Journal of Materials Chemistry A

Accepted Manuscript

This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this *Accepted Manuscript* with the edited and formatted *Advance Article* as soon as it is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](http://www.rsc.org/Publishing/Journals/guidelines/AuthorGuidelines/JournalPolicy/accepted_manuscripts.asp).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](http://www.rsc.org/help/termsconditions.asp) and the Ethical quidelines still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

www.rsc.org/materialsA

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxxx

ARTICLE TYPE

In situ **Preparation of Novel Organo-Inorganic (6, 13- Pentacenequinone: TiO2) coupled Semiconductor Nanosystem: A new Visible Light active Photocatalyst for Hydrogen Generation**

Vikram Pandit,^a Sudhir Arbuj,^a Ranjit Hawaldar,^a Pradnya Kshirsagar,^a Uttam Mulik,^aSuresh Gosavi,^b *s* **Chan-Jin Park^{*},^c Bharat Kale^{*}^{, a}**

Received (in XXX, XXX) XthXXXXXXXXX 20XX, Accepted Xth XXXXXXXXX 20XX **DOI: 10.1039/b000000x**

Studies related to synthesis of stable UV-Visible light active photocatalyst for hydrogen generation has ¹⁰been limited to only inorganic semiconductors, their nanostructures and hetero-structures. Hence, in the present investigation, organo-inorganic $(6, 13)$ - Pentacenequinone (PQ) : TiO₂) coupled semiconductor nanosystem has been demonstrated as an efficient visible light active photocatalyst for hydrogen production for the first time. The 3-5 nm anatase $TiO₂$ nanoparticles have been uniformly decorated on thin sheets of monoclinic PQ by in-situ solvothermal method. These as prepared PQ: $TiO₂$ coupled

¹⁵semiconductor nanosystems exhibited band gap in the range of 2.7-2.8 eV. The strong emission observed at 590 nm can be attributed to transfer of electrons from the LUMO energy level of $TiO₂$ combine with the holes present at HOMO level of PQ. Furthermore, this electron-hole recombination makes availability of electrons and holes in LUMO of PQ and HOMO of $TiO₂$, respectively. Hence, this hybrid semiconductor coupled nanosystem conferred utmost hydrogen evolution i.e. 36,456 μ mol/h/g from H₂S

20 under UV-visible light which is four times higher than $TiO₂$ as well as earlier report of UV-visible light active photocatalyst. The enhanced activity is obtained for the PQ: TiO₂nanosystem has been discussed, thoroughly. Our results open up a new path to explore various inorganic systems coupled with PQ as a new photoactive hybrid catalyst for various chemical and physicochemical processes.

25

Introduction

Nanomaterial's have aroused considerable interest due to their fascinating size-dependent optoelectronic, magnetic, chemical, 30 and other physical properties.¹⁻³ Now, it is renowned that for the fabrication of next-generation optoelectronic devices and for fulfilling the energy demand, the use of nanoscale semiconductor nanostructures are absolutely essential. $4-6$ In this context, great efforts have been dedicated by researchers for the synthesis of ³⁵multi-functional nanomaterial's. Nanostructured semiconductor materials are known to act as efficient photocatalyst due to their high surface and different morphologies.^{7,8} Numerous strategies have been employed to achieve efficient photocatalysts by using surface-tuning strategies for synthesis of various oxides at 40 nanoscale.⁹ In the past decade, though there has been noteworthy progress in the field of photocatalysis¹⁰ but most of it has been limited due to use of ultraviolet (UV) light instead of most useful visible–light region. Consequently, scientific and engineering interests in semiconductor photocatalysis have grown

 45 significantly in researchers.¹⁰ The researchers have explored visible light (renewable source) active photocatalyst for hydrogen production, which is essential for economical hydrogen production.^{10, 11} Hydrogen has been recognized as a future fuel and capable to meet the global energy demand. Using the 50 photocatalysis process, the H_2 can be produced from H_2O as well as H_2S , economically. Since, H_2S is abundantly available as a waste gas from oil refineries and alkali chemical industries, presently, the ubiquitous Claus process is used for $H₂S$ utilization to produce sulphur instead of hydrogen. Hence, photocatalytic hydrogen production from H2S has immense importance**.** 55 Considering the importance of hydrogen as a future fuel, hydrogen production by photocatalytic process using UV-Visible light has immense importance. Photocatalytic hydrogen production from hydrogen sulphide and water are economically ⁶⁰more viable as compared to conventional hydrogen production methods such as steam reforming, electrolysis etc.⁸⁻¹² Hence, we have utilized the abundant waste $H₂S$ to produce eco-friendly hydrogen under UV-Visible light^{8,13-16} Few reports are available on visible light active photocatalyst for hydrogen production from H2S. The main efforts were taken to develop the visible light active photocatalyst. In this context, metal oxide semiconductors have been doped with transition metal as well as anions for

- $\frac{1}{2}$ s narrowing the band gap suitable for UV-Visible light.¹⁷⁻²³ Considering the $TiO₂$ as one of the best stable, eco-friendly and economical photocatalyst, massive work has been performed on doping of $TiO₂$ with cations and anions for narrowing the bandgap.^{6,9} Recently, $TiO₂$ graphene nanocomposites and
- 10 decoration of Ag, Au, and Pt on TiO₂ have also been reported as a visible light active photocatalyst.^{11, $24-34$}

In this article, considering the good stability of $TiO₂$ and our earlier report of 6, 13- Pentacenequinone (PQ) organic semiconductor as a best photocatalyst for energy harvesting and

- 15 environmental remediation, we have investigated efficient organo-inorganic hybrid (PQ: TiO₂) nanosystem as a visible light active photocatalyst for hydrogen production for the first time.¹⁶ The PQ has good thermal stability and band gap around 2.8 eV. It has also simple synthesis procedure along with higher molar
- ²⁰absorption coefficient. This makes PQ as an efficient visible light active photocatalyst. The in-situ preparation of novel organoinorganic $(PQ: TiO₂)$ coupled semiconductor nanosystem has been investigated using facile solvothermal method. The interesting optical properties of this organio–inorganic coupled ²⁵nanosystem have been discussed and justified the enhanced
- photocatalytic activity. During the solvothermal preparation of $TiO₂$, PQ has been
- introduced which confer PQ : TiO₂ hybrid nanosystem. The PQ : $TiO₂$ nanosystems are prepared using 5, 10 and 17 mmol of
- 30 titanium tetra-isopropoxide precursor and termed as PT-5, PT-10 and PT-17, respectively, throughout the manuscript. This unique nanosystem has been characterized further for phase identification.

Results and Discussion

 35 These as prepared PQ: TiO₂ nanosystems were characterized by X-ray diffractometry (XRD) and depicted in Figure 1. XRD patterns pertaining to pure PQ and $TiO₂$ are also shown in Figure 1a, e).

Figure 1.XRD patterns of PQ: TiO₂ hybrid nanosystem where a) PQ, b) 55 PT-5, c) PT-10, d) PT-17 and e) $TiO₂$

In the XRD patterns of nanosystems (Figure 1b-d), the diffraction

peaks at $2\theta = 25.3$, 37.8, 48, 53.9 and 55° can be indexed as (101), (004), (200), (105) and (211) corresponds to anatase $TiO₂$ (JCPDS No. 21-1272). The diffraction peaks at $2\theta = 14.7, 23.6$ 60 and 27.7° represent (012), (112) and (104) planes which indicates the existence of crystalline monoclinic phase of PQ (JCPDS No.47-2123). The broadening of the XRD peaks of $TiO₂$ clearly shows nanocrystalline nature of the material. The crystallite size of $TiO₂$ calculated in PT-5, PT-10 and PT-17 hybrid nanosystem ⁶⁵using Scherrer's formula and was observed to be 5.8, 6.6 and 7.5 nm, respectively. However, with increase in $TiO₂$ concentration, the overall intensity of the other $TiO₂$ peaks also increased slightly with retaining the broadness. The difference is quite marginal and difficult to compare visibly because the crystallite π ⁰ size difference is not much (\sim 1 nm). However, it has been also studied using SAED pattern and discussed in morphological study. The existence of monoclinic PQ and anatase $TiO₂$ in the XRD clearly indicates the formation of organo-inorganic (PQ: TiO²) hybrid nanosystem.

Figure 2.FESEM of PQ: TiO₂ hybrid nanosystem (a, b) PQ, (c, d) PT-5, (e, f) PT-10, and (g, h) PT-17

FESEM images of as prepared PQ: $TiO₂$ hybrid nanosystems i.e. ⁸⁰PT-5, PT-10 and PT-17 are shown in Figure 2. The pristine PQ observed in the form of micron-sized sheets with smooth surface having thickness in the range of 50 to 200 nm (Figure 2a, b). FESEM images of PT-5 nanosystem indicate the formation of nanosized $TiO₂$ particles having size \leq 5 nm on the surface of PQ ⁸⁵sheets (Figure 2c, d). Whereas, PT-10 and PT-17 hybrid

50

55

nanosystems showed 5-7 and 8-10nm sized $TiO₂$ particles uniformly layered on PQ sheets (Figure 2e-h). All the three prepared hybrid nanosystems depict the same morphology having nanosized $TiO₂$ particles on the surface of PQ sheets. More

- ⁵FESEM images with high resolution are given in supporting information (Fig. S1). It is also observed that the density of the nanoparticles increases from PT-5 to PT-17 which is quite obvious because of the increase in the $TiO₂$ concentration. Moreover, the $TiO₂$ nanoparticle clusters were also seen
- 10 sandwiched between the PQ sheets. Overall, FESEM images clearly show the uniform distribution of the $TiO₂$ nanoparticles on the PQ sheet for all the prepared nanosystems. The TEM images with SAED pattern of prepared PQ: $TiO₂$
- hybrid nanosystems are depicted in Figure 3. The hybrid PT-5
- ¹⁵nanosystem (Figure 3a, b) confirms the formation of spherical shaped $TiO₂$ nanoparticles having size in the range of 3-5 nm (inset of Figure 3b). Also, TEM images of PT-10 and PT-17 hybrid nanosystems validates the existence of $TiO₂$ nanoparticles having sizes 6-8 and 9-10 nm, which are evenly distributed on PQ
- ²⁰sheets (Figure 3c-f). High resolution TEM images are depicted in supporting information indicating size and shape of $TiO₂$ particles (Fig. S2).

Figure 3.TEM images of PQ: $TiO₂$ nanosystem (a, b) PT-5, (c, d) PT-10, and (e, f) PT-17, inset of Figure 3(b, d, and f) indicates the HRTEM and SAED pattern of PQ: TiO2.

 The well-defined lattice fringes are seen in the HRTEM ³⁰images (as a inset of Figure 3b, d& f) and the inter-planer

distance between the adjacent planes was observed to be 0.34nm which corresponds to 101 plane of anatase $TiO₂$, thereby confirming the formation of anatase phase of $TiO₂$ in all the three hybrid PQ: $TiO₂$ nanosystems. The SAED pattern (inset of Figure ³⁵3b, d, and f) also shows d spacing of 0.24 and 0.34 nm attributed to (103) and (101) reflection planes matching with anatase $TiO₂$ in all three nanosystems, respectively. The ED pattern clearly shows that the intensity of bright rings increases from PT-5 to PT-17 indicates slight increase in crystallinity which was very

- ⁴⁰imprecise to see from XRD. The HRTEM clearly depicts the perfect interface between $TiO₂$ nanoparticles and PQ nanosheets. This interface could provide the proper channel for electrons and holes transfer between PO and TiO₂ semiconductors. The detail study about the electron transfer is given in optical study section.
- ⁴⁵The diffuse reflectance UV-Visible absorbance spectra of PQ loaded $TiO₂$ are depicted in Figure 4. Pristine PQ shows absorbance edge at 449 nm indicating band gap around 2.8 eV.

Figure 4.Diffuse reflectance UV-Visible absorbance spectra of PQ: TiO₂ hybrid nanostructures a) PQ, b) PT-5, c) PT-10 and d) PT-17

- ⁶⁰The nanosystem PT-5 shows absorption edge at 451nm having band gap of 2.74 eV, however, PT-10 and PT-17 nanosystem shows two distinct absorption edges at 380 nm (band gap 3.26 eV), and 455 nm (band gap 2.73 eV), corresponds to $TiO₂$ and PQ, respectively. The higher band gap of $TiO₂$ in PT-10 and PT-
- ⁶⁵17 are due to blue shift in the absorbance as a function of particle size. The band gap values are verified using Tauc's plot (Supporting information Fig. S3). This confirms the formation of coupled hybrid nanosystem of PQ: TiO₂ (Figure 4c, d).

 The increase in absorbance in the range of 370 to 400 nm π_0 attributed to increased concentration of TiO₂ in prepared nanosystem. In case of sample PT-5, we observed clear single absorption peak due to higher concentration of PQ. Overall, the absorbance peak in the range of 400 to 460 nm corresponds to PQ and 370 to 400 nm corresponds to $TiO₂$ in PT-10 and PT-17 ⁷⁵hybrid nanosystem.

 The photoluminescence spectra are often employed to study surface processes involving the electron-hole fate of the semiconductors. Figure 5 represents the photoluminescence spectra of hybrid nanosystems PT-5, PT-10, PT-17 along with δ ⁸⁰ pure PQ and TiO₂, respectively. The prepared nanosystem show strong emission peaks at 550 nm with satellite peak at 590 nm. Increase in PL intensity as a function of $TiO₂$ concentration is observed.35-38

Figure 5.Photoluminescence spectra of PQ: TiO₂ hybrid nanosystem a) PQ, b) PT-5, c) PT-10 and d) PT-17

 The peak at 550nm might be due to the certain defects generated because of oxygen vacancies. Such defects acts as self ¹⁵trap centres for the excitons recombination owing to combine effect of defects canters generated from oxygen vacancies and lower particle size. PT-5 is showing slightly higher PL emission at 550 nm compared to PT-10, this might be because of the lower particles size of $TiO₂$ which leads the self-trapped exciton

²⁰recombination as discussed earlier. *Lin et al* reported that the emission peaks at 550nm is attributed to electronic transitions involving traps states.⁴⁰ A report by *Zhang et.al*. on the photoluminescence study of anatase $TiO₂$ nanoparticle is in good agreement with our results. 32 The satellite peak at 590 nm ²⁵might be due to electron-hole recombination.

 For simplicity, we studied the HOMO and LUMO energy levels of PQ and $TiO₂$. Figure 6 shows the band diagram of coupled nanosystems.

Figure 6.Schematic band structures of PQ: TiO₂ hybrid nanosystem

The LUMO energy levels of PQ and $TiO₂$ have values of 3.29 and 4.20 eV, whereas the HOMO level values are 6. 29 and 7.40 eV, respectively.^{16, 39} Upon excitation with 350 nm light, the electron-hole pairs were generated in both the semiconductors, PQ and TiO² ³⁵. Due to coupling of these two semiconductors, the

electron from LUMO energy level of PQ transfer to the LUMO of $TiO₂$ which may later on combine with the hole present in the HOMO level of PQ. This electron-hole recombination might be responsible for the emission peak at 590 nm confirmed by PL

In the present nanosystem, spherical nanoparticles of $TiO₂$ with 3-5 nm (PT-5) size are intact on plate like structure of PQ (organic semiconductor); hence, the generated electron-hole pairs can be easily moved across the organic-inorganic interface.

⁴⁵Reports are also available mentioning that the emission peak at 550 nm may be due to mid trap states.³⁴ The pristine PQ indicates the emission peaks at 408, 432 and 574 nm, the peak at lower wavelength are due to π to π^* and n to π^* whereas the emission peak at 574 nm might be due to certain mid trap centers. The ⁵⁰enhance PL intensity in the prepared hybrid system is result of proper electron transfer between organic-inorganic semiconductor which could inhibits the non-radiative electron-hole pair recombination's. The observed results are in agreement with the reported one wherein for the hetero-systems of $TiO₂$, with particle 55 size less than 10 nm shows intense yellow-green emissions. $37-39$

Photocatalytic study

Photocatalytic study was performed using synthesized PQ: $TiO₂$ hybrid nanosystems i.e. PT-5, PT-10 and PT-17 under UV-Visible light for hydrogen generation using H_2S .

Figure 7.Photocatalytic activity of PQ: TiO₂ hybrid nanosystems (250) mL reactor, 0.1 g catalyst, feeding rate of H2S gas 0.5 mL/min.)

Figure 7 represents the time dependent photodecomposition of H2S which clearly shows stable linear increase in the hydrogen ⁶⁵evolution. The utmost hydrogen evolution rate i.e. 36,456 µmol/h/g was obtained for PT-5 hybrid nanosystem (Table1) and slightly lower for PT-10 and PT-17.The present hydrogen evolution rate obtained is four times higher than $TiO₂$ as well as also superior than earlier reports. $13-16$, 24

⁷⁰Table 1: Photocatalytic Hydrogen evolution

The hydrogen evolution obtained is much higher than earlier reports as well as P-25 Degussa $TiO₂$ and PQ for the feeding rate of H_2S at 0.5 ml/min. At this minimum H_2S flow rate, 75 continuous hydrogen production has been monitored. As the

initial rate of H_2S feeding is higher, the hydrogen evolution rate is also increases, accordingly and hence 0.5 ml/min feeding rate is maintained throughout the experiment.¹⁶ It is noteworthy that excellent hydrogen evolution rates observed at this feeding rate.

- $5 \text{ In } 0.25 \text{ M KOH solution (pH 12.5), the weak diprotic acid H₂S.}$ (pKa values are 7.0 and 11.96) dissociates and maintains equilibrium with hydrosulfide HS^- ions (1). The PQ: TiO_2 hybrid nanosystem absorbs light and generates electron-hole pairs (2). The photogenerated valence band hole $(h⁺VB)$ upon band gap
- 10 excitation of PQ: TiO₂ hybrid nanosystem oxidizes the HS^{$-$} ion to disulfide ion (S_2^2) , liberating a proton from the HS⁻ ion (3). The conduction band electron (e^-CB) from PQ:TiO₂ hybrid nanosystem photocatalyst reduces protons to produce molecular hydrogen (4).

$$
{}_{15} H_2 S + O H \qquad \Longleftrightarrow \quad H S + H_2 O \tag{1}
$$

Semiconductor $PQ: TiO_2 \leftrightarrow h^+VB + e^-CB$ (2)

Oxidation: $2HS^+ + 2h^+VB \longrightarrow S_2^2 + 2H^+$ (3)

Reduction: $2H^+ + 2e^-CB \longrightarrow H_2$ (4)

²⁰We did not observe any hydrogen evolution without catalyst and in dark; this clearly indicates the hydrogen evolution is due to PQ: TiO2hybridnanosystems.

The mechanistic approach for photocatalytic H_2 generation via H₂S decomposition over prepared PQ: TiO₂hybrid nanostructures ²⁵are depicted in Figure 8. The PT-5 exhibits higher photocatalytic activity as compared to other prepared nanosystems. The conduction band (CB) of $TiO₂$ is 0.91 eV lower than the CB of PQ and valence band (VB) of $TiO₂$ is \sim 1.11 eV lower than VB of PQ, respectively. Upon excitation with light, there are two ³⁰possibilities: (1) the electron-hole pairs generated in both the semiconductors and generated species acts as a reducing as well as oxidizing agents to generate H_2 and (2) the CB electrons of $TiO₂$ recombine with VB hole of PQ. The $H₂$ generation takes place at CB of PQ and oxidation reaction occurred at VB of $TiO₂$.

Figure 8.Mechanism of H₂ generation

This process results in the availability of electron for H_2 generation by inhibiting the electron-hole recombination in the individual semiconductor. The PL study supports the reaction ⁴⁰mechanism and reveals that activity of nanosystem found to be decreased with increase in PL emission. The PL study clearly support that the emission peak around 550 nm confer the recombination of electron-hole pairs where large number of electrons transfer from CB of PQ to CB of $TiO₂$ and recombine

45 with holes present in VB of PQ or $TiO₂$. The PT- 17 nanosystem

is having higher PL intensity and hence shows lower H_2 generation. Additionally, the reduction in the nanoparticles size of $TiO₂$ intact on PQ in the sample PT-5 and PT-10 also affects the H_2 generation. The reduction in PL intensity in these samples $50 \text{ may be due to lower size of TiO}_2$ which creates defects. These defects are responsible for suppression of charge carrier recombination which ultimately enhances the photocatalyticactivity.¹⁵ However, though PT-5 has slightly higher intensity than PT-10 but combine effect with lower 55 particle size enhance the inhibition in the charge carrier recombination (as discussed in PL study) which ultimately confer utmost activity.⁴⁰ Another noteworthy advantage of coupled semiconductor systems is that it allows inter-particle electron transfer which enhances charge separation and help for ⁶⁰enhancement in photocatalytic activity. Considering the very small nanoparticles of $TiO₂ (3-5 nm)$ in intimate contact with PQ, the surface trapping of electrons and holes before recombination will also be more efficient.⁴⁰ The photo generated electron hole pairs have a much shorter distance to travel to reach the surface in ⁶⁵a small cluster. Once, the charged carriers (electron and holes) trapped at the interface, and then they can participate in redox

reactions and contribute in enhancement of photocatalytic activity.

Conclusions

 70 In nutshell, the novel organo-inorganic PQ: TiO₂ hybrid semiconductor nanosystems have been successfully synthesized by facile solvothermal method. The nanosystems have been used as a photocatalyst for the photodecomposition of H_2S to produce hydrogen under UV-visible light. The utmost hydrogen evolution

- 75 i.e. 36,456 μ mol/h/g has been observed over a naked (without co catalyst) PQ: TiO₂ hybrid nanosystem under UV-Visible light. In this hybrid nanosystem, very small nanoparticles of $TiO₂$ are intact with sheets of PQ which exhibit most efficient inhibition of charge carrier recombination which is responsible for enhanced
- ⁸⁰photocatalytic activity. The prima-fascia result shows that organo-inorganic hybrid nanosystem providing a promising choice of efficient UV-Visible light active photocatalyst. It is noteworthy that other hybrid nanosystems with other semiconductor oxide can be synthesized with the present 85 technique.

Experimental

Catalyst Preparation

For the synthesis of PQ: $TiO₂$ nanosystem, 5 ml of titanium tetraisopropoxide (17 mmol) in methanol, 2 ml of hydrazine hydrate ⁹⁰was stirred in beaker. To this solution 3.05 g (17 mmol) of guanidine carbonate solution in acetic acid was added. The solution was further stirred for 15 min and appropriate amount (0.25 g) of freshly prepared PQ was added and continued the stirring for 15 min. PQ was prepared as per the previously 95 reported method. The resultant reaction mixture was transferred to a Teflon-lined autoclave; furthermore, this Teflon lined reactor was sealed and heated at 150ºC for 15 h in oven. The obtained precipitate of PQ: $TiO₂$ was washed and dried at 80 $^{\circ}$ C. Similar procedure was followed using different concentrations of 100 titanium tetra-isopropoxide i.e. 10 and 5 mmol. The PQ:

 $TiO₂$ nanosystem prepared using 5, 10 and 17 mmol of titanium tetra-isopropoxide precursor were termed as PT-5, PT-10 and PT-17, respectively.

Catalyst Characterization

- ⁵Structural studies of the as-synthesized products were performed using an X-ray diffractometer (XRD-D8, Advance, Bruker-AXS). Morphological study of the samples was performed using Field Emission Scanning Electron Microscopy (FESEM, Hitachi S-4800) and a Transmission Electron Microscope (TEM, Philips,
- ¹⁰EM-CM-12). UV-Visible absorbance spectra were recorded using Shimadzu UV-vis-NIR spectrophotometer (Model UV-3600) over a wavelength range of 200 to 800 nm.

Photocatalytic Study

The hydrogen evolution was performed in a quartz photo-reactor 15 using 0.1 g of PQ: TiO₂ under the irradiation of 450W Xe lamp light source (Oriel). The cylindrical quartz reactor was filled with 250 mL, 0.25 M aqueous KOH and purged with argon for 30

- min. Hydrogen sulphide (H_2S) was bubbled through the solution for 60 min. at the rate of 0.5 mL min⁻¹. 0.1 g of photocatalyst
- $_{20}$ (PQ: TiO₂) was introduced in to the reactor and irradiated with the visible-light source (Xe-lamp Oriel, 450 W) at a constant stirring with continuous H_2S bubbling (0.5 mL/min). The excess H2S was trapped in (0.5M) NaOH solution. The amount of evolved H₂ was measured using graduated gas burette and
- ²⁵analyzed using Gas Chromatograph (Model Shimadzu GC-14B, MS-5 Å column, TCD, Ar carrier).

Acknowledgements

BBK would like to thank Department of Electronics and ³⁰Information Technology (DeitY), Govt. of India for financial support and C-MET for providing research facilities. VUP gratefully acknowledges Council of Scientific and Industrial Research, CSIR, New Delhi, India for financial support.BBK and CJP would like to acknowledge the support by MSIP (Ministry of

³⁵Science, ICT & future planning), Korea through Brain Pool program.

Notes and references

^aCentre for Materials for Electronics Technology (C-MET), Department of Electronics and Information Technology (DeitY), Govt. of India. Off

⁴⁰*Pashan Road, Panchwati, Pune-411008, India (kbbb1@yahoo.com) ^bDepartment of Physics, University of Pune, Pune-411007, India ^cDepartment of Materials Science and Engineering, Chonnam National University77, Yongbongro, Bukgu, Gwangju, South Korea.(parkcj@jnu.ac.kr)*

- 1 H. Kind, H. Yan, B. Messer, M. Law, P. D. Yang, *Adv. Mater.,* 2002, **14**, 158.
- 2 M. H. Huang, S. Mao, H. Feick, H. Yan, Y. Wu, H. Kind, E. Weber, ⁵⁰R. Russo, P. D. Yang, *Science.,* 2001, **92**, 1897.
- 3 X. Fang, Y. Bando, M. Liao, U. K. Gautam, C. Zhai, B. Dierre, B. Liu, T. Sekiguchi, Y. Koide, D. Golberg, *Adv. Mater.,* 2009, **21**, 2034.
- 4 Z. B. He , J. S. Jie , W. J. Zhang , W. F. Zhang , L. B. Luo , X. Fan , ⁵⁵G. D. Yuan , I. Bello, S. T. Lee, *Small.,* 2009, **5**, 345.
- 5 X. Chen, C. Li, M. Gratzel, R. Kosteckid, S. S. Mao, *Chem. Soc. Rev*., 2012, **41**, 7909.
- 6 M. Gratzel, *Nature.,* 2001, **414**, 338.
- 7 T. Y. Zhai, Z. J. Gu, H. Z. Zhong, Y. Dong, Y. Ma, H. B. Fu, Y. F.
- ⁶⁰Li, J. N. Yao, *Cryst. Growth Des.,*2007, **7**, 488.
- 8 B. B. Kale, J. O. Baeg, S. M. Lee, H. Chang, S. J. Moon, C. W. Lee,*Adv. Funct. Mater.,* 2006,**16**, 1349.
- 9 P. Zhang, J. Zhang, J. Gong, *Chem. Soc. Rev*., 2014, **43**, 4395.
- 10 K. E. deKrafft, C. Wang, W. Lin, *Adv. Mater.,* 2012, **24**, 2014. ⁶⁵11 L. Liu, Z. Liu, A. Liu, X. Gu, C. Ge, F. Gao, L. Dong,
	- *ChemSusChem.,* 2014, **7**, 618.
	- 12 X. Zong, J. Han, B. Seger, H. Chen, G. Lu, C. Li, L. Wang, *Angew. Chem*. *Int. Ed*., 2014, **53**, 4399.
	- 13 S. K. Apte, S. N. Garaje, G. P Mane, A. Vinu, S. D. Naik, D. P. Amalnerkar, B. B. Kale, *small.*, 2011, 7, 957.
- 14 N. S. Chaudhari, A. P. Bhirud, R. S. Sonawane, L. K. Nikam, S. S. Warule, V. H. Rane, B. B. Kale,*Green Chem*., 2011, **13**, 2500.
- 15 N. S. Chaudhari, S. S. Warule, S. A. Dhanmane, M. V. Kulkarni, M. Valant, B. B. Kale, *Nanoscale*., 2013, **5**, 9383.
- ⁷⁵16 V. U. Pandit, S. S. Arbuj, U. P. Mulik, B. B. Kale, *Environ. Sci. Technol.,* 2014, **48**, 4178.
	- 17 P. V. Kamat,*J. Phys. Chem. B.,* 2002,**106**, 7729.
- 18 A. Gasparotto, P. Fornasiero, G. V. Tendeloo, D. Barreca, V. Gombac, D. Bekermann, O. I. Lebedev, E. Tondello, A. Devi, C. ⁸⁰Maccato, R. A. Fischer, T. Montini, *J. Am. Chem. Soc*., 2011, **133**, 19362.
- 19 D. Barreca, P. Fornasiero, A. Gasparotto, V. Gombac, C. Maccato, A. Pozza, E. Tondello, *Chem. Vap. Deposition,* 2010, **16**, 296.
- 20 M. Cargnello, A. Gasparotto, D. Barreca, V. Gombac, P. Fornasiero, ⁸⁵T. Montini, *Eur. J. Inorg. Chem.,* 2011, **28**, 4309.
- 21 C. Maccato, D. Barreca, G. Carraro, A. Gasparotto, V. Gombac, P. Fornasiero, *Surface & Coatings Technology*, 2013, **230** , 219.
- 22 G. Carraro, C. Maccato, A. Gasparotto, T. Montini, S. Turner, O. I. Lebedev, V. Gombac, G. Adami, G. Tendeloo, D. Barreca, P. ⁹⁰Fornasiero, *Adv. Funct.Mater.* 2014, **24**, 372.
- 23 D. Barreca, G. Carraro, V. Gombac, A. Gasparotto, C. Maccato, P. Fornasiero, E. Tondello, *Adv. Funct. Mater.* 2011, **21**, 2611.
- 24 N. Li, G. Liu, C. Zhen, F. Li, L. Zhang, H. M. Cheng, *Adv. Funct. Mater.,* 2011, **21**, 1717.
- ⁹⁵25 L. L. Tan, S. P. Chai, A. R. Mohamed, *ChemSusChem.,* 2012, **5**, 1868.
- 26 G. Xie, K. Zhang, B. Guo, Q. Liu, L. Fang, J. R. Gong, *Adv. Mater.,* 2013, **25**, 3820.
- 27 J. S. Lee, K. H. You, C. B. Park, *Adv. Mater.,* 2012, **24**, 1084.
- ¹⁰⁰28 W. Qian, P. A. Greaney, S. Fowler, S. K. Chiu, A. M. Goforth, J. Jiao, *ACS Sustainable Chem. Eng*., 2014, **2**, 1802.
	- 29 T. Hirakawa, P. V. Kamat, *J. Am. Chem. Soc*., **2005**, **127**, 3928.
- 30 R. Liu, P. Wang, X. Wang, H. Yu, J. Yu, *J. Phys. Chem. C*, 2012, **116**, 17721.
- ¹⁰⁵31 C. T. Dinh, H. Yen, F. Kleitz, T.O. Do, *Angew. Chem. Int. Ed*., 2014, **53**, 6618.
	- 32 Z. Bian, T. Tachikawa, P. Zhang, M. Fujitsuka, T. Majima, *J. Am. Chem. Soc*., 2014, **136**, 458.
- 33 T. Kamegawa, S. Matsuura, H. Seto, H. Yamashita, *Angew. Chem.* ¹¹⁰*Int. Ed*., 2013, **52**, 916.
	- 34 B. Seger, P. V. Kamat, *J. Phys. Chem. C,* 2009, **113***,* 18946.
	- 35 C. Lin, Y. Song, L. Cao, S. Chen, *Nanoscale.*, 2013, **5**, 4986.
- 36 S. Pradhan, D. Ghosh, S. W. Chen, *ACS Appl. Mater. Interfaces*., 2009, **1**, 2060.
- ¹¹⁵37 M. M. Khan, S. A. Ansari, D. Pradhan, M. O. Ansari, D. H. Han, J. Lee, M. H. Cho, *J. Mater. Chem. A*, 2014, **2**, 637.
	- 38 J. Y. Kim, D. Lee, H. J. Kim, I. Lim, W. I. Lee, D. J. Jang, *J. Mater. Chem. A*, 2013, **1**, 5982.
- 39 R. Zhu, C. Y. Jiang, B. Liu, S. Ramakrishna, *Adv. Mater*., 2009, **21**, $120994.$
- 40 W. Ho, J.Yu, J. Lin, J. Yu, P. Li, *Langmuir*, 2004, **20**, 5865.

⁴⁵