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Page 1 of 23 **RSC Advances**

Controlled assembly of Bi2S3 architectures for the all-solid-state supercapacitor electrodes and highly efficient photocatalysts

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RSC Advances Page 2 of 23

Abstract: Bismuth sulfide (Bi_2S_3) microflowers have been successfully fabricated through one-pot hydrothermal method. The structures and morphologies of the as-obtained products are characterized by X-ray diffraction (XRD), field-emission scanning electron microscopy (FESEM), transmission electron microscopy (TEM) and Raman spectroscopy. The experimental results show that $Bi₂S₃$ microflowers are composed of many microrods with the length of $18{\sim}20$ μ m. Metal/semiconductor/metal (MSM) sandwich structures are fabricated, and the current-voltage (*I-V*) characteristics exhibit a clear back-to-back Schottky-diode behavior. The galvanostatic charge/discharge performance illustrates that the prepared Bi₂S₃ microflowers has the good performance of discharge efficiency at current density from 1 mA cm⁻² to 10 mA cm⁻². Furthermore, the as-synthesized $Bi₂S₃$ microflowers are also used as the efficient UV-light photocatalysts for the photocatalytic degradation of methylene orange (MO) under light illumination, which shows almost complete degradation $(\sim 95\%)$ of MO dye.

1. Introduction

The fabrication and assemblies of high-efficiency energy storage devices and high-efficiency photocatalysts have attracted widespread attention because of ever-increasing demands for renewable energy and environmental issues.¹⁻⁴ Supercapacitors (SCs), known as electrochemical capacitors, are promising energy storage devices due to their higher power densities, longer operating lifespans, and better safety tolerances than batteries. The past few decades have witnessed substantial progress on the development of high-capacitive-performance transition metal oxides (TMOs), such as $RuO₂$, MnO₂, Co₃O₄, and NiO.⁵⁻⁷

Semiconducting chalcogenides, especially metal sulfides, selenides, and tellurides, have received considerable attention in many applications, including lithium-ion batteries, $\frac{8}{3}$ supercapacitors, $\frac{9}{3}$ solar cells, $\frac{10}{3}$ photocatalysts, $\frac{11}{3}$ resulting from their unique structural features. Moreover, much more attention has been paid to metal chalcogenides, such as Bi_2S_3 ,¹² ZnS,¹³ and CuS,¹⁴ due to their high photocatalytic activities in the degradation of pollutants. Among these metal chalcogenides, $Bi₂S₃$ which is a direct band gap (1.4 eV) layered semiconductor and crystallizes in the orthorhombic system, 15 has received considerable attention based on its unique electric and optical properties.16-18 Great efforts have been made to obtain Bi_2S_3 with different morphologies and structures, such as nanorods,¹⁹ nanowires, 20 and microstructures. $21,22$ Furthermore, its compounds such as urchin-shaped Fe₃O₄@Bi₂S₃ core–shell hierarchical structures, are also obtained by a sonochemical method, showing 100% degradation ratio of rhodamine B after irradiating for $6h^{23}$ Many methods, such as sonochemical techniques, 24 template-directed procedures, 25 organometallic complex decomposition, $26,27$ electrochemical deposition,^{28,29} chemical vapor deposition(CVD),³⁰ hydrothermal and solvothermal routes, $31-34$ and the crystallization of amorphous colloids, 35 have successfully been developed for generating 1D metal sulfides. Among these methods, hydrothermal route can synthesize good crystalline and special structure materials with the merits of simpleness, less pollution, economy and high purity.

In this paper, we report a facile template-free hydrothermal process to

RSC Advances Page 4 of 23

synthesize $Bi₂S₃$ microflowers assembled by many nanorods, which are characterized by energy-dispersive X-ray spectroscopy (EDS), transmission electron microscopy (TEM), selected-area electron diffraction (SAED), powder X-ray diffraction (XRD), scanning electron microscopy (SEM), high resolution transmission electron microscopy (HRTEM) , etc. We also made two kinds of devices with either aluminum or titanium foil as the bottom electrode for $(I-V)$ characteristics, which are not only the stable substrate to support $Bi₂S₃$ microflowers but also the bottom electrode to transport the electrical charges from the film. Meanwhile, another metal foil is fixed on the microflowers film as the top electrode. A metal/semiconductor/metal (MSM) sandwich structure is fabricated, and the *I-V* characteristics are investigated, which exhibit clear back-to-back Schottky-diode behavior. The method can be easily developed for an economic mass-production process.

2. Experimental Section

2.1 Synthesis Process

To fabricate the microflowers, 0.930g (2.5 mmol) of ethylenediaminetetraacetic acid disodium salt (EDTA-Na) is dissolved in 50mL of deionized (DI) water, followed by the addition of 0.388 g (0.8 mmol) of $Bi(NO₃)₃·5H₂O$. The mixture is ultrasonicated to dissolve the white precipitate hydrolyzed from $Bi(NO₃)$ ³. Then, 0.360 g (2.4 mmol) of mercaptosuccinic acid (MSA) is added with stirring. This solution is transferred into a 50mL Teflon-lined stainless-steel autoclave with placing an Al or Ti foil inside, which is maintained at 180° C for 8h and subsequently cooled down to room temperature naturally. The final gray solid product is collected by centrifugation, washed with distilled water and absolute ethanol for several times, and finally is dried in air at 60° C overnight.

2.2 Preparation of Sandwich Structures

MSM sandwich structures are fabricated with Al and Ti foil as bottom electrodes, on which the material $Bi₂S₃$ are grown directly. The top electrodes are through mechanically pressing a flat indium granule on $Bi₂S₃$ microflowers.

2.3 Material Characterizations

The morphology and composition of $Bi₂S₃$ microflowers are characterized by a

Page 5 of 23 RSC Advances

field emission scanning electron microscope (FESEM, FEI QUANTA FEG250) equipped with an energy dispersive x-ray spectroscopy (EDS, INCA MAX-50) and a high-resolution transmission electron microscope (HRTEM, JEM-2100F, JEOL) and selected-area electron diffraction (SAED). The phase of $Bi₂S₃$ microflowers is checked with X-ray diffraction (XRD, D8-Advance, Bruker), and is further measured by a Raman spectroscopy (Nomadic). Continuous photocurrent investigations are performed on a spectral response device (44939A, U.S.A). A diode laser (white light) is chosen as the light source.

2.4 Electrochemical measurement

A typical three-electrode experimental cell is used for measuring the electrochemical properties of the working electrode. The $Bi₂S₃$ microflowers serve as the working electrode, while a platinum foil electrode and a standard calomel electrode (SCE) are used as the counter and reference electrodes, respectively. All electrochemical measurements are carried out in 1.0 M $Na₂SO₄$ solution as an electrolyte. Cyclic voltammetry (CV), galvanostatic charge–discharge, and electrochemical impedance spectroscopy (EIS) are carried out on the CHI660 electrochemical work station (Chenhua, Shanghai).

2.5 Photocatalytic activity

Photocatalytic activities of the samples are measured by the degradation of methyl orange (MO) under UV irradiation with photocatalytie reaction apparatus (XPA series-7, Nanjing), 500W mercury lamp. Typically, a sample of $Bi₂S₃$ microflowers on Ti foil as a photocatalyst is added into 15 mL of MO (20 mg/L) aqueous solution. The solution is stewing for 10 min in the dark to ensure the adsorption–desorption equilibrium between the photocatalyst and the MO and then shined under UV irradiation. The photocatalytic degradation of the organic dye is estimated by measuring the absorbance of dye solution in the presence of photocatalyst exposed at different time intervals. The degradation efficiencies of the organic dyes are analyzed by monitoring the dye decolorization at the maximum absorption wavelength using UV-vis spectrophotometer (TU-1900/1901, Beijing). All the experiments are conducted at room temperature.

RSC Advances Accepted Manuscript RSC Advances Accepted Manuscript

3. Results and discussion

Fig. 1 depicts the morphologies of as-prepared $Bi₂S₃$ microflowers. The SEM images with low magnification (Fig. 1(a) and (b)) indicate the formation of $Bi₂S₃$ microflowers structures. Bundles of $Bi₂S₃$ microflowers are observed as shown in Fig. 1(a), and the typical architectures are presented in Fig. 1(b) in which the length of the branches is estimated to be $18{\sim}20$ µm. Interestingly, the enlarged SEM image (Fig. $1(c)$) indicates that the microstructures of Bi₂S₃ microflowers are composed of many microrods, which have sharp apexes and flaky surfaces with the width of about 1.7 µm.

Fig. 2 shows the XRD pattern of the obtained $Bi₂S₃$ microflowers. The diffraction peaks correlate very well with the reference diffractions of a pure orthorhombic Bi_2S_3 phase (JCPDS NO.65-2435), and no additional peaks are observed indicating that no impurities or secondary phases exist. Raman spectroscopy is further used for the structure investigation, as shown in Fig. 3. Only one peak at about 250cm^{-1} is observed, belonging to Bi-S bond Raman shift peak, which is consistent with previous report.³⁶ No other peaks are observed indicating that the obtained products are pure $Bi₂S₃$ phase without other impurities.

The morphology and structure of the $Bi₂S₃$ microflowers are further detected by TEM. Fig. 4(a) shows a typical TEM image of $Bi₂S₃$ microflower. It can be seen that the diameters of nanorods range from 100 nm to 200 nm and the length reach to 40 µm. The individual nanorod with uniform diameter of 141 nm is presented in Fig. 4(b). The EDS image (inset in Fig. 4(b)) exhibits the elements of Bi, S, O and Cu, indicating that the nanorods are composed of elements of Bi and S. The elements of O and Cu come from Cu grid for TEM observation. The HRTEM image (Fig. $4(c)$) demonstrates that the interplanar spacing values are of about **~**0.31 and 0.35 nm, corresponding to the spacing of the (211, 230) and (111) planes of the orthorhombic $Bi₂S₃$, respectively. The electron diffraction pattern in Fig. 4(d) confirms that these nanostructures are pure $Bi₂S₃$ crystalline phase, which is in good agreement with XRD pattern shown in Fig. 2.

Fig. 5 shows the absorption spectrum of $Bi₂S₃$ microflowers and the inset shows

a plot of $(\alpha h v)^{2/3}$ verses hv, where α is the absorption coefficient (which can be obtained by the absorbance, A, following Beer's law, $A = \alpha I$, l is the optical path length), h is Planck's constant and ν is the frequency of the optical radiation. Following the relation of α hv=(hv-E_g)^{3/2}, the plot in the inset has an intercept of nearly 1.65 eV, indicating the optical bandgap (E_g) of 1.65 eV, which is consistent with the onset of the absorption spectrum $(\sim 751 \text{ nm}, \text{Fig.5}).$

To investigate the electrical-transport properties of the oriented $Bi₂S₃$ microflowers, a MSM sandwich structure is fabricated by mechanically pressing a indium granule as a top electrode on the microflowers. In our work, we made two kinds of devices with Al and Ti foil as the bottom electrode, on which the material is grown directly. *I*-*V* curves under dark and light illumination are measured with Al and Ti bottom electrode at room-temperature by applying a bias voltage with the range from -0.5 to +0.5 V. The *I-V* characteristics of the $Bi₂S₃$ array devices exhibit rectifying behavior, as shown in Fig. 6. Because the $Bi₂S₃$ -metal contact is usually a Schottky contact, then the as-fabricated device can be considered as a back-to-back Schottky diode, of which the schematic view is shown in Fig. $6(c)$. On occasion that the electrical field is high enough to overcome the Schottky barrier height at the Bi₂S₃-metal interface, one of the two Schottky contacts can be reversely biased at an applied bias voltage, and the current of the MSM diode is mainly determined by the reverse-biased diode. In Fig. 6, all $I-V$ curves of $In/Bi₂S₃/Al$, and $In/Bi₂S₃/Ti$ structures display nonlinear and asymmetric behavior. Especially in the case of $In/B₁$, S₃/Al structure, the current with a negative bias is much lower than that with a positive bias, which is similar to the results of Bao et al^{37} . They made three kinds of device with either tungsten, copper, or silver foil as the top electrode and in case of $Ag / Bi₂S₃/W$ structure, the current with a positive bias is much higher than that with a negative bias. The asymmetric *I*-*V* curves are attributed to the different contacts of the two Bi_2S_3/M devices, as the Bi_2S_3 structures are fabricated on the Al substrate during the reaction, then the firm contact with the $Bi₂S₃$ structures can form a better electrical contact with a lower electron-transfer barrier height, compared with a higher electron-transfer barrier caused by the mechanical contacts between $Bi₂S₃$ structures

RSC Advances Page 8 of 23

RSC Advances Accepted Manuscript RSC Advances Accepted Manuscript

and In. The experimental results show that *I*-*V* behavior can be modified by changing the growth substrates (Ti and Al), which can be explained by the change in the barrier height of the semiconductor/metal contact. From the geometrical length $(L_{Al}=L_{Ti}=2$ cm) of the substrates, the apparent conductivities of 8.5**×**10-8 S/cm,1.15**×**10-8 S/cm in dark, and 5×10^{-7} S/cm, 3.4×10^{-7} S/cm under illumination (V_{applied} = 5 V) for the microstructure with Al and Ti foil as the bottom electrodes, respectively. Li et al.³⁸ made similar sandwich cells for nano/microcrystals and observed the enhanced conductivity when light was on, and a fast response time was observed, indicating its potential application in photoelectronic switches. Furthremore, Bao et al³⁷ obtained the apparent relative conductivities of 1.5×10^{-6} , 2.7×10^{-7} , and 1.9×10^{-4} S/cm (V_{applied}= 6 V) for the array with W, Cu, and Ag foil as the top electrodes, respectively. Our results show similar order of magnitude with them. Konstantatos et al³⁹ reported the current-voltage characteristics of the device in dark and illuminated conditions, they found that no photoconductive gain was observed below 1 V and asignificant dark conductance was found in bias regime beyond 1 V. From Fig. 6, it can be found that the electrical conductance changes about six and thirty times under illumination at either a negative bias or a positive bias. It can be explained that many electrons had been excited from the valence band to the conduction band. Furthermore, two Schottky contacts are formed between Bi_2S_3 microflowers and metal contacts.

The photocatalytic activity of the as-synthesized $Bi₂S₃$ microflowers is evaluated by the degradation of methyl orange (MO) under UV irradiation. Fig. 7(a) presents the absorption spectra of MO aqueous solution in the presence of $Bi₂S₃$ microflowers under UV irradiation at different time intervals. As the exposure time prolonging, the intensities of absorption peaks rapidly decrease as a result of the degradation of MO. About 95% of MO is degraded within 390 min by the $Bi₂S₃$ microflowers, suggesting the almost complete degradation of MO. The result shows that the $Bi₂S₃$ samples possessed good photocatalytic activity in the degradation of MO. In order to check the photodegradation activity easily, the C/C_o vs time curves of the photodegradation of MO with and without $Bi₂S₃$ sample under UV irradiation are plotted in Fig. 7(b). Under similar experimental conditions, the MO with $Bi₂S₃$

microflowers photocatalyst is degraded very fast compared with the control (MO without Bi_2S_3) confirms that Bi_2S_3 microflowers have good photodegradation activity in the degradation of MO under UV irradiation. Chen et al.⁴⁰ fabricated $Bi₂S₃$ hierarchical nanostructure and the similar results were observed. They found that that Bi₂S₃ hierarchical nanostructure exhibited superior photocatalytic performance to pure Bi2S3, and about 97% of MO was degraded within 4 h, they ascribed them to large specific surface area, hierarchical nanostructure.

The electrochemical performances of $Bi₂S₃$ microflowers are evaluated as the working electrode for supercapacitor applications. Fig. 8(a) presents the cyclic voltammetry (CV) curves of the $Bi₂S₃$ microflowers electrode at scan rates of 10, 20, 30, 40, 50, 80 and 100 mV s^{-1} , respectively with a potential ranging from -0.001V~-0.7V. The CV curves are different from the ideal rectangular shape of the typical electric double-layer capacitance. There are a couple of redox peaks of $Bi₂S₃$, which suggest that the capacitance is mainly from the pseudocapacitance of $Bi₂S₃$. In addition, increasing the scan rate leads to further augment of the CV curve and the redox peaks, indicating that the redox reactions of $Bi₂S₃$ are rapid. Fig. 8(b) depicts the charge–discharge behavior of the $Bi₂S₃$ microflowers electrode between 0 V and -0.7 V at different current densities by cathodic and anodic current of 0.001A. Areal and specific capacitance can be calculated as follows: Ca=I× $\Delta t/(S \times \Delta V)$, where I is the constant discharge current, ∆t represents the discharge time, ∆V designate potential drop during discharge time, and S is the geometrical area of the electrode. According to the above equation, the specific capacitance of the $Bi₂S₃$ microflowers electrode is calculated to be about 185.7, 94.3, 72.9, 57.1, 50 and 36 F cm⁻² at the current densities of 1, 2, 3, 4,5 and 10 mA cm⁻², respectively (Fig. 8(c)), and its high frequency and low frequency are 1×10^5 Hz and 0.01Hz. Electrochemical impedance spectroscopy (EIS) is also employed to characterize the $Bi₂S₃$ microflowers electrodes. As shown in the Nyquist plots (Fig. 8(d)), with a potential ranging from $-0.007V \sim 0.185V$, the equivalent series resistance (ESR) of the Bi₂S₃ microflowers electrode is about 1.96 Ω , indicating a lower diffusion resistance and charge-transfer resistance.

4. Conclusions

In conclusion, $Bi₂S₃$ rod-formed flowers have been synthesized using a facile hydrothermal approach. This template-free approach may provide a new universal method for the growth of $Bi₂S₃$ microflowers on a conductive substrate. MSM back-to-back Schottky diodes are formed and the *I*-*V* characteristics of as-fabricated devices exhibit a unique rectifying behavior, indicating its great potential application in electronic nanodevices photocatalytic. The galvanostatic charge/discharge performance illustrates that as-synthesized $Bi₂S₃$ microflowers has the well performance of discharge efficiency at current density from 1mA cm^{-2} to 10 mA cm^{-2} . The results presented herein highlight the significance of deliberate interfacial engineering in the development of low-cost, large-scale, and practical supercapacitor electrodes and manipulation of the supercapacitor performance.

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Page 13 of 23 RSC Advances

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Fig.1 SEM images of $Bi₂S₃$ microflowers with low (a, c) and enlarged (b, d) magfications parepared at 180 °C for 8h.

Fig.2 XRD patterns of the as-synthesized $Bi₂S₃$ microflowers. (JCPDS: 65-2435)

Fig.3 Raman pattern of as-prepared $Bi₂S₃$ microflowers.

Fig.4 (a) Typical TEM image of $Bi₂S₃$ finger-like nanostructures with multiple branches and brush-like single nanorod in (b); (c) HRTEM image of multiple $Bi₂S₃$ nanorods as individual branch; (d) SAED pattern and EDAX spectra of $Bi₂S₃$ nanostructure (inset in Fig. 4(b)).

Fig.5 The absorption spectrum of the Bi₂S₃ microflower.

Fig.6 I-V curves of Bi₂S₃ microflowers devices measured at room temperature in the dark and with illumination

Fig.7 (a) Absorption spectra of MO aqueous solution (20 mg/L) in the presence of $Bi₂S₃$ sample at different time intervals(b) the C/C_o vs time curves of MO and $Bi₂S₃$ sample photodegration under UV irradiation.

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Table of contents

Controlled assembly of Bi2S3 architectures for the all-solid-state supercapacitor electrodes and highly efficient photocatalysts

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Bi₂S₃ microflowers were controllably fabricated, and their improved electrochemical and photocatalytic properties were observed.

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