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## ARTICLE

# Co<sub>9</sub>S<sub>8</sub> nanoflakes on Graphene (Co<sub>9</sub>S<sub>8</sub>/G) nanocomposites for high performance supercapacitors

Cite this: DOI: 10.1039/x0xx00000x

Received 00th January 2012,  
Accepted 00th January 2012

DOI: 10.1039/x0xx00000x

www.rsc.org/

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Co<sub>9</sub>S<sub>8</sub>/graphene nanocomposites (Co<sub>9</sub>S<sub>8</sub>/G) at various concentrations of graphene and Co<sub>9</sub>S<sub>8</sub> were prepared by a simple chemical route from cobalt nitrate and graphene as precursor in the presence of PVP as surfactant and thioacetamide (TAA) as sulfur source. To further know about the structural, morphological and physical properties, the composite material was analyzed by X-ray diffraction (XRD), Field Emission Scanning Electron Microscope (FE-SEM), X-ray photoelectron spectroscopy (XPS) and thermogravimetric Analysis (TGA). SEM measurements showed the presence of well dispersed ~300 nm sized Co<sub>9</sub>S<sub>8</sub> nanoflakes. To assess the properties of the nanocomposites for its applicability towards supercapacitors, electrochemical analysis was carried out in 6 M KOH electrolyte. A maximum specific capacitance of 808 F/g was observed for Co<sub>9</sub>S<sub>8</sub>/G-d at 5 mV/s scan rate. Galvanostatic charge/discharge curves showed the excellent cyclic stability of Co<sub>9</sub>S<sub>8</sub>/G-d composite with high charge/discharge duration than pure Co<sub>9</sub>S<sub>8</sub>. The excellent electrochemical performance of the composite could be due to the better electrical conductivity behavior of graphene on Co<sub>9</sub>S<sub>8</sub> nanoflakes.

## Introduction

Supercapacitors also known as ultracapacitors are energy storage devices that combine the high power density and high energy storage capability of batteries and conventional capacitors and has long cyclic life and fast dynamic charge propagation.<sup>1-2</sup> In general, supercapacitors fall under two categories viz. Electric Double Layer Capacitors (EDLC) and pseudocapacitors. In EDLC energy is stored through charge accumulation at electrode and electrolyte interface whereas, the energy is stored through the surface faradic redox reactions in pseudocapacitors.<sup>3</sup> Currently, three types of electrode materials have been used as supercapacitor electrodes; first carbon based materials like activated carbon, carbon nanotube, carbon aerogel; second conducting polymers like polyaniline and polypyrrole and finally metal oxides and hydroxides such as RuO<sub>2</sub>, MnO<sub>2</sub>, Co(OH)<sub>2</sub>, NiO, ZnO, etc.. In general, carbon based materials have been used in EDLC due to its high specific surface area and easy processability. But it provides relatively low specific capacitance (100-300 F/g) [1]. Though metal oxides give higher specific capacitance (>2000 F/g), they

are limited by its high cost and poor conductivity. Many of the metal oxides like MnO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>, NiO, though they have high theoretical capacitance and low cost, the electrical conductivity of these metal oxides limits on its power density and cycling stability. Among the transition metal oxide, RuO<sub>2</sub> exhibits high specific capacitive performance, but due to its high cost and toxicity, its utility is limited and cannot be commercialized. To achieve high specific capacitance and high power density with low cost, a plethora of interest is laid on the combined usage of carbon based materials with metal oxides or polymers.<sup>4</sup> Graphene, the parent of all graphitic structures, offer high specific surface area, high conductivity and outstanding mechanical properties. These factors make graphene as a promising electrode material in supercapacitors. Recently graphene based metal oxide and polymer composites like G-MnO<sub>2</sub>,<sup>5</sup> G-Co<sub>3</sub>O<sub>4</sub>,<sup>6</sup> G-NiO,<sup>7</sup> RGO-CuO,<sup>8</sup> G-SnO<sub>2</sub>,<sup>9</sup> G-V<sub>2</sub>O<sub>5</sub>,<sup>10</sup> G-PANI<sup>11</sup> and G-PPY<sup>12</sup> has shown an enhanced specific capacitance than their individual form due to the synergistic effect between the composite materials.<sup>13</sup> Similar to metal oxides, metal sulfide based active electrode materials is also of tremendous interest for energy storage applications. Among the various transition metal chalcogenides, cobalt sulfide is a material of immense interest. Cobalt sulfides exist in different stoichiometric ratios like Co<sub>1-x</sub>S, CoS, CoS<sub>2</sub>, Co<sub>9</sub>S<sub>8</sub> and Co<sub>3</sub>S<sub>4</sub>.

ZnS and CoS electrodes have been studied for Li ion batteries and supercapacitor applications.<sup>14-15</sup> A maximum specific capacitance of 508 F/g for CoS<sup>16</sup> and 200 F/g for ZnS<sup>14</sup> has been reported. Though cobalt sulfide has higher thermal conductivity and electrical conductivity than other metal sulfides however, they have a number of drawbacks like capacity fading, low conductivity and poor cyclability, which will result in the decay of the specific capacity and the formation of polysulfide anions.<sup>17</sup> In this view, to resolve this, graphene was used as a conducting matrix where the presence of graphene not only increased the electrical conductivity but also improves the contact between electrode and electrolyte due to its large surface area.<sup>18</sup> More recently, a high specific capacitance of 1535 F/g for CoS-Graphene composite was reported.<sup>19</sup> Recently we have reported the role of N doped graphene based Co<sub>3</sub>S<sub>4</sub> for supercapacitor applications.<sup>20</sup>

In the current report, the electrochemical testing and supercapacitive property of Co<sub>9</sub>S<sub>8</sub>-Graphene nanocomposites are investigated. The nanocomposites were prepared by a simple chemical method from cobalt nitrate, thioacetamide and poly (N-vinylpyrrolidone) (PVP) with different loadings of graphene. To understand the effect of Co<sub>9</sub>S<sub>8</sub> and graphene, the nanocomposites were synthesized at various loadings of Co<sub>9</sub>S<sub>8</sub> and graphene material and a systematic investigation has been done to optimize a particular ratio. The fabricated electrodes have been studied by cyclic voltammetry and charge-discharge technique. Results show a good capacitive behavior of 808 F/g with good reversible charge/discharge behavior.

## Experiment method

### Materials

AR grade of cobalt nitrate (Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O) sodium nitrate (NaNO<sub>3</sub>), potassium permanganate (KMnO<sub>4</sub>), sulfuric acid (H<sub>2</sub>SO<sub>4</sub>, 99%), sodium borohydride (NaBH<sub>4</sub>), hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>, 30%), sodium hydroxide (NaOH), potassium hydroxide (KOH), poly(N-vinylpyrrolidone) (C<sub>6</sub>H<sub>9</sub>NO)<sub>n</sub> and thioacetamide (C<sub>2</sub>H<sub>5</sub>NS) were purchased from Sigma-Aldrich. All the solutions were prepared by using Milli-Q water (pH 7.2).

### Preparation of Graphene Oxide (GO)

Graphene oxide was prepared through modified Hummer's method according to the previous literature.<sup>21</sup> Graphite powder (0.5 g) and sodium nitrate (0.5 g) was mixed with 23 ml of concentrated sulfuric acid (98% pure) under stirring in an ice bath. After four hours, potassium permanganate (3 g) was added into the mixture. The reaction mixture was maintained at 35°C for two hours with continuous stirring in water bath, which was later diluted with 46 ml of water. After that the temperature was raised to 98°C and maintained for two hours. Then the reaction mixture was diluted with 100 ml warm water and 10 ml hydrogen peroxide under stirring. After 1 hour, the solution turned light yellow in color. The powders were collected after repeated washing in deionized water by centrifugation and the collected powder was dried at 60°C for 24 h.

### Preparation of Graphene

Graphene was synthesized using sodium borohydride as a reducing agent.<sup>21</sup> Initially, GO (40 mg) and water (40 ml) was

ultrasonicated for 1 hour. Then, this GO dispersed water was taken in an airtight container (60 ml capacity). To it was added sodium borohydride (0.24 g) and 5 ml of 1M sodium hydroxide solution. The mixture was kept at 90°C for 1 hour. After cooled room temperature, the powder was separated by centrifuging at 7500 rpm. Finally the powder was dried at 60°C for 24 hrs.

### Preparation of Co<sub>9</sub>S<sub>8</sub> nanoflakes/Graphene (Co<sub>9</sub>S<sub>8</sub>/G) nanocomposites

Co<sub>9</sub>S<sub>8</sub>/G nanocomposites were prepared according to the literature with a slight modification.<sup>22</sup> Typically Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (0.04 M), thioacetamide (0.06 M) and PVP (0.01 M) were added to 3 wt% of graphene in 40 ml of ultra-pure water. After the addition of NaOH (9.6 ml, 0.5 M), the solution was heated in an oil bath at 100°C for 1 h. The color of the mixture became black indicating the formation of Co<sub>9</sub>S<sub>8</sub>/G nanocomposite. The solution was then sonicated for 1 h, centrifuged and washed several times with ethanol. The final powder was dried at 60°C for 12 h. The powder was named as Co<sub>9</sub>S<sub>8</sub>/G-a. Similarly, other nanocomposites viz. Co<sub>9</sub>S<sub>8</sub>/G-b, Co<sub>9</sub>S<sub>8</sub>/G-c, Co<sub>9</sub>S<sub>8</sub>/G-d and Co<sub>9</sub>S<sub>8</sub>/G-e were synthesized by a similar method where a-e represents 3, 5, 7, 9, and 11 wt% of graphene loading. To know the effect of Co<sub>9</sub>S<sub>8</sub> on the electrochemical activity, the synthesis was repeated at three different cobalt concentration of 0.03 M, 0.04 M and 0.05 M Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O keeping 9 wt% graphene loading constant. The samples were named as Co<sub>9</sub>S<sub>8</sub>/G-x, Co<sub>9</sub>S<sub>8</sub>/G-y and Co<sub>9</sub>S<sub>8</sub>/G-z. Here Co<sub>9</sub>S<sub>8</sub>/G-d and Co<sub>9</sub>S<sub>8</sub>/G-y are the same composite i.e. Co<sub>9</sub>S<sub>8</sub> prepared from 0.04 M and 9 wt% graphene loading. Pure Co<sub>9</sub>S<sub>8</sub> was prepared by the above procedure without graphene.

### Materials characterization

X-ray diffraction system (BRUKER, D8 Advance, Germany) was used for the X-ray analysis with Cu-K $\alpha$  radiation ( $\lambda=1.540$  Å). Step scanning was done with 2 $\theta$  intervals from 10° to 65°. Fourier transform infrared spectra were recorded using Spectrum one: FTIR-spectrometer in the range 450-4500 cm<sup>-1</sup> to characterize the functional groups of the composites. UV-Vis spectroscopy (JASCO, V-670 spectrometer) was used for further studying the electronic structure and optical properties of the samples. Thermogravimetric analyses (TGA) of the samples were performed in air atmosphere with SDT Q600 (TA Instruments, Korea). X-ray photoelectron spectroscopy (XPS) measurements were carried out by using a Thermo Scientific Multilab 2000 spectrometer with Mg source. The surface morphologies and EDX mappings of the composites were obtained with FESEM by using a Hitachi S-4800 and EDS respectively. The AFM images were taken from the system (Nanosurf EasyScan 2 AFM, version 1.3).

### Electrical conductivity measurements

Electrical conductivity of pure Co<sub>9</sub>S<sub>8</sub> and Co<sub>9</sub>S<sub>8</sub>/G-d composite was measured using Four Probe Set up (Model- DFP 02) at different temperatures. The active material was pelletized with the thickness of 0.45 mm and radius 6.25 mm.

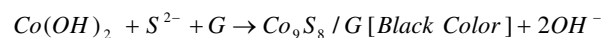
### Electrochemical measurements

All the electrochemical measurements were carried out with an electrochemical analyzer (CHI 600C work station, version 5.01) using a three electrode system in 6 M KOH as electrolyte solution under ambient conditions. The potentials and current were measured with respect to Ag/AgCl (sat.KCl) as the reference. Carbon paper (Purchased from Cabot, USA) and Pt wire were used as the working and counter electrode respectively. The cyclic voltammetry measurements were performed at various scan rates in the potential range from -0.1 to 0.3 V. The working electrode was prepared as follows: In brief, a known amount of active material was dispersed in 5 wt% of nafion and the mixture was coated on carbon paper (3 cm x 1cm x 0.1 cm). The total surface of active material coated area was 0.5 cm x 0.5 cm. The mass of the active material on electrode was 2 mg/cm<sup>2</sup>. Then, the electrode was dried for 3 h at room temperature.

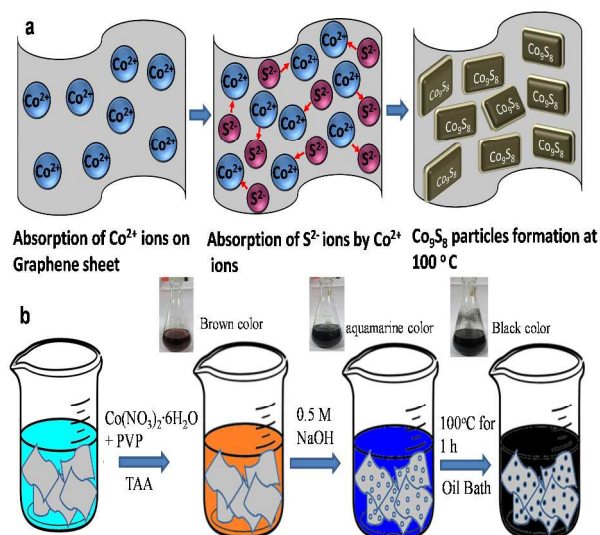
## Results and Discussion

### Formation mechanism

The formation mechanism of Co<sub>9</sub>S<sub>8</sub> is given as Scheme 1.<sup>23</sup> During the synthesis of the nanocomposites, cobalt nitrate and thioacetamide in the presence of PVP gave a brown color suspension initially and after the addition of NaOH, the color of the suspension turned aquamarine due to the reaction of Co<sup>2+</sup> ions with OH<sup>-</sup> to form Co(OH)<sub>2</sub>. Later, S<sup>2-</sup> ions obtained from thioacetamide decomposition at 100<sup>o</sup> C react with the formed Co(OH)<sub>2</sub> to form Co<sub>9</sub>S<sub>8</sub> nanoflakes on graphene sheets via noncovalent π-π interaction between delocalized π electrons of graphene and Co<sub>9</sub>S<sub>8</sub>.<sup>24</sup> The formation reaction is as follows<sup>16</sup>,



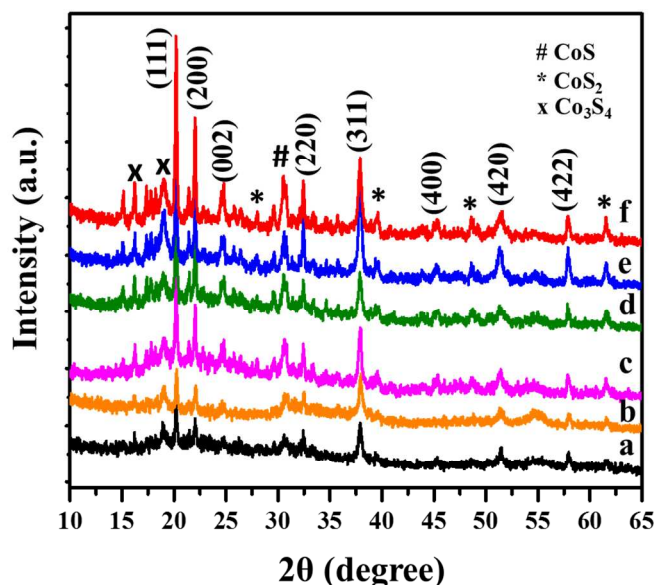
A schematic diagram of the formation mechanism and color changes during the course of the reaction is given in Scheme 1.



**Scheme. 1** Schematic diagram of (a) Co<sub>9</sub>S<sub>8</sub>/G nanocomposite formation with experimental photographic images and (b) Co<sub>9</sub>S<sub>8</sub> formation mechanism.

### Microstructural characterization

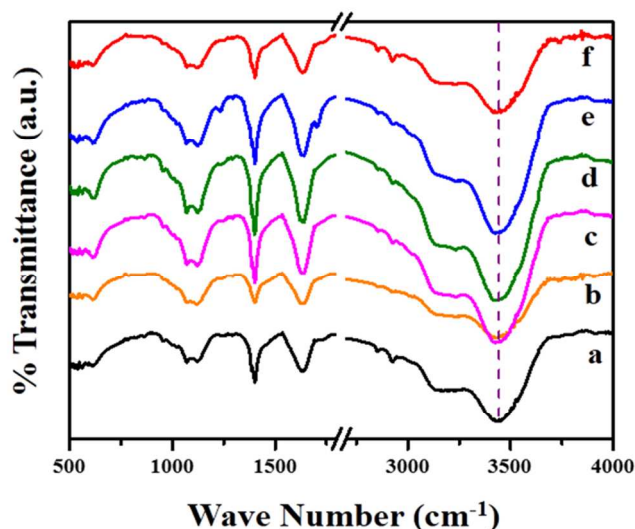
The structure and phase of the samples were identified by X-ray diffraction and the corresponding pattern of Co<sub>9</sub>S<sub>8</sub>/G nanocomposites and pure Co<sub>9</sub>S<sub>8</sub> is shown in Fig.1. As seen in Fig.1a, the diffraction peaks of pure Co<sub>9</sub>S<sub>8</sub> were observed at 20.2, 22.4, 32.2, 37.8, 45.2, 51.7 and 57.1<sup>o</sup> and can be indexed (111), (200), (220), (311), (400), (420) and (422) of cubic phase nature with lattice constant of 9.93 Å as in agreement with the JCPDS card no 19-0364.



**Fig. 1** XRD pattern of Co<sub>9</sub>S<sub>8</sub>/G composite at different concentrations of graphene (a) pure Co<sub>9</sub>S<sub>8</sub> (b) Co<sub>9</sub>S<sub>8</sub>/G-a (c) Co<sub>9</sub>S<sub>8</sub>/G-b (d) Co<sub>9</sub>S<sub>8</sub>/G-c (e) Co<sub>9</sub>S<sub>8</sub>/G-d (f) Co<sub>9</sub>S<sub>8</sub>/G-e.

Apart from the Co<sub>9</sub>S<sub>8</sub> peaks, other cobalt sulfide complex phases like CoS, CoS<sub>2</sub> and Co<sub>3</sub>S<sub>4</sub> were observed, suggesting that the cobalt sulphides are mixed phases of Co<sub>9</sub>S<sub>8</sub>, CoS<sub>2</sub>, CoS and Co<sub>3</sub>S<sub>4</sub> respectively. This fact was supported by Wang et al where it has been suggested that it was difficult to obtain pure phase of cobalt sulphide because of the complex stoichiometric of cobalt chalcogenides.<sup>25</sup> Though, the other phases are mixed in Co<sub>9</sub>S<sub>8</sub>, the major phases corresponds to Co<sub>9</sub>S<sub>8</sub>. In Co<sub>9</sub>S<sub>8</sub>/G nanocomposite, the peak at 24.9<sup>o</sup> clearly shows the crystal plane (002) of graphene sheets<sup>26</sup> in the composite. All the diffraction peaks are sharp indicating that the high crystallinity of Co<sub>9</sub>S<sub>8</sub> in Co<sub>9</sub>S<sub>8</sub>/G nanocomposite.





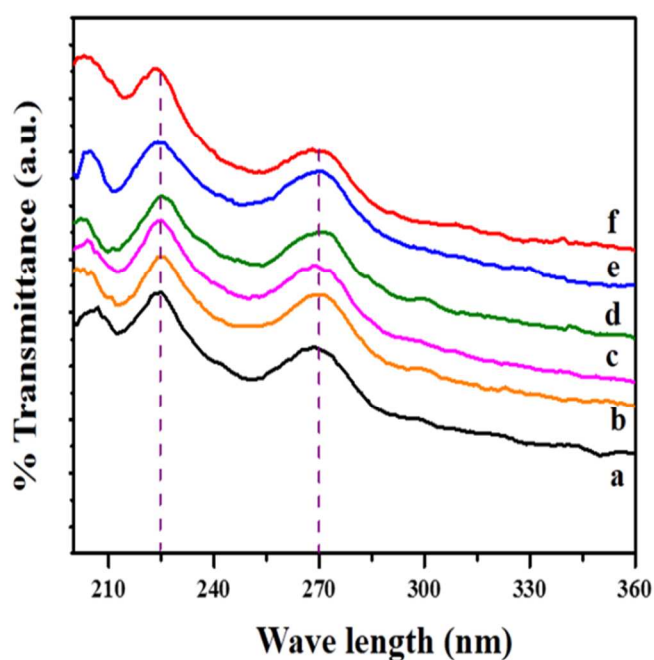
**Fig. 2** FTIR spectrum of  $\text{Co}_9\text{S}_8/\text{G}$  nanocomposites (a) pure  $\text{Co}_9\text{S}_8$  (b)  $\text{Co}_9\text{S}_8/\text{G}$ -a (c)  $\text{Co}_9\text{S}_8/\text{G}$ -b (d)  $\text{Co}_9\text{S}_8/\text{G}$ -c (e)  $\text{Co}_9\text{S}_8/\text{G}$ -d (f)  $\text{Co}_9\text{S}_8/\text{G}$ -e.

Fig. 2 shows the FT-IR spectrum of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G}$  nanocomposites. In the spectrum of  $\text{Co}_9\text{S}_8$  nanoflakes, the peak at 1130 indicates the S-O bending vibration of sulfonated groups of cobalt sulfide.<sup>27</sup> The peak at 1632  $\text{cm}^{-1}$  could be due to the presence of PVP on the surface of cobalt sulfide and a small peak at 619.9  $\text{cm}^{-1}$  is related to the stretching vibration of Co atoms in the surface of  $\text{Co}_9\text{S}_8$ .<sup>28</sup> In the case of  $\text{Co}_9\text{S}_8/\text{G}$  nanocomposite, the peak at 1397  $\text{cm}^{-1}$  corresponds to the epoxide group of graphene.<sup>29-30</sup>

To further know about the optical properties of  $\text{Co}_9\text{S}_8/\text{G}$  nanocomposites, DRS-UV was recorded as given in Fig.3. There are two major transmittance peaks observed. The peak at 270 nm indicates that the electronic conjugation is restored within the graphene sheets upon  $\text{NaBH}_4$  reduction<sup>31-32</sup> and the peak at 225 nm might be due to the vibrations of S molecules in  $\text{Co}_9\text{S}_8$  nanoflakes. The optical band gap of the composite material was calculated using the below equation from UV spectrum:

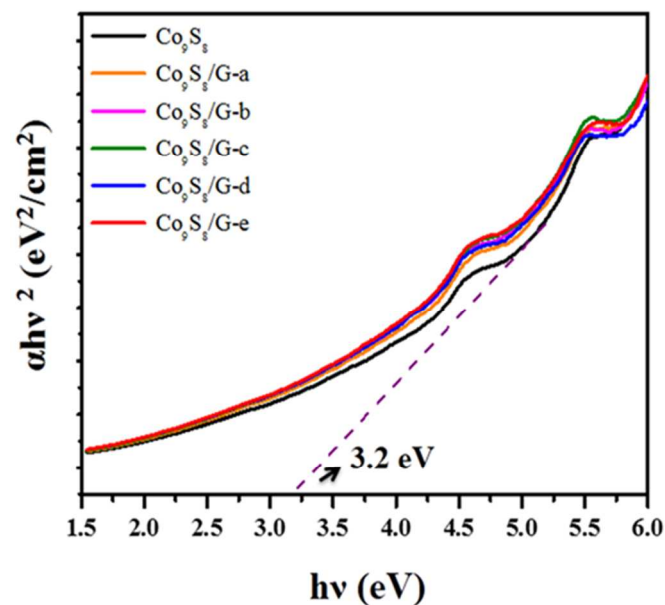
$$(\alpha h\nu)^n = A(h\nu - E_g)$$

To find the energy band gap, a plot between  $(\alpha h\nu)^2$  vs.  $(\alpha\nu)$  was drawn, where ' $\alpha$ ' is the absorption coefficient, ' $h\nu$ ' is the photon energy, ' $A$ ' is a constant, ' $E_g$ ' is the band gap, and ' $n$ ' is either  $\frac{1}{2}$  for an indirect transition or 2 for a direct transition. Thus, a plot of  $(\alpha h\nu)^2$  vs.  $(\alpha\nu)$  is a straight line whose intercept on the energy axis gives the energy gap.



**Fig. 3** DRS UV spectrum of (a) pure  $\text{Co}_9\text{S}_8$  (b)  $\text{Co}_9\text{S}_8/\text{G}$ -a (c)  $\text{Co}_9\text{S}_8/\text{G}$ -b (d)  $\text{Co}_9\text{S}_8/\text{G}$ -c (e)  $\text{Co}_9\text{S}_8/\text{G}$ -d (f)  $\text{Co}_9\text{S}_8/\text{G}$ -e.

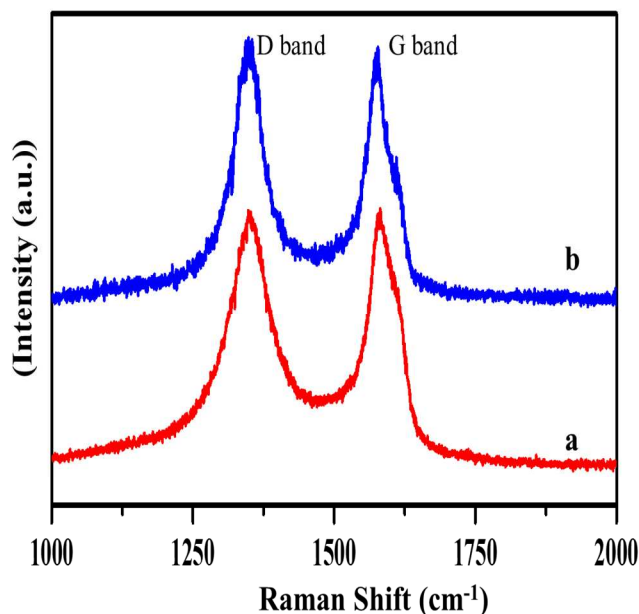
According to the above equation, based on the direct transition, a band gap of 3.2 eV was obtained for pure  $\text{Co}_9\text{S}_8$ , which matched well with the literature reports.<sup>33</sup> In the case of composites, the band gap value was decreased, which could be due to the fact that graphene sheets provide better electrical conductivity in the composite.



**Fig. 4** Optical band gap spectrum of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G}$  composites

Raman spectroscopy is widely used to study the crystalline structure of graphene and graphene composites. Fig. 5 shows the Raman spectra of pure graphene and  $\text{Co}_9\text{S}_8/\text{G}$ -d composite. For pure graphene, the two peaks at 1350  $\text{cm}^{-1}$  and 1580  $\text{cm}^{-1}$

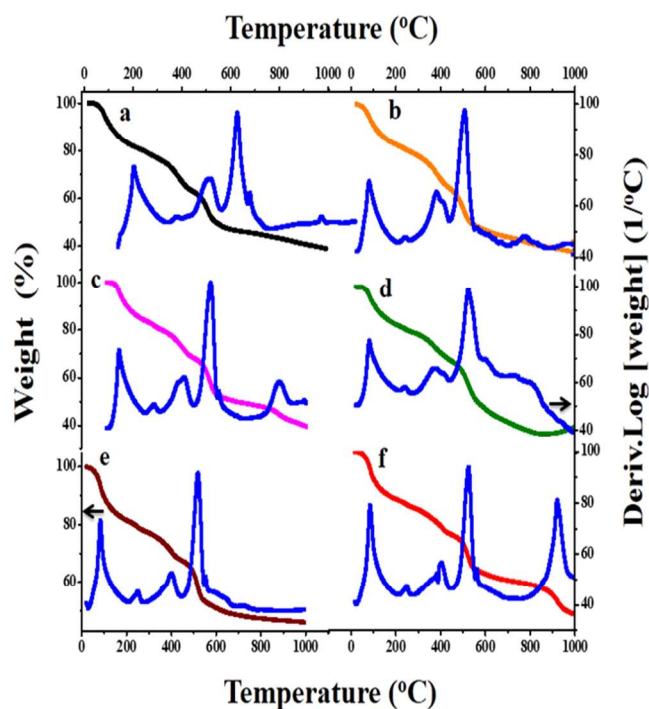
can be assigned to the D and G band, respectively. The D-band is related to the vibrations of  $sp^3$  carbon atoms of disordered graphene, while the G band is a result of the first order scattering of  $E_{2g}$  mode of  $sp^2$  carbon domains from graphene. The D/G ratio of the composite has increased than pristine graphite<sup>34</sup>, which confirmed the presence of graphene sheets.



**Fig. 5** Raman spectrum of (a) Graphene and (b)  $Co_9S_8/G$  composite

Both the intensity and position of D and G bands lead to the structural changes in graphene matrix, which are affected by doping, layer number, defects etc.<sup>35</sup> The  $Co_9S_8/G$ -d composite showed a red shift of G band at  $1576\text{ cm}^{-1}$  compared with graphene, strongly suggesting the electronic interactions between graphene and  $Co_9S_8$ .<sup>35</sup> Also, a higher D/G intensity ratio (1.01) of  $Co_9S_8/G$ -d composite than pure graphene (0.90) is due to the decreased  $sp^2$  carbon domains when  $Co_9S_8$  are decorated between graphene sheets.<sup>36</sup>

Thermogravimetric analysis was performed in air from room temperature to  $800^\circ\text{C}$  to quantify the amount of graphene, decomposition details and phase changes of the composite. Fig. 6 shows the TGA and DTA curves of pure  $Co_9S_8$  and  $Co_9S_8/G$  nanocomposites. Initially, 8% weight loss of material was observed for pure graphene from  $50^\circ$  to  $125^\circ\text{C}$  attributed to the evaporation of water and other organic molecules (Fig. S1). A gradual weight loss was observed from the temperature  $125^\circ$  to  $800^\circ\text{C}$  in graphene. The actual weight loss of the material could be observed after  $800^\circ\text{C}$  due to burning of graphene and the weight of graphene after  $1000^\circ\text{C}$  was 60%. There are three stages of weight loss observed for pure  $Co_9S_8$  (Fig.6a).

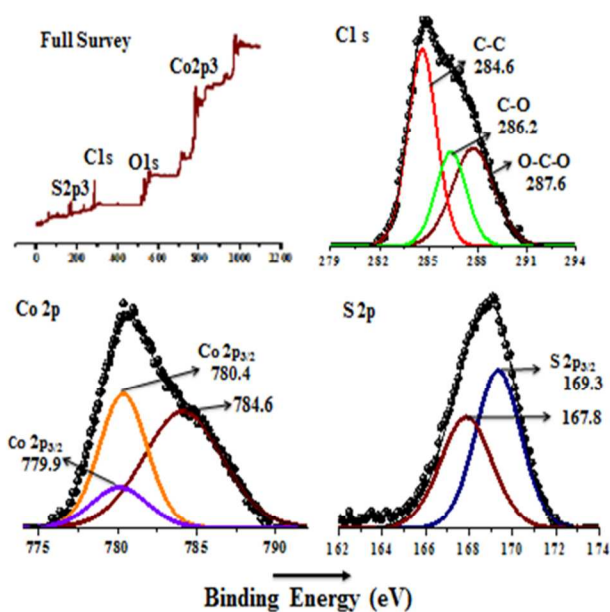


**Fig. 6** Thermogravimetric TGA and DTA graph of (a) pure  $Co_9S_8$  (b)  $Co_9S_8/G$ -a (c)  $Co_9S_8/G$ -b (d)  $Co_9S_8/G$ -c (e)  $Co_9S_8/G$ -d (f)  $Co_9S_8/G$ -e.

In DTA curve, the first two endothermic peaks at  $87^\circ$  and  $400^\circ\text{C}$  might be due to the evaporation of solvent and PVP molecules in cobalt sulfide. The exothermic peak at  $518^\circ\text{C}$  is attributed to oxidation and conversion of  $Co_9S_8$  to  $Co_3O_4$ .<sup>37</sup> In  $Co_9S_8/G$ -d nanocomposite, the second exothermic peak is due to the burning of graphene and weight loss of graphene in this temperature is about 10% as observed in pure graphene. From Fig.6 (e), the weight % of graphene in  $Co_9S_8/G$ -d nanocomposite was calculated to be around 12%. When the concentration of graphene was below 11 wt %, there was no second exothermic peak indicating that the weight loss of graphene was negligible in such composites.

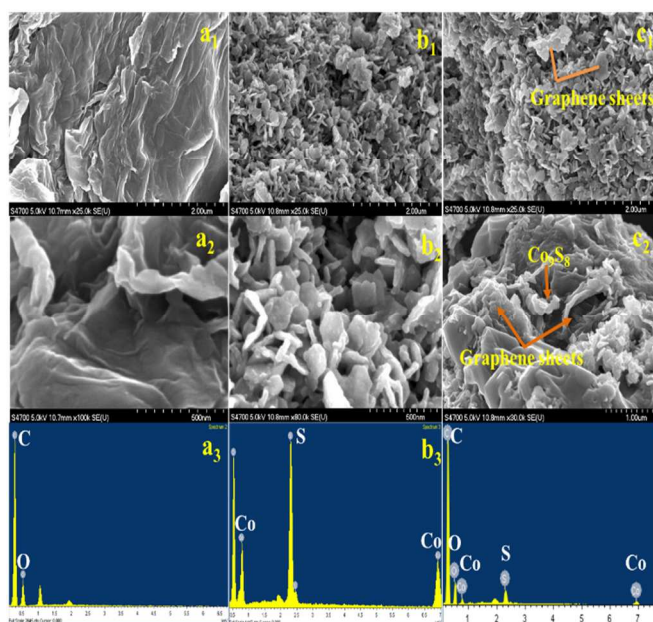
The chemical composition of  $Co_9S_8/G$ -d nanocomposite was further investigated by using XPS analysis. An overall spectrum shows the presence of Co, S, C and O in the composites with no other element incorporated. In the deconvoluted spectra of C1s (Fig.7b), a main peak was observed at  $284.6\text{ eV}$  due to graphitic  $sp^2$  carbon atoms. The other two peaks located at  $286.2$  and  $287.6$  are due to C-O and O-C-O, suggesting the presence of oxygen in graphene sheets.<sup>38</sup> The Co 2p spectrum was deconvoluted further, which gave major and minor peaks at  $780.4$  and  $779.9\text{ eV}$ , corresponding to Co  $2p_{3/2}$  characteristic peak of cobalt sulfide. Another peak at  $784.6\text{ eV}$  indicates the formation of octahedrally coordinated  $Co^{2+}$  ions in cobalt.<sup>39</sup> A trace amount of oxygen was also observed from XPS, which is due to the residual oxygen-containing groups (such as OH and COOH) bonded with C atoms in graphene. The peaks at  $167.8$  and  $169.3\text{ eV}$  from S 2p spectrum is attributed to the sulphite and sulphates in cobalt sulfide. No peaks of sulfide are found in S 2p spectrum. This suggesting that the cobalt does not bind with sulfur atoms directly without involvement of oxygen in sulphite/sulphates.<sup>40-</sup>

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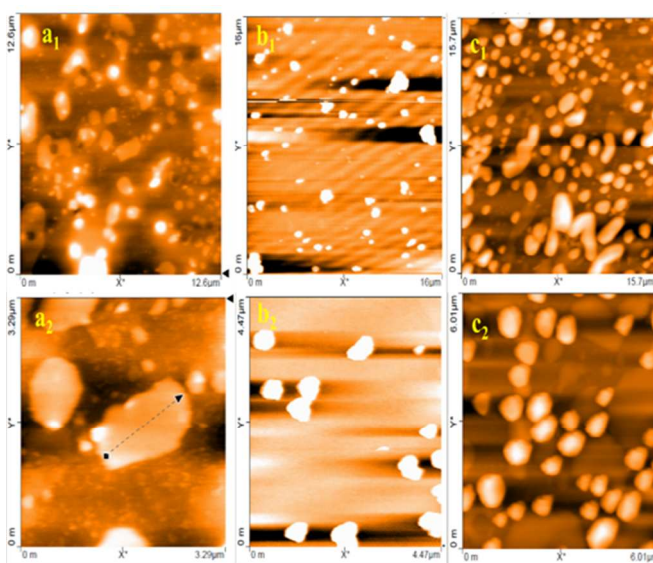
**Fig. 7** Wide scan XPS survey and core-level XPS spectrum of Co2p, Cls and S2p for Co<sub>9</sub>S<sub>8</sub>/G-d composite

A homogeneous distribution of catalyst is essential for storage applications and in this regard, FESEM was recorded and the corresponding images of graphene, Co<sub>9</sub>S<sub>8</sub> and Co<sub>9</sub>S<sub>8</sub>/G-d nanocomposites at various magnifications are shown in Fig. 8. Few layers of graphene with crumpled sheet like morphology could be seen in Fig. 8a<sub>1</sub>, a<sub>2</sub> indicates the graphene sheets were exfoliated during graphene synthesis<sup>42</sup> and in the case of pure Co<sub>9</sub>S<sub>8</sub>, homogeneously shaped nanoflakes of Co<sub>9</sub>S<sub>8</sub> with approximately 300 nm size could be seen (Fig. 8b<sub>1</sub>, b<sub>2</sub>). Also could be seen from the images of Co<sub>9</sub>S<sub>8</sub>/G-d nanocomposite is that there is no sign of aggregation of Co<sub>9</sub>S<sub>8</sub> nanoflakes and they are uniformly dispersed on the graphene sheets (Fig. 8c<sub>1</sub>, c<sub>2</sub>). These individual nanoflakes can facilitate fast electron transfer between graphene and Co<sub>9</sub>S<sub>8</sub> in electrochemical reactions rather than in their aggregation form. The corresponding EDS spectrum of Co<sub>9</sub>S<sub>8</sub>/G-d is given in Fig. 8c<sub>3</sub>, which shows the presence of Co and S with 1:0.8 atomic wt % ratios. Fig. S2-S4 shows the mapping images of C in graphene, Co and S in Co<sub>9</sub>S<sub>8</sub> and C, Co, S in Co<sub>9</sub>S<sub>8</sub>/G-d nanocomposite.



**Fig. 8** FESEM and EDS spectrum of (a<sub>1</sub>, a<sub>2</sub>, a<sub>3</sub>) graphene (b<sub>1</sub>, b<sub>2</sub>, b<sub>3</sub>) pure Co<sub>9</sub>S<sub>8</sub> (c<sub>1</sub>, c<sub>2</sub>, c<sub>3</sub>) Co<sub>9</sub>S<sub>8</sub>/G-d nanocomposites.

The morphology of the samples was further analyzed by AFM technique. As given in Fig. 9a<sub>1</sub>, a<sub>2</sub>, few layers of graphene sheets could be seen. The thickness of graphene is about 10 nm as observed from Fig. 9a<sub>2</sub> revealing the presence of few layer graphene sheets, which is in well agreement with SEM images. The typical thickness of individual graphene sheet is ~0.34 nm.<sup>43</sup> Thus, the number of sheets in our study to be approximately 30 layers. The average size of Co<sub>9</sub>S<sub>8</sub> was measured to be ~500 nm from Fig. 9b<sub>1</sub> & b<sub>2</sub>, which indicates the aggregation of Co<sub>9</sub>S<sub>8</sub>. In the case of Co<sub>9</sub>S<sub>8</sub>/G composite, a uniform size distribution of Co<sub>9</sub>S<sub>8</sub> was observed, which is due to the prevention of agglomeration by graphene sheets.

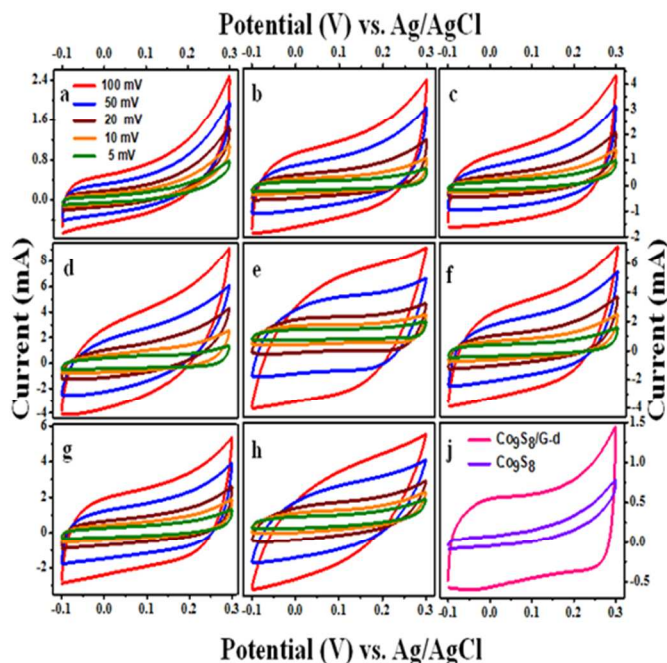


**Fig. 9** AFM images (a<sub>1</sub>-a<sub>2</sub>) Graphene (b<sub>1</sub>-b<sub>2</sub>) pure Co<sub>9</sub>S<sub>8</sub> and (c<sub>1</sub>-c<sub>2</sub>) Co<sub>9</sub>S<sub>8</sub>/G-d composite.

### Electrochemical behavior and specific capacitance measurements



Performance of the prepared composite materials for supercapacitors was estimated by cyclic voltammetry (CV), galvanostatic charge-discharge and electrochemical impedance measurements in 6M KOH aqueous solutions. The CV of a material assures the capacitive feature where in a large magnitude of current, rectangular shape of voltammograms and symmetry in anodic and cathodic directions are features of an ideal capacitive behavior. The working cell potential range for the samples ranged between -0.1 to 0.3 V as obtained from CV measurements. Fig. 10 shows typical cyclic voltammograms of pure  $\text{Co}_9\text{S}_8$  and composites at different scan rates. As evidenced from the Fig, CV curves of all the electrode materials are symmetrical in nature, characteristic of ideal capacitive behavior<sup>44</sup> with quasi rectangular shape indicating the excellent capacitive behavior and a low contact resistance in the supercapacitor electrode.<sup>45</sup> No peaks were observed at different scan rates, suggesting that the electrode is charged and discharged at a pseudo-constant rate. As observed from the Fig, cyclic voltammetry curves of all the composites are slightly different from pure  $\text{Co}_9\text{S}_8$  as the overall specific capacitance for composites is enhanced due to the combined contributions from EDLC and pseudocapacitance of graphene, while the capacitance of  $\text{Co}_9\text{S}_8$  is from pseudocapacitance only. The electric double layer capacitance of composites is influenced by the oxygen groups, which is present in the edges of graphene. These oxygen groups enhances the accessibility to hydrophilic surface in aqueous electrolytes.<sup>46</sup> The pseudocapacitance is due to the interactions between oxygen functional groups in graphene sheets and metal ions in electrolyte.<sup>47</sup> Also, the electrochemical capacitive behavior depends on the rate of charge transfer, i.e. the diffusion of anions and cations towards the electrode/electrolyte interfaces.



**Fig.10** Cyclic voltammetric behavior of (a) pure  $\text{Co}_9\text{S}_8$  (b)  $\text{Co}_9\text{S}_8/\text{G-a}$  (c)  $\text{Co}_9\text{S}_8/\text{G-b}$  (d)  $\text{Co}_9\text{S}_8/\text{G-c}$  (e)  $\text{Co}_9\text{S}_8/\text{G-d}$  (f)  $\text{Co}_9\text{S}_8/\text{G-e}$  (g)  $\text{Co}_9\text{S}_8/\text{G-x}$  (h)  $\text{Co}_9\text{S}_8/\text{G-z}$  (j) Cyclic voltammetry response of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G-d}$  composite at 5 mV scan rate.

It could be seen that both the cathodic and anodic current response of pure  $\text{Co}_9\text{S}_8$  is almost above the zero current level, suggesting poor electrochemical performance of the synthesized  $\text{Co}_9\text{S}_8$  nanoflakes. The addition of graphene into  $\text{Co}_9\text{S}_8$  can improve the dispersion of  $\text{Co}_9\text{S}_8$  on the surface of graphene and provide more paths for ion diffusion due to the increased interface area between electrode and electrolyte. Like oxygen, sulfur ions in  $\text{Co}_9\text{S}_8$  also provide pseudocapacitance activity due to the fact that sulfur and oxygen belong to the same group.<sup>16</sup> CV measurements were conducted at various scan rates. The relative increment in current response with the scan rate constitutes an ideal capacitive behavior.<sup>48</sup> The specific capacitance was calculated from the CV curves according to the following formula<sup>49, 21</sup>

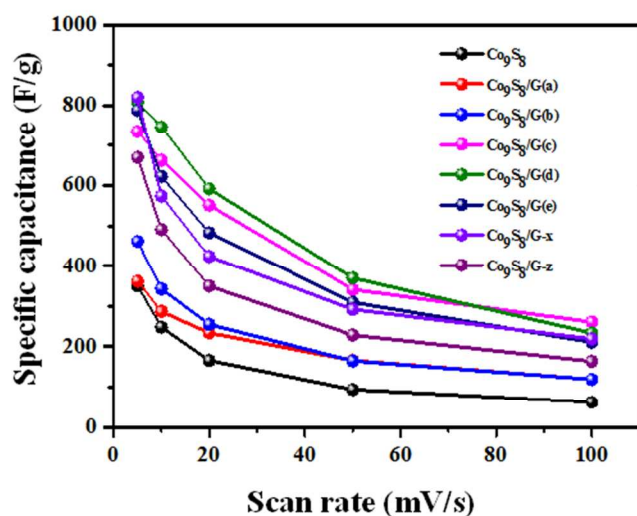
$$C = \frac{(I_+ - I_-)}{mv}$$

where,  $I_+$  is the maximum current in the positive scan (amp),  $I_-$  is the maximum current in the negative scan (amp), 'm' the mass of the active material coated on working electrode (gm) and 'v' is the scan rate (V/s).

To know the effect of graphene, CV was run for pure  $\text{Co}_9\text{S}_8$  and the same has been compared with  $\text{Co}_9\text{S}_8/\text{G-d}$  composite (Fig. 10j). As clearly seen, the sample with graphene has a well-defined cyclic voltammetric behavior. Plot of scan rate versus specific capacitance of all samples shown in Fig. 11. From this plot, a maximum specific capacitance of 361.1, 460.8, 734.9, 808 and 788.2 F/g was obtained for  $\text{Co}_9\text{S}_8/\text{G-a}$ ,  $\text{Co}_9\text{S}_8/\text{G-b}$ ,  $\text{Co}_9\text{S}_8/\text{G-c}$ ,  $\text{Co}_9\text{S}_8/\text{G-d}$  and  $\text{Co}_9\text{S}_8/\text{G-e}$  composite electrodes respectively, at 5 mV/s scan rate. At the same scan rate, pure  $\text{Co}_9\text{S}_8$  electrode exhibited a specific capacitance of 350 F/g, which is lower than that of all other composite electrode materials. It is believed that such a large capacitance difference is caused by the introduction of graphene into  $\text{Co}_9\text{S}_8$ , which is the reason for obtaining high specific capacitances. The specific capacitance values decreased as the scan rate increased. At low scan rate, positive ions ( $\text{K}^+$ ) can easily diffuse into almost all available spaces of materials leading to sufficient insertion reaction. On the other hand, at higher scan rate, positive ions can approach only outer surface of the electrode and the material embedded in the inner space has little contribution to the capacitance behavior thus leading to a slight deviation from ideality and less capacitance.<sup>44</sup> On the basis of the CV results, it is well evidenced that the loading of graphene sheets in  $\text{Co}_9\text{S}_8$  can provide better electrical conductivity in the network. Among the various composites, 9 wt% of graphene loaded ( $\text{Co}_9\text{S}_8/\text{G-d}$ ) showed the highest capacitance. Thus,  $\text{Co}_9\text{S}_8/\text{G-d}$  composite provided large and better electrochemical activities towards the energy storage devices.

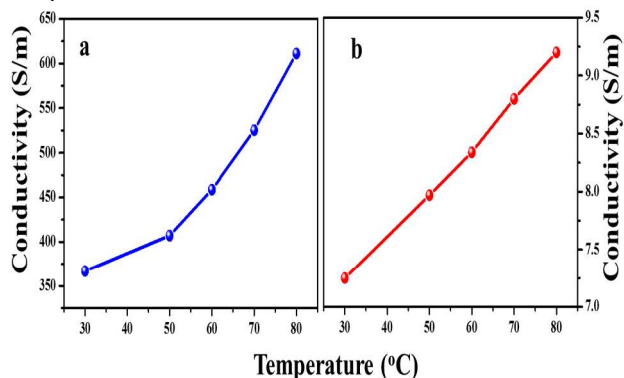
To further know the effect of cobalt sulfide on graphene, CV measurements were done at various loadings of  $\text{Co}_9\text{S}_8$ . The composite was synthesized at three different concentrations of Co source with details given in the experimental section. Investigations showed that a maximum specific capacitance of 820 F/g and 670 F/g for  $\text{Co}_9\text{S}_8/\text{G-x}$  and  $\text{Co}_9\text{S}_8/\text{G-z}$  composite respectively. Thus, a high concentration (0.05 M) of cobalt sulfide in graphene leads to poor electrochemical performance.





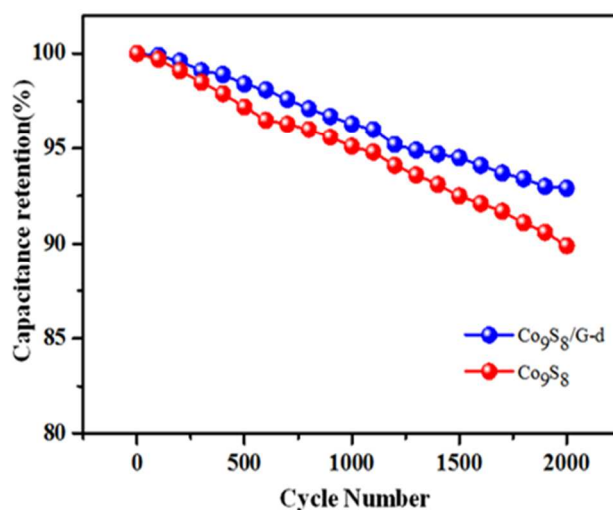
**Fig. 11** Plot of scan rate versus specific capacitance graph.

Note that, the specific capacitance of  $\text{Co}_9\text{S}_8/\text{G-x}$  electrode has slightly higher specific capacitance than  $\text{Co}_9\text{S}_8/\text{G-y}$  electrode material at 5 mV/s scan rate, but the values of specific capacitance decreased for higher scan rates due to the diffusion effect of ions. These results suggested the difficult movement of ions into electrode from electrolyte. Thus, it can be concluded that  $\text{Co}_9\text{S}_8/\text{G-y}$  electrode (viz. the composite prepared from 0.04 M Co precursor and 9 wt% graphene loading) showed the highest activity among the other composites. This increased specific capacitance of  $\text{Co}_9\text{S}_8/\text{G-d}$  or  $\text{Co}_9\text{S}_8/\text{G-y}$  electrode is due to the higher electrical conductivity than pure  $\text{Co}_9\text{S}_8$ .<sup>50</sup> This fact is supported by the electrical conductivity analysis given in Fig. 12. As seen from the figure, the presence of graphene increased the electrical conductivity of the composite, which further aided the increase in capacitance value.



**Fig. 12** Electrical conductivity of (a)  $\text{Co}_9\text{S}_8/\text{G-d}$  composite and (b) pure  $\text{Co}_9\text{S}_8$  at different temperatures.

The capacitance retention property of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G-d}$  electrodes were investigated by repeating the CV measurements at 100 mV/s scan rate for 2000 cycles. As shown in Fig. 13, the capacitance retention of  $\text{Co}_9\text{S}_8/\text{G-d}$  composite electrode was around 92.9% even after 2000 cycles, whereas pure  $\text{Co}_9\text{S}_8$  exhibited retention of 89.9%, indicating high electrochemical cyclic stability of composite electrode.



**Fig. 13** Capacitance retention of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G-d}$  electrodes as function of cycle number measured at 100 mV/s scan rate.

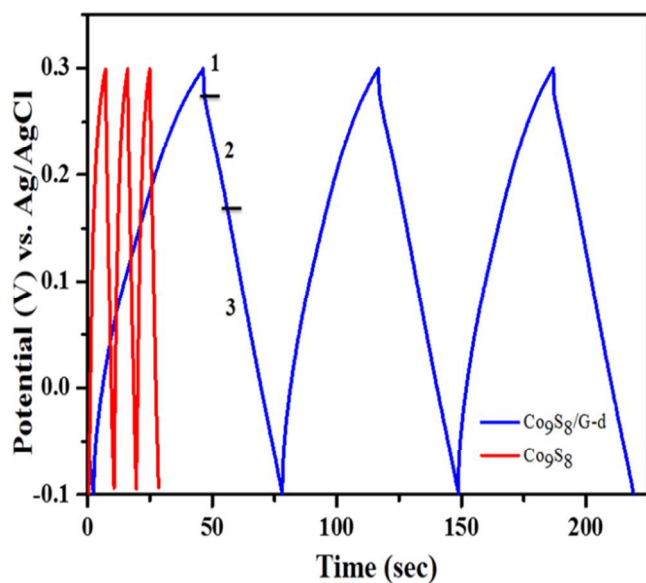
A comparison is made with the current material and the reported as given in Table 1 and as seen from the table, it can be seen that  $\text{Co}_9\text{S}_8/\text{G}$  reported in this paper has good activity and could be used for supercapacitor applications.

**Table 1:** Comparison of specific capacitance of various electrode materials towards the supercapacitor applications.

Material	E[a]	SR[b]	SC[c]	R[d]
Graphene / $\text{Mn}_2\text{O}_4$	6 M KOH	5	159	51
F-rGO-Ni(OH) <sub>2</sub>	1 M KCl	5	750	52
NiO/CNT	6 M KOH	-	523	53
$\text{MnO}_2/\text{PPY}/\text{Carbon}$ nanofibre	2 M KCl	2	705	54
Graphitic petals/ $\text{MnO}_2$	1 M $\text{Na}_2\text{SO}_4$	2	580	55
Graphene/PANI	3 M KOH	1	463	56
Graphene/Sn	2 M $\text{KNO}_3$	1	320	57
MWCNT/ $\text{Co}_3\text{O}_4$	1 M $\text{Na}_2\text{SO}_4$	-	720.2	58
CoS	3 M KOH	-	508	23
$\text{CoS}_2$ -Graphene	6 M KOH	5	253	59
$\text{Co}_9\text{S}_8$	6 M KOH	5	350	This work
$\text{Co}_9\text{S}_8/\text{G-d}$	6 M KOH	5	808	This work

[a] Electrolyte, [b] Scan rate (mV/s), [c] Specific capacitance (F/g), [d] Reference

To further assess the potential of  $\text{Co}_9\text{S}_8/\text{G-d}$  nanocomposite and  $\text{Co}_9\text{S}_8$  electrodes, galvanostatic charge/discharge measurements were carried out. The charge-discharge test is a reliable method for the evaluation of electrochemical performances of electrode material under constant current condition. Fig.14 shows the galvanostatic charge/discharge curve of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G-d}$  composite. The charge/discharge experiment was carried out at a current density of  $0.5 \text{ A g}^{-1}$  in the potential range  $-0.1$  to  $0.3 \text{ V}$ . As can be observed, during the charging-discharging process, the charge curve is almost symmetric to its corresponding discharge counterpart with a slight curvature. The well symmetric and triangular nature of charge-discharge curve reveal the electric double layer capacitive and ideal capacitive behavior of the material.<sup>60-61</sup> Apart from double layer capacitance, pseudocapacitance behavior was also noted in charge-discharge curve suggesting that the overall capacitance of composite electrodes was a mixture of electric double layer and pseudocapacitors. Typically, three regions could be seen in the discharge curve. A sudden drop of the current at the starting point of discharge current (region I), due to the electrode/electrolyte interface resistance or internal resistance.<sup>62</sup> The IR drop is attributed to the internal resistance of the electrode associated with the electrical connection resistance, bulk solution resistance and resistance of ion migration in electrode materials. The IR drop of pure  $\text{Co}_9\text{S}_8$  electrode was much higher than that of  $\text{Co}_9\text{S}_8/\text{G-d}$  electrodes which is clearly seen in Fig. 14, indicating that the internal resistance of  $\text{Co}_9\text{S}_8$  is much higher than the composite electrodes. This small internal resistance (IR) drop, indicate the presence of pseudocapacitance behavior along with the double layer capacitance.<sup>63</sup> The double layer capacitance behavior is due to the charge separation of electrode and electrolyte interfaces as seen in region II (a linear variation of the time dependence of the potential). The slope variation of time dependence of potential (region III) indicates the pseudocapacitance behavior that might be due to electrochemical adsorption/desorption or redox reaction at electrode/electrolyte interfaces.<sup>64-66</sup> A discharge time of 3.8 sec was noticed for pure  $\text{Co}_9\text{S}_8$  electrode, which is too low as compared to the composite electrode.

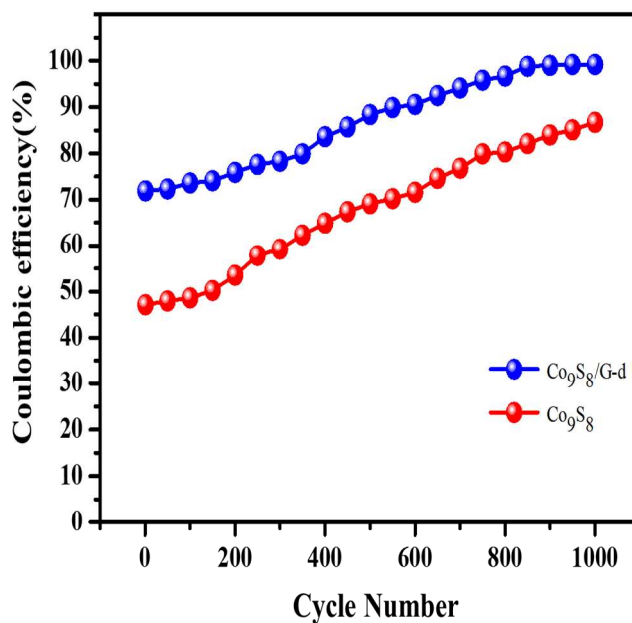


**Fig. 14** Galvanostatic charge/discharge behavior of pure  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G-d}$  composite at current density  $0.5 \text{ A g}^{-1}$ .

Coulombic efficiency is an important factor which describes the stability of electrode materials through charge/discharge process. The coulombic efficiency of electrode is calculated according to the formula<sup>67</sup>,

$$\eta = \frac{t_d}{t_c} \times 100$$

where,  $t_d$  is discharge duration and  $t_c$  is charging duration. As seen in Fig. 15, the initial coulombic efficiency of both electrodes is low, which is due to the re-stacking of graphene layers in  $\text{Co}_9\text{S}_8/\text{G-d}$  composite and side reactions between graphene sheets and electrolyte arising from abundant functional groups such as  $\text{COOH}$  and  $\text{OH}$ , etc in graphene during charging of electrode.<sup>68</sup> Also, the coulombic efficiency of electrodes were gradually increased from the 1<sup>st</sup> cycle to 1000<sup>th</sup> cycle, which could be attributed due to the activation effect of electrodes during long term charging/discharging process.<sup>69</sup> Notably, the  $\text{Co}_9\text{S}_8/\text{G-d}$  composite electrode shows the coulombic efficiency of  $\sim 99.2\%$  even after 1000 cycles, which is higher than that of bare  $\text{Co}_9\text{S}_8$  (86.7%). This result suggested that the energy loss of  $\text{Co}_9\text{S}_8/\text{G-d}$  electrode during discharge process was low. Fig. S5 & S6 shows the galvanostatic charge/discharge of  $\text{Co}_9\text{S}_8/\text{G-d}$  composite and pure  $\text{Co}_9\text{S}_8$  at different current densities. It can be seen that the IR drop is negligible at low discharge current density as both inner and outer pore surfaces of electrode could be utilized for electrolyte propagation during the charge/discharge process. Thus, the electrode has stored a high charge at low discharge current density.<sup>70</sup>



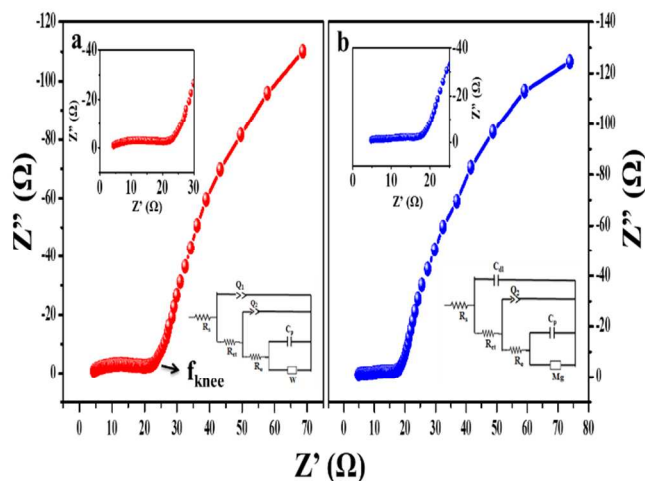
**Fig.15** Coulombic efficiency of  $\text{Co}_9\text{S}_8$  and  $\text{Co}_9\text{S}_8/\text{G-d}$  electrodes at current density of  $0.5 \text{ A g}^{-1}$

EIS analysis was used to investigate the fundamental behavior of the electrode materials for supercapacitors like charge transport phenomenon at the electrode/electrolyte interface. Electrochemical Impedance Spectroscopy (EIS) was measured with an AC perturbation of  $0.01 \text{ V}$  in the frequency range of  $0.01 \text{ Hz}$  to  $10^5 \text{ Hz}$  at an open circuit potential. Fig.16 shows the Nyquist plots of  $\text{Co}_9\text{S}_8$  and composite electrodes

with their corresponding equivalent circuit. All spectra shows a semicircle in the high frequency region (inset pictures), a straight line inclined at a constant phase in the mid frequency region and an almost vertical capacitive line in the low frequency region. The impedance plot of  $\text{Co}_9\text{S}_8$  was increased sharply at low frequency and tends to become vertical due to non-faradic charge storage mechanism. This vertical straight  $45^\circ$  line shows the pure capacitive behavior of cobalt sulfide.<sup>71</sup> The intersection of curve at real axis in high frequency region gives the solution resistance ( $R_s$ ) which reflects the ionic conductivity of electrolyte system and the diameter of the semicircle represents interfacial charge transfer resistance ( $R_{ct}$ ), which is due to the electron conductivity of electrode material.<sup>67</sup> It is well known that a large semicircle indicates high interfacial charge transfer resistance, which is due to the poor conductivity of the electrode. All the parameters extracted from EIS spectrum are given in Table 2 and as seen, both solution and charge transfer resistance of  $\text{Co}_9\text{S}_8/\text{G-d}$  or  $\text{Co}_9\text{S}_8/\text{G-y}$  composite electrode were less than pure  $\text{Co}_9\text{S}_8$ , which could be attributed to the enhanced conductivity of graphene sheets.<sup>72</sup>

**Table 2:** Comparison of observed results of  $\text{Co}_9\text{S}_8/\text{G-d}$  and  $\text{Co}_9\text{S}_8$  electrodes from EIS analysis

Sample	$R_s$ ( $\Omega$ )	$R_{ct}$ ( $\Omega$ )	$\tau$ (s)	$f_{knee}$ (Hz)
$\text{Co}_9\text{S}_8$	4.1	26.3	$5 \times 10^{-2}$	$9.766 \times 10^{-2}$
$\text{Co}_9\text{S}_8/\text{G-d}$ or $\text{Co}_9\text{S}_8/\text{G-y}$	3.5	19	$1.64 \times 10^{-5}$	$4.542 \times 10^{-1}$



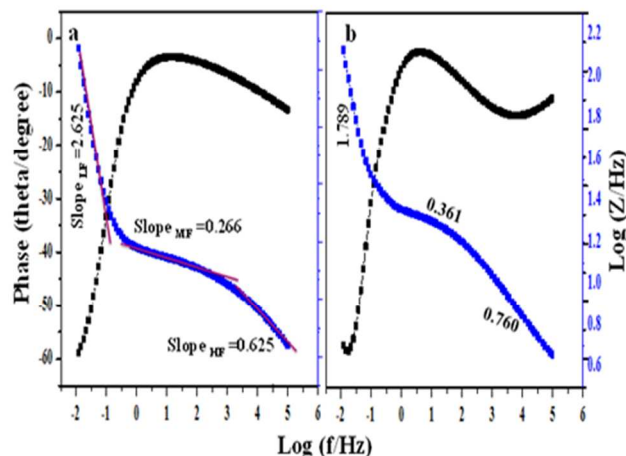
**Fig. 16** Nyquist Plots of (a) pure  $\text{Co}_9\text{S}_8$  and (b)  $\text{Co}_9\text{S}_8/\text{G-d}$  composite.

The time constant of electrode materials can be calculated from the frequency ( $f^*$ ) corresponding to the maximum imaginary component ( $-Im Z''$ ) of the semicircle by using expression<sup>59</sup>

$$\tau = \frac{1}{2\pi f^*}$$

Low time constant is an important characteristic for fast charge-discharge functioning of the supercapacitor electrodes. The time constant of  $\text{Co}_9\text{S}_8/\text{G-d}$  was calculated to be  $1.96 \times 10^{-5}$  seconds, which is lower than that of pure  $\text{Co}_9\text{S}_8$  electrode

suggesting the fast charge-discharge characteristic behavior of composite due to the presence of graphene.



**Fig. 17** Bode plots of phase angle and bode plots of total impedance versus frequency of (a)  $\text{Co}_9\text{S}_8/\text{G-d}$  composite and (b) pure  $\text{Co}_9\text{S}_8$ .

Bode plots of phase angle and bode plots of total impedance versus frequency are shown in Fig. 17. The phase angle of both electrodes were observed to be close to  $-60^\circ$  in low frequency, which indicates the capacitor behavior of electrodes. The amplitude of the total impedance versus frequency plots show three slopes at different frequency regions. The slope of both electrodes showed large value at low frequency region (1-0.01 Hz), which is probably due to the resistive components of the interfaces. The present study emphasizes the combination of both resistive and capacitive behavior at high frequency region ( $10^5$ -100 Hz) and only capacitive behavior at mid frequency region (100-1 Hz). Compared with pure  $\text{Co}_9\text{S}_8$ , the slope value of  $\text{Co}_9\text{S}_8/\text{G-d}$  electrode at both mid and high frequencies region was small, which evidences for more capacitive behavior of composite than  $\text{Co}_9\text{S}_8$  electrode.

“Knee frequency ( $f_{knee}$ )” is another important characteristic of energy storage devices. It refers to the transition point between the high frequency and low frequency component, which is a measure of the power capability of capacitors. The electrochemical capacitor can be charged more rapidly and high power could be achieved with higher knee frequency.<sup>73</sup> A high knee frequency of  $4.542 \times 10^{-5}$  Hz have been obtained for  $\text{Co}_9\text{S}_8/\text{G-d}$  electrode, which indicated that the developed composite is a promising electrode material for supercapacitor applications.

## Conclusions

In summary, a rapid, low cost route for the synthesis of the  $\text{Co}_9\text{S}_8$  nanoflakes/ graphene composites have been developed and a systematic investigation was carried out on the effects of graphene loading on supercapacitor performance of the composite materials. The composite electrode showed enhanced specific capacitance than pure  $\text{Co}_9\text{S}_8$ , which is due to contribution of high electrical conductivity and pseudocapacitance of graphene and  $\text{Co}_9\text{S}_8$ . A maximum specific capacitance of 808 F/g was obtained for  $\text{Co}_9\text{S}_8/\text{G}$  composites with 9 wt% graphene loading. The galvanostatic charge/discharge measurement shows the excellent cyclic stability of both electrodes. The EIS measurement showed that



the prepared electrode materials has low impedance and hence such hybrid Co<sub>9</sub>S<sub>8</sub>/G composite electrodes are promising materials with potential applications in high performance supercapacitors.

## Acknowledgements

This work was carried out with the help of VIT management through research scholarship. This work was also conducted under the framework of Research and Development Program of the Korea Institute of Energy Research (KIER) (B3-2467-07).

## Notes and references

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Electronic Supplementary Information (ESI) available: [Further details of Graphene TGA graph, SEM mapping images and galvanostatic charge/discharge behaviour of composite and pure Co<sub>9</sub>S<sub>8</sub>]. See DOI: 10.1039/b000000x/

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