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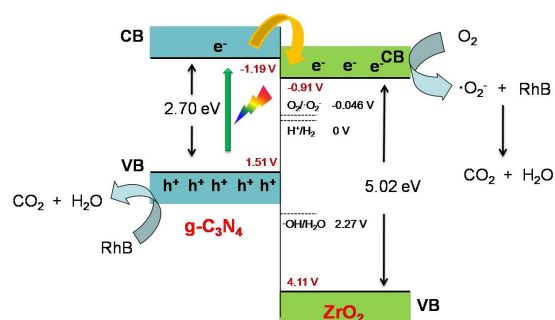
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$\text{ZrO}_2/\text{g-C}_3\text{N}_4$ composite was prepared by directly heating ZrO_2 -melamine mixture and showed excellent activity in rhodamine photodegradation.

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ARTICLE TYPE

Synthesis and characterization of ZrO₂/g-C₃N₄ composite with enhanced visible-light photoactivity for rhodamine degradation

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The visible-light-driven ZrO₂/g-C₃N₄ hybrid photocatalysts were prepared by direct heating of ZrO₂ and melamine. Compared to pure g-C₃N₄ or ZrO₂, the synthesized ZrO₂/g-C₃N₄ exhibited much higher photocatalytic activity for rhodamine (RhB) degradation under visible light irradiation. In order to reveal the origin of the high photoactivity, the ZrO₂/g-C₃N₄ composites were characterized by various techniques including N₂ adsorption, thermogravimetric analysis (TG), X-ray diffraction (XRD), Fourier transform infrared spectroscopy (FT-IR), scanning electron microscopy (SEM), transmission electron microscopy (TEM), X-ray photoelectron spectroscopy (XPS), UV-vis diffuse reflectance spectroscopy (DRS), photoluminescence spectroscopy (PL), and electrochemical method. The characterization results demonstrated that ZrO₂ nanoparticles were well distributed on the surface of g-C₃N₄. Although the anchoring of ZrO₂ on g-C₃N₄ increased the surface area and light absorption ability, the formed hetero-junctions between the two semiconductors which retarded the recombination of electron and holes were believed to result in the enhanced photoactivity of ZrO₂/g-C₃N₄ composite. In addition, it was found that holes and •O₂⁻ generated in the photocatalytic process played a key role in RhB degradation over the ZrO₂/g-C₃N₄ hybrids.

1. Introduction

Since 1972, heterogeneous photocatalysis technique has received considerable attention as an attractive strategy in water purification and converting solar energy into the form of hydrogen by splitting water. A large number of metal oxide semiconductors, including TiO₂, ZnO and ZrO₂, have been investigated and reported¹⁻⁵. However, their application in a large scale is greatly limited since they can only absorb the ultraviolet (UV) light which occupies no more than 4% of the solar spectrum. Therefore, several approaches, such as metal ions or non-metal ions doping, dye sensitization, and semiconductor doping, have been reported to extend their light response⁶⁻¹¹. The last approach is considered as the most efficient way, and many semiconductors have been applied to improve their photocatalytic activity under visible light. An interesting phenomenon is that the modification is mainly focused on TiO₂ and ZnO⁶⁻¹¹, while few research concerned on ZrO₂ has been reported. The extra wide band gap might be the main reason, which induces the difficulty in shortening the band gap of ZrO₂. Another possible reason is its high conduction band. Based on the reported literature, the CB potential of ZrO₂ is estimated to be about -1.0 eV¹² which is more negative than that of TiO₂ (-0.12 eV)³ and ZnO (-0.45 eV)¹³. Few semiconductors contain such negative CB that can be used to sensitize ZrO₂.

Recently, Wang and his coworkers reported a novel photocatalyst, graphitic carbon nitride (g-C₃N₄), which exhibits a

high photocatalytic activity for the splitting of water into hydrogen using solar energy¹⁴. The metal-free semiconductor also exhibits activity for the degradation of organic pollutants and photocatalytic CO₂ reduction under visible light irradiation^{15,16}. Combined with the merits of low cost, thermal and chemical stability, g-C₃N₄ semiconductor is considered as a valuable material for photocatalysis-driven applications. However, to date bare g-C₃N₄ still suffers the disadvantage of low quantum efficiency, which limits the photocatalytic performance. Therefore, a variety of semiconductors were decorated on g-C₃N₄ to improve its photoactivity¹⁷⁻²². For example, Wang et al. reported a Ag₃VO₄/g-C₃N₄ photocatalyst and its application for triphenylmethane dye degradation¹⁷. The loading of Ag₃VO₄ greatly promoted the degradation rate of g-C₃N₄. Huang et al. synthesized CeO₂/g-C₃N₄ composite by mixing-calcination method and investigated its photoactivity in RhB photodegradation¹⁸. The results indicated that the photocatalytic activity of the composite material was enhanced by the loading of CeO₂. Other similar catalysts, such as LnVO₄ (Ln=Sm, La, Y, Bi)/g-C₃N₄¹⁹⁻²², WO₃/g-C₃N₄²³, MoO₃/g-C₃N₄²⁴, graphene/g-C₃N₄²⁵, Ag₃PO₄/g-C₃N₄²⁶, AgX/g-C₃N₄²⁷, and CdS/g-C₃N₄²⁸, were also reported. However, to the best of our knowledge, there has been no report regarding the coupling of g-C₃N₄ with ZrO₂ to enhance the photocatalytic activity. Actually, the conduction band (CB) of g-C₃N₄ is located at ~-1.2 eV¹⁹, indicating g-C₃N₄ can sensitize ZrO₂. The ZrO₂ coupled g-C₃N₄ composite might be an efficient photocatalyst.

In this study, we present the first example of a ZrO₂-

hybridized g-C₃N₄ photocatalyst fabricated by a mixing-calcination method. ZrO₂ is prepared by hydrothermal method and shows the morphology of nano spindle. After hybridization with the spindle-like ZrO₂, the visible photocatalytic activity of g-C₃N₄ for RhB photodegradation is effectively enhanced, demonstrating the hybrid effect existed between ZrO₂ and g-C₃N₄. On the basis of the investigation results, the origin of the hybrid effect is discussed.

2. Experimental Section

2.1 Catalysts preparation

All chemicals were analytical pure reagent and purchased commercially without further purification. Pure g-C₃N₄ was prepared by directly heating melamine at 520 °C in a muffle furnace for 4 h at a heating rate of 10 °C/min. The preparation of ZrO₂ referred to the previous literature²⁹. Typically, 2.204 g cetyltrimethylammonium bromide (CTAB) was first dissolved in 40 mL water with stirring at 40 °C to obtain a clear micellar solution. Then 3.6 g zirconyl chloride was added to the solution. This combined solution was stirred for 15 min and then NaOH (1 mol/L) was added until the pH reached 11.5. After that, the mixture was transferred into a 100 mL autoclave with an inner Teflon lining and maintained at 100 °C for 24 h. The resulting white precipitate was collected by centrifugation, washed several times with ethanol and deionized water, and dried in an oven at 80 °C for 12 h.

The ZrO₂/g-C₃N₄ composites were prepared as follows: ZrO₂ and melamine with different ratio were mixed in a mortar and then grounded for 30 min. The resultant mixed powder was put into a crucible with a cover and then heated at 520 °C in a muffle furnace for 4 h with a heating rate of 10 °C/min. After the temperature decreased to room temperature, the ZrO₂/g-C₃N₄ hybrids with various ZrO₂ contents were obtained. Thermogravimetric analysis (TG) was investigated to determine the weight contents of ZrO₂ in ZrO₂/g-C₃N₄. Based on the characterization results (Fig. S1), the ZrO₂/g-C₃N₄ composites with different ZrO₂ concentrations were named as 10wt.% ZrO₂/g-C₃N₄, 20.9wt.% ZrO₂/g-C₃N₄, 29.7wt.% ZrO₂/g-C₃N₄ and 36.1wt.% ZrO₂/g-C₃N₄, respectively.

2.2 Characterizations

The TG analysis (Netzsch STA449) of the catalysts was carried out in a flow of air (20 mL/min) at a heating rate of 10 °C/min. The N₂ adsorption measurement was performed on an Autosorb-1 (Quantachrome Instruments) by the Brunauer-Emmett-Teller (BET) method at 77 K. The XRD characterization of the catalysts was carried out on a Philips PW3040/60 using Cu K α radiation. The SEM pictures were taken on a field emission scanning electron microscope (Hitachi S-4800). The TEM images were collected with a JEOL-2100F transmission electron microscope at an accelerating voltage of 200 kV. The DRS spectra of catalysts were recorded on a UV-vis spectrometer (Thermo Nicolet Evolution 500) using BaSO₄ as a reference sample. The XPS spectra were obtained by using a Quantum 2000 Scanning ESCA Microprobe instrument using AlK α . The PL spectra were collected on FLS-920 spectrometer (Edinburgh Instrument),

using a Xe lamp (excitation at 365 nm) as light source.

The electrochemical impedance spectroscopy (EIS) and photocurrent responses (PR) measurements were performed using a CHI 660B electrochemical workstation with a standard three-electrode cell at room temperature. The prepared sample, Ag/AgCl (saturated KCl), and Pt wire were used as the working electrode, the reference electrode, and the counter electrode, respectively. The working electrodes was prepared as follows. An indium tin oxide (ITO) glass pieces with a size of 1.5×5 cm was cleaned successively by acetone, boiling NaOH (0.1 mol/L), deionized water, and dried in an air stream. Then, 0.018g sample and 0.002g polyvinylidene fluoride was mixed and ground for three minutes. After the addition of three drops of 1-Methyl-2-pyrrolidinone and the subsequently ultrasonic treatment for 20 min, the obtained suspension was coated onto the ITO glass substrate. The coated area on the ITO glass was controlled to be 0.8×0.8 cm. Finally, the coated ITO glass was dried at 50 °C to obtain the working electrode. The EIS spectra were recorded by applying an AC voltage of 10 mV amplitude in the frequency range of 10⁵ Hz to 10⁻² Hz with the initial potential (0 V) in 0.01 M Na₂SO₄. For PR measurement, a 350 W Xe arc lamp served as the light source and Na₂SO₄ (0.5 M) aqueous solution was used as the electrolyte.

2.3 Photocatalytic reaction

The photocatalytic activities of ZrO₂/g-C₃N₄ hybrids were tested by RhB (10 mg/L) degradation under visible-light irradiation in an apparatus. Visible light (420 < λ < 800 nm) generated by a 500 W Xe lamp equipped with two optical filters was used as the light source. The power density of visible light at the position of reactor was approximately 7.4 mW/cm² and the catalyst content was 0.2 g/100 mL. Prior to irradiation, the suspensions were magnetically stirred in dark for 60 min. During the RhB photodecomposition, samples were withdrawn at regular intervals and centrifuged to separate solid particles for analysis. The concentration of the RhB was determined by a UV-vis spectroscopy at its maximum absorption wavelength (about 554 nm). The examination experiment of reactive species is similar to the photodegradation experiment. A quantity of scavengers was introduced into the RhB solution prior to addition of the catalyst. The concentration of scavengers was controlled to be 0.01 mol/L according to the previous studies³⁰⁻³³.

3. Results and discussion

3.1 Characterizations of ZrO₂/g-C₃N₄ composites.

The morphologies of pure g-C₃N₄, ZrO₂, and ZrO₂/g-C₃N₄ composite were investigated by SEM and TEM. ZrO₂ shows as spindle nanoparticles with an average particle size of 50 nm (Figures 1a and 1b), which leads to its high surface area of 43.2 m²/g. Different from ZrO₂, pure g-C₃N₄ has a low surface area of 13 m²/g, which can be attributed to its special morphology. As shown in Figures 2a and 2b, the polymer is shown as an aggregation of many sheets due to its layered structure¹⁴. The different morphologies of ZrO₂ and g-C₃N₄ make the two semiconductors be easily distinguished in the SEM image of ZrO₂/g-C₃N₄ composite. It can be observed that ZrO₂

nanoparticles are coated on the surface of g-C₃N₄ in the composite (Fig. 1e). The hybrid structure of ZrO₂/g-C₃N₄ composite can be verified further by the TEM analysis (Fig. 1f). ZrO₂ particles, which are darker than g-C₃N₄ due to its heavier atom, are observed to disperse finely on g-C₃N₄ sheets. In addition, considering that the ZrO₂/g-C₃N₄ was ultrasonicated for 30 min before the TEM analysis, the result in Fig. 1f indicates that ZrO₂ particles are closely adhered on g-C₃N₄ sheets and the interaction between ZrO₂ and g-C₃N₄ is strong, which is beneficial for the formation of ZrO₂-g-C₃N₄ hetero-junction. The BET surface areas result obtained via the method of N₂ adsorption shows that the addition of ZrO₂ increases the surface area of g-C₃N₄. The BET surface area of 10wt.% ZrO₂/g-C₃N₄, 20.9wt.% ZrO₂/g-C₃N₄, 29.7wt.% ZrO₂/g-C₃N₄ and 36.1wt.% ZrO₂/g-C₃N₄ samples are 21.7, 25.8, 33.1 and 36.4 m²/g, respectively.

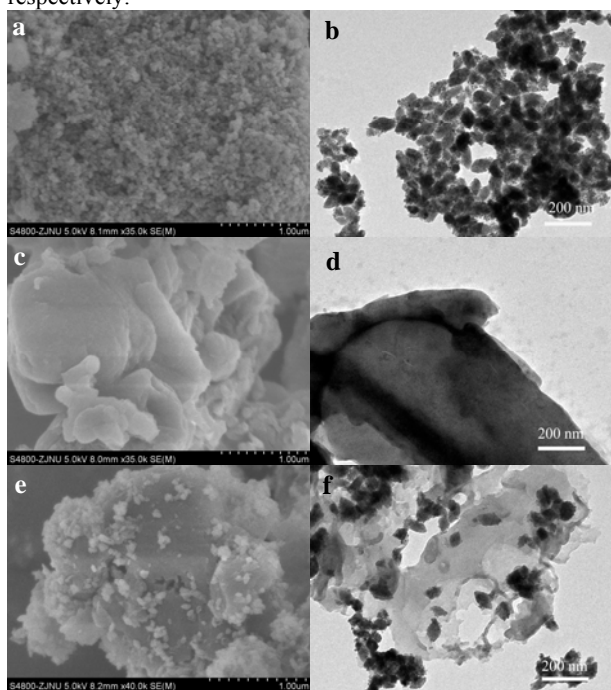


Fig. 1 SEM and TEM images of ZrO₂ (a,b), g-C₃N₄ (c,d), and 20.9wt.% ZrO₂/g-C₃N₄ composite (e,f).

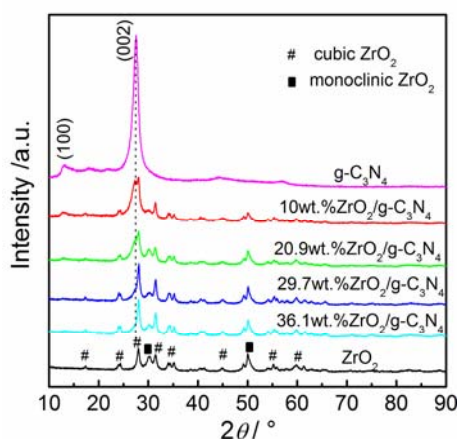


Fig. 2 XRD patterns of ZrO₂/g-C₃N₄ composites with different ZrO₂ concentration.

The crystalline structure of ZrO₂/g-C₃N₄ was investigated by XRD. The synthesized ZrO₂ consists of both monoclinic phase (JCPDS No. 37-1484) and body-centered cubic phase (JCPDS No. 49-1642). g-C₃N₄ shows its characterization peaks at 27.4 ° and 13.0 °, which can be indexed to (002) and (100) diffraction plane of the graphite-like carbon nitride¹⁵. For ZrO₂/g-C₃N₄ composite, it exhibits diffraction peaks corresponding to both g-C₃N₄ and ZrO₂. The diffraction peaks of ZrO₂ intensify gradually as the increase of ZrO₂ content, while the g-C₃N₄ peaks decrease. With the exception of ZrO₂ and g-C₃N₄, no other phase is detected, which reflects the hybrid structure of ZrO₂/g-C₃N₄.

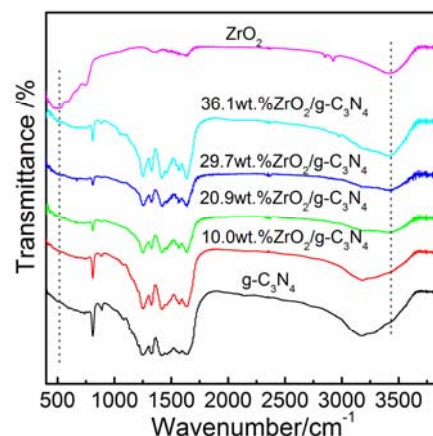


Fig. 3 FT-IR spectra of ZrO₂/g-C₃N₄ composites with different ZrO₂ concentration.

The FT-IR spectra of ZrO₂, g-C₃N₄, and ZrO₂/g-C₃N₄ composite are shown in Fig. 3. The broad absorption bands at around 509 cm⁻¹ and 750 cm⁻¹ are attributed to the Zr-O vibration absorption³⁴, while the broad absorption region with a maximum around 3446 cm⁻¹ can be attributed to the hydroxyl groups of hydrated oxide surface and the adsorbed water³⁴. For pure g-C₃N₄, the observed strong peaks in the range of 1200-1700 cm⁻¹ can be ascribed to the typical stretching vibration of CN heterocycles¹². The peak at 808 cm⁻¹ can correspond to the breathing mode of triazine units¹². In addition, a broad absorption band with a maximum around 3220 cm⁻¹ corresponding to the N-H groups is also observed¹². The FT-IR spectra of ZrO₂/g-C₃N₄ are similar with that of pure g-C₃N₄. However, the change in the bands of 509 and 3446 cm⁻¹ with the variation of ZrO₂ content can still demonstrate the existence of ZrO₂ phase. The FT-IR result shows good coincidence with XRD analysis and indicates the hybrids structure production of ZrO₂/g-C₃N₄ composite.

Fig. 4 shows the XPS spectra of g-C₃N₄, ZrO₂, and ZrO₂/g-C₃N₄ composite. All the signals of C, N, Zr, O are detected in the survey XPS spectrum of ZrO₂/g-C₃N₄ composite (Fig. 4a), indicating its hybrids structure, which is consistent with the XRD and FT-IR experiments. The high-resolution C1s XPS spectrum of ZrO₂ shows one peak at 284.6 eV which can be attributed to the contaminated carbon^{16,17}. In the case of g-C₃N₄ and ZrO₂/g-C₃N₄, another strong C1s peak is observed, which can be ascribed to the sp²-bonded carbon (N=C=N)^{16,17}. Meanwhile, the addition of ZrO₂ affects the binding energy (BE) of C1s of g-C₃N₄. The C1s BE of ZrO₂/g-C₃N₄ shifts from 287.7 eV to 287.4 eV,

indicating the interaction of ZrO_2 and $\text{g-C}_3\text{N}_4$, just as the previous literatures^{35,36}. The same phenomenon is also observed in the Zr 3d XPS spectra. As shown in Fig. 4c, Zr 3d_{5/2} and 3d_{3/2} BE of ZrO_2 are located at 181.0 eV and 184.4 eV, respectively, which are very close to those reported in the literature for pure ZrO_2 ³⁷. As compared to pure ZrO_2 , Zr 3d peak of $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ exhibits a slight positive shift due to the interaction between ZrO_2 and $\text{g-C}_3\text{N}_4$. This kind of interaction is beneficial for the formation of hetero-junction between the two semiconductors and the subsequently promotion in separation of electron-hole pairs. Fig. 4d shows the valence band (VB) XPS spectra of $\text{g-C}_3\text{N}_4$ and ZrO_2 . The position of the valence band edges of $\text{g-C}_3\text{N}_4$ and ZrO_2 are 1.51 eV and 4.11 eV, respectively.

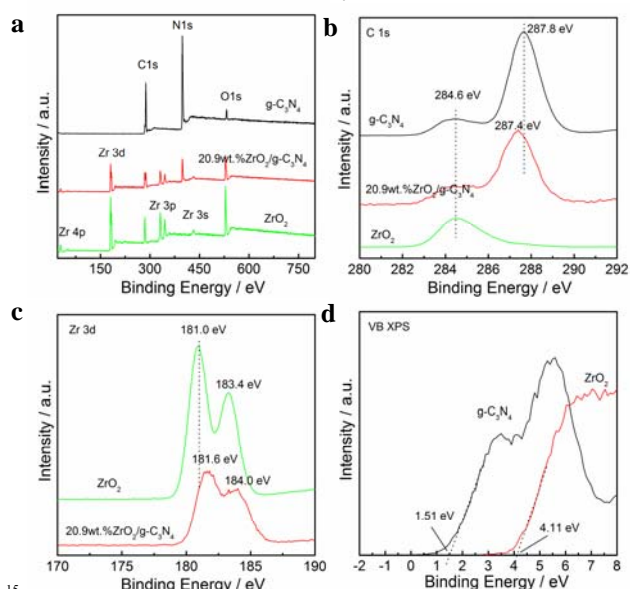


Fig. 4 XPS spectra of $\text{g-C}_3\text{N}_4$, ZrO_2 , and 20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ composite. (a) survey spectra, (b) C1s, (c) Zr 3d, (d) VB XPS spectra of ZrO_2 and $\text{g-C}_3\text{N}_4$.

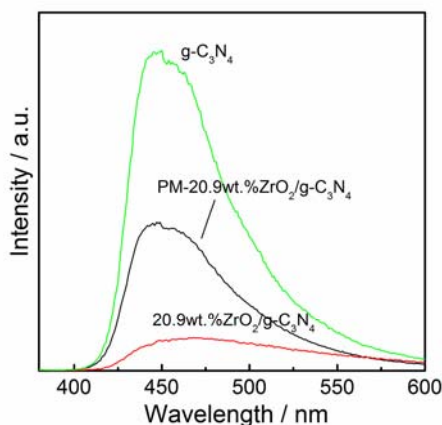


Fig. 5 PL spectra of 20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$, PM-20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ and pure $\text{g-C}_3\text{N}_4$ excited at 365 nm.

The effect of the formed $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ hetero-junction on the separation efficiency of electron-hole pairs was investigated by PL spectra. As shown in Fig. 5, a strong emission peak at around 460 nm is observed in pure $\text{g-C}_3\text{N}_4$, which can be attributed to the bandgap transition emission with the energy of emission light

approximately equal to the bandgap energy of $\text{g-C}_3\text{N}_4$. However, in the PL spectrum of $\text{ZrO}_2/\text{g-C}_3\text{N}_4$, a considerable fluorescence quenching in the same position is detected, indicating that the addition of ZrO_2 significantly inhibits the recombination of electrons and holes³⁸. For comparison, the PL spectrum of the physical mixture PM-20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ is also displayed in Fig. 5. The PL peak of PM-20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ is weaker than that of $\text{g-C}_3\text{N}_4$, but stronger than that of PM-20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$, indicating that the decreased concentration of $\text{g-C}_3\text{N}_4$ contributes the weakened PL peak. However, there definitely exists charge transfer between $\text{g-C}_3\text{N}_4$ and ZrO_2 component, which further retards the recombination of electrons and holes.

In addition to the PL technique, EIS and transient photocurrent experiments are also effective methods to investigate the separation efficiency of electron-hole pairs in a photocatalyst. Fig. 6a shows the EIS changes of $\text{g-C}_3\text{N}_4$, ZrO_2 , and 20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ electrodes. It can be observed that the arc size of the three electrodes is 20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ < ZrO_2 < $\text{g-C}_3\text{N}_4$. In general, the decreased semicircle diameter indicates the smaller charge-transfer resistance on the electrode surface which results in an effective electron-hole separation³⁹⁻⁴⁰. Hence, the data in Fig. 6a suggests that $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ composite has the highest efficiency in charge separation. The same result is also obtained by the transient photocurrent responses experiment. As shown from Fig. 6b, the photocurrent of 20.9 wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ is much higher than that of $\text{g-C}_3\text{N}_4$ or ZrO_2 , which proves the $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ hybrid holds stronger ability in generating and transferring the photoexcited charge carrier under light irradiation^{41,42}.

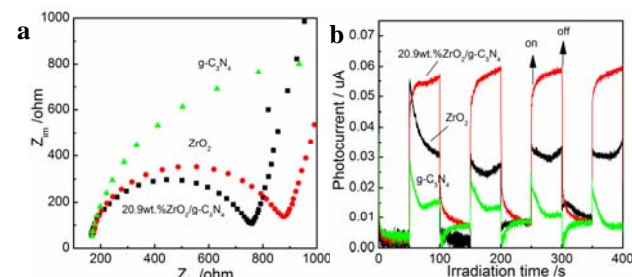


Fig.6 EIS changes (a) and transient photocurrent responses (b) of $\text{g-C}_3\text{N}_4$, ZrO_2 , and 20.9wt.% $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ electrodes.

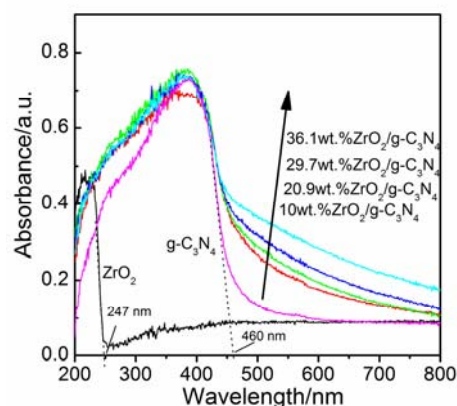


Fig. 7 UV-vis spectra of ZrO_2 , $\text{g-C}_3\text{N}_4$ and $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ composites.

The optical properties of $\text{g-C}_3\text{N}_4$, ZrO_2 , and $\text{ZrO}_2/\text{g-C}_3\text{N}_4$ composites were measured via the UV-vis DRS technique. As shown in Fig. 7, white ZrO_2 can only absorb the light with a

wavelength shorter than 247 nm, and the band gap energy is about 5.02 eV. For pure g-C₃N₄, the absorbance edge locates at around 460 nm, and the band-gap energy is determined to be 2.7 eV which is in excellent agreement with the previous value^{14,15}. The ZrO₂/g-C₃N₄ composite exhibits better photoabsorption performance than g-C₃N₄, which might be due to the interaction between ZrO₂ and g-C₃N₄. Some formed chemical bonds between the two semiconductors might result in the promoted optical property, similar as that in N-doped ZrO₂⁴³ and graphene doped TiO₂ photocatalyst^{44,45}. With increasing ZrO₂ content, the light absorption ability of ZrO₂/g-C₃N₄ is gradually strengthened. The results from DRS imply that the ZrO₂/g-C₃N₄ hybrids should possess visible-light photocatalytic activity.

3.2 Photocatalytic activities of ZrO₂/g-C₃N₄

The degradation of RhB under visible light irradiation was carried out to evaluate the photocatalytic activity of as-prepared ZrO₂/g-C₃N₄ composites. Fig. 8a displays the changes of the RhB concentration versus the reaction time over ZrO₂/g-C₃N₄ photocatalysts with different ZrO₂ concentrations. The blank test shows that RhB is stable under visible light irradiation, which indicates that the contribution of RhB photolysis can be neglected. Pure ZrO₂ has a large surface area and exhibits good adsorption for RhB. Nearly half of RhB is adsorbed on ZrO₂ after one hour adsorption in the dark. Under visible light irradiation, some RhB are desorbed due to the increased temperature on the ZrO₂ surface, resulting in the enhanced RhB concentration at the beginning of the light-on. When the irradiation time is longer than 20 min, the concentration of RhB decreases slowly. This result indicates that a part of UV light may pass through the cutoff filter and induce the poor photoactivity of ZrO₂. Compared to ZrO₂, pure g-C₃N₄ has worse adsorption of RhB, but better photoactivity under visible light irradiation. The degradation rate is 0.012 min⁻¹, which is six times higher than that of ZrO₂. The decoration of ZrO₂ on g-C₃N₄ can promote the separation efficiency of electron-hole pairs and the light absorption ability, the catalytic activity for RhB degradation is thus enhanced. With the increase of ZrO₂ concentration from 10 wt.% to 31.6 wt.%, the photocatalytic activity of ZrO₂/g-C₃N₄ increases gradually and then decreases. 20.9wt.% ZrO₂/g-C₃N₄ sample exhibits the highest degradation rate of 0.028 min⁻¹, which is 2.33 times higher than that of g-C₃N₄.

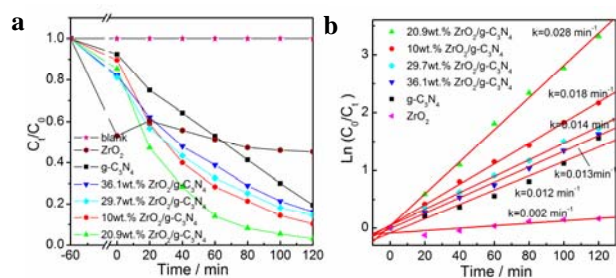


Fig. 8 Photodegradation of RhB over ZrO₂/g-C₃N₄ composites under visible light irradiation (a) and the corresponding reactive constant (b).

Although the high photoactivity of ZrO₂/g-C₃N₄ composite has been proven, the stability is important in view of its practical

applications. Hence, the stability of the 20.9wt.% ZrO₂/g-C₃N₄ composite was investigated by a six-run cycling test under the same condition. For each run, the photocatalyst was recycled, cleaned, and dried. From Fig. 9a, the photodegradation efficiency of 20.9wt.% ZrO₂/g-C₃N₄ decreases slightly in the six cycles, which indicates that the ZrO₂/g-C₃N₄ photocatalysts can be reused completely for wastewater treatment. Fig. 9b shows the photocatalytic activity of 20.9wt.% ZrO₂/g-C₃N₄ with different scavengers. From Fig. 9b, it can be observed that 2-propanol (•OH quencher)^{30,31} showed little effect on the reaction rate (k) of RhB degradation. However, the addition of benzoquinone (BQ, •O²⁻ quencher)^{32,33} leads to a great decrease of the k from 0.028 min⁻¹ to 0.001 min⁻¹. The k also had an obvious drop to 0.05 min⁻¹ in the presence of KI (h⁺ and •OH quencher)^{30,31}. This result indicates that h⁺ and •O²⁻ are two main reactive species in the photocatalytic process of ZrO₂/g-C₃N₄ hybrids.

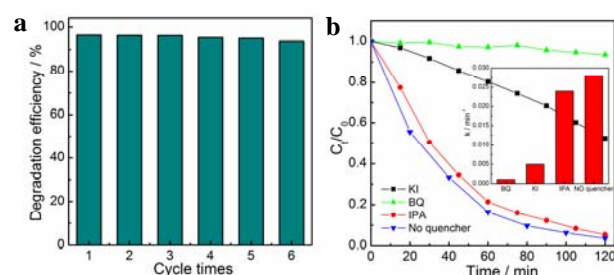


Fig. 9 The cycle test (a) and the reactive species trapping experiment (b) on 20.9wt.% ZrO₂/g-C₃N₄ composite.

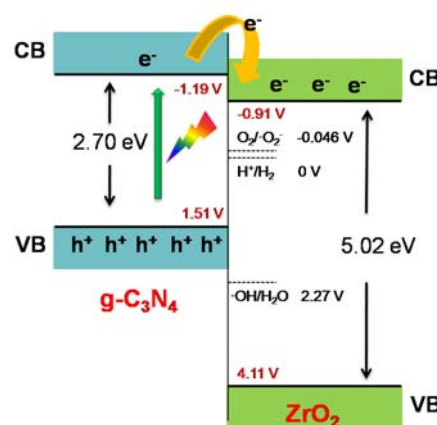
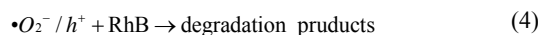
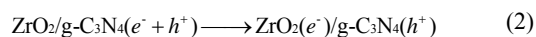
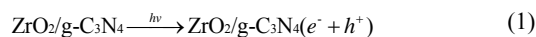


Fig. 10 Photocatalytic mechanism scheme of ZrO₂/g-C₃N₄ composite.

Therefore, it is clearly that the decoration of ZrO₂ on g-C₃N₄ generates an effective photocatalyst with high stability for the RhB degradation under visible light irradiation. The introduction of ZrO₂ can promote the surface area and light absorption capability of g-C₃N₄, which is beneficial for the photocatalytic reaction. However, the data in Fig. 8a shows that the photocatalytic activity of ZrO₂/g-C₃N₄ does not always increase with the enhancement of the two characters. For example, 36.1wt.% ZrO₂/g-C₃N₄ exhibits the highest surface area and light absorption ability. However, its photoactivity is much lower than that of 20.9wt.% ZrO₂/g-C₃N₄. This result indicates that the surface area and light absorption ability are not the dominant factor in affecting the photocatalytic activity of ZrO₂/g-C₃N₄. The key factor is the separation efficiency of electron-hole pairs. The

VB edges of ZrO₂ and g-C₃N₄ have been determined to be 4.11 eV and 1.51 eV via the VBXPS experiment, respectively. The CB edge potentials of the two semiconductors are thus obtained by using the equation of $E_{CB} = E_{VB} - E_g$. As shown in Fig. 10, the CB potential of g-C₃N₄ is -1.19 eV, which is more negative than that of ZrO₂. Under visible light irradiation, the electrons are excited from VB to CB in g-C₃N₄, which generates holes in VB of the semiconductor. The photogenerated electrons on g-C₃N₄ surfaces can easily transfer to ZrO₂ due to the difference in CB edge potentials, whereas the holes stay on the VB of g-C₃N₄. This process can effectively improve the separation of photogenerated electron-hole pairs and greatly decrease the possibility of charge recombination, resulting in the high photoactivity of ZrO₂/g-C₃N₄ composites, as proven in the PL, EIS, photocurrent and photocatalytic tests. Meanwhile, based on the mechanism in Fig. 10, the interface between ZrO₂ and g-C₃N₄ is important for the charge transfer and separation of electron-hole pairs. Excess ZrO₂ would aggregate on the g-C₃N₄ surface, which reduces the interface area between ZrO₂ and g-C₃N₄ and thereby lowers the charge separation efficiency. That's why there exists an optimal concentration of ZrO₂. The photogenerated electrons and holes are the origin of the photocatalytic reaction. Because the CB edge potential of ZrO₂ is more negative than $E_{O_2/\cdot O_2^-}$ (-0.046 V)⁴⁶, the electrons in ZrO₂ can capture O₂ and reduce it to $\cdot O_2^-$ which has been proven to be one of the main reactive species. For the holes in g-C₃N₄, the reaction of h^+ with H₂O to generate $\cdot OH$ cannot occur due to the high VB position^{1,31}. Instead, the holes directly oxidize the adsorbed RhB to inorganic products. The possible mechanism may be described as follows:



4. Conclusion

In this work, an enhanced visible-light-driven photocatalyst of ZrO₂/g-C₃N₄ was prepared by directly calcination of ZrO₂ and melamine. Among the prepared hybrid photocatalysts, the 20.9 wt.% ZrO₂/g-C₃N₄ sample exhibited the optimal photocatalytic activity for RhB degradation under visible-light irradiation. The synergetic effect between ZrO₂ and g-C₃N₄ which resulted in efficient separation of electron-hole pairs was believed to be the origin of the high photocatalytic activity. Furthermore, the reusability experiments suggested that the photocatalytic material possessed good stability. This work might provide a promising approach for treatment of dye wastewater.

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† Electronic Supplementary Information (ESI) available: TG profiles of pure g-C₃N₄ and ZrO₂/g-C₃N₄ composites (Fig. S1). See DOI: 10.1039/b000000x/

Notes and references

1. A. Fujishima, T.N. Rao, D.A. Tryk, *J. Photochem. Photobiol. C: Photochem. Rev.* 2000, **1**, 1–21.
2. X.B. Chen, S.H. Shen, L.J. Guo, S.S. Mao, *Chem. Rev.*, 2010, **110**, 6503–6570.
3. K. Mori, H. Yamashita, M. Anpo, *RSC Adv.*, 2012, **2**, 3165–3172.
4. A. McLaren, T. Valdes-Solis, G.Q. Li, S.C. Tsang, *J. Am. Chem. Soc.*, 2009, **131**, 12540–12541.
5. Y. Kohno, T. Tanaka, T. Funabiki, S. Yoshida, *Chem. Commun.*, 1997, 841–842.
6. W. Choi, A. Termin, M.R. Hoffmann, *J. Phys. Chem.*, 1994, **98**, 13669–13679.
7. R. Ashi, T. Morikawa, T. Ohwaki, K. Aoki, Y. Taga, *Science*, 2001, **293**, 269–271.
8. G.H. Qin, Z. Sun, Q.P. Wu, L. Lin, M. Liang, S. Xue, *J. Hazard. Mater.*, 2011, **192**, 599–604.
9. X.C. Wang, J.C. Yu, Y.L. Chen, L. Wu, X.Z. Fu, *Environ. Sci. Technol.*, 2006, **40**, 2369–2374.
10. X. Yang, J. Qin, Y. Li, R. Zhang, H. Tang, *J. Hazard. Mater.*, 2013, **261**, 342–350.
11. X. Yang, J. Qin, Y. Jiang, R. Li, Y. Li, H. Tang, *RSC Adv.*, 2014, **4**, 18627–18636.
12. K. Sayama, H. Arakawa, *J. Phys. Chem.*, 1993, **97**, 531–533.
13. A. Kushwaha, M. Aslam, *Electrochim. Acta*, 2014, **130**, 222–231.
14. X.C. Wang, K. Maeda, A. Thomas, K. Takanabe, G. Xin, J.M. Carlsson, K. Domen, M. Antonietti, *Nat. Mater.*, 2009, **8**, 76–80.
15. S.C. Yan, Z.S. Li, Z.G. Zou, *Langmuir*, 2009, **25**, 10397–10401.
16. J. Mao, T.Y. Peng, X.H. Zhang, K. Li, L.Q. Ye, L. Zan, *Catal. Sci. Technol.*, 2013, **3**, 1253–1260.
17. S.M. Wang, D.L. Li, C. Sun, S.G. Yang, Y. Guan, H. He, *Appl. Catal. B: Environ.*, 2014, **144**, 885–892.
18. L.Y. Huang, Y.P. Li, H. Xu, Y.G. Xu, J.X. Xia, K. Wang, H.M. Li, X.N. Cheng, *RSC Adv.*, 2013, **3**, 22269–22279.
19. T.T. Li, L.H. Zhao, Y.M. He, J. Cai, M.F. Luo, J.J. Lin, *Appl. Catal. B: Environ.*, 2013, **129**, 255–263.
20. Y.M. He, J. Cai, L.H. Zhang, X.X. Wang, H.J. Lin, B.T. Teng, L.H.

- Zhao, W.Z. Weng, H.L. Wan, M.H. Fan, *Ind. Eng. Chem. Res.*, 2014, **53**, 5905–5915.
21. J. Cai, Y.M. He, X.X. Wang, L.H. Zhang, L.Z. Dong, H.J. Lin, L.H. Zhao, X.D. Yi, W.Z. Weng, H.L. Wan, *RSC Adv*, 2013, **3**, 20862–20868.
22. Y.X. Ji, J.F. Cao, L.Q. Jiang, Y.H. Zhang, Z.G. Yi, *J. Alloy Compd.*, 2014, **590**, 9–14.
23. Y.P. Zang, L.P. Li, Y. Zuo, H.F. Lin, G.S. Li, X.F. Guan, *RSC Adv.*, 2013, **3**, 13646–13650.
24. Y.M. He, L.H. Zhang, X.X. Wang, Y. Wu, H.J. Lin, L.H. Zhao, W.Z. Weng, H.L. Wan, M.H. Fan, *RSC Adv.*, 2014, **4**, 13610–13619.
25. G.Z. Liao, S. Chen, X. Quan, H.T. Yu, H.M. Zhao, *J. Mater. Chem.*, 2012, **22**, 2721–2726.
26. P.Z. He, L.M. Song, S.J. Zhang, X.Q. Wu, Q.W. Wei, *Mater. Res. Bull.*, 2014, **51**, 432–437.
27. H. Xu, J. Yan, Y.G. Xu, Y.H. Song, H.M. Li, J.X. Xia, C.J. Huang, H.L. Wan, *Appl. Catal. B: Environ.*, 2013, **129**, 182–193.
28. J.Y. Zhang, Y.H. Wang, J. Jin, J. Zhang, Z. Lin, F. Huang, J.G. Yu, *ACS Appl. Mater. Interf.*, 2013, **5**, 10317–1032.
29. P. Trems, M.J. Hudson, R. Denoyel, *J. Mater. Chem.*, 1998, **8**, 2147–2152.
30. G.T. Li, K.H. Wong, X.W. Zhang, C. Hu, J.C. Yu, R.C.Y. Chan, P.K. Wong, *Chemosphere*, 2009, **76**, 1185–1191.
31. Y.M. He, J. Cai, T.T. Li, Y. Wu, H.J. Lin, L.H. Zhao, M.F. Luo, *Chem. Eng. J.*, 2013, **215–216**, 721–730.
32. X. Yang, H. Cui, Y. Li, J. Qin, R. Zhang, H. Tang, *ACS Catal.*, 2013, **3**, 363–369.
33. H. Cui, X. Yang, Q. Gao, H. Liu, Y. Li, H. Tang, X. Yan, *Mater. Lett.*, 2013, **93**, 28–31.
34. Q. Zhang, Y. Zhang, H.T. Li, C.G. Gao, Y.X. Zhao, *Appl. Catal. A: Gen.*, 2013, **466**, 233–239.
35. C.S. Pan, J. Xu, Y.J. Wang, D. Li, Y.F. Zhu, *Adv. Fun. Mater.*, 2012, **22**, 1518–1524.
36. X.X. Wang, S.S. Wang, W.D. Hu, J. Cai, L.H. Zhang, L.Z. Dong, L.H. Zhao, Y.M. He, *Mater. Lett.*, 2014, **115**, 53–56.
37. S. Velu, K. Suzuki, C.S. Gopinath, H. Yoshida, T. Hattori, *Phys. Chem. Chem. Phys.*, 2002, **4**, 1990–1999.
38. L.Q. Jing, Y.C. Qu, B.Q. Wang, S.D. Li, B.J. Jiang, L.B. Yang, W. Fu, H.G. Fu, J.Z. Sun, *Sol. Energ. Mat. Sol. C.*, 2006, **90**, 1773–1787.
39. H.T. Yu, X. Quan, S. Chen, H.M. Zhao, Y.B. Zhang, *J. Photochem. Photobiol. A: Chem.*, 2008, **200**, 301–306.
40. J. Lim, D. Monllor-Satocaa, J.S. Jang, S. Lee, W. Choi, *Appl. Catal. B: Environ.*, 2014, **152–153**, 233–240.
41. L.Z. Dong, Y.M. He, T.T. Li, J. Cai, W.D. Hu, S.S. Wang, H.J. Lin, M.F. Luo, X.D. Yi, L.H. Zhao, W.Z. Weng, H.L. Wan, *Appl. Catal. A: Gen.*, 2014, **472**, 143–151.
42. Y.P. Bi, S.X. Ouyang, J.Y. Cao, J.H. Ye, *Phys. Chem. Chem. Phys.*, 2011, **13**, 10071–10075.
43. Y.Y. Zhao, Y.C. Zhang, J. Li, X.H. Du, *Mater. Lett.*, 2014, **130**, 139–142.
44. K. Woan, G. Pyrgiotakis, W. Sigmund, *Adv. Mater.*, 2009, **21**, 2233–2239.
45. R. Leary, A. Westwood, *Carbon*, 2011, **49**, 741–772.
46. D.F. Wang, T. Kako, J.H. Ye, *J. Am. Chem. Soc.*, 2008, **130**, 2724–2725.