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**PAPER**

# Electrochemically induced Fenton reaction of few-layer MoS<sub>2</sub> nanosheets: Preparation of luminescent quantum dots via a transition of nanoporous morphology

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Electrochemically induced Fenton (electro-Fenton) reaction was used to the efficient and controllable preparation of hydroxyl radical, leading to the generation of luminescent quantum dots through the etching of as-exfoliated MoS<sub>2</sub> nanosheets. The observable morphologic changes of MoS<sub>2</sub> nanosheets in electro-Fenton reaction were monitored using transmission electron microscopy, indicating that the etching of MoS<sub>2</sub> nanosheets induced by hydroxyl radicals resulted in rapid homogeneous fracturing of the sheets into small dots via a transition of nanoporous morphology. The as-generated dots with the vertical dimensional thickness of ca. 0.7 nm and plane size of ca. 5 nm were demonstrated to be MoS<sub>2</sub> quantum dots (MoS<sub>2</sub>-QDs) and their photoluminescence properties were explored based on the quantum confinement, edge effect, and intrinsic characteristics. Moreover, the degree of etching and the concomitant porosity of MoS<sub>2</sub> nanosheets could be conveniently tuned by electro-Fenton reaction time and a new morphology, nanoporous MoS<sub>2</sub> nanosheets, was obtained, raising new issues concerning MoS<sub>2</sub> nanosheets applications in various significant areas.

## Introduction

Two-dimensional (2D) nanomaterials, derived from layered bulk crystals analogous to graphite, have aroused a great deal of consideration, because of their unusual properties associated with ultra-thin thickness and 2D morphology.<sup>1-4</sup> These developments have been spearheaded by researches of graphene, a 2D nanomaterial that is unique due to its combination of thermal, electronic, optical, and mechanical properties.<sup>5-10</sup> With the penetration of graphene, other 2D materials with a nano-layered structure such as hexagonal boron nitride, transition metal dichalcogenides, and metal halides have also attracted a large amount of interest.<sup>11,15</sup> Especially, MoS<sub>2</sub> nanosheets, 2D transition metal dichalcogenides, were obtained through efficient exfoliation of three-dimensional crystals and breaking the van der Waals interactions between nanosheets.<sup>16-19</sup> Owing to the specific 2D confinement of electron motion and absence of interlayer perturbation, the MoS<sub>2</sub> nanosheets possess a direct band gap and show some remarkable properties offering new opportunities in various areas, such as sensors,<sup>18-22</sup> single-layer transistors,<sup>17</sup> integrated circuits,<sup>23</sup> phototransistors,<sup>24</sup> nanometer thick photovoltaics,<sup>25</sup> and materials of batteries.<sup>26,27</sup> Recently, MoS<sub>2</sub> nanosheets were found to be promising materials for hydrogenation catalysis, attributed to active sites located along the edges of their layered crystal structure.<sup>28-31</sup> To achieve the potential applications of MoS<sub>2</sub> nanosheets limited by the insufficient active sites, defect-rich MoS<sub>2</sub> nanosheets,<sup>29</sup> oxygen-incorporated MoS<sub>2</sub> nanosheets,<sup>30</sup> and metallic MoS<sub>2</sub> nanosheets<sup>31</sup>

representing some novel morphologies of MoS<sub>2</sub> nanosheets were presented.

Following the research of 2D and one-dimensional materials, zero-dimensional materials obtained more and more consideration. When compared to graphene sheets, zero-dimensional graphene quantum dots (GQDs) possess strong quantum confinement and edge effects, which induce new physical properties.<sup>32</sup> Due to their excellent properties and ongoing applications, the preparation of GQDs has been highly demanded and several strategies have been developed so far, such as hydrothermal route,<sup>33</sup> wave-assisted hydrothermal method,<sup>34</sup> ultrasonication breaking,<sup>35</sup> enzymatic oxidations,<sup>36</sup> steam etching,<sup>37</sup> and size fractionation.<sup>38</sup> Nevertheless, the mass scale manufacture of GQDs is limited by the harsh conditions, time consumption, low yield, and damage of aromatic carbon framework. To develop effective strategies for GQDs preparation, Fenton reaction with the assistance of ultraviolet light irradiation (photo-Fenton reaction) was proposed based on the etching of graphene oxide (GO).<sup>39</sup> However, regardless of conventional Fenton and photo-Fenton reaction, the initial addition of numerous Fe<sup>2+</sup> and H<sub>2</sub>O<sub>2</sub> makes an over-fast reaction occur in the beginning. Following the reaction, the consumption of Fenton reagent makes the rate of reaction down, which is detrimental to control the reaction rate and effectively prepare GQDs. Decades ago, electrochemically induced Fenton (electro-Fenton) reaction, an important advanced oxidation technology, was proposed and mainly used to the available degradation of pollutants in environment via generating highly powerful chemical oxidants hydroxyl radicals.<sup>40</sup> Electro-Fenton process is an attractive

method owing to its high effectiveness, fast treatment rate, and environmental compatibility in which hydrogen peroxide is generated in situ and the addition of ferrous ions allows an enhancement of the oxidation activity according to the following Fenton reaction through formation of  $\bullet\text{OH}$ . Meanwhile, the ferrous ions are re-generated at the cathode, making the hydroxyl radicals be produced in a catalytic and controlled mode, tuned by an electrochemical process.<sup>40</sup>

So far, most attention is focused on the preparation, application, and property studies of  $\text{MoS}_2$  nanosheets, leading to the relevant exploration of  $\text{MoS}_2$  quantum dots ( $\text{MoS}_2$ -QDs) be ignored. Zero-dimensional  $\text{MoS}_2$ -QDs, owning the smaller plane size than  $\text{MoS}_2$  nanosheets, possess strong quantum confinement and edge effects, which can induce numerous unique and outstanding properties. With the successful preparation of graphene quantum dots (GQDs) from the morphologic change of 2D graphene sheets,<sup>33-39</sup>  $\text{MoS}_2$  nanosheets owing the specific 2D structure similar to graphene were also taken into consideration to prepare  $\text{MoS}_2$ -QDs. Herein, a simple, cost-effective, efficient, and controllable method for the production of hydroxyl radicals was used to cut GO into GQDs based on electro-Fenton reaction. Meanwhile, the reaction between  $\text{MoS}_2$  nanosheets and hydroxyl radicals was explored, indicating the generation of zero-dimensional  $\text{MoS}_2$ -QDs from the etching of  $\text{MoS}_2$  nanosheets (Scheme 1). The whole process of reaction was monitored by transmission electron microscopy (TEM) and a new  $\text{MoS}_2$  nanosheets morphology, nanoporous  $\text{MoS}_2$  nanosheets, was obtained via tuning the reaction time to control the degree of etching and concomitant porosity. Compared to  $\text{MoS}_2$  nanosheets, both  $\text{MoS}_2$ -QDs and nanoporous  $\text{MoS}_2$  nanosheets can provide much more active sites at the edge of plane, taken into a great deal of consideration for  $\text{MoS}_2$  nanomaterials applications in various areas.

## Experimental Section

### Materials

Molybdenum(IV) sulfide ( $\text{MoS}_2$  crystalline powder, < 2  $\mu\text{m}$ , 99%) was purchased from Sigma-Aldrich Co., USA. Graphite powder (99.85%) was obtained from Shanghai Huayuan Chemical Engineering Co., Ltd., Shanghai, China. Sodium cholate (98%) and iron(II) sulfate heptahydrate ( $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$ )

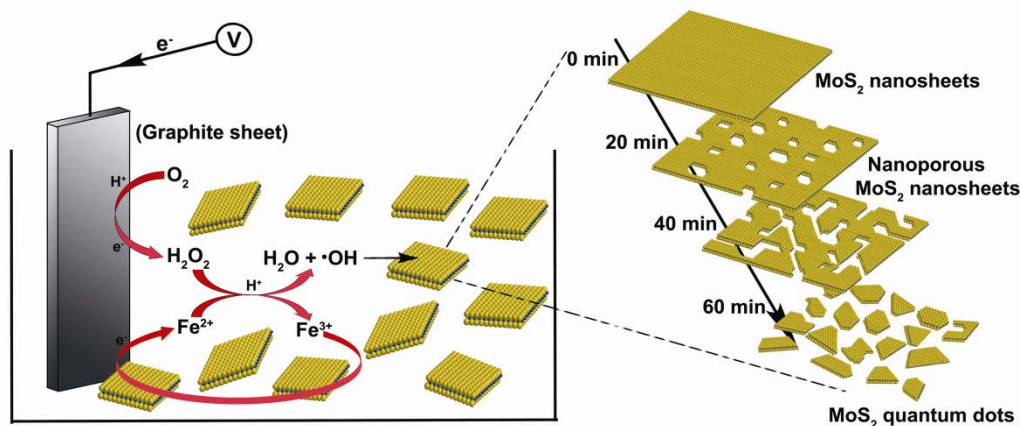
were purchased from Aladdin Chemistry Co., Ltd., Shanghai, China. Other chemicals employed in this work were of analytical reagent grade and purchased from Kelong Chemical Reagent Co., Ltd., Chengdu, China.

### Experimental instrumentations

All electrochemical processes were conducted using a CHI 660B electrochemical workstation (CHEN HUA Instruments Co., China). A three-electrode system consisting of a 5  $\text{cm}^2$  graphite sheet working electrode, a saturated calomel electrode (SCE), and an auxiliary electrode made of platinum was adopted. All potentials were given with respect to the SCE. The KQ-250B ultrasonic bath (250 W, Kun Shan Ultrasonic Instruments Co., Ltd, China) was adopted to exfoliate  $\text{MoS}_2$  powder. TEM, high resolution TEM (HRTEM), and high-angle annular dark-field scanning TEM (HAADF-STEM) measurements were performed on a Tecnai G2 F20 S-TWIN transmission electron microscope (FEI, USA) operated at 200 kV. The samples were prepared by dropping the aqueous suspension of GQDs and  $\text{MoS}_2$  nanomaterials on the carbon grids and drying under ambient condition. Scanning electron microscope (SEM) images were obtained using an S-4800 field emission scanning electron microscope (Hitachi, Japan). Atomic force microscope (AFM) data were obtained on a Dimension Icon atomic force microscope (Bruker, Germany). X-ray diffraction (XRD) patterns were obtained with a D8 DISCOVER X-ray diffractometer (Bruker, Germany). UV-visible spectra were recorded on a UV-2450 UV-vis spectrophotometer (Shimadzu, Japan) at room temperature. The photoluminescence (PL) and photoluminescence excitation (PLE) spectra were recorded on an F-2700 fluorescence spectrophotometer (Hitachi, Japan) with Xe lamp as an excitation source; slit: 10 nm; PMT voltage: 400 V. The centrifugation was carried out using a TGL-16M high-speed refrigerated centrifuge (Xiangyi, China).

### Preparation of $\text{MoS}_2$ nanosheets and graphene oxide

$\text{MoS}_2$  nanosheets were prepared through the sonication-assisted exfoliation of  $\text{MoS}_2$  crystal in aqueous surfactant solution.<sup>41</sup> A mixed water solution, containing 5  $\text{mg mL}^{-1}$   $\text{MoS}_2$  powder and 1.5  $\text{mg mL}^{-1}$  sodium cholate, was sonicated (KQ-250B 250 W) at normal temperature for 10 h, making the preparation of black dispersion. The dispersion was centrifuged at 3000 rpm for 30



**Scheme 1.** Schematic illustration of the preparation of nanoporous  $\text{MoS}_2$  nanosheets and  $\text{MoS}_2$  quantum dots from electro-Fenton process.

min, followed by the separation of yellow-green supernatant to remove bulk MoS<sub>2</sub> powder. Furthermore, the separated supernatant was centrifuged at 12000 rpm for 30 min, making as-exfoliated MoS<sub>2</sub> nanosheets be obtained. To remove sodium cholate, the MoS<sub>2</sub> nanosheets were dispersed in ultrapure water with the assistance of sonication. Similarly, the regenerated dispersion was centrifuged at 12000 rpm for 30 min, followed by the collection of sediments to complete the washing process. The washing process was repeated for other two times to remove sodium cholate completely. Ultimately, the sediments were dispersed in a certain amount of ultrapure water to prepare uniform MoS<sub>2</sub> nanosheets dispersion. Graphene oxide was synthesized by the oxidation of graphite powder according to the hummers method and characterized in our pervious work.<sup>42</sup>

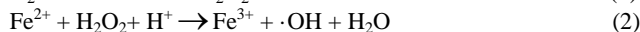
### 15 Electro-Fenton reaction

The electro-Fenton reaction of MoS<sub>2</sub> nanosheets (or graphene oxide) was conducted in an electrolytic cell equipped with an electrochemical workstation. A three-electrode system was adopted to connect the electrolytic cell with electrochemical workstation. A portion of 50 mL mixed solution containing 0.3 mg mL<sup>-1</sup> exfoliated MoS<sub>2</sub> nanosheets (or graphene oxide) and 0.05 M FeSO<sub>4</sub> was added into the electrolytic cell. The initial pH of the mixture was adjusted to 3 by the dropwise addition of 0.5 M H<sub>2</sub>SO<sub>4</sub> solution and it remained almost constant over the whole reaction. The potential of -0.5 V was applied and continuous saturation of the solution by O<sub>2</sub> at atmospheric pressure was ensured through bubbling of compressed air. Solutions were continuously stirred using a magnetic bar (500 rpm) to allow mass transfer. After various reaction times, the production was collected and dialyzed (Mw = 3500) in H<sub>2</sub>SO<sub>4</sub> solution (pH = 3) for one day to remove iron ions and some other impurities, avoiding the production of sediments induced by hydrolysis of iron, and in ultrapure water for another day to remove H<sup>+</sup> and SO<sub>4</sub><sup>2-</sup> ions.

## 35 Results and Discussion

### Electro-Fenton process of graphene oxide

In electro-Fenton process, the electrochemical generation of H<sub>2</sub>O<sub>2</sub> takes place at the cathode based on Equation 1 and with the addition of a certain amount of ferrous ions, the Fenton reagent thus formed leads to the formation of •OH according to Equation 2.



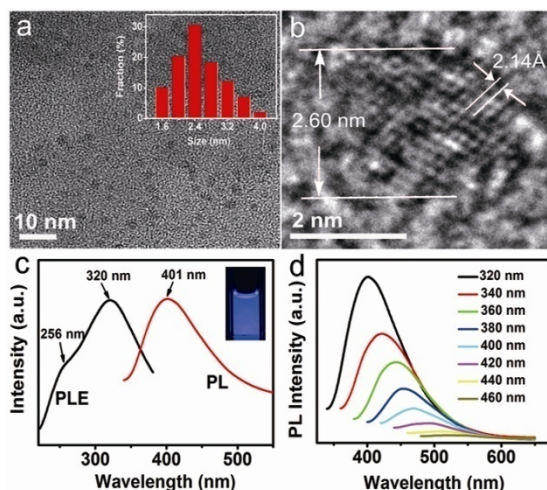
Meanwhile, O<sub>2</sub> is fed into the electrochemical system making the concentration of H<sub>2</sub>O<sub>2</sub> keep almost constant (Fig. S1). The regeneration of Fe<sup>2+</sup> can occur by a direct cathodic reaction (Equation 3) under the conditions of electro-generation of H<sub>2</sub>O<sub>2</sub>, and contribute to further production of a great number of hydroxyl radicals.



Compared to conventional Fenton and photo-Fenton reaction, electro-Fenton reaction can be conveniently controlled by tuning the electrochemical process, making the generation of hydroxyl radicals more efficient and controllable. The controllability of electro-Fenton process was confirmed through conducting the experiment about the degradation of rhodamine B (Fig. S2).

Meanwhile, when a specific condition for electro-Fenton process was confirmed, the impact of •OH on the substrate only depended on the reaction time.

60 Hydroxyl radicals-mediated oxidation of aromatic molecules has been established before. The initial attack of hydroxyl radical produces an •OH adduct or undergoes electron transfer to produce cation radicals and continued attack of hydroxyl radicals causes complete oxidation of aromatic molecules to CO<sub>2</sub> and water. Graphene and GO sheets could be considered as super aromatic molecules and the reaction between graphene sheets and hydroxyl radicals has been discussed.<sup>39,43</sup> The hydroxyl radicals can make an impact on aromatic molecules and further break up parts of the hexatomic ring structure of GO sheets. On the basis of the exploration about electro-Fenton process, the route that using electro-Fenton process to prepare GQDs from GO was proposed. Under the optimal conditions (Fig. S3), the electro-Fenton process of GO was conducted and the progresses were monitored using AFM. From the results (Fig. S4), it was obviously observed that bulk GO sheets turned into a great deal of small dots possessing a vertical dimensional thickness of ca. 0.8 nm after reaction for 60 min, which were demonstrated to be GQDs with great photoluminescence properties. As shown in the TEM image (Fig. 1a), the plane sizes of as-prepared GQDs are uniform, ranging from 1.6 to 4.0 nm with the average value of approximately 2.53 nm. The lattice spacing of the GQDs (plane size: ca. 2.60 nm) in the HRTEM image (Fig. 1b) is ca. 2.14 Å, which is similar to the hexagonal pattern of graphene with d<sub>1100</sub>.<sup>44</sup> It was confirmed that hydroxyl radicals generated from electro-Fenton process could break the 2D structure of GO and played a positive impact on the conversion of GO to GQDs. In electro-Fenton reaction, the rates of hydroxyl radical production were influenced by electrochemical process, making it effective to control the whole process of GQDs preparation. The UV-vis spectra of GO before and after the electro-Fenton reaction were recorded. As shown in Fig. S5, the GO has two absorption peaks at 230 and 304 nm, corresponding to π to π\* transition of aromatic sp<sup>2</sup> domains and n to π\* transition of C=O bond, respectively. Meanwhile, two UV-vis absorption peaks centered at 210 and 276 nm are observed after GO solution reacted with the electro-Fenton reagent (Fig. S5). The origin of these absorptions can be ascribed to π electron transition in oxygen-containing GQDs. The stronger absorption at 210 nm corresponds to a good deal of π to π\* transition of aromatic sp<sup>2</sup> domains, and the absorption at 276 nm is due to n to π\* transition of C=O bond, located in the edge of GQDs.<sup>33</sup> The PL and PLE spectra of the GQDs solution were recorded (Fig. 1c). The GQDs were excited by a Xe lamp from the fluorescence spectrometer and an optimal emission peak centered at 401 nm was observed when the sample was excited at 320 nm with a Stokes shift of 81 nm. The PLE spectrum recorded with the optimal luminescence showed two peaks at 256 nm, which was regarded as a shoulder peak, and 320 nm. The theory used to account for this phenomenon was proposed by Pan et al.<sup>33</sup> The PLE peaks at 256 and 320 nm correspond to the absorption from the transitions of π to π\* and n to π\*, related to the absorption peaks at 210 and 276 nm in absorption spectrum, respectively. Furthermore, the PLE spectrum demonstrates that the luminescence observed from GQDs is directly correlated with both of the two kinds of transitions rather than only transition of π to π\*, which can be used to explain the unique optical properties of GQDs. The as-



**Fig. 1** (a) TEM image of the GQDs prepared from electro-Fenton reaction and diameter distribution of as-prepared GQDs. (b) HRTEM image of an individual GQD shown in panel a. (c) PLE spectra of GQDs with the detection wavelength of 401 nm and PL spectra excited at 320 nm. (d) PL spectra of the GQDs excited by various wavelengths ranging from 320 to 460 nm.

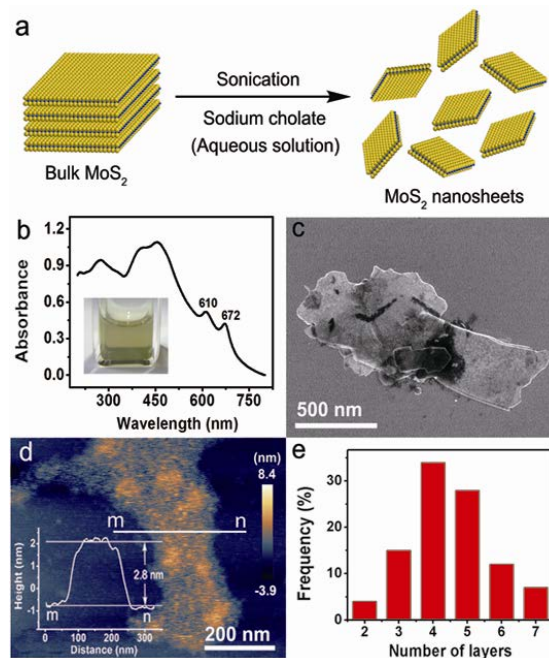
prepared GQDs also exhibits an excitation-dependent PL behavior. When the excitation wavelength changes from 320 to 460 nm, the PL peak shifts to longer wavelengths and the PL intensity decreases significantly (Fig. 1d).

With the successful preparation of GQDs from morphologic change of GO sheets induced by hydroxyl radicals in electro-Fenton reaction, it was proposed that other 2-dimensional materials like graphene in structure might react with hydroxyl radicals as well. Therefore, due to the unique and excellent properties, few-layer MoS<sub>2</sub> nanosheets, originally obtained from exfoliation of MoS<sub>2</sub> powder, were also taken into consideration. To the best of our knowledge, the discussion about the reaction between MoS<sub>2</sub> nanosheets and hydroxyl radicals has never been conducted before. Nevertheless, the morphologic changes of MoS<sub>2</sub> nanosheets under the influence of hydroxyl radicals generated from electro-Fenton reaction attracted our interests.

### Exfoliation of MoS<sub>2</sub> in aqueous surfactant solution

A more recent strategy for exfoliation is to expose the layered material to ultrasonic waves in a solvent.<sup>41,45</sup> Such waves generate cavitation bubbles that collapse into high-energy jets, breaking up the layered crystallites and producing exfoliated nanosheets. Sonication has been used to exfoliate bulk MoS<sub>2</sub> in water, so long as ionic surfactants are present to prevent rapid reaggregation of MoS<sub>2</sub> nanosheets due to their van der Waals binding to the exfoliated sheets and subsequent electrostatic stabilization.<sup>41</sup> In our experiment, MoS<sub>2</sub> nanosheets were obtained through the sonication-assisted exfoliation of MoS<sub>2</sub> crystal in aqueous surfactant solution (Fig. 2a). MoS<sub>2</sub> powder was added to sodium cholate aqueous solution for sonication-treatment, resulting in the preparation of dispersed solution. Centrifugation of dispersed solution at 3000 rpm was used to remove the bulk and thicker MoS<sub>2</sub>. The precipitation was collected at 12000 rpm to obtain exfoliated MoS<sub>2</sub> nanosheets. The surfactant adsorbed on the surface of MoS<sub>2</sub> nanosheets was

removed through re-dispersing the precipitation in ultrapure water and collecting the new precipitation at 12000 rpm in centrifugation. The results of thermogravimetric analysis were used to verify the purification of MoS<sub>2</sub> nanosheets through the process of washing and centrifugation, exhibiting that most of sodium cholate were removed (Fig. S6). Optical absorption is one characteristic which is related to the band structure of a semiconductor. Bulk MoS<sub>2</sub> is a semiconductor with an indirect bandgap of about 1.29 eV and no characteristic absorption peak can be observed in UV-vis absorption spectrum.<sup>16</sup> However, theoretical calculation has predicted an indirect to direct bandgap transition in the *d*-electron system when the bulk MoS<sub>2</sub> is thinned to a single layer, and the few angstrom-thin single-layered MoS<sub>2</sub> has recently been reported to exhibit a significant energy bandgap of 1.8 eV, while those of double- and triple-layer MoS<sub>2</sub> reduce to 1.65 and 1.35 eV, respectively.<sup>46</sup> The as-exfoliated MoS<sub>2</sub> nanosheets aqueous solution with yellow-green color has two characteristic absorption peaks at 610 and 672 nm (Fig. 2b), corresponding to B and A excitonic peaks, respectively, derived from the K point of the Brillouin zone in 2D MoS<sub>2</sub> with large lateral dimensions.<sup>25,45</sup> The sheet structure and morphology are obviously exhibited in SEM image (Fig. 2c), primarily showing the nanoscale of as-exfoliated MoS<sub>2</sub>. The AFM image of MoS<sub>2</sub> nanosheets is shown in Fig. 2d and the height of as-exfoliated MoS<sub>2</sub> nanosheets was measured to be around 2.8 nm (Inset of Fig. 2d), corresponding to about 4 monolayers of MoS<sub>2</sub>.<sup>25,47</sup> A statistical analysis based on AFM measurements indicated that the MoS<sub>2</sub> nanosheets prepared from sonication-assisted exfoliation of MoS<sub>2</sub> powder had various thicknesses with the



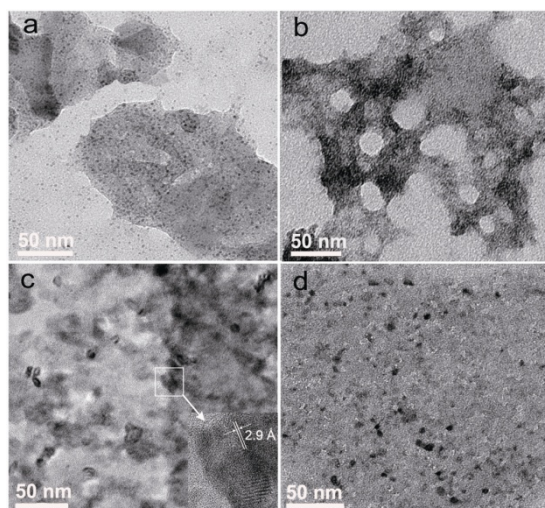
**Fig. 2** (a) Schematic illustration of the exfoliation of bulk MoS<sub>2</sub> for the preparation of MoS<sub>2</sub> nanosheets. (b) UV-vis absorption spectrum and photograph (inset) of exfoliated MoS<sub>2</sub> nanosheets aqueous solution. (c) SEM and (d) AFM images of a typical MoS<sub>2</sub> nanosheet with the inset representing the height profile along the white line overlaid on the image. (e) The thickness distribution based on 100 randomly selected MoS<sub>2</sub> nanosheets.

majority in the range of 3 to 5 monolayers (Fig. 2e).

During the past few years, MoS<sub>2</sub> based catalysts have been considered promising alternatives for platinum due to their high abundance and low cost. Promising catalytic activity from MoS<sub>2</sub> in the hydrogen evolution reaction (HER) is attributed to the active site located along the edges of its two-dimensional layered crystal. If MoS<sub>2</sub> based catalysts are to realize their potential, there is an urgent need to design MoS<sub>2</sub> nanostructures with more edge sites.<sup>28-31</sup> On the basis of electro-Fenton reaction of GO turning 2D graphene sheets into small quantum dots, the protocol that using electro-Fenton reaction to generate new morphologies of MoS<sub>2</sub> sheets was proposed.

### Electro-Fenton process of MoS<sub>2</sub> nanosheets

The progresses of electro-Fenton reaction of MoS<sub>2</sub> nanosheets were monitored using TEM. As shown in Fig. 3a, the basal plane of MoS<sub>2</sub> nanosheets is clearly observed, meanwhile, a number of ultrasmall nanoparticles, representing the fragments arising from sonication of bulk crystals, could also be found in the samples of as-obtained nanosheets. After 20 min of the reaction, a few small holes with a diameter of approximately 20 nm are distinctly observed on the basal plane of MoS<sub>2</sub> nanosheets (Fig. 3b). Undergoing a short time of electro-Fenton reaction less than 20 min, the layered morphology of the MoS<sub>2</sub> nanosheets is still preserved, but exhibits the nanoporous facets. Herein, a new morphology of MoS<sub>2</sub>, nanoporous MoS<sub>2</sub> nanosheet, was obtained through electro-Fenton treatment of exfoliated MoS<sub>2</sub> nanosheets. Indeed, considerable focus has been placed in developing unique morphologies of 2D material, including graphene sheets with nanoporous facets that exhibit excellent performance as supercapacitor electrodes.<sup>48,49</sup> By creating pores within the MoS<sub>2</sub> nanosheets, the active edge sites at the surface of sheets can increase and improvements in some unique properties of MoS<sub>2</sub> nanosheets are realized. Similar to the nanoporous graphene sheets, the nanoporous MoS<sub>2</sub> nanosheets should exhibit potential

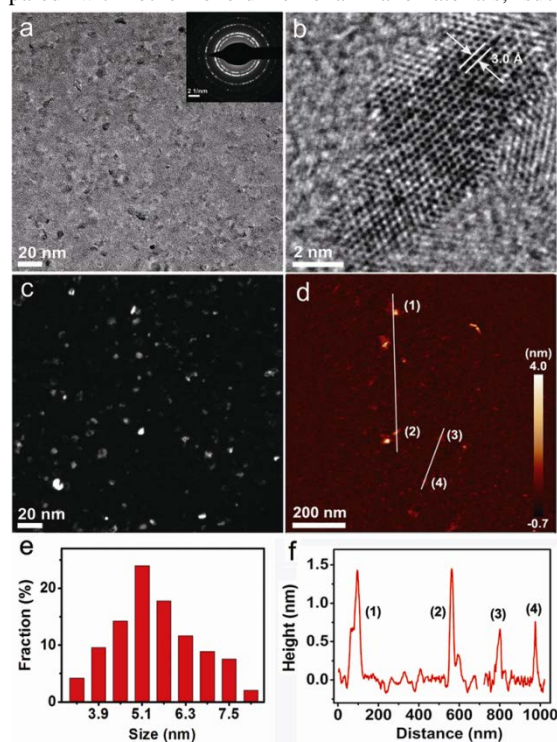


**Fig. 3** TEM images of MoS<sub>2</sub> nanosheets before (a) and after being reacted with the pre-Fenton reagents (O<sub>2</sub> and Fe<sup>2+</sup>) under the assistance of electrochemical process (applied potential of -0.5 V) for 20 (b), 40 (c), and 60 (d) min, respectively. The inset in Figure 3c is the HRTEM image of single MoS<sub>2</sub>-QD shown in panel c.

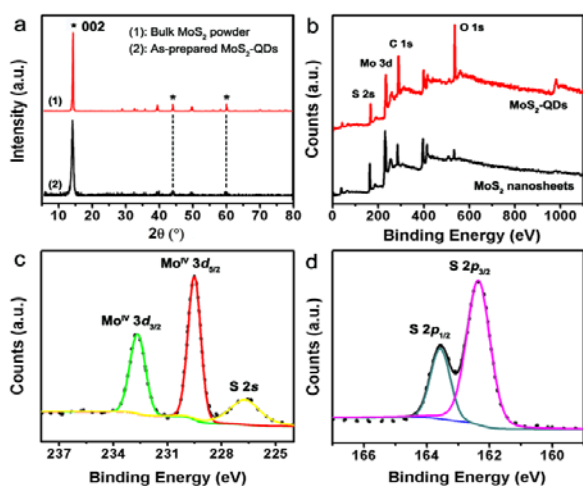
use in opto-electronic and renewable energy applications. Furthermore, with the reaction time increasing to 40 min, an increasing number of holes with various sizes appeared on the MoS<sub>2</sub> nanosheets, and a number of small nanosheets under the quantum scale were gradually obtained, which were derived from the spaces between holes (Fig. 3c). After 60 min of the electro-Fenton reaction, MoS<sub>2</sub> nanosheets were completely cut into small dots (Fig. 3d), which were demonstrated to be zero-dimensional MoS<sub>2</sub>-QDs later using HRTEM, AFM, and PL. As a control experiment, the same electrochemical process of MoS<sub>2</sub> nanosheets aqueous solution without the addition of Fe<sup>2+</sup> was conducted for 60 min, and no apparent changes were found on basal planes. Meanwhile, the reaction of MoS<sub>2</sub> nanosheets with conventional Fenton reagent was carried out, and no any apparent holes or changes were observed. It was verified that electrochemical process and pre-Fenton reagents (O<sub>2</sub> and Fe<sup>2+</sup>) were essential for the preparation of nanoporous MoS<sub>2</sub> monosheets and MoS<sub>2</sub>-QDs.

### Characterization of MoS<sub>2</sub> quantum dots

The as-prepared MoS<sub>2</sub>-QDs were subjected to TEM measurement (Fig. 4a), revealing that a few MoS<sub>2</sub>-QDs are uniformly distributed without agglomeration. The HRTEM image (Fig. 4b) shows an individual MoS<sub>2</sub> quantum dot with the length and width of approximately 11.5 and 5.3 nm, respectively, indicating that zero-dimensional MoS<sub>2</sub>-QDs have a different morphology compared with other one-dimensional nanomaterials, such as



**Fig. 4** (a) TEM image of the as-prepared MoS<sub>2</sub>-QDs and the inset presents a typical electron diffraction pattern of MoS<sub>2</sub>-QDs. (b) HRTEM image of an individual MoS<sub>2</sub>-QD with measured lattice spacing. (c) HAADF-STEM image of the dots. (d) AFM image of the dots deposited on freshly cleaved mica substrates. (e) Size distribution and (f) height distribution of as-prepared MoS<sub>2</sub>-QDs in panel d. Height: ca. 0.7 nm, one layer; ca. 1.4 nm, two layers.



**Fig. 5** (a) XRD patterns of MoS<sub>2</sub>-QDs prepared from electro-Fenton reaction and bulk MoS<sub>2</sub> powder (peaks correspond to [∗] 2H MoS<sub>2</sub>). (b) XPS survey of MoS<sub>2</sub> nanosheets before electro-Fenton reaction and as-prepared MoS<sub>2</sub>-QDs. High resolution Mo 3d (c) and S 2p (d) spectra of MoS<sub>2</sub>-QDs.

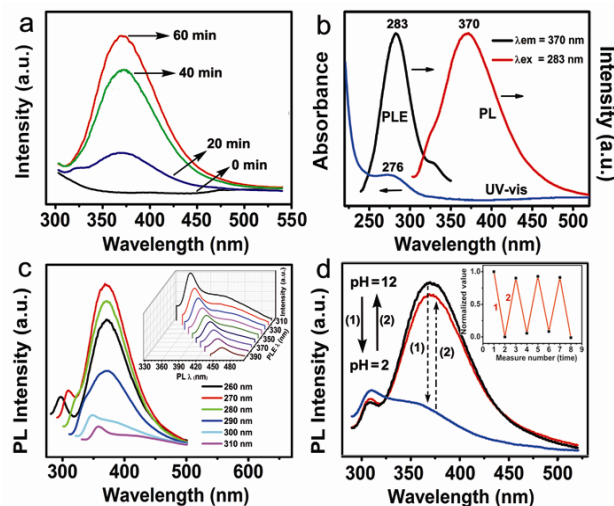
nanoribbon, nanotube, and nanowire.<sup>50</sup> Moreover, as expected, the HRTEM illustrates that the MoS<sub>2</sub>-QDs have 2D hexagonal symmetry crystalline structure with the lattice spacing of 3.0 Å, which is similar to the hexagonal pattern of MoS<sub>2</sub> monolayer.<sup>51</sup> This result was further verified by the electron diffraction pattern of MoS<sub>2</sub>-QDs (Inset of Fig. 4a). In order to reveal the morphology feature of MoS<sub>2</sub>-QD, HAADF-STEM image is shown clearly in Fig. 4c. Fig. 4d presents the AFM image of the MoS<sub>2</sub>-QDs on a fresh mica substrate and a few small dots with various scales of heights below 3 nm are observed. From the size distribution (Fig. 4e), we can see that the size of as-prepared MoS<sub>2</sub>-QDs was uniform ranging from 3 to 8 nm with the average value of 5.43 nm. Meanwhile, combining the AFM image and height distribution (Fig. 4f), it was shown that the heights of most of the MoS<sub>2</sub>-QDs were below 2 nm, that is, around 1.4 nm representing the height of MoS<sub>2</sub> bilayers and approximately 0.7 nm indicating the type of monolayer.

The crystal structure of these as-generated MoS<sub>2</sub>-QDs is investigated using XRD system, while the MoS<sub>2</sub> powder is used as the reference. From the XRD pattern shown in Fig. 5a, it can be observed that both MoS<sub>2</sub>-QDs and bulk powder are identified as 2H MoS<sub>2</sub> with a dominant peak appearing at 14.4°, representing the (002) plane (ICDD card no. 77-1716).<sup>52</sup> In order to explore the changes of chemical composition, XPS of MoS<sub>2</sub> nanosheets before and after electro-Fenton reaction were recorded. It can be seen from Fig. 5b that XPS survey exhibits that the proportion of oxygen element in MoS<sub>2</sub> nanosheets sample after electro-Fenton reaction increases due to the influence of hydroxyl radicals. Meanwhile, the atomic ratio of Mo:S increase from 1:2.30 (MoS<sub>2</sub> nanosheets) to 1:2.42 (MoS<sub>2</sub>-QDs), ascribed to an increasing number of unsaturated sulfur atoms located at the external edge of MoS<sub>2</sub>-QDs. As shown in Fig. 5c and 5d, Mo3d<sub>3/2</sub>, Mo3d<sub>5/2</sub>, S2s, S2p<sub>1/2</sub>, and S2p<sub>3/2</sub> peaks can be observed at 232.7, 229.5, 226.7, 163.6, and 162.3 eV, respectively, indicating the dominant 2H MoS<sub>2</sub> phase in the crystal structure.<sup>52</sup> Two characteristic peaks located at 229.5 and 232.7 eV arise from Mo

3d<sub>5/2</sub> and Mo 3d<sub>3/2</sub>, suggesting the dominance of +4 oxidation state.<sup>30</sup> Whereas the S 2p region (Fig. 5d) exhibits primarily a single doublet with the 2p<sub>3/2</sub> peak at 162.3 eV, corresponding to -2 oxidation state of sulfur.<sup>30</sup> The semiconducting phase of as-prepared MoS<sub>2</sub>-QDs (2H MoS<sub>2</sub>) was further verified based on the XPS analyses.

### Photoluminescence properties of MoS<sub>2</sub> quantum dots

Recent success in isolating MoS<sub>2</sub> monolayer has allowed the observation of strong photoluminescence, attributed to the direct gap electronic structure of monolayer MoS<sub>2</sub>. According to theoretical calculation,<sup>53,54</sup> the MoS<sub>2</sub> nanostructure exhibited some unique photoluminescence properties due to the effect of quantum confinement and edge. Therefore, the optical properties of as-prepared MoS<sub>2</sub>-QDs were explored. Fig. 6a shows the PL spectra of exfoliated MoS<sub>2</sub> nanosheets solutions at different reaction times in electro-Fenton process when the excitation wavelength is 283 nm. It can be seen that the as-exfoliated MoS<sub>2</sub> nanosheets before the electro-Fenton reaction are PL inactive, indicating the absence of MoS<sub>2</sub>-QDs in exfoliated MoS<sub>2</sub> nanosheets samples. However, after 20 min of electro-Fenton reaction, the as-generated nanoporous MoS<sub>2</sub> nanosheets (Fig. 3b) show weak PL peak centered at 370 nm, which continues to increase when the reaction time increases to 60 min. Hence, the morphology of quantum dot (Fig. 3d) was demonstrated to exhibit great PL property. The MoS<sub>2</sub>-QDs have an absorption band peaked at 276 nm (Fig. 6b), whereas the characteristic absorption peaks of exfoliated MoS<sub>2</sub> nanosheets at 610 and 672 nm disappeared, indicating the absence of 2D MoS<sub>2</sub> nanosheets with large lateral dimensions in as-prepared MoS<sub>2</sub>-QDs. It was suggested that the optical absorption of low-dimensional or



**Fig. 6** (a) PL spectra of exfoliated MoS<sub>2</sub> nanosheets suspensions before and after the electro-Fenton reaction for 20, 40, and 60 min using the excitation wavelength of 283 nm. (b) UV-vis absorption, PL, and PLE spectra of MoS<sub>2</sub>-QDs dispersed in water. (c) PL spectra of as-prepared MoS<sub>2</sub>-QDs aqueous solution at different excitation wavelengths ranging from 260 to 310 nm. Inset: excitation wavelengths ranging from 310 to 390 nm. (d) pH-dependent PL spectra when pH is switched between 12 and 2. Inset: reversible switching of PL intensity through the change of pH between 12 and 2.

quantum dot MoS<sub>2</sub> exhibited a strong blue shift when the lateral dimensions of the MoS<sub>2</sub> nanostructures were reduced to <50 nm, ascribed to the quantum size effect.<sup>52</sup> Hence, for MoS<sub>2</sub>-QDs with the plane size of ca. 5 nm, the optical absorption peak centered at 276 nm can be assigned to blue-shifted convoluted Z, C, and D excitonic peaks.<sup>52,55</sup> As we can see from the PL and PLE spectra (Fig. 6b), an optimal emission peak centered at 370 nm is observed when the sample is excited at 283 nm with a Stokes shift of 87 nm. The PLE spectrum recorded with the detection wavelength of 370 nm shows a peak at 283 nm, corresponding to the absorption peak at 276 nm in the UV-vis absorption spectrum. The PL spectra excited by different wavelengths ranging from 260 nm to 390 nm were recorded in Fig. 6c, showing that MoS<sub>2</sub>-QDs aqueous solution have an excitation-independent characteristic emission spectrum peaked at 370 nm and the PL intensity decreases with the excitation wavelengths changing from 270 to 310 nm. Meanwhile, there is another weaker emission spectrum being observed. When the excitation wavelength changes from 270 to 390 nm, the PL peak shifts to longer wavelengths peaked from 310 to 465 nm and the intensity decreases, showing the excitation-dependent characteristic. A pH-dependent PL with the peak at 370 nm is exhibited in MoS<sub>2</sub>-QDs aqueous solution (Fig. 6d). Under alkaline conditions, the MoS<sub>2</sub>-QDs emit strong PL, whereas, the PL is nearly completely quenched under acidic conditions. If pH is switched repeatedly between 12 and 2, the PL intensity varies reversibly. It has been suggested that a significant number of H<sup>+</sup> intercalates into the 2H MoS<sub>2</sub>-QDs, transforming it from the semiconducting 2H phase into the metallic MoS<sub>2</sub> (1T) phase that eventually leads to the loss of their PL properties.<sup>47</sup>

## Conclusion

In summary, a new route to effectively and controllably prepare GQDs from GO sheets was proposed using electro-Fenton reaction. Meanwhile, it was demonstrated that the exfoliated MoS<sub>2</sub> nanosheets could react with hydroxyl radicals induced by electro-Fenton process, achieving the generation of new morphologies, nanoporous MoS<sub>2</sub> nanosheets, and zero-dimensional MoS<sub>2</sub>-QDs, in a mass scale. The as-prepared MoS<sub>2</sub>-QDs showed unique PL properties due to the quantum confinement, edge effect and their intrinsic characteristics, which were of great promise to be a novel optical probe and used for optoelectronic systems. Compared to MoS<sub>2</sub> nanosheets exhibiting excellent catalytic activity for the hydrogen evolution reaction, nanoporous MoS<sub>2</sub> sheets and MoS<sub>2</sub> quantum dots possessing much more active edge sites could enable new opportunities for enhancing properties in hydrogen evolution catalysis and other important technological applications.

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## Notes and references

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