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ARTICLE TYPE

Branch-structured Bi₂S₃-CNT hybrids with improved lithium storage capability

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Bismuth sulfide (Bi_2S_3) is a promising Li-storage material due to the high gravimetric and volumetric capacities. However, this intrinsic merit has often been compromised by the poor cycle and rate capability due to lacking of structural integrity upon Li insertion/extraction process. Here, we engineer a branch-

¹⁰ structured bismuth sulfide-carbon nanotube (CNT) hybrid by growing Bi₂S₃ nanorods onto CNTs to mitigate this issue. The hierarchical Bi₂S₃-CNT hybrids possess high surface areas, rich porosity for electrolyte infiltration, and direct electron transport pathways, and can be employed as efficient electrode material for Li storage. This electrochemical results show that the Bi₂S₃-CNT hybrid exhibits a high reversible capacity (671 mAh g⁻¹ at 120 mA g⁻¹), stable cycling retention (534 mAh g⁻¹ after 90 cycles),

¹⁵ and remarkable rate capability (399 mAh g^{-1} at 3000 mA g^{-1}), notably outperforming other reported Bi₂S₃ materials. Such superb Li storage capabilities suggest that the Bi₂S₃-CNT branches could be potential electrodes for rechargeable batteries.

1 Introduction

Increasing concerns on depletion of fossil fuels and ²⁰ environmental pollution leads to a desire to use renewable energy such as solar and wind energies. However, the intermittent characteristic of such renewable energies presents tremendous challenges, and thus demands elaborate integration with energy storage system (ESS).¹ Amongst currently available ESSs, Li-ion ²⁵ battery is surely one of the most promising systems due to the bigh energy density large flavibility and environmental

- high energy density, large flexibility, and environmental benignity.² Despite of an ongoing advancement, current Li-ion batteries are limited by performance challenges of electrode materials in terms of low energy and power density. To meet the ³⁰ ever-growing demanding for higher energy, it is essential to
- utilize high-capacity anode materials such as metal oxides and sulfides to replace graphite. Through distinct conversion and/or alloying mechanisms these compounds could afford a capacity up to 600–1000 mAh g⁻¹,³⁻⁶ substantially beyond that of graphite ³⁵ counterpart (372 mAh g⁻¹).⁷

Among the chalcogenides for Li storage, bismuth sulfides (Bi_2S_3) has attracted growing attention in recent years. Featured with a direct band gap of 1.3 eV, Bi_2S_3 has been well explored as an important type of semiconductor for versatile applications ⁴⁰ such as optics, ⁸ magnetics, ⁹ biology¹⁰ and energy generation.^{11, 12}

- Specifically, Bi_2S_3 is regarded as an ideal host for hydrogen^{13, 14} and lithium storage,¹⁵⁻¹⁸ owing to the unique laminar structure. In contrast to many other metal chalcogenides such as Fe_3O_4 ,¹⁹ MoO₃,^{20, 21} and MoS₂²² that store Li mainly through conversing
- $_{\rm 45}$ reaction, the Bi_2S_3 storing Li involves succession of conversion and alloying processes with maximum Li uptake of 6.25 Li per

unit formula of Bi_2S_3 . This leads to theoretical capacities of 625 mAh g⁻¹ by mass or ~4250 mAh cm⁻³ by volume, which are 70% or 420% greater than those of current graphite, respectively.



Fig. 1 Schematic illustration of electron and ion transport in Bi₂S₃-CNT branched structure.

However, the Bi₂S₃ materials often suffer from instable performance associated with the poor conductivity and structural ⁵⁵ integrity induced by huge volume expansion upon Li cycling. which severely compromises their potentiality in advanced Li-ion batteries. For instance, Ma *et al.* ¹⁶ reported that uniform Bi₂S₃ fabrics could deliver 1083 mAh g⁻¹ initially but only retained 366

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mAh g⁻¹ after 10 cycles. Jin et al. also demonstrated that flowerlike Bi₂S₃ retained a low capacity of 169 mAh g⁻¹ after 30 cycles.¹⁸ To address this challenge, researches engineered many Bi₂S₃ composites. Jung et al. fabricated fine Bi₂S₃/carbon s nanocomposite that retained ~450 mAh g^{-1} for Bi₂S₃ alone over 100 cycles.¹⁵ However, the excessive carbon (30 wt%) in the composite presents a significant challenge for real application. Zhang et al. demonstrated that the Bi₂S₃/RGO (reduced graphene

oxide) achieved a capacity of 400.5 mAh g⁻¹ over 50 cycles.²³ ¹⁰ Previously, we reported that the carbon coated Bi₂S₃ nanomeshes demonstrated 472 mAh g^{-1} at 120 mA g^{-1} over 50 cycles, and retained 301 mAh g⁻¹ at 600 mA g⁻¹ over 40 cycles.²⁴ However, both the cycling and rate capability still need further improvement to meet the stringent performance requirement for 15 practical application.

In this work, we demonstrate the synthesis and Li storage capability of a bismuth sulfide-carbon nanotube (CNT) branched hybrid (denoted as Bi₂S₃-CNT). This hybrid is fabricated via a facile sonochemical approach followed by crystallization in

20 dimethyl formamide (DMF). The hierarchical Bi₂S₃-CNT hybrid exhibits unique structure features such as high surface areas, rich porosity, intrinsic flexibility, and direct electron transport pathways. These features ensure rapid electron and ion movement and stable structural integrity upon Li cycling (Fig. 1).²⁵ As a 25 consequence, the Bi₂S₃-CNT hybrids display enhanced Li-storage

performance outperforming previously reported Bi₂S₃ materials.

2 Experimental

2.1 Sample preparation

The synthesis of Bi₂S₃-CNT hybrid involves sonochemical $_{30}$ hydrolysis of Bi(NO₃)₃ in the presence of thioacetamide (TAA) and CNTs. In a typical operation, 20 mg HNO₃-treated CNTs (Shenzhen Nanoport, 20-30 nm in diameter) were dispersed in 40 ml aqueous solution containing 75 mg TAA (Sinopharm Chemicals) by sonication. To this mixture a 5 mL of 0.4 M HNO₃

35 solution containing 0.243g Bi(NO₃)₃·5H₂O was slowly added, and the suspension was further agitated for 1 hour. The resultant precipitation was then dispersed in 20 mL DMF and solvothermally treated at 150 °C for 2 hours. Free Bi₂S₃ sample was also prepared via the identical route without CNTs.

2.2 Characterization

The Bi₂S₃-CNT samples were characterized by X-ray diffraction (XRD, Rigaku Dmax-2400 automatic diffractometer), scanning electron microscopy (SEM, Hitachi S-4800), transmission 45 electron microscopy (TEM, FEI Tecnai G2 T20), Fourier transform infrared spectroscopy (FTIR, Bruker Tensor 27), thermogravimetry and differential thermal analysis (TG-DTA, Seko TG/DTA-7300), and nitrogen adsorption and desorption (Micromeritics Tristar 3020).

Electrochemical Li storage performance of the Bi₂S₃-CNT was evaluated by coin-type 2032 cells using composite electrodes consisting of 70% active material, 20% Super-P-Li carbon black, and 10% polyvinylidene fluoride binder. The typical loading of the active material was 1.0-1.5 mg cm⁻². Cells were assembled in

55 an Ar-filled glove box (MBraun) with both water and oxygen concentration below 1 ppm. The counter and referent electrode

are Li metal foil, the electrolyte is 1 M LiPF₆ solution in ethylene carbonate and dimethyl carbonate (1:1 by volume), and the separator is Celgard 2320 membrane. Cyclic voltammetry (CV) 60 and electrochemical impedance spectroscopy (EIS) were measured on a Zennium electrochemical workstation (Zahner). Galvanostatic charge and discharge tests were performed on a LAND battery test system (Jinnuo) at room temperature.



Fig. 2 SEM images of Bi₂S₃-CNT before (a) and after (b) solvothermal 65 treatment. TEM images of Bi₂S₃-CNT before (c, d) and after (e-g) solvothermal treatment. Poorly crystalline areas of Bi₂S₃ are highlighted by circles in (d). (h) EDS of Bi₂S₃-CNT.

3 Results and discussion

70 Morphologies of the Bi₂S₃-CNT hybrid were observed by SEM and TEM, and the images are illustrated in Fig. 2. The Bi₂S₃-CNT hybrid after sonochemical reaction shows a branched structure, where the CNT backbones are decorated with numerous Bi₂S₃ nanorods (Fig. 2a). The Bi₂S₃ branches are uniform and pretty 75 dense, as clearly shown by TEM (Fig. 2c). A high-resolution TEM image of Bi2S3 nanorods shown in Fig. 2d reveals clear lattice fringe spacings of 0.51 and 0.40 nm, corresponding to (120) and (001) facets of orthorhombic Bi₂S₃, respectively. The (001) plane is perpendicular to the elongation direction of the ⁸⁰ nanorod, suggesting that each rod is grown along the [010] direction.^{13, 24} However, at this stage, the Bi₂S₃ is still in poorly crystalline state and show certain structure disorder, as pointed out by the circles in Fig. 2d. Additionally, the lattice fringe of $0.34\ \mathrm{nm}$ that corresponds to (002) graphene planes confirm the presence of CNT backbone.

After solvothermal reaction in DMF, the branched structure is well preserved (Fig. 2b). Nonetheless, the density of Bi_2S_3

- s nanorods is slightly reduced (Fig. 2e, f), suggesting that some Bi_2S_3 rods may be detached from CNTs in DMF. It is noted that the use of aprotic solvent is critical to maintaining the branched structure, otherwise most Bi_2S_3 rods will detach CNT if protic solvents such as water or ethanol are employed. The Bi_2S_3
- ¹⁰ nanorods are 5–10 nm in width and 30–80 nm in length, and their growth direction is roughly perpendicular to the side surfaces of CNTs to minimize the lattice mismatch at heterojunction interfaces.²⁶ The high-resolution TEM image confirms that the crystallinity of Bi₂S₃ rods is much improved after DMF
 ¹⁵ solvothermal process (Fig. 2g). In the absence of CNTs, the obtained Bi₂S₃ sample exhibits a spherical morphology with particle size about 400 nm (Fig. S1 in supporting information). The microsphere is actually composed of numerous Bi₂S₃ nanorods, consistent with Zhang's report.²⁷ Energy dispersive
 ²⁰ spectroscopy (EDS) shown in Fig. 1h reveals that the Bi₂S₃-CNT
- hybrid is composed of Bi, S, and C elements.



Fig. 3. (a) XRD patterns and (b) FTIR of Bi₂S₃-CNT and acid treated CNT. (c) TG-DTA of Bi₂S₃-CNT in air. (d) N₂ adsorption and desorption isotherms and pore size distribution (inset) of Bi₂S₃-CNT.

The structure of the Bi₂S₃-CNT was identified by XRD, as shown in Fig. 3a. The diffraction peaks in the pattern can be indexed into orthorhombic Bi₂S₃ phase (PDF#17-0320),¹³ while the (002) peak of CNTs is barely visible, possibly being masked ³⁰ by the surface Bi₂S₃ phase. The structure of the Bi₂S₃-CNT was further characterized by FTIR spectroscopy (Fig. 3b). The bands at 1713 and 1633 cm⁻¹ are due to C=O and C=C stretching modes, respectively. The peak at 1385 cm⁻¹ is assigned to C–OH stretching vibrations, while the peaks in the range of 1124–1025 ³⁵ cm⁻¹ is due to C–O vibrations.²⁸ Besides these bands associated

with CNTs, additional bands at 620 and 517 cm⁻¹ might be attributed to C–S and O–Bi, respectively, confirming the anchoring of Bi and S species onto CNT. The band shift of C=O from 1728 to 1713 cm⁻¹ also suggests possible charge transfer ⁴⁰ between functional CNT and Bi_2S_3 .

Fig. 3c displays the TG-DTA profile of the Bi_2S_3 -CNT hybrid in air. The mass loss of 1.0 wt% below 150 °C is due to dissipation of adsorbed water. The variation between 300 and 480 °C may be attributed to oxidation of Bi_2S_3 , when the sulfur species may be lost as SO_2 gas or deposited as SO_4^{2-} species. The following loss of 13.7 wt% can be ascribed to burning out of CNTs.²⁴ Therefore, the loading of Bi_2S_3 in the hybrid is calculated to be 86.3 wt%. Despite a heavy loading of Bi_2S_3 , the branched hybrid shows rich porous characteristic. Fig. 3d ⁵⁰ displays N₂ adsorption and desorption isotherms of the Bi_2S_3 -CNT. The surface area of the Bi_2S_3 -CNT is estimated to be 24.7 m² g⁻¹ using the Brunauer-Emmett-Teller method, while the Barrrett-Joyner-Halenda pore volume derived from the desorption part of the isotherm is 0.13 cm³ g⁻¹. The peaks of the pore size are mesopores.



Fig. 4. (a) Initial cyclic voltammogram of Bi_2S_3 -CNT at scan rate of 0.1 mv s⁻¹. (b) Initial charge and discharge profiles of Bi_2S_3 -CNT at current rate of 60 MA g⁻¹. (c) Cyclic voltammogram of Bi_2S_3 -CNT at various scan rates. (d) Cycling performance of Bi_2S_3 -CNT at various current rates. (e) EIS of Bi_2S_3 -CNT electrode before and after cycling using an AC signal amplitude of 5 mV between 0.1 and 100k Hz.

The Li storage performance of the Bi2S3-CNT was 65 electrochemically evaluated by CV and galvanostatic tests. Fig. 4a shows the CV profiles of the Bi2S3-CNT upon initially three cycles at 0.1 mV s⁻¹, which reveal a high and stable reversibility of the hybrid towards Li reaction. The cathodic peaks at 1.79 and 1.70 V are due to conversion reaction, where Bi_2S_3 is reduce to 70 metallic Bi and Li₂S. The peaks at 0.70 and 0.59 V may be ascribed to alloying process, where LiBi and Li₃Bi are formed sequentially. Compared with previous Bi₂S₃ nanomaterial, the Bi₂S₃-CNT hybrid shows appreciable peak shift towards higher potential, suggesting a reduced polarization.²⁴ In the reverse 75 process, the dealloying of Li₃Bi occurs at 0.97 V, while the recovery of Bi₂S₃ at 1.83 and 2.11 V.¹⁵ As the anodic peak of conversion reaction is much weaker than the cathodic one, substantial portion of Bi2S3 cannot be recovered and contribute to the irreversible capacity loss. An extra redox pair at 2.04/2.35V

may be associated with Li adsorption/desorption in oxygen groups of CNTs, which is generally less reversible.

Galvanostatic curves of the ${\rm Bi}_2{\rm S}_3$ -CNT are presented in Fig. 4b. In the voltage window of 0.01–3.0 V, the initial capacities of

- s the Bi₂S₃-CNT branches reach 1146 and 705 mAh g⁻¹, and the Coulombic efficiency is 61.5%. The irreversible loss may be associated with deactivation of conversion product and formation of solid electrolyte interphase, as frequently observed in oxide electrodes.^{19, 20} After two cycles, the capacity retreats to 723 and
- ¹⁰ 667 mAh g⁻¹ for the discharge and recharge, respectively. The actual capacity of Bi_2S_3 would be 775 and 710 mAh g⁻¹ if the contribution of the CNTs (~400 mAh g⁻¹) was ruled out.²¹ This capacity is evidently higher than free Bi_2S_3 microspheres (Fig. S2a in supporting information) and previously reported Bi_2S_3
- 15 nanomaterials, $^{15,\ 16}$ and even beyond the theoretical value (625 mAh g^{-1}), which could be explained by taking the rich porosity into consideration. Li storage in mesopores through adsorption/ desorption is a quite common phenomena and has been frequently reported. 29
- ²⁰ Besides such a high capacity delivery, the Bi₂S₃-CNT branches exhibit a remarkable rate behavior. The CV presented in Fig. 3c indicates that the Bi₂S₃-CNT can sustain rapid potential sweep. Even at a fast sweep rate of 2.0 mV s⁻¹, the Bi₂S₃-CNT retains the basic CV profile, suggesting that the electrochemical Li storage
- ²⁵ in the hybrid has barely been limited by the transport of electrons and Li ions. In addition, the galvanostatic performance further confirms that the Bi₂S₃-CNT is high-rate capable (Fig. 3d). At identical charge-discharge rates of 120, 300, 600, and 3000 mA g^{-1} , the Bi₂S₃-CNT is capable of delivering capacities of 671,
- ³⁰ 585, 527, and 399 mAh g^{-1} , respectively (taking the 2nd cycle value at each rate). The capacity delivery is also stable for 10 cycles at each rate. At an extremely high rate of 6000 mA g^{-1} , the Bi₂S₃-CNT hybrid still affords 264 mAh g^{-1} . In contrast, the free Bi₂S₃ microspheres only deliver capacities of 385 mAh g^{-1} at 600
- 35 mA g⁻¹ and 250 mAh g⁻¹ at 2000 mA g⁻¹ (Fig. S2b in supporting information). More importantly, when the rate is decreased, the capacity of the Bi₂S₃-CNT increases accordingly, and a high value of 534 mAh g⁻¹ is recovered at the 90th cycle at 120 mA g⁻¹. These results indicate that the Bi₂S₃-CNT branches
- ⁴⁰ significantly outperforms versatile reported Bi₂S₃ nanostructures^{15, 16, 18, 24} and Bi₂S₃-CNT²⁷ and Bi₂S₃-RGO²³ nanocomposite. It is also worth noting that the Li storage performance of such hybrids is comparable or even superior to some well-known sulfide composites such as SnS₂-CNT³⁰ and ⁴⁵ MoS₂-CNT,³¹ suggesting the great potentiality of such branched
- materials.

Fig. 4e compares the EIS of the Bi_2S_3 -CNT electrodes before and after 90 cycles. Both spectra are composed of two semicircle in the high and middle frequencies, and a spike in the low

- ⁵⁰ frequency. The interception at Z_{real} axis refers to R_s including solution resistance and contact resistance. The high- and middlefrequency semicircles reflect the interphase resistance (R_f) and charge transfer resistance (R_{ct}), respectively.^{32, 33} It is clearly seen that the Bi₂S₃-CNT electrodes exhibit little variation in the
- ⁵⁵ spectroscopy profile after 90 cycles, implying a marked stability. This stability ensures more active material particles participating in the Li storage process, resulting in a higher level of capacity. ²¹

Such an excellent Li storage capability can be correlated with

the unique branched structure of the Bi₂S₃-CNT, in which the fine 60 Bi₂S₃ branches tightly anchored on the CNT backbone. The Bi₂S₃ nanorods, 5-10 nm in width and 30-80 nm in length, drastically reduce diffusion length of electrons and Li ions and ensure their rapid transport.^{20, 34} The CNT backbone serves as a flexible and express path for rapid charge transfer and thus lowers the 65 electrode reaction resistance.^{35, 36} In addition, the CNT could efficiently hold cracked Bi2S3 particles during Li cycling and thus minimize their break of electrical contacts with current collector. Similar phenomena have also reported in graphene supported nanomaterials.^{37, 38} Moreover, the branched structure allows free ⁷⁰ penetration of electrolyte into the bottom of Bi₂S₃ rods and thus avoids depletion of Li ions. Furthermore, the Bi₂S₃-CNT branches are intrinsically flexible, which can efficiently reduce mechanical stress and strain of the electrode upon lithiation and delithiation, thereby warranting maximum electrode stability. As $_{75}$ a consequence, the $\mathrm{Bi}_2\mathrm{S}_3\text{-}\mathrm{CNT}$ branches exhibit a robust and stable electrochemical behavior towards Li storage.

4 Conclusions

A branch-structured Bi₂S₃-CNT hybrid was readily prepared by a facile sonochemical approach followed by solvothermal ⁸⁰ crystallization at 150 °C for 2 h. The Bi₂S₃-CNT hybrid is composed of uniform Bi₂S₃ nanorods, 5–10 nm in width and 50–100 nm in length, growing roughly perpendicular to the flexible CNT backbone. This unique Bi₂S₃-CNT hybrid can be employed as an ideal material for electrochemical Li storage. The ⁸⁵ galvanostatic test results show that the Bi₂S₃-CNT demonstrates high reversible capacity (671 mAh g⁻¹), robust rate capability (399 mAh g⁻¹ at 3000 mA g⁻¹), and stable cycleability (534 mAh g⁻¹ after 90 cycles at various rates). Therefore, this work provides a facile approach to fabricate branched structure to improve the ⁹⁰ Li storage performance of Bi₂S₃ electrodes. Due to simplicity and

efficiency, this approach is intrinsically extendable and tuneable to engineering other chalcogenide materials.

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A hierarchical, branched Bi₂S₃-CNT hybrid was fabricated through a facile sonochemical approach and exhibited outstanding Li-storage capability.