RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This Accepted Manuscript will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/advances

Journal Name

RSCPublishing

COMMUNICATION

Simple electrochemical synthesis of ultra-long silver telluride nanotubes†

Cite this: DOI: 10.1039/x0xx00000x

Kee-Ryung Park,^a Seil Kim,^a Nosang V. Myung,^b Sung-Oong Kang,^{*a} and Yong-Ho Choa^{*a}

Received 00th January 2014, Accepted 00th January 2014

DOI: 10.1039/x0xx00000x

www.rsc.org/

A simple synthetic route for ultra-long silver telluride (Ag_2Te) nanotubes is presented. Silver (Ag) nanofibers electro-spun on a plastic substrate could be electrochemically transformed into Ag_2Te nanotubes. The electrochemical transformation were processed through the electrodeposition of tellurium (Te) on the surface of Ag nanofibers, which induces the Ag atoms to diffuse into the layers of Te, and thus leads to a structural and morphological change from nanofibers to hollow nanotubes.

Silver telluride (Ag₂Te) is an intriguing phase-change material with a mixed ionic-electronic conductivity in a solid, which exhibits a structural phase transition at 150°C from a low-temperature monoclinic phase (β -Ag₂Te) of narrow bandgap semiconductor to a high-temperature cubic phase (α -Ag₂Te) of superionic conductor.^{1,2} The semiconducting and magnetic properties of Ag₂Te have been found to possess large magnetoresistance, high carrier mobility and low thermal conductivity.^{3,4} In particular, the monoclinic β -Ag₂Te with low thermal conductivity has been concerned as one of promising thermoelectric materials and also employed to be a second-step host material chemically transformed into a series of metal tellurides.^{4,5} In addition, recent theoretical and experimental studies investigated a topological insulating phenomenon in the β -Ag₂Te nanostructures, which consist of a fully filled insulating bulk state and a highly anisotropic massless Dirac cone of twodimensional surface states from the periodic quantum interference of electron phase.6,7

For enhancement in thermoelectric efficiency of the metal telluride nanostructures, researchers have predicted that onedimensional (1D) telluride nanostructures, including nanofibers, nanotubes and nanowire-based composites, would be advantageous over their bulk counterparts mainly due to the reduction of lattice thermal conductivity based on the stronger phonon-surface scattering.^{8,9} Various synthetic routes have been developed to prepare the 1D telluride nanostructures. Yang et al. reported a tellurium-templated chemical bath reaction for a series of metal tellurides with the composition modulation, closely related to the resultant thermoelectric properties.4,10 Electrochemical growth in anodic alumina nanopores yielded single-crystalline Ag2Te nanowire arrays with a narrow diameter distribution.¹¹ In addition, different types of solvothermal methods synthesized a bunch of Ag₂Te nanowires based on the topotactic transformation and the composite-hydroxidemediated redox reaction, respectively.^{12,13} It has been generally believed that the hollow 1D nanostructures with ultra-long lengths would be beneficial to enhance the thermoelectric properties stemming from the stronger phonon scattering and the coverage of the wider temperature gradients.^{8,9} However, the synthetic approaches mentioned above have been mostly effective in producing the specific morphology of nanowires with relatively short length up to tens of micrometers. To this end, we previously reported a two-step synthetic method for the ultra-long 1D telluride nanostructures from the hollow nanofibers to the hierarchical nanotubes using a combination of electro-spinning process for the sacrificial nickel (Ni) nanofibers and the following galvanic displacement of Ni nanofibers.14-17

In this work, we report the simpler and more facile electrochemical synthesis of ultra-long Ag_2Te hollow nanotubes, which could be directly transformed from the electro-spun Ag nanofibers without the conventional step of the sacrificial Ni nanofibers. The electrodepostion of tellurium (Te) on the surface of Ag nanofibers induced the chemical transformation into the Ag_2Te nanotubes with the ultra-long lengths spanning over the substrate. Structural and microscopic study demonstrated the morphological and structural changes during the electrochemical transformation. We believe that this synthetic approach would open an effective way to produce a wide range of 1D metal telluride nanostructures.

Detailed experiments from the electro-spinning of Ag nanofibers to the transformation into the Ag_2Te nanotubes are described in supplementary information (See the ESI†). Silver nitrate/polyvinylpyrrolidon (AgNO₃/PVP) composites were first electro-spun and thermally reduced to the Ag nanofiber on a plastic substrate (polyimide) in a reducing condition. Structural transition

and morphological change between the AgNO₃/PVP and the Ag nanofibers could be confirmed by scanning electron microscopy (SEM) and x-ray diffraction (XRD). As shown in Fig. 1a and b, the diameters of electro-spun AgNO₃/PVP nanofibers (125 ± 25 nm) decreases to 85 ± 15 nm in the Ag nanofibers due to evaporation of PVP binder and formation of Ag nanoparticles: the diameters and a standard deviation were counted from SEM images. The thermal reduction of AgNO₃/PVP composite leads to a structural change into the Ag phase from the AgNO₃ crystallites, as verified in XRD patterns of samples (Fig. 1c). All diffraction peaks in the XRD pattern of Ag nanofibers are indexed to the cubic Ag (JCPDS no.01-07-4613).



Fig. 1 SEM images of (a) AgNO₃/PVP nanofibers and (b) Ag nanofibers.(c) XRD pattern indicates the strctrual change from the AgNO₃ crystallites to the Ag phase induced by the thermal reduction.

Microstructure and crystallinity of the Ag nanofibers were observed using transmission electron microscopy (TEM). A bright field TEM image in Fig. 2a visualizes the bumpy surface of Ag nanofiber made of multigrains of Ag nanoparticles, which are nucleated and reduced from AgNO₃ crystallites in a PVP polymer matrix. The randomly oriented crystal lattices visualized in a highresolution TEM (HR-TEM) micrograph show the interplanar spacing of 0.238 nm, corresponding to (111) planes of cubic Ag structure. A ring pattern of selected area electron diffraction (SAED) is displayed in the inset of Fig. 2b. The XRD, HR-TEM and SAED results indicates the polycrystalline nature of as-prepared Ag nanofiber.



Fig. 2 (a) TEM image of a typical Ag nanofiber and (b) HR-TEM image of lattic fringes randomly oriented in the Ag nanofiber.

The growth mechanism on the specific morphology of Ag_2Te hollow nanofibers can be explained by a well-known Kirkendall effect or topotactic transformation. First, the Kirkendall effect may influence the growth of hollow nanostructure based on a mutual diffusion process with different diffusion coefficients of Ag and Te atoms.^{18,19} During the reaction, the so-called 'Kirkendall voids' are created from the condensation of excess vacancies at the original interface and within the faster diffusion side. For the formation of

hollow nanostructure, the Ag nanofibers are coated by the electrodeposited Te layers and the diffusion Ag and Te atoms occur in opposite directions at the different diffusion rates. In order to confirm the Kirkendall effect, we characterized the phases of sample collected at an early stage of reaction using XRD (Fig. S1 in the ESI[†]). Once the Kirkendall effect determines the growth of hollow nanofibers, the product at the early stage should consist of three phases: unreacted Ag inside wall, Ag2Te phase in the mid of nanotubes and the undiffused Te outer wall. However, the XRD result of the sample reacted for 10 m indicates that the product is composed of only two phases of the unreacted Ag and Ag₂Te. As a result, the phases identified in this early-staged sample may rule out the Kirkendall effect on the growth of hollow nanofibers because the growth of hollow nanostructures affected by the Kirkendall effect generally follow the growth steps from the early-staged core/shell structure, through the movement of interface between a diffusion couple and to the final hollow morphology.

Instead, the growth mechanism of topotactic transformation appears to be more plausible. Due to the topotactic lattice matching between hexagonal t-Te and metal telluride crystal structures, such a structural conversion system has been reported to be thermodynamically favoured, and thus derive the self-templated transformation of Te nanowires into metal telluride nanowires.^{4,5} Here, we should note that the specific morphology of metal telluride nanowires, not hollow nanofibers, were chemically transformed from the Te nanowires in a conventional chemical bath reaction.⁵ This structural and morphological preservation in the chemical transformation from Te nanowires to Ag₂Te nanowires may indicate the diffusion of Ag atoms into the layer structure of Te nanowires. In a given contrary case of Te diffusion into Ag structure, the chemical transformation is expected to result in Ag₂Te hollow nanofibers from Te nanowires. As a result, in our electrochemical scheme, it is expectable that Te4+ ions in the electrochemical solution reduce to Te on the surface of Ag nanofibers (a working electrode) by applying a negative potential. The Te atoms reduced from Te^{4+} ions are envisaged to nucleate and grow on the surfaces of Ag nanofibers in the forms of thin layers or porous sheath.¹⁵ In parallel with the deposition of Te layers, the Ag atoms from the nanofibers diffuse into the layers of Te and chemically transform into the monoclinic Ag₂Te structure. Through this electrochemical reaction, the core Ag nanofibers are sourced and exhausted for the chemical transformation and finally produce the hollow nanotubes. A schematic for the electrochemical transformation through the electrodeposition of Te is depicted in Fig. 3. Cyclic voltammetry was used to monitor the cathodic reduction of Te⁴⁺ ions on the working electrode of Ag nanofibers. A scan in an electrochemical solution containing 1M HNO₃ and 10 mM TeO₂ indicated that the Te⁴⁺ ions reduce at a potential of -0.05 V (Fig. S2 in the ESI⁺).



Fig. 3 A schematic of electrochemical transformation from the Ag nanofiber into the Ag_2Te nanotubes.

The electrochemical transformation from the Ag nanofibers to the Ag₂Te nanotubes resulted in the changes in morphologies and crystal structure. The structure identification of Ag₂Te nanotubes using XRD demonstrates the structure transformation from the cubic Ag into the monoclinic Ag₂Te, which was electrodeposited for 30 m with an applied potential of -0.05 V (Fig. 4a). The XRD pattern of Ag₂Te nanotubes exhibits the diffraction peaks that can be indexed to the crystal phase of monoclinic Ag_2Te (JCPDS no. 34-0142, a = 8.16 Å, b = 8.93 Å, c = 8.01 Å) without by-product (e. g., Ag₇Te₄) or Ag phase. The different reaction periods longer and shorter than 30 m produced the Te-rich Ag_{4.53}Te₃ phase and the unreacted Ag phase in the final products, respectively. The structural transformation should be accompanied with the morphological changes in the Ag₂Te nanotubes. Being discernible in SEM image (Fig. 4b), the diameters of nanotubes enlarge to 143 ± 21 nm in comparison with those of Ag nanofibers (85 ± 15 nm). The ultra-long morphology of Ag₂Te nanotubes in lengths is depicted in Fig. S3 (See the ESI[†]), which is inherited from the electro-spun Ag nanofibers. The lowmagnification SEM image (Fig. S3) indicates that the nanotubes were fabricated to span over the substrate with the lengths longer than at least 200 µm and a large aspect ratio higher than 1400. The diameters of as-prepared nanotubes are slightly larger than those of hydrothermally rolled-up lead telluride (PbTe) nanotubes but considerably smaller than those of hydrothermally prepared Ag₂Te nanotubes.^{20,21} However, these hydrothermally produced metal telluride nanotubes were reported to have relatively short lengths up to several micrometers and wide size distributions. In comparison with our previous work on bismuth telluride (Bi₂Te₃) nanotubes prepared through a combinational route of electrospinning of sacrificial Ni metal and its galvanic displacement,¹⁵ the diameters and size distributions of nanotubes fabricated in this work could be notably reduced and improved.



Fig. 4 (a) XRD and (B) SEM identification of the Ag₂Te nanotubes.

The electrochemical transformation into the nanotubes was further investigated by a TEM technique. Figure 5a displays a representative TEM image of Ag_2Te nanotubes, which are straight in length with rough surfaces. Differences in brightness and contrast apparently visualize the hollow morphology of nanotubes. Chemical composition of nanotubes was studied using energy-dispersive x-ray spectroscopy (EDS). The high-angle annular dark-field scanning TEM (HAADF-STEM) and EDS mapping results indicate the homogeneous distribution of Ag and Te elements within the nanotubes, which signifies the complete electrochemical transformation into the Ag₂Te nanotubes (Fig. 5b-e). EDS measurement to check the stoichiometry of nanotubes presents the stoichiometric ratio of 2:1 between the elements of Ag and Te (Fig. 5f). EDS peaks for Cu originates from the TEM grid. The morphological transformation into nanotubes could be also evidenced by the cross-sectional SEM observation (Fig. S4 in the ESI \dagger).



Fig. 5 TEM study on the Ag_2Te nanotubes. (a) TEM image, (b) HAADF-STEM image and (c - e) EDS mapping results of the Ag_2Te nanotubes. (f) EDS pattern for the stoichiometry of product.

The complete electrochemical transformation into the Ag₂Te nanotubes could be further confirmed through a cross-sectional TEM study. As recognized in the cross-sectional TEM image (Fig. 6a), the nanotube has an apparent hollow morphology. The HR-TEM image (obtained from the area marked in a red circle in Fig. 6a) shows the randomly oriented polycrystalline grains with the interplanar spacing of 0.21, 0.23 and 0.28 nm that correspond to the (200), (212) and (312) planes in monoclinic Ag₂Te (Fig. 6b).



Fig. 6 (a) a cross-sectional TEM image of nanotube and (b) HR-TEM image representing the crystal lattices of polycrystalline grains in the nanotube.

In summary, we presented a facile and high-throughput synthetic route to produce the ultra-long Ag_2Te nanotubes. The structural and microscopic characterization demonstrated that the electro-spun Ag nanofibers are readily and completely transformed into the Ag_2Te nanotubes *via* the post-electrodeposition of Te. This synthetic approach is believed to significantly contribute to direct production of various 1D metal telluride nanostructures. Furthermore, the Ag_2Te nanotubes prepared in this work could be a useful host material for the topotactic transformation into a wide range of metal telluride nanotubes, which have great applications from sensors to thermoelectrics.

Acknowledgements

This work was supported by the Industrial Strategic Technology Development Program (10045177, Development of Resistive Ceramic Thin Film using Solution Process and Low Temperature Thin Film Vacuum Getter) and No. 20112010100100) funded by the MOTIE (Ministry of Trade, Industry, and Energy), Republic of Korea.

Notes and references

^a Department of Fusion Chemical Engineering, Hanyang University, Ansan 426-791, Republic of Korea E-mail: choa15@hanyang.ac.kr

^b Department of Chemical and Environmental Engineering, University of California-Riverside, Riverside, CA 92521, USA.

† Electronic supplementary information (ESI) available: Experimental and supplementary characterizations. See DOI: 10.1039/b000000x/

- 1 R. Dalven, R. Gill, J. Appl. Phys. 1967, 38, 753
- 2 R. Chen, D. Xu, G. Guo, L. Gui, J. Mater. Chem. 2002, 12, 2435.

- 3 R. Xu, A. Husmann, T. F. Rosenbaum, M. L. Saboungi, J. E. Enderby, P. B. Littlewood, *Nature* 1997, **390**, 57
- 4 H. Yang, J. –H. Bahk, T. Day, Amr M. S. Mohammed, B. Min, G. Jeffrey Snyder, A. Shakouri, Yue Wu, Nano Lett. 2014, 14, 5398
- 5 G. D. Moon, S. Ko, Y. Xia, U. Jeong, ACS Nano. 2010, 4, 2307.
- A. Sulaev, P. Ren, B. Xia, Q. H. Lin, T. Yu, C. Qiu, S. -Y. Zhang, M. Y. Han, Z. P. Li, W. G. Zhu, Q. Wu, Y. P. Feng, L. Shen, S. -Q. Shen,
 L. Wang, *AIP Advances* 2013, 3, 032123.
- S. Lee, J. In, Y. Yoo, Y. Jo, Y. C. Park, H. –J. Kim, H. C. Koo, J. Kim,
 B. Kim, K. L. Wang, *Nano Lett.* 2012, **12**, 4194.
- 8 R. Yang, Gang. Chen, Nano Lett. 2005, 5, 1111.
- 9 H. Fang, Y. Wu, J. Mater. Chem. A. 2014, 2, 6004.
- 10 H. Yang, S. W. Finefrock, J. D. Albarracin Caballero, Y. Wu, J. Am. Chem. Soc. 2014, 136, 10242.
- 11 R. Chen, D. Xu, G. Guo, L. Gui, J. Mater. Chem. 2002, 12, 2435.
- 12 F. Xiao, Q. Wang, L. Wnag, J. Pei, N. Zhou, J. Solid State Chem. 2010, 183, 2382.
- 13 F. Li, C. Hu, Y. Ziong, B. Wan, W. Yan, M. Zhang, J. Phys. Chem. C. 2008, 112, 16130.
- 14 F. Xiao, B. Yoo, K. H. Lee, N. V. Myung, J. Am. Chem. Soc. 2007. 129, 10068.
- 15 K. –J. Lee, H. Song, Y. –I. Lee, H. Jung, M. Zhang, Y. –H. Choa, N. V. Myung, *Chem. Commun.* 2011, 47, 9107.
- 16 H. Park, H. Jung, M. Zhang, C. H. Chang, N. Ndifor-Angwafor, Y. H. Choa, N. V. Myung. *Nanoscale*, 2013, 5, 3058.
- 17 C. M. Hangarter, Y. –I. Lee, S. C. Hernandez, Y. –H. Choa, N. V. Myung, *Angew. Chem. Int. Ed.* 2010, **49**, 7081.
- 18 A. D, Smigelskas, E. O. Kirkendall, *Trans. AIME* 1947, **171**, 130.
- 19 Y. Yin, R. M. Rioux, C. K. Erdonmez, S. Hughes, G. A. Somorjai, A. Paul Alivisatos, *Science* 2001, **304**, 711.
- 20 L. Zhang, J. C. Yu, M. Mo, L. Wu, K. W. Kwong, Q. Li, *Small* 2005, 1, 349.
- 21 A. Qin, Y. Fang, P. Tao, J. Zhang, C. Su, Inorg. Chem. 2007, 46, 7403.

Graphical Abstract



Silver (Ag) nanofibers electro-spun on a plastic substrate could be electrochemically transformed into Ag_2Te nanotubes. The electrochemical transformation from Ag nanofibers into Ag_2Te nanotubes were processed through the post-electrodepostion of tellurium (Te) on the surface of Ag nanofibers, which induces the Ag atoms to diffuse into the layers of Te and thus leads to a morphology change from nanofibers to hollow nanotubes.