

# RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

*Accepted Manuscripts* are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

1 **Theoretical studies on the dynamics of DNA fragment translocation through multilayer**  
2 **graphene nanopores**

3 Lijun Liang<sup>1,2</sup>, Zhisen Zhang<sup>1</sup>, Jiawei Shen<sup>3</sup>, Kong Zhe<sup>4</sup>, Qi Wang<sup>1,\*</sup>, Tao Wu<sup>1</sup>, Hans Ågren<sup>2</sup>,  
4 and Yaoquan Tu<sup>2,\*</sup>

5 <sup>1</sup>Department of Chemistry and Soft Matter Research Center, Zhejiang University, Hangzhou  
6 310027, People's Republic of China

7 <sup>2</sup>Division of Theoretical Chemistry and Biology, School of Biotechnology, KTH Royal  
8 Institute of Technology, SE-10691 Stockholm, Sweden

9 <sup>3</sup>School of Medicine, Hangzhou Normal University, Hangzhou 310016, People's Republic of  
10 China

11 <sup>4</sup>College of Materials and Environmental Engineering, Hangzhou Dianzi University,  
12 Hangzhou, Zhejiang 310018, China

13

14

15 \*Corresponding authors.

16 Fax: +86-571-87951895.

17 E-mail addresses:

18 qiwang@zju.edu.cn (Q. Wang)

19 tu@theochem.kth.se (Yaoquan Tu)

20

21

22

23

24

25

26

## 1 **Abstract**

2 Motivated by several potential advantages over common sequencing technologies, solid-state  
3 nanopores, in particular graphene nanopores, have recently been much explored as biosensor  
4 material for DNA sequencing. Studies carried out on monolayer graphene nanopores aiming  
5 at single-base resolution have recently been extended to multilayer graphene (MLG) films,  
6 indicating that MLG nanopores are superior to their monolayer counterparts for DNA  
7 sequencing. However, the underlying dynamics and current change in the DNA translocation  
8 to thread MLG nanopores remain poorly understood. In this paper, we report a molecular  
9 dynamics study of DNA passing through graphene nanopores of different layers. We show  
10 that the DNA translocation time could be extended by increasing the graphene layers up to a  
11 moderate number (7) under a high electric field and that the current in DNA translocation  
12 undergoes a stepwise change upon DNA going through an MLG nanopore. A model is built to  
13 account for the relationship between the current change and the unoccupied volume of the  
14 MLG nanopore. We demonstrate that the dynamics of DNA translocation depends  
15 specifically on the interaction of nucleotides with the graphene sheet. Thus, our study  
16 indicates that the resolution of DNA detection could be improved by increasing the number of  
17 graphene layers in a certain range and by modifying the surface of the graphene nanopores.

18 **Keywords:** DNA sequencing, multilayer graphene; solid state nanopores; DNA translocation;  
19 interaction energy

20

21

22

23

24

25

## 1 **1. Introduction**

2 In the presence of an external bias voltage, a DNA or RNA molecule dispersed in a salt  
3 solution can be driven through a nanopore, thereby interrupting the flow of the salt ions and  
4 triggering a detectable change of the ion current which can be used to probe the identity of the  
5 bases in the molecule. DNA sequencing with nanopores, such as solid-state<sup>3-8</sup> or biological  
6 nanopores<sup>9-11</sup>, is believed to be superior to other sequencing technologies and has  
7 experienced an exceptionally rapid development in recent years. For example, by using a  
8 mutant MspA nanopore and phi29 DNA polymerase, Manrao *et al.* were able to read DNA at  
9 single-nucleotide resolution<sup>12</sup>.

10 An advantage of using biological nanopores is that they can be chemically engineered  
11 through advanced molecular biology techniques. However, the lipid membrane to fix the  
12 biological nanopores is delicately sensitive to temperature, pH and salt concentration, which  
13 makes biological nanopore difficult to control their stability. In contrast, with established  
14 technologies, very stable and functionally useful solid-state nanopores can be fabricated using  
15 silicon nitride, silicon oxide or metal oxide<sup>18</sup>. Thanks to the robustness and the ability to tune  
16 the size and shape of the nanopores used, different types of nanopore have been used to  
17 sequence DNA<sup>5</sup>. However, solid-state nanopores are typically tens of nanometers in thickness  
18 which makes it difficult to sequence DNA with low-noise detection<sup>19</sup>.

19 Recently, solid-state nanopores fabricated from graphene sheets<sup>20</sup> have attracted intensive  
20 interest due to the unique properties of graphene<sup>21-24</sup>. Reading a DNA molecule at single-  
21 nucleotide resolution with a monolayer graphene nanopore has though been hampered by the  
22 fast translocation speed of the DNA<sup>25</sup>. Many theoretical and experimental studies have been  
23 carried out to solve this problem, such as those studying decreasing temperature, decreasing  
24 applied voltage or increasing solvent viscosity<sup>26, 27</sup>. However, such methods are unable to  
25 change the translocation dynamics of DNA through a nanopore. Recently, multilayer

1 graphene (MLG) films of less than 10 layers have been fabricated and tested in this respect,  
2 and their properties were found to be superior to those of monolayer graphene sheets<sup>30</sup>. Kim  
3 et al pointed out the few layer graphene possessed low noise ratio as compared to single layer  
4 graphene<sup>31</sup>. Although DNA sequencing with MLG nanopores has been investigated  
5 intensively<sup>32</sup>, the dynamics and details of the sequencing process remain unclear.

6 The aim of our work is to apply Molecular Dynamics (MD) simulations to study the  
7 atomistic details of a DNA molecule translocation through a nanopore. MD simulations have  
8 been successfully applied to the study of DNA translocation driven by electric fields<sup>33-35</sup><sup>36</sup>.  
9 In this work, we carried out MD simulations to investigate DNA translocation through MLG  
10 nanopores. Due to its simplicity, poly(A-T)<sub>45</sub> was used as a model DNA fragment. We  
11 demonstrate that the translocation time can be extended by increasing the graphene layers to a  
12 certain range. By studying the current change of the DNA fragment going through the MLG  
13 nanopores, we built a model to explore the relationship between the unoccupied volume of the  
14 nanopore and the signal current and investigated the relationship between the translocation  
15 speed of the nucleotides and the interaction of the nucleotides with the graphene sheets.

## 16 2. Computational method

### 17 System setup

18 Table 1 lists the systems studied in this work. For each system, an MLG sheet was placed in  
19 the  $x$ - $y$  plane with its center of mass in the origin (0, 0, 0) of a Cartesian coordinate system. A  
20 nanopore was constructed by deleting the atoms with their coordinates satisfying  $x^2 + y^2 < D^2$ ,  
21 where  $D$  is the radius of the graphene nanopore and was set to 1.5 nm, the bond length is  
22 1.42Å in multilayer graphene sheet, and the separation distance is 3.4Å between graphene  
23 sheets. Poly(A-T)<sub>45</sub> was constructed by using the Hyperchem software (Version 7.0,  
24 Hypercube, Inc). The nanopore and poly(A-T)<sub>45</sub> were placed in a box and solvated with  
25 45340 TIP3P water molecules<sup>37</sup>. The TIP3P water model is compatible with the CHARMM

1 force field which is used in this work to model the DNA fragment. The system then  
2 underwent a 10000-step energy minimization. Thereafter, KCl was added to make its  
3 concentration equal to 1.0 M as in experiment<sup>38</sup> by replacing the water molecules randomly.  
4 The system was then subject to a 200000-step energy minimization. The length of the  
5 simulation box is 90Å in the  $x$  and  $y$  directions in all the simulations, while in the  $z$  direction  
6 the box length changes with the number of layers of the graphene nanopore, and the box  
7 length is varied from 250 Å to 270 Å in  $z$  direction in different simulations with different  
8 graphene layers. Fig.1 shows the initial setup of the system with poly(A-T)<sub>45</sub> and a three-layer  
9 graphene nanopore.

10 All MD simulations were performed three times by the Gromacs program<sup>39</sup> with a time step  
11 of 2.0 fs, and all bonds that involve H-atoms were fixed. The density in all the simulations  
12 varied from 1.051 to 1.054 g/cm<sup>3</sup>. The DNA fragment and KCl were modeled by the  
13 Charmm27 force field<sup>40</sup>. All the carbon atoms in the graphene sheets were set to be neutral.  
14 The Lennard-Jones parameters of the graphene carbon atoms were  $\sigma_{CC} = 0.385$  nm and  $\epsilon_{CC} = -$   
15 0.439 kJ/mol. Periodic boundary condition was used in all directions. The cutoff for the non-  
16 bonded van der Waals interaction was set by a switching function starting at 1.0 nm and  
17 reaching zero at 1.2 nm. The Langevin method was employed to keep the simulation  
18 temperature at 298.0 K, and the pressure was set to 101.3 kPa in all directions. The particle  
19 mesh Ewald summation was used to recover the long range electrostatic interaction, with a  
20 cutoff of 1.3 nm for the separation of the direct and reciprocal space summation. A bias  
21 voltage of 100mV/nm was applied to drive the ions and the DNA fragment passing through  
22 the nanopores in all the simulations. This bias voltage has the same magnitude as that used in  
23 our previous work<sup>36</sup>.

#### 24 **Analysis method**

25 To describe the blockade current of a DNA molecule through a nanopore and to explain the

1 phenomenon, the time-dependent ionic current  $I(t)$  was calculated as<sup>26</sup>,

$$2 \quad I(t) = \frac{1}{\Delta t L_z} \sum_{i=1}^N q_i [z_i(t + \Delta t) - z_i(t)] \quad (1)$$

3 where  $L_z$  is the length of the system in the  $z$ -direction,  $z_i(t)$  is the  $z$  coordinate of atom  $i$  at time  
 4  $t$ ,  $\Delta t$  is set to 10.0 ps,  $N$  is the total number of atoms, including those of the DNA and ions,  
 5 and  $q_i$  is the charge of atom  $i$ , respectively. The interaction of a nucleotide with each graphene  
 6 layer was calculated by:

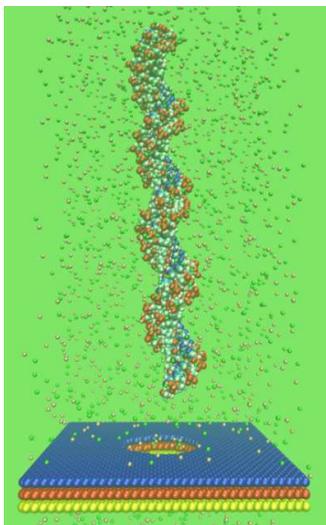
$$7 \quad E_{int} = E_{gra+} E_{nuc} - E_{gra+nuc} \quad (2)$$

8 where  $E_{int}$  is the interaction of the nucleotide with the graphene layer,  $E_{gra}$ ,  $E_{nuc}$ ,  $E_{gra+nuc}$  are  
 9 potential energies of graphene layer, nucleotide, and graphene with nucleotide, respectively.

10 Table 1. Systems studied\*

	Number of atoms	Number of layers	Temperature (K)	Voltage (mV/nm)	DNA fragment	Simulation time (ns)
Sim1	134,235	1	298	100		10
Sim2	134,543	3	298	100		10
Sim3	134,257	5	298	100		10
Sim4	134,387	7	298	100		10
Sim5	134,556	9	298	100		10
SimD1	175,336	1	298	100	poly(A-T) <sub>45</sub>	10
SimD2	192,232	3	298	100	poly(A-T) <sub>45</sub>	5
SimD3	193,213	5	298	100	poly(A-T) <sub>45</sub>	10
SimD4	195,433	7	298	100	poly(A-T) <sub>45</sub>	10
SimD5	201,775	9	298	100	poly(A-T) <sub>45</sub>	20
SimN1	12,982	1	298		poly(dA) <sub>5</sub>	20
SimN2	13,282	1	298		poly(dT) <sub>5</sub>	20
SimN3	12,793	1	298		poly(dC) <sub>5</sub>	20
SimN4	12,874	1	298		poly(dG) <sub>5</sub>	20
SimV1	175,336	1	298	30	poly(A-T) <sub>45</sub>	50
SimV2	192,232	3	298	30	poly(A-T) <sub>45</sub>	50
SimV3	193,213	5	298	30	poly(A-T) <sub>45</sub>	50
SimV4	195,433	7	298	30	poly(A-T) <sub>45</sub>	50
SimV5	201,775	9	298	30	poly(A-T) <sub>45</sub>	50

11 \* the concentration of KCl is 1.0 M in all the simulations.



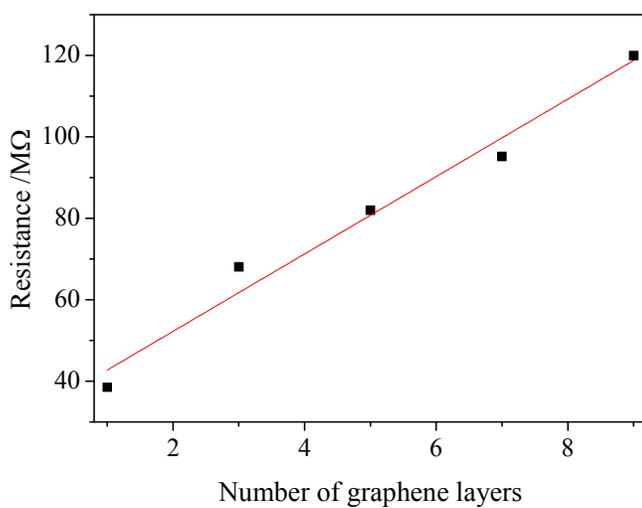
1

2 Fig.1. Initial setup of the system with poly(A-T)<sub>45</sub> and a three-layer graphene nanopore. The  
3 DNA fragment is placed above the top of the graphene nanopore, as shown by the vdW (van  
4 der Waals) model. K<sup>+</sup> (pink) and Cl<sup>-</sup> (green) ions are shown by the CPK (Corey-Pauling-  
5 Kortum) model. Water molecules are not shown for clarity.

6

### 7 3. Results and discussion

#### 8 3.1. Open nanopore resistance



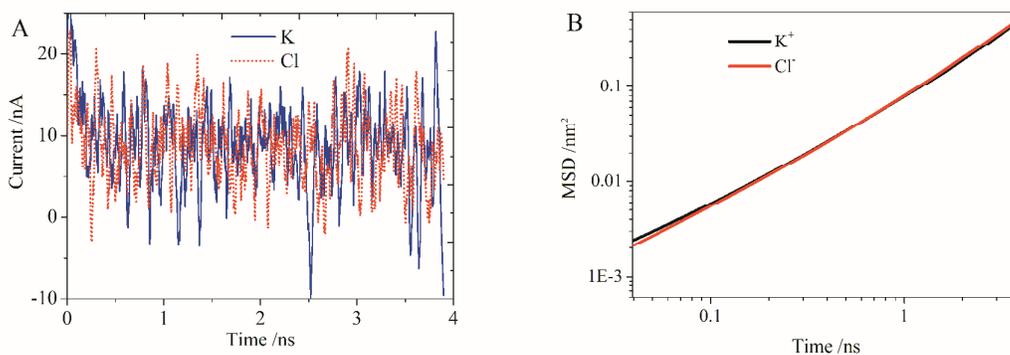
9

10 Fig.2. Change of the open nanopore resistance with the number of graphene layers.

1 In our previous work, the relationship between the diameter of a nanopore and its open  
2 nanopore resistance was investigated<sup>42</sup>. Here, the resistance (Res) of the nanopores was  
3 evaluated, with the diameter of the nanopores set to 3nm. A series of MD simulations were  
4 performed with the number of graphene layers varying from 1 to 9 under the same KCl  
5 concentration (1.0 M) as used in experiments<sup>38</sup>. Due to the applied external electric field, K<sup>+</sup>  
6 and Cl<sup>-</sup> ions were driven to move in opposite directions and the average ionic current  $\langle I \rangle$  was  
7 calculated from Eq.(1). The slope of the  $V/\langle I \rangle$  curve was determined as the resistance of a  
8 nanopore. Based on the experiment data<sup>25</sup>, the resistance on diameter 3nm monolayer  
9 graphene nanopore should be 273.02 M $\Omega$ , and it is 38.2 M $\Omega$  in our simulation. This indicates  
10 that the pore resistance in our simulations is lower than the experimental results. One reason  
11 is that the voltage we used is larger than that in experiment, because a low voltage similar to  
12 those used in experiment could require a simulation time too long to be practical with our  
13 computational resources. Another one is the charge distribution of graphene nanopore is not  
14 considered in the simulations, and the force field to describe the interaction of ions and  
15 graphene should be improved as mentioned in others work<sup>26</sup>. We note here that the thickness  
16 of a graphene sheet is directly proportional to the number of graphene layers, which is defined  
17 as  $L$ . As plotted in Fig.2, the resistance of the graphene nanopores depends closely on the  
18 number of graphene layers, which can be expressed as  $Res \sim L$ . This reflects that the open  
19 nanopore resistance of a graphene sheet is also directly proportional to its thickness and the  
20 detected current accordingly decreases with the increase of graphene layers. The relationship  
21 between the current and the layers of graphene nanopores was also discussed by Lv et al<sup>32</sup>.  
22 They demonstrated that the ionic current is sensitive to the number of graphene layers, which  
23 is in accordance with our result in this work. Because it is difficult to control the thickness of  
24 a graphene sheet experimentally, the dependence of the resistance of the graphene sheet on its

1 thickness remains unclear. Here, the relationship between the open nanopore resistance and  
2 the thickness was explored qualitatively.

3 The movement of  $K^+$  and  $Cl^-$  ions under the applied field contributes to the measured  
4 currents. For Sim2, the contributions of  $K^+$  and  $Cl^-$  ions to the current were found to be almost  
5 the same (see Fig.3A). To interpret the phenomenon more deeply, the mean square  
6 displacements (MSDs) of  $K^+$  and  $Cl^-$  ions in the Z direction of the Cartesian coordinate system,  
7 i.e. in the direction of the applied field, were calculated. As shown in Fig.3B, the MSDs of the  
8 two ions in the same simulation time are almost the same.

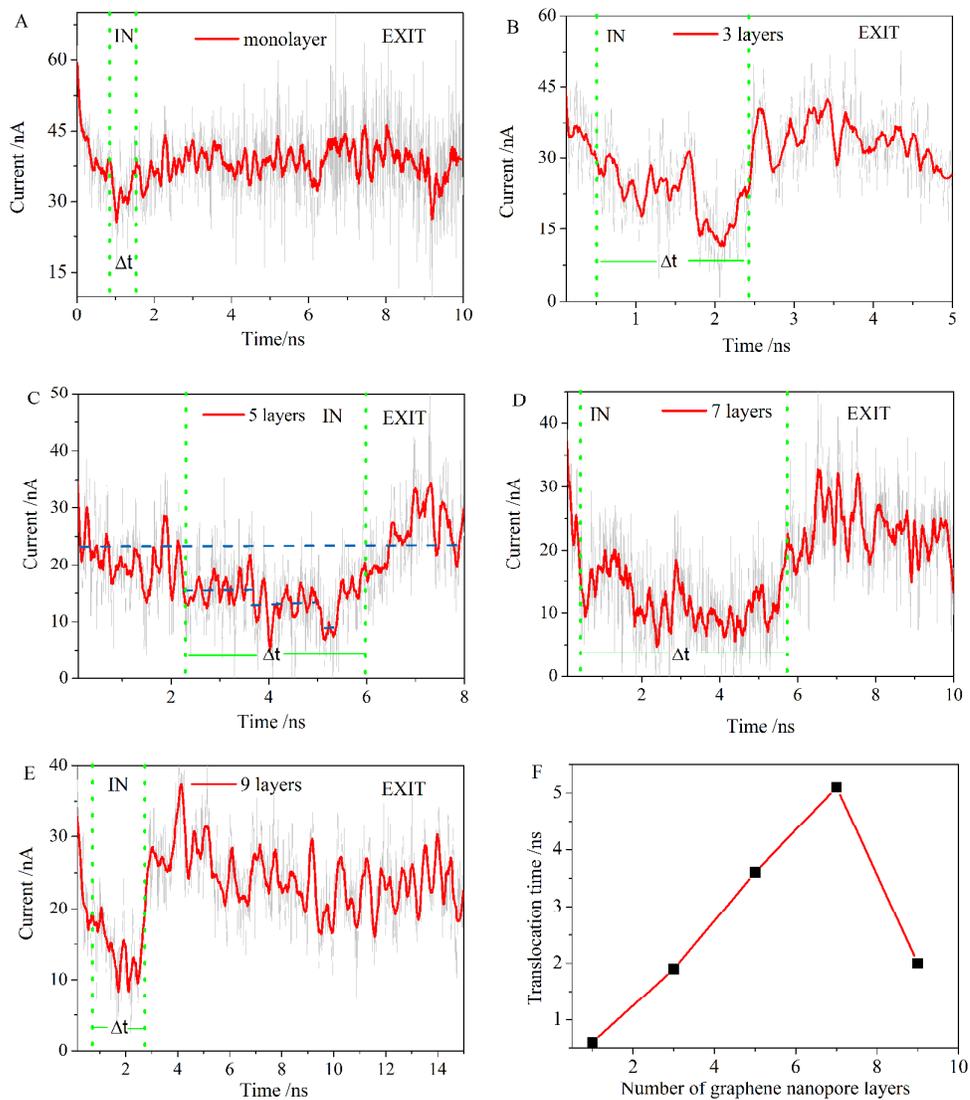


9

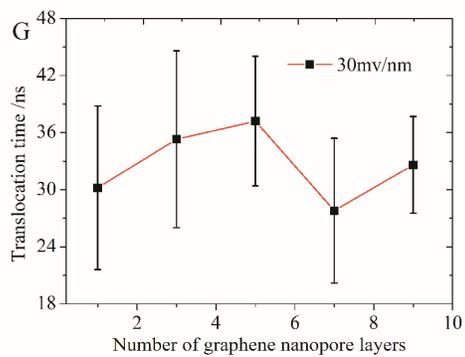
10 **Fig.3. A. Contributions of  $K^+$  and  $Cl^-$  to the current in Sim2. B. MSDs of  $K^+$  and  $Cl^-$  in**  
11 **Sim2.**

12

13 3.2. Effect of the number of graphene layers



1



2

3

4

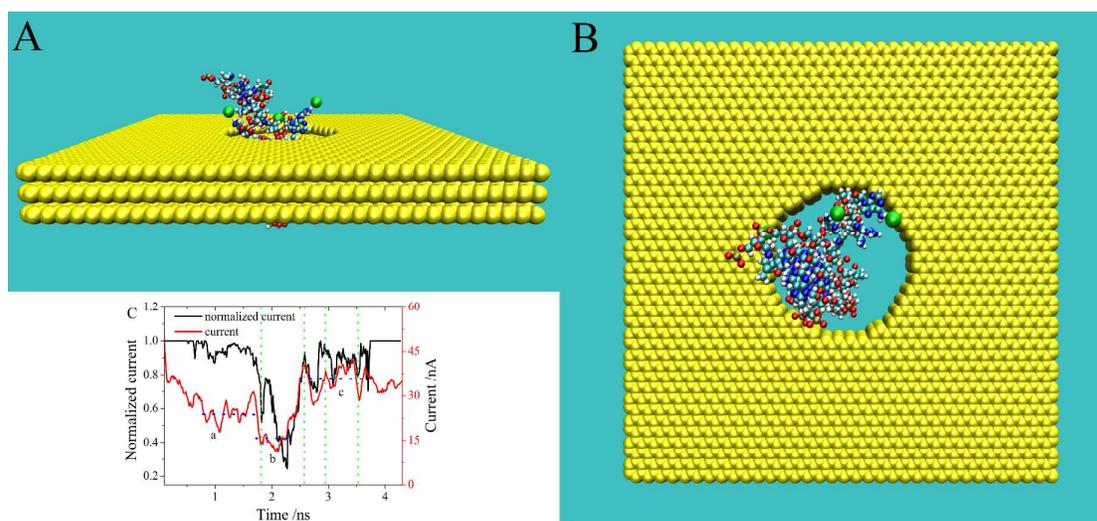
1 Fig.4 Change of the currents with the time for the DNA fragment translocation through  
2 graphene nanopores of 1-layer (A), 3-layer (B), 5-layer (C), 7-layer (D) and 9-layer (E) under  
3 100mv/nm electric field, respectively.  $\Delta t$  represents the translocation time for poly(A-T)<sub>45</sub> to  
4 pass through the nanopores. The blue dashed line in (C) corresponds to the current change  
5 with poly(A-T)<sub>45</sub> translocation through the nanopore. EXIT means that all the residues of  
6 poly(A-T)<sub>45</sub> are outside of the nanopore and IN indicates that some segments of poly(A-T)<sub>45</sub>  
7 are passing through the nanopore. Change of the translocation time with the number of  
8 graphene layers under electric fields of 100mv/nm (F) and 30mv/nm (G), respectively.

9 Fig.4 shows the change of the current with the DNA fragment passing through the graphene  
10 nanopores of 1, 3, 5, 7, and 9 layers. As seen in the figure, the translocation time is about  
11  $0.6\pm 0.2$ ,  $1.9\pm 0.8$ ,  $3.6\pm 0.9$ , and  $5.1\pm 1.2$ ns for the nanopores of 1, 3, 5, and 7 layers,  
12 respectively, and the translocation time for 45bp in our simulation is much faster than that in  
13 the experiment<sup>25</sup>. One reason is the applied voltage is 100mv/nm, and the total voltage is 2.5v,  
14 which is much larger than that in experiment. The other reason is the force field used here  
15 could not describe the charge distribution of graphene nanopore, which should be improved.  
16 Compared with the monolayer graphene sheet, the translocation time was clearly prolonged.  
17 This implies that increasing the number of graphene layers could increase the translocation  
18 time. One of the major challenges in applying a graphene nanopore to detect a DNA molecule  
19 is to reduce the speed at which the DNA molecule passes through the nanopore<sup>43</sup>. Our results  
20 thus suggest that the translocation time of the DNA fragment can be greatly extended by using  
21 MLG nanopores. However, if the number of graphene layers grows large, for example more  
22 than 10<sup>30,44</sup>, the electronic structure of a graphene nanopore approaches to the 3D limit of  
23 graphite<sup>45</sup> and can influence the electric properties of the graphene nanopore and produce  
24 high noise in the DNA sequencing. Therefore, the number of graphene layers should be less  
25 than 10. As we can see from Fig.4, the translocation time is almost directly proportional to the

1 number of graphene layers if the number varies from 1, 3, 5 and 7, while the translocation  
2 time through the 9-layer nanopore becomes very short. As reported by others, the longer a  
3 carbon nanotube, the deeper the potential well in the tube, and accordingly a protein can be  
4 spontaneously encapsulated into a longer carbon nanotube. This is also the case for a  
5 nanopore: with the increase of the thickness of a nanopore, the potential well in the nanopore  
6 becomes deep. Thus, increasing the number of graphene layers also means deepening the  
7 potential well in the nanopore, leading to that the DNA quickly threads to the nanopore.  
8 However, the DNA translocation time is not related to the nanopore thickness in the  
9 experiment<sup>25</sup>. The main reason is the difference of the electric fields applied in the experiments and  
10 simulations. With a lower electric field applied in the experiments, the translocation time is greatly  
11 extended. As seen in Fig.4G, the translocation time of poly(A-T)<sub>45</sub> through the graphene nanopores of  
12 different layers is independent of the number of layers under a low electric field (30mv/nm). Under  
13 such a low electric field, the driving force for the DNA fragment to pass through a nanopore is very  
14 small, while the interaction of the graphene nanopore with the DNA fragment governs the  
15 translocation time since the DNA fragment tends to adsorb onto the graphene nanopore. The sticking  
16 time and trapping time in the graphene nanopore are so long that the effect of the graphene nanopore  
17 thickness on the translocation time is hardly observable. Thus, the translocation time is not related to  
18 the number of graphene nanopore layers in the experiment. However, the sticking and trapping times  
19 are very short under a high electric field since the driving force is very large. This means that the  
20 effect of the graphene nanopore thickness on the translocation time is very important under a high  
21 electric field (100mv/nm) (see Fig.4F). Thus, the translocation time is related to the graphene  
22 nanopore thickness under a high electric field but not a low electric field. To increase the  
23 translocation time and decrease the translocation speed of a DNA molecule, the number of  
24 graphene layers should be increased but not exceed a certain number, which is 7 according to  
25 our calculations. The average current decreases with the increase of the number of graphene  
26 layers since the nanopore resistance is directly proportional to the number of graphene layers.

1 As shown by the blue dashed line in Fig.4C, the current undergoes a stepwise change upon  
2 the DNA fragment passing through an MLG nanopore. In the first step, the current is ca.  
3 22.5nA, which is the open ionic current corresponding to that all the residues of poly(A-T)<sub>45</sub>  
4 are outside of the nanopore. In the second step, the current is ca. 15.1nA, which is the blocked  
5 current corresponding to that one layer of the nanopore is occupied by poly(A-T)<sub>45</sub>. In the  
6 third step, the blocked current becomes ca.12.3nA when 2-3 layers of the nanopore are  
7 occupied by poly(A-T)<sub>45</sub>. In the fourth step, the blocked current is ca. 7.6nA when all the  
8 layers of the nanopore are occupied by poly(A-T)<sub>45</sub>. In the last step, the current recovers to ca.  
9 22.5nA since all the residues of poly(A-T)<sub>45</sub> are outside of the nanopore. The impact of  
10 thermal noise on DNA sequencing has been studied by Lv's group<sup>48</sup> with single-layer rigid or  
11 flexible graphene nanopores. It seems that freezing the carbon atoms of a graphene nanopore  
12 has essentially no effect on the impact of thermal noise. The impact of thermal noise on  
13 multilayer graphene nanopores will be the potential goal of further studies.

### 14 3.3. The theoretical model



15

16 Fig.5. Translocation of poly(A-T)<sub>45</sub> in the three-layer graphene nanopore. The atoms of  
17 poly(A-T)<sub>45</sub> captured by the nanopore (with the distance to the pore center < 1.2nm) are  
18 shown by the CPK model, the graphene nanopore is shown in yellow by the vdW model, and

1 the ions captured by the nanopore (with the distance to the pore center < 1.2nm) are shown in  
 2 green by the vdW model: (A) lateral view; (B) top view. (C) The black and red solid lines  
 3 are the normalized current from the theoretical model and the current from the simulation for  
 4 the SimD2 system, respectively.

5 Here, we take the 3-layer graphene nanopore as an example to explore the dependence of the  
 6 current on the unoccupied volume of the nanopore. First, only those DNA atoms with the  
 7 distance to the nanopore less than  $|z|$  are considered to influence the movement of ions to pass  
 8 through the nanopore as described in the previous work<sup>36</sup>. The  $z$  parameter ( $|z| \leq 1.2$  nm)  
 9 depends on the cutoff for the non-bonded van der Waals interaction used in the simulation. As  
 10 shown in Fig. 5A and 5B, the captured atoms of poly(A-T)<sub>45</sub> occupy the nanopore and prevent  
 11 the ions from passing through it. To calculate the normalized current, a nanopore was  
 12 considered as a cylinder and was divided into  $N$  parts of equal length in the  $Z$  direction. As  
 13 shown in Fig.S1, we found that the model normalized current is independent of  $N$  when  $N$  is  
 14 larger than 10 for the 3-layer graphene nanopore (see Supporting Information). Here, the 3-  
 15 layer graphene nanopore was divided into 12 parts, with each part being 0.6 Å in thickness.  
 16 The across area in each part was calculated and the method for calculating  $A_i$  for part  $i$  is the  
 17 same as that for the monolayer graphene nanopore described in our previous work<sup>36</sup>. The  
 18 normalized current is in direct proportion to  $1/R_{total}$  under the same voltage, where  $R_{total}$  is the  
 19 total resistance, with  $R_{total} = \sum_1^N R_i$ . Since  $R_i$  is directly proportional to  $1/A_i^2$ , we have

$$20 \quad I = V/R \quad (3)$$

$$21 \quad R_{total} = \sum_1^N R_i \quad (4)$$

$$22 \quad R_i \propto \frac{1}{A_i^2} \quad (5)$$

23 Under the same voltage

$$1 \quad I \propto \frac{1}{R_{\text{total}}} \quad (6)$$

$$2 \quad I \propto \frac{1}{\sum_1^N \frac{1}{A_i^2}} \quad (7)$$

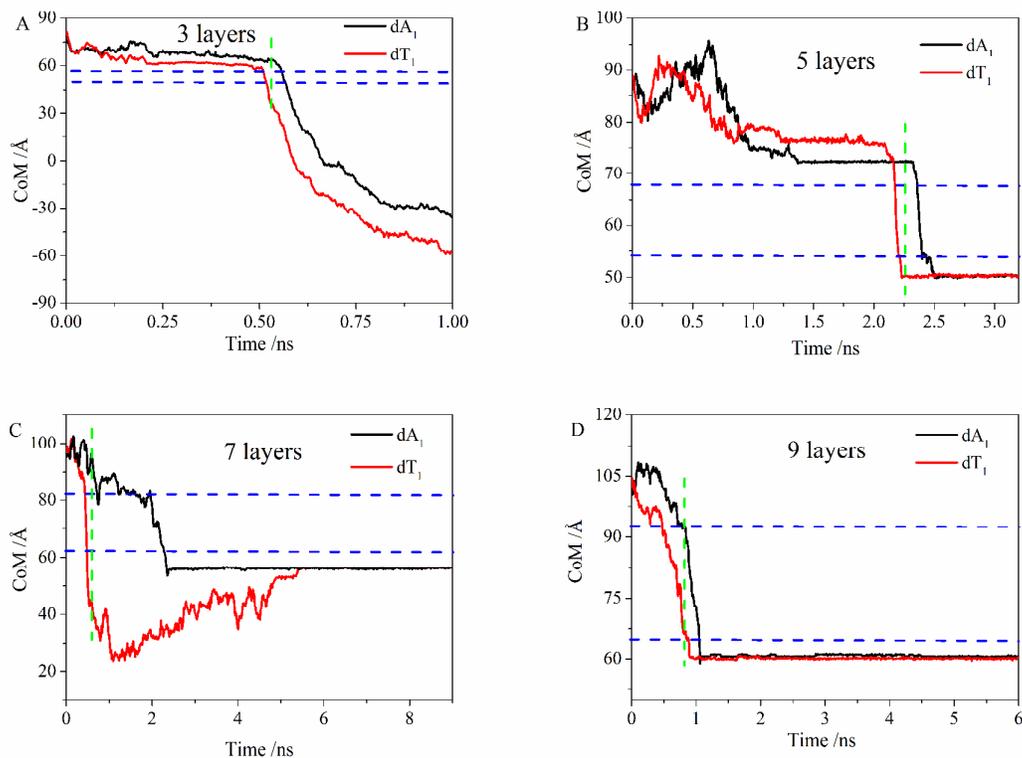
3 The current calculated from Eq.(7) is not the current observed in the simulation. To compare  
4 these two currents, the current from the model was normalized by dividing the open nanopore  
5 current which was also calculated from the model. As shown in Fig.5c, the changes of the  
6 normalized current accord with that of the current in the 3-layer system with DNA fragment.  
7 This reflects that the model is able to describe correctly the current change in the DNA  
8 translocation. Since the ratio of the blockade current to open nanopore current is considered as  
9 the signal for distinguishing nucleotides, the change of normalized current could be thought as  
10 the real signal current. Thus, our model implies that a nucleotide could be distinguished with  
11 different unoccupied nanopore volumes and the resolution of DNA sequencing could be  
12 improved by modifying the nanopore volumes.

13 The evolution of the average current corresponds to three intervals, i.e. intervals a, b, and c.  
14 The current was blocked when the DNA fragment entered partially into the nanopore (interval  
15 a). With the majority of the DNA atoms entering into the nanopore, more current was blocked  
16 (interval b). Once the whole DNA molecule passed through the nanopore, the current was  
17 recovered (interval c). As mentioned above, this stepwise process was also observed with the  
18 5-layer graphene nanopore. The stepwise change of the current comes as the result of using  
19 MLG nanopores. Compared with a monolayer system, an MLG nanopore can display more  
20 details about the current change. Therefore, increasing the thickness of a nanopore may  
21 improve the resolution in DNA sequencing.

22 3.4. Dynamics of dA and dT

23

1



2

3

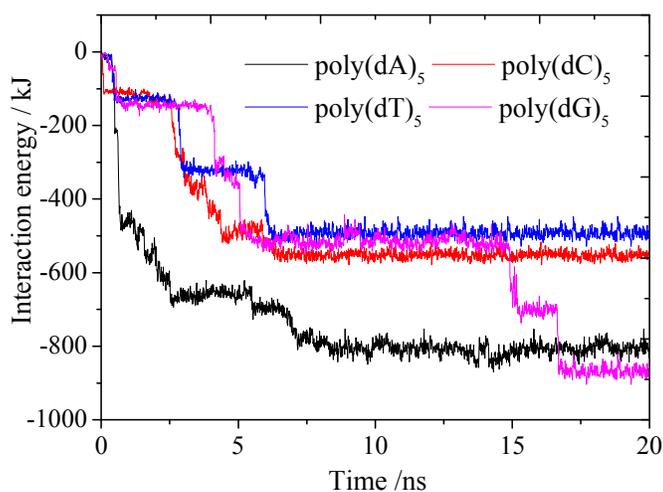
4 Fig.6. Time evolution of the position of the center of mass of the first residue of poly(dA)<sub>45</sub>  
 5 (black line) and the first residue of poly(dT)<sub>45</sub> (red line) in the *z* direction (A) 3 layers, (B) 5  
 6 layers, (C) 7 layers and (D) 9 layers. The two blue lines correspond to the upper and lower  
 7 limits of the graphene sheets in the *z* direction, respectively. The green line indicates the time  
 8 when dT has left the nanopore while dA has not entered into it.

9 In view of the importance of the translocation time in DNA sequencing, the dynamics of  
 10 poly(A-T)<sub>45</sub> was investigated. Here, the movement of the CoM (center of mass) of the first  
 11 residue of poly(A-T)<sub>45</sub> in the *z* direction was calculated. As we can see in the animation  
 12 trajectory, the first four or five residues of poly(A-T)<sub>45</sub> were unzipped before entering into the  
 13 nanopore. Unzipping of the double stranded DNA fragment upon going through a nanopore  
 14 was also observed by other researchers, and the unzipping strand poly(dA)<sub>45</sub> or poly(dT)<sub>45</sub>  
 15 means dA and dT strand of the poly(A-T)<sub>45</sub>, in respectively. In Fig.6A, the first residue of

1 poly(dA)<sub>45</sub> is denoted as dA<sub>1</sub> and the first residue of poly(dT)<sub>45</sub> is denoted as dT<sub>1</sub>, respectively.  
2 We can see that the time interval between dA<sub>1</sub> entering into and leaving the nanopore is  
3 almost the same as that for dT<sub>1</sub>. The time for poly(dA)<sub>45</sub> and poly(dT)<sub>45</sub> translocation through  
4 a nanopore of the same number of layers was essentially the same. However, an interesting  
5 point is that our simulations showed that poly(dT)<sub>45</sub> entered into the nanopore earlier than  
6 poly(dA)<sub>45</sub>, as reflected by the vertical green dashed line in Fig.6. The time for the first  
7 residue of poly(dA)<sub>45</sub> and poly(dT)<sub>45</sub> passing through the identical nanopore was thus virtually  
8 the same although their sequences are different.

9 Since the DNA translocation dynamics depends on the DNA-pore interaction, we carried  
10 out further a series of simulations to study the adsorption of different nucleotides on a  
11 monolayer graphene sheet to gain insight into the DNA-pore interaction. In the simulation, the  
12 final state could be seen as the absorption state of graphene with nucleotide, and they are apart  
13 in the primary state. The enthalpy of adsorption could be seen as the interaction energy in  
14 final state of the simulation to minus that of the primary state. Since the interaction energy in  
15 the primary state are all zero in all systems, the value of interaction energy in the final state  
16 could be seen as the enthalpy of adsorption. As shown in Fig.7, the interactions of the  
17 graphene with poly(dT)<sub>5</sub> and poly(dA)<sub>5</sub> are ca. 500 and 800kJ/mol, respectively, which means  
18 that the interaction of poly(dA)<sub>5</sub> with the graphene sheet is much stronger than of poly(dT)<sub>5</sub>.  
19 The strong interaction of poly(dA)<sub>5</sub> with the graphene sheet leads to that the time of poly(dA)<sub>5</sub>  
20 sticking to the graphene nanopore becomes much longer than that of poly(dT)<sub>5</sub> and thus  
21 explains why poly(dA)<sub>45</sub> enters into the nanopore later than poly(dT)<sub>45</sub>. Therefore, the  
22 translocation time can be different for nucleotides with distinctive nucleotide-pore  
23 interactions. The interactions of dC and dT with the graphene sheet are almost the same,  
24 leading to poor resolution of dC and dT in the DNA sequencing, which is consistent with the  
25 results of Qiu et.al<sup>52</sup>. The interaction between dG and the graphene sheet is the largest, in

1 accordance with the result of Zhao's et al<sup>53</sup>. The interaction of nucleotides with a graphene  
2 nanopore can be of crucial importance in distinguishing the nucleotides, and the resolution of  
3 DNA detection could therefore be improved by modifying the graphene nanopore surface.



4

5 Fig.7. Interaction energy of different nucleotides with the monolayer graphene.

6

#### 7 **4. Conclusion**

8 In this work we carried out molecular dynamics simulations to investigate the DNA  
9 translocation process through multi-layer graphene (MLG) nanopores. In particular, we  
10 addressed the importance of the dynamics and current change in DNA translocation to thread  
11 MLG nanopores, which so far have not received a detailed analysis. We found that the open  
12 nanopore resistance is directly proportional to the nanopore length. The contribution of K<sup>+</sup>  
13 ions to the current was almost the same as that of Cl<sup>-</sup>. The translocation time increases with  
14 the increase of the number of graphene layers under a high electric field and reaches a  
15 maximum at a few layers (7), but decreases thereafter. This behavior was associated with the  
16 potential well in the nanopore. Based on the analysis of the DNA translocation through MLG  
17 nanopores, a model was constructed to explore the relationship between the current and the

1 unoccupied volume of the nanopores. It was demonstrated that the blockade current is closely  
2 related to the unoccupied volume of the nanopores. We found that the ionic current underwent  
3 a stepwise change with the DNA passing through an MLG nanopore. Our study showed that  
4 the stepwise current came as the result of the increase of the nanopore thickness, meaning that  
5 the resolution could be improved by increasing the thickness of the nanopore to a certain  
6 range. We also found that due to the difference in the interaction between the nucleotides and  
7 the graphene sheet, the translocation process of poly(dT)<sub>45</sub> was earlier than that of poly(dA)<sub>45</sub>  
8 under the same conditions. Our work indicates that the interaction of the nucleotides with the  
9 graphene nanopore is of crucial importance in improving the discrimination of the nucleotides  
10 and that the resolution of DNA sequencing could be improved by modifying the nanopore  
11 surface. It could be helpful to applied graphene as a promising biosensor material for DNA  
12 sequencing.

### 13 **Acknowledgements**

14 This work was financially supported by the National Natural Science Foundation of China  
15 (Grant Nos. 21273200 and 21074115), MOE (J20091551), Zhejiang Provincial Natural  
16 Science Foundation of China (No. LQ12F05001) and Zhejiang University (2011XZZX002,  
17 2011QNA3014). The computations were performed on resources provided by the Swedish  
18 National Infrastructure for Computing (SNIC) at the parallel computer centre (PDC), through  
19 the project "Multiphysics Modeling of Molecular Materials", SNIC 020/11-23.

20

21

22

23

### 24 **References**

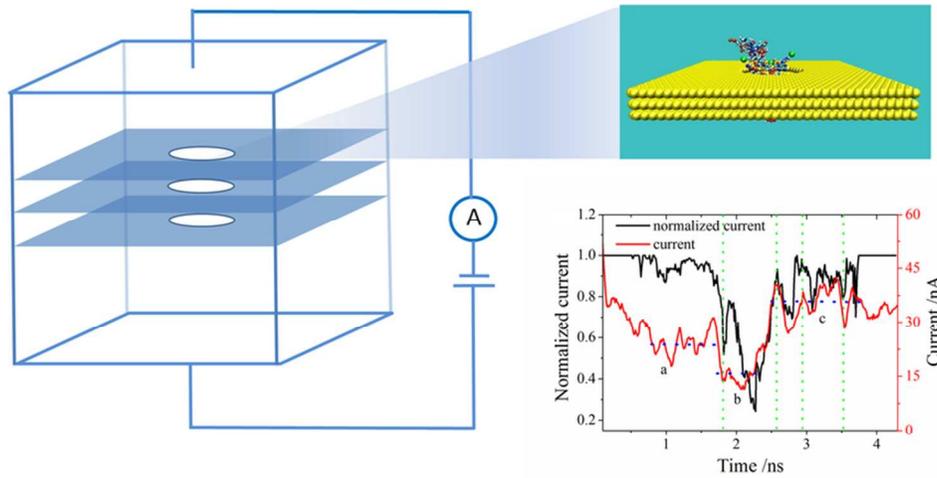
- 25 1. S. K. Min, W. Y. Kim, Y. Cho and K. S. Kim, *Nat Nano*, 2011, **6**, 162-165.  
26 2. H. W. C. Postma, *Nano Letters*, 2010, **10**, 420-425.  
27 3. K. Healy, B. Schiedt and A. P. Morrison, *Nanomedicine (Lond)*, 2007, **2**, 875-897.

- 1 4. C. Dekker, *Nature Nanotechnology*, 2007, **2**, 209-215.  
2 5. H. Yan and B. Q. Xu, *Small*, 2006, **2**, 310-312.  
3 6. D. M. Vlassarev and J. A. Golovchenko, *Biophys J*, 2012, **103**, 352-356.  
4 7. Q. Liu, H. Wu, L. Wu, X. Xie, J. Kong, X. Ye and L. Liu, *Plos One*, 2012, **7**, e46014.  
5 8. B. McNally, M. Wanunu and A. Meller, *Biophys J*, 2007, 652a-652a.  
6 9. T. Z. Butler, J. H. Gundlach and M. A. Troll, *Biophys J*, 2006, **90**, 190-199.  
7 10. R. F. Purnell and J. J. Schmidt, *Acs Nano*, 2009, **3**, 2533-2538.  
8 11. L. Franceschini, E. Mikhailova, H. Bayley and G. Maglia, *Chem Commun*, 2012, **48**, 1520-  
9 1522.  
10 12. E. A. Manrao, I. M. Derrington, A. H. Laszlo, K. W. Langford, M. K. Hopper, N. Gillgren, M.  
11 Pavlenok, M. Niederweis and J. H. Gundlach, *Nat Biotechnol*, 2012, **30**, 349-353.  
12 13. M. Wanunu and A. Meller, *Nano Lett*, 2007, **7**, 1580-1585.  
13 14. A. J. Storm, J. H. Chen, X. S. Ling, H. W. Zandbergen and C. Dekker, *Nat Mater*, 2003, **2**,  
14 537-540.  
15 15. M. J. Kim, M. Wanunu, D. C. Bell and A. Meller, *Advanced Materials*, 2006, **18**, 3149-3153.  
16 16. A. J. Storm, C. Storm, J. Chen, H. Zandbergen, J. F. Joanny and C. Dekker, *Nano Lett*, 2005,  
17 **5**, 1193-1197.  
18 17. A. Aksimentiev, J. B. Heng, G. Timp and K. Schulten, *Biophys J*, 2004, **87**, 2086-2097.  
19 18. B. M. Venkatesan, J. Polans, J. Comer, S. Sridhar, D. Wendell, A. Aksimentiev and R. Bashir,  
20 *Biomed Microdevices*, 2011, **13**, 671-682.  
21 19. H. Bayley, *Nature*, 2010, **467**, 164-165.  
22 20. S. Garaj, W. Hubbard, A. Reina, J. Kong, D. Branton and J. A. Golovchenko, *Nature*, 2010,  
23 **467**, 190-U173.  
24 21. C. A. Merchant, K. Healy, M. Wanunu, V. Ray, N. Peterman, J. Bartel, M. D. Fischbein, K.  
25 Venta, Z. T. Luo, A. T. C. Johnson and M. Drndic, *Nano Letters*, 2010, **10**, 2915-2921.  
26 22. S. Liu, Q. Zhao, J. Xu, K. Yan, H. Peng, F. Yang, L. You and D. Yu, *Nanotechnology*, 2012,  
27 **23**, 085301.  
28 23. C. A. Merchant and M. Drndic, *Methods Mol Biol*, 2012, **870**, 211-226.  
29 24. B. M. Venkatesan, D. Estrada, S. Banerjee, X. Jin, V. E. Dorgan, M. H. Bae, N. R. Aluru, E.  
30 Pop and R. Bashir, *Acs Nano*, 2012, **6**, 441-450.  
31 25. G. F. Schneider, S. W. Kowalczyk, V. E. Calado, G. Pandraud, H. W. Zandbergen, L. M. K.  
32 Vandersypen and C. Dekker, *Nano Lett*, 2010, **10**, 3163-3167.  
33 26. C. Sathe, X. Q. Zou, J. P. Leburton and K. Schulten, *Acs Nano*, 2011, **5**, 8842-8851.  
34 27. D. B. Wells, M. Belkin, J. Comer and A. Aksimentiev, *Nano Letters*, 2012, **12**, 4117-4123.  
35 28. D. Fologea, J. Uplinger, B. Thomas, D. S. McNabb and J. L. Li, *Nano Letters*, 2005, **5**, 1734-  
36 1737.  
37 29. K. J. Freedman, C. W. Ahn and M. J. Kim, *Acs Nano*, 2013, **7**, 5008-5016.  
38 30. G. Jo, M. Choe, C. Y. Cho, J. H. Kim, W. Park, S. Lee, W. K. Hong, T. W. Kim, S. J. Park, B.  
39 H. Hong, Y. H. Kahng and T. Lee, *Nanotechnology*, 2010, **21**, 175201.  
40 31. A. Kumar, K.-B. Park, H.-M. Kim and K.-B. Kim, *Nanotechnology*, 2013, **24**, 495503.  
41 32. W. Lv, M. Chen and R. a. Wu, *Soft Matter*, 2013, **9**, 960-966.  
42 33. B. Q. Luan, H. B. Peng, S. Polonsky, S. Rossnagel, G. Stolovitzky and G. Martyna, *Physical*  
43 *Review Letters*, 2010, **104**.  
44 34. B. Q. Luan, G. Martyna and G. Stolovitzky, *Biophys J*, 2011, **101**, 2214-2222.  
45 35. D. Y. Lu, A. Aksimentiev, A. Y. Shih, E. Cruz-Chu, P. L. Freddolino, A. Arkhipov and K.  
46 Schulten, *Phys Biol*, 2006, **3**, S40-S53.  
47 36. L. Liang, P. Cui, Q. Wang, T. Wu, H. Agren and Y. Tu, *RSC Advances*, 2013, **3**, 2445-2453.  
48 37. W. L. Jorgensen, J. Chandrasekhar, J. D. Madura, R. W. Impey and M. L. Klein, *J. Chem.*  
49 *Phys.*, 1983, **79**, 926-935.  
50 38. D. Fologea, M. Gershow, B. Ledden, D. S. McNabb, J. A. Golovchenko and J. L. Li, *Nano*  
51 *Lett*, 2005, **5**, 1905-1909.  
52 39. B. Hess, *Abstr Pap Am Chem S*, 2009, **237**.  
53 40. A. D. MacKerell, D. Bashford, M. Bellott, R. L. Dunbrack, J. D. Evanseck, M. J. Field, S.  
54 Fischer, J. Gao, H. Guo, S. Ha, D. Joseph-McCarthy, L. Kuchnir, K. Kuczera, F. T. K. Lau, C.  
55 Mattos, S. Michnick, T. Ngo, D. T. Nguyen, B. Prodhom, W. E. Reiher, B. Roux, M.

- 1 Schlenkrich, J. C. Smith, R. Stote, J. Straub, M. Watanabe, J. Wiorcikiewicz-Kuczera, D. Yin  
2 and M. Karplus, *J Phys Chem B*, 1998, **102**, 3586-3616.
- 3 41. L. J. Liang, Q. Wang, T. Wu, J. W. Shen and Y. Kang, *Chinese Journal of Chemical Physics*,  
4 2009, **22**, 627-634.
- 5 42. P. C. Lijun Liang, Qi Wang, Tao Wu, Hans Ågren , Yaoquan Tu, *RSC Advance* 2013, DOI:  
6 10.1039/C2RA22109H.
- 7 43. B. M. Venkatesan and R. Bashir, *Nat Nanotechnol*, 2011, **6**, 615-624.
- 8 44. A. K. Geim and K. S. Novoselov, *Nat Mater*, 2007, **6**, 183-191.
- 9 45. B. Partoens and F. M. Peeters, *Physical Review B*, 2006, **74**, 075404.
- 10 46. Y. Kang, Y. C. Liu, Q. Wang, J. W. Shen, T. Wu and W. J. Guan, *Biomaterials*, 2009, **30**,  
11 2807-2815.
- 12 47. Q. Chen, Q. Wang, Y.-C. Liu, T. Wu, Y. Kang, J. D. Moore and K. E. Gubbins, *The Journal*  
13 *of chemical physics*, 2009, **131**, -.
- 14 48. A. P. Ivanov, Novel nanopore/nanoelectrode architectures for biomolecular analysis  
15 [electronic resource].
- 16 49. A. Ferrantini and E. Carlon, *Journal of Statistical Mechanics: Theory and Experiment*, 2011,  
17 **2011**, P02020.
- 18 50. B. McNally, M. Wanunu and A. Meller, *Nano Lett*, 2008, **8**, 3418-3422.
- 19 51. A. F. Sauer-Budge, J. A. Nyamwanda, D. K. Lubensky and D. Branton, *Phys Rev Lett*, 2003,  
20 **90**, 238101.
- 21 52. H. Qiu and W. Guo, *Applied Physics Letters*, 2012, **100**, 083106.
- 22 53. C.-L. Cheng and G.-J. Zhao, *Nanoscale*, 2012, **4**, 2301-2305.

23

24



DNA translocation through multilayer graphene nanopore with a change of current  
38x24mm (600 x 600 DPI)