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# <sup>7</sup> Engineering three-dimensional structures using bio <sup>8</sup> inspired dopamine and strontium on titanium for <sup>9</sup> biomedical application

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<sup>12</sup> The excellent mechanical properties and chemical stability of titanium and its alloys have <sup>13</sup> led to their wide use as a material for dental and orthopaedic implants. However, the bio-<sup>14</sup> inert nature of these materials must be overcome to enhance cell affinity and cell function <sup>15</sup> following implantation. Effective implants require strong interfacial bonding, mechanical <sup>16</sup> stability, osteoblast attachment, enhanced spreading and growth during early stages, and <sup>17</sup> induced differentiation and mineralization in later stages. This study developed an organic-<sup>18</sup> inorganic multilayer coating process for the modification of titanium implants in order to <sup>19</sup> improve cell responses. A three-dimensional structure comprising strontium and micro-arc <sup>20</sup> oxidized (MAO) titanium was covered with a film of poly(dopamine) to form a multilayer <sup>21</sup> coating. The titanium surface formed a uniform hydrophilic oxide coating, which was firmly <sup>22</sup> adhered to the surface. The poly(dopamine) film facilitated the initial attachment and <sup>23</sup> proliferation of cells. Cell differentiation was enhanced by the release of strontium from the <sup>24</sup> coatings. Our results demonstrate the efficacy of the proposed coating process in enhancing <sup>25</sup> the multi-biological function of implant surfaces.

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#### 28 1. Introduction

Titanium and its alloys have been widely used as a 29 30 framework material in dental and orthopedic implants, due to 31 their excellent mechanical properties and chemical stability. 32 However, the natural oxide thin film which forms on titanium 33 is bio-inert, which makes it difficult to achieve a chemical 34 bond with living tissue.<sup>1</sup> The effectiveness of an implant 35 depends on the development of strong anchorage between the 36 material comprising the implant and existing bone tissue. 37 Various forms of surface modification have been used to 38 accelerate the initial osseointegration immediately after 39 implantation in order to enhance the reactivity of the tissue 40 and thereby shorten the healing period of the bone. 41 Considerable research has been devoted to the development <sup>42</sup> of techniques for coating titanium, such as plasma spraying,<sup>2</sup> 43 electro-deposition,<sup>3</sup> and sputtering.<sup>4</sup> Currently, titanium and 44 its alloys are commonly coated with hydroxyapatite (HA) 45 through plasma-spraying; however surface coverage is low 46 and a lack of uniformity in the coating hinders adhesion to 47 the substrate.<sup>5</sup>

Ideally, implant surface coatings should mimic themorphology and function of natural bone in order to optimizethe integration of the implant and existing human bone tissue.

51 Porous coatings provide more surface area; therefore, 52 microporous coatings appear to be a promising approach to 53 the further development of dental and orthopedic implants. 54 Micro-arc oxidation (MAO) (also commonly called "plasma 55 electrolytic oxidation (PEO)", "microplasma oxidation 56 (MPO)" and "spark anodizing") is an effective technique used 57 to produce ceramic-like coatings with porous structures. 58 Three-dimensional structures with numerous craters can 59 improve surface properties, such as wear and corrosion 60 resistance, as well as enhance biological performance. The 61 dense, strongly adherent coating produced by MAO can 62 minimize spalling and reduce osteolysis, which is generally 63 triggered by the loosening of particles on implants. Rough 64 surfaces can also improve early fixation and long-term 65 mechanical stability.<sup>6,7</sup> The bioactivity of MAO coatings can 66 be further modified by combining MAO with other activation 67 methods, such as electric polarization and ultraviolet (UV) 68 irradiation.<sup>8, 9</sup> The MAO method makes it possible to 69 incorporate multiple ions within an oxide layer, simply by 70 adjusting the compositions of the electrolyte solutions. It has 71 even been proposed that the introducing of Ag into an MAO 72 coating could provide antibacterial properties.<sup>10</sup> In a previous 73 study, we investigated the incorporation of manganese ions

1 into MAO coatings, the subsequent release of which 2 enhanced cell differentiation and mineralization.<sup>11</sup>

<sup>3</sup> Strontium (Sr) has been attracting considerable attention <sup>4</sup> for the clinical treatment of osteoporosis. Like calcium (Ca), <sup>5</sup> Sr is a bone-seeking element and approximately 98% of the <sup>6</sup> total Sr in the body is localized in bone tissue. Strontium-<sup>7</sup> substituted hydroxyapatite (Sr-HA) has the same charge as <sup>8</sup> calcium, and has therefore been adopted as a filler material to <sup>9</sup> improve the biocompatibility of bone cement.<sup>12, 13</sup> The <sup>10</sup> presence of Sr in coatings can enhance osteoblast activity and <sup>11</sup> differentiation, while inhibiting the production and <sup>12</sup> proliferation of osteoclast.<sup>14, 15</sup> Previous studies have <sup>13</sup> demonstrated the efficacy of MAO coatings incorporating <sup>14</sup> strontium, calcium, and phosphate (P) in enhancing cell <sup>15</sup> responses.<sup>16</sup>

Considerable improvements have been made 16 in 17 modifying the topography of implants; therefore, it is 18 expected that the immobilization of bioactive molecules 19 (biochemical modification) will be the next major focus. A 20 number of naturally occurring adhesives have recently 21 attracted attention. For example, the 3,4-22 dihydroxyphenethylamine (dopamine) molecule is an 23 adhesive protein that allows mussels to adhere to a variety of 24 materials.<sup>17</sup> Poly(dopamine)-based film could be used in 25 biomedical applications as an intermediate layer for the 26 immobilization of other biofunctional molecules.<sup>18-20</sup> In a 27 previous study, we were inspired by the mussels of genus 28 Mytulis to develop a synthetic adhesive platform for tissue-<sup>29</sup> implant applications.<sup>21</sup> In the current study, we proceeded on 30 the assumption that a poly(dopamine) film could provide a 31 bioreactive surface with which to enhance the cell response to 32 MAO coatings.

In the present study, the MAO method was employed to modify a titanium surface in aqueous electrolytes, such that Ca, P, and Sr could be incorporated into the porous oxide ocating to improve the differentiation of osteoblast cells. We r then employed self-polymerization to generate a bioactive film of poly(dopamine) on the surface of the resulting coating. The addition of dopamine and Sr was shown to alter the local chemistry and in so doing change the biological properties of the MAO coating. The result is an increase in cell response, capable of altering cell morphology, cell proliferation, and bone-related gene expression.

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#### 45 2. Materials and methods

#### 46 2.1 Preparation of specimens

<sup>47</sup> Medical grade titanium (commercially pure titanium, <sup>48</sup> Grade 2, ASTM F-67 S-Tech Co., Taiwan) was used as a <sup>49</sup> substrate. Substrate discs 12.7 mm (diameter)  $\times$  2 mm <sup>50</sup> (thickness) were ground and polished using 1500-grit <sup>51</sup> followed by ultrasonic cleaning in acetone, ethanol, and <sup>52</sup> distilled water prior to MAO processing. The MAO reaction <sup>53</sup> apparatus was constructed in a two-electrode electrochemical 54 cell with DC power supply (GPS-60H15S, Good Will 55 Instrument Co., Taiwan) using stainless steel as the cathode 56 and the titanium specimen as the anode. The electrolyte was 57 prepared by dissolving sodium phosphate monobasic 58 monohydrate (NaH<sub>2</sub>PO<sub>4</sub>·H<sub>2</sub>O, 99.7%, J.T. Baker, USA), 59 calcium acetate hydrate (Ca(CH<sub>3</sub>COO)<sub>2</sub>·H<sub>2</sub>O, 99%, Acros, 60 USA) and strontium acetate hemihvdrate 61 (Sr(CH<sub>3</sub>COO)<sub>2</sub>·0.5H<sub>2</sub>O, 98%, Alfa Aesar, USA) in distilled 62 water. The compositions of the electrolyte solutions are 63 presented in Table 1. The samples were treated with an 64 applied voltage of 350 V for 1 min with the temperature 65 maintained at 25°C by circulating water through the 66 electrochemical cell. As shown in Table 1, the MAO 67 specimens are denoted as CaP and SrCaP, in accordance with 68 the preparation of the materials in the electrolytic solution. 69 As-prepared specimens were ultrasonically cleaned in 70 acetone, ethanol, and distilled water prior to dopamine 71 modification. The as-prepared specimens then underwent 72 dopamine polymerization involving immersion in a solution 73 of 10 mM Tris-HCl solution adjusted to pH 8.5 with added 74 dopamine at a concentration of 2 mg/ml. The colour of the 75 solution changed from clear to dark brown resulting from pH-76 induced oxidative reaction. The reaction was performed at 77 room temperature, under mixing at 300 rpm for 12 h. The 78 CaP and SrCaP specimens were also immersed in Tris-HCl 79 solution to act as a control group with the same ion 80 concentration as the test samples. All of the specimens were 81 then rinsed using deionized water and dried with N<sub>2</sub> gas. The 82 two resulting MAO substrates with poly(dopamine) coating 83 were denoted as CaPD and SrCaPD, respectively. All of the 84 specimens were sterilized using 70% ethanol and then 85 divided evenly into a 24-well plate for cell assays.

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## 87 2.2 Characterization of poly(dopamine) film coated on MAO88 coatings

The surface morphology of the specimens was observed 90 using a scanning electron microscope (SEM, JSM-6390LV, 91 JEOL) equipped with an energy- dispersive X-ray 92 spectrometer (EDX, INCA/350, Oxford) for chemical 93 analysis. The surface wettability of the specimens was 94 determined using measurements of static contact angle. We 95 employed the sessile drop method using a 5 µl drop of 96 double-distilled water as probing liquid.

<sup>97</sup> The surface chemistry was analysed using X-ray <sup>98</sup> photoelectron spectroscopy (XPS, Thermo K-Alpha, Thermo <sup>99</sup> Scientific, USA), recorded using a monochromatic Al K $\alpha$ <sup>100</sup> source (1486.6 eV). The photoelectron take-off angle was 45° <sup>101</sup> and survey spectra were collected over a range of 0-1200 eV. <sup>102</sup> The C 1s spectra was set to 284.6 eV as a reference to <sup>103</sup> calibrate the scale of binding energy. Intensity ratios were <sup>104</sup> converted into atomic concentration ratios using sensitivity <sup>105</sup> factors.

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Specimen		Se/(Se) Ce)	DOPA		
	NaH <sub>2</sub> PO <sub>4</sub> ·H <sub>2</sub> O	Ca(CH <sub>3</sub> COO) <sub>2</sub> ·H <sub>2</sub> O	Sr(CH <sub>3</sub> COO) <sub>2</sub> •0.5H <sub>2</sub> O	- Sr/(Sr+Ca)	(mg/mL)
Ti					
TiD					2
CaP	0.06	0.13	0	0	
CaPD	0.06	0.13	0	0	2
SrCaP	0.06	0.1170	0.0130	10	
SrCaPD	0.06	0.1170	0.0130	10	2

2 Table 1 Composition of the Sr-MAO electrolyte and DOPA solutions

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#### 4 2.3 Cell culture

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<sup>5</sup> Osteoblastic cells (MC3T3-E1, ATCC number: CRL <sup>6</sup> 2593<sup>TM</sup>) were obtained from the American Type Culture <sup>7</sup> Collection (ATCC, USA) and maintained in alpha Minimum <sup>8</sup> Essential Medium (alpha MEM, Gibco Invitrogen, USA) <sup>9</sup> supplemented with 10% heat-inactivated fetal bovine serum <sup>10</sup> (FBS, Gibco Invitrogen, USA), antibiotics (100 U/ml <sup>11</sup> penicillin, 0.1 mg/ml streptomycin, and 0.25  $\mu$ g/ml <sup>12</sup> amphotericin B, Cashmere Biotech, Taiwan) at 37°C in a <sup>13</sup> humidified atmosphere of 5% CO<sub>2</sub> in air. The medium was <sup>14</sup> replaced twice weekly and cultured until confluence. The <sup>15</sup> cells were washed twice with phosphate-buffered saline <sup>16</sup> (PBS), detached using 0.05% trypsin-EDTA (Gibco <sup>17</sup> Invitrogen, USA), and then centrifuged at 1000 rpm for 5 <sup>18</sup> min.

#### 19 2.4 Initial cell spreading and cytoskeleton development

The specimens were seeded with cells at a density of 3.5 20  $_{21} \times 10^3$  cells/cm<sup>2</sup>. A fluorescence microscope (Axio Observer 22 Z1, Zeiss, Germany) and image analyser (Axio Vision, Zeiss, 23 Germany) were used to examine cell morphology and 24 cytoskeletal arrangement. After culturing for 3 h, cells were 25 fixed using 4% paraformaldehyde (Sigma, USA) for 30 min 26 and permeabilized with 0.1% Triton-X-100 (JT Baker, USA) 27 for 15 min. After being washed with PBS, the cells were 28 incubated in 2% BSA for 40 min. Vinculin (1:150) was 29 labeled with monoclonal antibodies (V9131, Sigma, USA) 30 and anti-mouse IgG conjugated with Alexa flour 594 31 (A11005, Molecular Probes, USA). F-actin (1:200) was 32 stained with Alexa flour 488-labled phallodin (A12379, 33 Molecular Probes, USA). Finally, the coverslips were 34 incubated with DAPI (4',6-Diamidino-2-phenylindole 35 dihydrochloride) nuclear dye and mounted in ProLong® Gold 36 antifade reagents (Molecular Probes, USA) for analysis using 37 a fluorescence microscope.

#### 38 2.5 Cell proliferation

<sup>39</sup> Cell suspensions of all specimens were seeded at a cell <sup>40</sup> density of  $5 \times 10^3$  cells/cm<sup>2</sup>. Following 1, 3, and 7 days of <sup>41</sup> culturing, the number of cells was determined using a 3-[4,5-<sup>42</sup> dimethylthiazol-2-yl]- 2,5-diphenyltetrazolium bromide assay <sup>43</sup> (MTT, Sigma, USA). For this assay, 100 µl of MTT working <sup>44</sup> solution was added to each well. The MTT working solution <sup>45</sup> was then removed after an incubation period of 4 h and <sup>46</sup> combined with insoluble formazan crystal dissolved in <sup>47</sup> dimethyl sulfoxide (DMSO). Analysis was performed using <sup>48</sup> an enzyme-linked immunosorbent assay (ELISA) plate reader <sup>49</sup> (Sunrise, Austria) at 570 nm. The OD<sub>570</sub> were plotted against <sup>50</sup> a series of known cell numbers ( $10^3$ ,  $5 \times 10^3$ ,  $10^4$ ,  $5 \times 10^4$ , <sup>51</sup>  $10^5$ ,  $5 \times 10^5$  cell/well) to establish a standard calibration <sup>52</sup> curve.

#### 53 2.6 ALP activity

All specimens were seeded with cell suspensions in 24-54 s well plates at a cell density of  $5 \times 10^3$  cells/cm<sup>2</sup>. After 7 days 56 of culturing, ALP activity was measured colorimetrically 57 using SIGMAFAST<sup>TM</sup> p-Nitrophenyl phosphate (p-NPP) 58 tablets (Sigma, USA). 100µl p-NPP working solution was 59 added to the supernatant and the reaction was stopped using 60 0.05M NaOH following incubation for 30 min at 37 °C. 61 Analysis was performed using an ELISA plate reader 62 (Sunrise, Austria) at 405 nm. ALP activity was normalized 63 from the total protein content and total protein concentrations 64 were determined using the micro bicinchoninic acid method 65 (BCA, Sigma, USA) according to the instructions of the 66 manufacturer. The concentration of each protein was 67 calibrated using a standard curve according to the instructions 68 of the manufacturer. Further analysis was then performed 69 using the ELISA plate reader at 562 nm.

#### 70 2.7 Initial cell spreading and cytoskeleton development

The messenger RNA (mRNA) expression of integrin-72 binding sialoprotein (IBSP), alkaline phosphatase (ALP), 73 osteopontin (OPN), and osteocalcin (OCN) were 74 quantitatively evaluated by real-time quantitative polymerase 75 chain reaction (Q-PCR) (7500 fast real-time PCR System, 76 Applied Biosystems, USA). After culturing for 7 days, the 77 total RNA was extracted using the total RNA isolation kit 78 (Geneaid, USA) according to the instructions of the 79 manufacturer. Purified RNA was used to synthesize cDNA 80 with MMLV reverse transcriptase (Invitrogen, USA). The 81 resulting cDNA was used for the PCR reaction using Fast 82 SYBR<sup>®</sup> Green Master Mix. The amplification profile 83 involved denaturation at 95 °C for 20 s, followed by 40 84 cycles of 95 °C for 3 s, and 60 °C for 30 s following a

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1 melting curve to check for amplicon specificity. Gene 2 expression was normalized by glyceraldehyde 3-phosphate 3 dehydrogenase (GAPDH) using the comparative cycle 4 threshold (ddCt) method. The following primer sets were 5 used to amplify each target sequence: GAPDH (forward: 5'-6 GGA GTA AGA AAC CCT GGA-3'; reverse: 5'-CTG GGA 7 TGG AAA TTG TGA G-3'), IBSP (forward: 5'-CCG AGC 8 TTA TGA GGA TGA ATA CA-3', reverse: 5'-GGT AGC 9 CAG ATG ATA AGA CAG AAT-3'), ALP (forward: 5'-10 CTG CCT TGC CTG TAT CTG-3', reverse: 5'-GGT GCT 11 TTG GGA ATC TGT-3'), OPN (forward: 5'-CTT TCA CTC 12 CAA TCG TCC CTA-3'; reverse: 5'-GTC CTC ATC TGT 13 GGC ATC A-3'), OCN (forward: 5'-TCG GCT TTG GCT 14 GCT CTC-3'; reverse: 5'-CCT GTG ACA TCC ATA 15 CTT G-3').

#### 16 2.8 Statistical analysis

At least five samples were obtained at each time point and the results are presented as the mean ± standard deviation (SD). One-way analysis of variance (ANOVA) and Duncan's 20 post-hoc tests were performed for all assessments.

#### 21 3. Results and discussion

## 22 3.1 Morphology and characteristics of poly(dopamine) film on23 MAO coatings

Figure 1 presents the surface morphologies of the 24 25 specimens in this study. Compared to the untreated titanium 26 substrate, the MAO process produced a much rougher surface 27 with crater-like structures (Figs. 1c-f). The inset images were 28 recorded at a tilt angle of 30° and clearly illustrate the three-29 dimensional structure of the coatings. All specimens 30 presented coatings with uniform pore distribution and no 31 obvious differences were observed among the specimens. 32 The specimens were immersed directly in the dopamine 33 solution, which resulted in the spontaneous formation of a 34 self-polymerization layer on the specimens. As shown in Fig. 35 1b, specimen TiD was smoother than the Ti substrate with a 36 covering of poly(dopamine). Compared with the pristine 37 porous structure prior to dopamine treatment, a number of 38 granular lumps comprising dopamine molecules can be seen <sup>39</sup> on the surface (Figs. 1d and f, indicated by white arrow).

The topographies of the CaPD and SrCaPD specimens are similar to that of the CaP coating, suggesting that the poly(dopamine) formed an ultra-thin layer with little to no a effect on the three-dimensional structure beneath. These results are consistent with several previous studies that reported the homogeneous coating of poly(dopamine) across the surface.<sup>20, 22, 23</sup> Under alkaline conditions, dopamine rolecules are believed to undergo adsorption and selfpolymerization simultaneously, which involves the oxidation of catechol to quinone, which then react with amines and o other catechols/quinones to form an adherent polymer.<sup>24</sup> As a <sup>51</sup> result, dopamine may provide strong covalent as well as <sup>52</sup> noncovalent interactions with a variety of substrates without <sup>53</sup> noticeably altering the topography.

<sup>55</sup> **Figure 1.** Surface micrographs of specimens without and <sup>56</sup> with added dopamine: (a) Ti; (b) TiD; (c) CaP; (d) CaPD; (e)



<sup>57</sup> SrCaP; (f) SrCaPD. The insets in (d)-(g) are highly magnified
<sup>58</sup> images obtained at a tilt of 30°. White arrows indicate
<sup>59</sup> granular lumps which comprising dopamine molecules.

The surface hydrophilicity of biomaterials has been for the surface hydrophilicity of biomaterials has been for shown to be an important factor influencing cell responses.<sup>25</sup> for this study we used static water contact angle to evaluate for the wettability of the specimens (Fig. 2). The static contact for angle of droplets on as-polished titanium was  $52.9^{\circ} \pm 5.6^{\circ}$ . for Following the MAO process, the water contact angle was for approximately 10°, due to the hydrophilic nature of the for coating. After being coated with poly(dopamine), the two for MAO coatings differed only slightly, with an increase in ro contact angle of approximately 15°.

The MAO process with dopamine modification resulted r2 in a surface with relatively hydrophilic properties. The r3 hydrophilic nature of titanium surfaces has been shown to r4 enhance protein adsorption, and thus promote the r5 proliferation and differentiation of osteoblastic cells, and r6 ultimately the strength of bone–implant integration.<sup>26</sup> In our r7 previous study, the accumulation of adsorbed serum protein r8 corresponded to an increase in DOPA content.<sup>21</sup> It is r9 reasonable to speculate that a poly(dopamine) layer could be s0 used to interconnect serum proteins and modulate cell s1 attachment.<sup>27</sup> Journal Name



Figure 2. Water contact angle of all specimens in the study:
(a) Ti; (b) TiD; (c) CaP; (d) CaPD; (e) SrCaP; (f) SrCaPD.

4 XPS analysis was used to investigate the surface 5 chemistry in order to confirm the effectiveness of the coating 6 process. Figure 3a (left) shows the XPS spectra of Ti, CaP, 7 and SrCaP specimens, respectively. The survey spectra of the 8 CaP specimen reveals peaks associated with Ti2p, O1s, C1s, 9 Ca2p, and P2s. The addition of strontium to the electrolyte 10 during MAO resulted in peaks associated with the presence 11 of Sr3p1/2 and Sr3p3/2, confirming the incorporation of 12 strontium into the oxide layer (Fig. 3b).

To ensure that the control group had the same ion release the characteristics as the test samples, CaP and SrCaP specimens to were also immersed in the Tris-HCl solution without the dopamine. Our results suggest that differences in the trontium content could be attributed to the thickness of the the poly(dopamine) coating. Following the coating process, we the observed a nitrogen signal from the amino groups of the presence of such a nitrogen peak is an indication that the the poly(dopamine) coating process was successful.

23 Table 2 presents a summary of the surface chemical 24 composition obtained using XPS analysis. The nitrogen 25 content for specimens TiD, CaPD, and SrCaPD was 7.9%, 26 7.9%, and 7.5%, respectively, indicating that the surface had 27 indeed been coated with poly(dopamine) film, with only a 28 negligible effect on the surface topography. Further, the 29 composition of the substrate did not appear to have a 30 significant effect on the amount of poly(dopamine) adhered 31 to the substrate. The relative atomic concentration of C was 32 18.0% (TiD), 18.9% (CaPD), and 14.7 % (SrCaPD). After the 33 specimens were coated with a poly(dopamine) film, the <sup>34</sup> relative atomic concentrations of C increased to 68.4% (TiD), 35 68.3% (CaPD), and 66.7% (SrCaPD). The dopamine 36 molecule contains amide and benzyl functional groups; 37 therefore, the N1s and C1s signals in the XPS spectra can be

<sup>38</sup> attributed to the poly(dopamine) film on the MAO coatings. <sup>39</sup> This also explains why the CaPD and SrCaPD specimens <sup>40</sup> presented increased N/C ratios. The N/C ratios of the three <sup>41</sup> specimens were close to the theoretical values of the <sup>42</sup> dopamine molecule (0.125).<sup>17</sup> Analysis of surface chemical <sup>43</sup> composition suggests that the strontium was incorporated <sup>44</sup> within the coatings and that a film of poly(dopamine) had <sup>45</sup> been successfully attached to the specimens.



<sup>47</sup> Figure 3. (a) Broad-range XPS spectra of specimens; (b)
<sup>48</sup> high resolution Sr3p spectra of specimens; (c) high resolution
<sup>49</sup> N1s spectra of specimens

## 51 3.2 Cell morphologies and cytoskeleton on poly(dopamine)52 film over MAO coatings

<sup>53</sup> Understanding how osteoblasts interact with artificial <sup>54</sup> materials is of crucial importance. The behaviour of <sup>55</sup> osteoblast cells on the six specimens in this study was <sup>56</sup> systematically examined in terms of initial cell spreading, <sup>57</sup> cytoskeletal organization, proliferation, ALP activity, and <sup>58</sup> bone-related gene expression. Cytoskeletal reorganization is <sup>59</sup> particularly important in cell attachment, proliferation, and <sup>60</sup> differentiation, all of which are essential to the initial success <sup>61</sup> of an implant.<sup>28, 29</sup> In this study, we seeded MC3T3-E1 cells <sup>62</sup> on each specimens to examine the cell-material interactions, <sup>63</sup> using a fluorescence microscope was used to provide an <sup>64</sup> accurate indication of cell spreading and cytoskeletal <sup>65</sup> reorganization.

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Table 2. Surface element composition of MAO coatings, as determined by XPS									
Specimen	Atomic %								
	Ti2p	O1s	C1s	Ca2p	P2p	Sr3p	N1s	N/C	
Ti	16.3	56.2	27.6						
TiD	0.3	17.7	74.1				7.9	0.11	
CaP	4.4	51.4	18.0	14.1	12.1				
CaPD	0.6	20.0	68.4	1.6	1.5		7.9	0.12	
SrCaP	6.4	46.8	18.0	9.6	16.0	1.6			
SrCaPD	0.6	20.2	68.3	1.3	2.0	0.1	7.6	0.11	

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MC3T3-E1 showed the characteristic pattern of complex 4 5 focal adhesion with a dense phalloidin stained f-actin (green) 6 surrounded by vinculin (red) staining at the leading edge of 7 cells (Figs. 4 and 5). Figure 4 presents the morphologies of 8 the cells grown on specimens for 3 h. Clearly, the MC3T3-E1 9 cells rapidly attached to the surface of all of the specimens; 10 however, the attachment rates varied according to the features 11 of the specimens. The MC3T3-E1 cells appeared to have 12 spread well and formed focal adhesion to the flat Ti 13 specimen. Cells cultured on TiD specimens showed 14 spreading over a greater area than on the Ti specimens (Figs. 15 4a and b). Cells cultured on the CaP specimen presented no 16 evidence of stress fibers, which is an indication of limited 17 interaction with the substrate (Fig. 4c).

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19 Figure 4. Fluorescence images of MC3T3-E1 cells cultured 20 on various specimens after 3h of incubation: (a) Ti; (b) TiD; 21 (c) CaP; (d) CaPD; (e) SrCaP; (f) SrCaPD. F-actin (green) 22 was stained with Alexa Flour 488 phalloidin, vinculin (red) 23 was stained with Alexa Fluor 594, and the nucleus (blue) was 24 stained with DAPI.

25 Cells covered a greater area of the SrCaP specimens than the 26 CaP specimens (Fig. 4e). The presence of poly(dopamine) 27 film appears to have promoted cell attachment on the CaPD 28 and SrCaPD specimens, resulting in long extensions of 29 cytoplasmic membranes, which spread out to assume a fully 30 flattened shape (Figs. 4d and f). The shape and morphology 31 of the cells indicated that poly(dopamine) film promoted the 32 attachment and spreading of osteoblastic cells.

Following an incubation period of 24 h, we observed the 33 34 expansion of actin filaments in cell cultures grown on 35 poly(dopamine) modified specimens. The filaments exhibited 36 a flattened polygonal morphology spread over a wider area 37 (Figs. 5b, d, and f). Compared to cells cultured on 38 unmodified specimens, those cultured on poly(dopamine)-39 coated surfaces presented well-stretched actin bundles and 40 highly organized actin stress fibers, which is a clear 41 indication of strong cell adhesion at the interface. The 42 poly(dopamine) film was also found to induce the extension 43 of lamellipodia on SrCaPD specimens (Fig. 5f). A number of 44 previous studies have reported that the roughness of 45 substrates strongly influences MC3T3-E1 cell adhesion; 46 however, in those studies, rough surfaces resulted in less 47 proliferation and adhesion than did smooth surfaces.30, 31 48 This difference in surface topography may explain why the 49 cells cultured on Ti specimens exhibited attachment 50 characteristics superior to those cultured on CaP specimens. 51 In contrast, cells cultured on CaPD and SrCaPD presented 52 expansion characteristics far exceeding those on the CaP and 53 SrCaP specimens, which suggests that the poly(dopamine) 54 film had improved MC3T3-E1 cell adhesion, particularly on 55 samples with a more porous structure.

Controlling cell adhesion is crucial in this type of 57 organic-inorganic system, because adhesion provides a 58 physical link to the environment and influences all of the 59 major cell fate decisions, including cellular signal 60 transduction, differentiation, and gene expression.<sup>32</sup> A higher 61 density of actin stress fibres in the cytoskeleton of a cell 62 indicates a firmer, stronger link between the cells and 63 substrate. This, in turn, indicates that a given biomaterial is 64 suitable for cell attachment and the preservation of biological 65 functions.<sup>33</sup> As previously reported, bioactive functional 66 groups, such as the hydroxyl and amine groups that form 67 dopamine, may be beneficial to the process of cell 68 attachment. In this study, dopamine appears to have played a

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- 1 major role in this process. It appears that combining 2 dopamine and strontium ions significantly enhances initial
- 3 cell attachment, thereby strengthening the link between cells
- 4 and MAO coatings.



<sup>6</sup> Figure 5. Fluorescence images of MC3T3-E1 cells cultured
<sup>7</sup> on various specimens after 24 h of incubation: (a) Ti; (b)
<sup>8</sup> TiD; (c) CaP; (d) CaPD; (e) SrCaP; (f) SrCaPD. F-actin
<sup>9</sup> (green) was stained with Alexa Fluor 488 phalloidin, vinculin
<sup>10</sup> (red) was stained with Alexa Fluor 594, and the nucleus
<sup>11</sup> (blue) was stained with DAPI.

## 13 3.3 Cell proliferation and differentiation on MAO coatings14 further coated with strontium and dopamine

Figure 6 presents the number of MC3T3-E1 cells on the surface of each specimen after culturing for 1, 3, and 7 days, respectively. All of the specimens presented an increase in the number of cells, indicating that the MC3T3-E1 cells proliferated continuously throughout the 7 days. After days 1, 3, and 7, the number of cells on the TiD, CaPD and SrCaPD provide the specimens. Nonetheless, after 7 days, the number of cells on pure titanium was significantly higher than on specimens of ACaP and SrCaP. The number of cells on the CaPD and 25 SrCaPD specimens exceeded the numbers observed on any26 samples without poly(dopamine) modification.

ALP is a marker of early stage osteoblast differentiation, which undergoes a sequence of processes eventually resulting in the formation of bone. Figure 7 presents the ALP activity of MC3T3-E1 cell on the surface of each specimen over the 17-day incubation period. Previously, the presence of strontium ions was reported to be advantageous to cell differentiation.<sup>14</sup> In this study, the ALP activity of MC3T3-E1 cells on SrCaP and SrCaPD specimens was significantly more pronounced than that observed on CaP and CaPD specimens, indicating that the incorporation of strontium may prove the initiation of osteogenic differentiation in MC3T3-E1 cells.



39

40 Figure 6. MTT cell proliferation assay results for MC3T3-E1
41 cells after culturing on various specimens for 1, 3, and 7
42 days: (a) Measurements from the MTT cell proliferation
43 assay, (b) Duncan grouping of the 1-day MTT assay, (c)
44 Duncan grouping of the 3-day MTT assay, and (d) Duncan
45 grouping of the 7-day MTT assay

46 47 1.2

Ti

TiD

CaP

CaPD

SrCaP

SrCaPD

CaPD

TiD

Ti

(a)

Normalized ALP activity

(b)

4 activity.

5

6

(μ**mol p-NP/ μg protein)** <sub>70</sub> 80

0.

Journal Name

SrCaP



7 measured the expression of osteogenic marker genes 8 including integrin-binding sialoprotein (IBSP), alkaline 9 phosphatase (ALP), osteopontin (OPN) and osteocalcin 10 (OCN) using quantitative reverse transcription PCR. Our 11 results show that the SrCaP and SrCaPD specimens promoted 12 bone-related gene expression, such as IBSP, ALP, OPN, and 13 OCN (Fig. 8). Strontium-based artificial materials, including 14 strontium-substituted hydroxyapatite cement and bioactive 15 glasses, have been attracting interest due to the appearance of 16 a link between strontium and the stimulation of osteoblast 17 differentiation and the inhibition of osteoclast formation and 18 resorption in vitro. The results indicate that strontium 19 enhances cell differentiation, as evidenced by the up-20 regulation of bone-related mRNA.<sup>14</sup>

**SrCaP** 

SrCaPD

The surface chemistry and topography of materials can 21 22 influence the response of osteoblastic cells. Surfaces with a 23 moderately rough microstructure, such as MAO, have been 24 shown to enhance the maintenance of implant stability.<sup>34, 35</sup> 25 Several studies have demonstrated that strontium can increase 26 the quality of bone structure and promote bone ingrowth, 27 thereby accelerating implant osseointegration.<sup>36, 37</sup> The 28 SrCaPD coating altered the topography as well as the surface 29 chemistry. The resulting increase in cell adhesion and ALP 30 activity are in good agreement with previous reports. Our 31 results suggest that dopamine and strontium both play 32 important roles in stimulating the proliferation and 33 differentiation of MC3T3-E1 cells. These effects are most 34 pronounced on the SrCaPD sample with the best cell 35 attachment, proliferation, and differentiation of MC3T3-E1 36 cells.

SrCaP SrCaPD SrCaPD SrCaP

<sup>38</sup> Figure 8. Bone-related gene expression of osteoblast cells 39 cultured on various specimens in terms of (a) IBSP, (b) ALP, 40 (c) OPN, and (d) OCN.

CaP Ti TiD CaPD

Ti

TID

CaP CaPD

SrCaP

SrCaPD

TiD CaPD Ti CaP

(d)

**OCN** gene expression

fold change)

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37

#### 1 4. Conclusions

This study developed an innovative coating for titanium surfaces combining MAO with bio-inspired surface modification. The proposed method creates an excellent three-dimensional structure, the topography of which is in no way jeopardized by the application of a bioactive layer to promote cell attachment and cytoskeletal development. Moreover, the incorporation of strontium-ions in the MAO ocating was shown to greatly induce osteoblasts cell differentiation. This novel dual-setting material provides many of the benefits anticipated in the next generation of biomodified implants.

13

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#### **19 Notes and references**

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- 27
- L. Zhu, X. Ye, G. Tang, N. Zhao, Y. Gong, Y. Zhao, J. Zhao and
   X. Zhang, J. Biomed. Mater. Res., Part A, 2006, 78A, 515-522.
- X. Zhao, G. Wang, H. Zheng, Z. Lu, X. Zhong, X. Cheng and H.
   Zreiqat, ACS Appl. Mater. Interfaces, 2013, 5, 8203-8209.
- 32 3. K. P. Ananth, S. Suganya, D. Mangalaraj, J. M. Ferreira and A.
   Balamurugan, *Mater. Sci. Eng.*, *C*, 2013, 33, 4160-4166.
- 34 4. E. R. Urquia Edreira, J. G. C. Wolke, A. A. Aldosari, S. S. Al-
- Johany, S. Anil, J. A. Jansen and J. J. J. P. van den Beucken, J.
   *Biomed. Mater. Res., Part A*, 2014, DOI: 10.1002/jbm.a.35173.
- 37 5. C. W. Yang and T. S. Lui, Mater. Trans., 2007, 48, 211-218.
- 38 6. L. H. Li, Y. M. Kong, H. W. Kim, Y. W. Kim, H. E. Kim, S. J.
  39 Heo and J. Y. Koak, *Biomaterials*, 2004, 25, 2867-2875.
- 40 7. M. Degidi, D. Nardi and A. Piattelli, *Clin Implant Dent Relat Res*,
  41 2012, 14, 828-838.
- 42 8. C. Ma, A. Nagai, Y. Yamazaki, T. Toyama, Y. Tsutsumi, T.
- Hanawa, W. Wang and K. Yamashita, *Acta Biomater.*, 2012, 8,
  860-865.
- 45 9. Y. Han, D. Chen, J. Sun, Y. Zhang and K. Xu, *Acta Biomater.*,
  2008, 4, 1518-1529.
- 47 10. W. H. Song, H. S. Ryu and S. H. Hong, J. Biomed. Mater. Res.,
   48 Part A, 2009, 88A, 246-254.
- <sup>49</sup> 11. Y. T. Liu, K. C. Kung, T. M. Lee and T. S. Lui, *J. Alloys Compd.*,
  <sup>50</sup> 2013, **581**, 459-467.
- 51 12. G. X. Ni, W. W. Lu, K. Y. Chiu, Z. Y. Li, D. Y. T. Fong and K. D.
   52 K. Luk, J. Biomed. Mater. Res., Part B, 2006, 77B, 409-415.
- 53 13. Y. W. Li, J. C. Y. Leong, W. W. Lu, K. D. K. Luk, K. M. C.
- 54 Cheung, K. Y. Chiu and S. P. Chow, J. Biomed. Mater. Res., 2000,
- **55 52**, 164-170.

- 56 14. C. Capuccini, P. Torricelli, F. Sima, E. Boanini, C. Ristoscu, B.
  57 Bracci, G. Socol, M. Fini, I. N. Mihailescu and A. Bigi, *Acta Biomater.*, 2008, 4, 1885-1893.
- <sup>59</sup> 15. Y. Zhang, L. Wei, J. Chang, R. J. Miron, B. Shi, S. Yi and C. Wu,
   *J. Mater. Chem. B*, 2013, 1, 5711-5722.
- 61 16. K. C. Kung, T. M. Lee, J. L. Chen and T. S. Lui, *Surf. Coat.*62 *Technol.*, 2010, **205**, 1714-1722.
- 63 17. H. Lee, S. M. Dellatore, W. M. Miller and P. B. Messersmith,
   *Science*, 2007, **318**, 426-430.
- 65 18. S. Saidin, P. Chevallier, M. R. Abdul Kadir, H. Hermawan and D.
  Mantovani, *Mater. Sci. Eng.*, *C*, 2013, **33**, 4715-4724.
- 67 19. T. Liu, Z. Zeng, Y. Liu, J. Wang, M. F. Maitz, Y. Wang, S. Liu, J.
  68 Chen and N. Huang, *ACS Appl. Mater. Interfaces*, 2014, DOI:
  69 10.1021/am5015309.
- 70 20. C. Y. Chien, T. Y. Liu, W. H. Kuo, M. J. Wang and W. B. Tsai, J. *Biomed. Mater. Res., Part A*, 2013, **101A**, 740-747.
- 72 21. Y. T. Liu, T. M. Lee and T. S. Lui, *Colloids Surf.*, B, 2013, 106, 37-45.
- 74 22. N. G. Rim, S. J. Kim, Y. M. Shin, I. Jun, D. W. Lim, J. H. Park
  and H. Shin, *Colloids Surf.*, *B*, 2012, 91, 189-197.
- 76 23. W. Wang, R. Li, M. Tian, L. Liu, H. Zou, X. Zhao and L. Zhang,
   77 ACS Appl. Mater. Interfaces, 2013, 5, 2062-2069.
- 78 24. R. A. Zangmeister, T. A. Morris and M. J. Tarlov, *Langmuir*,
  2013, 29, 8619-8628.
- 80 25. P. Thevenot, W. Hu and L. Tang, *Curr. Top. Med. Chem.*, 2008, 8, 270-280.
- 82 26. J. H. Park, C. E. Wasilewski, N. Almodovar, R. OlivaresNavarrete, B. D. Boyan, R. Tannenbaum and Z. Schwartz, *Biomaterials*, 2012, 33, 7386-7393.
- 85 27. Y. Ding, Z. Yang, C. W. C. Bi, M. Yang, J. Zhang, S. L. Xu, X.
  Lu, N. Huang, P. Huang and Y. Leng, *J. Mater. Chem. B*, 2014,
  DOI: 10.1039/C4TB00386A.
- 88 28. M. Ghibaudo, A. Saez, L. Trichet, A. Xayaphoummine, J.
  Browaeys, P. Silberzan, A. Buguin and B. Ladoux, *Soft Matter*,
  2008, 4, 1836-1843.
- 91 29. K. Anselme, Biomaterials, 2000, 21, 667-681.
- 92 30. K. Anselme, P. Linez, M. Bigerelle, D. Le Maguer, A. Le Maguer,
  93 P. Hardouin, H. F. Hildebrand, A. Iost and J. M. Leroy,
  94 *Biomaterials*, 2000, 21, 1567-1577.
- 95 31. K. Anselme, M. Bigerelle, B. Noel, E. Dufresne, D. Judas, A. Iost
  and P. Hardouin, *J. Biomed. Mater. Res.*, 2000, 49, 155-166.
- 97 32. O. Guillame-Gentil, O. Semenov, A. S. Roca, T. Groth, R. Zahn, J.
  Vörös and M. Zenobi-Wong, *Adv. Mater.*, 2010, 22, 5443-5462.
- <sup>99</sup> 33. W. Thein-Han and H. H. K. Xu, *Tissue Eng.*, *Part A*, 2011, 17, 2943-2954.
- 101 34. W. Zechner, S. Tangl, G. Fürst, G. Tepper, U. Thams, G. Mailath
  and G. Watzek, *Clin Oral Implants Res*, 2003, 14, 150-157.
- 103 35. A. Rocci, M. Martignoni, P. M. Burgos, J. Gottlow and L.
  Sennerby, *Clin Implant Dent Relat Res*, 2003, 5, 88-98.
- 105 36. O. Z. Andersen, V. Offermanns, M. Sillassen, K. P. Almtoft, I. H.
  106 Andersen, S. Sørensen, C. S. Jeppesen, D. C. E. Kraft, J. Bøttiger,
  107 M. Rasse, F. Kloss and M. Foss, *Biomaterials*, 2013, 34, 5883108 5890.
- 109 37. L. Maïmoun, T. C. Brennan, I. Badoud, V. Dubois-Ferriere, R.
  Rizzoli and P. Ammann, *Bone*, 2010, 46, 1436-1441.



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