RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This Accepted Manuscript will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/advances



This novel layered hybrid aerogel will pave the way for development of drug delivery devices with superior properties.

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxxx

ARTICLE TYPE

A Novel Hybrid Material: An Inorganic Silica Aerogel Core Encapsulated with a Tunable Organic Alginate Aerogel Layer

Zeynep Ulker,^{*a*} Can Erkey^{*a**}

Received (in XXX, XXX) Xth XXXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX 5 DOI: 10.1039/b000000x

A novel layered material consisting of a silica aerogel core encapsulated by an alginate aerogel layer was developed. The components of the hybrid aerogel had the high surface area and high porosity of pure aerogels which should lead to 10 development of new layered systems for a wide variety of applications.

Over the past few decades, nanostructured materials have shown a great promise in a variety of application areas like catalysis, insulation, sensors or drug delivery¹⁻³. Among them, nanoporous ¹⁵ aerogels were found to be potential candidates for the encapsulation and adsorption of a variety of molecules due to their large inner surface areas, high surface to volume ratios,

- large pore volumes and uniform pore sizes⁴⁻⁶. The possibility of designing a multi layered material based on aerogels is even more ²⁰ exciting to provide new opportunities for intended applications such as increased and/or controlled loading/adsorption and controlled release. Moreover, the possibility of tuning the properties of the layers by controlling the preparation conditions
- should make these materials very attractive for a wide variety of ²⁵ applications. There have already been several attempts to synthesize layered structures based on different porous materials such as hydrogels. Ladet et al. prepared the first multi layered hydrogel from the periphery towards the core by controlling the kinetics of the gelation process⁷. Dai et al. prepared a multi
- ³⁰ layered hydrogel structure with and without internal space between separated layers by a precise control of the processing conditions via a dynamic self assembly method⁸. They suggested that a layered structure from the core to the surface might facilitate the drug loading and cell immobilization Along similar
- ³⁵ lines, Duan et al. synthesized a new multi layered hydrogel using agarose gel-core as the template by sequential cross-linking ⁹. They showed that the layer formation was dependent on the type of cross-linker, its concentration and/or soaking time. Alternatively, layered aerogel based materials such as silica
- ⁴⁰ aerogel coated with PEG hydrogel ¹⁰ or with polymeric materials ¹¹ were also prepared. However, to our knowledge there are no reports in the literature of a layered material prepared from two distinct aerogels with high surface areas and porosities. This approach may lead to development of new materials with
- ⁴⁵ desirable properties for different applications. The incorporation of different compounds with high loadings can be achieved in one single material. Loading of two different compounds on different

layers could prevent the interaction between them. Moreover, it may be possible to use layered materials for sequential release of 50 two different compounds.

- In this study, we designed and prepared a new layered aerogel system comprising a biocompatible silica aerogel core and a biodegradable alginate aerogel layer while retaining their porous characteristics intact. The new alginate layer was shown to
- ⁵⁵ protect the hydrophilic silica aerogel core when immersed in liquid solutions which otherwise collapse easily. It was also demonstrated that the kinetics of drug release was changed due to the existence of a thin alginate layer around the silica aerogel core.
- ⁶⁰ Silica aerogel was prepared by the conventional two step sol-gel synthesis and the alginate aerogel was synthesized using the internal setting method¹². The alginate layer formation was triggered by the diffusion of cross-linking ions from the silica alcogel core into a sodium alginate solution in which it was
 ⁶⁵ immersed. The crosslinking ions were shown to create an electrostatic binding between macromolecules and form an eggbox structure ^{8, 13}. To synthesize the silica alcogel core, the silica precursor, tetraethylorthosilicate (TEOS), was primarily hydrolyzed in a mixture of CaCl₂ solution (instead of pure water)
 ⁷⁰ and ethanol. The rate of hydrolysis reactions was accelerated by the addition of an acid catalyst as shown in Figure 1.



Figure 1: The scheme for the preparation of silica aerogel coated with an alginate aerogel

This journal is © The Royal Society of Chemistry [year]

After the hydrolysis, the base catalyst was added to increase the rate of condensation reactions which would eventually lead to gelation. The gelled samples which are called alcogels were then placed in Na-alginate solution. The diffusion of Ca^{2+} ions from

s the porous silica alcogel structure to the the alginate solution resulted in the formation of an alginate layer around the silica alcogel core as shown in Figure 1. The shape of the initial silica alcogel was found to determine the final shape of the layered materials due to the homogeneous diffusion of the Ca²⁺ ions from ¹⁰ the alcogel core.

The growth of the alginate layer continued as long as the silica alcogel core had sufficient free Ca^{2+} . The separation of the newly formed layered structure from the sodium alginate solution

- ¹⁵ stopped the growth of the alginate layer. The resulting material could be further cross-linked (or cured) in the same divalent cation solution which in turn was found to reduce the thickness of the alginate layer ⁸. After gelation, if the layered sample was primarily dipped in CaCl₂ solution and then placed in a different
- ²⁰ Na-alginate solution following similar principles of dip coating for membrane formation, a new layer formed leading to a multi layered aerogel structure. Due to the difference in the rate of layer formation, two successive layers would be expected to have different pore properties. The same strategy might also be
- ²⁵ followed for the initial layer formation; however it would be more difficult to control the layer thickness. After curing, the resulting layered aerogel was subjected to successive aging steps prior to supercritical drying. The samples were successively placed in a series of ethanol-water baths with increasing alcohol
- $_{30}$ concentrations (10, 30, 50, 70, 90 and 100). The reason of using this kind of aging was to prevent the structural changes which could have occurred if 100% ethanol was immediately added ¹⁴. The ethanol in the pores was then removed by supercritical drying using CO₂ at 90 bar and 313K.
- ³⁵ As shown in Figure 2A, the layer got thicker with increasing crosslinking time and increasing $CaCl_2$ concentration in the pores of the silica alcogel due to the existence of more free divalent cations. The change in Ca^{2+} ion concentration in the silica alcogel core with a diameter of 1.13cm changed the time required for the
- ⁴⁰ growth of the alginate layer to a certain thickness.



Figure 2A: The change in the thickness of the alginate layer with time using different concentrations of CaCl₂ (0.5 wt.% alginate).



⁴⁵ Figure 2B: The thickness of the alginate layer versus CaCl₂ concentration at specific times (10 min and 60 min)

- The thickness reached almost 4 mm in 60 minutes for the highest $CaCl_2$ concentration versus 0.5 mm for the lowest $CaCl_2$ concentration for experiments carried out using the same ⁵⁰ composition of alginate solution. The rate of layer growth was reduced after 60 minutes which can be attributed to the reduced rate of diffusion of Ca^{2+} ions through the silica alcogel and then the newly formed alginate hydrogel layer. As seen in Figure 2B, there was a linear correlation between the concentration of Ca^{2+}
- ⁵⁵ ions and the thickness of the alginate layer at a certain time for a fixed alginate concentration. As the initial CaCl₂ concentration increased, the alginate layer was found to be thicker at a constant alginate concentration due to the existence of more free divalent cations. Moreover, as the alginate concentration was increased,
- ⁶⁰ the growth of the layer was shown to be reduced at a constant CaCl₂ concentration due to the formation of more cross links creating a tighter layer. This linear relation enables the selection of the appropriate concentration for a desired thickness over a specified time period.
- ⁶⁵ The pore properties of the samples were characterized using N_2 physisorption. The samples were primarily degassed at 60°C under vacuum prior to analysis. The effect of increasing Naalginate weight ratio in the solution on the pore properties of the alginate aerogels was also investigated. The pore structure was
- ⁷⁰ then analyzed with Zeiss Ultra Plus Field Emission Scanning Electron Microscope after gold coating. The pore size distribution of the samples was shown to be consistent among the BET results and SEM images.
- N_2 physisorption results as shown in Figures 3A and 3B indicated 75 that the formation of an alginate layer around the silica alcogel did not significantly affect the pore characteristics of the silica aerogel. The small reduction in its surface area (which is usually around ~950-1000 m²/g) could be related to its volume shrinkage during the layer formation due to the absence of the conventional
- ⁸⁰ first aging step which should have strengthened the network. In addition, the surface area of alginate aerogel layers was sufficiently high indicative of high loading capacities. Furthermore, the difference in pore sizes could be used to enable the control over the loading and release of different sized, ⁸⁵ especially large biomolecules.

with different concentrations				
Sample	BET Surface area (m ² /g)	Pore volume (cm ³ /g)	Desorption average pore radius (BJH) (nm)	t-plot micropore volume (cm ³ /g)
Silica aerogel core	905	4,1	8,0	0,02
Alginate aerogel layer (%1.5w/w)	345	0,8	16,5	0,043
Alginate aerogel layer (%1w/w)	400	0,9	11,7	0,031
Alginate aerogel layer (%0.5w/w)	407	1,1	10,3	0,036

Table 1: BET Results for silica aerogel core and alginate aerogel layers

 with different concentrations

- ⁵ As shown in Table 1, the surface areas of alginate layers which were prepared with different concentrations ranging from 0.5 g of alginate per 100 grams of water to 1.5 g of alginate per 100 grams of water were found to be similar ^{14, 15} and even higher ¹⁶ compared to the values in the literature for alginate aerogels.
- ¹⁰ Veronovski et al. showed that increasing the concentration of the alginate solution resulted in a higher degree of cross-linking leading to more compact and stable gels ¹⁴. This finding when combined with our BET results suggests that alginate aerogels prepared using different alginate concentrations have similar pore
- ¹⁵ properties. Increasing the concentration of the polymeric chains during the synthesis was expected to create smaller pores and thus reduced pore volumes. Even though the data given in Table 1 are in agreement with this hypothesis, the magnitude of the difference among the pore volumes is smaller than expected. This
- ²⁰ can perhaps be attributed to a higher degree of cross-linking which enables a higher resistance to shrinkage during supercritical drying due to the higher strength of the solid network. Moreover, the presence of smaller pores of the alginate aerogel layers compared to the silica aerogel core could enable
- 25 the use of this system for molecules with different sizes.



Figure 3A: Adsorption Isotherms of N₂ on Alginate Aerogel Layers and Silica aerogel Core



Figure 3B: The pore size distribution of different alginate layers and silica aerogel core



⁵⁰ Figure 4A: SEM image of the alginate layer prepared with %0.5 w/w Naalginate in water



Figure 4B: SEM image of the alginate layer prepared with %1 w/w Naalginate in water

55



Figure 4C: SEM image of the alginate layer prepared with %1.5 w/w Naalginate in water

- ²⁰ As shown in Figure 3B, alginate layers had wide pore size distributions. This can also be seen from the SEM images of the alginate layers which were prepared with different percent of Naalginate in water as shown in Figure 4A, B and C. SEM images also indicate that the alginate aerogel layers have a fibrous cross-
- ²⁵ linked structure with different sized pores. From the average pore size and specific surface area data, the alginate aerogel layers can be considered to be mesoporous with a small fraction of microporosity compared to the silica aerogel which has less micropore volume. The increasing initial alginate concentration
- ³⁰ also resulted in a slight increase in micropore volume as shown in Table 1.

To show the effect of the alginate layer on the drug release kinetics, a layered system with paracetomol as the model drug was prepared. Paracetamol was loaded to the silica alcogel phase

³⁵ and this matrix was then covered with an alginate layer. (See supplementary materials for the details of the procedure). The drug loaded hybrid aerogels were shown to be stable even after 4 months of their synthesis as shown in Figure 5.



45

50

55 Figure5: Paracetamol loaded silica aerogels encapsulated with alginate aerogel layers. (Hybrid aerogel at the right was synthesized after 4months the hybrid aerogel at the left was synthesized)

The release of paracetamol from a native silica aerogel and silica aerogels encapsulated with alginate aerogel layer was compared 60 in PBS buffer (pH: 5.8). Nanodrop 1000 spectrophotometer was used to monitor the release of the paracetamol at 244 nm. The alginate layer prevented the collapse of the silica aerogel when contacted with the buffer solution, swelled with time, and thus inhibited the burst drug release and slightly retarded the drug 65 release as shown in Figure 6. As the thickness of the alginate aerogel layer was around 1-2 mm, the release was not significantly affected but comparably slowed down indicating that these hybrid aerogels are promising as drug delivery vehicles. Increasing the concentration of the alginate in the initial 70 solution and increasing the thickness of the layer would lead to a slower release. Thus, more work will be carried out in order to analyze the effect of the thickness of alginate layers and the concentration of the alginate in the layers on the drug release kinetics.



Figure6: The release profiles of paracetamol in PBS buffer with a pH of 5.8

Conclusions

- A novel layered material consisting of two distinct aerogels was ⁸⁰ prepared for the first time. The approach of combining different porous structures in one single material is expected to widen the application area of these materials especially in biomedical field for controlled release applications. The layers were not only shown to protect the core and to slightly retard the release of the ⁸⁵ drug but are also expected to be used as a delivery device capable
- of carrying different molecules at the same time for sequential release.

Notes and references

- ^a Koç University, Rumelifeneri Yolu,34450 Sarıyer,Istanbul,TURKEY. 90 Tel: 90 2123381866; *E-mail: cerkey@ku.edu.tr
- † Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/b000000x/
- M. Goldberg, R. Langer and X. Jia, *Journal of biomaterials science*. Polymer edition, 2007, 18, 241-268.

- 2. F. Zaera, Chemical Society Reviews, 2013, 42, 2746-2762.
- Z. Ulker and C. Erkey, *Journal of Controlled Release*, 2014, 177, 51-63.
- 4. I. Smirnova, S. Suttiruengwong and W. Arlt, *Journal of Non-Crystalline Solids*, 2004, **350**, 54-60.
- C. A. García-González, M. Alnaief and I. Smirnova, *Carbohydrate* Polymers, 2011, 86, 1425-1438.
- S. Bozbag, S. Kostenko, M. Kurykin, V. Khrustalev, A. Khokhlov, L. Zhang, M. Aindow and C. Erkey, *J Nanopart Res*, 2012, 14, 1-13.
 - 7. S. Ladet, L. David and A. Domard, *Nature*, 2008, 452, 76-79.
 - H. Dai, X. Li, Y. Long, J. Wu, S. Liang, X. Zhang, N. Zhao and J. Xu, Soft Matter, 2009, 5, 1987-1989.
- J. Duan, R. Hou, X. Xiong, Y. Wang, Y. Wang, J. Fu and Z. Yu, Journal of Materials Chemistry B, 2013, 1, 485-492.
- S. Giray, T. Bal, A. M. Kartal, S. Kızılel and C. Erkey*, Journal of Biomedical Materials Research Part A, 2012, 100A, 1307-1315.
- M. Alnaief, I. Smirnova, S. Antonyuk, S. Heinrich and S. Roth, *Chemie Ingenieur Technik*, 2010, 82, 1466-1467.
- C. M. Silva, A. J. Ribeiro, I. V. Figueiredo, A. R. Gonçalves and F. Veiga, *International Journal of Pharmaceutics*, 2006, **311**, 1-10.
- 13. M. Robitzer, L. David, C. Rochas, F. Di Renzo and F. Quignard, ²⁵ *Langmuir*, 2008, **24**, 12547-12552.
- A. Veronovski, Ž. Knez and Z. Novak, *The Journal of Supercritical Fluids*, 2013, **79**, 209-215.
- R. R. Escudero, M. Robitzer, F. Di Renzo and F. Quignard, Carbohydrate Polymers, 2009, 75, 52-57.
- 30 16. T. Mehling, I. Smirnova, U. Guenther and R. H. H. Neubert, *Journal of Non-Crystalline Solids*, 2009, 355, 2472-2479.