

This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This Accepted Manuscript will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/advances

1	Electrochemical mechanism of Cr(III) reduction for preparing
2	crystalline chromium coatings based on 1-butyl-3-
3	methylimidazolium hydrogen sulfate ionic liquid
4	Xinkuai He, Chen Li, Qingyun Zhu, Bailong Hou, Yumei Jiang, Luye Wu *
5	
6	School of Packaging and Materials Engineering, Hunan University of Technology,
7	Zhuzhou 412007, P. R. China.
8	
9	*Corresponding author
10	E-mail: lyxk999@163.com (L.Y. Wu) and h-xk@163.com (X.K. He)
11	Phone: +86 731 22182088, Fax: +86 731 22182168

**RSC Advances Accepted Manuscript** 

13	Chromium coatings can be prepared from ionic liquids containing Cr(III) ions.
14	However, these coatings are almost amorphous due to the fact that the deposits
15	include metalloid atoms (such as P, C, N, O). The present work reports the direct
16	preparation of crystalline chromium coatings by electrodeposition based on 1-butyl-3-
17	methylimidazolium hydrogen sulfate ([BMIM]HSO <sub>4</sub> ) ionic liquid. The Cr(III)
18	electrochemical reduction mechanism and chromium nucleation/growth process on a
19	glassy carbon (GC) electrode in [BMIM]HSO4 are investigated. These results from
20	cyclic voltammetry and linear sweep voltammetry reveal that the Cr(III) reduction
21	occurs by a two-step process, Cr(III) to Cr(II), and Cr(II) to Cr(0), respectively, the
22	first step is irreversible with a diffusion coefficient of Cr(III) in solution of $2.03 \times 10^{-7}$
23	$cm^{2} \cdot s^{-1}$ at 353 K, and the two-step process has been confirmed by
24	chronopotentiometry. The chromium coatings are characterized by SEM, EDS and
25	XRD. XRD pattern of the coatings shows the characteristic peak of crystal Cr.
26	Chronoamperometry results reveal that chromium electrodeposition in [BMIM]HSO <sub>4</sub>
27	can be attributed to a three-dimensional instantaneous nucleation and diffusion-
28	controlled growth mechanism. These results observed in this work indicate that
29	[BMIM]HSO <sub>4</sub> ionic liquid may be a useful electrolyte for chromium
30	electrodeposition.

31 1. Introduction

The electrodeposition of chromium and its alloys is of great practical importance due to the fact that these coatings are widely used in industrial fields such as corrosion prevention, electronic materials, and functional coatings.<sup>1-5</sup> Chromium plating is

traditionally prepared in aqueous solutions.<sup>6-12</sup> However, the chromium 35 electrodeposition from Cr(III)-based aqueous baths is generally accompanied by 36 intensive hydrogen evolution reaction. Thus, the current efficiency and quality of the 37 chromium deposits can be obviously affected by the reaction.<sup>6, 13, 14</sup> Another important 38 issue concerning the technique is the participation in the chromium deposition process 39 of organic additives, which is normally used in Cr(III)-based aqueous baths.<sup>15-18</sup> The 40 participation results in the formation of chromium carbide compounds in these 41 deposits, suggesting that they are normally amorphous.<sup>15-17</sup> However, the crystal 42 43 chromium coatings present higher hardness, better wear resistance, and then more extensive applications. For these reasons it is of great interest to find alternative 44 electrolytes for crystalline chromium electrodeposition. Compared to aqueous 45 46 solutions, ionic liquids are an ideal kind of solvents for the electrodeposition of metals and alloys due to the fact that they can provide a wide electrochemical window, good 47 electrical conductivity and low volatility for electrolysis without evolution of 48 hydrogen.19-26 49

In the last two decades, the electrodeposition of chromium and its alloys from ionic liquids is popular. For instance, Al-Cr alloy coatings were prepared by Moffat<sup>27</sup> and Ali et al.<sup>28</sup> in AlCl<sub>3</sub>-trimethylphenylammonium chloride (TMPAC) or in AlCl<sub>3</sub>-N-(n-butyl)pyridinium chloride containing Cr(II) ions. Moreover, chromium concentrations of these Al-Cr alloys, between 0 and 94 at.%, could be adjusted by the applied process parameters. Whereas, the reductive Cr(II) as a chromium source was used for these Al-Cr alloys electrodeposition in the above two ionic liquids.<sup>27, 28</sup> In

**RSC Advances Accepted Manuscript** 

57	contrast to Cr(II), the electrodeposition of chromium and its alloys using Cr(III) is
58	more popular due to its better chemical stability. For instance, the crack-free
59	chromium coatings were electrodeposited by Abbott and his co-workers from choline
60	chloride/chromium chloride system. <sup>29</sup> But its hardness is only 242 Vickers, which is
61	less than that of the chrome hydride electrodeposited from chromic acid process
62	(typically 800 to 900 Vickers). Similarly, microcrack chromium coatings could be
63	prepared based on choline chloride/chromium chloride system. <sup>30</sup> To improve these
64	chromium coatings quality, a new ionic liquid, 1-butyl-3-methylimidazolium
65	tetrafluoroborate ([BMIM]BF <sub>4</sub> ), was applied to prepare black chromium coatings by
66	Eugénio et al. <sup>31, 32</sup> . It should be noted that these chromium coatings were composed of
67	a mixture of chromium oxide/hydroxide and metallic chromium with a sub-
68	micrometric granular structure. Thus, these coatings are amorphous. In addition, they
69	found that the Cr(III) reduction is quasi-reversible and occurs in a two-step process,
70	Cr(III) to $Cr(II)$ and $Cr(II)$ to $Cr(0)$ . <sup>32</sup> However, the $Cr(III)$ reduction in [BMIM]PF <sub>6</sub> is
71	an irreversible one-step reaction, $Cr(III)$ to $Cr(0)$ . <sup>33</sup> It can be concluded that the
72	mechanism of Cr(III) reduction may be remarkably different in different ionic liquids.
73	It should be noted that these chromium coatings electrodeposited from the two ionic
74	liquids are amorphous due to the fact that they include P, C and O atoms. It suggests
75	that $[BMIM]BF_4$ and $[BMIM]PF_6$ are not stable enough and can involve in the
76	electrodepositing process. Thus, it is important to develop a more stable ionic liquid
77	system from which the crystalline chromium coatings can be prepared directly.

Page 5 of 30

#### **RSC Advances**

78	Moreover, the electrocrystallization mechanism of chromium depositing from
79	ionic liquid systems has scarcely been investigated. The metallic phase growth is
80	preceded by the nucleation step. The nucleation theory of metals on surfaces proposed
81	by Scharifker and Hills <sup>34-36</sup> was acknowledged for the examination of the process of
82	metals electrodeposition from aqueous solutions <sup>37-39</sup> and ionic liquids <sup>40-42</sup> . For
83	instance, the nucleation and growth mechanism of sliver from $10^{-2}$ M Ag(NH <sub>3</sub> ) <sup>2+/1.6</sup>
84	M NH <sub>3</sub> , 1 M KNO <sub>3</sub> onto different carbon electrode substrates has been successfully
85	determined by Miranda et al. <sup>43</sup> and its behavior is in accordance with a three-
86	dimensional (3D) instantaneous nucleation mechanism followed by growth of the
87	nuclei controlled by silver ions diffusion kinetic.
88	1-butyl-3-methylimidazolium-hydrogen sulfate ([BMIM]HSO <sub>4</sub> ) ionic liquid may
89	not involve in the chromium electrodeposition process since it can provide a more
90	stable anion that is not subject to decomposition as has been observed with the
91	fluorinated anions such as $[BF_4]^-$ and $[PF_6]^-$ . <sup>44, 45</sup> In addition, the crystalline chromium
92	prepared directly from 1-butyl-3-methylimidazolium hydrogen sulfate still has not
93	been reported. Thus, in this work, [BMIM]HSO <sub>4</sub> is used as the electrolyte to prepare
94	the crystalline chromium coatings. The electrochemical reduction of Cr(III) on a GC
95	electrode is investigated by cyclic voltammetry, linear sweep voltammetry. Moreover,

96 the result is confirmed by chronopotentiometry, and the morphology and structural

97 properties of the obtained coatings are also characterized by SEM, EDS and XRD.

98 Finally, the chromium nucleation/growth process is investigated by

99 chronoamperometry.

# 100 2. Experimental

# 101 **2.1 Materials**

The [BMIM]HSO<sub>4</sub> was synthesized and purified according to the method published in a previous literature.<sup>46</sup> The 0.55 M CrCl<sub>3</sub>-[BMIM]HSO<sub>4</sub> solutions were prepared by heating the proper mixtures of anhydrous  $CrCl_3(99.9\%, Aldrich)$  and [BMIM]HSO<sub>4</sub> in a beaker for 72 h, the process was performed in a glove box. The working electrode was GC purchased from Alfa Aesar with a purity of 99.99% and the exposing area was 0.07 cm<sup>2</sup>.

# 108 2.2 Measurement procedures and apparatus

109 Electrochemical measurements were performed by a CHI 660B potentiostat using three-electrode configuration. The GC mentioned above was used as the 110 111 working electrode. A platinum foil (20 mm  $\times$  15 mm) and a platinum electrode were used as the counter-electrode and the reference electrode, respectively. The electrolyte 112 temperature was maintained at 353 K during the electrochemical measurements. The 113 114 working electrode was mechanically polished with 0.05 µm alumina slurries on 115 lapping pads. All the electrodes were rinsed by deionised water then carefully dried 116 with  $N_2$  before transported into the glove box. It should be noted that the Tafel curves were obtained at a scan rate of 1.0 mV·s<sup>-1</sup> around -150 mV against the equilibrium 117 118 potential.

## 119 2.3 Electrodeposition

For electrodeposition, a platinum foil (20 mm × 15 mm) was used as anode and a
Cu foil (10 mm × 5 mm, 99.9%) as cathode, respectively. A platinum electrode was

- 122 connected as the reference electrode, to which all potentials were referred. The
- 123 distance between the cathode and anode was maintained at 30 mm. After
- 124 electrodeposition, the samples were cleaned in acetone to eliminate the [BMIM]HSO<sub>4</sub>
- 125 remains, and then rinsed in deionized water. Thickness of Cr films was calculated
- 126 according to the gravimetric data  $\Delta M$  using the equation:  $T = \frac{\Delta M}{\rho \cdot S}$ , where T is the
- 127 coating thickness,  $\Delta M$  is the weight gained on the Cu substrate which can be
- 128 measured,  $\rho$  is the density of chromium(7.20 g·cm<sup>-3</sup>), S is the surface area of Cu
- 129 substrate that involved in the electrodeposition process.

#### 130 2.4 Characterization of the electrodeposited coatings

- 131 The surface morphology of the chromium coatings was observed by a SEM
- 132 (HITACHI S-3000N), provided with an energy dispersive spectrometer (EDS)
- 133 detector. X-ray diffractogram (XRD) was recorded using a Siemens D-5000
- 134 diffractometer with  $CuK_{\alpha}$  radiation.

# 135 3. Results and discussion

#### 136 **3.1 Cyclic voltammetry study**

137 Transient cyclic voltammogram is an effective technique applied to obtain

- 138 important information about the reaction on a GC electrode surface.<sup>32, 47</sup> To study the
- 139 electrochemical reduction of Cr(III), electrochemical measurements are performed at
- 140 353 K using this technique. A representative cyclic voltammogram of pure
- 141 [BMIM]HSO<sub>4</sub> is shown in Fig. 1 (curve a). No apparent reduction peak is observed in
- 142 the forward scan up to -3.10 V and no anodic peak is observed in the reverse scan.
- The result indicates that [BMIM]HSO<sub>4</sub> is electrochemically stable in the potential 7/29

144 ranging from -3.10 to -0.10 V, which means [BMIM]HSO<sub>4</sub> possesses an

- 145 electrochemical window of 3.0 V, in accordance with a previous literature.<sup>48</sup>
- 146 The typical cyclic voltammogram of 0.55 M Cr(III) in [BMIM]HSO<sub>4</sub> ionic liquid
- 147 on the GC electrode at 353 K is also presented in Fig. 1 (curve b). The potential scan
- 148 is negatively begun from 0.25 to -2.45 V, and then reversed to the beginning potential.
- 149 Two obvious peaks are observed at -1.60 and -2.25 V, implying that the Cr(III)
- 150 reduction occurs in two steps via Cr(II) ions:
- 151 peak A  $Cr(III) + 1e^- \rightarrow Cr(II)$
- 152 peak B  $Cr(II) + 2e^- \rightarrow Cr(0)$

153 It is a widely recognized mechanism for the Cr(III) reduction, in accordance with the
published literatures.<sup>32, 49</sup>

155 The effect of the potential scan rates on the voltammograms of the electrolyte 156 corresponding to different scan ranges is displaced in Fig. 2. As shown in Fig. 2, the voltammograms of the electrode is changed remarkably by scan rate. The cathodic 157 158 peak potential shifts negatively and the peak current increases with the increasing of 159 the scan rate. In addition, whether or not substep scan, no obvious anodic peaks are observed in the reverse scan. These results also indicate that the reduction of Cr(III) in 160 161  $[BMIM]HSO_4$  on a GC electrode is an irreversible two-step reaction, which is also verified by the linear relationship between the peak potential  $E_{\rm p}$  ( $E_{\rm pA}$  and  $E_{\rm pB}$ ) and  $\ln v$ 162 (Fig. 3a and b). However, the reduction of Cr(III) to Cr(II) in [BMIM][BF<sub>4</sub>] on a GC 163 electrode is a quasi-reversible reaction,<sup>34</sup> the difference may be attributed to the 164 165 diversity of properties between the  $[HSO_4]^-$  and  $[BF_4]^-$  in ionic liquid solutions.

166

#### **RSC Advances**

For an irreversible process, there is a relationship between  $n_{\alpha}$  and  $|E_{p}-E_{p/2}|$ :<sup>50</sup>

167	$ E_{\rm p}-E_{\rm p/2}  = 1.857RT/(\alpha n_{\alpha}F)$ (1)
168	where $E_p$ is the peak potential, $E_{p/2}$ is the half-peak potential, T is the absolute
169	temperature, R is the gas constant, F is the Faraday constant, $n_{\alpha}$ is the number of
170	electrons in the rate determining step and $\alpha$ is the charge transfer coefficient. Thus,
171	the value of $n_{\alpha}$ for the irreversible process can be determined by Eq.(1). The average
172	value of $\alpha n_{\alpha}$ is found to be 0.250 at 353 K. Thus, considered that $\alpha n_{\alpha} \leq 1$ , it is inferred
173	that the number of electrons participated in the kinetic step is one, with $\alpha = 0.250$ ,
174	which is in accordance with the previous literature. <sup>49</sup>
175	Fig. 4 presents the linear relationship between the cathodic peak current $(I_{pA})$ and
176	$v^{1/2}$ , implying the Cr(III) reduction is controlled by a diffusion process in
177	[BMIM]HSO <sub>4</sub> ionic liquid. Thus, based on the above discussion, it can be surmised
178	that the reduction of Cr(III) to Cr(II) is an irreversible and diffusion-controlled
179	process. By supposing the reaction is completely irreversible and ignoring the
180	electrode area change caused by chromium electrodeposition, the diffusion coefficient
181	of Cr(III) in [BMIM]HSO <sub>4</sub> solution at 353 K can be calculated using Eq.(2): <sup>51</sup>
182	$I_{\rm p} = 0.4958 n F^{3/2} C_0 A D^{1/2} v^{1/2} (\alpha n_{\alpha}/RT)^{1/2} $ <sup>(2)</sup>
183	where A is the geometric area of the electrode, $C_0$ is the Cr(III) concentration, D is the
184	diffusion coefficient, $\alpha$ is the charge transfer coefficient, $n_{\alpha}$ is the number of electrons
185	in the rate determining step, $v$ is the potential scan rate and $n$ is the number of
186	electrons transferred. The average value of $\alpha n_{\alpha}$ can be estimated using Eq.(1) and
187	according to Eq.(2), the diffusion coefficient of Cr(III) in CrCl <sub>3</sub> -[BMIM]HSO <sub>4</sub>

188 solution is calculated. Thus, the diffusion coefficient of Cr(III) obtained at 353 K is 189  $2.03 \times 10^{-7}$  cm<sup>2</sup> s<sup>-1</sup>.

# 190 3.2 Linear sweep voltammetry study

191 A linear sweep voltammetry measurement using a lower sweep rate is carried out 192 in order to further confirm the electrochemical mechanism of Cr(III) reduction. Fig. 5a presents a typical linear sweep voltammogram of GC electrode in 0.55 M Cr(III)-193 194 [BMIM]HSO<sub>4</sub> solution with a scan rate of 1 mV·s<sup>-1</sup> at 353 K. As shown in Fig. 5a, almost no increase of cathodic current occurs before -0.89 V (region 'A'). The current 195 196 begins rising remarkably from -0.89 to -1.67 V (region 'B'), suggesting that the Cr(III) reduction has happened. In order to obtain more information about the 197 reduction reaction, another electrodepositing experiment on a copper substrate based 198 199 on 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> solution at a constant potential of -1.67 V for 30 min is carried out at 353 K. No deposits are observed at the potential. The result can be 200 201 attributed to the reduction of Cr(III) to Cr(II), implying that the reduction of Cr(II) to 202 Cr(0) does not happen. The cathodic current begins rising more sharply from -1.67 to 203 -2.15 V (region 'C'), which indicates that another reduction reaction occurs. To further confirm the reduction reaction, another electrodepositing experiment under the 204 same conditions is carried out at -2.00 V. Deposits are observed at the potential, 205 206 implying that the reduction reaction of Cr(II) to Cr(0) also happens. Thus, these 207 results are consistent with cyclic voltammetry. In addition, these deposits are 208 investigated by SEM, EDS and XRD. As shown in Fig. 6a, a ball-like structure 209 without obvious micro cracks and pinholes is observed. The structure is clearly

210	defined at a higher resolution (Fig. 6b). EDS analysis shows that the deposits are
211	composed of Cr without other elements (Fig. 6c), and the XRD pattern shows two
212	strong peaks for (210) and (200) planes locate at $2\theta = 44.02^{\circ}$ and $64.58^{\circ}$ , respectively,
213	which reveals that the Cr electrodeposits present a crystal structure with preferential
214	orientation (210) and (200) (Fig. 6d). It should be noted that the thickness of the
215	deposits can be up to 9.2 µm, which is calculated by the equation: $T = \frac{\Delta M}{\rho \cdot S}$ .
216	However, these chromium deposits obtained from $[BMIM]BF_4$ or $[BMIM]PF_6$ ionic
217	liquid containing Cr(III) ions are generally amorphous. <sup>32, 33</sup> To prepare the crystalline
218	coatings, further annealing to the amorphous deposits at high temperature is
219	integrant. <sup>15-17</sup> Thus, the technique using [BMIM]HSO <sub>4</sub> as Cr(III) electrolytes to
220	prepare crystalline chromium coatings directly is simpler, which also demonstrates
221	that [BMIM]HSO <sub>4</sub> ionic liquid in commercial chromium plating is more promising. In
222	addition, as shown in Fig. 5b, a Tafel line is observed in the potential ranging from -
223	0.98 to -1.03 V when $\log I$ is plotted against <i>E</i> , corresponding well to the result of the
224	Cr(III) reduction. Thus, the cathode transfer coefficient can be determined by Eq.
225	(3). <sup>52</sup>

$$226 \quad b_{\rm c} = 2.3 RT/anF \tag{3}$$

where  $b_c$  is the slope of Tafel line, *R* is the ideal gas constant, *F* is the Faraday constant, *T* is the absolute temperature,  $\alpha$  is the cathode transfer coefficient, *n* is the number of electrons transferred. The cathode transfer coefficient is 0.241, which is basically in accordance with that obtained from cyclic voltammetry.

231 **3.3 Chronopotentiometry study** 11/29

232 The characteristics of chronopotentiometry about a two-step reaction have been reported by Plonski<sup>53</sup> and its most acknowledged characteristics can be described 233 234 using the theoretical curves given in Fig. 7a. Three main observations are made: (i) as 235 a result of the competition between the charging of the double layer and the transitory 236 electron transfer process, the  $\eta$ -t curves are peak-shaped between  $i_{cr1}$  and  $i_{cr2}$  ( $\eta$  is the overvoltage, *i* is the current); (ii) at first, the difference  $\eta_{\text{peak}}$ - $\eta_x$  increases and passes 237 238 through a maximum value, when  $i \rightarrow i_{cr2}$ , the value decreases toward zero; (iii) it is 239 generally valid that the time necessary to reach the steady state decreases with the 240 increasing of *i*, except for  $i_{cr2}$  when the steady state of  $\eta$  attains sooner. However, for a one-step reaction,  $\eta$  is only monotonous change with the time.<sup>53</sup> 241 242 In order to further investigate whether the Cr(III) reduction is a two-step process,

243 chronopotentiometric measurements are carried out in 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> 244 solutions by stepping the current density from  $5.0 \times 10^{-3}$  to  $5.6 \times 10^{-3}$  A, which are 245 within the theoretical value ranging between  $2.09 \times 10^{-4}$  and  $5.80 \times 10^{-3}$  A (Fig. 7a). 246 Representative potential-time transients resulting from these measurements are 247 depicted in Fig. 7b. These transient plots present the above similar characteristics of 248 the theoretical curves, which implies that Cr(III) reduction is a two-step process.

249 3.4 Chronoamperometry study

250 Chronoamperometric experiments are performed to reveal the mechanism of 251 chromium nucleation/growth process in 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> system by 252 stepping the potentials from -1.90 to -2.00 V. Fig. 8a shows three representative 253 current-time transients resulting from these experiments that exhibit a typical shape

254	for a diffusion-limited nucleation process with three-dimensional (3D) growth of
255	nuclei. Upon the potential step, the current-transients are characterized by a shape
256	double-layer charging current decay followed by a rising current due to the formation
257	and growth of Cr nuclei until a current maximum, $I_{max}$ , is reached at a time, $t_{max}$ (i.e.
258	the Cr nuclei begin to overlap, then follow by a decaying portion, converging to a
259	limiting current corresponding to linear diffusion of the electroactive ions to a planar
260	electrode as per the Cottrell equation). The $I_{max}$ increases while the $t_{max}$ shortens when
261	increasing the applied nucleation potential (Fig. 8a). This can be attributed to the
262	decreasing in the time required for the diffusion layer to overlay due to an increased
263	nucleation density.54 The electrodeposition of metals onto foreign surfaces often
264	presents some type of 3D nucleation. Two limiting cases for this type of metal
265	deposition have been classified by Scharifker and Hills as progressive nucleation,
266	where the number of nuclei increases during the whole depositing process, and
267	instantaneous nucleation, where all nuclei are formed immediately after the potential
268	step. <sup>34-36</sup> To confirm the nucleation behaviors, the $I$ versus $t$ transients are converted
269	to dimensionless $(I/I_{max})^2$ versus $(t/t_{max})$ curves and compared with the theoretical
270	dimensionless curves developed by Scharifker et al. <sup>53</sup> for the 3D instantaneous
271	(Eq.(4)) and progressive (Eq.(5)) nucleation. <sup>34</sup>

272 
$$\frac{I^2}{I_{\text{max}}^2} = \frac{1.9542}{t/t_{\text{max}}} \left[ 1 - \exp\left(-1.2564\frac{t}{t_{\text{max}}}\right) \right]^2$$
(4)

273 (Instantaneous nucleation)

**RSC Advances Accepted Manuscript** 

274 
$$\frac{I^2}{I_{\text{max}}^2} = \frac{1.2254}{t_{\text{max}}} \left[ 1 - \exp\left(-2.3367 \frac{t^2}{t_{\text{max}}^2}\right) \right]^2$$
(5)

#### 275 (Progressive nucleation)

where I and t are the current and time, respectively, and  $I_{max}$  is the maximum current 276 at time  $t_{\text{max}}$ . As the experimental and theoretical plots represented in Fig. 8b, the 277 chromium electrodeposition on a GC electrode in [BMIM]HSO<sub>4</sub> ionic liquid most 278 279 likely involves in the 3D instantaneous nucleation mode. 280 As depicted in Fig. 8c, the current-time transients in Fig. 8a suggest that the 281 growth of chromium nuclei on a GC electrode is typical for a diffusion-controlled process since after  $I_{max}$ , all of the decreasing currents I present a good linear relation 282 with  $t^{-1/2}$ . It can be concluded from the above results that the Cr(III) reduction on a GC 283 284 electrode corresponds to a 3D instantaneous nucleation with diffusion-controlled 285 growth. In addition, the diffusion coefficient (D) can be calculated using the Cottrell 286 equation:55

287 
$$I = nFAD^{1/2}C_0/(\pi^{1/2}t^{1/2})$$
 (6)

where *F* is the Faraday constant, *n* is the number of electrons transferred, *A* is the geometric area of the electrode,  $C_0$  is the Cr(III) concentration, *D* is the diffusion coefficient. Thus, the diffusion coefficient determined by the slope of the plot of *I* versus  $t^{1/2}$  (line 1 in Fig. 8c) is about  $1.98 \times 10^{-7}$  cm<sup>2</sup>·s<sup>-1</sup>, which is basically consistent with that obtained from cyclic voltammetry.

# 293 **4. Conclusions**

294	The Cr(III) electrochemical reduction on a GC electrode in [BMIM]HSO <sub>4</sub> ionic
295	liquid is studied. The cyclic voltammograms of Cr(III) at different scan rates are
296	measured. The result reveals that the Cr(III) reduction occurs in a two-step, which is
297	an irreversible and diffusion-controlled process. The diffusion coefficient of Cr(III) in
298	[BMIM]HSO <sub>4</sub> solution obtained from cyclic voltammetry is $2.03 \times 10^{-7} \text{ cm}^2 \cdot \text{s}^{-1}$ at 353
299	K. The two-step reaction is also confirmed by chronopotentiometry. Moreover, it can
300	be concluded that the chromium electrodeposition proceeds through a 3D nucleation
301	growth mechanism as deduced by analysis of chronoamperometry data. XRD pattern
302	of the electrodeposited chromium layer presents the characteristic peak of crystal Cr.
303	These results obtained in this work indicate that the [BMIM]HSO <sub>4</sub> ionic liquid may
304	be a useful electrolyte for electrodeposition of metals.

# 305 Acknowledgments

The authors acknowledge the financial supports of the Natural Science Foundation of China (No. 21476067), the Natural Science Foundation of Hunan Province, China (Nos.2015JJ2044 and 13JJ3107), and the Project Sponsored by the Scientific Research Foundation for the Returned Overseas Chinese Scholars, State Education Ministry.

## 311 References

- 312 1 C. E. Lu, N. W. Pu, K. H. Hou, C. C. Tseng and M. D. Ger, *Appl. Surf. Sci.*, 2013,
  313 282, 544-551.
- 314 2 Z. X. Zeng, Y. L. Sun and J. Y. Zhang, *Electrochem. Commun.*, 2009, 11, 331-
- 315 334.

- 316 3 K. N. Sun, X. N. Hu, J. H. Zhang and J. R. Wang, *Wear*, 1996, **196**, 295-297.
- 317 4 X. K. He, B. L. Hou, Y. X. Cai, C. Li, Y. M. Jiang and L. Y. Wu, J. Nanosci.
- 318 *Nanotechno.*, 2013, **13**, 4031-4039.
- 319 5 X. K. He, B. L. Hou, Y. X. Cai and L. Y. Wu, J. Nanosci. Nanotechno., 2013, 13,
- 320 2193-2200.
- 321 6 Y. B. Song and D. T. Chin, *Electrochim. Acta*, 2002, **48**, 349-356.
- 322 7 V. Protsenko and F. Danilov, *Electrochim. Acta*, 2009, **54**, 5666-5672.
- 323 8 Z. X. Zeng, Y. X. Zhang, W. J. Zhao and J. Y. Zhang, Surf. Coat. Tech., 2011,
- **205**, 4771-4775.
- 325 9 V. S. Protsenko, F. I. Danilov, V. O. Gordiienko, S. C. Kwon, M. Kim and J. Y.
- 326 Lee, *Thin Solid Films*, 2011, **520**, 380-383.
- 327 10 R. Giovanardi and G. Orlando, Surf. Coat. Tech., 2011, 205, 3947-3955.
- 328 11 Z. A. Hamid, Surf. Coat. Tech., 2009, 203, 3442-3449.
- 329 12 N. V. Phuong, S. C. Kwon, J. Y. Lee, J. Y. Shin, B. T. Huy and Y. I. Lee,
- 330 *Microchem. J.*, 2011, **99**, 7-14.
- 331 13 N. V. Phuong, S. C. Kwon, J. Y. Lee, J. H. Lee and K. H. Lee, Surf. Coat. Tech.,
- 332 2012, **206**, 4349-4355.
- 333 14 V. S. Protsenko, V. O. Gordiienko, F. I. Danilov and S. C. Kwon, Metal Finishing,
- **334 2011**, **109**, **33-37**.
- 335 15 O. V. Safonova, L. N. Vykhodtseva, N. A. Polyakov, J. C. Swarbrick, M. Sikora,
- 336 P. Glatzel and V. A. Safonov, *Electrochim. Acta*, 2010, **56**, 145-153.
- 337 16 V. A. Safonov, L. N. Vykhodtseva, Y. M. Polukarov, O. V. Safonova, G.
- 338 Smolentsev, M. Sikora, S. G. Eeckhout and P. Glatzel, J. Phys. Chem. B, 2006,
- **110**, 23192-23196.

- 340 17 S. G. Eeckhout, O. V. Safonova, G. Smolentsev, M. Biasioli, V. A. Safonov, L. N.
- 341 Vykhodtseva, M. Sikora and P. Glatzel, *J. Anal. Atom. Spectrom.*, 2009, 24, 215342 223.
- 343 18 F. I. Danilov and V. S. Protsenko, Prot. Met., 2001, 37, 223-228.
- 344 19 A. P. Abbott and K. J. McKenzie, Phys. Chem. Chem. Phys., 2006, 8, 4265-4279.
- 345 20 K. R. Seddon, G. Srinivasan, M. Swadźba-Kwaśny and A. R. Wilson, Phys. Chem.
- 346 *Chem. Phys.*, 2013, **15**, 4518-4526.
- 347 21 Y. H. You, C. D. Gu, X. L. Wang and J. P. Tu, J. Electrochem. Soc., 2012, 159,
- 348 642**-**648.
- 349 22 C. D. Gu and J. P. Tu, *Langmuir*, 2011, 27, 10132-10140.
- 350 23 C. D. Gu, Y. H. You, Y. L. Yu, S. X. Qu and J. P. Tu, Surf. Coat. Tech., 2011,
- **205**, 4928-4933.
- 352 24 M. C. Buzzeo, R. G. Evans and R. G. Compton, Chem. Phys. Chem., 2004, 5,
- 353 1106-1120.
- 354 25 Q. B. Zhang and Y. X. Hua, Mater. Chem. Phys., 2012, 134, 333-339.
- 355 26 Q. B. Zhang, Y. X. Hua, Y. T. Wang, H. J. Lu and X. Y. Zhang, *Hydrometallurgy*,
- 356 2009, **98**, 291-297.
- 357 27 T. P. Moffat, J. Electrochem. Soc., 1994, 141, 115-117.
- 358 28 M. R. Ali, A. Nishikata and T. Tsuru, *Electrochim. Acta*, 1997, 42, 2347-2354.
- 359 29 A. P. Abbott, G. Capper, D. L. Davies and R. K. Rasheed, *Chem-Eur. J.*, 2004, 10,
  360 3769-3774.
- 361 30 Y. Cui and Y. X. Hua, T. Nonferr. Metal., 2011, 63, 92-96.
- 362 31 S. Eugénio, C. M. Rangel, R. Vilar and A. M. B. Rego, Thin Solid Films, 2011,
- **519**, 1845-1850.

- 364 32 S. Eugénio, C. M. Rangel, R. Vilar and S. Quaresma, *Electrochim. Acta*, 2011, 56,
- 365 10347-10352.
- 366 33 Y. S. J. Hasimu, R. Q. Liu and H. Y. Mi, Chem. J. Chinese U., 2014, 35, 140-145.
- 367 34 B. Scharifker and G. Hills, *Electrochim. Acta*, 1983, 28, 879-889.
- 368 35 G. Gunawardena, G. Hills and I. Montenegro, J. Electroanal. Chem., 1982, 138,
- 369 241-254.
- 370 36 B. R. Scharifker and J. Mostany, J. Electroanal. Chem., 1984, 177, 13-23.
- 371 37 M. M. Hernandez, I. Gonzalez and N. Batina, J. Phys Chem B, 2001, 105, 4214-
- 372
   4223.
- 373 38 K. Yao and Y. F. Cheng, Int. J. Hydrogen Energ., 2008, 33, 6681-6686.
- 374 39 T. Rapecki, Z. Stojek and M. Donten, *Electrochim. Acta*, 2013, **106**, 264-271.
- 375 40 R. Bomparola, S. Caporali, A. Lavacchi and U. Bardi, *Surf. Coat. Tech.*, 2007,
  376 201, 9485-9490.
- 377 41 M. Jayakumar, K. A. Venkatesan, T. G. Srinivasan and P. R. Vasudeva Rao,
- 378 *Electrochim. Acta*, 2009, **54**, 6747-6755.
- 379 42 A. Bakkar and V. Neubert, *Electrochim. Acta*, 2013, **103**, 211-218.
- 380 43 R. Bomparola, S. Caporali, A. Lavacchi and U. Bardi, Surf. Coat. Tech., 2007,
- **201**, 9485-9490.
- 382 44 R. P. Swatloski, J. D. Holbrey and R. D. Rogers, Green Chem., 2003, 5, 361-363.
- 383 45 A. J. Whitehead, A. G. Lawrance and A. McCluskey, *Green Chem.*, 2004, 6, 313384 315.
- 385 46 J. A. Whitehead, G. A. Lawrance and A. McCluskey, *Aust. J. Chem.*, 2004, 57,
  386 151-155.
- 387 47 E. S. C. Ferreira, C. M. Pereira and A. F. Silva, *J. Electroanal. Chem.*, 2013, 707,
  388 52-58.

- 389 48 F. Endres, S. Z. E. Abedin and S. Matter, Phys. Chem. Chem. Phys., 2006, 8, 2101-
- 390 2116.
- 391 49 X. K. He, B. L. Hou, C. Li, Q. Y. Zhu, Y. M. Jiang and L. Y. Wu, Electrochim.
- *Acta*, 2014, **130**, 245-252.
- 393 50 P. Kedzierzawski and Z. Szklarska-Smialowska, J. Electroanal. Chem., 1981, 122,
- 394 269-278.
- 395 51 M. Frank and J. Winnick, J. Electroanal. Chem., 1987, 238, 163-182.
- 396 52 R. S. Nicholson, I. Shain, Anal. Chem., 1964, 36, 706-723.
- 397 53 I. H. Plonski, J. Electrochem. Soc., 1970, 117, 1048-1052.
- 398 54 W. Le, K. Du, L. Zhang, J. Xiao, C. Zhang, Y. W. Zhang, L. Zhou and Q. Yin,
- *Electrochim. Acta*, 2013, **95**, 179-184.
- 400 55 L. Wang, Y. L. Zheng, X. P. Lu, Z. Li, L. L. Sun and Y. H. Song, Sensor. Actuat.
- 401 *B-Chem.*, 2014, **195**, 1-7.

- 404 Fig. 1. Cyclic voltammograms of pure [BMIM]HSO<sub>4</sub> (curve a) and 0.55 M Cr(III)-
- 405 [BMIM]HSO<sub>4</sub> (curve b) solutions on a GC electrode at 353 K. Scan rate is 50 mV·s<sup>-1</sup>.
- 406 Fig. 2. Cyclic voltammograms of 0.55 M Cr(III) on a GC electrode in [BMIM]HSO<sub>4</sub>
- 407 solution at 353 K with different potential scanning rates between 50 and 250 mV  $\cdot$ s<sup>-1</sup>
- 408 using the scan ranges from -0.2 to -2.45 V (a), -0.90 to -1.69 V (b) and -1.85 to -2.45
- 409 V (c).
- 410 Fig. 3. Peak potential  $E_p$  vs. lnv for Cr(III) reduction on a GC electrode in
- 411 [BMIM]HSO<sub>4</sub> solution at 353 K. (a-peak A, b-peak B)
- 412 Fig. 4. Peak current  $I_{pA}$  vs.  $v^{1/2}$  for Cr(III) reduction on a GC electrode in
- 413 [BMIM]HSO<sub>4</sub> solution at 353 K.
- 414 Fig. 5. Linear sweep voltammogram(a) and Tafel curve(b) for 0.55 M Cr(III) on a GC
- 415 electrode in [BMIM]HSO<sub>4</sub> solution at 353 K.
- 416 Fig. 6. SEM micrographs of chromium deposits on Cu substrate at 353 K for 30 min
- 417 from the 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> solution at -2.00 V:  $a \times 5000$ ,  $b \times 10000$ . EDS
- 418 analysis (c) and XRD pattern (d) of the deposited layer.
- 419 Fig. 7. (a) Theoretical  $\eta$ -t curves of two process reduction reaction,
- 420 (b) Chronopotentiograms of 0.55 M Cr(III) on a GC electrode in [BMIM]HSO<sub>4</sub>
- 421 solution at 353 K.
- 422 Fig. 8. (a) Current-time transients of the chronoamperometry experiments recorded on
- 423 a GC electrode in 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> solution at 353 K. (b) Dimensionless
- 424 plots for instantaneous and progressive nucleation model in comparison with
- 425 experimental transients. (c) Plots of *I* against  $t^{1/2}$  from the decreasing portion of **20/29**

- 426 current-time transients in Fig. 8 (a) for the deposition process of 0.55 M Cr(III) on a
- 427 GC electrode.



441



456 **Fig. 2** Cyclic voltammograms of 0.55 M Cr(III) on a GC electrode in [BMIM]HSO<sub>4</sub> 457 solution at 353 K with different potential scanning rates between 50 and 250 mV·s<sup>-1</sup> 458 using the scan ranges from -0.2 to -2.45 V (a), -0.90 to -1.69 V (b) and -1.85 to -2.45 459 V (c).

460	Xinkuai He, et al., Fig. 2
-----	----------------------------



478 Fig. 3 Peak potential  $E_p$  vs. lnv for Cr(III) reduction on a GC electrode in

479 [BMIM]HSO<sub>4</sub> solution at 353 K. (a-peak A, b-peak B)

480



494 Xinkuai He, et al., Fig. 4

495



515

516 Xinkuai He, et al., Fig. 5



531 Fig. 6 SEM micrographs of chromium deposits on Cu substrate at 353 K for 30 min

532 from the 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> solution at -2.00 V: a  $\times$  5000, b  $\times$  10000. EDS

533 analysis (c) and XRD pattern (d) of the deposited layer.

- 534
- 535 Xinkuai He, *et al.*, Fig. 6
- 536





**Fig. 8** (a) Current-time transients of the chronoamperometry experiments recorded on a GC electrode in 0.55 M Cr(III)-[BMIM]HSO<sub>4</sub> solution at 353 K. (b) Dimensionless plots for instantaneous and progressive nucleation model in comparison with experimental transients. (c) Plots of *I* against  $t^{-1/2}$  from the decreasing portion of current-time transients in Fig. 8 (a) for the deposition process of 0.55 M Cr(III) on a GC electrode.

582

Xinkuai He, et al., Fig. 8

# Electrochemical mechanism of Cr(III) reduction for preparing crystalline chromium coatings based on 1-butyl-3methylimidazolium hydrogen sulfate ionic liquid

Xinkuai He, Chen Li, Qingyun Zhu, Bailong Hou, Yumei Jiang, Luye Wu\*



We report Cr(III) electrochemical reduction mechanism and nucleation/growth process, and the electrodepositing directly crystalline chromium coatings based on [BMIM]HSO<sub>4</sub>.