Journal of Materials Chemistry B

Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/materialsB

One-step synthesis of biocompatible magnetite/silk fibroin core shell nanoparticles

Weiqin Sheng, ‡^a Jing Liu, ‡^b Shanshan Liu,^a Qiang Lu, ^{*a} David L Kaplan,^{a c} Hesun Zhu^d

^aNational Engineering Laboratory for Modern Silk & Collaborative Innovation Center of Suzhou Nano Science and Technology,

Soochow University, Suzhou 215123, People's Republic of China. Tel: (+86)-512-67061649; E-mail: Lvqiang78@suda.edu.cn

^bRegenerative Medicine Center, First Affiliated Hospital of Dalian Medical University, No.222 Zhongshan Road, Dalian

116011, People's Republic of China

^cDepartment of Biomedical Engineering, Tufts University, Medford, MA 02155, USA

^dResearch Center of Materials Science, Beijing Institute of Technology, Beijing, 100081, People's Republic of China

‡ These authors contributed equally to this work.



Core-shell Fe_3O_4/SF nanoparticles, prepared by silk fibroin with one step, could be widely used in biomedical areas, such as contrast agent and targets with some surface modification.

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxxx

ARTICLE TYPE

One-step synthesis of biocompatible magnetite/silk fibroin core-shell nanoparticles

Weiqin Sheng, ‡^{*a*} Jing Liu, ‡^{*b*} Shanshan Liu^{*a*}, Qiang Lu, * ^{*a*} David L Kaplan, ^{*a c*} and Hesun Zhu^{*d*}

s Received (in XXX, XXX) Xth XXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX DOI: 10.1039/b000000x

A one-step hydrothermal process with silk fibroin (SF) nanofiber as template and coating was developed to synthesize core-shell magnetite/SF nanoparticles with limited controllable sizes. The Fe_3O_4 nanoparticles gradually aggregated into nanospheres with increased sizes from 120 to 500 nm by increasing the SF content in the reaction system. The magnetic properties and biocompatibility of Fe_3O_4 /SF nanoparticles, as well as their functional ability with antibody were also discussed to assess their possible applications in MRI and bio-separation. Compared to previous two-step process, our one-step method provides a simpler and more cost-effective approach to preparing biocompatible core-shell magnetite nanoparticles.

Keywords: Fe₃O₄, silk, core-shell nanoparticles, biocompatibility, Magnetic resonance imaging (MRI)

15

Introduction

There is an urgent requirement to develop multifunctional nanocomposites that could combine sensing, diagnostic and therapeutic functions within a single nanostructure.¹⁻¹⁰ Among 20 these nanocomposites. superparamagnetic iron oxide nanoparticles with biocompatible shells are in great demand because of their low cost and environmentally benign nature, as well as their promising potential in drug delivery,¹¹⁻¹⁵ magnetic resonance imaging (MRI),¹⁶⁻²⁰ magnetic hyperthermia,²¹⁻²³ tissue 25 repair,²⁴⁻²⁶ and bio-separation.^{9, 27} In these applications, good dispersibility without scarify of biocompatibility is a prerequisite. Much effort has been devoted to the fabrication of core-shell superparamagnetic nanoparticles with uniform size and good dispersibility in aqueous solution.^{1, 2} Magnetic microspheres with 30 tailored functional polymer shell have attracted special attention.

Both synthetic and natural polymers, such as PEG, ^{12, 28} chitosan, ¹⁹ alginate, ^{29, 30} dextran, ³¹ poly (vinylpyrrolidone) (PVP) ³² or poly (vinyl alcohol) (PVA), ³³ have been used to stabilize Fe₃O₄ microspheres, offering long-term stability and biocompatibility. ³⁵ Different natural polymers are considered as preferred choice because of their better biocompatibility, nontoxicity and easy conjugation with various functional groups. Several synthetic or coating strategies have been developed to fabricate Fe₃O₄ microspheres with natural polymers as the shell. However, ⁴⁰ tedious preparation steps remain main challenges of their wide applications. Simple and effective processes are urgently needed for core-shell magnetic microsphere formation.

B mori SF, a naturally derived polymer, has been utilized in various technological fields including tissue regeneration, ³⁶ drug ⁴⁵ release, ^{37, 38} optical components and electronic applications ³⁹⁻⁴². Its nanostructures and hydrophobic/hydrophilic properties can be

Journal of Materials Chemistry B Accepted Manuscript

regulated through controlling its self-assembly process, providing promising templates for different functional nanoparticles such as copper oxide, ⁴³ silver ⁴⁴ and others. ^{35, 45} Several cost-effective methods have been developed to prepare silk nanoparticles in 5 aqueous solution, which further offered different templates for these applications.^{19,30,34,35} Although a few studies have reported the fabrication of SF-coated magnetic nanoparticles,^{35, 46} the success in achieving satisfactory water-dispersibility for these nanoparticles has been limited. Huang group⁴⁶ fabricated 10 Fe₃O₄/SF nanoparticles with irregular structures. However, further improvement is still required to control the morphology for biomedical applications such as drug release. Chen group³⁵ achieved the morphology control using SF as template. Unfortunately, they could only prepare Fe₂O₃/SF nanoparticles 15 whose magnetism is not strong enough for bio-separation and MRI applications. In our study, more SF was added in the Fe₃O₄ reaction system to achieve Fe₃O₄/SF nanoparticles with regular morphology and stronger magnetism, providing better candidate for bio-separation, MRI and drug release applications.

- ²⁰ Overall, we developed a one-step solvothermal process to synthesize core-shell Fe₃O₄/SF nanomaterial with homogeneous size and water-dispersibility. Our results suggested that SF could be used to control the morphology and size of Fe₃O₄ microspheres. The biocompatibility and transverse relaxivity ²⁵ (typical characterization of the MRI efficiency) of the Fe₃O₄/SF
- core-shell nanomaterial, as well as their easily coupling ability with antibody, suggest their future applications in different biomedical applications.

30 Experimental Section

Materials

Iron chloride hexahydrate (FeCl₃·6H₂O), Sodium acetate

trihydrate (NaAc·3H₂O), Ethylene glycol (EG), Ethanol and Na₂CO₃ were purchased from Sinopharm Chemical Reagent Co., ³⁵ Ltd., Shanghai, China. LiBr was bought from Sigma-Aldrich. All the reagents were used without further purification.

Preparation of SF solutions

SF solutions were prepared according to our previously published procedures.⁴⁷ *B mori* cocoons were boiled for 20 min in an ⁴⁰ aqueous solution of 0.02 M Na₂CO₃ and then rinsed thoroughly with distilled water to extract the sericin proteins. After drying, the extracted silk was dissolved in 9.3M LiBr solution at 60°C for 4 h, yielding a 20% (w/v) solution. This solution was dialyzed against distilled water using dialysis tube (MWCO 3500) for 72 h ⁴⁵ to remove the salt. Then the solution was centrifuged at 9000 rpm for 20 min at 4 °C to remove silk aggregations formed during the process. The final concentration of SF in water was about 7 wt%, determined by weighing the remained solid after drying. The prepared SF solution was stored at 4 °C for future use.

50 Synthesis of Fe₃O₄/SF nanomaterial

Core shell Fe₃O₄/SF nanoparticles were synthesized with a modified solvothermal procedure.⁴⁸ In a typical procedure, 1.35g iron chloride hexahydrate (FeCl₃·6H₂O) and 5.97g NaAc·3H₂O were dissolved in 60ml ethylene glycol (EG) under stirring for ⁵⁵ 0.5 h. Then 10ml of 7 wt% SF solution was added into the blend under continuous stirring for another 0.5 h to form transparent solution. The solution was transferred into a 100ml Teflon-lined stainless-steel autoclave. After maintained at 160 °C for 12 h, the autoclave was cooled at room temperature. The as-prepared ⁶⁰ Fe₃O₄/SF nanoparticles were collected and separated from free SF by centrifuging the solution at 8000 rpm for 30 min, and then washed by ethanol and water for three times, respectively. 10 ml of SF solution with different concentrations were also added into the above reaction system to achieve Fe₃O₄/SF nanoparticles with ⁶⁵ various morphologies and sizes.

Characterizations

X-ray diffraction (XRD) patterns of the samples were recorded on a X'Pert-Pro diffractometer (PANalytical, Holland) with Cu Kα radiation in the 20 range from 10 to 90° at 40 kV and 40 mA. The 5 hydrodynamic diameter and zeta potential of Fe₃O₄/SF were determined with a Malvern Zetasizer Nano ZS90 instrument. The magnetic properties of the products were characterized by vibrating sample magnetometry (VSM) in an applied magnetic field sweeping from -20 KOe to 20 KOe at 300 K. The phase of ¹⁰ Fe₃O₄/SF was determined by Tecnai FEI 20 transmission electron microscopy (TEM). SEM images were obtained with a field emission scan electron microscopy (SEM, SU8010, Hitachi, Japan) at 15kV. Additionally, the inductively coupled plasma atomic emission spectroscopy (ICP-AES, OPTIMA 8000, ¹⁵ PerkinElmer, USA) was used to analyze the element of Fe in the

samples. During this process, Fe solution with standard concentration (5 ppm, 25 ppm, 50 ppm) was used to determine a calibration curve.

Cellular experiment

²⁰ Human dermal fibroblasts (Hs 865.Sk cell, American Type Culture Collection, ATCC) and human lung cancer cells (NCI-H460 cells, Cell Bank of Chinese Academy of Sciences, Shanghai, China) were cultured with Fe₃O₄/SF nanoparticles to assess the influence of the nanoparticles on normal and cancel ²⁵ cells, respectively.

Hs 865.Sk cells were cultured in Dulbecco's Modified Eagle's Medium (DMEM medium) supplemented with 10% fetal bovine serum (FBS) and 1% IU ml⁻¹ streptomycin-penicillin (all from Invitrogen, Carlsbad, CA). NCI-H460 cells were grown in RPMI-

³⁰ 1640 culture medium supplemented with 10% fetal bovine serum (FBS) and 1% streptomycin-penicillin. All the cells were maintained in a humidified incubator containing 5% CO₂ at 37 °C.

CCK-8 assay is an effective way of determining the number of ³⁵ viable cells. ⁴⁹ Considering it could provide more sensitive results than that of MTT, XTT, MTS or WST-1 assays, the CCK-8 assay was used to assess the cytotoxicity of Fe₃O₄/SF and pure Fe₃O₄ samples. Typically, 100 μ L of cells (1×10⁴ cells per well) were seeded in 96-well plates and incubated for 24h. Then, Fe₃O₄/SF 40 with different concentrations (the equivalent Fe concentrations were 100, 80, 60, 40, 20, 10 and 0 µg Fe/ml) were added into each group and incubated for another 24h. After discarding the culture medium, the cells were treated with 10% CCK8 in MPC media (no phenol red) for 1 h at 37 ° C. Absorbance at 450 nm 45 was measured using a microplate reader (Bio-Tek synergy 4, BioTek, USA). Cell viability in eight parallel wells was evaluated for each dose above, and each experiment was repeated at least three times. Cell viability was calculated by means of the following formula:

Cell viability (%) = $\frac{\text{OD450(sample)} - \text{OD450(blank)}}{\text{OD450(control)} - \text{OD450(blank)}} * 100\%$

FITC-Avidin was used to couple with Fe₃O₄/SF nanoparticles to assess the possibility of specific cancer cell targeting ability. Typically, FITC-Avidin was firstly used to couple with Fe₃O₄/SF nanoparticles by EDC coupling reaction. Then, 200 µL of the ⁵⁵ stock solution (biotinylated anti-CD3 antibody was dissolved in PBS buffer to make a 0.5 mg /ml stock solution) was mixed with 250 µL of FITC-Avidin-coupled Fe₃O₄/SF nanoparticles suspension. The mixture was incubated for 20 min at room temperature and then centrifuged at 10000 rpm for 5 min. The ⁶⁰ obtained nanoparticles were washed two more times with 1 ml PBS buffer each time. Lastly, 2 ml of cell suspension (2×10⁵ cells /ml) was added to each well of a 12-well plate and incubated with 200 µg of Biotinylated anti-CD₃ antibody-FITC-Avidin-Fe₃O₄/SF. The plate was incubated at 37 °C for 2 h, and then the cells were ⁶⁵ spun down at 1000 rpm for 10 min and re-suspended in PBS

Journal of Materials Chemistry B Accepted Manuscrip

buffer. The cells were imaged using a confocal laser scanning microscope (AXIO OBSERVER A1, Zeiss, Germany).⁵⁰

Measurement of T_2 relaxivity

 T_2 relaxivity means the efficiency of spin-spin relaxation. Stronger T_2 relaxivity leads to more obvious decrease in signal intensity of various target organs on T_2 -weighted images.^{51, 52} In order to measure the T_2 relaxivity, Fe₃O₄/SF nanoparticles and Fe₃O₄ nanoparticles with different iron concentrations were dispersed in 1% agarose solution. The samples were scanned at 10 room temperature using a T_2 -weighted fast spin-echo multi-slice sequence (fSEMS) (TR/TE=3,000/13.6, 27.2, 54.4, 81.6, 136 ms, slice thickness=2.0 mm) by a 3T MRI scanner (3T Trio Tim, Siemens).¹⁶

15 Statistical analysis

Statistical analysis was conducted using the two-tailed Student's test. The data were presented as means \pm S.D. obtained from three independent experiments. Results were considered to be statistically significant at P<0.05.

20

Results and discussion

The superparamagnetic nanoparticles used in biomedical applications need to be coated with biocompatible polymers for its long-term stability and further functionalization. At least two ²⁵ steps including Fe₃O₄ nanoparticle formation and then coating process are generally required to form the final core shell structure. A one-step approach was developed to prepare superparamagnetic nanoparticles with SF as template and coating material in our present study, providing a simpler and more ³⁰ feasible way for core-shell Fe₃O₄/SF nanoparticle fabrication.

Figure 1 showed the SEM images of Fe_3O_4 nanoparticles prepared with different amounts of SF as template and coating

material. The formed pure Fe₃O₄ nanoparticles showed irregular morphologies with size of about 20 nm. Following the increase of 35 SF in the reaction system, the Fe₃O₄ nanoparticles became more regular and finally transformed into nanospheres when the concentration of SF was above 7%. The size of the Fe₃O₄ nanospheres could be further increased from 120 nm to 200 nm and 500 nm if the SF concentration was 10 % and 15 %, 40 respectively, which was consistent with the dynamic light scattering results (Figure S1 in the supporting information, SI). The size distribution of the Fe₃O₄/SF particles was also supplied in Figure S1. Although the diameters of the nanoparticles increased following the increase of SF concentration, more ⁴⁵ Fe₃O₄/SF particles with inhomogeneous diameters appeared when the SF concentrations were 10 wt% and 15 wt%. The results indicated that best morphology control of the nanoparticles was achieved when the SF concentration was 7 wt%. The highmagnification SEM image (Figure S2) indicated that these 50 nanospheres were composed of smaller nanoparticles (10 nm). Transmission electron microscopy (TEM) provided further insight into the morphology and microstructure of the different nanoparticles (Figure 2). The same morphology transition from irregular shape to nanospheres also appeared in TEM images, 55 which was consistent with the SEM observations. Some pale areas and dark smaller nanoparticles were also observed inside Fe₃O₄ nanospheres, indicating that the nanospheres were not totally compact and aggregated from smaller Fe₃O₄ particles. In addition, it also can be observed that the surface of nanospheres 60 became blurry because of SF coating. Zeta potential changes of our products also confirmed the formation of SF coating outside the nanospheres. The compact SF coating could gradually form and result in maximum negative value of zeta potential when the SF concentration increased from 0 to 7 wt%. After the SF 65 concentration further increased, the thickness of the compact SF

coating increased while the zeta potential became more positive, possibly due to the aggregation of SF in the reaction process.⁵³Although further study is still necessary to clarify the reason for the zeta potential change of the different Fe₃O₄ s nanospheres, the TEM and zeta potential results had confirmed that SF could be coated on the Fe₃O₄ particles to form core-shell structure through simple one-step process. Furthermore, SF coating formed outside the nanoparticles was uniform due to the strong coordinative effect of –COO-Fe.⁵⁴⁻⁵⁷ The coating thickness 10 slightly increased from 2.7 nm to 4.4 and 5.3 nm when the SF

concentrations were 7 wt%, 10 wt% and 15 wt%, respectively. The desired Fe₃O₄ phase formation was evaluated with X-ray powder diffraction (XRD). Figure 3 showed the XRD patterns of Fe₃O₄ and Fe₃O₄/SF nanoparticles. Six diffraction peaks appeared 15 at $2\theta = 30.3^{\circ}$, 35.6° , 43.3° , 53.7° , 56.9° and 62.9° , which correspond to the (220), (311), (400), (422), (511) and (440) planes of Fe₃O₄ (JCPDS card no. 01-089-2355), respectively. No evident differences were observed for pure Fe₃O₄ and Fe₃O₄/SF nanoparticles, suggesting that the crystalline structure of the 20 nanoparticles was not affected by SF coating. Following the increase of SF content in the reaction system, the intensities of the Fe₃O₄ diffraction planes in the Fe₃O₄/SF nanoparticles increased and finally achieved a platform when the SF concentration was above 7%. These results implied that suitable ²⁵ amount of SF could promote the crystallization of Fe₃O₄ in the reaction system.

Several studies have revealed that SF can induce crystallization and formation of Fe₃O₄ nanoparticles.^{46, 58-60} However, the morphology control of the Fe₃O₄/SF nanoparticles remains a ³⁰ challenge. It is generally believed that the tyrosine residues in SF macromolecular chains have a strong interaction with Fe (III), ⁶¹⁻ ⁶² making silk-Fe complexes possible in different reaction systems. More recently, Chen et al prepared hematite

nanoparticles by a SF-assisted hydrothermal method.³⁵ They 35 further clarified the multiple functions of SF in Fe₂O₃ formation process. SF chains firstly induced spherical Fe₂O₃ nanoparticle formation and then controlled further growth to achieve primary nanoparticles. Afterword, the self-assemble of SF chains on the surface of those primary nanoparticles led to the formation of 40 large nanospheres. Although Fe₃O₄ nanospheres were not synthesized by a similar SF-assisted hydrothermal method in their group and the shape of Fe₂O₃ nanospheres was not round, the study implied that the ratios, secondary conformations and nanostructures of SF in the reaction system were critical in the 45 morphology control of the different nanoparticles. Therefore, different to previous SF-assisted hydrothermal method, SF was firstly assembled into nanofibers with higher zeta potential (-33.1 mV) along with the addition of EG in our study (Figure S3), providing stronger interaction with Fe (III) and also stronger 50 aggregations of the primary nanoparticles. After the modification, Fe₃O₄/SF core-shell nanospheres with improved round shapes and bigger sizes were achieved through similar hydrothermal process (Scheme 1). The fabrication of magnetic nanoparticles with biocompatible and biodegradable shells has been developed 55 in recent years. Natural biomaterials such as chitosan and alginate have been used to coat the Fe₃O₄ nanoparticles.^{19, 30} However, the preparation of Fe₃O₄ nanoparticles and the coating formation were different processes in the previous studies, resulting in at least two steps for the core-shell nanoparticles. In our study, the 60 core-shell nanospheres with limited controllable sizes were easily prepared in a one-step process, which would facilitate further biomedical applications and modifications.

The Fe₃O₄/SF nanoparticles prepared form the reaction system containing 7wt% SF were then used to assess its possible ⁶⁵ biomedical applications since the formed nanoparticles had more uniform morphology, and better dispersibility in aqueous

Journal of Materials Chemistry B Accepted Manuscript

solution. The magnetism of Fe₃O₄/SF and Fe₃O₄ nanoparticles were measured by sweeping the external magnetic field between -20 and 20 KOe at room temperature (Figure 4a). The Fe₃O₄/SF nanoparticles showed insignificant coercivity, suggesting the ⁵ superparamagnetic nature of the particles. The saturation magnetization (M_s) of Fe₃O₄/SF nanoparticles was decreased to a value 35 emu/g, compared to 50 emu/g of the control (pure Fe₃O₄ nanoparticles), which could be ascribed to the weight contribution from SF coating. TGA results further confirmed the ¹⁰ assumption (Figure S4). Although the M_s was decreased after SF formation, it was still large enough to facilitate the quick separation of particles from solution, using a regular magnetic plate (Figure 4a insert).

The potential of Fe₃O₄/SF nanoparticles as a contrast agent was ¹⁵ examined by using a 3T MR scanner. An obvious darkening of *T*₂-weighted MRI was observed with the increasing concentration of Fe, showing a transverse relaxivity (*r*₂) 40 mM⁻¹s⁻¹ (Figure 4). Although the value of transverse relaxivity is still inferior to some results reported in other works,^{54, 55} a similar relaxivity with ²⁰ VSOP-C184 and Supravist was achieved in our Fe₃O₄/SF nanoparticles, which indicates the potential application of the Fe₃O₄/SF nanoparticles as MRI contrast agents.

The cytotoxicity of Fe₃O₄/SF and Fe₃O₄ nanoparticles was investigated with CCK8 assay. The Hs 865.Sk and NCI-H460 ²⁵ cells were incubated with Fe₃O₄/SF and Fe₃O₄ nanoparticles for

24h with concentrations of 10, 20, 40, 60, 80 and 100 μ g Fe /ml (measured by ICP-AES) at 37 °C (Figure 5). The results showed that above 80% of cells maintained viable even at highest concentration of 100 μ g Fe /ml, indicating the good ³⁰ biocompatibility of the core-shell Fe₃O₄/SF nanoparticles.

The surface area of the Fe_3O_4/SF nanoparticles was measured with nitrogen adsorption-desorption isotherms (Figure S5). The Brunauer-Emmett-Teller (BET) surface area of the Fe_3O_4/SF 35 Fe₃O₄ nanoparticles (6 m²g⁻¹). The results suggest a potential application of the Fe₃O₄/SF nanoparticles as a drug carrier. Then, FITC-Avidin that could bind Biotinylated anti-CD₃ antibody through Avidin-biotin interaction was used to couple with Fe₃O₄/SF nanoparticles, offering the Fe₃O₄/SF nanoparticles ⁴⁰ specific cancer cell targeting ability.⁵⁰ As shown in Figure 6, after incubation with the T-lymphocytic cell line Jurkat and human bone marrow-derived mesenchymal stem cell, stronger green fluorescence (Biotinylated anti-CD3 antibody-FITC-Avidin-Fe₃O₄/SF) attaching to the cell was achieved in positive group, 45 suggesting successful couple of Biotinylated anti-CD₃ antibody-FITC-Avidin with the nanoparticles.⁵⁰ The results indicated that the NPs prepared in our study could be further designed as targeting vehicle by surface modification, and then used to distinguish cancer cells from normal cells. However, it was also 50 found that not all T-lymphocytes were labeled and the aggregation of nanoparticles appeared, which might be due to the neutralization of negative charge in the system. Coating the nanoparticless with polymers with positive charge through layerby-layer process might be a feasible way to resolve the problem 55 but makes the method more complicated, which should be further studied in our next work. Overall, the biocompatibility, magnetism, and higher surface area of the Fe₃O₄/SF nanoparticles imply their potential applications in MRI contrast agents and drug carriers.

nanoparticles was 46 m^2g^{-1} , significantly higher than that of pure

60 Conclusions

Biocompatible core-shell Fe_3O_4/SF nanoparticles was successfully prepared via a one-step solvothermal process. Changing the contents of SF in reaction system could modulate the sizes of the nanoparticles in the range of 120-500 nm. The 65 suitable magnetism, biocompatibility as well as further fabrication ability of the nanoparticles implied their promising applications in different biomedical fields. Therefore, our present study provides a new way to design functional Fe_3O_4/SF nanoparticles.

5 Notes and references

- ^aNational Engineering Laboratory for Modern Silk & Collaborative Innovation Center of Suzhou Nano Science and Technology, Soochow University, Suzhou 215123, People's Republic of China. Tel: (+86)-512-67061649; E-mail: Lvqiang78@suda.edu.cn.
- 10 b
 Regenerative Medicine Center,
 First

 Affiliated Hospital of Dalian Medical
 University,
 No.222
 Zhongshan

 Road, Dalian 116011, People's Republic of China
 View Providential
 View Providential
 View Providential
 - ^cDepartment of Biomedical Engineering, Tufts University, Medford, MA 02155, USA
- ¹⁵ ^dResearch Center of Materials Science, Beijing Institute of Technology, Beijing, 100081, People's Republic of China
 - † Electronic Supplementary Information (ESI) available: [DLS and zeta potential of the products with different SF concentrations; highmagnification SEM of the products with 7% SF; AFM images of SF
- $_{20}$ before and after addition of EG; TGA curves of the Fe₃O₄/SF and pure Fe₃O₄]. See DOI: 10.1039/b000000x/
 - ‡ These authors contributed equally to this work.
 - 1. J. E. Lee, N. Lee, T. Kim, J. Kim, and T. Hyeon, Acc. Chem.
- 25 Res., 2011, 44, 893.
 - D. E. Lee, H. Koo, I. C. Sun, J. H. Ryu, K. Kim, and I. C. Kwon, *Chem. Soc. Rev.*, 2012, 41, 2656.
 - 3. J. Gao, H. Gu, and B. Xu, Acc. Chem. Res., 2009, 42, 1097.
 - 4. Y. D. Jin, C. X. Jia, S. W. Huang, M. O'Donnell, and X. H.
- 30 Gao, Nat. Commun., 2010, 1, 41.
 - 5. S. H. Hu, and X. Gao, J. Am. Chem. Soc., 2010, 132, 7234.
- Z. Y. Xiao, E. Levy-Nissenbaum, F. Alexis, A. Lupták, B. A. Teply, J. M. Chan, J. J. Shi, E. Digga, J. Cheng, R. Langer, and O. C. Farokhzad, *ACS Nano*, 2012, 6, 696.

- 35 7. D. Kim, Y. Y. Jeong, and S. Jon, ACS Nano, 2010, 4, 3689.
- T. D. Schladt, M. I. Shukoor, K. Schneider, M. N. Tahir, F. Natalio, I. Ament, J. Becker, F. D. Jochum, S. Weber, O. Köhler, P. Theato, L. M. Schreiber, C. Sönnichsen, H. C. Schröder, W. E. G. Müller, and W. Tremel, *Angew. Chem. Int.*
- *Ed.*, 2010, **49**, 3976.
- J. Wang, G. Z. Zhu, M. X. You, E. Song, M. I. Shukoor, K. J. Zhang, M. B. Altman, Y. Chen, Z. Zhu, C. Z. Huang, and W. H. Tan, *ACS Nano*, 2012, 6, 5070.
- 10. J. F. Lovell, C. S. Jin, E. Huynh, H. Jin, C. Kim, J. L.
- Rubinstein, W. C. W. Chan, W. Cao, L. V. Wang, and G. Zheng, *Nat. Mater.*, 2011, 10, 324.
- M. I. Majeed, Q. W. Lu, W. Yan, Z. Li, I. Hussain, M. N. Tahir, W. Tremele, and B. Tan, *J. Mater. Chem. B*, 2013, 1, 2874.
- ⁵⁰ 12. C. Wang, H. Xu, C. Liang, Y. M. Liu, Z. W. Li, G. B. Yang, L. Cheng, Y. G. Li, and Z. Liu, *ACS nano*, 2013, 7, 6782.
- Q. A. Pankhurst, J. Connolly, S. K. Jones, and J. Dobson, J. Phys. D : Appl. Phys., 2003, 36, R167.
- 14. J. Kim, J. E. Lee, J. Lee, J. H. Yu, B. C. Kim, K. An, Y.
- Hwang, C. H. Shin, J. G. Park, J. Kim, and T. Hyeon, J. Am. Chem. Soc., 2006, **128**, 688.
- B. Luo, S. Xu, A. Luo, W. R. Wang, S. L. Wang, J. Guo, Y. Lin, D. Y. Zhao, and C. C. Wang, *ACS nano*, 2011, 5, 1428.
- 16. Z. H. Zhao, Z. J. Zhou, J. F. Bao, Z. Y. Wang, J. Hu, X. Q.
- ⁶⁰ Chi, K. Y. Ni, R. F. Wang, X. Y. Chen, Z. Chen, and J. H. Gao, *Nat. Commun.*, 2013, 4, 2266.
- H. X. Wu, S. J. Zhang, J. M. Zhang, G. Liu, J. L. Shi, L. X.
 Zhang, X. Z. Cui, M. L. Ruan, Q. J. He, and W. B. Bu, *Adv. Funct. Mater.*, 2011, 21, 1850.
- ⁶⁵ 18. X. Y. Shi, S. H. Wang, S. D. Swanson, S. Ge, Z. Y. Cao, M. E. V. Antwerp, K. J. Landmark, and J. R. Baker. Jr, *Adv. Mater.*, 2008, **20**, 1671.

Page 10 of 19

- J. L. Arias, L. H. Reddy, and P. Couvreur, J. Mater. Chem., 2012, 22, 7622.
- P. Kucheryavy, J. B. He, V. T. John, P. Maharjan, L. Spinu, G.
 Z. Goloverda, and V. L. Kolesnichenko, *Langmuir*, 2013, 29, 710.
- X. J. Song, H. Gong, S. N. Yin, L. Cheng, C. Wang, Z. W. Li,
 Y. G. Li, X. Y. Wang, G. Liu, and Z. Liu, *Adv. Funct. Mater.*,
 2014, 24, 1194.
- 22. L. S. Lin, Z. X. Cong, J. B Cao, K. M. Ke, Q. L. Peng, J. H.
- Gao, H. H. Yang, G. Liu, and X. Y. Chen, *ACS Nano*, 2014, **8**, 3876.
- Z. W. Li, S. N.Yin , L. Cheng , K. Yang , Y. G. Li , and Z. Liu, *Adv. Funct. Mater.*, 2014, 24, 2312.
- 24. N. Tran, D. Hall, and T. J. Webster, Nanotechnology, 2012,
- 15 **23**, 455104.
 - I. Bajpai, K. Balani, and B. Basu, J. Am. Ceram. Soc., 2013, 96, 2100.
 - 26. W. Chen, T. Long, Y. J. Guo, Z. A. Zhu, and Y. P. Guo, J. Mater. Chem. B, 2014, 2, 1653.
- ²⁰ 27. Y. H. Deng, D. W. Qi, C. H. Deng, X. M. Zhang, and D. Y. Zhao, *J. Am. Chem. Soc.*, 2008, **130**, 28.
 - J. F. Zeng, L. H. Jing, Y. Hou, M. X. Jiao, R. R. Qiao, Q. J. Jia, C. Y. Liu, F. Fang, H. Lei, and M. Y. Gao, *Adv. Mater.*, 2014, 26, 2694.
- ²⁵ 29. S. Y. Gao, Y. G. Shi, S. X. Zhang, K. Jiang, S. X. Yang, Z. D. Li, and E. Takayama-Muromachi, *J. Phys. Chem. C*, 2008, **112**, 10398.
 - X. Liu, X. Chen, Y. F. Li, X. Y. Wang, X. M. Peng, and W.
 W. Zhu, ACS Appl. Mater. Interfaces, 2012, 4, 5169.
- 30 31. E. A. Osborne, T. M. Atkins, D. A. Gilbert, S. M. Kauzlarich1, K. Liu, and A. Y. Louie, *Nanotechnology*, 2012, 23, 215602.
 - 32. X. Y. Lu, M. Niu, R. R. Qiao, and M. Y. Gao, J. Phys. Chem.

B, 2008, 112, 14390.

- 35 33. M. E. Khosroshahi, and L. Ghazanfari, J. Magn. Magn. Mater., 2012, **324**, 4143.
 - B. B. Nathwani, M. Jaffari, A. R. Juriani, A. B. Mathur, and
 K. E. Meissner, *IEEE T Nanobiosci.*, 2009, 8, 72.
- 35. X. Fei, Z. Z. Shao, and X. Chen, *J. Mater. Chem. B*, 2013, **1**, 213.
- 36. Y. G. Chung, K. Algarrahi, D. Franck, D. D. Tu, R. M. Adam,D. L. Kaplan, C. R. E. Jr., and J. R. Mauney, *Biomaterials*, 2014, 35, 7452.
- E. Wenk, A. J. Wandrey, H. P. Merkle, and L. Meinel, J. Controlled Release, 2008, 132, 26.
- 38. F. P. Seib, E. M. Pritchard, and D. L. Kaplan, *Adv. Funct. Mater.*, 2013, 23, 58.
- 39. N. C. Tansil, L. D. Koh, and M. Y. Han, *Adv. Mater.*, 2012, 24, 1388.
- ⁵⁰ 40. D. H. Kim, Y. S. Kim, J. Amsden, B. Panilaitis, D. L. Kaplan, F. G. Omenetto, M. R. Zakin, and J. A. Rogers, *Appl. Phys. Lett.*, 2009, **95**, 133701.
- 41. P. Domachuk, K. Tsioris, F. G. Omenetto, and D. L. Kaplan, *Adv.Mater.*, 2010, **22**, 249.
- 55 42. F. G. Omenetto, and D. L. Kaplan, Nat. Photon., 2008, 2, 641.
- 43. X. Fei, Z. Z. Shao, and X. Chen, Nanoscale, 2013, 5, 7991.
- 44. X. Fei, M. H. Jia, X. Du, Y. H. Yang, R. Zhang, Z. Z. Shao, X. Zhao, and X. Chen, *Biomacromolecules*, 2013, 14, 4483.
- 45. S. J. Xu, L. Yong, and P. Y. Wu, *Appl. Mater. Interfaces*, 2013,
 5, 654.
- 46. M. Deng, Z. B. Huang, Y. W. Zou, G. F. Yin, J. Liu, and J. W. Gu. *Colloids Surf.*, *B*, 2014, **116**, 465.
- 47. Q. Lu, X. L. Wang, S. Z. Lu, M. Z. Li, D. L. Kaplan, and H. S. Zhu, *Biomaterials*, 2011, **32**, 1059.
- 65 48. H. Deng, X. L. Li, Q. Peng, X. Wang, J. P. Chen, and Y. D. Li, Angew. Chem., Int. Ed., 2005, 44, 2782.

- J. E. Frith, D. J. Menzies, A. R. Cameron, P. Ghosh, D. L.
 Whitehead, S. Gronthos, A. C. W. Zannettino, and J. J.
 Copper-White, *Biomaterials*, 2014, 35,1150.
- X. Q. Wang, and D. L. Kaplan, *Macromol. Biosci.*, 2011, 11, 100.
- 51. R. N. Muller, P. Gillis, F. Moiny, and A. Roch, *Magn. Reson. Med.*, 1991, **22**, 178.
- 52. A. Bjørnerud, and L. Johansson, NMR Biomed., 2004, 17, 465.
- 53. S. M. Bai, S. S. Liu, C. C. Zhang, W. Xu, Q. Lu, H. Han, D. L.
- ¹⁰ Kaplan, and H. S. Zhu, *Acta Biomaterialia*, 2013, **9**, 7806.
- 54. S. H. Xuan, F. Wang, J. M. Y. Lai, K. W. Y. Sham, Y. X. J. Wang, S. F. Lee, J. C. Yu, C. H. K. Cheng, and K. C. F. Leung, ACS Appl. Mater. Interfaces, 2011, 3, 237.
- 55. S. H. Xuan, F. Wang, Y. H. J. Wang, J. C. Yu, and K. C. F.
- Leung, J. Mater: Chem., 2010, 20, 5086.
 - 56. K. X. Yao, and H. C. Zeng, J. Phys. Chem. C, 2007, 111, 13301.
 - 57. T. He, D. R. Chen, and X. L. Jiao, *Chem. Mater.*, 2004, **16**, 737.
- ²⁰ 58. Y. F. Ma, Q. L. Feng, and X. Bourrat, *Mater. Sci. Eng. C*, 2013, **33**, 2413.
 - X. L. Zhang, Z. H. Fan, Q. Lu, Y. L. Huang, D.L. Kaplan, and H. S. Zhu, *Acta Biomater.*, 2013, 9, 6974.
 - 60. B. B. Peksen, C. Uzelakcil, A. Gunes, O. Malay, and O.
- 25 Bayraktar, J. Chem. Technol. Biotechnol., 2006, 81, 1218.
- X. H. Lin, G. B. Ji, Y. S. Liu, Q. H. Huang, Z. H. Yang, and
 Y. W. Du, *CrystEngComm.*, 2012, 14, 8658.
- 62. Q. Song and Z. J. Zhang, J. Am. Chem. Soc., 2004, 126, 6164.

30

Figure 1 SEM images of the products prepared under different silk fibroin contents. 10 ml of silk fibroin aqueous solutions added into the reaction system had different concentrations as follows: (a) pure water, (b) 0.1wt%, (c) 2.0 wt%, (d) 4.0 wt%, s (e)7.0 wt%, (f)10.0 wt%, (g)15.0 wt%

Figure 2 TEM images of the products prepared under different silk fibroin contents. 10 ml of silk fibroin aqueous solutions added into the reaction system had different concentrations as follows: (a) pure water, (b) 0.1 wt%, (c) 2.0 wt%, (d) 4.0 wt%, ¹⁰ (e)7.0 wt%, (f)10.0 wt%, (g)15.0 wt%.

Figure 3 XRD patterns of Fe₃O₄/SF nanomaterial prepared under different silk fibroin contents. 10 ml of silk fibroin aqueous solutions added into the reaction system had different concentrations as follows: (a) pure water, (b) 0.1 wt%, (c) 2.0
¹⁵ wt%, (d) 4.0 wt%, (e)7.0 wt%, (f)10.0 wt%, (g)15.0 wt%.

Figure 4 (a) Magnetic hysteresis loops of Fe₃O₄/SF and Fe₃O₄ nanoparticles measured at room temperature by VSM, Inserts were the photographs of Fe₃O₄/SF suspension and Fe₃O₄/SF with an external magnetic field, (b) T_2 -weighted MRI image of Pe₃O₄/SF in aqueous solution (containing 1% agarose) with different Fe concentrations, (c) The linear fitting of relaxation rates (R_2) versus Fe concentrations of Fe₃O₄/SF. Additionally, R_2 is equal to 1/ T_2 , and the relaxivity value (r_2) was obtained from the slopes of the line.

²⁵ Figure 5 Relative viabilities of Hs 865.sk cells (a) and NCI-H460 cells (b) determined by the CCK8 assay after incubation with Fe₃O₄/SF and Fe₃O₄ nanoparticles with different Fe concentrations at 37 °C for 24h. The cells cultured in same culture medium without Fe as positive control. Cell viability in ³⁰ eight parallel wells was evaluated for each dose, and each experiment was repeated at least three times. *Statistically

Figure 6 Linkage of biotinylated anti-CD₃ antibody to the surface

of FITC-Avidin-Fe₃O₄/SF and specific targeting of the modified ³⁵ Fe₃O₄/SF to CD₃ positive T-lymphocytic cell line Jurkat. (a, b): positive groups (CD₃ + lymphocytes); (c, d) negative groups (human bone marrow-derived mescenchymal stem cells). (a, c): the overlay of DAPI fluorescence, FITC fluorescence and the bright field images of cells (the DAPI fluorescence associated ⁴⁰ with nucleus, and the FITC fluorescence associated with Biotinylated anti-CD₃ antibody-FITC-Avidin-Fe₃O₄/SF); (b, d) the bright field images of cells.

Scheme 1 Possible formation mechanism of Fe_3O_4/SF 45 nanomaterial.

significant P<0.05.

5

10

15



20 Figure 1



Figure 2



Figure 3



Figure 4



10

Figure 5



Figure 6

20



Scheme 1