

# RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

*Accepted Manuscripts* are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxxx

ARTICLE TYPE

# Spontaneous formation of giant vesicles with tunable sizes based on jellyfish-like graft copolymers

Ke-Jing Gao,<sup>\*a</sup> Xiao-Zhou Liu,<sup>b</sup> Guangtao Li,<sup>c</sup> Bo-Qing Xu<sup>c</sup> and Jianjun Yi<sup>a</sup>

Received (in XXX, XXX) Xth XXXXXXXXX 20XX, Accepted Xth XXXXXXXXX 20XX

DOI: 10.1039/b000000x

For self-assembly studies, a series of “jellyfish-like” graft copolymers with shorter backbone and longer branch chain were adopted in this work. Chitooligosaccharide (COS), the oligomer of chitosan, was chosen as the hydrophilic short rigid backbone and poly( $\epsilon$ -caprolactone) (PCL) as the hydrophobic long branch chain. It was found that these special graft copolymers in 1,4-dioxane/water mixture could self-assemble into giant vesicles with diameter in the range of 0.5–54  $\mu\text{m}$ . The structure parameters of COS-g-PCLs have significant effect on their self-assembly behavior. The average number of PCL chains per COS unit and the average degree of polymerization of PCL branch chain are the key factors for the vesicles obtained. The more the number of PCL chains and the longer PCL chains there are on COS unit, the larger the vesicles become. In comparison to other copolymer vesicles, the obtained giant vesicles show unusual properties. One is that the polymer vesicles undergo sequential morphology changes upon heating. With the temperature increasing, the vesicles gradually disappear in the mix solvent. But with the temperature is down, the giant vesicles appear again, with the sharper size distribution compared to the original vesicle. The other property is that the polymer vesicles have a great number of amino and hydroxyl groups on their surface. These groups could be utilized for the further modification of the obtained giant vesicles, such as cross-linking of vesicles, chelating of metal ion to vesicles and so on.

## 1. Introduction

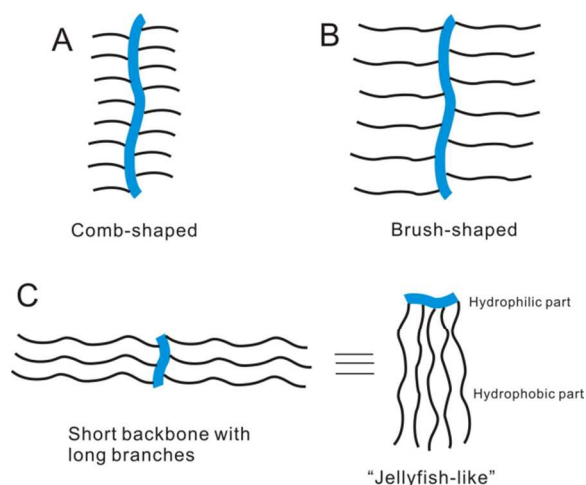
Supramolecular self-assembled vesicles have attracted great attention for their potential applications in drug delivery, gene therapy and model systems of biomembranes.<sup>1–3</sup> In biology, lipids and proteins can spontaneously assemble into vesicles (liposomes) or vesicle-like structures, because of the driving forces such as electrostatics and hydrophobicity.<sup>4,5</sup> Vesicles can also be formed from synthetic amphiphilic polymers.<sup>6–10</sup> Compared to liposomes, polymer vesicles possess unique properties such as good stability and broader range of accessible solvents.<sup>11</sup> Polymer vesicles have thus become attractive and promising research objects since the first observation of block copolymer vesicles by Eisenberg and co-workers,<sup>12</sup> who called the aggregations of the block copolymer with small hydrophilic fractions (<20%) “crew-cut” micelles. Discher et al.<sup>13</sup> reported a type of polymer giant vesicles formed from block copolymer and these were termed polymersomes. Polymer vesicles have also been prepared from other linear block copolymers such as polypeptides, rod-coil and rod-rod copolymers.<sup>13–19</sup> Compared to linear block copolymers, relatively few works were conducted on amphiphilic graft copolymers. In most cases, amphiphilic graft copolymers tended to form large compound spherical micelles (LCMs).<sup>20–23</sup> unimolecular micelles.<sup>24, 25</sup> In fact, the phenomena are due to the chemical structure of graft copolymers. Most of graft copolymers which were adopted to self-assemble usually focused on the traditional comb-shaped or brush-shaped graft

copolymer based on longer main chain as backbone and shorter side chains as graft segments. It is believed that these structures easily undergo inter-molecular entanglement among the longer main chains or intra-molecular association so that they tend to form large compound spherical micelles (LCMs). If the structure of amphiphilic graft copolymer could be properly manipulated to decrease the above mentioned influence, it should be possible for a graft copolymer to organize into abundant morphologies in selective solvents.<sup>26,27</sup>

So far, some attention has been devoted to developing the new kind of amphiphilic graft copolymer with a structure suitable for generating self-assembled morphologies other than LCMs.<sup>28–34</sup> By adjustment of grafting density or side chain length, it was found that some graft copolymers can self-assemble successfully into vesicles at certain condition.

Herein, we synthesized a new type of amphiphilic graft copolymer based on a shorter rigid chain as backbone and a longer chain as branch (Scheme 1c). Chitooligosaccharide (COS), the oligomer of chitosan, was chosen as the hydrophilic short rigid main chain and poly( $\epsilon$ -caprolactone) (PCL) as the hydrophobic long side chain. The chemical structure of COS-g-PCL obtained is quite different from the conventional comb-shaped or brush-shaped graft copolymers (Scheme 1a, b). We termed it “jellyfish-like” structure.<sup>34–38</sup> It was found that this type of graft copolymers in water/1,4-dioxane mixtures can self-assemble into vesicles with diameter in the range of 0.5–54  $\mu\text{m}$ . As far as we know, our research group is the first one to

demonstrate the giant vesicles self-assembled from the graft copolymers in solution.<sup>37</sup> In the present paper, the influences of the structure parameters of COS-g-PCL molecule on the vesicle morphologies have been examined. The properties of polymer vesicles obtained and the mechanism of self-assembly have also been investigated.



**Scheme 1** The illustration of the graft copolymers with different branch lengths.

## 2. Experimental section

### 2.1 Materials

Chitooligosaccharide (COS,  $M_n = 1800$ , the degree of deacetylation is 96%), purchased from Shenzhen Bright Way Novel Biomaterials Tech. Co. Ltd. (P. R. China), was dried at 60 °C under vacuum.  $\epsilon$ -Caprolactone (Acros Organics, 99%) was purified by vacuum distillation over  $\text{CaH}_2$  and the fraction collected at 96 - 98 °C was used in polymerization. Triethylaluminum was obtained from Fluka Company and used as a 1.0M solution in toluene. Chloroform and pyridine were purified by usual distillation method. Hexamethyldisilane (HMDS) and other reagents were used as received without further purification.

### 2.2 Synthesis of COS-g-PCL copolymers

The graft copolymer of COS-g-PCL was synthesized according to the previous literature.<sup>36,37</sup> Briefly, Chitooligosaccharide (COS, 5.0 g, 30.9 mmol pyranose) was placed in a dried glass reactor that was previously flushed with high purity  $\text{N}_2$  for 30 min. and added 50ml of Dimethyl sulfoxide (DMSO). Under vigorous stirring, the dispersion was heated at 70 °C for 6h to form a clear solution and then allowed to cool down to room temperature. Predetermined amount of hexamethyldisilazane (the molar ratio of hexamethyldisilazane to pyranose unit was in the range of 2.0 - 4.0) were added, and after being heated at 80 °C for 12h with stirring, the mixture was poured into 150ml of acetone. The precipitate was filtered, washed with water, and dried in vacuum to give pale tan powder. The degree of substitution (DS) of COS, i.e. the average number of  $-\text{Si}(\text{CH}_3)_3$  per COS unit, was calculated from the C/N value of elemental analysis.

1.0g trimethylsilylchitooligosaccharide (TMSCOS) with a certain degree of substitution was dissolved in Chloroform of 50

ml and then a certain amount of triethylaluminum in toluene (the molar ratio of  $\text{AlEt}_3$  to OH contained in TMS-COS was 1.2:1) was added dropwise via a syringe through a rubber septum. The reaction was kept for two hours and then a desired amount of the  $\epsilon$ -caprolactone monomer was transferred into the reactor. Under  $\text{N}_2$ , the graft copolymerization was allowed to proceed for 24h at ambient temperature while stirred. When the setting reaction time of 24 h elapsed, 5ml mixture of isopropyl alcohol/ $\text{H}_2\text{O}$ / $\text{HCl}$  was added and the reactor was stirred for further 4h to move the TMS group. The resulting polymer was purified by repeated precipitation (3 times) using 10ml of chloroform and 300 ml alcohol as a solvent and precipitant pair. It was dried for 72h at 50 °C in vacuum to a constant weight.

### 2.3 Measurement of COS-g-PCL copolymers

$^1\text{H-NMR}$  spectra were taken at room temperature in tetrahydrofuran-*d*8 or  $\text{D}_2\text{O}/\text{THF-}d_8$  in by a Bruker AV400 spectrometer operating at 400. Elemental analyses were performed using ThermoQuest EA1112 elemental analyzer with a combustion temperature at 950 °C. The molecular weight ( $M_n$ ) and molecular weight distribution ( $M_w/M_n$ ) were measured with a Waters 515 GPC instrument equipped with Styragel columns (101-, 102-, and 103-nm pore sizes) with tetrahydrofuran (THF) as the mobile phase. Polystyrene and Waters Millennium 32 were used as calibration standards and data-processing software, respectively.

### 2.4 The self-assembly of COS-g-PCL copolymers

The obtained COS-g-PCL copolymers were initially dissolved in 1,4-dioxane. Deionized water was added dropwise at a rate of 0.2% per minute to the COS-g-PCL solutions under vigorous stirring until the water contents were reached to ca. 25wt%. The solution were stirred for at least 2h except otherwise noted, and then observed with optical microscopy or TEM.

### 2.5 Characterization of Vesicle Morphology and Size

Environmental Scanning Electron Microscopy (ESEM) Measurements were recorded with model XL 30 ESEM FEG from Micro FEI Philips at room temperature. The lyophilized aggregate powders were deposited on a silicon wafer. A thin layer of gold was coated on the sample surface before measurement. Transferring Electron Microscopy (TEM) was performed on a JEOL JEM-1200 electron microscope at an acceleration voltage of 120KV. A drop of the dialyzed was deposited on a copper grid coated with a carbon film and stained with a drop of phosphorous tungstic acid in water (2 wt%). The copper grid was then dried at room temperature prior to measurement. Confocal Laser Scanning Microscopy (LSCM) measurements were conducted as follow: The pyrene solution in acetone (50mM, prepared prior to use) was added to the suspension of GVs in dioxane/water mixture to make a pyrene concentration of 5mM. Fluorescence images were recorded using the microscopy (Olympus BX 61, Olympus Optical Co., Tokyo, Japan) with a confocal optical scanner (Yokogawa Inc., Tokyo, Japan) and an argon ion laser with 405nm excitation wavelength. Phase contrast micrographs were recorded using the phase contrast microscopy (Olympus BX 51, Olympus Optical Co., Tokyo, Japan) with a digital CCD camera, and operated by the software provided by the camera manufacturer. The average size of vesicle

was calculated from 100 vesicles selected randomly.

## 2.6 Thermal Response Experiments

The GVs suspension was transferred into an observation chamber and observed with a phase-contrast microscopy (Olympus BX-51), equipped with a JVC color video camera and connected to an image-recording system. To determine the effect of temperature on the stability of GVs. Slow uniform heating was achieved with a resistive heating stage constructed in our lab, consisting of a backside metal coated microscope coverslip and a current limiting standard power supply. Cooling was achieved by simple heat dissipation after turning off the heating stage. Temperature in the droplet close to the vesicle was measured using a calibrated micron-sized Cu/Ni thermoelement. The temperature measurement was time synchronized with the time readings on the videotape.

## 3. Results and discussion

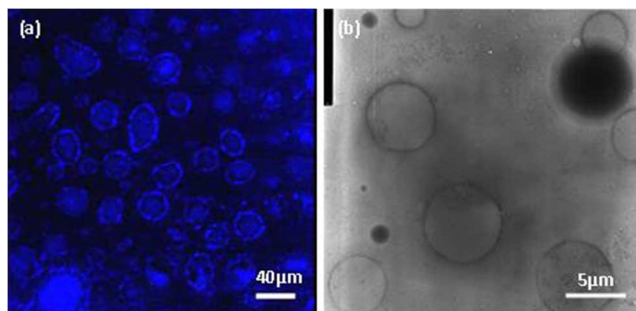
### 3.1 Vesicle formation dependence on the structure parameters of COS-g-PCLs

According to the synthesis method reported in the previous literature,<sup>36</sup> six COS-g-PCL samples with different structure parameters were synthesized (denoted as g1-6). The details of the characterization of COS-g-PCLs are listed in Table 1.

**Table 1** The molecular characteristics of COS-g-PCLs

Sample	DS <sup>(a)</sup> of COS	Mn <sup>(b)</sup>	DP <sup>(c)</sup> of CL	Diameter <sup>(d)</sup> /μm
g1	1.0	1.4×10 <sup>4</sup>	61	0.5
g2	1.0	2.6×10 <sup>4</sup>	100	35
g3	1.0	7.7×10 <sup>4</sup>	178	54
g4	0.5	1.4×10 <sup>4</sup>	71	Cluster
g5	0.5	2.9×10 <sup>4</sup>	118	31
g6	0.5	4.5×10 <sup>4</sup>	187	43
g7	0.2	5.2×10 <sup>4</sup>	261	Precipitation

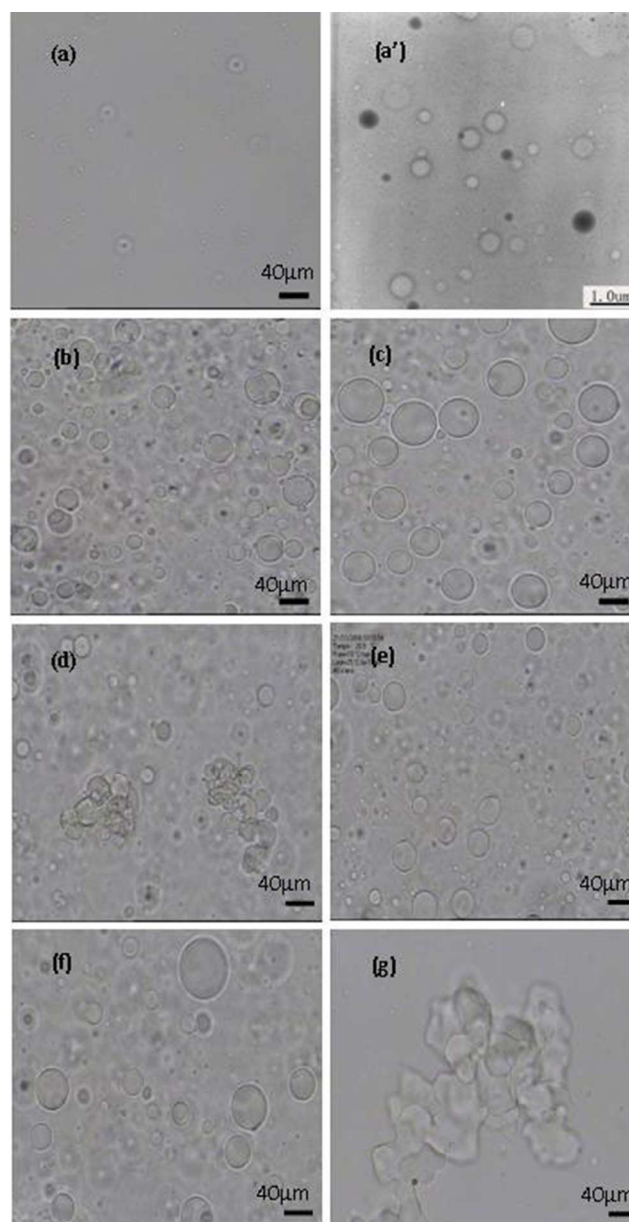
<sup>a</sup> The average number of PCL chains per COS unit, determined by elemental analysis. <sup>b</sup> Number average molecular weight of COS-g-PCL, determined by GPC as described in the text. <sup>c</sup> DP: the average degree of polymerization of PCL single branch attached to COS chain, calculated by <sup>1</sup>H NMR. <sup>d</sup> Determined visually from images for the average diameter of vesicles, based on the 100 vesicles.



**Fig. 1** Typical morphologies of COS-g-PCL vesicles. (a) LSCM images of fluorescence-labeled g2 vesicles, containing membrane-encapsulated pyrene in 1, 4-dioxane/water mixture (5mM, peak excitation = 405 nm). (b) TEM micrographs of the g2 vesicles. The samples for TEM measurement were negatively stained with 2% aqueous phosphorous tungstic acid solution.

Using water as selective solvent and 1, 4-dioxane as common solvent, the self-assembly behavior of COS-g-PCLs was

investigated under stirring according to the methodology of Eisenberg et al. to create “crew-cut” aggregates.<sup>39-44</sup> It was found that most of the jellyfish-like graft copolymers can self-assemble into vesicles with diameter great than 0.5 μm. Fig. 1a typically displays the micrographs of the fluorescence-labeled g2 aggregates observed by laser scanning focal microscope (LSCM). These images clearly illustrate the hollow interior of vesicular morphology, as evidenced by a significant decrease in fluorescence intensity toward the center of the spheres.<sup>45</sup> The argument is also supported by the TEM measurements. Fig. 1b shows a representative TEM image of the g2 aggregates negatively stained by phosphorous tungstic acid in water (2wt%). The contrast difference between the aggregate skin and the inner pool proved that the aggregates are vesicular in nature.



**Fig. 2** Phase contrast micrographs (a, b, c, d, e, f, g) and TEM image (a') of vesicles obtained from COS-g-PCL with different structure parameter. (a), (a') vesicles from g1 sample; (b) vesicles from g2 sample; (c) vesicles from g3 sample; (d) vesicle clusters from g4 sample; (e) vesicles from g5 sample; (f) vesicles from g6 sample; (g) precipitation from g7 sample.

The samples for TEM measurement were negatively stained with 2% aqueous phosphorous tungstic acid solution.

The morphologies of the aggregates are heavily dependent on the structure parameters of COS-g-PCL molecules. When DS of COS (the average number of PCL chains per COS unit) was 1.0, COS-g-PCLs can self-assemble into vesicles and the vesicle diameter increases with DP of CL (Table 1, Fig. 2a-c). When DP of CL was lower (sample g1 in Table 1), the diameters of vesicles obtained were less than 1 $\mu$ m. The low resolution of microscopy limits direct imaging of the aggregates (Fig. 2a). So, TEM was used to observe the vesicles (Fig. 2a'). When DP of CL was higher (sample g3 in Table 1), the diameter of vesicles grew up to more than 50 $\mu$ m. For COS-g-PCLs with DS of 0.5, the results were different slightly. When DP of CL was lower (sample g4 in Table 1), the vesicle-like aggregates were formed, accompanied frequently by clusters of vesicle-like aggregates (Fig. 2d). With the further increase in DP, the clusters disappeared and the vesicle diameter increased obviously (Fig. 2e-f). The appearing of cluster, resulting from vesicle aggregation, might attribute to the intervesicle hydrogen bonds due to a relative more -OH and -NH<sub>2</sub> groups embedded in their vesicles. For the COS-g-PCLs with higher DP of CL (sample g5, g6), the hydrophobic interaction among PCL branches may become the main driving force for self-assembly and the intervesicle hydrogen bonds are hard to build up. As a consequence, the discrete vesicles are the only observed assemblies (Fig. 2e-f). In addition to DP of CL, the DS of COS was crucial for the formation of vesicle. When the DS was decreased to 0.2, no vesicles were obtained although the copolymers have the higher DP of CL (sample g7 in table 1, Fig. 2g).

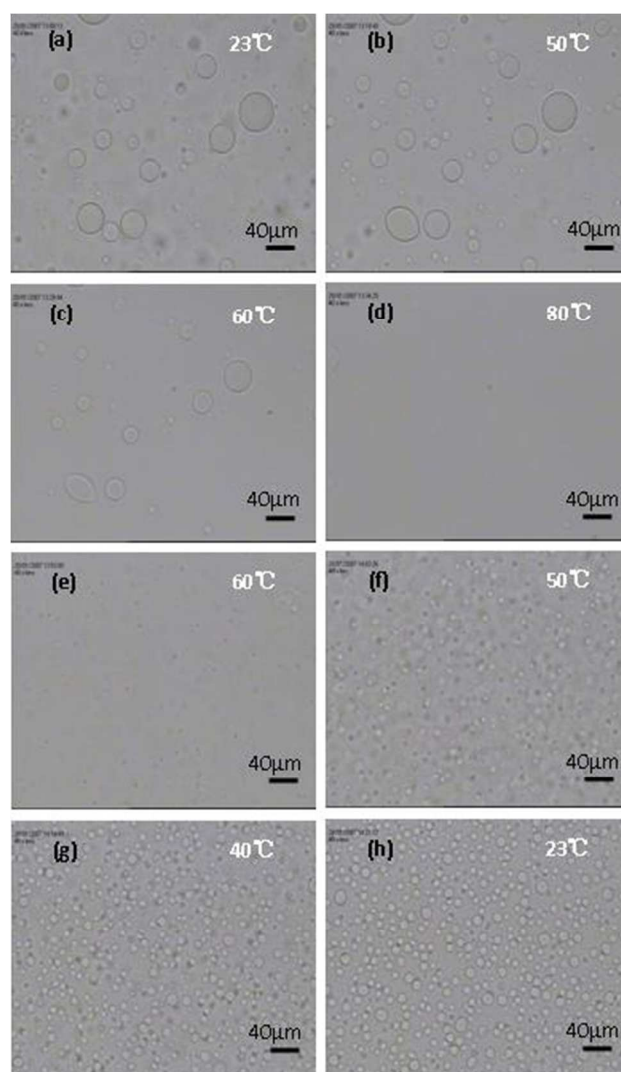
In short, the structure parameters of these copolymers exert a significant effect on formation of the aggregates. Only the COS-g-PCLs with the higher DP of CL and the higher DS of COS can self-assemble into giant vesicles with the diameter greater than 0.5 $\mu$ m.

### 3.2 Properties of the vesicles in solution

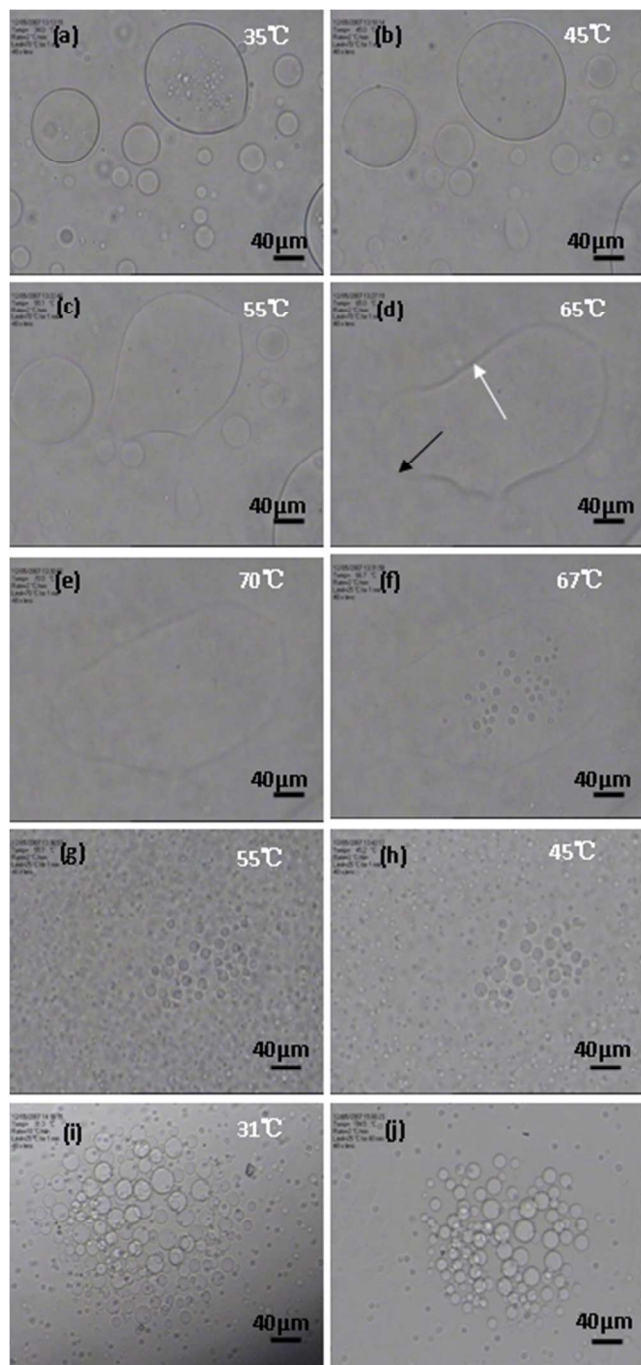
#### 3.2.1 Thermal response of COS-g-PCL vesicles

The stability of the giant vesicles originated from COS-g-PCLs was examined by measuring the vesicle diameter as a function of standing time. As a matter of fact, it remains unchanged after 3 months. Furthermore, the effect of temperature on the vesicle diameter was investigated. As shown in Fig. 3 and Fig. 4, when the temperature was below 50 °C, the vesicle diameter remains almost unchanged (Fig. 3a-b, Fig. 4a-b). Observed closely, however, the vesicles obtained from COS-g-PCL with different DP of CL demonstrate different response to heating. As the temperature was raised up to beyond 50 °C, the vesicles obtained from copolymer g2 with lower DP gradually dissolve in 1, 4-dioxane/water mixtures (Fig. 3c-d). When the temperature was cooled down to 50 °C, the vesicles were formed spontaneously and grew to bigger ones with the decrease of temperature (Fig. 3e-h). After cycling of heating and cooling, the vesicles with the sharper size distribution were obtained (Fig. 3h). In contrast, the vesicles prepared from copolymer g3 with higher DP show a different response to temperature. When the temperature was raised, the vesicles fused each other and precipitated from the solution onto the bottom of the vessel to form a membrane (Fig.

4a-e). The image in Fig. 4d provides an intermediate between vesicle and membrane, in which the part without a clear rim (black arrow) has changed into membrane and the part with a clear rim (white arrow) still keeps the vesicle shape. Interestingly, the deposited polymer membranes can self-assemble into the vesicles again when the solution temperature was dropped back (Fig. 4f-i). The similar sequential morphology transformation of polymer vesicles was only reported by Yan and co-worker, who study the LCST (the lower critical solution temperature) behavior of amphiphilic hyperbranched star copolymers having a hydrophobic hyperbranched poly[3-ethyl-3-(hydroxymethyl)oxetane] (HBPO) core and many hydrophilic polyethylene oxide (PEO) arms.<sup>46</sup> Fig. 4j displays the vesicle images after 3 rounds of heating and cooling. It demonstrates that the size of vesicles remain unchanged, which indicates that the deformation and formation of the polymer vesicles is reversible. It also means that we can produce the polymer vesicles with nearly monodisperse size by means of heating and cooling.



**Fig. 3** Caption sequence showing the heating (image a-d) and the cooling (image e-h) of g2 vesicles, which was observed with phase-contrast microscopy.



**Fig. 4** Sequence showing the heating (image a-e) and cooling (image f-i) of the vesicles from copolymer g3 and the morphologies (image j) of the vesicles after 2 rounds of heating and cooling. The images were observed with phase-contrast microscopy.

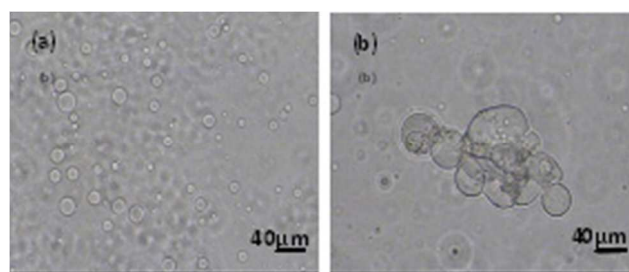
### 3.2.2 Reactive ability of COS-g-PCL vesicles

COS-g-PCL molecule consists of an amphiphilic COS backbone and many hydrophobic PCL branches, along with a great number of  $-OH$  and  $-NH_2$  groups on the COS backbone. By adding the selective solvent of water into COS-g-PCL solution in 1, 4-dioxane, the graft copolymers spontaneously self-organize into vesicles. In these vesicles, every COS-g-PCL molecule is speculated to separate into a hydrophobic PCL core and a hydrophilic corona carrying  $-OH$  and  $-NH_2$  groups.

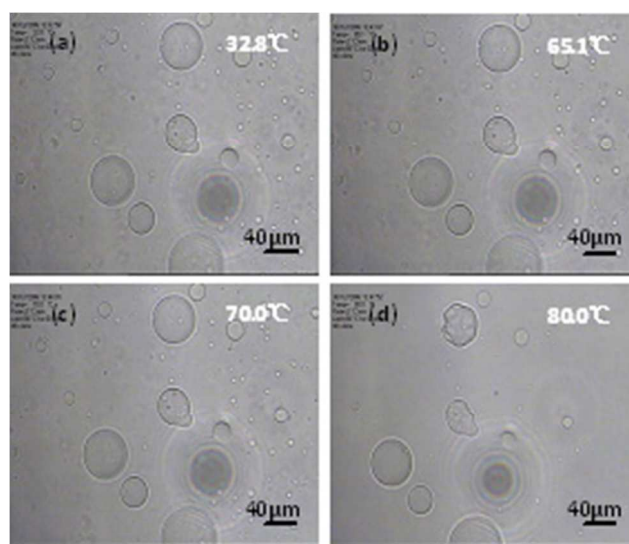
As mentioned above, the vesicles from copolymer g4

aggregate into clusters due to the intervesicle hydrogen bonds among the  $-OH$  and  $-NH_2$  groups embedded in these vesicles. These  $-OH$  and  $-NH_2$  groups have a strong affinity to  $Cu^{2+}$ . The intervesicle hydrogen bonds will be broke up in the presence of  $Cu^{2+}$ , and clusters will disappear. As expected, with the presence of  $Cu^{2+}$  in the vesicle suspension, no clusters were observed, only leading to discrete vesicles (Fig. 5a).

Cucurbit[n]uril (CB[n]  $n=5-8$ ), a family of macrocyclic compounds comprising  $n$  glycoluril units, has a hydrophobic cavity that is accessible through two identical carbonyl-fringed portals.<sup>47</sup> Supramolecular chemistry of CB[n], including their host-guest chemistry, has been studied extensively.<sup>48,49</sup> In particular, CB[6] and its derivatives have been found to form exceptionally stable host-guest complexes with compound containing  $-NH_2$  groups. Many supramolecular assemblies such as one- (1D), two- (2D), three-dimensional (3D) structures and molecular necklaces have been reported.<sup>50</sup> In our case, CB(6) in aqueous solution was introduced into the copolymer g2 vesicle suspension and stirred for 12h at room temperature. It was found that the vesicle clusters were generated owing to the strong affinity of CB(6) toward  $-NH_2$  groups with their two identical carbonyl-fringed portals (Fig. 5b).



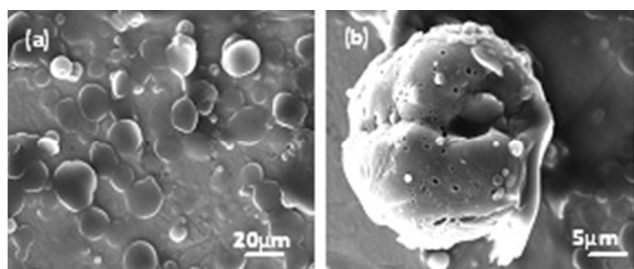
**Fig. 5** Phase contrast micrographs of COS-g-PCL aggregates under various conditions. (a) the aggregates in the presence of  $Cu^{2+}$  at 0.3 mM in the vesicle suspension, (b) the copolymer g2 aggregates in the presence of CB(6) at 3  $\mu M$  in the vesicle suspension.



**Fig. 6** Phase contrast micrographs (a, b, c, d) at different temperature after copolymer g2 vesicles were cross-linked with Glutaraldehyde. (a) 32.8  $^{\circ}C$ ; (b) 65.1  $^{\circ}C$ ; (c) 70.0  $^{\circ}C$ ; (d) 80.0  $^{\circ}C$ .

The  $-OH$  and  $-NH_2$  groups on the surface of vesicles could be used for further modification through covalent cross-linking,

which is a acceptable means of stabilization. To achieve this, a certain amount of Glutaraldehyde was added into the vesicle suspension and stirred for 4 h at temperature 40 °C. After cross-linking, the vesicle stability against heating was verified to promote greatly (Fig. 6). Even at high temperature of 70 °C, the vesicles remained their shapes (Fig. 6c). When temperature was increased to above 80 °C, the vesicles from copolymer g2 did not still dissolve in 1, 4-dioxane/water mixture in spite of the fact that all of them collapsed due to the escape of dioxane/water mixture inside the vesicles (Fig. 6d). SEM images also support the enhanced stability of vesicles. Due to the enhanced stability of vesicle, the vesicles could stand the tough condition during the TEM preparation and the most of vesicles were not destroyed although collapsed, which are shown in Fig. 7



**Fig. 7** The SEM images (a, b) after copolymer g2 vesicles were cross-linked with Glutaraldehyde. (a) the image in large-scale, (b) the image of the individual vesicle.

### 3.3 The mechanism of self-assembly

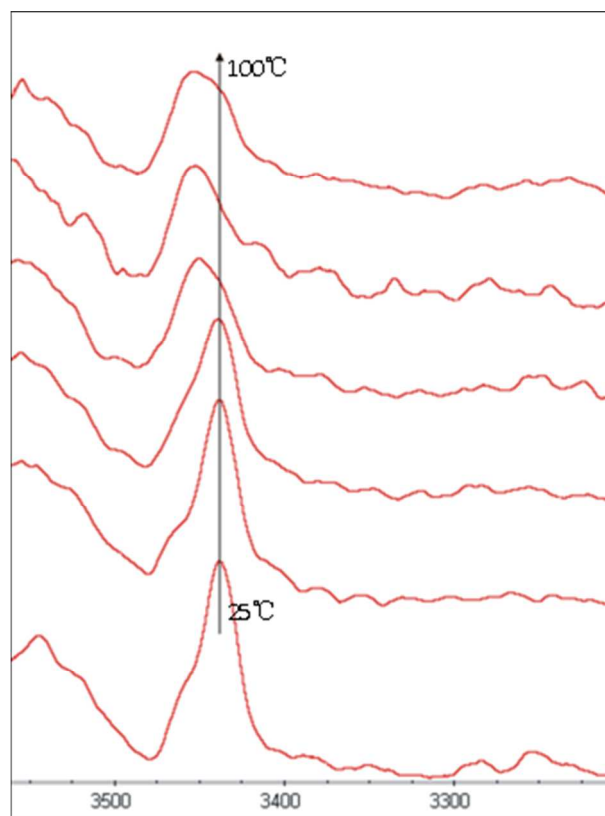
Taking account of the structure nature of COS-g-PCL, the new kind of graft copolymer has hydrophilic COS backbones and pendant hydrophobic PCL polyester chains. During the course of COS-g-PCL self-assembly driven by the hydrophobic interaction in 1,4-dioxane/water mixtures, the PCL branches would form the middle layer of the vesicle wall, and the COS backbones would form the outer layers.

On the other hand, there are a lot of hydroxyl and amino groups along the backbones of COS, the formation of hydrogen bonds is another potential interaction for the self-assembly. To verify this argument, trimethylsilyl chitooligosaccharide-graft-poly(caprolactone) (TMS-COS-g-PCL) were utilized to self-assemble. When the amino and hydroxyl groups were protected by trimethylsilylation, the formation of vesicles was not observed. Only ill-defined precipitation was detected (not shown).

Variable temperature FTIR was used to detect the hydrogen bonds in the resulting dried vesicles. The results are provided in Fig. 8. It was found that with increasing temperature from 25 °C to 80 °C, the peak of the hydroxyl groups gradually becomes weaker and shifts to the high wavelength side, which clearly indicates that a certain amount of hydroxyl groups have formed hydrogen bonds in the dried vesicles originated from the self-assembly of COS-g-PCL molecules.

As for the conformation of COS backbone in the aggregate, it is believed that they exist in parallel array to achieve as many hydrogen bonds as possible and the lowest energy.<sup>51</sup> Conventionally, amphiphilic graft copolymers with the same molecular weight might form self-assemblies with small size (on nanometer scale). For our system, the graft copolymers can form large vesicles (on micrometer scale) spontaneously. This may be

due to the tendency for the COS backbones to form parallel array driven by intra- and intermolecular hydrogen bonds originated from the groups of –OH and –NH<sub>2</sub>. Therefore, in the present case, the self-assembly mechanism should include two driving forces: one is the hydrophobic interaction which is attributed to the amphiphilic character of COS-g-PCL consisting of hydrophilic COS backbones and hydrophobic PCL branches, and the other is the formation of hydrogen bonds among the amino and hydroxyl groups remained on the COS backbones.



**Fig. 8** Variable temperature FTIR spectra of dried giant vesicles from copolymer g2. The temperature of the related curve from the bottom to the top is 25 °C, 50 °C, 60 °C, 70 °C, 80 °C and 100 °C, respectively.

## 4. Conclusions

The size of the giant vesicles can be tuned in the range of 0.5–54 μm through the structure adjustment of COS-g-PCL molecules. Surprisingly, the formed vesicles exhibit high stability and can be stored at ambient temperature, even at elevated temperature (<50 °C) for 3 months. When the vesicle suspension is heated to above 50 °C, the vesicles dissolve or collapse in the mixture of 1,4-dioxane/water. But with temperature cooled down, the copolymer can self-assembled into giant vesicles again. Taking into account the structure of COS-g-PCL molecule, there are a great number of –OH and –NH<sub>2</sub> groups along COS hydrophilic backbone. These groups, which exist on the surface of vesicles, could be utilized for the further modification of the obtained giant vesicles, such as cross-linking of vesicles, chelating of metal ion to vesicles and so on. In general, the work will shed new light on designing new type of polymer giant vesicles, self-assemble from chitooligosaccharide-based graft copolymer in 1,4-dioxane/water, with properties different from other vesicles.

## Acknowledgements

This work was supported by National Natural Science Foundation of China (20473044, 20533050, 50673048 and 973 Program 2006CB806200). The authors thank Q. F. Zhao and B. Q. Xing for technical assistance, and S. Y. Tao for useful discussions.

## Notes and references

- <sup>a</sup> Petrochina Petrochemical Research Institute, 100195-Beijing, China. Fax: +86 10 62792905; Tel: +86 10 62978296; E-mail: [gaokejing@petrochina.com.cn](mailto:gaokejing@petrochina.com.cn)
- <sup>b</sup> PetroChina Refining & Chemicals Company, 100007-Beijing, China  
 Key Lab of Organic Optoelectronics & Molecular Engineering,  
 Department of Chemistry, Tsinghua University, 100084-Beijing, China
- 1 S. Jain, F. S. Bates, *Science*, 2003, **300**, 460.
  - 2 S. Forster and T. Plantenberg, *Angew. Chem. Int. Ed.*, 2002, **41**, 688.
  - 3 M. W. Neiser, S. Muth, U. Kolb, J. R. Harris, J. Okuda, M. Schmidt, *Angew. Chem. Int. Ed.*, 2004, **43**, 3192.
  - 4 T. Ruyschaert, A. F. P. Sonnen, T. Haefele, W. Meier, M. Winterhalter, D. Fournier, *J. Am. Chem. Soc.* 2005, **127**, 6242.
  - 5 Q. Wang, T. W. Lin, L. Tang, J. E. Johnson, M. G. Finn, *Angew. Chem., Int. Ed.* 2002, **41**, 459.
  - 6 D. E. Discher, A. Eisenberg, *Science* 2002, **297**, 967.
  - 7 A. Ranquin, W. Versées, W. Meier, J. Steyaert, P. Van, Gelder, *Nano Lett.* 2005, **5**, 2220.
  - 8 A. Choucair, C. Lavigneur, A. Eisenberg, *Langmuir* 2004, **20**, 3894.
  - 9 E. P. Holowka, D. J. Pochan, T. J. Deming, *J. Am. Chem. Soc.* 2005, **127**, 12423.
  - 10 J. Rodriguez-Hernández, S. Lecommandoux, *J. Am. Chem. Soc.* 2005, **127**, 2026.
  - 11 P. P. Ghoroghchian, G. Li, D. H. Levine, K. P. Davis, F. S. Bates, D. A. Hammer, M. J. Therien, *Macromolecules* 2006, **39**, 1673.
  - 12 L. Zhang, A. Eisenberg, *Science* 1995, **268**, 1728.
  - 13 B.M. Discher, Y.Y. Won, D.S. Ege, J.C. M. Lee, F.S. Bates, D. E. Discher, D. A. Hammer, *Science* 1999, **284**, 1143; M. Antonietti, S. Förster, *Adv. Mater.* 2003, **15**, 1323.
  - 14 F. S. Chécot, G. Lecommandoux, H. A. Klok, *Angew. Chem. Int. Ed.* 2002, **41**, 1339.
  - 15 S. I. Stupp, V. LeBonheur, K. Walker, L. S. Li, K. Huggins, M. Keser, A. Amstutz, *Science* 1997, **276**, 384.
  - 16 A. Kros, W. Jesse, G. A. Metselaar, J. J. L. M. Cornelissen, *Angew. Chem. Int. Ed.* 2005, **44**, 4349.
  - 17 P. Ghoroghchian, G. Li, D. Levine, K. Davis, F. Bates, D. Hammer, M. Therien, *Macromolecules* 2006, **39**, 1673.
  - 18 J. Sun, X. Chen, C. Deng, H. Yu, Z. Xie, X. Jing, *Langmuir* 2007, **23**, 8308.
  - 19 F. Meng, C. Hiemstra, G. Engbers, F. Jan, *Macromolecules* 2003, **36**, 3004.
  - 20 J. H. Jeonga, H. S. Kangb, S. R. Yanga, J. D. Kim, *Polymer*, 2003, **44**, 583.
  - 21 K. Kuroda, K. Fujimoto, J. Sunamoto, K. Akiyoshi, *Langmuir*, 2002, **18**, 3780.
  - 22 O. E. Philippova, E. V. Volkov, N. L. Sitnikova, A. R. Khokhlov, J. Desbrieres, M. Rinaudo, *Biomacromolecules*, 2001, **2**, 483.
  - 23 B. Nottelet, A.E. Ghzaoui, J. Coudane and M. Vert, *Biomacromolecules*, 2007, **8**, 2594.
  - 24 S. Yusa, A. Sakakibara, T. Yamamoto, Y. Morishima, *Macromolecules*, 2002, **35**, 5243.
  - 25 A. Kikuchi, T. Nose, *Macromolecules*, 1996, **29**, 6770; A. Kikuchi, T. Nose, *Polymer*, 1996, **37**, 5889.
  - 26 H. Duan, M. Kuang, J. Wang, D. Chen, M. Jiang, *J. Phys. Chem. B*, 2004, **108**, 550.
  - 27 K. H. Kim, J. Huh and W. H. Jo, *Macromolecules* 2004, **37**, 676.
  - 28 N. Xu, F. S. Du and Z. C. Li, *J. of Polym. Sci. Part A: Polym. Chem.*, 2007, **45**, 1889.
  - 29 C. Cai, J. Lin, T. Chen and X. Tian, *Langmuir*, 2010, **26**, 2791.
  - 30 Y.-C. Huang, M. Arham, J.-S. Jan, *Soft Matter.*, 2011, **7**, 3975.
  - 31 K.-H. Liu, S.-Y. Chen, D.-M. Liu, T.-Y. Liu, *Macromolecules*, 2008, **41**, 6511.
  - 32 B.-Y. Chen, Y.-F. Huang, Y.-C. Huang, T.-C. Wen, J.-S. Jan, *ACS Macro Lett.*, 2014, **3**, 220.
  - 33 K.-H. Liu, B.-R. Chen, S.-Y. Chen, D.-M. Liu, *J. Phys. Chem. B*, 2009, **113**, 11800.
  - 34 Y.-C. Huang, J.-S. Jan, *Polymer*, 2014, **55**, 540.
  - 35 C. Wang, G. Li, R. Guo, *Chem. Commun.* 2005, 3591.
  - 36 K.-J. Gao, J. Yi, J. Mao, G. Li, B.-Q. Xu, *J. Appl. Polym. Sci.* 2013, **128**, 1687.
  - 37 K.-J. Gao, G. Li, H. Shi, X. Lu, Y. Gao, B.-Q. Xu, *J. Polym. Sci. Part A: Polym. Chem.* 2008, **46**, 4889.
  - 38 K.-J. Gao, G. Li, X. Lu, Y. G. Wu, B.-Q. Xu, J.-H. Fuhrhop, *Chem Commun*, 2008, 1449.
  - 39 L. Zhang, A. Eisenberg, *J. Am. Chem. Soc.* 1996, **118**, 3168.
  - 40 L. Zhang, K. Yu, A. Eisenberg, *Science* 1996, **272**, 1777.
  - 41 K. Yu, A. Eisenberg, *Macromolecules* 1996, **29**, 6359.
  - 42 Y. Yu, L. Zhang, A. Eisenberg, *Macromolecules* 1998, **31**, 1144.
  - 43 L. Zhang, A. Eisenberg, *Polym. Adv. Technol.* 1998, **9**, 667.
  - 44 P. L. Soo, A. Eisenberg, *J. Polym. Sci. Part B: Polym. Phys.* 2003, **42**, 923.
  - 45 R. J. Thibault, T. H. Galow, E. J. Turnberg, M. Gray, P. J. Hotchkiss, V. M. Rotello, *J. Am. Chem. Soc.* 2002, **124**, 15249.
  - 46 Y. Zhou, D. Yan, W. Dong, Y. Tian, *J. Phys. Chem. B* 2007, **111**, 1262.
  - 47 J. W. Lee, S. Samal, N. Selvapalam, H. J. Kim, K. Kim, *Acc. Chem. Res.* 2003, **36**, 621.
  - 48 Y. H. Ko, K. Kim, J. K. Kang, H. Chun, J. W. Lee, S. Sakamoto, K. Yamaguchi, J. C. Fettinger, K. Kim, *J. Am. Chem. Soc.* 2004, **126**, 1932.
  - 49 P. Mukhopadhyay, A. Wu, L. J. Isaacs, *J. Org. Chem.* 2004, **69**, 6157.
  - 50 J. Lagona, P. Mukhopadhyay, S. Chakrabarti, L. Isaacs, *Angew. Chem., Int. Ed.* 2005, **44**, 4844.
  - 51 K. Naka, R. Yamashita, T. Nakamura, A. Ohki, S. Maeda, *Macromol. Chem. Phys.* 1997, **198**, 89.



## Graphic Abstract

## Spontaneous Formation of Giant Vesicles with tunable sizes based on Jellyfish-like Graft Copolymers

*Ke-Jing Gao, Xiaozhou Liu, Guangtao Li, Bo-Qing Xu and Jianjun Yi*

For self-assembly studies, a series of “jellyfish-like” graft copolymers with short backbone and long branch chain were adopted. Chitooligosaccharide (COS), the oligomer of chitosan, was chosen as the hydrophilic short rigid backbone and poly( $\epsilon$ -caprolactone) (PCL) as the hydrophobic long branch chain. It was found that these special graft copolymers in 1, 4-dioxane/water mixtures could self-assemble into giant vesicles with diameter in the range of 0.5-54 $\mu\text{m}$ .

