

Journal of Materials Chemistry A

Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this *Accepted Manuscript* with the edited and formatted *Advance Article* as soon as it is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

1 **Facile Fabrication of Stable Monolayer and Few-layer Graphene**
2 **Nanosheets as Superior Sorbents for Persistent Aromatic**
3 **Pollutant Management in Water**
4

5 Kaijie Yang,^{ab} Jun Wang,^{ab} and Baoliang Chen^{*ab}

6 ^a Department of Environmental Science, Zhejiang University, Hangzhou 310058, China

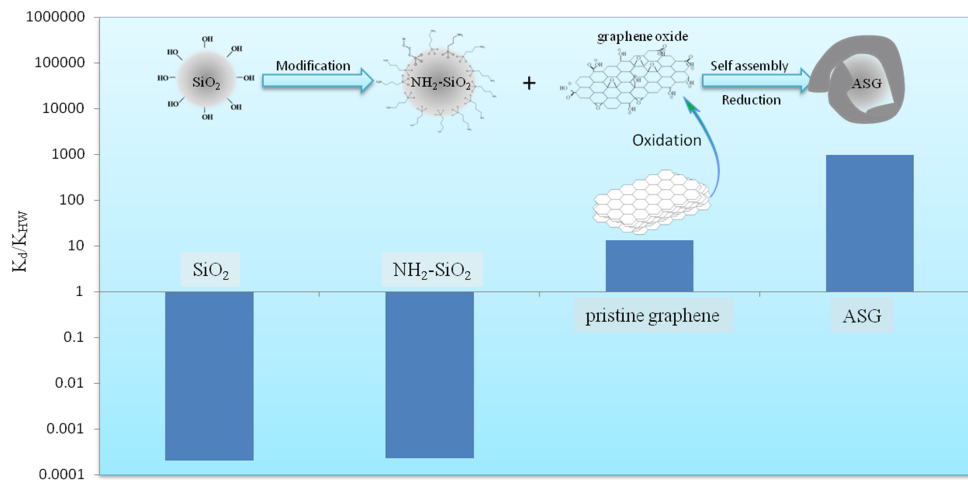
7 ^b Zhejiang Provincial Key Laboratory of Organic Pollution Process and Control, Hangzhou 310058, China

8
9
10
11 * Corresponding Author E-mail: blchen@zju.edu.cn

12 Phone: 0086-571-88982587

13 Fax: 0086-571-88982587
14
15
16
17
18
19
20
21
22

Table of Content (TOC)



45 **Abstract**

46 A facile method was employed to prepare stable monolayer and few-layer graphene (ASG)
47 nanosheets by loading on nanosilica substrates. The adsorption capability of ASG was
48 enhanced by a factor of up to 10^2 for phenanthrene compared with pristine graphene, which
49 was higher than that in previous reports for nanomaterials. Interaction mechanisms of
50 monolayer graphene with organic solute were discussed.

51

52 **Introduction**

53 Graphene is a 2D, single-atom thick carbon sheet with the carbon atoms arranged in a
54 honeycomb pattern.¹ Its discovery has attracted enormous attention because of its remarkable
55 electronic properties, ultra strong mechanical strength, optical transmittance and large surface
56 area, which are important for various applications, including energy storage,² catalysis,³
57 electronics,⁴ sensing,^{5,6} biointerface transport^{7,8} and environmental pollution management.⁹⁻¹⁹
58 The majority of chemical and biological processes in the environment and in biology occur at
59 the liquid/solid interface of monolayer graphene/water systems.^{5,8,11,17} However, it is a critical
60 challenge to obtain stable monolayer graphene sheets in water for investigating nanoscale
61 interactions with organic solutes on the graphene plane because significant aggregation of the
62 nanosheets occurs via large area π - π interactions and strong van der Waals forces.^{11,12,17}
63 Furthermore, the aggregation hides the huge surface area of graphene nanosheets,¹¹ and then
64 may hinder the adsorption capability of graphene to environmental pollutants.¹⁶ Therefore,
65 most knowledge on the adsorption capacity of graphene in aqueous solutions has been
66 obtained from pristine stacked graphene,^{10-13,20} and the largely microscopic mechanisms have
67 been derived from theoretical simulations.²¹ Thus, the true adsorption potential of monolayer

68 graphene remains poorly understood. Moreover, the mechanism determined via theoretical
69 simulations must be verified by experimental methods. Herein, we report a facile method to
70 prepare stable monolayer and few-layer graphene in water by controlling the loading of
71 graphene on nanosilica (nano-SiO₂) as a framework, reveal the truly powerful adsorption
72 potential of graphene, and provide insight on nanoscale interactions between monolayer
73 graphene and organic solutes at the water/solid interface.

74

75 **Results and discussion**

76 In this study, a facile and low-cost approach to stabilize monolayer and few-layer graphene in
77 water was employed using a core-shell structure (SiO₂@graphene). Nanosilica was selected
78 because of its considerable realistic significance due to the unavoidable interaction between
79 graphene and silica when graphene is discharged into environments where silica is
80 widespread. The fabrication progress is presented in Figure 1. Briefly, the supporting SiO₂
81 was first modified with 3-aminopropyltriethoxysilane (APTES) to bind with -NH₂ and make
82 it positively charged. Monolayer graphene oxide (GO) was synthesized using the modified
83 Hummer's method followed by ultrasonication.^{22,23} The self-assembly between GO and
84 aminosilica (NH₂-SiO₂) was conducted at proper pH conditions and controlled concentrations
85 under mechanical stirring. After obtaining the aminosilica-support-graphene oxide (ASGO),
86 aminosilica-support-graphene (ASG) was produced by traditional chemical reduction. The
87 detailed process of the fabrication, structural characterization, and adsorption experiments are
88 presented in the electronic supplementary information (ESI).

89

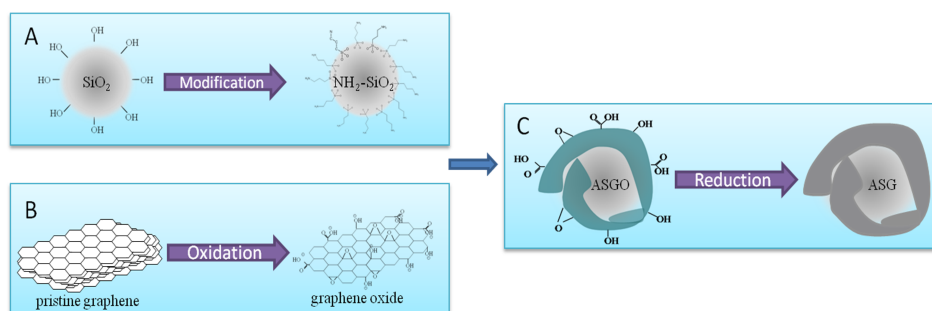


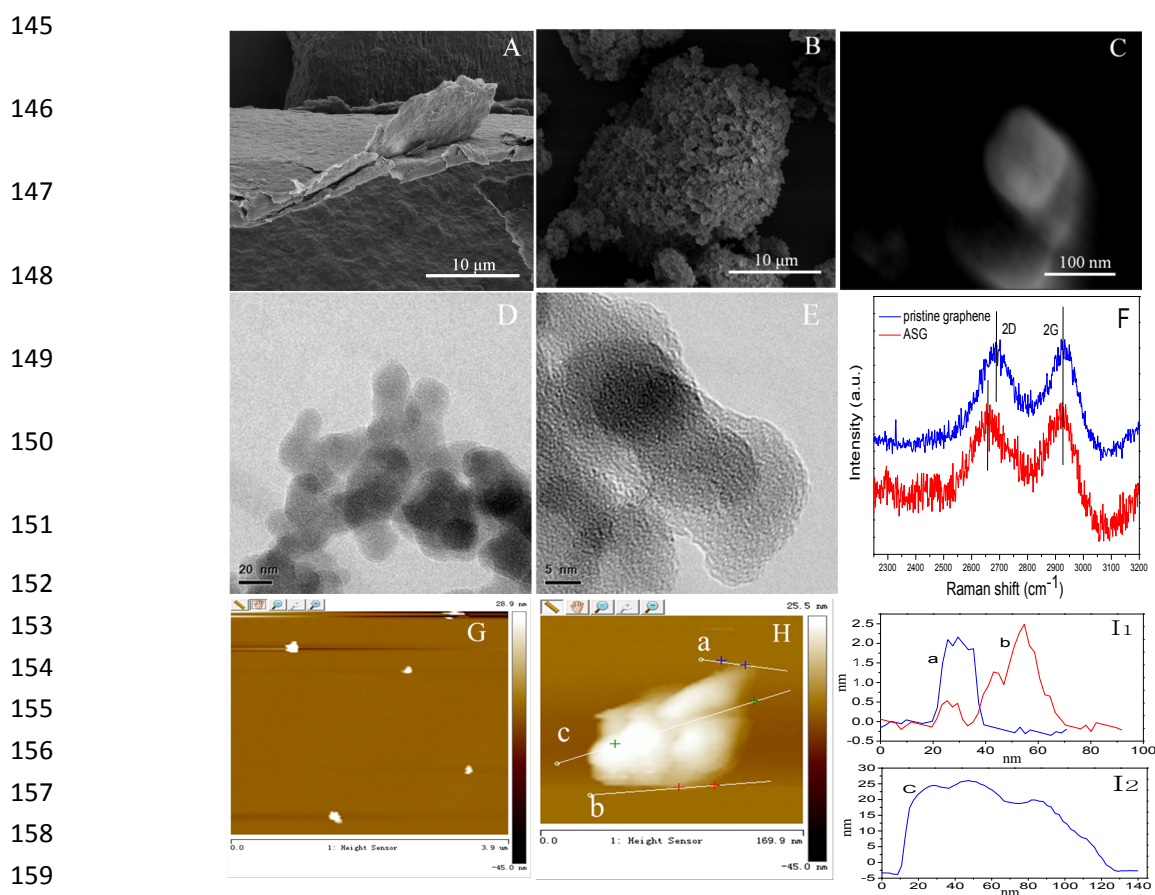
Figure 1. The chemical route for the fabrication of aminosilica-support-graphene (ASG). (A) Modification of SiO₂ with -NH₂. (B) Cleavage graphene nanosheets via oxidation. (C) Coating of negative charged graphene oxide on the positive charged NH₂-SiO₂, and then conversion to ASG via chemical reduction.

The prepared products were analyzed by Fourier transform infrared spectroscopy (FTIR) (Figure S-1). Two additional adsorption peaks appeared in NH₂-SiO₂ at 2854 and 2926 cm⁻¹, which were assigned to the C-H vibration of the grafted APTES, indicating the successful synthesis of NH₂-SiO₂. The elemental analysis results (see Table S-1 in ESI) suggested that NH₂-SiO₂ had an additional N percentage of 0.4258% compared with SiO₂. The surface area (Table S-2) and zeta potential (Figure S-2) of NH₂-SiO₂ and GO were measured to verify the appropriate fabrication conditions. NH₂-SiO₂ had a surface area of 108.5 m²/g. Through theoretical calculations, one gram of NH₂-SiO₂ can completely support 41.25 mg of graphene (2630 m²/g) in a monolayer state at full coverage. To supply sufficient loading regions on the NH₂-SiO₂ during fabrication, the proportion of graphene in ASG was maintained below 4.13%. From the zeta potential analysis, NH₂-SiO₂ and GO were oppositely charged in the pH range from 1 to 8; thus, the GO and aminosilica (NH₂-SiO₂) were able to self-assemble via electrostatic interactions. Once attached, the positive charges on the surface of NH₂-SiO₂ were neutralized by the negatively charged graphene. After coating with monolayer GO nanosheets,

116 no additional GO sheets could bind to the surface of SiO₂ due to the electrostatic repulsion
117 between the adlayer GO and the free GO. In addition, the disappearance of the FTIR peak for
118 C=O (1725 cm⁻¹) in ASGO compared with GO suggested the occurrence of a chemical
119 reaction between -COOH and -NH₂. The final product of ASG was obtained through the
120 chemical reduction of ASGO. The surface charge of ASG was highly positive (Figure S-2)
121 and was dominated by the core of NH₂-SiO₂ rather than by the outer region of the graphene
122 adlayer. Even under ultrasonication conditions, no graphene nanosheets dissolved from the
123 ASG, suggesting that the combination of graphene and NH₂-SiO₂ was stable.

124 The microstructures of the pristine as-prepared graphene and ASG were examined by
125 SEM and HR-TEM. The SEM image (Figure 2A) indicates that the pristine graphene was
126 seriously assembled, as reported.^{11,12} After wrapping the surface of the NH₂-SiO₂, the stacked
127 interlamination was loosened, and the morphology was fluffy and porous (Figure 2B) due to
128 the electrostatic repulsion between the positively charged ASG particles. The large area
129 face-to-face interaction between the graphene nanosheets caused by the π - π interactions and
130 van der Waals forces was transformed into a point-to-point interaction with the support of
131 SiO₂. In addition, the ASG exhibited a core-shell structure of SiO₂@graphene with an
132 ultrathin graphene layer (Figure 2C), which was confirmed by HR-TEM (Figures 2D and 2E).
133 From the AFM images (Figure 2G-2H) and Raman spectra (Figure 2F), we observed that the
134 graphene layer loading on the surface of SiO₂ was only a few layers or even a monolayer.
135 Because of the large area of the graphene sheet, part of the graphene layer extended outside
136 the edge of the SiO₂ (Figure 2E). As reflected in the AFM image, the height of the extended
137 graphene was approximately 2 nm, as observed in Figure 2(I₁), representing approximately

138 two or three layers of overlapped, extended graphene.²⁴⁻²⁷ In some areas, the height of the
 139 extended graphene layer was only 0.5 nm (Figure 2I₁), which was nearly equal to the
 140 theoretical thickness of monolayer graphene (0.34 nm).²⁶ In the Raman spectra (Figure S-3),
 141 the 2D peak of ASG exhibited an obvious down-shift compared with pristine graphene
 142 (Figure 2F), which also demonstrated that the graphene nanosheets in the ASG comprised less
 143 than 5 layers or even only a monolayer.^{25,28} The size of the ASG particle was approximately
 144 100 nm (Figures 2C and 2I₂).



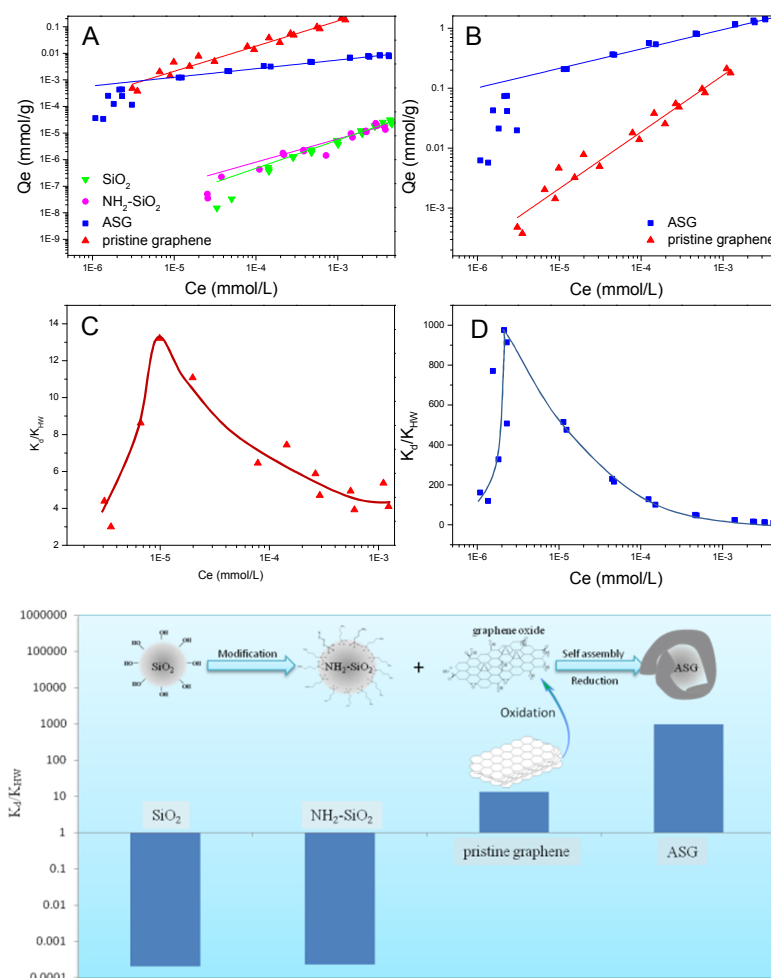
160 **Figure 2.** SEM images of pristine graphene sheets (A), the ASG bulk view (B) and an
 161 ASG single particle (C). HR-TEM images of the ASG in bulk view (D) and at high
 162 magnification (E). 2D peak in the Raman spectra of the ASG and of pristine graphene
 163 (F). AFM images showing a bulk view (G) and a single ASG particle (H). The height
 164 of each testing line in Figure (H) was presented in Figure (I₁) and Figure (I₂).

165

166 To probe the superior adsorption capacity of the monolayer and few-layer graphene
167 nanosheets, batch tests of the adsorption performance were conducted to determine the
168 adsorption capability of SiO₂, NH₂-SiO₂, ASG and pristine graphene (Figure 3). Phenanthrene
169 was selected as a model solute because it is a typical persistent organic pollutant. The
170 isotherms of SiO₂, NH₂-SiO₂, ASG and pristine graphene were further fitted by Freundlich
171 model (the regression parameters are presented in Table S-3). The clear structure only
172 containing three benzenes without other complex groups indicates that the interactions
173 between the graphene nanosheets and phenanthrene should consist of no more than the
174 hydrophobic effect and π - π interactions.^{11,29} The rapid equilibration (60 min) (Figure S-4)
175 illustrated the high adsorption efficiency of this material. Clearly, the phenanthrene adsorption
176 to SiO₂ and NH₂-SiO₂, which possess hydrophilic surfaces, was extremely low. Notably, once
177 the silica materials were wrapped by the monolayer and few-layer graphene nanosheets (the
178 content of graphene was only 0.5876%), the phenanthrene adsorption capacity improved
179 sharply, by a factor of up to 3×10^4 . The graphene sheets, in a manner similar to a large cloth,
180 tightly attached to the surface of SiO₂ (Figures 2E and 2H), which transformed the
181 hydrophilic interface of SiO₂ into an aromatic hydrophobic interface. The interaction
182 mechanism with phenanthrene was therefore altered. As SiO₂ is widely existed in the
183 environment, simple combination of released graphene and SiO₂ will dramatically improve
184 the adsorption capacity of SiO₂ with environmental pollutants. The resulting drastic change
185 will deeply effect the fate and transport of pollutants in the environment, which may be then
186 aggravated environmental risks of the released graphene.

187 The adsorption capacities with different adsorbents by mass are presented in Figure 3A.
188 Obviously, the adsorption capacity of ASG is lower than the pure pristine graphene, because

189 SiO₂ has little affinity to phenanthrene and the content of graphene is just 0.5876%. During
 190 fabrication, SiO₂ was added as auxiliary material to excavate the potential capacity of
 191 graphene. For ASG, the well-dispersed graphene nanosheets displayed a large improvement,
 192 by a factor of up to 10², in adsorption quantity compared with agglomerated graphene for the
 193 same graphene content (Figure 3B). The monolayer and few-layer graphene nanosheets



212 **Figure 3.** Adsorption isotherms of phenanthrene on SiO₂, NH₂-SiO₂, ASG and pristine
 213 graphene (A). Comparative adsorption capacity of ASG and pristine graphene at the
 214 same graphene content (B). The variation curves of K_d/K_{HW} of pristine graphene (C)
 215 and ASG (D) with equilibrium concentrations of phenanthrene. The maximum values
 216 of the K_d/K_{HW} ratio for phenanthrene adsorption to various materials (E). The initial
 217 concentration of phenanthrene ranged from 0.0056 to 0.56 $\mu\text{mol/L}$. The water-to-solid
 218 ratio was 200 mg/8 mL for SiO₂, 200 mg/8 mL for NH₂-SiO₂, 1 mg/240 mL for
 219 pristine graphene, and 1 mg/8 mL for ASG.

220 exhibited superior adsorption performance for phenanthrene, with capacities of up to 1.43
221 mmol/g. We ascribe this strong adsorption capability to two sources: (1) Graphene
222 aggregation was avoided, and the graphene was present as monolayer and few-layer graphene
223 on an nano-SiO₂ support. After that, the abundance of potential active adsorption sites of the
224 pristine graphene, which were concealed on the inner sheets and occupied by interfacial water
225 molecules,²⁶ were released and functioned as superior affinity regions. (2) There are two
226 adsorption sites on the graphene ad-layer, i.e., the flat surface and the grooves.^{11,29} A groove is
227 a high surface energy site, and molecules will prefer to adsorb to the groove, while the flat
228 surface remains unoccupied. Benefitting from the rugged surface of the monolayer graphene
229 on nano-SiO₂, many additional wrinkles will be created to form large and powerful groove
230 regions in a stable state.^{30,31}

231 To gain further insight into the adsorption mechanism, the hexadecane-water partition
232 coefficient ($K_{HW}=34581$ L/kg for phenanthrene) was introduced to highlight the interaction
233 strength beyond the hydrophobic effect.³² K_{HW} has been widely used as a surrogate for
234 modeling the transport and partition of organic molecules on biological interfaces.³³ As
235 observed in Figure 3C, the affinity of the organic solute to pristine graphene was much higher
236 than K_{HW} , and the peak K_d/K_{HW} ratio was as high as 13 at the phenanthrene equilibrium
237 concentration (C_e) of 1.2×10^{-5} mmol/L. The affinity of the monolayer and few-layer graphene
238 nanosheets was much higher than K_{HW} , and the maximum K_d/K_{HW} reached 1,000 at $C_e =$
239 2.4×10^{-6} mmol/L (Figure 3D). That is, the enrichment capacity of the monolayer and
240 few-layer graphene was 10^3 times higher than that of hexadecane, which acted as a surrogate
241 phospholipid bilayer. These data are much higher than the K_d/K_{HW} ratio of naphthalene on

242 sulfonated graphene, which has been one of the highest adsorption capabilities reported for
243 nanomaterials.¹⁷ Compared with SiO₂, NH₂-SiO₂ and pristine graphene, the monolayer and
244 few-layer graphene loading on nanosilica acted as a superior sorbent for persistent aromatic
245 organic pollutants in water (Figure 3E). The maximum enrichment factor ($K_d=Q/C_e$) of
246 phenanthrene in water for the monolayer and few-layer graphene nanosheets reached
247 3.46×10^7 . Furthermore, the facile preparation and collection of ASG make SiO₂@graphene an
248 ideal preconcentration material for analytical chemistry, catalysis reactions and environmental
249 pollutant management.

250 More surprisingly, an underlying variation in K_d/K_{HW} was observed by varying the initial
251 concentration of phenanthrene from 5.6×10^{-3} to 5.6×10^{-6} mmol/L. A sharp peak formed in the
252 curve of K_d/K_{HW} for the monolayer and few-layer graphene as a function of the C_e of the
253 solute (Figures 3C and 3D), consistent with our previously reported results for the adsorption
254 of naphthalene, phenanthrene and pyrene to pristine graphene and graphene oxide.¹¹ This
255 unusual variation has not been recognized in theoretical simulations of the interaction of
256 graphene with small organic molecules.²¹ Therefore, in addition to the hydrophobic effect and
257 π - π interactions, additional adsorption mechanisms should be considered because the actual
258 interaction between monolayer graphene and a solute is much more complicated than
259 previously reported. Recent studies revealed that small organic molecules will affect the
260 morphology of graphene nanosheets.¹¹ Additionally, water molecules are arranged in an
261 ice-like structure³⁴⁻³⁷ on the liquid/solid interface of water/graphene through non-H bonding.³⁸
262 Upon interacting with low concentrations of phenanthrene, more wrinkles are generated on
263 the extended graphene nanosheets. The ice-like water molecules attached to the flat graphene

264 can be replaced by phenanthrene though the replacement of the non-H bond with a stronger
265 π - π interaction, which is a possible mechanism for increase of K_d in the low solute
266 concentration ranges. This exchange process will reinforce the hydrophobic character of
267 graphene, which is affected by the attachment of water molecules and the release of free
268 energy upon phenanthrene adsorption. Thus, K_d/K_{HW} increased with C_e at low concentration
269 ranges. The turning point of K_d/K_{HW} for ASG appeared at relatively lower C_e than that of
270 pristine graphene. Because the monolayer and few-layer graphene nanosheets on ASG stand
271 well in water in comparison with the piled pristine graphene, the interaction of phenanthrene
272 with graphene nanosheets on ASG was more effective.

273 In summary, stable monolayer and few-layer graphene nanosheets in water were
274 successfully prepared by a facile and low-cost method by controlling the loading of
275 negatively charged GO on positively charged nanosilica. The true adsorption capability of
276 monolayer and few-layer graphene nanosheets in water was reported for the first time and was
277 extremely high compared with pristine graphene. The high efficiency and easy collection of
278 ASG make it a promising material for the detection and treatment of environmental pollutants.
279 Additional interaction mechanisms at the water/solid interface between monolayer graphene
280 and organic solutes should be recognized in addition to hydrophobic effects and π - π
281 interactions.

282

283 **Electronic supplementary information (ESI) available:** Experimental methods
284 for the preparation of GO, graphene, NH_2 - SiO_2 , ASGO and ASG and for the adsorption
285 experiment and characterization results including FTIR, zeta potential, Raman spectra,

286 element analysis, surface area and adsorption kinetics.

287

288 **Acknowledgments**

289 This project was supported by the National Science Foundation for Distinguished Young
290 Scholars of China (Grant 2142500268), the National Basic Research Program of China (Grant
291 2014CB441106), the National Natural Science Foundation of China (Grants 21277120,
292 41071210, and 20890111).

293

294 **Notes and references**

295 1 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dobonos, I. V.
296 Grigorieva, A. A. Firsov, *Science*, 2004, **306**, 666.

297 2 J. B. Goodenough, K. S. Park, *J. Am. Chem. Soc.*, 2013, **135**, 1167.

298 3 S. J. Guo, S. H. Sun, *J. Am. Chem. Soc.*, 2012, **134**, 2492.

299 4 H. L. Li, S. P. Pang, S. Wu, X. L. Feng, K. Mullen, C. Bubeck, *J. Am. Chem. Soc.*, 2011,
300 **133**, 9423.

301 5 F. Schedini, A. K. Geim, S. V. Morozov, E. W. Hill, P. Blake, M. L. Katsnelson, K. S.
302 Novoselov, *Nature Materials*, 2007, **6**, 652.

303 6 Q. F. Xia, Y. J. Huang, X. Yang, Z. Li, *Acta Chimica Sinica*, 2012, **70**, 1315.

304 7 O. Akhavan, E. Ghaderi, *ACS Nano*, 2010, **4**, 5731.

305 8 C. X. Guo, X. T. Zheng, Z. S. Lu, X. W. Lou, C. M. Li, *Adv. Mater.*, 2010, **22**, 5164.

306 9 K. C. Kemp, H. Seema, M. Saleh, N. H. Le, K. Mahesh, V. Chandra, K. S. Kim, *Nanoscale*,
307 2013, **5**, 3149.

- 308 10 L. L. Ji, W. Chen, Z.Y. Xu, S. R. Zheng, D. Q. Zhu, *J. Environ. Qual.*, 2013, **42**, 191.
- 309 11 J. Wang, Z. M. Chen, B. L. Chen, *Environ. Sci. Technol.*, 2014, **48**, 4817.
- 310 12 J. Zhao, Z. Y. Wang, Q. Zhao, B. S. Xing, *Environ. Sci. Technol.*, 2014, **48**, 331.
- 311 13 S. M. Maliyekkal, T. S. Sreeprasad, D. Krishnan, S. Kouser, A. K. Mishra, U. V.
- 312 Waghmare, T. Pradeep, *Small*, 2013, **9**, 273.
- 313 14 Q. Liu, J. B. Shi, J. T. Sun, T. Wang, L. Zeng, G. B. Jiang, *Angew. Chem. Int. Ed.*, 2011, **50**,
- 314 5913.
- 315 15 Q. L. Fang, B. L. Chen, *J. Mater. Chem. A*, 2014, **2**, 8941.
- 316 16 X. T. Liu, H. Y. Zhang, Y. Q. Ma, X. L. Wu, L. X. Meng, Y. L. Guo, G. Yu, Y. Q. Liu,
- 317 *J. Mater. Chem. A*, 2013, **1**, 1875.
- 318 17 G. X. Zhao, L. Jiang, Y. D. He, J. X. Li, H. L. Dong, X. K. Wang, W. P. Hu, *Adv. Mater.*,
- 319 2011, **23**, 3959.
- 320 18 H. C. Bi, X. Xie, K. B. Yin, Y. L. Zhou, S. Wan, R. S. Ruoff, L. T. Sun, *J. Mater. Chem. A*,
- 321 2014, **2**, 1652.
- 322 19 G. X. Zhao, J. X. Li, X. M. Ren, C. L. Chen, X. K. Wang, *Environ. Sci. Technol.*, 2011, **45**,
- 323 10454.
- 324 20 Y. B. Sun, S. B. Yang, G. X. Zhao, Q. Wang, X. K. Wang, *Chem. Asian J.*, 2013, **8**, 2755.
- 325 21 P. Lazar, F. Karlicky, P. Jurecka, K. Kocman, E. Otyepkova, K. Safarova, M. Otyepka, *J.*
- 326 *Am. Chem. Soc.*, 2013, **135**, 6372.
- 327 22 W. S. Jr. Hummers, R. E. Offeman, *J. Am. Chem. Soc.*, 1958, **80**, 1339.
- 328 23 N. I. Kovtyukhova, P. J. Ollivier, B. R. Martin, T. E. Mallouk, S. A. Chizhik, E. V.
- 329 Buzaneva, A. D. Gorchinskiy, *Chem. Mater.*, 1999, **11**, 771.

- 330 24 D. Chen, H. B. Feng, J. H. Li, *Chem. Rev.*, 2012, **112**, 6027.
- 331 25 K. P. Loh, Q. L. Bao, P. K. Ang, J. X. Yang, *J. Mater. Chem.*, 2010, **20**, 2277.
- 332 26 G. Cicero, J. C. Grossman, E. Schwegler, F. Gygi, G. Galli, *J. Am. Chem. Soc.*, 2008, **130**,
- 333 1871.
- 334 27 A. C. Ferrari, J. C. Meyer, V. Scardaci, C. Casiraghi, M. Lazzeri, F. Mauri, S. Piscanec, D.
- 335 Jiang, K. S. Novoselov, S. Roth, A. K. Geim, *Phys. Rev. Lett.*, 2006, **97**, 187401.
- 336 28 Z. H. Ni, Y.Y. Wang, T. Yu, Z. X. Shen, *Nano Res.*, 2008, **1**, 273.
- 337 29 M. Roos, D. Kunzel, B. Uhl, H. H. Huang, O. B. Alves, H. E. Hoster, A. Gross, R. J. Behm,
- 338 *J. Am. Chem. Soc.*, 2011, **133**, 9208.
- 339 30 Z. H. Pan, N. Liu, L. Fu, Z. F. Liu, *J. Am. Chem. Soc.*, 2011, **133**, 17578.
- 340 31 M. Ishigami, J. H. Chen, W. G. Cullen, M. S. Fuhrer, E. D. Williams, *Nano Lett.*, 2007, **7**,
- 341 1643.
- 342 32 M. H. Abraham, H. S. Chadha, G. S. Whiting, R. C. Mitchell, *J. Pharmaceutical Sci.*, 1994,
- 343 **83**, 1085.
- 344 33 S. Natesan, Z. B. Wang, V. Lukacova, M. Peng, R. Subramaniam, S. Lynch, S. Balaz, *J.*
- 345 *Chem. Inf. Model.*, 2013, **53**, 1424.
- 346 34 T. O. Wehling, A. L. Lichtenstein, M. I. Katsnelson, *Appl. Phys. Lett.*, 2008, **93**, 202110.
- 347 35 O. Leenaerts, B. Partoens, F. M. Peeters, *Physical Review B.*, 2008, **77**, 125416.
- 348 36 M. I. Katsnelson, *Science*, 2010, **329**, 1157.
- 349 37 D. E. Lee, G. Ahn, S. Ryu, *J. Am. Chem. Soc.*, 2014, **136**, 6634.
- 350 38 A. Politano, A. R. Marino, V. Formoso, G. Chiarello, *Carbon*, 2011, **49**, 5180.