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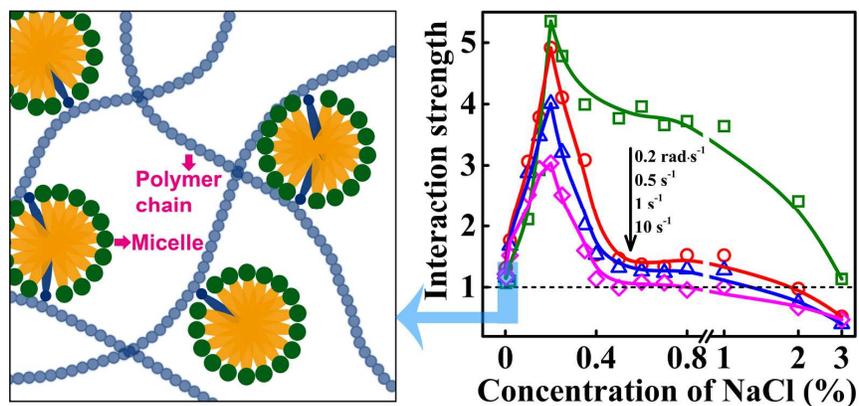


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The hydrophobic interaction between laurel alkanolamide micelles and hydrophobically associating polyacrylamide enhances the formation of network structure and the viscosity of the mixed solution increases in large electrolyte concentration range.

24

25 **Abstract**

26 Synergism of water-soluble hydrophobically associating polyacrylamide
27 (HA-PAM) with nonionic surfactant laurel alkanolamide (LAA) was investigated via
28 the rheology, fluorescence spectroscopy and dissipative particle dynamics (DPD)
29 simulation methods. The viscosity and elasticity of HA-PAM solutions increased in
30 large LAA concentration range, which was principally ascribed to the crosslinking
31 effect of LAA by aggregating to the hydrophobic chains of HA-PAM molecules as
32 confirmed by the DPD simulation and experimental results. The crosslinking effect
33 was enhanced in the presence of appropriate amount of electrolyte or with increasing
34 temperature in the studied temperature range of 20–70 °C. Thus, the LAA not only
35 significantly enhanced the salt resistance of HA-PAM but also retarded the decrease
36 of the viscosity and elasticity of the HA-PAM solutions at high temperature. The
37 HA-PAM/LAA binary systems exhibit great potential for application in tertiary oil
38 recovery of oil fields with high salinity.

39 **Keyword:** hydrophobically associating polyacrylamide, laurel alkanolamide, salt
40 resistance, synergism

41

42

43 **1. Introduction**

44 Polymer and polymer/surfactant flooding systems have been extensively used in
45 enhanced oil recovery (EOR),¹ in which polymers are being used as thickeners to
46 increase the viscosity of the flooding system and adjust the mobility ratio of water and
47 oil phase. Polymers are indispensable for the high oil displacement efficiency of the
48 flooding systems. Partially hydrolyzed polyacrylamide (HPAM)² is the most widely
49 used polymer in EOR, but the poor salt and shear resistance of HPAM limit its
50 application in reservoirs with high salinity.

51 In recent years, polymers containing salt resistance components such as
52 amphoteric or nonionic side groups^{3,4} have been reported with strong salt and shear
53 resistance. More exciting progress comes from the advance of the comb-shaped,⁵
54 hydrophobically associating⁶ polymers, which could form reversible networks,
55 resulting in a dramatic increase in apparent viscosity even at a low concentration or
56 under high salinity, and attracted the most interest of researchers. According to the
57 literature reports,⁷ hydrophobically associating polyacrylamide (HA-PAM) exhibited
58 much better salt resistance than HPAM and have showed great potential for
59 application in reservoirs with high salinity.

60 In most of the actual practices, EOR for example, polymers were used, combined
61 with surfactant. The interaction between surfactants and polymers was an interesting
62 topic, attracting much more attention for a long time. Surfactants have strong

63 influences on the rheological properties⁸ and salt resistance of polymeric solutions.⁹⁻¹¹
64 The polymer/surfactant systems with opposite charge¹²⁻¹⁵ have been paid much
65 attention. The viscosity of the solution could be greatly improved because of the inter-
66 or intra- molecular cross-links driven by the electrostatic attracting interactions. But
67 the synergistic effect is very sensitive to the concentration ratio of the surfactants and
68 polymers, and these mixed solutions always exhibit a bad salt resistance, because the
69 electrostatic interactions are sensitive to electrolytes, and minor electrolytes lead to a
70 sharp decrease of the viscosity of the solutions.¹⁶ The combination of the
71 polyelectrolytes and ionic surfactants with the same charge or nonionic surfactants
72 were also reported. Hydrophobic interaction was the driving force to enhance the
73 formation of molecular network, and some complex exhibited better salt resistance.¹¹
74 ¹⁷ The synergetic effect between HA-PAM and surfactants is still unclear, and the
75 related reports were rare. To seek appropriate surfactants to cooperate with HA-PAM
76 for further enhancing its salt resistance and oil displacement capability is desired in
77 practical areas.

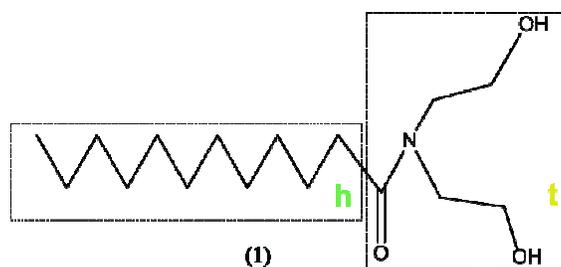
78 In this study, the rheological properties of mixed solutions of HA-PAM and
79 anionic, amphoteric and nonionic surfactants were investigated. The salt resistance of
80 HA-PAM solutions was greatly enhanced in the presence of nonionic surfactant laurel
81 alkanolamide (LAA). The synergism between HA-PAM and LAA were investigated
82 by combining experimental and molecular simulation methods. The HA-PAM/LAA
83 (H/L) binary system exhibits high thickening capacity in large concentration ratio or

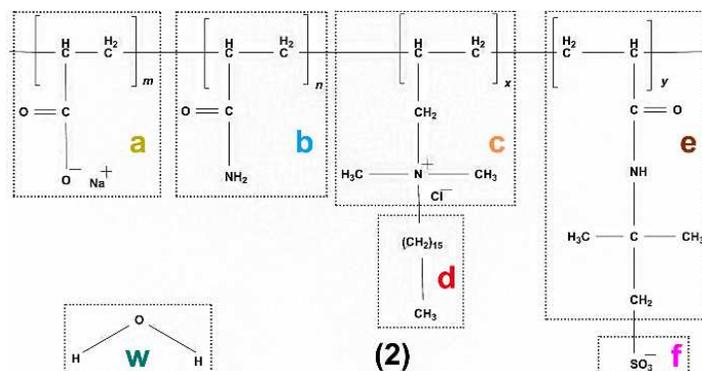
84 under high salinity. This system has great potential for application as a
85 polymer/surfactant flooding system in high salinity reservoirs.

86 2. Materials and methods

87 2.1. Materials

88 HA-PAM with a viscosity-averaged molecular weight of 3.6×10^5 and
89 cocamidopropyl betaine (CAB, 30% aqueous solution, 99% purity) were provided
90 by Geological Scientific Research Institute of Shengli oil field (Sinopec, China).⁷
91 Laurel alkanolamide (LAA) was synthesized in our lab and purified according to Ref.
92 18.¹⁸ Sodium dodecyl sulfate (SDS, 99% purity) was obtained from Cxbio
93 Biotechnology Co. Ltd. (China). Dodecyl sulphobetaine (DSB, analytical pure) was
94 synthesized and purified by Dr. J. Chen in Jin Ling Petrochemical Co. (Sinopec,
95 China). Nonaethylene glycol monododecyl ether (C₁₂E₉, 99% purity) was purchased
96 from Sigma-Adrich. Deionized water was used in all the experiments. Chemical
97 structures of HA-PAM and LAA are shown in Fig. 1. Structures of the other
98 surfactants are shown in Fig. S1.





100

101 **Fig. 1** Structures of LAA (1) and HA-PAM (2) with mole percent of m , n , x and y to

102 be 10, 80, 0.5 and 9.5 mol%, respectively. In DPD simulation, segments of HA-PAM

103 and LAA molecules are represented by beads, labeled as 'a', 'b', 'c', 'd', 'e', 'f', 'h'

104 and 't', respectively. The water molecule is labeled as 'w'.

105

106 The HA-PAM and LAA stock solutions with concentrations of 0.2% and 1%

107 were prepared by dissolving 0.2 g HA-PAM and 1.0 g LAA in 99.8 and 99.0 g

108 deionized water, respectively. Solutions with low concentrations of HA-PAM and/or

109 LAA were obtained via the dilution of the stock solutions. The pH values of all the

110 solutions are 7.28 ± 0.19 . The NaCl concentrations of solutions were adjusted by

111 adding 10 or 1% NaCl bulk solutions. The HA-PAM concentration is 0.1% in all the

112 related solutions in this paper unless special explanation. HA-PAM solutions

113 containing pyrene (about $3.03 \times 10^{-4} \text{ g}\cdot\text{L}^{-1}$) were prepared by adding 20 μL of 0.15114 $\text{g}\cdot\text{L}^{-1}$ pyrene ethanol solution to 10 ml of 0.1% HA-PAM aqueous solution.115 **2.2. Rheological measurement**

116 Rheological measurements were carried out on a Haake RS75 Rheometer

117 (Germany) with a Z41 Ti coaxial cylinder sensor system. All of the samples were
118 rested for more than 12 h to eliminate the bubbles before measurement.

119 In the steady-state shearing experiment, the viscosity was measured at the shear
120 rates ($\dot{\gamma}$) ranging from 0 to 1000 s⁻¹ with a check gradient of 0.5 ($\Delta\tau/\tau$)/ $\Delta t\%$ where τ
121 and t is the shear stress and time, respectively, and the maximum waiting time for
122 each shear rate step was 20 s. The frequency sweep measurements were carried out at
123 the angular frequency (ω) of 0.05–100 rad·s⁻¹ and stress of 0.04 Pa (in the linear
124 viscoelastic region). The equilibrium values of dynamic viscosity ($|\eta^*|$) and moduli
125 (G' and G'') were measured at $\omega = 0.2$ rad·s⁻¹ in the linear viscoelastic region. The
126 experiment temperature was controlled to be 25.0 ± 0.1 °C, unless the temperature
127 effect was investigated.

128 The “gain in viscosity”,¹⁹ S_η is defined to denote the interacting strength between
129 HA-PAM and LAA:

$$130 \quad S_\eta = \frac{\eta_{H/L}}{\eta_H} \quad (1)$$

131 where $\eta_{H/L}$ and η_H are the viscosity of H/L and HA-PAM solutions,⁷ respectively.
132 Herein, $S_\eta > 1$ means the interaction increases the solution viscosity; on the contrary,
133 the interaction causes the decrease of solution viscosity.

134 2.3. Fluorescence and surface tension measurements

135 Fluorescence spectra of solutions were detected with a Perkin-Elmer LS-55
136 spectrofluorometer using a quartz cuvette (1.0 cm × 1.0 cm). The excitation and
137 emission slit widths were set at 7.5 nm. The emission spectra were recorded from 350

138 to 650 nm with excitation wavelength fixed at 335 nm. The fluorescence intensity
139 ratio at 373 and 384 nm, namely, I_1/I_3 served as a parameter representing the
140 micropolarity of aggregates. Pyrene was used as a fluorescent probe with the
141 concentration of $3.03 \times 10^{-4} \text{ g}\cdot\text{L}^{-1}$.

142 Surface tension measurements of various concentrations of LAA solutions were
143 performed on a K100 Processor Tensiometer (Krüss Co., Germany) using the
144 Wilhelmy ring at $30.0 \pm 0.1 \text{ }^\circ\text{C}$. Before each measurement, the plate was carefully
145 cleaned with deionized water and flamed. The surface tension of deionized water was
146 measured to calibrate the tensiometer and to check the cleanliness of the sample pool.

147 **2.4. DPD simulation**

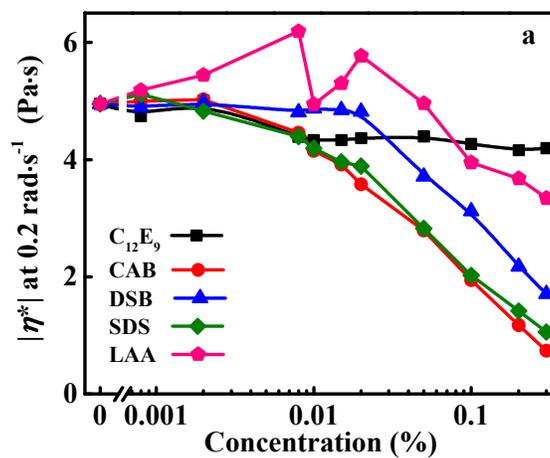
148 Simulations were carried out with the DPD package from Accelrys (version 5.0),
149 ²⁰. A cubic cell of size $20 \times 20 \times 20 R_c^3$ was built, where $R_c = 9.5$ was the cut-off
150 radius. 20000 time steps per simulation were set to assure the system attained
151 dynamic equilibrium. The number density and the spring constant between different
152 beads were 3.0 and 4.0, respectively. The temperature was 1.0 kBT according to
153 Groot's work.²¹ Flory-Huggins parameters (χ_{ij}) were obtained in Blend module. They
154 could be converted into the interaction parameters (a_{ij}) by the equation, $a_{ij} = \chi_{ij} \times 3.27$
155 + a_{ii} . Table 1 shows the interaction parameters a_{ij} used in this work. The diffusion
156 coefficient of a DPD particle was interpreted as the ratio of the distance that fluid
157 particles diffused through to the time they took.²⁰ Larger diffusion coefficient of water
158 molecules meant lower viscosity of solutions.

159

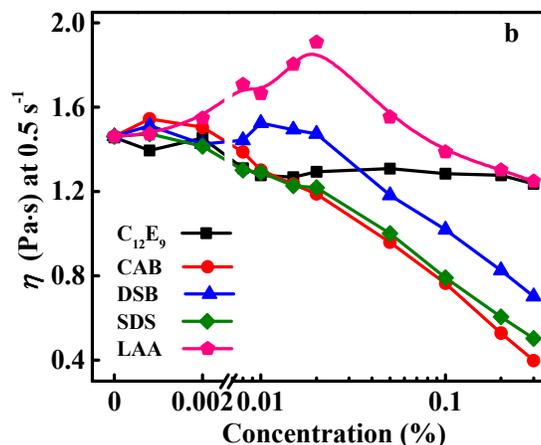
Table 1 Bead-Bead interaction parameters a_{ij} used in the simulation.

Bead	a	b	c	d	e	f	h	t	w
a	80								
b	22.9	25							
c	14	29.8	70						
d	75.4	79.1	47	25					
e	29	29.5	27.1	70.9	25				
f	90	45	16	73.3	55.3	100			
h	20	23.7	29.5	49.4	27.5	27.4	25		
t	51.7	55.9	32.3	33.1	53.5	66.1	36.2	25	
w	15	27	28	64.4	31.5	9.6	30.7	56.5	25

160

161 **3. Results and discussion**162 **3.1. Rheological properties of the mixed solutions of HA-PAM and various**163 **surfactants**

164



165

166 **Fig. 2** Influences of various surfactants on dynamic and shear viscosity of 0.1%

167

HA-PAM solutions.

168

169 Fig. 2 shows the variation of the viscosity of 0.1% HA-PAM solutions with

170 concentrations of CAB, DSB, LAA, C₁₂E₉ and SDS. The critical micelle171 concentrations of these surfactants are 0.045, 0.072, 0.010 (Fig. S2), 0.0058²² and172 0.23%,²³ respectively. C₁₂E₉ exhibits little effect on the viscosity of HA-PAM

173 solutions. The ionic surfactants (CAB, DSB and SDS) all induce a slight increase,

174 followed by a decrease of HA-PAM solution viscosity with increasing surfactant

175 concentrations. The slight increase of the viscosity results from the formation of

176 surfactant micelle bridges between different polymer chains. Compared with the

177 nonionic surfactant C₁₂E₉, the ionic surfactants induce a more abrupt viscosity

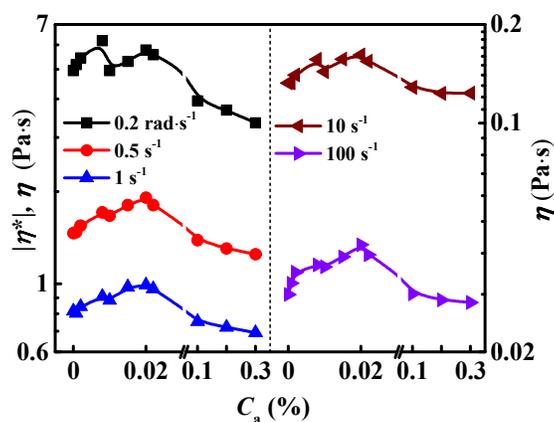
178 decrease of HA-PAM solutions at the surfactant concentration higher than 0.1%,

179 which might be because of the increase of inter- molecular electrostatic repulsion

180 induced by the co-aggregation of ionic surfactant and the screening effect of excess

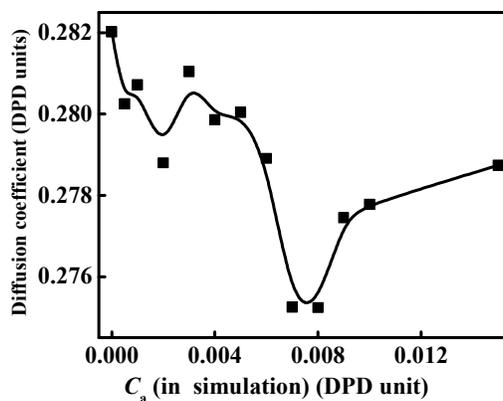
181 counter ions.²⁴⁻²⁶ Being different from the above systems, the HA-PAM/LAA (H/L)
 182 combined system exhibits the strongest thickening ability at the concentration (C_a)
 183 less than 0.08%. Two viscosity maximums are observed at $C_a = 0.008$ and 0.02%,
 184 respectively.

185 **3.2. Effects of LAA concentration C_a and $\dot{\gamma}$ on the rheological properties of H/L**
 186 **solutions**



187
 188 **Fig. 3** Effect of LAA concentration (C_a) on dynamic and steady viscosity of HA-PAM
 189 solutions at different shear rates.

190



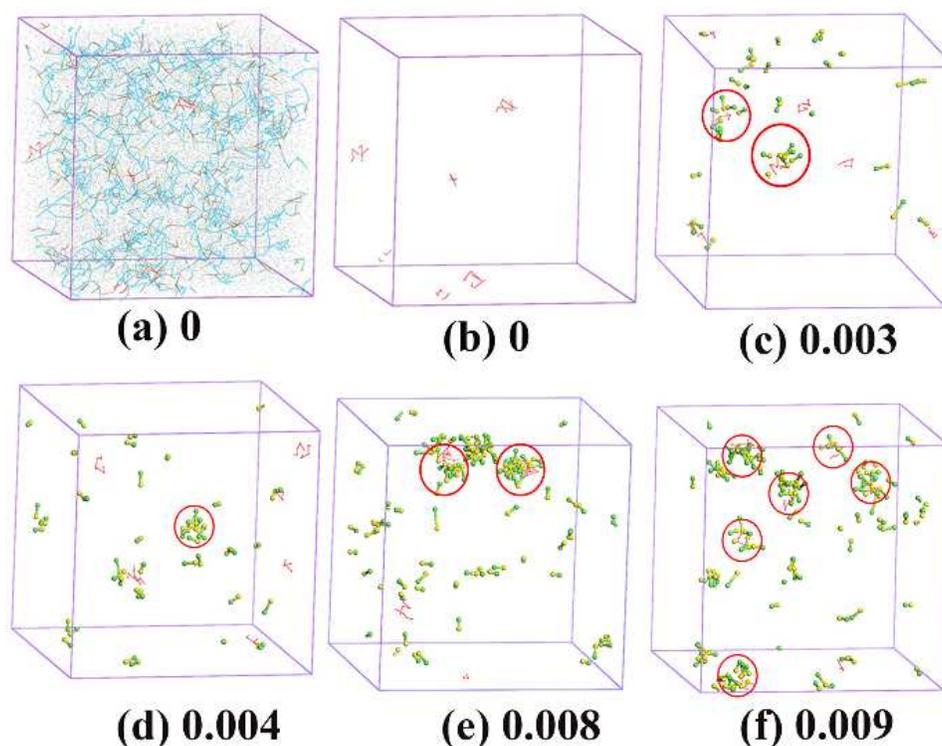
191

192 **Fig. 4** Water diffusivity in the aqueous system as a function of LAA concentration at a
193 shear rate of 0.005 in the simulation.

194

195 Fig. 3 shows the dynamic and steady viscosity of H/L solutions as a function of
196 $\dot{\gamma}$ and C_a . Similar viscosity variation trend with increasing C_a are observed at various
197 $\dot{\gamma}$ (Figs. 2 and 3), but the first viscosity maximum is less obvious under the steady
198 shear than the oscillation shear (Figs. 2 and 3). DPD simulation results of the H/L
199 solutions also show two minimums of the diffusion coefficient of water molecules in
200 the systems with increasing alkanolamide concentration in the simulation (C_a (in
201 simulation)) (Fig. 4), which agreed very well with the experiment results, indicating
202 that the bead-bead interaction parameters a_{ij} used in the simulation are reasonable.

203



204

205 **Fig. 5** Configurations of H/L at shear rate of 0.005 and various LAA concentrations:

206 (a) 0 with all the beads; (b) 0; (c) 0.003; (d) 0.004; (e) 0.008; (f) 0.009. Colors of the

207 beads are the same as those in Fig. 1: a (sodium acrylate, golden yellow); b

208 (acrylamide, light blue); c and d (hydrophobically associating monomer, orange and

209 red); and e and f (sodium 2-acrylamido-2-methylpropanesulfonate, brown and pink);

210 h and t (surfactant head and tail, green and yellow) in the simulation;

211

212 According to simulation results shown in Fig. 5, the hydrophobic side chain of

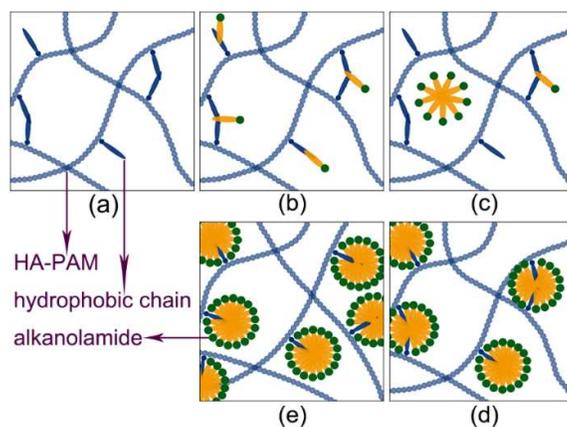
213 the HA-PAM molecules cluster to form microdomains (Fig. 5b). When LAA was

214 added to the HA-PAM solutions, the LAA molecules aggregate around these

215 hydrophobic domains (Fig. 5c), and mixed micelle-like associations containing LAA

216 molecules and hydrophobic segments of different HA-PAM molecules were formed.
 217 Herein, the LAA acts as crosslinkers which contributes to the increase of the
 218 intermolecular hydrophobic interaction, resulting in the viscosity increase of mixed
 219 solutions (Fig. 3). Fig. S3 shows the fluorescence spectra of the H/L solutions at
 220 different C_a . The increase of I_3/I_1 values with increasing C_a also means that more
 221 hydrophobic microdomains were formed.

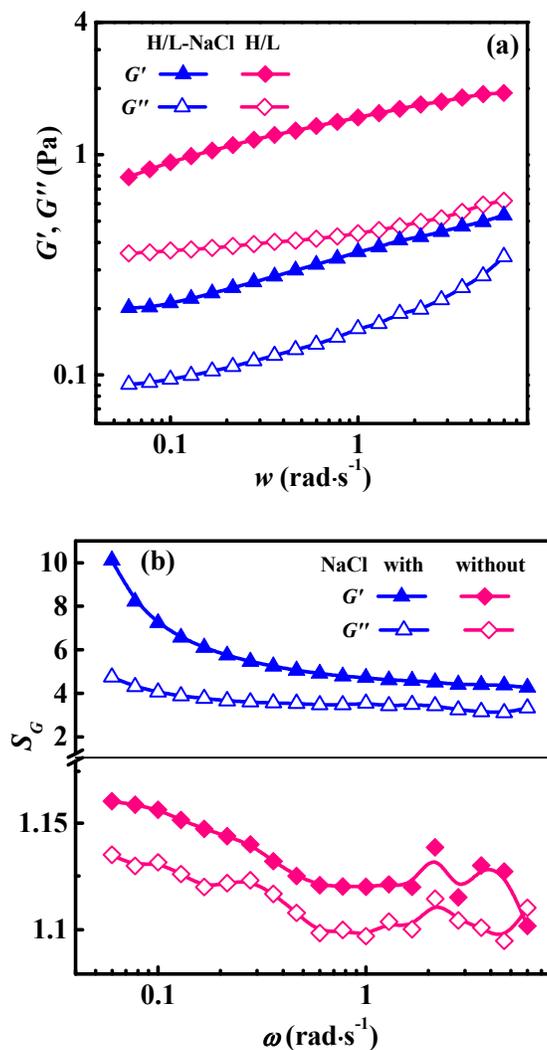
222 When the LAA concentration increases, independent LAA micelles were found
 223 in the bulk phase (Fig. 5d), the desorption of LAA from the hydrophobic
 224 microdomains of the polymer molecules may lead to the decrease of the viscosity of
 225 the mixed solutions, as shown in Fig. 3. As the LAA concentration increases further,
 226 independent micelle were formed around the hydrophobic side chains of the HA-PAM
 227 (Fig. 5f).²⁷⁻²⁹ The intermolecular association of HA-PAM molecules induced by
 228 coaggregation of the hydrophobic side groups³⁰ would be broken, leading to the
 229 viscosity decrease of the mixed solutions, corresponding well with the experimental
 230 results in Fig. 3.



232 **Fig. 6** Schematic illustration of H/L microstructures at different LAA concentrations.

233

234 By comparing the experimental and simulation results, we conclude that the
235 LAA molecules aggregate together with the hydrophobic side chain of the HA-PAM
236 molecule, and the simulation results show that the formed aggregates containing LAA
237 micelles and hydrophobic segments of HA-PAM molecules (Fig. 5e) behave like
238 crosslinkers to bind the HA-PAM molecules together and strengthen the network
239 structure, leading to an increase of the viscosity of the bulk solutions (Fig. 3).²⁷⁻²⁹ The
240 whole process of the effect of the LAA concentration on the micromolecular behavior
241 and the properties of the H/L mixed systems were shown in Fig. 6. At $C_a < 0.008\%$,
242 the LAA molecules get clustered with the hydrophobic segments of HA-PAM (Fig.
243 6b). At $0.008\% < C_a < 0.01\%$, the LAA molecules tend to form micelles in bulk
244 solution (Fig. 6c). At $0.01\% < C_a < 0.02\%$, the micelle-like associations containing
245 LAA molecules and hydrophobic segments from different HA-PAM molecules were
246 formed (Fig. 6d), and at $C_a > 0.02\%$, independent micelles were formed around each
247 hydrophobic side chains of the HA-PAM (Fig. 6e).²⁷⁻²⁹ The intermolecular association
248 of HA-PAM molecules induced by coaggregation of the hydrophobic side groups³⁰
249 would be broken, leading to the decrease of the viscosity of the mixed solutions. The
250 increasing I_3/I_1 value with C_a in Fig. S3 indicates the increase of the amount of
251 hydrophobic microdomains in solutions, corresponding to the variation of the
252 microstructure shown in Fig. 5 and 6.



253

254

255 **Fig. 7** (a) Frequency scan curves for H/L solutions containing 0.02% LAA and (b)

256 modulus ratios (S_G) of H/L to HA-PAM solutions⁷ with or without 0.2% NaCl.

257

258 In the EOR area, not only the viscosity but also the elasticity of the flooding

259 solutions are important in the oil displacement process.³¹ The frequency scan curves

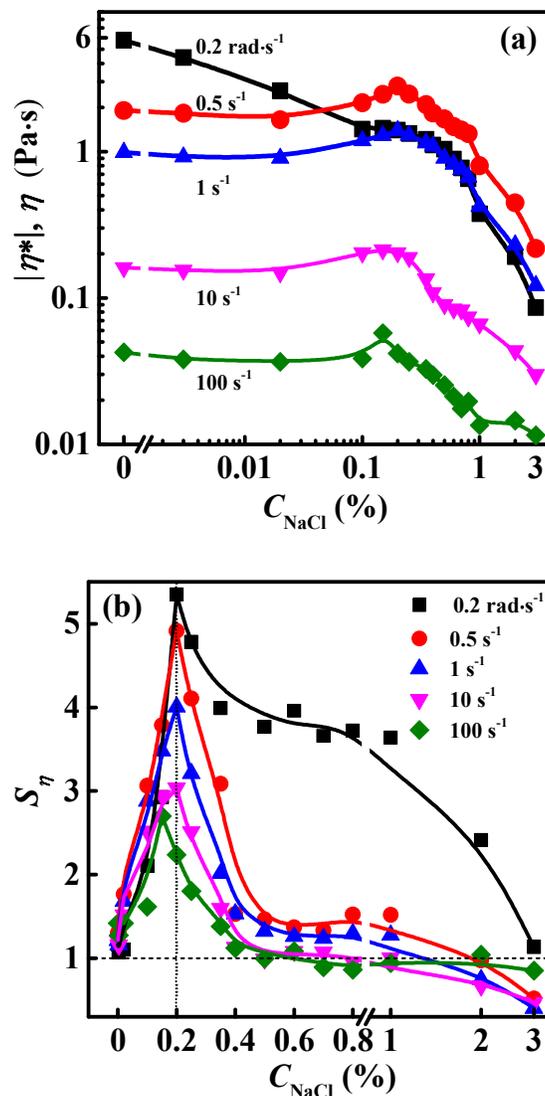
260 of H/L solutions at $C_a = 0.02\%$ are shown in Fig. 7a. The G' are higher than G'' in the

261 studied ω range, indicating the dominant elastic behavior of the solution. Fig. 7b

262 shows the modulus ratios (S_G) of the H/L mixed solutions to HA-PAM solutions. Both
263 $S_{G'}$ and $S_{G''}$ are larger than 1, and the phase angle tangent values of the H/L mixed
264 solutions are less than those of the HA-PAM solutions (Fig. S4), indicating the
265 hydrophobic association interaction between HA-PAM and LAA also induces the
266 enhancement of the elasticity of the solutions.

267 The coaggregation of the surfactants and hydrophobic side chains of the
268 HA-PAM plays the key role in formation of the network to maintain the high
269 viscosity and elasticity of the mixed solutions. Being different from the H/L mixed
270 system, the viscosity of the mixed solutions of $C_{12}E_9$ and HA-PAM changed slightly,
271 as shown in Figure 2. The HLB of $C_{12}E_9$ was higher than LAA, and the driven force
272 for hydrophobic association is weaker for $C_{12}E_9$ comparing with LAA. It is concluded
273 that suitable hydrophilic-lipophilic balance is needed for the surfactants to get
274 coaggregated beside the hydrophobic side chains of the HA-PAM.

275 3.3. Effect of NaCl on the rheological properties of H/L solutions



276

277

278 **Fig. 8** (a) Dynamic (■) and steady viscosity of H/L solutions containing 0.1%
 279 HA-PAM and 0.02% LAA and (b) interaction strength (S_η) of HA-PAM with LAA at
 280 various NaCl concentrations and shear rates.

281

282 Fig. 8a shows the variation of dynamic and steady viscosity of H/L solutions
 283 containing 0.02% LAA as a function of the NaCl concentration (C_{NaCl}). With
 284 increasing C_{NaCl} , the viscosity of the H/L solutions exhibits the same variation trend as

285 the HA-PAM solutions.⁷ But for the H/L solutions, a prominent viscosity
286 enhancement is observed in the presence of NaCl at $\dot{\gamma} = 0.5\text{--}100\text{ s}^{-1}$ (Fig. S5). The S_η
287 values between HA-PAM and LAA at different C_{NaCl} are shown in Fig. 8b. With
288 increasing C_{NaCl} , S_η increases first and then decreases, with a maximum observed at
289 about $C_{\text{NaCl}} = 0.2\%$ where the H/L solutions exhibit the largest viscosity (Fig. 8a). S_η
290 can decrease to be less than 1 only at very high C_{NaCl} . So the H/L binary system has
291 better salt-tolerance than HA-PAM.

292 The addition of NaCl enhanced the hydrophobic interaction between
293 hydrophobic chains of HA-PAM and LAA (Fig. 5e).³² With increasing NaCl
294 concentration, the salting-out effect disturbs the hydration of the polymer and
295 surfactant, and thereby promotes the hydrophobic association, so S_η increases
296 continuously with the increase of C_{NaCl} at $C_{\text{NaCl}} < 0.2\%$. Meanwhile, apart from the
297 hydrophobic units of the HA-PAM, there are several charged groups such as COO^- ,
298 SO_3^- , and N^+ in the molecules. The screening effect of NaCl can decrease the
299 intramolecular electrostatic repulsion caused by these charged groups, leading to the
300 curling of the polymer molecules.^{34, 35} The molecular curling enhances the
301 intramolecular and impairs the intermolecular hydrophobic interactions between
302 HA-PAM molecules³⁵, and simultaneously decreases the hydrophobic interaction
303 between HA-PAM and LAA molecules, which makes the viscosity of H/L solutions
304 decrease slightly at $C_{\text{NaCl}} < 0.02\%$ (Fig. 8a), and S_η decrease significantly with
305 increasing C_{NaCl} at $C_{\text{NaCl}} > 0.2\%$. $S_\eta < 1$ at higher C_{NaCl} is probably because the LAA

306 participates in the intramolecular hydrophobic interaction of HA-PAM molecules and
307 increases the curling of HA-PAM molecules.

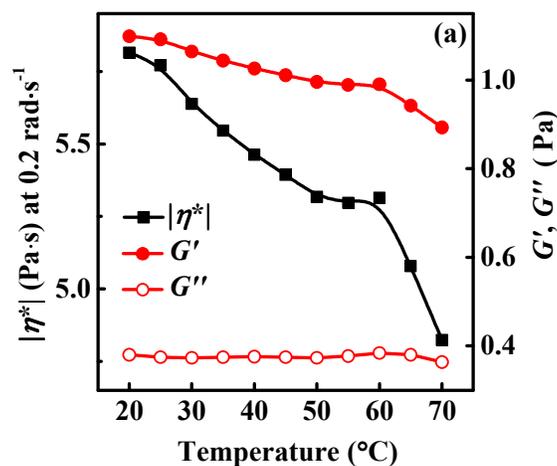
308 The $\dot{\gamma}$ also has a great effect on the viscosity of H/L solutions and S_η . With
309 increasing $\dot{\gamma}$, the steady viscosity of the solutions gradually decreases (Fig. 8a). At
310 $C_{\text{NaCl}} < 0.04\%$, the $|\eta^*|$ of the solutions is larger than the steady viscosity at $\dot{\gamma} =$
311 $0.5\text{--}100\text{ s}^{-1}$, while at $C_{\text{NaCl}} > 0.04\%$, the steady viscosity at $\dot{\gamma} = 0.5\text{ s}^{-1}$ is larger than
312 the $|\eta^*|$ of the solutions, which shows that the addition of NaCl enhances the
313 shear-tolerance of the system. The shear could induce the variation of the molecular
314 conformation and coaggregation behavior. On one hand, the shear can destroy the
315 inter/intra-molecular hydrophobic interactions³⁶. On the other hand, the shear can
316 inhibit the macromolecular curl. At low C_{NaCl} , the intramolecular hydrophobic
317 interaction in the H/L system is weak, so the shear principally destroy the
318 intermolecular hydrophobic interaction, which leads to the decrease of solution
319 viscosity. At high C_{NaCl} , the screening effect of NaCl induces the severe curl of
320 HA-PAM molecules, as discussed above. The shear mainly destroys the
321 intramolecular hydrophobic interaction and dissociates the LAA molecules from the
322 intramolecular hydrophobic microdomains, which makes the steady viscosities of
323 solutions at $\dot{\gamma} = 0.5\text{ s}^{-1}$ higher than the $|\eta^*|$.

324 The frequency scan curves of H/L solutions with $C_a = 0.02\%$ and $C_{\text{NaCl}} = 0.2\%$
325 and S_G values are shown in Fig. 7. $G' > G''$ indicates the solution exhibits a dominant
326 elastic behavior. The $S_{G'}$ and $S_{G''}$ are larger than 1, and the phase angle tangent values

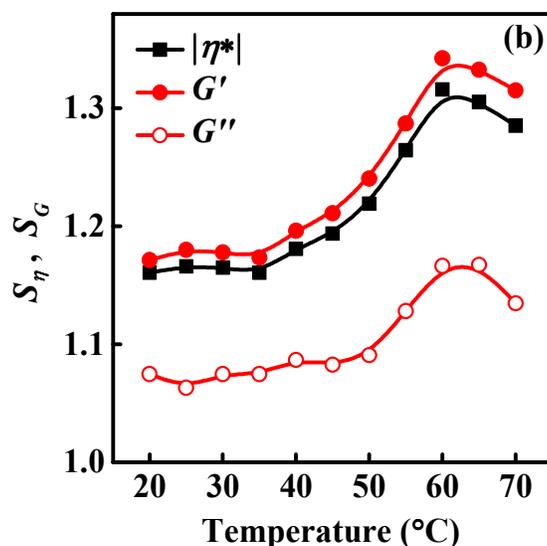
327 of the H/L solutions are less than those of the HA-PAM solutions (Fig. S4), indicating
328 the hydrophobic association between HA-PAM and LAA increases the elasticity of
329 solutions. The $S_{G'}$ and $S_{G''}$ values of solutions with NaCl are much larger than those
330 without NaCl (Fig. 7b), which reveals that in the presence of NaCl, the elasticity
331 enhancement of solutions induced by the hydrophobic interaction between HA-PAM
332 and LAA is more prominent.

333 Overall, the LAA contributes to the viscosity and elasticity enhancement of
334 HA-PAM solutions at $C_{\text{NaCl}} < 1\%$, indicating the salt tolerance of H/L solutions is
335 better than that of the HA-PAM solutions.

336 3.4. Effect of temperature on rheological properties of H/L solutions



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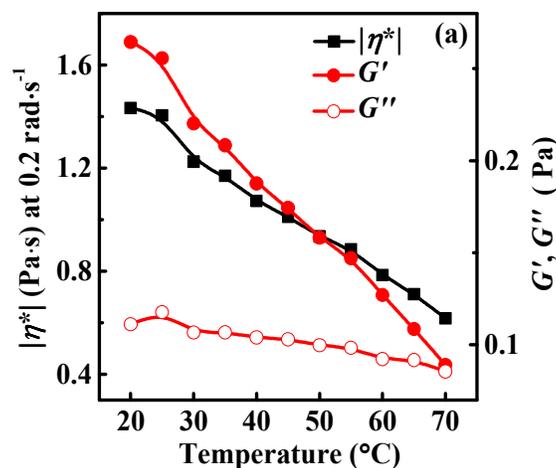
339 **Fig. 9** Effect of temperature on (a) dynamic viscosity and moduli of H/L solutions
 340 containing 0.02% LAA, and (b) viscosity and modulus ratios (S_η and S_G) of H/L to
 341 HA-PAM solutions

342

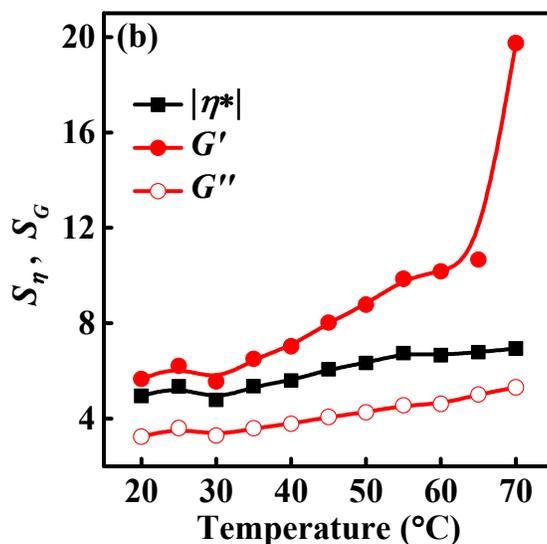
343 Temperature (T) can substantially influence the rheological properties of the
 344 hydrophobic associating polymer solutions.³⁷ Fig. 9a shows the variation of $|\eta^*|$, G'
 345 and G'' with T for the H/L solutions with $C_a = 0.02\%$. G' is always higher than G'' in
 346 the tested temperature range for all solutions, indicating a dominant elastic behavior.
 347 With increasing T , the $|\eta^*|$ and G' of H/L solutions exhibit the similar trend as the
 348 HA-PAM solutions,⁷ i.e. gradual decrease, except a plateau appears at $T = 50\text{--}60$ °C,
 349 respectively. The T has little effect on the G'' of solutions. The decrease of $|\eta^*|$ and G'
 350 of solutions with increasing T is because high temperature accelerates the
 351 macromolecular movement, thus the hydrated layer and destroys the intermolecular
 352 interaction.³⁸ The appearance of plateaus reveals the decrease of $|\eta^*|$ and G' is

353 retarded (Fig. S6), further indicating that the hydrophobic interaction between
 354 HA-PAM and LAA contributes to the viscosity enhancement of solutions.

355 In order to assess the effect of T on the interaction strength of HA-PAM with
 356 LAA, S_η and S_G of solutions are shown in Fig. 9b. In the studied T range, the S_η and S_G
 357 are always larger than 1, indicating the interaction between HA-PAM and LAA leads
 358 to a viscosity increase. With increasing T , both S_η and S_G first increase and then
 359 decrease with a maximum observed at $T = 60$ °C. As is reported, the hydrophobic
 360 association is an endothermic process,^{38, 39} and the increase of temperature is
 361 favorable for hydrophobic interaction enhancement. Thus, The S_η and S_G increase with
 362 T at $T < 60$ °C. The reduction of S_η and S_G at $T > 60$ °C is caused by the strong thermal
 363 motion of the molecules which destroys the interaction between HA-PAM and LAA.



364



365

366 **Fig. 10** Effect of temperature on (a) dynamic viscosity and moduli of H/L solutions
 367 containing 0.02% LAA, (b) viscosity and modulus ratios of H/L to HA-PAM solutions,
 368 with NaCl concentration of 0.2%.

369

370 In the presence of NaCl, the effects of T on the rheological properties of H/L
 371 solutions, S_η and S_G are different from the results without NaCl. As shown in Figs. 10a
 372 and S7, at $C_{\text{NaCl}} = 0.2\%$, the $|\eta^*|$, G' and G'' decrease more prominently with
 373 increasing T than those without NaCl (Figs. 9a and S6), and no plateau is observed,
 374 which is similar with those for the pure HA-PAM solutions.⁷ But G' is always larger
 375 than G'' in the tested temperature range for the H/L solutions, indicating a dominant
 376 elastic behavior. While for the HA-PAM solutions, G' is larger than G'' at $T =$
 377 20–35 °C and less than G'' at $T = 35–70$ °C. Fig. 10b shows the changes of S_η and S_G
 378 with T . All S_η , S_G and $S_{G''}$ gradually increase with the T in the studied range, which is
 379 different from the results without NaCl (Fig. 9b). Besides, as the T increases from 20

380 to 60 °C, the S_{η} , $S_{G'}$ and $S_{G''}$ of solutions with NaCl increases by ~35, ~80 and ~42%,
381 respectively, while those without NaCl increases by ~13, ~15 and ~9%, respectively,
382 indicating that the hydrophobic interaction between HA-PAM and LAA was
383 enhanced in the presence of NaCl.

384 4. Conclusions

385 The rheological behavior of binary mixed solution of a hydrophobically
386 associating polyacrylamide (HA-PAM) and surfactants was investigated. Compared
387 with anionic surfactant SDS, zwitterionic surfactant carbobetaine ($C_{19}H_{38}N_2O_3$)
388 sulfobetaine ($C_{17}H_{37}NSO_3$) and nonionic surfactant nonaethylene glycol monododecyl
389 ether ($C_{30}H_{62}O_{10}$), the nonionic surfactant laurel alkanolamide (LAA) exhibited
390 stronger thickening ability when added to the HA-PAM solutions. The DPD
391 simulation was used to interpret the synergistic mechanism. The obvious increase of
392 solution viscosity was induced by the hydrophobic interaction between HA-PAM and
393 LAA. The effects of electrolyte concentration and temperature on the interaction
394 between HA-PAM and LAA were also investigated. The interaction strength of
395 HA-PAM with LAA enhanced with the increase of C_a , C_{NaCl} and temperature. Overall,
396 the HA-PAM/LAA mixed solution exhibited stronger salt and temperature resistance
397 than the HA-PAM solution. The HA-PAM/LAA systems are appropriate for use in
398 EOR of oil field with high salinity.

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403 **References**

- 404 1. H. Zhang, M. Dong and S. Zhao, *Energ. Fuel.*, 2010, **24**(3), 1829-1836.
- 405 2. D. A. Z. Wever, F. Picchioni and A. A. Broekhuis, *Prog. Polym. Sci.*, 2011,
406 **36**(11), 1558-1628.
- 407 3. E. E. Kathmann, L. A. White and C. L. McCormick, *Polymer*, 1997, **38**(4),
408 871-878.
- 409 4. A. Sabhapondit, A. Borthakur and I. Haque, *Energ. Fuel.*, 2003, **17**(3),
410 683-688.
- 411 5. C. Zhong, L. Jiang and X. Peng, *J. Polym. Sci. Polym. Chem.*, 2010, **48**(5),
412 1241-1250
- 413 6. C. L. McCormick, J. C. Middleton and D. F. Cummins, *Macromolecules*,
414 1992, **25**(4), 1201-1206.
- 415 7. Q. Deng, H. Li, Y. Li, X. Cao, Y. Yang and X. Song, *Aust. J. Chem.*, 2014,
416 **67**(10), 1396-1402.
- 417 8. P. A. Ioannis S. Chronakis, *Macromolecules*, 2001, **34**(14), 5005-5018.
- 418 9. L. Y. C. Perry F. C. Lim, S. B. Chen, *J. Phys. Chem. B* 2003, **107**(206),
419 6491-6496.
- 420 10. M. Y. Thomas A. P. Seery, Thieo E. Hogen-Esch, Eric J. Amis',
421 *Macromolecules*, 1992, **25**(18), 4784-4791.

- 422 11. X. Xin, G. Xu, D. Wu, Y. Li and X. Cao, *Colloid Surf. A*, 2007, **305**(1–3),
423 138-144.
- 424 12. I. Hoffmann, P. Heunemann, S. Prevost, R. Schweins, N. J. Wagner and M.
425 Gradzielski, *Langmuir*, 2011, **27**(8), 4386-4396.
- 426 13. K. Fukada, E. Suzuki and T. Seimiya, *Langmuir*, 1999, **15**(12), 4217-4221.
- 427 14. P. Deo, N. Deo, P. Somasundaran, A. Moscatelli, S. Jockusch, N. J. Turro, K.
428 Ananthapadmanabhan and M. F. Ottaviani, *Langmuir*, 2007, **23**(11),
429 5906-5913.
- 430 15. A. Hersloef, L. O. Sundeloef and K. Edsman, *J. Phys. Chem.*, 1992, **96**(5),
431 2345-2348.
- 432 16. N. Plucktaveesak, A. J. Konop and R. H. Colby, *J. Phys. Chem. B*, 2003,
433 **107**(32), 8166-8171.
- 434 17. D. Zhuang, T. E. Hogen - Esch and Y. Zhang, *J. Appl. Polym. Sci.*, 2004,
435 **92**(2), 1279-1285.
- 436 18. S. Zhang, P. Zhu, Y. Sun, Y. Yang, X. Cao, X. Song and Y. Li, *RSC Adv.*,
437 2014, **4**(79), 41831-41837.
- 438 19. I. Iliopoulos, J. L. Halary and R. Audebert, *J. Polym. Sci. Polym. Chem.*, 1988,
439 **26**(1), 275-284.
- 440 20. X. Cao, G. Xu, Y. Li and Z. Zhang, *J. Phys. Chem. A*, 2005, **109**(45),
441 10418-10423.
- 442 21. Y. Zhao, L.-Y. You, Z.-Y. Lu and C.-C. Sun, *Polymer*, 2009, **50**(22),

- 443 5333-5340.
- 444 22. K. S. Sharma, P. A. Hassan and A. K. Rakshit, *Colloid Surf. A*, 2007, **308**(1 -
- 445 3), 100-110.
- 446 23. S. Ghosh, S. Mondal, S. Das and R. Biswas, *Fluid. Phase. Equilibr*, 2012,
- 447 **332**(0), 1-6.
- 448 24. K. Y. Mya, A. Sirivat and A. M. Jamieson, *J. Phys. Chem. B*, 2003, **107**(23),
- 449 5460-5466.
- 450 25. W. Seng, K. Tam and R. Jenkins, *Colloid Surf. A*, 1999, **154**(3), 365-382.
- 451 26. M. Amiri and A. A. Dadkhah, *Colloid Surf. A*, 2006, **278**(1), 252-255.
- 452 27. K. Tam, W. Seng, R. Jenkins and D. Bassett, *J. Polym. Sci. Poly. Phys.*, 2000,
- 453 **38**(15), 2019-2032.
- 454 28. G. Zhao, C. C. Khin, S. B. Chen and B.-H. Chen, *J. Phys. Chem. B*, 2005,
- 455 **109**(29), 14198-14204.
- 456 29. X. Xie and T. E. Hogen-Esch, *Macromolecules*, 1996, **29**(5), 1734-1745.
- 457 30. H. Bu, A.-L. Kjøniksen, K. D. Knudsen and B. Nyström, *Colloid Surf. A*,
- 458 2007, **293**(1), 105-113.
- 459 31. W. Wang, Y. Liu and Y. Gu, *Colloid Polym. Sci.*, 2003, **281**(11), 1046-1054.
- 460 32. J. Ma, B. Liang, P. Cui, H. Dai and R. Huang, *Polymer*, 2003, **44**(4),
- 461 1281-1286.
- 462 33. R. Sadeghi and F. Jahani, *J. Phys. Chem. B*, 2012, **116**(17), 5234-5241.
- 463 34. C. Zhong and P. Luo, *J. Polym. Sci. Poly. Phys.*, 2007, **45**(7), 826-839.

- 464 35. R. D. Wesley, C. A. Dreiss, T. Cosgrove, S. P. Armes, L. Thompson, F. L.
465 Baines and N. C. Billingham, *Langmuir*, 2005, **21**(11), 4856-4861.
- 466 36. K. Tam, W. Ng and R. Jenkins, *Polymer*, 2005, **46**(12), 4052-4059.
- 467 37. V. D. Alves, F. Freitas, N. Costa, M. Carvalheira, R. Oliveira, M. P.
468 Gonçalves and M. A. M. Reis, *Carbohydr. Polym.*, 2010, **79**(4), 981-988.
- 469 38. J. Ma, P. Cui, L. Zhao and R. Huang, *Eur. Polym. J.*, 2002, **38**(8), 1627-1633.
- 470 39. J. Desbrieres, *Polymer*, 2004, **45**(10), 3285-3295.
- 471