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La₂O₃ spherical catalyst.

Communication

Monodisperse Sr-La₂O₃ hybrid nanofibers for oxidative coupling of methane to synthesize C₂ hydrocarbons

Jianjun Song,^{*a,b*} Yongnan Sun,^{*a,c*} Rongbin Ba,^{*a,b*} Shangshang Huang,^{*a,b*} Yonghui Zhao,^{*a*} Jun Zhang,^{*a*} Yuhan Sun^{*a,**} and Yan Zhu^{*a,**}

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The synergistic effects from combinations of each component's functionality in hybrid Sr-La₂O₃ nanofibers brought about an improved catalytic behaviour for oxidative ¹⁰ coupling of methane carried out under high temperature, which cannot be achieved over the conventional Sr doped

Hybrid nanocatalysts with different component arranged in a controlled structure are widely used in industry since they can ¹⁵ provide entirely novel activity and selectivity for many important chemical reactions via the coupling between components within the hybrids.¹⁻³ This class of hybrid nanomaterials combines dissimilar materials into a single multifunctional fashion, and

- facilitates the electron or charge and energy flow between ²⁰ constituent hybrids.⁴⁻⁶ Different functionalities from each component of hybrids can be integrated to bring about a synergistic effect and new functionality, so the hybrid nanocatalysts are desirable for catalytic reaction.^{7,8} Progress in nanomaterial synthesis has made it possible for synthetic ²⁵ strategies of hybrid nanomaterials, such as metal oxide-
- metallic,^{6,9} semiconductor-metallic¹⁰ and semiconductor-magnetic hybrid nanomaterials.¹¹ However, most of these hybrid nanomaterials are used under mild reaction conditions and it remains unclear whether hybrid nanomaterials are stable when
- ³⁰ they are used as catalysts for the high temperature reactions since one component as a dopant is very mobile and easy to remove from the hybrids under high temperature conditions.¹² Especially, direct conversion of CH₄ to form chemicals such oxidative coupling of methane (OCM) is difficult succeeded at mild ³⁵ conditions, and the activation of CH₄ to methyl radical by the
- abstraction of H-atom at the aid of O_2 and the coupling of methyl radicals to form C_2 hydrocarbons (ethane/ethene) is carried out

^a CAS Key Laboratory of Low-Carbon Conversion Science and Engineering, Shanghai Advanced Research Institute, Chinese Academy of Sciences, Shanghai 201210, China. E-mail: zhuy@sari.ac.cn; sunyh@sari.ac.cn

^bUniversity of Chinese Academy of Sciences, Beijing 100049, China. ^CDepartment of Materials and Science and Engineering, Shanghai University, shanghai, 200444, china. gh of he in on gal-Meters O₃ he, of he at O₃ he, of he at O₃ he, ity, at ald me Monoscole Accepted Manuscript

under high temperature ($\sim 800 \, {}^{\circ}C$).¹²⁻¹⁵ The severe high temperature condition could lead to the complete oxidization of ⁴⁰ methane to produce CO and CO₂, and moreover suppress the selectivity of C₂ hydrocarbons.^{16,17} Thanks to the discovery in previous investigation that La₂O₃ nanorods exhibited low ignition temperature (450 $\,^{\circ}C$) of CH₄ compared with corresponding nanoparticles,¹⁸ it is now possible for precise synthesis of metal-⁴⁵ inserted La₂O₃ hybrid nanofibers by a facile route for OCM reaction.

In the present work, we report, for the first time, the targetspecific synthesis of monodisperse hybrid $Sr-La_2O_3$ nanofibers through an incipient wetness impregnation route. $Sr-La_2O_3$ ⁵⁰ nanofibers acted as a catalyst for oxidative coupling of methane, unprecedented in the literatures, the initial ignition temperature of methane is about 500 °C, and the conversion of methane and the selectivity of C_2 hydrocarbons could be up to ~35% and 47% at 500 °C (Figure 1), respectively. Within these hybrid $Sr-La_2O_3$

⁵⁵ nanofibers, 8.6 wt% Sr-La₂O₃ nanofibers (the content of Sr is 8.6 wt%) exhibited the highest methane conversion and C₂ selectivity, in particular, 20% yield towards C₂ hydrocarbons was obtained at 650 °C (Figure 1b), which cannot be achieved over conventional Sr doped La₂O₃ spherical catalysts (Figure S1). More ethene and state of the selection of the select

⁶⁰ ethane can be produced on the Sr-La₂O₃ nanofiber than pure La₂O₃ (Figure 1c). The complete oxdiation products like CO and CO₂ can be suppressed on Sr-La₂O₃ nanofibers (Figure 1d). Other metals such as Ba, Ca and K modified La₂O₃ nanofibers can not exhibit superior catalysis to Sr-La₂O₃ nanofibers for OCM
 ⁶⁵ reactions (Figure S2). Additionally, the hybrid Sr-La₂O₃

nanofibers showed a good stability during the oxidative coupling of methane at temperatures tested, which can be inferred from the stability test under different reaction conditions. From Figure S3, it was found that the conversion of CH_4 and yield of C_2 decreased

⁷⁰ slightly during the 100 hours test. The improved activity and selectivity over Sr-La₂O₃ nanofibers are attributed to the synergistic effect of material properties of individual Sr and La₂O₃ optimized independently, such Sr might tailor the lattice structure and electronic structure of La₂O₃, change the oxygen ⁷⁵ species and basic sites on the surface of La₂O₃, and lower the reaction energy of CH₄ on the surface of hybrid catalysts, which have been revealed by our studies.

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Figure 1. Catalytic performances of pure La2O3 and hybrid Sr-La2O3 nanofibers for OCM reaction. Upper panel and lower panel of c are C2H6 5 selectivity and C2H4 selectivity. Upper panel and lower panel of d are CO selectivity and CO2 selectivity.

- We now turn to the studies to find the reasons why the hybrid Sr-La₂O₃ nanofibers have an improved catalytic 10 performance and good stability, compared to pure La₂O₃ nanofibers. Firstly, TEM, XRD and theoretic studies confirmed that the lattice structure of La₂O₃ can be indeed tailored by the Sr insert. Pure La2O3 nanofibers were 20-30 nm in diameter and several micrometers in length with smooth surfaces (Figure 2a). 15 Herein, 8.6 wt% Sr-La₂O₃ nanofibers as an example, hybrid Sr-
- La2O3 nanofibers with rough surface were in a diameter of 40-50 nm with hundreds nanometers in length (Figure 2b and 3a). The corresponding EDS analysis shown in the inset of Figure 2b, was a direct evidence to let us believe the existence of Sr in hybrid
- 20 nanofibers. The homogeneous distribution of Sr in hybrid nanofibers was presented in the element mapping (Figure 3d), which clearly revealed the existence and distribution of Sr, La and O elements in hybrid nanofibers, wherein different colors represented different elements (Figure 3b-d). Unlike conventional
- 25 Sr doped La₂O₃ spherical catalyst, Sr easily leaking from the doped catalyst under high temperature conditions, hybrid Sr-La₂O₃ catalysts are much stable during the OCM reaction and Sr element was still homogeneous distributed in hybrid nanofibers after reaction, as shown in Figure 3e-h. Especially, TG-MS data
- 30 of spent Sr-La₂O₃ catalyst indicated that no coke was formed on hybrid Sr-La₂O₃ nanofibers, verified by no weight loss measured from spent catalyst (Figure S4). It is of paramount importance to attain the catalytic materials which can resist sintering and coking at high temperature.



Figure 2. SEM images of (a) pure La₂O₃ and (b) Sr-La₂O₃ nanofibers, and the inset is corresponding EDS data. HRTEM images of (c) pure La2O3 nanofibers 40 and (d) Sr-La2O3 nanofibers.



Figure 3. (a) TEM image of fresh Sr-La₂O₃ nanofibers and (b, c, d) the 45 corresponding element maps for La, O and Sr. (e) TEM image of spent Sr-La2O3 nanofibers and (f, g, h) the corresponding element maps for La, O and and Sr.

Moreover, the introduction of Sr widened the lattice fringes 50 of La₂O₃ nanofibers. In Figure 2c, the crystal lattice fringes of La₂O₃ are 0.19, 0.30, 0.31 and 0.33 nm apart, which agree with the d value of (110), (101), (002) and (100) lattice planes of hexagonal La₂O₃ with growth along the (101) direction. For Sr-La₂O₃ nanofibers, the crystal lattice fringes of La₂O₃ are 0.20, 55 0.30, 0.32 and 0.34 nm apart, which are corresponding to the dvalue of (110), (101), (002) and (100) lattice planes of hexagonal La₂O₃. The lattice fringes of (110), (101) and (002) lattice planes broaden after introducing Sr, which is due to the ionic larger radii of Sr (0.118Å) than La ions (0.1032Å).

0	• SrCO ₃ • La ₂ O ₂ CO ₃	
	12.1 wt% Sr-La _. O ₃	catalyst
Î	8.6 wt% Sr-La ₂ O ₃	La ₂ O ₃
ity (a.	6.0 wt% Sr-La ₂ O ₃	0.8 wt % Sr/ La ₂ O ₃
ntens	itile in 3.4 wt% Sr-La_O3	3.4 wt % Sr/ La ₂ O ₂
-	• 10.8 wt% Sr-La ₂ O ₃	6.0 wt % Sr/ La ₂ O ₃
	100 100 110 103 La_2O_3	8.6 wt % Sr/ La2O3
2	0 30 40 50 60 70 80 2 0 (degree)	12.1 wt % Sr/ La ₂ O ₃

catalyst	a (Å)	b (Å)	c (Å)
La ₂ O ₃	0.3930	0.3930	0.6109
0.8 wt % Sr/ La ₂ O ₃	0.3930	0.3930	0.6109
3.4 wt % Sr/ La ₂ O ₃	0.3934	0.3934	0.6117
6.0 wt % Sr/ La ₂ O ₃	0.3934	0.3934	0.6117
8.6 wt % Sr/ La ₂ O ₃	0.3934	0.3934	0.6117
12.1 wt % Sr/ La2O3	0.3934	0.3934	0.6117

Figure 4. XRD patterns of La₂O₃ and Sr-La₂O₃ nanofibers (left panel) and lattice parameters (a, b and c) of La₂O₃ and Sr-La₂O₃ nanofibers (right panel).

XRD studies further corroborated the change of space lattice of La₂O₃ nanofibers via the insert of Sr ions into the crystal lattice of La₂O₃ (right panel of Figure 4). Without Sr inserted, the lattice parameters of hexagonal La₂O₃ is 0.3930 Å (a), 0.3930 Å (b) and 0.6109 Å (c), while hybrid Sr-La₂O₃, in particular the Sr content $_{70}$ is equal to or more than 3.4 wt%, the lattice parameters of La₂O₃ are broadened such 0.3934 Å (a), 0.3934 Å (b) and 0.6117 Å (c),

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respectively. Interestingly, XRD patterns of pure La_2O_3 and hybrid Sr-La₂O₃ nanofibers showed the similar diffraction peaks at $2\theta = 26.11^\circ$, 29.96°, 46.08°, and 52.13°, corresponding to the characteristic diffractions of the (100), (101), (110), and (103)

- ⁵ planes of La₂O₃ (JCPDS 05-0602). It is noted that black rhombs in the XRD of Figure 4 referred to the La₂O₂CO₃ phases, which cannot be avoided for La₂O₃ to transform into La₂O₂CO₃ in moist air. Red dots in the left panel of Figure 4 stood for the SrCO₃ phases, which might be generated during the calcination of
- ¹⁰ Sr(NO₃)₂-La(OH)₃ in the air (experimentals in Supporting Information). Thus these catalysts must be pretreated at 800°C under N₂ for 2 h before OCM test to remove these impurities. There is no indication of Sr or related oxide phases, suggesting the single component of Sr-La₂O₃ themselves rather than the mix ¹⁵ of Sr and La₂O₃, except for small amount of SrCO₃ and
- $La_2O_2CO_3$. After reaction, the spent Sr-La_2O_3 showed similar XRD diffractions of fresh catalysts (Figure S5).



Figure 5. The side and top views of (a) pure La₂O₃(101) surfaces and (b) 30 hybrid Sr-La₂O₃(101) surfaces.

The change of La₂O₃ structure caused via inserting Sr was also confirmed by DFT + U method (the periodic plane wave DFT implementation in the Vienna ab initio simulation program ³⁵ (VASP) code ^{19,20}). For the La f orbitals, a value of 7.5 eV for the U parameter for all calculations was presented herein as suggested in the literature.²¹ The side and the top views of the pure and hybrid Sr-La₂O₃ (101) surfaces, and the optimized bond length for the top layer atoms (eg. La-O and O-Sr) are shown in ⁴⁰ Figure 5. The optimized La atom in the surface layer of the La₂O₃

- (101) face is coordinated with four O atoms, with La-O distances of 2.23, 2.23, 2.34 and 2.32 Å (Figure 5a). With Sr introduction, there are corresponding four Sr-O, with Sr-O distances of 2.28, 2.28, 2.46 and 2.46 Å (Figure 5b). From the values, the
- ⁴⁵ expanding in the structure caused by Sr is obvious and such change in the structure should result in a difference in the activity of hybrid Sr-La₂O₃ and pure La₂O₃.



50 Figure 6. CO2-TPD profiles of pure La2O3 and hybrid Sr-La2O3 nanofibers.

Besides the surface structure of La₂O₃ tailored by Sr, the surface species and surface basic sites on the Sr-La₂O₃ face made such a difference to that of pure La₂O₃. The characterization from 55 CO₂-TPD allowed us to demonstrate the nature of surface strong basic site on Sr-La₂O₃ nanofibers (Figure 6). There are three types of basic sites: the weak, medium and strong basic site, which are associated with desorption of CO₂ at the temperatures 100 °C, 300-600 °C and >600 °C, respectively. Among the three 60 distinct basic sites, the strong basic sites are crucial for the OCM process carried out above 600 °C. The addition of Sr, influencing sharply the strength and quantity of base of La₂O₃, resulted in the creation of more strong basic sites compared with intermediate basic sites. For pure La₂O₃, most of CO₂ was desorbed at 300-600 65 °C. For 0.8 wt% Sr-La₂O₃, partial of CO₂ was desorbed at >600 °C, and most CO₂ was desorbed at 300-600 °C. When the Sr content exceeded 3.4 wt% within hybrids, almost all of CO₂ was desorbed at 700-800 °C. It suggests that the surfaces of La2O3 and 0.8 wt% Sr-La₂O₃ are dominated by medium basic site, whereas 70 the surface of Sr-La₂O₃ with Sr content above 3.4 wt% are dominated by strong basic site. The distinct difference of basic sites on the surfaces is one of reasons for improved catalysis of hybrid Sr-La₂O₃ on OCM reaction.

Table 1. Curve-Fitting results from XPS Data

Catalysts	O 1s binding energies (eV)			$(0^{-1}, 0^{-1})/(0^{2})$	
	O ²⁻	O ⁻	CO32-	O_2^-	$(0+0_2)/0$
La ₂ O ₃	529.1	531.5	532.7	534.6	5.1
0.8 wt% Sr-La ₂ O ₃	529.1	531.0	532.1	533.6	3.4
3.4 wt% Sr-La ₂ O ₃	528.7	531.1	531.9	534.0	4.3
6.0 wt% Sr-La ₂ O ₃	528.7	531.1	532.0	533.9	4.6
8.6 wt% Sr-La2O3	529.1	531.6	533.0	534.6	7.5
12.1 wt% Sr-La ₂ O ₃	528.9	531.1	532.1	533.4	5.1

The roles played by surface oxygen species are, in addition to surface basic sites, investigated by XPS analyses. The O1s spectra of samples can be deconvoluted into four peaks ⁸⁰ corresponding to four different oxygen species: superoxide O₂; carbonate $CO_3^{2^-}$; peroxide ions O⁻; and lattice oxygen O²⁻, seen in Table 1. The electron deficient species such O_2^- and O^- on the surface of catalysts, as is well-known, are effective for the C₂ selectivity, while the lattice oxygen O2- is responsible for 85 complete oxidation to form CO and CO₂.^{18,22} The ratio of the peak intensities of the electron deficient species to lattice oxygen, $(O_2^{-}+O^{-})/O^{2^{-}}$, can be calculated using the 80% Gaussian-20% Lorentzian peak shape (Figure S6). From the ratio values of (O $+O_2$) to O² shown in Table 2, 8.6 wt % Sr-La₂O₃ nanofibers have 90 the highest ratio value (7.5) and achieved the highest C_2 selectivity among the hybrid catalysts, which is in accordance with literatures.18,22



Figure 7. (a) XPS La $3d_{5/2}$ spectra of La₂O₃ and 8.6 wt% Sr-La₂O₃ nanofibers and (b) Sr 3d XPS spectra of 8.6 wt% Sr-La₂O₃.

- Figure 7 (curve a) showed the typical La $3d_{5/2}$ of pure La₂O₃ and 8.6 wt% Sr-La₂O₃ nanofibers. After the introduction of Sr, the binding energy peak located at 834.3 eV moved to a lower binding energy (833.1 eV), and the energy peak at 837.4 eV shifted to a higher binding energy (837.5 eV), revealing that the
- ¹⁰ electronic properties of La was changed by Sr insert. Figure 7b showed that the Sr 3d spectra could be divided into two peaks at the binding energy of 132.1 eV and 133.9 eV, which are corresponding to SrO and SrCO₃,²³ suggesting that Sr occurred on the surface of Sr-La₂O₃ in the form of SrO and SrCO₃.
- The comparison of the structure and energetics of the associative and dissociative adsorption of CH_4 on pure and hybrid Sr-La₂O₃ could be made through DFT study, since the reaction pathway of CH_4 on the two catalysts is difficultly traced. The optimized structures for the CH_4 , CH_3 , H and co-adsorbed
- ²⁰ CH₃+H on Sr-La₂O₃(101) surfaces were shown in Figure 8, and distances for H-O and C-O were listed in Table 3. In Figure 8, CH₃, H are preferable to be adsorbed at the three bonded oxygen near Sr atom. Corresponding optimized structures for the CH₄, CH₃, H and co-adsorbed CH₃+H on pure La₂O₃(101) surfaces can
- $_{25}$ get from our former work.¹⁸ The adsorption energies for all the species on the pure La₂O₃ surface are all weaker than those of Sr-La₂O₃ surface (-0.11 vs. -0.12 eV for CH₄, -2.34 vs. -3.54 eV for CH₃, and -3.40 vs. -4.67 eV for H). The interatomic distances between O and H, O and C on the Sr-La₂O₃(101) surfaces are all
- ³⁰ shorter than the pure La₂O₃ surface (2.23 vs. 2.33 Å for CH₄, 0.96 vs. 0.97 Å for H, and 1.39 vs. 1.40 Å for CH₃). The same trends are observed in the CH₃ + H co-adsorption, with adsorption energies of -4.54 eV and -6.97 eV, respectively. Thus, the corresponding reaction energies of CH₄ activation on (101) of
- ³⁵ La₂O₃ and Sr-La₂O₃ are 0.30 and -2.13 eV, respectively. According to the Brønsted–Evans–Polanyi relationship, such significant difference (2.43 eV) in reaction energy suggests that methane activation is preferred over the (101) surface of Sr-La₂O₃, even when taking the actual reaction conditions into account.





Figure 8. The optimized structures for the CH₄, CH₃, H and co-adsorbed

CH3+H on Sr-La2O3(101) surfaces.

Table 2. A	dsorption energie	s, structure parame	eters of CH ₄ , CH ₃ , l	H and co-
45 adsorbed C	H ₃ +H on pure La	$_{2}O_{3}$ (101) surfaces	and Sr-La ₂ O ₃ (101) surfaces.

Adsorbates	Surface	Eads/eV	d _{O-H} (Å)	d _{C-H} (Å)
CH	La ₂ O ₃ (101)	-0.11		2.33
CH_4	Sr-La ₂ O ₃ (101)	-0.12		2.23
CH ₃	La ₂ O ₃ (101)	-2.34		1.40
	Sr-La ₂ O ₃ (101)	-3.54		1.39
Н	La ₂ O ₃ (101)	-3.47	0.97	
	Sr-La ₂ O ₃ (101)	-4.67	0.96	
CIL II	La ₂ O ₃ (101)	-4.54	0.98	1.41
CH_3+H	Sr-La ₂ O ₃ (101)	-6.97	0.97	1.39

Conclusions

In summary, hybrid Sr-La₂O₃ nanofibers enhanced significantly conversion of CH₄ and the selectivity of C₂ hydrocarbons. Especially, for 8.6 wt% Sr-La₂O₃ hybrid catalyst, ⁵⁰ the initial ignition temperature of methane was about 500°C, and methane conversion and C_2 selectivity were up to ${\sim}35\%$ and 47% at 500°C. The synergistic effects of material properties of individual Sr and La2O3 optimized independently, i.e. lattice structure and electronic properties of La₂O₃ are tailored by Sr, 55 surface basic sites of La₂O₃ are strengthened by Sr, oxygen species on the surface of La₂O₃ are modified by Sr, and methane activation is preferred over Sr-La₂O₃ surface, are believed to be primarily responsible for the improved activity and selectivity over Sr-La₂O₃ nanofibers. Especially, Sr-La₂O₃ nanofibers have 60 the ability to resist sintering and further avoid the loss of Sr, as well as no coke is formed on the surface of Sr-La₂O₃ catalysts during the high temperature reaction. As we have seen, the Sr-La₂O₃ nanocatalysts may open an avenue for a new type hybrid catalyst with high activity and selectivity for oxidative coupling 65 of methane carried out at high temperature, and are promising in that they will provide new opportunities for studying the nature of hybrid catalysts for high temperature reaction.

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Notes and references

- 75 1. A. Azetsu, H. Koga, A. Isogai and T. Kitaoka, *Catalysts*, 2011, 1, 83-96.
 - F. Alosfur, M. H. H. Jumali, S. Radiman, N. J. Ridha, M. A. Yarmo and A. A. Umar, *Int. J. Electro. Sci.*, 2013, 8, 2977-2982.
- 3. Y. Lu, Y. Jiang, X. Gao, X. Wang and W. Chen, *J. Am. Chem. Soc.*, 2014, **136**, 11687-11697.
- L. L. Chng, N. Erathodiyil and J. Y. Ying, Acc. Chem. Res., 2013, 46, 1825-1837.
- M. N. Luwang, S. Chandra, D. Bahadur and K. Srivastava, J. Mater. Chem., 2012, 22, 3395-3403.
- 85 6. R. Ferrando, G. Rossi, F. Nita, G. Barcaro and A. Fortunelli, ACS Nano, 2008, 2, 1849-1856.

Nanoscale

- 7. J. Guo, H. Li, H. He, D. Chu and R. Chen, *J. Phys. Chem. C*, 2011, **115**, 8494-8502.
- S. Chen, R. Si, E. Taylor, J. Janzen and J. Chen, J. Phys. Chem. C, 2012, 116, 12969-12976.
- ⁵ 9. S. Liu, S.-Q. Bai, Y. Zheng, K. W. Shah and M.-Y. Han, *ChemCatChem*, 2012, **4**, 1462-1484.
- 10. K. Maeda, K. Sekizawa and O. Ishitani, *Chem. Commun.*, 2013, **49**, 10127-10129.
- M. Casavola, V. Grillo, E. Carlino, C. Giannini, F. Gozzo, E.
 Fernandez Pinel, M. Angel Garcia, L. Manna, R. Cingolani and P. Davide Cozzoli, *Nano Lett.*, 2007, 7, 1386-1395.
- S. Arndt, G. Laugel, S. Levchenko, R. Horn, M. Baerns, M. Scheffler, R. Schloegl and R. Schomaecker, *Catal. Rev.*, 2011, **53**, 424-514.
- 13. T. Ito, J. X. Wang, C. H. Lin and J. H. Lunsford, J. Am. Chem. Soc., 1985, **107**, 5062-5068.
- 14. J. E. France, A. Shamsi and M. Q. Ahsan, *Energy & Fuels*, 1988, 2, 235-236.
- 15. M. T. Xu and J. H. Lunsford, Catal. Lett., 1991, 11, 295-300.
- 16. J. H. Lunsford, Angew. Chem. Int. Ed., 1995, 34, 970-980.
- 20 17. J. L. Dubois and C. J. Cameron, Appl. Catal., 1990, 67, 49-71.
 - 18. P. Huang, Y. Zhao, J. Zhang, Y. Zhu and Y. Sun, *Nanoscale*, 2013, **5**, 10844-10848.
- 19. G. Kresse and J. Furthmuller, Phys. Rev. B, 1996, 54, 11169-11186.
- 20. G. Kresse and J. Hafner, *Phys. Rev. B*, 1994, **49**, 14251-14269.
- 25 21. B. Li and H. Metiu, J. Phys. Chem. C, 2010, 114, 12234-12244.
- 22. W. P. Ding, W. P. Ding, Y. Chen and X. C. Fu, *Catal. Lett.*, 1994, **23**, 69-78.
- 23. M. V. Bukhtiyarova, A. S. Ivanova, L. M. Plyasova, G. S. Litvak, V. A. Rogov, V. V. Kaichev, E. M. Slavinskaya, P. A. Kuznetsov and I. A.
- ³⁰ Polukhina, *Appl. Catal. A*, 2009, **357**, 193-205.