

# RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

*Accepted Manuscripts* are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

# Prediction of Quantum Anomalous Hall Effect on Graphene Nanomesh

Cite this: DOI: 10.1039/x0xx00000x

Xiaoming Zhang, Mingwen Zhao\*

Received 06 December 2014,

Quantum anomalous Hall effect (QAHE) has been realized in the Cr-doped magnetic topological insulator at an extremely low temperature ( $\sim 30$  mK). From first-principles, we predict that QAHE can also be achieved in a well-designed graphene nanomesh without the requirement of transition metal doping. The ferromagnetic ordering arises mainly from the in-plane  $p_{xy}$  orbitals, leading to a Curie temperature of 830 K. The bulk band gap due to the intrinsic spin-orbital coupling (SOC) of the  $p_{xy}$  orbitals is about 3.7 meV, corresponding to an operating temperature of 43 K. This work reveals a viable approach for realizing QAHE in metal-free graphene nanostructures.

www.rsc.org/advances

## 1. Introduction

The quantum anomalous Hall effect (QAHE) is a topological nontrivial phase characterized by a finite Chern number and helical edge states within the bulk band gap. Without an external magnetic field, internal magnetization breaks time-reversal symmetry and spin-orbit coupling (SOC) opens a SOC-gap, giving rise to a quantized anomalous Hall conductivity. The Hall conductivity  $\sigma_{xy}$  develops plateaus at values (Chern number) quantized in units of  $e^2/h$ . The helical edge states are robust against backscattering and thus promising for low power consumption device applications.

Haldane first proposed a honeycomb-net model (known as Haldane model) for achieving QAHE. He demonstrated theoretically that the break of time-reversal invariance does not necessarily require external magnetic fields, but could occur as a consequence of magnetic ordering in a quasi-two-dimensional lattice.<sup>1,2</sup> The internal ferromagnetic (FM) ordering splits the spin-degenerated bands of topological insulators (TIs), leading to time-reversal symmetry-breaking. Magnetic topological insulators (TIs) are therefore expected to be promising for achieving QAHE.<sup>3-7</sup> In order to induce stable FM ordering, heavy transition metals are included in these systems, such as mercury-based quantum wells,<sup>8</sup> topological insulator thin films containing heavy elements,<sup>9,10</sup> graphene<sup>11-13</sup> and silicone<sup>14,15</sup> based systems, and kagome lattice<sup>16,17</sup> systems. However, due to the difficulty in controlling magnetization and SOC, the QAHE has not been observed experimentally until very recently, when it was finally realized in a Cr-doped  $(\text{Bi,Sb})_2\text{Te}_3$  topological insulator at extremely low temperature ( $\sim 30$  mK).<sup>18</sup>

Meanwhile, increasing attention has been paid to searching for metal-free topological insulators, since the first prediction on the quantum spin Hall (QSH) states of graphene.<sup>19</sup> However, the SOC in graphene is extremely weak, leading to a unobservably small topologically nontrivial band gap, which

limits the operating regime to unrealistically low temperatures. Several two-dimensional (2D) materials beyond graphene, such as silicene, germanene,<sup>20</sup> silicon carbide,<sup>21</sup> graphyne,<sup>22</sup> carbon nitride,<sup>23</sup> have been proposed as candidates for achieving QSH effect at an experimentally accessible temperature. These works enrich the database of 2D TIs and offer additional candidate materials for achieving QAHE by inducing stable FM ordering. Notably, theoretical works have showed that other than transition metal atoms, nitrogen atom can induce stable FM ordering in semiconductors, such as graphitic carbon nitrides<sup>24-26</sup> and silicon carbide<sup>27</sup>.

In this work, based on first-principles calculations, we propose a nitrogen-doped graphene kagome nanomesh as a promising metal-free magnetic topological insulator, which can be used to realize QAHE at high temperature. Combined with a Monte Carlo (MC) simulation and anomalous Hall conductivity calculation, we demonstrate that this nanomesh has stable FM ordering with a Curie temperature of 830 K and a topologically nontrivial band gap of 3.7 meV ( $\sim 43$  K) characterized by a Chern number of -1. The stable FM and relatively strong SOC are both related to the in-plane  $p_{xy}$  atomic orbitals rather than the  $p_z$  orbital. We also propose a single orbital tight binding (TB) model on an extended ruby lattice (named as extended ruby model) to reproduce the exotic properties of the nanomesh. The high Curie temperature and moderate SOC gap indicate the feasibility of the QAHE in the metal-free magnetic TI at experimentally accessible temperature.

## 2. Method and Computational Details

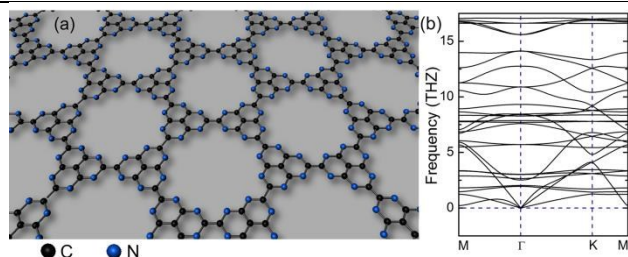
Our first principles calculations are performed in the framework of density-functional theory (DFT) as implemented in the Vienna ab initio simulation package (VASP).<sup>28</sup> The electron-electron interactions are treated within a generalized gradient approximation (GGA) in the form of Perdew-Burke-Ernzerhof (PBE) for the exchange-correlation functional.<sup>29</sup> The electron-ion interaction is described by projector-augmented-wave (PAW) potentials. The electron wavefunctions are expanded

School of Physics and State Key Laboratory of Crystal Materials, Shandong University, Jinan 250100, Shandong, China. E-mail: zmw@sdu.edu.cn

using the plane-waves with the energy cutoff of 500 eV. Self-consistent solution of the Kohn-Sham equation is carried out with the convergence of  $10^{-8}$ . The supercells are repeated periodically on the  $x$ - $y$  plane while a vacuum region of about 15 Å is applied along the  $z$ -direction to avoid mirror interaction between neighboring images. A symmetry reduced  $13 \times 13 \times 1$  Monkhorst-Pack sampling is used for the Brillouin zone (BZ) integration. Structural optimizations are performed using a conjugate gradient (CG) method until the remanent force on each atom is less than 0.002 eV/Å. Electron spin-polarization is involved in all calculations. The phonon spectra are calculated using a finite displacement method, as implemented in the Phonon code<sup>30, 31</sup> combined with the VASP code. The anomalous Hall conductivity was calculated using the Kubo formula based on the maximally localized Wannier functions.<sup>32</sup>

### 3. Result and discussion

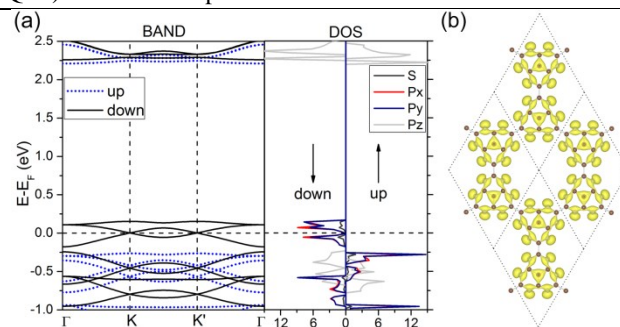
The optimized nitrogen-doped graphene kagome nanomesh is shown in Fig. 1a. It can be regarded as a porous graphene monolayer with the marginal carbon (C) atoms around the pores being substituted by nitrogen (N) atoms. We chose N dopant here considering that N have been widely used for chemical doping of carbon materials due to a comparable atomic size and five valence electrons for bonding with carbon atoms.<sup>33</sup> Meanwhile, N-doping of graphene was highly effective in modulating its band gap to achieve new properties for device applications.<sup>34</sup> It is clearly that the planar configuration and six-fold rotational symmetry of the nanomesh are both preserved, displaying characteristics of a structurally perfect kagome lattice. The equilibrium lattice constant is found to be 11.77 Å with an angle of 60° or 120° between the primitive vectors from present calculations. A vibrational analysis of phonon spectrum calculated from first-principles confirms the stability of the hypothetic nanomesh, because no modes with imaginary frequencies are found in the phonon spectrum along the highly symmetric points in BZ as shown in Fig. 1b.



**Fig. 1** (a) Schematic representation and (b) Phonon spectrums of the nitrogen-doped graphene kagome nanomesh.

Certainly, the experimental realization of the nanomesh remains challenging for chemists, though it is kinetically stable. Both unified pores in graphene and exactly N-doping are difficult to control in experiments. However, the recent refinements in the techniques of fabricating porous graphene nanomeshes (GNMs)<sup>35-38</sup> offer potential pathways to synthesizing a GNM with well-ordered, appropriately-sized pores. Meanwhile, started from the functionalization of graphene with nitrogen achieved by Wang *et al*<sup>39</sup>, abundant strategies to prepare N-doped graphene has been reported in controlled ways.<sup>40-43</sup> The doping nitrogen tend to siting at the edge of pores<sup>44, 45</sup> provides a promising prospect to controllable synthesis of the nanomesh proposed in Fig. 1a.

Notably, the above top-down strategy involves the generation of well-ordered pores in graphene sheets, which may be limited by the quality of GNMs. Bottom-up approaches refer to the construction of nanomesh from small organic precursor molecules through catalytic or thermal treatment. The power of this synthetic approach lies in the fact that it is possible to design molecular precursors leading to uniquely defined structures. This approach was successfully applied to construct a two-dimensional framework from building blocks created via the trimerization of terephthalonitrile.<sup>46</sup> Hence, the N-doped graphene kagome nanomesh may also be synthesized by the controlled bottom-up approach with appropriate precursor monomers, such as N-doped graphene quantum dots (N-GQDs)<sup>47</sup> and melem pre-treated with absolute ethanol<sup>48-50</sup>.

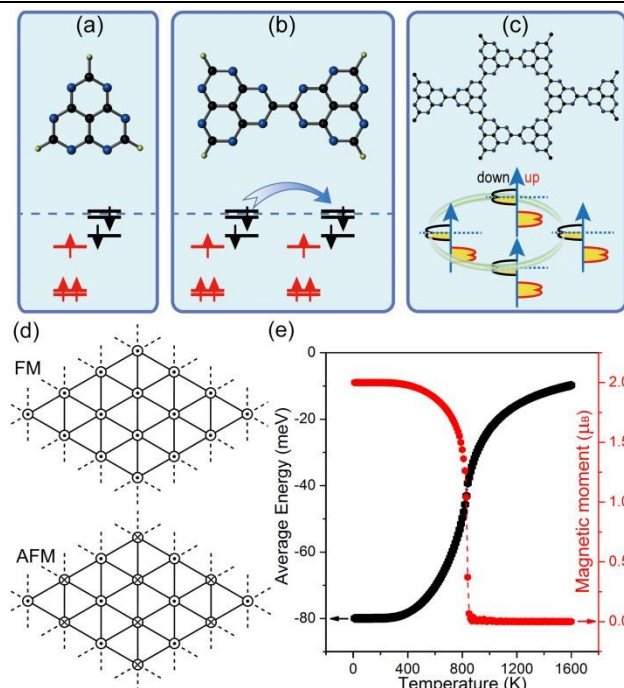


**Fig. 2** (a) Spin-resolved band lines (without SOC) and orbital-resolved electron density of states of the nanomesh. (b) The isosurfaces of Kohn-Sham wave functions of the three bands in proximity of the Fermi level with the isosurface value of  $0.004 \text{ \AA}^{-3}$ .

To understand the electronic properties of nanomesh, we performed electronic band structure calculations. Self-consistent calculations without SOC gave a stable spin-polarized state with a magnetic moment of  $2.0 \mu_B$  in one unit cell (containing a  $C_{14}N_{12}$  formula). The spin-resolved band structures are plotted in Fig. 2a along the highly-symmetric points (K and K') in BZ. We can clearly see the characteristic of kagome-like bands consisting of one flat band above two Dirac bands. Notably, because of the magnetization, the spin degeneracy of the kagome-like bands is lifted with spin-down channel being well separated from other bands and the Fermi level crossing the Dirac point exactly (a spin-gapless semiconductor). The Dirac points of the two spin channels remain touching at the K or K' points in BZ, but with a split of about 0.42 eV. The features can be seen more clearly from the orbital-resolved electron density of states (DOS) shown in the right panel of Fig. 2a. Meanwhile, the DOS demonstrates that the local magnetic moments come mainly from the  $p_x$  and  $p_y$  electrons and minor from the  $s$  electrons, rather than the  $p_z$  electrons. This feature is further confirmed by the isosurfaces of Kohn-Sham wave functions of the three spin-polarized kagome-like bands. The shapes of the isosurfaces shown in Fig. 2b display clearly characteristics of hybridized  $p_{xy}$  orbitals, which are expected to have stronger SOC effect than  $p_z$  orbital.<sup>23, 51</sup>

Since the nanomesh is magnetic, it is of interest to study the preferred magnetic coupling between local magnetic moments. We first carefully studied the magnetic interaction in one unit cell and found that there is no stable solution with an antiferromagnetic configuration (AFM), and the ferromagnetic (FM) ordering is the ground state which is more stable than the nonmagnetic (NM) state by about 145.6 meV. This result can

be explained from the perspective of the energy levels of the isolated molecules (see Fig. 3a) constituting the nanomesh. After structural relaxation, self-consistent calculations demonstrated that the molecule favors a spin-polarized ground state. For the three energy levels nearest to the Fermi level, the spin-up branch is fully occupied and the spin-down branch is only partially (2/3) occupied, producing a net magnetic moment of  $1.0 \mu_B$  (see Fig. 3a). These partially-occupied states can act as acceptor levels and mediate a virtual hopping between adjacent magnetic moments for the FM arrangement as shown in Fig. 3b, leading to a stable FM state.<sup>52</sup>

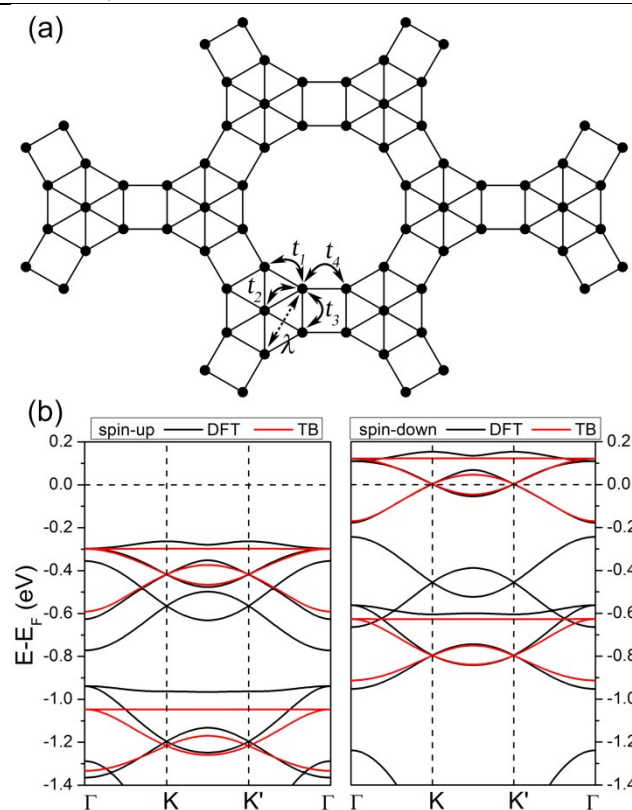


**Fig. 3** Schematic diagrams of spin-resolved energy levels exchange mechanisms for (a) isolated molecule and (b) molecular dimer cutting from the nanomesh with the corner carbon passivated by hydrogen. (c) The exchange mechanisms for the graphene nanomesh. (d) FM and AFM orderings of the graphene nanomesh. The direction of magnetic moments are represented by dot or cross in circle. (e) Temperature-dependent average energy and magnetic moments per unit cell.

Consequently, we adopted large-size supercells containing four primitive cells (donated as  $2 \times 2$ ) to further confirm the magnetic coupling and estimate Curie temperature. Starting from different initial spin arrangements of the local magnetic moments of  $2.0 \mu_B$ , self-consistent calculations lead to two types of magnetic orderings, FM and AFM, as shown in Fig. 3d, respectively. The AFM states can be seen as an antiferromagnetic combination of collinear ferromagnetic configuration. We found that the AFM phase is less stable than the FM phase by about 426.3 meV per  $2 \times 2$  cell. Hence, the nearest-neighbor exchange parameter of the local magnetic moments can be evaluated using the *Ising* model without an external field,  $H = -J_0 \sum_{\langle i,j \rangle} \hat{m}_i \times \hat{m}_j$ , where  $\hat{m}_i$  and  $\hat{m}_j$  are

local magnetic moments at the nearest-neighbor sites  $i$  and  $j$ . By comparing the DFT total energies of FM and AFM phase, the values of  $J_0$  are extracted as 6.66 meV using the following

formula:  $J_0 = \Delta E / 16m^2$ . We then estimate the Curie temperature ( $T_C$ ) from a Monte Carlo (MC) simulation within the *Ising* Hamiltonian. Starting from the initial spin arrangements of FM (see Fig. 3d) in a  $100 \times 100$  supercell containing  $10^4$  local magnetic moments, we get the variation of average energy and magnetic moment as a function of temperature shown in Fig. 3e. It can be seen that the average energy gradually rises with the increase of temperature and the magnetic moment per unit cell rapidly falls to  $0.0 \mu_B$  at the temperature of  $T_C = 830$  K, indicating the system undergoes a transition from FM to paramagnetic (PM) at this temperature. Although this MC simulation only gives a rough estimation of the Curie temperature, it is consistent with the high stability of the FM state superior to the AFM state. This is related to the large split between the two spin channels in the region near the Fermi level. The partially-occupied acceptor levels allow virtual hopping for the FM arrangement,<sup>27</sup> as shown in Fig. 3c, whereas the AFM arrangement is highly energetically disadvantageous.



**Fig. 4** (a) Schematic representation of the extended ruby lattice.  $t_1$ ,  $t_2$ ,  $t_3$ , and  $t_4$  represent the nearest-neighbor hopping parameters and  $\lambda$  is the next-nearest-neighbor intrinsic SOC parameter. (b) A comparison between the band structures calculated from first-principles (DFT) and the extended ruby model.

We now go back to the band structures of the ferromagnetic N-doped graphene kagome nanomesh and investigate its topological properties. In addition to the atomic orbitals discussed earlier in this article, the isosurfaces of Kohn-Sham wave functions shown in Fig. 2b also suggests the kagome-like bands come mainly from the marginal N and center C atoms and minor from the remaining atoms. Based on the features, we adopted a single-orbital tight-binding (TB) model with an

exchange field and onsite energy to reproduce the spin-polarized kagome-like bands. In this model, we only consider the nearest-neighbor (NN) hopping between marginal N and center C atoms, leading to an extended ruby lattice, as shown in Fig. 4a. According to this approximation, the Hamiltonian is expressed as follows:

$$H = \varepsilon_{on} - \sum_{\langle i,j \rangle, \alpha} (t_{ij} c_{i\alpha}^+ c_{j\alpha} + h.c.) - M \sum_{i, \alpha, \beta} c_{i\alpha}^+ s_{\alpha\beta}^z c_{i\beta}$$

here,  $c_{i\alpha}^+$  and  $c_{i\alpha}$  are creation and annihilation operators, respectively, for an electron with spin  $\alpha$  on site  $i$ .  $\varepsilon_{on}$  is onsite energy for both spin up and down channels.  $t_{ij}$  is the distance-dependent nearest-neighbor hopping parameters labeled  $t_1$ ,  $t_2$ ,  $t_3$ , and  $t_4$  in Fig. 4a. In addition to the nearest-neighbor hopping term (the first term), the exchange field with  $M$  is involved (the second term). Diagonalizing the above Hamiltonian in reciprocal space, we obtain its band structures, as shown in Fig. 4b. The band structures reproduce well the spin-polarized kagome-like bands given by DFT calculations with the optimal parameters of  $\varepsilon_{on} = 1.7032$  eV,  $t_1 = -0.180$  eV,  $t_2 = -0.100$  eV,  $t_3 = -2.250$  eV,  $t_4 = -0.375$  eV, and  $M = -0.21$  eV, respectively. This implies that the exchange field causes the spin-polarized kagome-like bands and adjusts the Dirac points to the Fermi level.

Next, we investigated the SOC effects. The band structures with SOC are almost identical to those without SOC (see Fig. 2a), except a SOC band gap of 3.7 meV at K or K' points in proximity of the Fermi level, which can be seen clearly from the magnified bands in Fig. 5a. The SOC band gap can be well described using the above extended ruby model combined with spin-orbital interaction, and the TB Hamiltonian is written as:

$$H_{SO} = H + i\lambda \sum_{\langle\langle i,j \rangle\rangle} c_i^+ \mathbf{s} \cdot \hat{\mathbf{e}}_{ij} c_j$$

The spin-orbital interaction is defined as spin-dependent second-nearest-neighbor hopping.  $\lambda$  is spin-orbital coupling strength shown in Fig. 4a and  $\mathbf{s}$  is the spin Pauli matrix. The unit vector  $\hat{\mathbf{e}}_{ij}$  is defined as:

$$\hat{\mathbf{e}}_{ij} = \frac{\mathbf{d}_{ij}^1 \times \mathbf{d}_{ij}^2}{|\mathbf{d}_{ij}^1 \times \mathbf{d}_{ij}^2|}$$

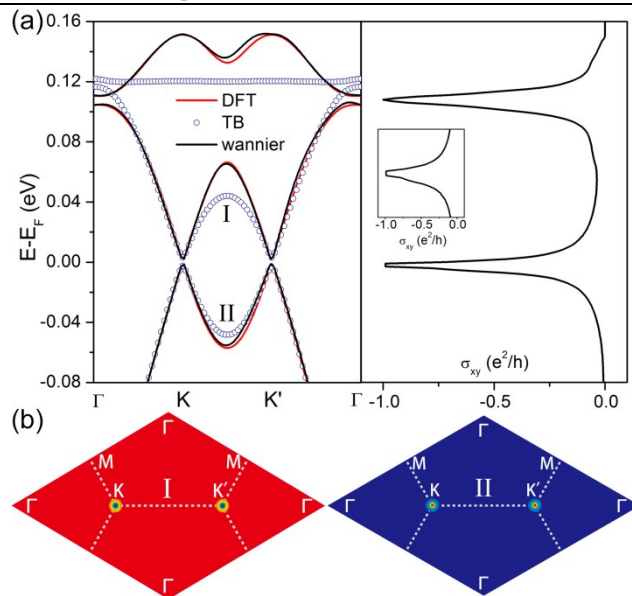
where  $\mathbf{d}_{ij}^1$  and  $\mathbf{d}_{ij}^2$  are bond vectors along the two bonds that the electron traverses when going from atom  $j$  to  $i$ . According to this definition,  $\mathbf{s} \cdot \hat{\mathbf{e}}_{ij} = \pm s^z$ . We note that this intrinsic SOC term forbids mixing of spin-up and spin-down channels due to the special nature of the 2D extended ruby model. Based on the above Hamiltonian with  $\lambda = 0.0013$  eV, the two Dirac bands are well reproduced as shown in Fig. 5a. This implies that the SOC gap of the Dirac bands in the N-doped graphene kagome nanomesh is opened due to intrinsic SOC.

The nontrivial topology of the SOC gaps can be confirmed by the Chern number ( $C$ ) calculated from the  $\mathbf{k}$ -space integral of Berry curvature ( $\Omega(\mathbf{k})$ ) of all the states below the Fermi level using the Kubo formula:

$$C = \frac{1}{2\pi} \int_{BZ} \Omega(\mathbf{k}) d^2k \quad \text{with}$$

$$\Omega(\mathbf{k}) = \sum_{n < E_F} \sum_{m \neq n} 2 \text{Im} \frac{\langle \psi_{nk} | v_x | \psi_{mk} \rangle \langle \psi_{mk} | v_y | \psi_{nk} \rangle}{(\varepsilon_{mk} - \varepsilon_{nk})^2}$$

where  $\psi_{nk}$  is the spinor Bloch wave function of band  $n$  with the corresponding eigenenergy  $\varepsilon_{nk}$  and  $v_i$  is the  $i$ th Cartesian component of the velocity operator. The right panel of Fig. 5a shows the calculated anomalous Hall conductivity as a function of energy based on maximally localized Wannier functions, which develops plateaus at the value of  $-1.0 e^2/h$  within the energy window of SOC gaps. This implies that the N-doped graphene kagome nanomesh has a topologically nontrivial state at the Fermi level with a Chern number of  $C = -1$ . The well match between Wannier and DFT bands shown in the left panel of Fig. 5a ensures the accuracy of the anomalous Hall conductivity. Based on the extended ruby model, we also calculated the distribution of Berry curvatures for the lowest conduction band (I) and highest valence band (II) around the Fermi level, respectively, and plotted them in Fig. 5b. It is obvious that the Berry curvature of the two Dirac bands resides mainly around the K or K' point in BZ. More importantly, the nontrivial SOC gap in the graphene nanomesh is well separated from other bands and thus implementable in achieving the QAHE at low temperature ( $\sim 43$  K).<sup>53</sup>



**Fig. 5** (a) Enlarged view of the kagome-like bands with SOC nearest to the Fermi level calculated from DFT, TB, and Wannier90 (left panel), and anomalous Hall conductivity calculated from the maximally localized Wannier functions (right panel). (b) The 2D distribution of the Berry curvature for the Dirac bands (I and II) obtained from TB in the momentum space. The dashed lines mark the first Brillouin zone.

## 4. Conclusions

Using first-principles calculations combined with a tight-binding model, we propose a nitrogen-doped graphene kagome nanomesh which has electronic structures characterized by kagome-like bands composing of one flat band above two Dirac bands. A moderate band gap of 3.7 meV is opened due to intrinsic SOC at the Fermi level. Anomalous Hall conductivity and Berry curvatures calculations show that the graphene nanomesh is a two-dimensional magnetic topological insulator

characterized by a Chern number of -1, which may be implementable for achieving the QAHE at low temperature (~43 K). The physical origins of QAHE are the relatively strong SOC effect and stable FM state provided by  $p_{xy}$  atomic orbitals. These features can be described by using an extended ruby model. This work suggests a promising approach to search for metal-free magnetic TIs for realizing QAHE at high temperatures.

## Acknowledgements

This work is supported by the National Basic Research Program of China (No.2012CB932302), the National Natural Science Foundation of China (Nos.91221101 and 21433006), the 111 project (No. B13209), the Taishan Scholar Program of Shandong, and the National Super Computing Centre in Jinan.

## References

- F. D. M. Haldane, *Phys. Rev. Lett.*, 1988, **61**, 2015-2018.
- M. Onoda and N. Nagaosa, *Phys. Rev. Lett.*, 2003, **90**, 206601.
- H. Zhang, C. Lazo, S. Blügel, S. Heinze and Y. Mokrousov, *Phys. Rev. Lett.*, 2012, **108**, 056802.
- H. Zhang, F. Freimuth, G. Bihlmayer, M. Ležaić, S. Blügel and Y. Mokrousov, *Phys. Rev. B*, 2013, **87**, 205132.
- Z. F. Wang, Z. Liu and F. Liu, *Phys. Rev. Lett.*, 2013, **110**, 196801.
- Y. Zhang and J. Shi, *Phys. Rev. Lett.*, 2014, **113**, 016801-016801.
- R. Yu, W. Zhang, H. J. Zhang, S. C. Zhang, X. Dai and Z. Fang, *Science*, 2010, **329**, 61-64.
- C.-X. Liu, X.-L. Qi, X. Dai, Z. Fang and S.-C. Zhang, *Phys. Rev. Lett.*, 2008, **101**, 146802.
- K. Garrity and D. Vanderbilt, *Phys. Rev. Lett.*, 2013, **110**, 116802.
- Y. Ma, Y. Dai, X. Li, Q. Sun and B. Huang, *Carbon*, 2014, **73**, 382-388.
- W.-K. Tse, Z. Qiao, Y. Yao, A. H. MacDonald and Q. Niu, *Phys. Rev. B*, 2011, **83**, 155447.
- Z. Qiao, W. Ren, H. Chen, L. Bellaiche, Z. Zhang, A. H. MacDonald and Q. Niu, *Phys. Rev. Lett.*, 2014, **112**, 116404.
- Z. Qiao, S. Yang, W. Feng, W.-K. Tse, J. Ding, Y. Yao, J. Wang and Q. Niu, *Phys. Rev. B*, 2010, **82**, 161414.
- H. Pan, Z. Li, C.-C. Liu, G. Zhu, Z. Qiao and Y. Yao, *Phys. Rev. Lett.*, 2014, **112**, 106802.
- X.-L. Zhang, L.-F. Liu and W.-M. Liu, *Sci. Rep.*, 2013, **3**, 2908.
- M. Zhao, A. Wang and X. Zhang, *Nanoscale*, 2013, **5**, 10404-10408.
- Z. Y. Zhang, *Journal of physics. Condensed matter : an Institute of Physics journal*, 2011, **23**, 365801.
- C. Z. Chang, J. Zhang, X. Feng, J. Shen, Z. Zhang, M. Guo, K. Li, Y. Ou, P. Wei, L. L. Wang, Z. Q. Ji, Y. Feng, S. Ji, X. Chen, J. Jia, X. Dai, Z. Fang, S. C. Zhang, K. He, Y. Wang, L. Lu, X. C. Ma and Q. K. Xue, *Science*, 2013, **340**, 167-170.
- C. L. Kane and E. J. Mele, *Phys. Rev. Lett.*, 2005, **95**, 146802.
- C.-C. Liu, W. Feng and Y. Yao, *Phys. Rev. Lett.*, 2011, **107**, 076802.
- M. Zhao and R. Zhang, *Phys. Rev. B*, 2014, **89**, 195427.
- M. Zhao, W. Dong and A. Wang, *Sci. Rep.*, 2013, **3**, 3532.
- A. Wang, X. Zhang and M. Zhao, *Nanoscale*, 2014, **6**, 11157-11162.
- A. Du, S. Sanvito and S. C. Smith, *Phys. Rev. Lett.*, 2012, **108**, 197207.
- X. Zhang, M. Zhao, A. Wang, X. Wang and A. Du, *J. Mater. Chem. C*, 2013, **1**, 6265.
- X. Li, J. Zhou, Q. Wang, Y. Kawazoe and P. Jena, *J PHYS CHEM LETT*, 2013, **4**, 259-263.
- M. W. Zhao, F. C. Pan and L. M. Mei, *Appl. Phys. Lett.*, 2010, **96**, 012508.
- G. Kresse and J. Furthmüller, *Phys. Rev. B*, 1996, **54**, 11169-11186.
- J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865-3868.
- K. Parlinski, Z. Q. Li and Y. Kawazoe, *Phys. Rev. Lett.*, 1997, **78**, 4063-4066.
- K. Parlinski, Cracow.
- A. A. Mostofi, J. R. Yates, Y.-S. Lee, I. Souza, D. Vanderbilt and N. Marzari, *Comput. Phys. Commun.*, 2008, **178**, 685-699.
- H. Liu, Y. Liu and D. Zhu, *J. Mater. Chem.*, 2011, **21**, 3335.
- Y. Li, Z. Zhou, P. Shen and Z. Chen, *ACS Nano*, 2009, **3**, 1952-1958.
- G. Ning, Z. Fan, G. Wang, J. Gao, W. Qian and F. Wei, *Chem. Commun. (Camb.)*, 2011, **47**, 5976-5978.
- Z. Fan, Q. Zhao, T. Li, J. Yan, Y. Ren, J. Feng and T. Wei, *Carbon*, 2012, **50**, 1699-1703.
- Y. Zhu, S. Murali, M. D. Stoller, K. J. Ganesh, W. Cai, P. J. Ferreira, A. Pirkle, R. M. Wallace, K. A. Cychoz, M. Thommes, D. Su, E. A. Stach and R. S. Ruoff, *Science*, 2011, **332**, 1537-1541.
- J. Bai, X. Zhong, S. Jiang, Y. Huang and X. Duan, *Nature nanotechnology*, 2010, **5**, 190-194.
- X. Wang, X. Li, L. Zhang, Y. Yoon, P. K. Weber, H. Wang, J. Guo and H. Dai, *Science*, 2009, **324**, 768-771.
- X. Li, H. Wang, J. T. Robinson, H. Sanchez, G. Diankov and H. Dai, *J. Am. Chem. Soc.*, 2009, **131**, 15939-15944.
- L. Qu, Y. Liu, J.-B. Baek and L. Dai, *ACS Nano*, 2010, **4**, 1321-1326.
- Y. Wang, Y. Shao, D. W. Matson, J. Li and Y. Lin, *ACS Nano*, 2010, **4**, 1790-1798.
- Z.-H. Sheng, L. Shao, J.-J. Chen, W.-J. Bao, F.-B. Wang and X.-H. Xia, *ACS Nano*, 2011, **5**, 4350-4358.
- W. Zhao, O. Höfert, K. Gotterbarm, J. F. Zhu, C. Papp and H. P. Steinrück, *J. Phys. Chem. C*, 2012, **116**, 5062-5066.
- Q. Min, X. Zhang, X. Chen, S. Li and J. J. Zhu, *Anal. Chem.*, 2014, **86**, 9122-9130.
- P. Kuhn, A. Forget, D. Su, A. Thomas and M. Antonietti, *J. Am. Chem. Soc.*, 2008, **130**, 13333-13337.
- T. V. Tam, N. B. Trung, H. R. Kim, J. S. Chung and W. M. Choi, *Sensors Actuators B: Chem.*, 2014, **202**, 568-573.
- Y. Wang, X. Wang and M. Antonietti, *Angew. Chem. Int. Ed.*, 2012, **51**, 68-89.
- T. Mu, J. Huang, Z. Liu, B. Han, Z. Li, Y. Wang, T. Jiang and H. Gao, *J. Mater. Res.*, 2011, **19**, 1736-1741.
- G. Dong, K. Zhao and L. Zhang, *Chem. Commun.*, 2012, **48**, 6178-6180.
- C. Wu and S. Das Sarma, *Phys. Rev. B*, 2008, **77**, 235107.
- E. Kan, W. Hu, C. Xiao, R. Lu, K. Deng, J. Yang and H. Su, *J Am Chem Soc*, 2012, **134**, 5718-5721.
- Z. F. Wang, N. Su and F. Liu, *Nano Lett.*, 2013, **13**, 2842-2845.