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A vanadium(IV) pyrazolate metal organic polyhedron with permanent porosity and adsorption selectivity

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A vanadium(IV) pyrazolate-based open metal organic polyhedron of $[V_3(\mu_3-O)O(OH)_2(\mu_4-BPD)_{1.5}(\mu-HCOO)_3]$ (BDP = benzene-1,4-bipyrazolate) formulation gives rise to a porous crystal structure exhibiting micro and mesoporosity which is useful for selective adsorption of CO_2 over N_2 and benzene over cyclohexane.

Metal-organic polyhedra (MOPs) are discrete molecular architectures constructed via the coordination of metal ions and organic linkers to yield highly structured polyhedra. These systems usually possess a cavity able to incorporate host molecules thereby modifying their reactivity and dynamics.¹ These features put these system in close resemblance to biological self-assembled container supramolecules such as ferritin which play fundamental roles in regulating major physiological functions.² MOPs are also interesting since they can be considered as the building units of MOFs³, however, their discrete nature confer them a highly processable nature.⁴⁻⁶ On the other hand, vanadium ions are very attractive building units since its coordination diversity and redox activity gives rise to a rich variety of predefined SBUs and catalytic activity.⁷⁻¹⁷ Noteworthy, the hard acidic nature of vanadium gives rise to the preferred use of O-donor linkers (i.e. phosphonate and carboxylates) for the construction of MOPs and MOFs.^{18,19-24} However, the enhanced basicity of some nitrogen-donor pyrazolate ligands has proven to give rise to highly stable constructs with $Ni_3(BTP)_2$ (BTP = 1,3,5-tris(1H-pyrazol-4-yl)benzene) being a remarkable example of high chemical and thermal stability making them potentially useful for application in harsh experimental conditions.²⁵⁻³¹

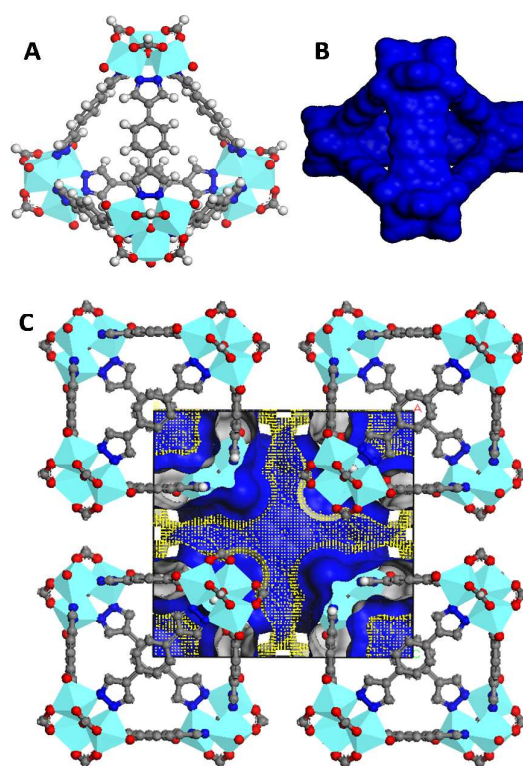


Fig. 1. View of the molecular structure of tetrahedral metal organic polyhedron found in the crystal structure of **1** in the form of polyhedra (A), van der Waals surface (B) and Connolly surface of the crystal packing (C).

In this communication, we show that the reaction of VCl_3 and H_2BDP (1,4-bi(1H-pyrazol-4-yl)benzene), in DMF under solvothermal conditions at 423 K, gives rise to single crystals of a system of $[V_3(\mu_3-O)O(OH)_2(\mu_4-BPD)_{1.5}(\mu-HCOO)_3]$ (**1**) formulation. The formation of formate anions should be regarded as a consequence of the hydrolysis of DMF solvent molecules during the solvothermal reaction. The crystallographic study of **1** reveals that its molecular structure

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is based on triangular trinuclear vanadium(IV) SBUs of $[V_3(\mu^3-O)O(OH)_2(\mu-HCOO)_3]$ type in which the vanadium centres exhibit a typically distorted octahedral VN_2O_4 stereochemistry.¹⁴ These triangular SBUs are connected by exotetradentate benzene-1,4-bipyrazolate (BDP) to give rise to the formation of tetrahedral MOPs with 1.2 nm inner cavities accessible through 0.85 nm wide triangular windows (see Fig. 1).[†] Noteworthy, MOPs packing in the crystal structure follows a *bcc* arrangement which gives rise to 0.85 nm inter nanocage voids. Calculation of the Connolly surface of the crystal structure gives rise to high values of potentially accessible surface area for a probe molecule of 1.4 Å, namely $1660 \text{ m}^2 \text{ g}^{-1}$ and accessible pore volume of $0.130 \text{ cm}^3 \text{ cm}^{-3}$ (Fig. 1C). In view of both the intra and inter cage porosity, we thought that this material might be useful for gas separation and purification purposes (see below). The physico-chemical nature of **1** was further studied by means of IR, EPR, TGA and XRPD (see ESI). Both room temperature EPR spectra of **1**, exhibiting a sharp isotropic signal with *g* value of 1.996, and IR spectra showing the apparition of band at 953 cm^{-1} characteristic of $V=O$ agree with the observed octahedral environment of vanadium(IV) centres. X-ray powder diffraction (XRPD) patterns of **1** are indicative of the phase purity of the bulk material. The thermogravimetric analysis of **1** in an atmosphere of air, (Fig. S4), reveals a significant weight loss (34 %) of the material in the 373 - 503 K temperature range which corresponds to the loss of water and DMF solvent molecules. Above this temperature the desolvated material remains thermally stable up to 568 K. The permanent porosity of **1** was verified by N_2 adsorption isotherm at 77 K (Fig. 2), which reveals a type-I isotherm with a moderate value of BET surface area of $260 \text{ m}^2 \text{ g}^{-1}$ and an adsorption capacity of 4.6 mmol g^{-1} at 0.8 P/P_0 . Additionally, the presence of a (type H4) hysteresis is indicative of textural mesoporosity arising from interparticle mesopores. Indeed, porosity distribution by density functional theory indicates, as implemented in Micromeritics software, is indicative of the presence of both micropores of ca. 1.5 nm (inside cages) and mesopores of 5 nm (interparticle).

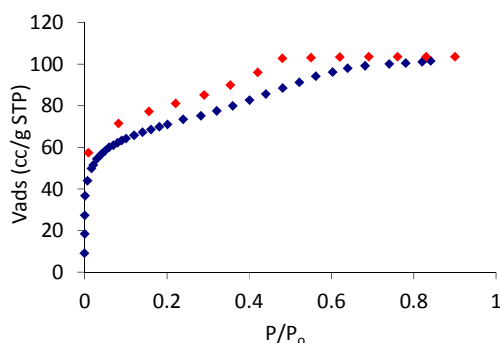


Fig. 2. N_2 adsorption (blue symbols) and desorption (red symbols) measured at 77 K.

The formation of mesoporosity, which is not observed in the crystal structure of **1** might be responsible for the observed higher pore volume accessibility than the calculated one from crystal structure (experimental accessible volume of $0.150 \text{ cm}^3 \text{ cm}^{-3}$ vs the calculated $0.130 \text{ cm}^3 \text{ cm}^{-3}$), should be related to a partial loss of supramolecular ordering after sample desolvation. By contrast, the observed BET surface is considerably lower than the calculated one from the Connolly Surface for a probe molecule of 1.4 Å ($260 \text{ m}^2 \text{ g}^{-1}$ vs $1660 \text{ m}^2 \text{ g}^{-1}$) which might be indicative of poor accessibility of the probe molecule to the whole crystal porosity after sample activation (see below).

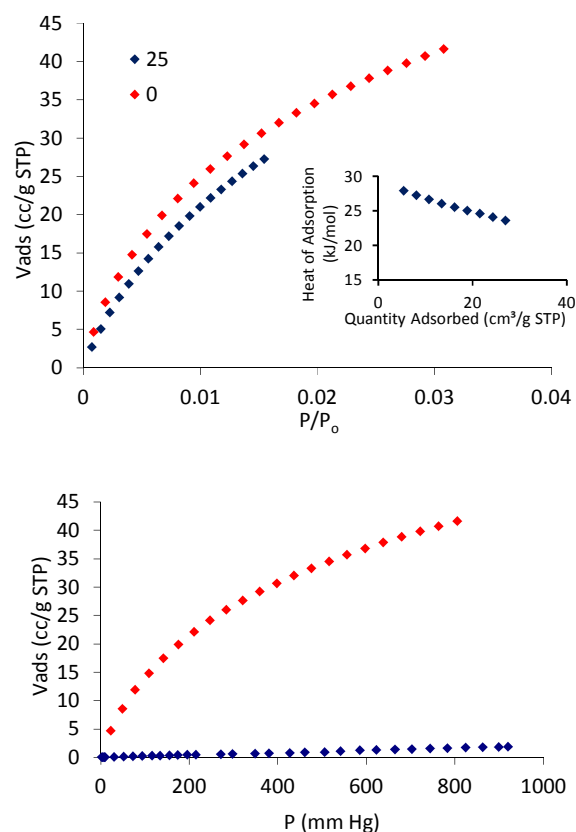


Fig. 3. CO_2 adsorption isotherms measured at 273 K (red) and 298 K (blue) (top), CO_2 isosteric heat of adsorption (inset in top) and CO_2/N_2 selectivity at 273 K (down).

Indeed, XRPD of **1** after thermal activation is indicative that although the original XRPD pattern is maintained a certain degree of amorphization is obvious which may give rise to the observed hierarchical pore structure and to the observed slow diffusion of N_2 molecules inside the pore structure which is manifested by long equilibration times during the adsorption measurements.

Once we proved the accessible nature of the porous structure of **1** we proceed to evaluate its possible utility in gas and vapor separation processes, namely CO_2 capture from flue gas and

benzene/cyclohexane separation. Fig. 3 gather the CO₂ adsorption isotherms of **1** at 273 and 298 K as well as a comparative of N₂ and CO₂ adsorption isotherms at 273 K. The CO₂ adsorption isotherms of **1** follow a type-I behavior in the studied pressure range. The isosteric heats of CO₂ adsorption were evaluated according to the Clausius-Clapeyron (equation 1) from the adsorption isotherms measured at 273 and 298 K:

$$q_{st} = -R \left(\frac{d \ln P}{d \left(\frac{1}{T} \right)} \right)_N \quad (1)$$

where q_{st} , R , P , and N represent isosteric heat of adsorption, gas constant, pressure, and amount of adsorption of CO₂, respectively. The results show that CO₂ molecules interact with the porous structure of **1**. Indeed, the adsorption heat values in the -30 to -25 kJ mol⁻¹ range (Fig. 3b) are indicative of a moderate interaction of the guest CO₂ molecules with the inner voids of the MOP cages since there is no strong interacting sites available and consequently the interaction is probably driven by the size of both intra and inter cage microporosity. In order to prove the adsorption selectivity of this system, we proceeded to estimate the N₂/CO₂ selectivity by measuring single component isotherms of both gases at 273 K (Fig. 3c) which shows that selectivity α_{CO_2/N_2} values vary from 60 in the low pressure range to 25 at 1 bar.

In order to prove that the adsorption selectivity of **1** towards CO₂ is maintained towards gas mixtures we have prepared a chromatographic column of this material (0.35 g) packed in a 15 cm long glass column with a 5 mm inner diameter and tested the separation process towards a simulated flue gas (86 % of N₂ and 14 % CO₂). For this purpose, we proceeded to activate the packed chromatographic column under an He flow (10 mL min⁻¹) at 403 K and after cooling down to 273 K we flowed a gas mixture containing 1.4 mL min⁻¹ of CO₂ and 8.6 mL min⁻¹ of N₂. The results show (Fig. 4) that CO₂ is selectively retained during 160 s which corresponds to adsorption capacity 0.5 mmol of CO₂ g⁻¹ while N₂ does not interact with the adsorbent.

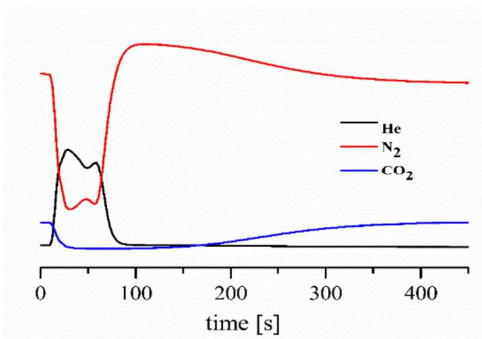


Fig. 4. N₂/CO₂ separation breakthrough experiments at 273 K of a column packed MOP **1**.

In order to further prove the adsorption selective properties of this system we have also essayed the possible suitability of this material for the shape selective separation of benzene (b.p. 80.1 °C) and cyclohexane (b.p. 80.7 °C) since separation by distillation is not suitable since the bp of both compounds differ in less than 1 K.³¹ With this aim we have carried out variable-temperature pulse gas chromatography experiments in the 383–423 K temperature range with a complex gas mixture (methane, cyclohexene and benzene) in order to find out the possible resolution of this gas mixture by **1** (Fig. 5). The results show that **1** give rise to negligible interactions with methane in the studied temperature range which is useful as a reference.

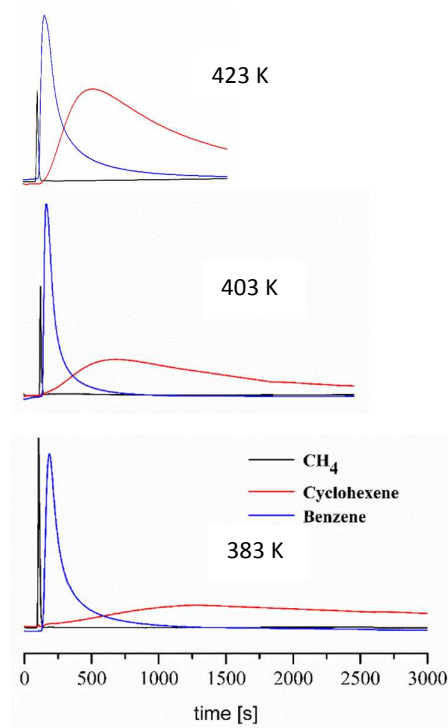


Fig. 5. Variable-temperature pulse gas chromatography experiments of an equimolecular CH₄/benzene/cyclohexene gas mixture passed through a chromatographic column packed with vanadium nanocages using a He flow of 10 mL min⁻¹.

Noteworthy, while the benzene interaction seems to be strong as manifested by long retention times, broad peaks and high adsorption heat value (-28.3 kJmol⁻¹), the interaction with cyclohexene appears to be much weaker as manifested by lower retention times, sharper chromatographic peaks and low adsorption heat value (-2.0 kJmol⁻¹). Indeed, $\alpha_{benzene/cyclohexene}$ selectivity values of 10 are found which should be attributed to the smaller size of benzene compared to cyclohexane which may facilitate the diffusion of benzene inside the MOP cavities justifying its higher retention while cyclohexane molecules are probably interacting with the external surface of the MOP microcrystalline powder.

In summary, a new vanadium MOP has been synthesized and fully characterized. It should be noted that in spite of the discrete nature of 1 MOP, the crystallographic study as well as the static and dynamic adsorption experiments reveal the accessibility of both the intra and intercage space to gas and VOC molecules giving rise to selective adsorption processes which are useful for carbon capture applications as well as shape separation of challenging mixtures benzene/cyclohexane.

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Notes and references

† Crystal Data for 1: $C_{84}H_{68}N_{24}O_{40}V_{12}$, $M = 2664.87 \text{ g mol}^{-1}$, cubic, space group $I-43m$, $a = b = c = 20.084(5) \text{ \AA}$, $V = 8101(3) \text{ \AA}^3$, $Z = 2$, $\rho_{\text{calc}} = 1.084 \text{ mg m}^{-3}$, $T = 100 \text{ K}$, $\lambda_{\text{MoK}\alpha} = 0.71073 \text{ \AA}$, $R(F, F^2 > 2\sigma) = 0.0683$, $R_w(F^2, \text{ all data}) = 0.1648$ for 497 unique reflections, goodness-of-fit = 1.250. CCDC 1412183.

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