by organic linkers.¹⁻⁵ Increasingly, MOFs with unusual thermal and chemical stability are being made, and investigated in many applications such as catalysis and gas storage.6-10 MOFs based solely on sodium are few because of the tendency of sodium to coordinate a large ratio of solvent ligands to organic linkers, forming unstable frameworks with discrete or 1D sodium oxide building blocks that are highly sensitive to pore evacuation.¹¹⁻¹⁸ This instability is brought about by the loss of these solvent ligands, leading to structural collapse.¹⁹ In this contribution, we show how the use of multiple carboxylate functionalities of L-aspartate to coordinate sodium gives two MOFs (Figure 1), MOF-705 $[Na_4(BDA)(CH_3OH)(H_2O)]$, and the extended version MOF-706 $[Na_4(BPDA)(H_2O)_2]$, in which the terminal ligands can be removed without collapse of their overall structure. We believe this is possible because of the ligand design of flexible L-aspartate carboxylates that facilitate the unique 2D sodium oxide secondary building units (SBUs) formation (Figure 1). This is supported by previous report using a similar link backbone but lacking the aspartate, which yielded a 1D SBU wherein the aspartate causes the polyhedra to share edges and/or corners to make the extended sheets.¹² These compounds represent the first examples of porous sodium

MOFs which are air stable and retain their structure upon removal of guests even after a month at ambient conditions. The structures of both MOFs were determined by single crystal X-ray diffraction (SXRD) analysis, and further characterized \sim elemental analysis, NMR, and powder X-ray diffraction (PXRD). The gas adsorptive properties of these compounds indicate permanent porosity and their ability to uptake nitrogen an carbon dioxide.

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The designed linkers H4BDA (**1**) and H4BPDA (**2**) (Figure 1a) were synthesized by coupling 1,4-benzenedicarboxylic acid $(H₂BDC)$ or $(1,1'-biphenyl)-4,4'-dicarboxylic$ acid $(H₂BPDC)$ respectively, to the α -amine of L-aspartic acid (see Electron ϵ Supporting Information (ESI), Section S1, for the detailed synthesis). MOF-705 was synthesized by the addition of H₄BD $\sqrt{ }$ to a methanolic solution of sodium hydroxide and heating to 50 °C while MOF-706 with H₄BPDA required heating to 70 \degree (Section S2). The syntheses afforded needle-shaped colorless single crystals, with the dimensions of $0.01 \times 0.01 \times 0.05$ mm³ for MOF-705 and 0.005 \times 0.005 \times 0.03 mm³ for MOF-706.

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In addition to single crystals, microcrystalline powder samples of these MOFs were obtained for higher yields by simply heating the reaction mixture to 85 °C, where the PXRD pattern of the microcrystalline product matched that simulated from the single crystal structure.

MOF-705 crystallizes in the chiral monoclinic P2₁ space group bearing the infinite 2D sodium oxide sheets extended in the [100] and [010] directions (Figure 1a and Table 1), with a repeat unit of four edge-sharing sodium atoms (Figure 1b). The first of these has square pyramidal geometry ($Na_{(1)}$), while the two in the middle are distorted trigonal bipyramids ($Na₍₂₎$ and $Na_{(3)}$, and the last is a distorted octahedron ($Na_{(4)}$). All the coordinating moieties are structural (an integral part of the MOF backbone) except for the octahedral Na $_{(4)}$, which is completed by the coordination of one methanol molecule. The Na₍₁₎ centers are bridged by two μ^2 water molecules in the [100] direction (Figure 1b), while μ^3 carboxylates connect all sodium atoms in the [010] direction. Completing the 3D structure, the linker joins these sodium oxide sheets in the [001] direction. The same applies for MOF-706, except that water molecules replace methanol coordinated to $Na_{(4)}$.

MOF-705 was immersed in a variety of organic solvents to explore the most suitable activation conditions (Figure S10). However, conventional methods to remove solvent from the pores were not successful causing both MOFs to collapse and form amorphous solids, so we turned to the use of supercritical $CO₂$ (SC-CO₂) activation from methanol. This yielded a crystalline material that was air stable for over a month in ambient conditions, as proven by PXRD and scanning electron microscopy (SEM, Figures 2 and S7). \ SXRD that methanol is still coordinated after but can be removed to widen the pores by simp

Figure 2. Observed reflection (purple) and PXRD patterns of MOF-705. Simulated from SXRD (red), as-synthesized (blue), SC-CO₂ activated (orange), heat activated (green).

[e \AA^{-3}]

^a For the data of SC-CO₂ activated MOF-705 and MOF-706 please refer to the ESI, section S4.

SC-CO₂ activated sample at 70 $^{\circ}$ C under vacuum for 6 h. This evacuation of methanol guests was validated by thermogravimetric analysis (TGA) which showed no weight loss below 350 °C (Table S1 and Figure S12). Maximal surface area for MOF-706 can be achieved using $SC\text{-}CO₂$ from acetone, with no increase in surface area upon heating, unlike MOF-70, (Figure S11).

The pores of $SC-CO₂$ activated MOF-705 were on accessible to $CO₂$ gas at 298 K (52 cm³ cm⁻³) while inaccessibl for N_2 at 77 K (Figures S14 and S16). However the hea activated sample (following loss of the methanol) does uptake N_2 gas at 77 K, with a BET (Langmuir)surface area of 132 (231) m^2 g⁻¹ (Figure S10) and with a steeper slope in the low. pressure region of the $CO₂$ isotherm, indicating a strong ga framework interaction (Figure S16). On the other hand despite the larger unit cell of MOF-706, the activated sample has a similar BET (Langmuir) surface area of 126 (216) m^2 g^{-1} and lower $CO₂$ uptake, probably due to guest molecules st. coordinated to $Na_{(4)}$ (Figures S15 and S18).

The CO₂ uptake and slit-like pores of both MOFs led us to

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*R*_{int} 0.0889 0.1134 θ range (degrees) 2.55 - 74.52 2.164 - 21.724

] 0.366 / -0.241 0.535 / -0.457

Sample^a MOF-705 as MOF-706 as

Table 1. Crystallographic data of as-synthesized MOF-705 and MOF-706.

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Figure 3. CO₂ and N₂ isotherms for MOF-705. N₂ isotherms are in orange (273 K) and green (298 K) while the $CO₂$ isotherms are in blue (273 K) and red (298 K). Filled and open symbols represent adsorption and desorption branches respectively. The connecting curves are guides for the eye.

test their selectivity towards $CO₂$ over N₂, with the objective of potentially employing these materials in $CO₂$ capture from post-combustion flue gas.20-23 Indeed, we found good selectivity and affinity of the two MOFs for the $CO₂$ over N₂ at both 273 and 298 K (Figures 3, S18 and Table 2). The selectivity, of the MOFs which was assessed based on the uptake ratio of $CO₂$ over N₂ at different pressures, is high in comparison with other leading MOFs in this regard e.g. SIFSIX-2-Cu (13.7), ZIF-300 (22), Ni-MOF-74 (>10) and IRMOF-74-III (35) , 17 , $^{23-25}$ and is enhanced at the lower pressure region to reach 200 and 40 times, for MOF-705 and MOF-706 respectively, at 298 K and 80 Torr (Table 2).

In order for a material to be a successful candidate for an industrial application of gas separation, dynamic separation capacity of $CO₂$ over N₂ is desirable because $CO₂$ isotherms only show the thermodynamic capacity.^{21-22, 27-28} To this end, breakthrough experiments were performed. A typical experiment involves passing a mixture of gases over the tested material to evaluate the selective capture of one gas, in this case $CO₂$, relative to the injection time compared to the other gases, specifically N_2 . Performance is evaluated by the retention time; the longer the time needed for $CO₂$ to breakthrough (retention time), the higher the uptake and the better the separation capacity. Both MOFs were subjected to a binary gas mixture comprising 16% CO₂ and 84% N₂ (v/v). The breakthrough experiments (Figures 4, S19 and S20) showed that $CO₂$ breakthrough was delayed and that the MOFs successfully captured $CO₂$ in the presence of N₂. Moreover, the

capture of CO₂ was efficient as more than 98% of the inlet CC \pm was captured (i.e. delayed breakthrough time), while N_2 gas passed through without interacting with the framework Based on the breakthrough time, the uptake of MOF-705 was greater than MOF-706, 53 compared to 29 cm³ c \sim respectively, which is in agreement with the capacity displayed by $CO₂$ sorption measurements of 58 vs. 35 cm³ cm⁻³. Finally, the regeneration of any material for $CO₂$ separation is an indispensable property, and both MOFs showed that they ca be purged of $CO₂$ simply by flowing pure N₂ at ambient temperature. Within five minutes, 99.5% of adsorbed $CO₂$ was expelled from the framework. This method of desorption wa effective over at least three cycles (Figure 4), confirming $t^{\frac{1}{2}}$ very promising properties of this material for flue gas separation.

In summary, we present two new sodium-based MOFs, MOF-705 and MOF-706, that comprise a stable 2D SBU of sodium oxide sheets. The design of the linkers proved to be crucial for the formation and stabilization of these MOFs \mathbf{b} taking advantage of the flexible tetratopic aspartate ends. This is evidenced by previous reports of sodium MOFs with similar but rigid carboxylate linkers resulting in discrete and 1D rod SBUs.¹¹⁻¹⁹ The combination of an air stable SBU and compact nature of the pore makes them strong candidates for C_2 separation applications.

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CCDC 1420344-1420347 contain the supplementary crystallographic data for this paper. These data can be obtained free of charge from The Cambridge Crystallographic Data Centre.

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