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2	Enhanced visible light photocatalytic activity of bismuth
3	oxybromide lamellas with decreasing lamella thicknesses
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22	Running title: Enhanced photoactivity of BiOBr lamellas
23	

### 24 Abstract

25 BiOBr lamellas were synthesized at different reaction pH values via a hydrothermal 26 process. X-ray diffraction, scanning electron microscopy, transmission electron microscopy, 27 X-ray photoelectron spectroscopy, UV-vis diffuse reflectance spectroscopy and N<sub>2</sub> sorption 28 measurements were used to characterize the BiOBr samples. BiOBr samples have the same 29 lamella structures and band gaps, but different lamella sizes and thicknesses. Adjusting the pH 30 of the reaction system tunes the BiOBr lamella thickness from 42 to 21 nm. The 31 photodegradation efficiencies of the BiOBr lamellas for rhodamine B (RhB) and methylene 32 blue (MB) in aqueous solution were examined. The degradation efficiency for RhB is much 33 higher than that for MB. The decrease in BiOBr lamella thickness significantly enhances the 34 photocatalytic activity for the dye degradation, despite the decrease in exposed photoactive 35 (001) facet percentage. Decreasing the lamella thickness from 42 to 21 nm yields a more than 36 fourfold enhancement in photodegradation efficiency of BiOBr samples for RhB. The most 37 important factor influencing the photocatalytic activity of the BiOBr samples is their lamella 38 thickness, rather than the exposed (001) facet percentage. Thus, even for flaky 39 semiconductors with high exposed photoactive facet contents, the influence of lamella 40 thickness on photocatalytic activity should be preferentially considered.

41 Keywords: BiOBr, lamella, nanosheet, thickness, photocatalysis

42

#### 43 **1. Introduction**

Photocatalytic degradation is one of the most effective energy-saving solutions to environmental pollution. Nanometer-scale semiconductors are promising photocatalysts for this goal<sup>1, 2</sup>. Many factors influence the activity of these photocatalysts, such as band gap<sup>3</sup>, particle size<sup>4</sup>, specific surface area<sup>5</sup> and surface morphology<sup>6, 7</sup>. The surface structures of photocatalysts largely affect their photocatalytic reactions, since reaction is caused by photoinduced electrons and holes on the surface<sup>6, 7</sup>.

50 The facet-dependent photocatalytic activity of semiconductors has received much recent interest<sup>8-22</sup>. Crystalline semiconductors with a high percentage of photoactive facets can 51 52 function better than those with lower contents. Many semiconductors have been prepared with 53 high percentages of exposed photoactive facets. Kuang et al. synthesized TiO<sub>2</sub> nanosheets and 54 hollow  $TiO_2$  boxes with high percentages of exposed (001) facets, and found that the  $TiO_2$ 55 samples with a higher percentage of (001) facets exhibited higher catalytic activity in the degradation of methyl orange (MO)<sup>15</sup> and splitting water<sup>19</sup>. Ye et al. prepared 56 single-crystalline Ag<sub>3</sub>PO<sub>4</sub> rhombic dodecahedrons with only (110) facets exposed, and cubes 57 bounded entirely by (100) facets<sup>17</sup>. The rhombic dodecahedrons exhibited much higher 58 59 activities than the cubes for the degradation of MO and rhodamine B (RhB), which was 60 attributed to the higher surface energy of (110) facets. Wu et al. synthesized high (001) facet 61 content BiOBr lamellas without capping agents, and found that higher (001) facet contents led to enhanced photodegradation efficiencies for RhB and MO<sup>8</sup>. BiOI<sup>9</sup>, BiOCl<sup>18</sup>, BiVO4<sup>10</sup>, 62  $Bi_2O_3^{23}$  and  $Bi_2MoO_6^{11}$  with high percentages of exposed (001), (001), (001), (001) and (010) 63 64 facets, respectively, also exhibit high photocatalytic activities. However, such reports 65 investigating the influence of active facet percentage on the photoactivity have usually 66 neglected the effects of other factors such as particle size, specific surface area and morphology. It is usually difficult to retain a constant particle size, specific surface area or 67

68 morphology when exposing more photoactive facets. Thus, this should be considered when69 exploring the influence of active facet percentage on photoactivity.

70 Bismuth oxyhalides (BiOX, where X = Cl, Br or I) are a new family of visible light-driven photocatalysts. They have attracted much attention because of their 71 72 structure-dependent photocatalytic performance, which arises from their layered structure interleaved with  $[Bi_2O_2]^{2+}$  and double halogen atom slabs<sup>8, 9, 24, 25</sup>. The layered structure 73 74 benefits the separation of photoinduced electrons and holes under the effect of an internal electric field between  $[Bi_2O_2]^{2+}$  and halogen layers<sup>24</sup>, and also the relatively large distortion of 75 the B–O polyhedron<sup>9, 11</sup>. BiOBr has a desirable band gap ( $\sim 2.7$  eV) for visible light 76 adsorption<sup>24, 26</sup>. BiOBr reportedly has two separate valence bands, which have different 77 oxidation abilities and respond to UV and visible light, respectively<sup>27</sup>. To enhance its 78 79 photocatalytic activity, BiOBr has been prepared with various morphologies such as hierarchical flower-like microspheres<sup>28, 29</sup>, hollow microspheres<sup>30</sup> and lamellas with a high 80 percentage of exposed (001) facets<sup>8</sup>. Photocatalytic activity has also been enhanced by 81 combining BiOBr with other elements to form solid solutions<sup>31</sup>, or with compounds to form 82 heterostructures<sup>32</sup>. To our knowledge, the influence of lamella thickness of BiOBr nanosheets 83 84 on their photocatalytic efficiency has not been systematically investigated.

In the current study, a series of BiOBr lamellas were synthesized via a hydrothermal route<sup>24, 33, 34</sup>. The lamella size, especially the lamella thickness, of BiOBr samples was adjusted by changing the pH of the reaction systems. The BiOBr lamella samples have specific surface areas ( $S_{BET}$ ) within one order of magnitude and similar band gaps, but different lamella thicknesses and percentages of exposed active (001) facets ( $F_{001}$ ). The dependence of photocatalytic activity on BiOBr sample thickness and  $F_{001}$  were examined, to distinguish which factors are the most important in improving photocatalytic performance.

<sup>92</sup> **2.** Experimental

93 2.1. Materials

Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O, hexadecyltrimethylammonium bromide (CTAB), NaOH, RhB, MB and
ethanol were of A.R. grade, and were purchased from Sinopharm Chemical Reagent Co., Ltd.
(Shanghai, P. R. China) and used without further purification. Commercial N doped anatase
TiO<sub>2</sub> (N-TiO<sub>2</sub>) with N content of about 4% was purchased from Xuan Cheng Jing Rui New
Material Co., Ltd. Water used in experiments was obtained from a Hitech-Kflow water
purification system (Hitech, Shanghai, P. R. China).

100 2.2. Synthesis of BiOBr crystals

BiOBr lamellas were synthesized as reported with minor modifications<sup>24, 33, 34</sup>. 101 102 Specifically, 9.0 g of Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O and 9.0 g of CTAB were dissolved in 450 mL water by 103 stirring for 10 min. The dispersion was divided equally into six parts. The initial pH of these 104 dispersions was about 1.4. Aqueous NaOH (1 mol/L) was added dropwise to five of the 105 dispersions, to adjust their pH values to 2.0, 3.0, 4.0, 5.0 and 7.0, respectively. The resulting 106 dispersions were stirred for 1 h at room temperature, then transferred into 100-mL autoclaves 107 and heated at 160 °C for 18 h. Finally, the precipitates were collected by filtration, washed 108 thoroughly with water and ethanol, and dried at 60 °C in an oven. The BiOBr lamella samples 109 obtained at pH values of 1.4, 2.0, 3.0, 4.0, 5.0 and 7.0 were denoted as H1.4, H2, H3, H4, H5 110 and H7, respectively.

111 2.3. Characterization

Powder X-ray diffraction (XRD) was carried out using a D8 Advance diffractometer (Bruker, Germany), with Cu K $\alpha$  radiation ( $\lambda = 1.54184$  Å). UV-vis diffuse reflectance spectra (UV-vis DRS) were obtained on a Cary 100 spectrophotometer (Agilent, USA), with a BaSO<sub>4</sub> reference. X-ray photoelectron spectroscopy (XPS) was performed on a Phi 5300 esca system (Perkin-Elmer, USA), with Mg K $\alpha$  radiation (photoelectron energy 1253.6 eV). The C 1s peak at 284.6 eV was used to calibrate peak positions. The composite structure was probed with

118 transmission electron microscopy (TEM), using a JEM-2100F microscope (Jeol, Japan). 119 Morphologies were examined with a Supra55 field emission-scanning electron microscopy 120 (SEM) (Zeiss, USA). Atomic force microscopic (AFM) measurements were performed in air 121 with a Nanoscope IIIa instrument (Digital Instruments Inc., Santa Barbara, CA) operating in 122 the tapping mode (TM). S<sub>BET</sub> values and pore structures were probed by measuring volumetric 123 N<sub>2</sub> adsorption-desorption isotherms at liquid nitrogen temperature, using an Autosorb IQ-MP 124 instrument (Quantachrome, USA). Samples were degassed at 150 °C for 3 h under vacuum 125 before measurements.

126 2.4. Photocatalytic activity

127 The photocatalytic performance of as-prepared catalysts under visible light irradiation was 128 evaluated by the degradation of RhB and MB at room temperature, on a XPA-7 photocatalytic 129 reaction apparatus (Xujiang Electromechanical Plant, Nanjing, P. R. China). A 350 W Xe 130 lamp was used as the light source, and was equipped with an ultraviolet cutoff filter ( $\lambda \ge 420$ 131 nm) to provide visible light. The distance between the lamp and glass tubes containing dye 132 solutions was about 10 cm. A water filter around the lamp was used to adsorb infrared light. 133 The reaction tubes around the lamp were soaked in a constant temperature bath. For each run, 134 0.02 g of photocatalyst was added to 50 mL of RhB (10 mg/L) or MB (5 mg/L) solution. Prior 135 to irradiation, the suspension was treated by ultrasonication for 5 min and then stirred in the 136 dark for 1 h, to ensure sorption equilibrium. After a given irradiation time, ~4 mL aliquots 137 were collected and filtered through 1.0-µm nylon syringe filters. The absorbance of filtrates 138 was measured by a SP-1105 visible spectrophotometer (Shanghai Spectrum Instrument Co., 139 Ltd., P. R. China), at wavelengths of 554 nm for RhB and 664 nm for MB. The ratio of 140 remaining RhB or MB concentration to its initial concentration,  $C/C_0$ , was obtained by 141 calculating the ratio of the corresponding absorbance intensity.

142 **3. Results and discussion** 

143 *3.1. XRD analysis* 

144 Fig. 1 shows XRD patterns of the BiOBr samples. The intense and clear diffraction peaks 145 imply good crystallinity of the as-prepared samples. All samples exhibit the same diffraction peak positions. The peaks at 10.9, 22.0, 25.3, 31.8, 32.3, 33.3, 39.4, 44.9, 46.3, 47.0, 50.8, 146 147 53.5, 56.4, 57.3, 62.1 and  $69.8^{\circ}$  correspond to the (001), (002), (011), (012), (110), (003), 148 (112), (004), (020), (113), (014), (211), (114), (212), (015) and (006) planes, respectively, and 149 are indexed as pure tetragonal phase BiOBr (JCPDS file 73-2061, space group: P4/nmm, unit 150 cell parameters: a = 3.923 Å, b = 3.923 Å and c = 8.105 Å). A significant decrease in the 151 relative intensity of the (001), (002), (003), (004) and (006) peaks is observed from H1.4 to H7, as marked with '\*' in the figure. This implies a decrease in  $F_{001}$  content. The relative 152 153 intensities of the peaks for H7 match well with those of the standard pattern shown at the 154 bottom of Fig. 1.

155 The average *c*-axis crystallite size, *D*, of the BiOBr samples was calculated from the 156 (001) peak of the XRD pattern using the Scherrer formula:

157 
$$D = \frac{0.9\lambda}{B\cos\theta}$$
(1)

158 where  $\lambda$ , *B* and  $\theta$  are the Cu K $\alpha$  wavelength, full width at half-maximum intensity in radians, 159 and Bragg diffraction angle, respectively<sup>35</sup>. The results are shown in Table 1. From H1.4 to 160 H7, *D* initially decreases and then subsequently increases, with a minimum *D* observed for H4. 161 H1.4 has the largest *D* of 42 nm. These results demonstrate that the *c*-axis crystallite size can 162 be controlled through adjusting the pH of the reaction.

163 *3.2. Morphology* 

164 The general morphologies of BiOBr samples prepared at different pH values are revealed 165 by SEM (Fig. 2). A flaky structure is apparent for all samples, and their lamella size 166 distributions are shown in Fig. S1 (Supporting Information). The average lamella sizes (*L*) of 167 the six BiOBr samples are listed in Table 1. From H1.4 to H7, *L* decreases from 963 to 230

168 nm and then increases to 554 nm, with H4 exhibiting the lowest L. The average lamella 169 thickness (*H*) of H1.4, H2, H4, H5 and H7 is measured to be about 39, 34, 32, 24, 28 and 34 170 nm, respectively, as shown in the insets in Fig. 2 and in Table 1. H and L exhibit the same 171 trend from H1.4 to H7. The lamella thickness  $(H_A)$  was also measured from AFM images (Fig. S2 in Supporting Information). The obtained  $H_A$  values are similar to D and H (Table 1).  $H_A$ 172 173 exhibits the same trend from H1.4 to H7 as D and H. Furthermore, the H and  $H_A$  of every 174 sample are very close to its D (Table 1). This indicates that the obtained lamellas are single 175 crystals, and the transverse planes are the (001) plane, which is also confirmed by TEM 176 observation. Fig. 3 shows TEM images of H4, in which a lamella structure is also observed 177 (Fig. 3a). Fig. 3b and c show HRTEM lattice images of crystal planes magnified from Fig. 3a. 178 The interplanar spacings of 0.274 and 0.197 nm match well with the (110) and (020) planes of 179 tetragonal BiOBr (JCPDS file 73-2061), respectively. The selected area electron diffraction 180 (SAED) pattern (Fig. 3d) indicates that the BiOBr nanosheet is a single crystal of high 181 crystallinity. These results indicate that the highly exposed planes of BiOBr lamellas are (001) planes, consistent with published reports<sup>8</sup>. 182

The nanosheet morphology of BiOBr has been widely reported<sup>8, 24, 36</sup>. The formation of 183 BiOBr nanosheets may result from the laminar atomic structure of BiOBr<sup>8</sup>. The unit cell of 184 185 BiOBr with its tetragonal matlockite-type structure is shown in Scheme 1. One O atom and 186 one Br atom coordinate with four Bi atoms, respectively. One Bi atom coordinates with four O atoms and four Br atoms in different bases, respectively. One  $[Bi_2O_2]^{2+}$  layer is sandwiched 187 188 between two bromine slabs. Different layers are stacked together via weak van der Waals interactions<sup>8, 36</sup>. However, Bi-O bonds are thought to be covalent<sup>37</sup>. Scheme 1 shows that (001) 189 190 facets are rich in Bi and O atoms. Under irradiation, Bi-O square anti-prisms could generate 191 many oxygen defects because of the unstable bonding between Bi and  $O^9$ .

192 *3.3. XPS analysis* 

193 The surface chemical composition and structure of H1.4, H4 and H7 were analyzed by 194 XPS. Fig. 4 shows survey scan and high-resolution XPS spectra of the Bi 4f, Br 3d and O 1s 195 regions. The spectra of these three samples exhibit similar peak intensities and binding 196 energies, which indicates that they consist of the same elemental composition and surficial structure. The C 1s peaks in the survey scan spectra result from adventitious carbon<sup>38</sup>. Peaks 197 198 at 159.3 and 164.6 eV are ascribed to Bi  $4f_{7/2}$  and Bi  $4f_{5/2}$ , respectively, and arise from the 199 Bi-O bonds in BiOBr. The O 1s peaks are fitted by two peaks at 531.4 and 529.9 eV, and are 200 related to oxygen in BiOBr and other components (such as -OH and H<sub>2</sub>O) adsorbed on the 201 BiOBr surface, respectively. The peaks at 69.6 and 68.6 eV are attributed to Br  $3d_{3/2}$  and Br  $3d_{5/2}$ , respectively<sup>28</sup>. These XPS results are consistent with previously reported results<sup>28</sup>. 202

203 *3.4. Photocatalytic activity* 

204 The visible light photocatalytic activity of the as-prepared BiOBr samples was evaluated 205 for the degradation of RhB and MB in aqueous solutions (Fig. 5). For all tested photocatalysts, 206 adsorption equilibrium is reached after stirring for 1 h in the dark (Figs. S3 and S4 in 207 Supporting Information). The RhB and MB solutions were irradiated with visible light for 160 208 and 450 min, respectively, in the absence of catalyst, and little change in dye concentrations 209 was observed. This indicates that the photodegradation of RhB and MB in the absence of catalyst is negligible<sup>8, 39</sup>. The photocatalytic activities of these BiOBr samples are quite 210 211 different, despite their similar lamella structures. From H1.4 to H7, the photocatalytic activity 212 gradually enhances and then decreases, with H4 exhibiting the highest activity (Fig. 5a and 213 5b). N-TiO<sub>2</sub> exhibits a low photocatalytic efficiency. The total photoactivity follows the sequence  $H4 > H5 > H7 > H3 > H2 > H1.4 > N-TiO_2$ . After irradiation for 100 min, about 214 215 99% of RhB is degraded by H4 sample, while less than 50% and 15% are degraded by H1.4 216 and N-TiO<sub>2</sub>, respectively (Fig. 5a).

217 To quantitatively investigate the reaction kinetics of RhB photodegradation by the BiOBr

samples, the experimental data were fitted to the pseudo-first-order model, as expressed bythe equation (2):

220 
$$-\ln(C/C_0) = kt$$
 (2)

221 where k is the apparent first-order rate constant. As shown in Fig. 5c, all plots of  $-\ln(C/C_0)$ 222 against t exhibit linear trends, indicating that RhB photodegradation is well described by the 223 pseudo-first-order model. The k values for RhB degradation by H1.4, H2, H3, H4, H5 and H7 224 are 0.007, 0.010, 0.017, 0.036, 0.030 and 0.020 min<sup>-1</sup> (Table 1), respectively. The k value of H4 is more than four times larger than that of H1.4 (Table 1).  $S_{\rm BET}$  can influence the 225 photocatalytic activity of catalysts. To rule out the effect of  $S_{\text{BET}}$ , the k values are normalized 226 with respect to  $S_{\text{BET}}^{24}$ , and the normalized k for H4 is still more than four times larger than 227 228 that for H1.4 (Table S1). Therefore, the higher photocatalytic activity of H4 arises from its 229 intrinsic structure. These results show that reaction pH can significantly influence the visible 230 light photocatalytic activity of BiOBr for dyes in aqueous solution.

The six BiOBr samples exhibit much lower photodegradation efficiencies for MB than for RhB, in accordance with observations by Wu et  $al^8$ . N-TiO<sub>2</sub> also exhibits lower photocatalytic activity for MB. Less than 60% of MB is degraded over H4 and no obvious MB degradation over N-TiO<sub>2</sub> is observed after irradiation for 450 min (Figs. 5b and S4). The adsorption capability of the BiOBr samples for MB is also somewhat lower than that for RhB (Figs. S5 and S4 in Supporting Information). The MB photodegradation process also does not obey the pseudo-first-order model (Fig. S5 in Supporting Information).

Specific surface area<sup>40</sup> and band gaps<sup>41</sup> are two significant factors influencing photocatalytic performance of semiconductors, which were determined using N<sub>2</sub> adsorption-desorption isotherms and UV-vis DRS, respectively, for the prepared BiOBr samples. All of the N<sub>2</sub> adsorption-desorption isotherms are of type II without hysteresis loops (Fig. S6 in Supporting Information), indicating features of nonporous materials<sup>42</sup>. The  $S_{BET}$ 

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values of H1.4, H2, H3, H4, H5 and H7 obtained by the Brunauer-Emmett-Teller (BET) method are 43.6, 93.4, 62.9, 51.7, 56.1 and 103.8 m<sup>2</sup>/g, respectively, being within one order of magnitude (Table 1). UV-vis DRS of the BiOBr samples exhibit obvious photoabsorption from the UV to visible wavelengths shorter than 460 nm (Fig. S7 in Supporting Information). All the BiOBr samples have comparable  $E_g$  values of 2.70 ± 0.02 eV (Table 1). Similar results were reported by Zhang and coworkers for similar flaky BiOBr<sup>24</sup>.

The six BiOBr samples have similar lamella structures, surface morphologies, band gaps and specific surface areas. The difference among them is their lamella size and thickness. Presuming the BiOBr lamellas to be circular plates, the  $F_{001}$  values of the BiOBr samples can be estimated from the equation  $F_{001} = 1/(1 + 2H/L)$ , as shown in Table 1.  $F_{001}$  is determined by H/L, rather than solely *L*. From H1.4 to H7, the trend in  $F_{001}$  is the same as those of *L* and *H*.

An increase in photoactive facet percentage and decrease in particle size can enhance the activity of photocatalysts<sup>8, 43</sup>. The changing tendency of photocatalytic activity from H1.4 to H7 coincides with the anticipated result from the change in *H*, but is opposite to that from the change of  $F_{001}$ . This indicates that for the six BiOBr samples, the effect of *H* on photocatalytic activity is more significant than that of  $F_{001}$ . The decrease in lamella thickness plays a key role in enhancing the photocatalytic performance of BiOBr lamellas.

Fig. 6 shows a plot of *k* versus *H*, in which a linear relationship is observed. *k* increases
linearly with decreasing *H*. The formula of this linear relationship is:

$$263 k = -0.0020H + 0.084$$

(3).

The correlation coefficient is higher than 0.99. Similar results are obtained when fitting the curves of *k vs. D* (Fig. S8 in Supporting Information). In addition, similar BiOBr lamellas with lamella thicknesses from 32 to 9 nm were synthesized by Wu and coworkers<sup>8</sup>. We fitted their photodegradation data for RhB to the pseudo-first-order model, and obtained *k* values for their various BiOBr samples (Fig. S9a in Supporting Information). A linear relationship between k and lamella thickness is again obtained (Fig. S9b in Supporting Information). BiOI single-crystal nanosheets with dominant exposed (001) facets were prepared by Zan and coworkers<sup>9</sup>. Photocatalytic performance enhancement of prepared BiOI nanosheets was thought to be related to increasing percentage of exposed (001) facets, but meanwhile, obvious decrease of the nanosheet thickness was observed.

The above discussion indicates that decreasing the BiOBr lamella thickness is an efficient way to enhance their photocatalytic activity. Decreasing the lamella thickness decreases the bulk recombination of photoinduced electrons and holes, as they transfer by a shorter distance from the bulk to surface<sup>43</sup>, as shown in Scheme 2. Therefore, the photocatalytic efficiency of BiOBr lamellas is strongly dependent on their thickness.

279 Holes, electrons, superoxide radicals ( $\cdot$ O<sub>2</sub>) and hydroxyl radicals ( $\cdot$ OH) are all possible active species for the photodegradation of organic pollutants<sup>44-47</sup>. To evaluate the role of these 280 active species, individual scavengers were added to the photodegradation system. The 281 scavengers used were tert-butyl alcohol for •OH<sup>45</sup>, sodium oxalate for holes<sup>44</sup>, K<sub>2</sub>Cr<sub>2</sub>O<sub>7</sub> for 282 electrons<sup>46</sup> and benzoquinone for  $\cdot O_2^{-45}$ . Fig. 7 shows that the addition of tert-butyl alcohol or 283 284 sodium oxalate causes a slight change in the photodegradation efficiency of H4. This indicates 285 that •OH and holes are not significant active species in the photocatalytic process. After 286  $K_2Cr_2O_7$  is added, the photodegradation activity of H4 declines to a large extent, indicating 287 that electrons play a role. A significant suppression of photocatalytic performance is observed 288 when benzoquinone is added, confirming the important role of  $\cdot O_2^-$  in the photodegradation process. Thus, the photocatalytic process is mainly governed by  $\cdot O_2$ , which is in agreement 289 with the previous studies<sup>28</sup>. 290

292 Six BiOBr lamella samples with similar lamella structures and band gaps but different

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293	lamella sizes and thicknesses were synthesized at different pH by a hydrothermal method. The							
294	BiOBr lamella thickness can be controlled through adjusting the reaction pH. Decreasing							
295	BiOBr lamella thickness can significantly enhance their photocatalytic activity for dye							
296	degradation. Lamella thickness is more important than exposed photoactive facet content in							
297	influencing the photocatalytic activity of the BiOBr samples. Even for flaky semiconductors							
298	with high exposed photoactive facet contents, the influence of lamella thickness on							
299	photocatalytic activity should be preferentially considered.							
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Photocatalytic activity of BiOBr samples is enhanced by more than four times by adjusting the reaction pH.

## Figure and table captions

- Fig. 1 XRD patterns of as-prepared BiOBr samples.
- **Fig. 2** SEM images of the BiOBr samples H1.4 H7, shown in a f, respectively. Insets indicate lamella thickness of samples.
- Fig. 3 (a) TEM, (b and c) HRTEM and (d) SAED images of BiOBr sample H4.
- Fig. 4 Survey scan and high-resolution XPS spectra of the Bi 4f, Br 3d and O 1s regions for BiOBr samples.
- **Fig. 5** Photodegradation of BiOBr samples for (a and c) RhB and (b) MB, under visible light irradiation.
- Fig. 6 Dependence of pseudo-first-order rate constant (k) on lamella thickness of BiOBr samples.
- Fig. 7 Photocatalytic degradation of RhB over the BiOBr sample H4 with different scavengers: 20 mM tert-butyl alcohol (*t*-BuOH), 0.2 mM sodium oxalate, 10 mM K<sub>2</sub>Cr<sub>2</sub>O<sub>7</sub> and 0.1 mM benzoquinone.
- Scheme 1 Structural model illustration of BiOBr crystals (3×3×3 lattices), showing (a) three-dimensional projection, (b) (001) facets, (c) (010) facets and (d) (100) facets.
- **Scheme 2** Schematic showing separation of a photoinduced electron (e<sup>-</sup>) and hole (h<sup>+</sup>), and transfer from the bulk to surface of BiOBr lamellas.
- Table 1 Physicochemical properties and photocatalytic activities of BiOBr samples.



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Table 11 hysicoenemical properties and photoeatarytic activities of DioDi samples.										
	Sample	Reaction	$D^{\mathrm{a}}$	$H^{\mathrm{b}}$	$L^{c}$	${H_{\rm A}}^{\rm d}$	$S_{ m BET}$	$E_{g}$	$k^{\rm e} \times 10^3$	$F_{001}{}^{\rm f}$
		pН	(nm)	(nm)	(nm)	(nm)	$(m^2 g^{-1})$	(eV)	$(\min^{-1})$	(%)
	H1.4	1.4	42	39	963	46	43.6	2.72	7.33	92.5
	H2	2.0	39	34	935	44	93.4	2.72	10.05	93.2
	Н3	3.0	36	32	389	33	62.9	2.71	16.62	85.9
	H4	4.0	21	24	230	23	51.7	2.71	36.33	82.7
	Н5	5.0	23	28	391	28	56.1	2.69	30.06	87.5
	H7	7.0	32	34	554	38	103.8	2.68	20.20	89.1

Table 1 Physicochemical properties and photocatalytic activities of BiOBr samples.

<sup>a</sup> Crystallite size calculated by Scherrer formula from (001) plane.

<sup>b</sup> Lamella thickness from SEM images.

<sup>c</sup> Average size of BiOBr lamellas from SEM images.

<sup>d</sup> Lamella thickness from SEM images.

<sup>e</sup> The degradation rate constant of RhB over BiOBr photocatalysts.

<sup>f</sup> Percentage of exposed (001) facets.