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2	Low Cost and Large-area Fabrication of Self-Cleaning coating on polymeric
3	surface Based on Electroless-Plating-Like Solution Deposition Approach
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16	Abstract
17	A novel low-cost controllable solution-based coating process for realization of
18	self-cleaning coating on large-area flexible polymeric substrates based on an
19	improved Electroless-Plating-Like Solution Deposition (EPLSD) approach was
20	developed. In a typical coating procedure, a layer of Aniline were first absorbed onto
21	the surface of a flexible polymeric substrate by dip-coating; and as the pretreated
22	Dereve Titenium Complex (BTC) modified gilies (P.Si) after the thermal treatment at
23	Peroxo-Intanum-Complex (FIC) modified since (F-SI), after the merinal dealinent at 20° C for 20 min, the flavible polymeric substrate costed by P Si composite thin films
24	$s_0 \in 10150$ mm, the flexible polymetric substrate coated by F-Si composite time mms
25	SEM_XRD_ETIR analysis and its self cleaning characteristic was also evaluated by
20	the contact angle and the methyl blue degradation test
28	the condict diffic and the methyl one degradation test.
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1 I. Introduction

Semiconducting metal oxides are an important class of functional materials promising a wide range of new technological applications. In particular, when it was prepared into thin films, the utilized efficiency and applicability of these materials can be significantly improved. For instance, titanium oxide (TiO₂) films which are coated on various substrates, such as glasses, polymers, and silicon wafers, have many important applications in catalytic decomposition of organic pollutants, anti-bacterial, self-cleaning, as well as solar cells, etc. ¹⁻⁶.

Among these applications, combining the photo-induced hydrophilicity and 9 photocatalytic properties of TiO₂ film, which make use of sunlight and natural rainfall 10 to keep surfaces self-cleaning, is one of the most attractive and promising fields, as it 11 can save significant cost for maintenance in industry and daily life⁷. However, the 12 hydrophilic surface of a simple TiO₂ film will slowly become hydrophobic again 13 14 without UV light irradiation, resulting in a deterioration or failure of its self-cleaning performance. Thus, a self-cleaning surface containing SiO₂ and TiO₂ is widely 15 adopted ⁸⁻¹⁰, as the SiO₂ component can preserve a lasting hydrophilicity without the 16 UV light irradiation. 17

On the other hand, considering the various substrates that have been tried for 18 supporting TiO_2 photocatalysts, although polymer substrate seems to be very 19 promising due to its several advantages such as flexible, low-cost, and ease of 20 availability, most of the as reported TiO₂ films were usually coated on the inorganic 21 22 substrates, such as glasses, tile, etc. Because, concerning the generally adopted preparing methods for fabrication TiO₂ self-cleaning coating, such as sol-gel method, 23 chemical vapor deposition (CVD), and hydrothermal methods, etc. ¹¹⁻¹⁴, a high 24 temperature sintering process was generally inevitable, as the sintering process can 25 26 benefit not only a high efficient photocatalytic activity which were generated only in well crystallized anatase TiO₂ phases, but also a strong adhesion between the coating 27 layer and the substrate due to a chemical bond formation between the TiO_2 and the 28 inorganic substrate during the high temperature process. Thus the low thermal 29 30 stability/tolerance of polymeric substrate precludes the use of the above mentioned 31 physical and chemical methods that need high sintering temperature for adherence of titania on the polymer surface. This calls for development of novel, low temperature 32 and economical methods for fabrication of polymer supported titania photocatalysts. 33

Alternative approaches for embedding of titania on the polymer surface without thermal sintering, such as binding ¹⁵, polymerization the organic monomer in the presence of TiO₂ nanoparticles ¹⁶, and thermal bonding (hot pressing)¹⁷ have then been explored. But a following loss catalytic activity, which was attributed to the partial embedment of TiO₂ particles on the polymer surface, clearly suggested a need in improvement of the deposition method. Recently, our group developed an

electroless-plating-like solution deposition (EPLSD) approach for the fabrication of 1 flexible metal oxide film ¹⁸, i.e., crystallized metal oxide (such as TiO₂, V₂O₅ and 2 MoO₃) can be attached on the surface of PET film adhesively by a redox reaction 3 between d⁰ configured Peroxo Metal Complex (Ti (IV), V (V), Mo (VI)) with the 4 conductive (electroactive/conjugated) polymer monomer (EDOT, Pyrrole, and 5 Aniline). In this EPLSD process, the formed metal oxide films showed highly 6 crystalline phase, all processes were performed under 80 °C conditions using 7 large-area polymer substrates, and the procedure is simple and cost-effective. 8 Moreover, conductive (electroactive/conjugated) polymer (PAni) formed during the 9 EPLSD procedure, characterized with an extended π -conjugated electron system, can 10 obtain a synergic effect with inorganic semiconductor metal oxides like TiO_2^{18} , i.e., 11 PAni causes rapid charge separation, slow charge recombination and thus an enhanced 12 Photocatalytic activity of the prepared TiO₂/PAni photocatalysts. Therefore, EPLSD 13 14 may suitable for the use in the polymer substrate coating of self-cleaning by enhanced self-cleaning ability instead of a catalytic activity loss due to the partial embedment of 15 16 TiO₂.

However, the as described EPLSD process was limited only early transition metal 17 series with d⁰ configuration, such as Ti, V, Mo, etc. While for the silica, which is 18 necessary for keeping the good self-cleaning effect with TiO₂ in the coating, as 19 described in above, was not involved in the candidates for EPLSD procedure, i.e., Si 20 element is not an early transition metal. This hindered the application of EPLSD 21 procedure on fabrication of the TiO_2 -SiO₂ composite self-cleaning coating on 22 polymeric surface. To overcome this limitation, in the present report, we developed an 23 improved EPLSD process which can be used to non-d⁰ configured nanoparticles to 24 fabricated an oxides composited film. This improved process includes modifying the 25 silica nanoparticles with the Peroxo-Titanium-Complex (PTC) aqueous solution, then 26 the PTC coated silica (P-Si) solution instead of Peroxo Metal Complex solution was 27 applied for the EPLSD approach, and a corresponding TiO₂-SiO₂ composite flexible 28 thin film was successfully coated onto the PET substrate. 29

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31 II. Experimental Section

Materials. All chemical reagents used in present experiments were analytical grade and purchased from Beijing Chem. Co. Ltd. The silica nano-sol (SiO₂) was purchased from Beijing Institute of Aeronautical Materials. A PET film as a flexible substrate was purchased from Tianjin Shiqi Co. Ltd.

A typical experiment process for the film preparation can be described in **Scheme** 1:



Scheme 1 Scheme of electroless-plating-like solution deposition (EPLSD) process for fabricating the TiO₂-SiO₂ self-cleaning polymeric film.

The coating solution for the EPLSD procedure was prepared as following: Firstly, 5 TiCl₄ was reacted with ammonia to form titanic acid precipitates, the precipitate was 6 dissolved into 30 wt. % H_2O_2 solution, and then the solution were diluted to [Ti] =7 $0.125 \text{ mol } L^{-1}$, and a detailed description for the preparation of 8 peroxo-tatinum-complex (PTC) solution can be found in the Ref. 18. Secondly, the as 9 10 prepared PTC aqueous solution was added to the as-purchased silica nanosol, the molar ratios of silica/Ti were 1:1. Finally, the mixtures were stirred and kept on 11 boiling for ca.4 h, and a final coating solution, named as PTC modified silica (P-Si) 12 solution, was thus achieved. 13

As for the pretreatment of the PET substrate, the swelling process of PET was carried out in a 2 L beaker, 20*40 cm PET film (10.98g) was immersed in the 1.5 L mixed organic solvent of Aniline and EtOH in the above flask for a 2h to reach the plateau value of 8 wt. % of aniline, and the swelling film was then taken out and performed a drying procedure in air for 10min, allowing nearly 0.87g aniline monomer was absorbed onto the PET surface. The swelling kinetics of PET film in aniline was discussed in supporting information Figure S1.

As for the EPLSD process, as was illustrated in scheme 1, it was performed as following: the P-Si coating solutions was firstly heated and remain at a stable temperature of 80 °C, and then the as-pretreated PET substrates were immersed into P-Si solution to fulfill coating; the reaction time can be varied from several minute to hours depending on requirement.

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27 Characterization

X-ray diffraction (XRD) experiments were performed with an XD-2 diffractometer
(Purkinje General Instrument Co. Ltd.), using monochromatic Cu Ka radiation at 36
kV and 30 mA. Fourier transform infrared (FT-IR) spectra were obtained by using an
Excalibur 3100 spectrometer with a resolution of 4 cm-1. Measurements were
performed in the transmission mode in spectroscopic grade KBr pellets for all the
powders. Transmission electron microscopy (TEM) studies were performed using

- Philips CM200/FEG field emission-gun transmission electron microscope, operating
 at 200 kV. The surface composition and elemental chemical state of the samples were
 examined by X-ray photoelectron spectroscopy (XPS) using a Model Axis Ultra
 (Kratos Analytical Ltd.) apparatus. The surface morphology of the sample was
 investigated by a scanning electron microscopy (Hitachi Ultra-High-Resolution
 S-4300).
- 7
- 8 III. Results and Discussion



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- **Fig.1** TEM images of the a and b are as-purchased silica sol, and c and d are P-Si composite solution, the insert photos are the corresponding digital photos of the samples.
- 12 13

Morphology of the as-purchased silica and the P-Si composite solution were 14 characterized by TEM observations (Fig. 1). Fig. 1a revealed that the as-purchased 15 silica nanosol was an aqueous solution of SiO₂ nanoparticles of with the size of about 16 17 4-6 nm in diameters, and the insert part of the digital photo indicated the solution was homogeneous and milky transparent. The armophous nature of the silica nanoparticles 18 19 can be indentified by the HRTEM image of Fig 1b. Fig. 1c and 1d are the TEM images of the samples taken from the P-Si composite solution, which was prepared as 20 21 discribed in the experimental section. Compared with the original silica sol, after 22 reaction with PTC, the latter P-Si solution shows a significant color variation, i.e., 23 from milky to yellow, but remained as a homogeneous and transparent aqueous sol. This P-Si sol can be kept in a sealed case for several months without precipitation. 24 25 Furthermore, as was observed in the Fig. 1c, this yellow transparent solution was proved to be a solution of nanoparticles with the size of about 8-10 nm in diameters. 26 27 Different from the amorphous nature of the original silica nanoparticles, a new

crystalline phase, as was shown in Fig. 1d (marked with the inset white arrow lines), 1 can be clearly distinguished, verified the existence of crystallized nanoparticles 2 attached on the silica spheres, which was characterized with a distinct lattice spacings 3 4 of 0.35 nm correspond to the (101) planes spacings of the anatase phase TiO_2 (JCPDS: No.02-0406). These results suggested the P-Si solution contains an Anatase 5 crystallized TiO₂, moreover, it might be a nanocomposite of SiO₂/TiO₂, concerning on 6 an increased of dimension in particle sizes, i.e., from 4-6 nm to 8-10 nm, compared 7 with the original silica nanoparticles. In order to further identify the structure of the as 8 prepared nanocomposite SiO2/TiO2, extensive TEM observations were made on 9 samples collected from the P-Si composite solution (see supporting information 10 Figure S3), and a random disordered structure of the as prepared nanocomposite 11 12 SiO_2/TiO_2 can be characterized.

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Fig.2 (a)zetal potential graph the original Silica, Peroxo-Titanium-Complex and the P-Si nanosols, and (b) XRD pattern of the the original Silica and the P-Si samples.

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As a part of efforts to obtain a better understanding of the formation of P-Si 19 20 nanocomposite, a serious of analysis were performed on the silica and P-Si solution. 21 The zeta potential analysis was shown in Fig. 2a, the mixture of a positively charged silica nanosols (2.7 mV) and negatively charged PTC aqueous solution (-7.9 mV) 22 would result in a negatively charged solution characterized with a relatively low zeta 23 potential of -2.1 mV, and this implies the modification of PTC on silica nanoparticles 24 may attribute to the electrostatic adsorption between the opposite charging of the PTC 25 26 precursor and silica nanoparticle. Fig. 2b shows the corresponding XRD patterns of the silica and the P-Si sapmle. Compared with the typical armophous result of the 27 silica sample, all the diffraction peaks of the PTC modified samples are the anatase 28 crystalline phase of titania (JCPDS no: 02-0406), confirming that the P-Si obtain the 29 component of TiO₂. 30

The samples were further investigated by FT-IR spectra and the results were shown in supporting information S4, not only the Ti-O-Ti stretching vibration of TiO_2 (462 cm^{-1} and 815 cm^{-1}) and the Si-O-Si stretching vibration of SiO₂ (1096 cm^{-1}), but

also a stretching vibration Ti-O-Si (958 cm⁻¹) can be clearly characterized, and this result revealed the existence of a chemical bond between TiO_2 and SiO_2 nanoparticle. Concerning on the fact the TiO_2 precursor, i.e. PTC, was absorbed on the silica surface, we may suggesting that TiO_2 were mostly nucleated on silica surface during the crystallization process and result in a nanocomposite of $TiO_2 \SiO_2$ connected by a chemical bond of Ti-O-Si.



Fig.3 TEM images of the P-Si samples prepared by refluxing P-Si solutions under the same conditions except that the reaction time was 0.5h (a); 4h (b) and 8h (c) respectively, the insert photos are the corresponding digital photos of the samples.

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In order to clarify the long term stability of the as prepared P-Si Nano sol, the 13 14 thermal treatment time of the P-Si solution was varied from 0.5h to 8h. The products of the as prepared P-Si samples were characterized by electron microscopy and the 15 result was shown in Fig. 3. As was shown in Fig. 3a, at the beginning of the thermal 16 treatment of the P-Si solution, some nanoparticles with the size of about 4-6 nm in 17 diameters surrounded by the amorphous materials can be indentified, and this may 18 attribute to the adsorption of PTC onto the surface of the silca due to the electrostatic 19 20 adsorption between the opposite charging of the PTC precursor and silica nanoparticle. As the thermal treating time of the P-Si solution increased to 4h, as was 21 22 shown in Fig. 3b, monodispersed nanocomposite SiO₂/TiO₂ can be observed. Meanwhile, the color of the as obtained nanosols changed from transparent orange to 23 transparent yellow (see the insert digital photo of Fig. 3a, 3b), which can be attribute 24 to the partly decomposition of PTC since it was the exclusive precursor of TiO_2 . 25 26 Finally, accompanied with the completely decomposition of PTC, identified by the 27 disappearing of PTC unique color (see the insert digital photo of Fig. 3c), agglomeration of the nanocomposite SiO₂/TiO₂ can be characterized in Fig. 3c, which 28 29 means the destruction of the long term stability and precipitation can also be observed in the as achieved P-Si solution. 30

The result discussed above suggested that the PTC groups absorbed on the nanoparticles surface may accounted for the long term stability of the as prepared P-Si nanosol. Actually, Ragai¹⁹ had also mentioned the long term stability of peroxidized

- 1 titanium solution for more than one year. According to Schwarzenbach's work ^{20, 21},
- 2 the condensation reaction of the PTC may result in a polynuclear product, and it is
- 3 thus reasonable to conclude that the polynuclear PTC absorbing onto the surface of
- 4 the nanocomposite SiO_2/TiO_2 may responsible for the long term stability of P-Si
- 5 solution due to a steric stabilization.
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Fig.4. SEM images of the flexible film samples of (a) Natural PET film; (b) and (c) Flexible PET films coated with TiO₂-SiO₂ were taken from different magnifications; (d) The cross-section image of flexible TiO₂-SiO₂ films obtained from EPLSD processes.

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Fig.4 were the SEM characterization result of the origal and P-Si coated PET films, 13 As shown in Fig.4, compared to a blank and smooth surface of the original PET film, 14 15 the surfaces of PET films treated by P-Si were covered with a layer of fine and closed nanoparticles (Fig. 4b, 4c), and this layer thickness is about 500 nm (Fig. 4d). The 16 17 EDS (see supporting information S5) spectra analysis suggested that the composition of these nanoparticles which formed on the surface of the PET films mainly contains 18 Ti, Si. C and O elements, and an atomic ratio of C (44.69%): Ti (2.96%): Si (2.22%) 19 can be identified. a saddle-shaped distribution of the element Ti and Si can be 20 observed (see supporting information S6), suggesting that the element Ti and Si were 21 mainly centralized on the surface of the PET film. 22

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Fig.5. TEM images of the samples that were scraped off from the PET substrate with a knife, the PET substrate was coated with P-Si sol based on EPLSD procedure.

5 We also scraped off the samples from the PET substrate with a knife and then dispersed the as-obtained sample in water to analysis its component. As shown in the 6 7 Fig. 5a, an amorphous aggregation can be observed from the TEM image, a further magnificated image (Fig.5b), some crystalline nanoparticles surrounded by the 8 amorphous materials with the size of about 4-6 nm in diameters can be indentified. 9 The lattice planes shown in the Fig.4b agree well with distances of (101) lattice planes 10 of anatase TiO₂ (Fig. 5b) (JCPDS no: 02-0406). Thus, this indicated that the improved 11 EPLSD process can truly be applied to deposite the as prepared P-Si nanoparticles 12 onto the PET surface. As to the amorphous component coming from the scraped off 13 coating samples as observed in Fig. 5, as was discussed in our previous report ^{18, 22}, it 14 might be the polymerization product of the Aniline monomer during the EPLSD 15 16 procedure.

Based on the above discussion, we can preliminarily interpret the mechanism of the 17 18 improved the EPLSD process which can realize the TiO₂-SiO₂ nanocomposite coating on the polymeric PET substrate surface. As was shown in scheme 2, firstly, by 19 refluxing the mixture of the negatively charged PTC precursor with the positively 20 21 charged silica nanoparticles solution, the surface of silica nanoparticle can be modified with TiO₂ nanocrystals, and a P-Si solution of TiO₂-SiO₂ nanocomposite 22 particles was thus obtained; secondly, by dipping the PET film which was pretreated 23 with aniline into the P-Si solution, triggered by the O-O group absorbed on the P-Si 24 nanoparticles surface, which was indentified by the feature yellow color in aqueous 25 solution and a O-O stretching vibration at 890 cm⁻¹ of the P-Si nanoparticles (see 26 27 supporting information S4), a radical polymerization reaction of the orgain aniline which absorbed onto the PET substrated begins; Finally, as was described in a typical 28 EPLSD procedure¹⁸, the *in-situ* produced polymer could be regarded as a self-binder 29 reagent, and this process makes the TiO₂-SiO₂ nanocomposite attached on the surface 30 of PET firmly. 31



Table 1. Durability test result of water static contact angles of PET film before and after treatment by TiO_2 -SiO₂ coating, with or without 365nm Uv light irradiation for 30min.

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11 The effectiveness of TiO_2 -SiO₂ as a hydrophilic coating on treated PET film 12 obtained was investigated by contact angle measurements. Table 1 depicts the durability test result of water static contact angles with or without UV light irradiation 13 for untreated and treated samples. The decrease of the water contact angle in the 14 treated surfaces (from 82.7 ° to 18.7 °) indicates that sufficient surface hydrophilicity 15 was achieved. Furthermore, UV light irradiation can enhances the decrease of the 16 water contact angle (from 18.7 ° to 9.7 °), and which are considered as the photo 17 induced hydrophilicity of the TiO₂. What more, a long time durability test result 18 revealed that the as achieved hydrophilicity of the as prepared PET film can be 19 20 retained more than one year.



Fig. 6. Photocatalytic degradation curves of aqueous MB irradiated under UV light (365 nm)
of (a) blank PET film and (b) TiO₂-SiO₂ coated PET film.

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5 The photocatalytic activities of the as-prepared TiO₂-SiO₂ film were evaluated by methylene blue degradation in an aqueous solution. In a typical experiment, the 6 films slides (20 mm \times 40 mm) were first dipped in 5 \times 10⁻⁵ M methylene blue for the 7 saturated adsorption process. The adsorption equilibrium testing film was then put 8 into a watch glass containing 20 mL of 1×10^{-5} M methylene blue solution and 9 exposed to the 365 nm UV light for 480 min. The result of methylene blue 10 degradation tests revealed that the TiO2-SiO2 coated PET film demonstrated an 11 obvious photocatalytic activity (33.21% methylene blue concentration decrease) 12 13 compared with a photocatalytic inactivity bare PET film.

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17 IV. Conclusions

By surface modification of SiO₂ nanoparticles with the d⁰ configured peroxo metal 18 19 complex precursor, self-cleaning coating of TiO₂-SiO₂ on large-area PET film via EPLSD procedure was successfully prepared. A durable hydrophilicity and color 20 21 fading test result revealed the as prepared polymeric film surface obtained a desired self-cleaning ability. The as reported novel low temperature and economical 22 23 solution-based procedure will representing not only a promising step toward polymer substrate self-cleaning, but also a widen application scope of the EPLSD from d⁰ 24 configured metal oxide nanoparticles to non- d⁰ configured nanoparticles. 25

26

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Table of Content:

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