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# High-Efficiency Transfer of Percolating Nanowire Film for Stretchable and Transparent Photodetectors

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Stretchable devices with good transparency offer exciting new applications over the existing technologies, but remarkable difficulties remain in the fabrication of transparent and stretchable devices. In this paper, we report an effective method to fabricate transparent elastic photodetectors which combines the merits of transparent Polydimethylsiloxane (PDMS) polymer with its stretchability and  $\text{Zn}_2\text{SnO}_4$  nanowire (NW) with its photodetection functionality. Zonyl fluorosurfactant is found to be critical which improves the bonding between the functional NWs and PDMS matrix, thus enabling the high efficient transfer of NW structures into PDMS. Highly conductive and thin percolating AgNW film were successfully embedded into PDMS mixed with ~11% Zonyl which are otherwise not achievable with pure PDMS. Transparent and stretchable photodetectors were fabricated with the developed method. The photocurrent was found to be reciprocal to the square of the channel length,  $I_{ph} \sim 1/l^2$ . The chemical bonded sensing materials in the PDMS matrix allow more NW exposure to the air. This lead to a fast switching operation of the photodetectors with a response time below 0.8 s and reset time around 3 s, which is significantly improved compared to reported stretchable NW photodetectors fully embedded in the polymer matrix.

## Introduction

Stretchable electronics have recently received great attention in the research community.<sup>1-6</sup> Stretchable devices are “soft” electronic devices which can be deformed and wrapped onto nonplanar curved surface. They can mimic the mechanical compliance of human skin while a mass of additional functionalities can be integrated. Among them, stretchable photodetectors provide the function of converting light stimuli information into electrical signal. They enable the integrations with biological systems such as wearable monitoring devices, electronic eye cameras conformed to special curvilinear shapes, and infrared detectors for night vision<sup>33-35</sup> as well as many other applications. However, very limited works can be found on stretchable photodetectors. Hemispherical imaging systems were demonstrated by utilizing thin and narrow wavy lines for connection between conventional rigid silicon sensing elements in their micrometer size.<sup>7,8</sup> Nanowires (NWs) have also been used to fabricate stretchable photodetectors with good stretchability up to 100%.<sup>9</sup> However, stretchable and meanwhile transparent photodetectors have not been achieved,

despite the fact that good optical transparency will enable exciting novel applications, such as integrating with foldable screens, smart windows and other transparent electronics.

Polydimethylsiloxane (PDMS) is one of the most promising materials for stretchable electronics with good biocompatibility, high stretchability at room temperature, excellent optical transparency and ease of molding down to the nanometer scale.<sup>10,11</sup> Elastic conductors based on PDMS and conductive fillers (such as AgNWs) have been adopted as stretchable electric connections.<sup>9,14</sup> However, these elastic conductors were based on AgNW films up to several micrometer thick and thus were opaque. Thin and transparent AgNW films have not been demonstrated for stretchable conductors in these reports, probably due to the fact that thin AgNW network could not be effectively transferred and embedded into the PDMS matrix. Large conductivity degradation might occur in the thin conducting network after the transfer process. Though AgNW embedded in poly(acrylate) has been demonstrated to achieve transparency and stretchability at the same time, the conductor required to be heated above its glass transition temperature (above room temperature) to become stretchable.<sup>12,13</sup> Other

approaches such as directly depositing thin NW networks on the top surface of PDMS might help to circumvent the problem,<sup>15,16</sup> but the exposed structures of the NW films make them mechanically vulnerable and tend to be easily displaced under repeated stretching.<sup>17</sup> The protruded NWs out of the substrate plane are also detrimental when a smooth surface is required for subsequent deposition of active layers in devices like light-emitting diodes (LEDs).<sup>18,19</sup> Here, we report a surfactant-assisted method to tackle the low-efficiency transfer of thin NW film into PDMS elastomer matrix. High-efficiency transfer of thin percolating NW film enables the successful fabrication of stretchable and transparent NW photodetectors. The innovative approach can be widely applied for the fabrication of other stretchable or wearable devices.

## Experimental

### Transparent and stretchable conductor fabrication

AgNWs were purchased from Seashell Technology with diameters of 120-150 nm and lengths of 20-50  $\mu\text{m}$ . The AgNWs solution in isopropyl alcohol was diluted into a concentration of 0.5 mg/ml before use. The AgNWs solution was deposited onto pre-cleaned glass substrates by a spray gun in the fume hood. Due to the fast evaporation rate of isopropyl alcohol, the spray-coated solution was immediately dried after deposited onto the substrates without any heating. Pure PDMS was prepared by mixing the base and curing agent (Sylgard 184, Dow Corning) with a ratio of 10:1. The liquid mixture was degassed and thermally cured at 60  $^{\circ}\text{C}$  overnight. Zonyl FS-300

fluorosurfactant was purchased from Sigma-Aldrich and used as received. The PDMS base, curing agent and Zonyl was mixed in different weight ratios of 10:2:1, 10:2:1.5, and 10:2:2. The mixture composition of 10:2:1.5 showed to create good bonding between the PDMS matrix and AgNWs without affecting the stretchability of PDMS. SEM images on the surfaces of AgNWs/PDMS mixed with different Zonyl concentrations were shown in the Supporting Information Fig. S1.

Fig. 1a illustrates the fabrication procedure of a stretchable transparent conductor. Initially, the AgNW solution was spray-coated onto a pre-cleaned glass substrate to form a homogenous interconnecting NW network, as shown in the SEM image of Fig. 1b. The AgNW network was annealed at 200  $^{\circ}\text{C}$  for 20 min to remove the polyvinylpyrrolidone (PVP) capping agent on the NW surface and create fusion between the AgNWs.<sup>20</sup> Sheet resistivity of the AgNW film was significantly reduced after the annealing process. Liquid PDMS mixed with different amount of fluorosurfactant Zonyl-FS300 (Zonyl) was poured onto the AgNW coated substrate, degassed for 30 min in a vacuum desiccator to allow the liquid to penetrate and wrap around the NW network. After that, the samples were transferred into an oven to cure the PDMS overnight at 110  $^{\circ}\text{C}$ . The resultant AgNWs/PDMS conductors were peeled off from the glass substrate for further characterization. Fig. 1c shows a transparent stretchable conductor with homogeneously coated AgNW film.

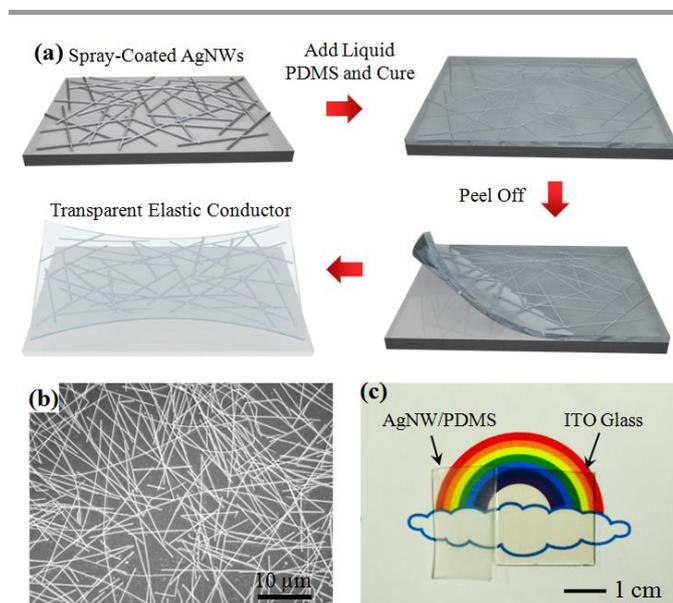
### Photodetector fabrication

$\text{Zn}_2\text{SnO}_4$  NWs were synthesized by a chemical vapor deposition method.<sup>32</sup> The synthesized  $\text{Zn}_2\text{SnO}_4$  NWs were removed from the silicon substrate and dispersed into isopropanol alcohol solution with a concentration of  $\sim 0.1$  mg/ml. The  $\text{Zn}_2\text{SnO}_4$  NWs solution was sonicated for 15 minutes before used. Morphologies of the  $\text{Zn}_2\text{SnO}_4$  NWs and AgNWs were coated onto glass substrates through the shadow masks subsequently to form the photodetector electrode and detection channel arrays. The devices were then transferred into the PDMS matrix by the method developed above.

### Characterization

SEM images of the samples were taken using a field-emission SEM (FE-SEM, JSM 7600F). The transmittance spectra were measured by a Shimadzu spectrometer (UV-3600) equipped with an optical integration sphere. The strain tests of the stretchable conductors and photodetectors were both carried out on a home-made stretching stage at room temperature. The electrical properties of the photodetector were characterized by a Keithley 4200-SCS parameter analyzer. The UV light with a central wavelength of 254 nm was generated by a portable UV lamp.

## Result and discussion

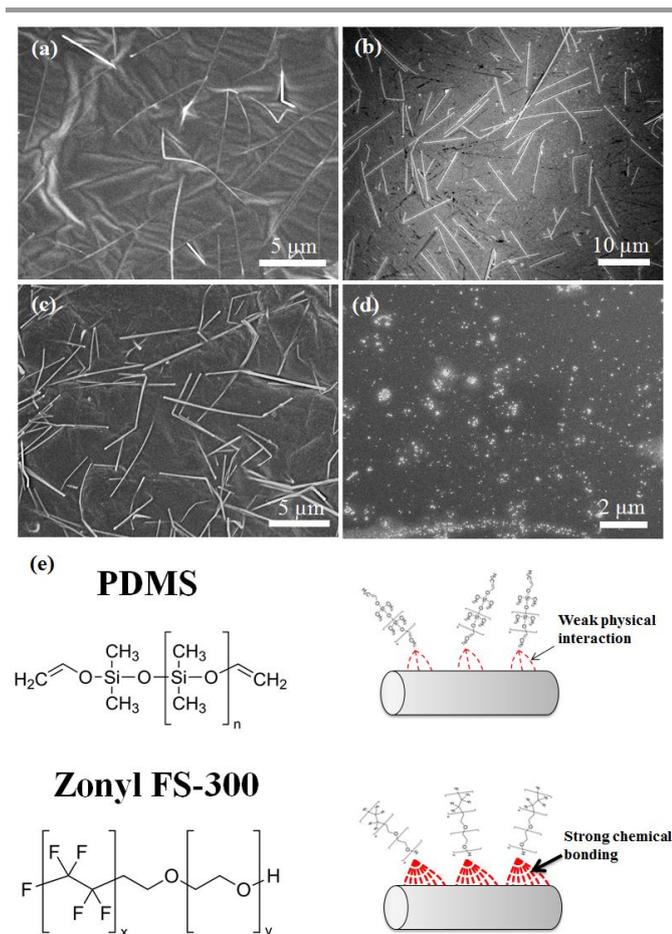


**Fig. 1** (a) Fabrication procedure of the transparent stretchable conductor. (b) SEM image of a highly conductive AgNW film coated on glass substrate. (c) Optical photograph of the AgNWs/PDMS stretchable conductor with a sheet resistivity of 4.5  $\Omega/\text{sq}$ , comparing with ITO glass with a sheet resistivity of 10  $\Omega/\text{sq}$ .

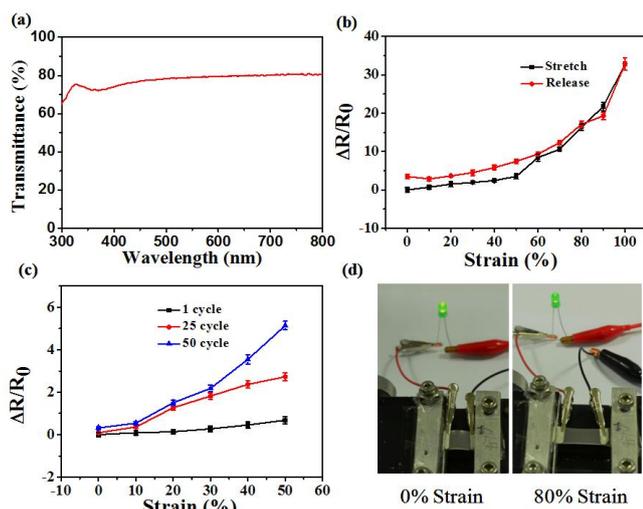
Glass substrates coated with similar densities of AgNWs were used to study the effect of Zonyl on the NW transfer efficiency. First we studied the transfer into pure PDMS matrix without Zonyl surfactant. Surfaces of the AgNWs/PDMS conductors were characterized with SEM after being peeled off from the glass substrates. The peeled surface of AgNWs embedded in pure PDMS, as shown in Fig. 2a, has significantly reduced wire density compared to the as-coated network on glass substrate (Fig. 1b). The AgNWs distribute sparsely in the PDMS matrix with limited interconnection nodes. The resultant conductor had a poor conductivity, which was beyond the measurement range of our equipment. Though the thickness of the AgNWs films were increased with the sample transparency dramatically reduced below ~40%, no substantial improvement in the conductivity could be observed. To investigate the cause of

weak conductivity, we inspected the surface of the exposed glass substrate after the transfer process. As represented in Fig. 2b, the SEM image shows that substantial amount of AgNWs remain on the glass substrate, revealing clearly a non-ideal NW transfer efficiency. Highly conductive AgNWs/PDMS conductors have been reported by embedding AgNW film with thickness of several micrometers.<sup>9,14</sup> In these cases, the conductive pathways are established through the massive NW-NW junctions. Although peeling off the polymer composite from the substrate would sacrifice NWs on the top surface, the NW network can still retain its good conductivity with the large amount of interconnecting AgNWs in the bulk. To achieve an optically transparent conductor, however, a much thinner film of AgNWs is required (monolayer or few layer as shown in Fig. 1b). Consequently, damage to the conductive film during the embedding process will result in severe conductivity degradation. PDMS is well-known for its very low surface energy of ~20 mN/m.<sup>21</sup> The transfer can only be effective when most part of the NW is embedded within the PDMS polymer, as schematically shown in Fig. S2, Supporting Information. SEM observations also verified that most of the NWs successfully transferred into the pure PDMS matrix (without Zonyl surfactant) were fully embedded inside the PDMS (see comparison in Fig. 2c with plenty of exposed NWs). Those AgNWs that are close to the interface will remain on the supporting substrate due to the weak physical interaction between the AgNWs and PDMS. To achieve a high-efficiency transfer of the transparent AgNW film, stronger bonding between the NWs and PDMS is required. Zonyl fluorosurfactant has an amphiphilic nature with both fluorinated and ethylene glycol segments (Fig. 2e). It has been used to effectively improve the interaction between PEDOT:PSS and PDMS.<sup>22,23</sup> The ethylene glycol group enables chemical coupling with the AgNWs surface through O-Ag bonding<sup>24,25</sup> and is expected to improve the bonding strength between the PDMS matrix and AgNWs. Indeed, with the addition of ~11 wt% Zonyl, significantly improved transfer efficiency for AgNWs at the interface was observed as shown in Fig. 2c. The surfactant-assisted highly efficient transfer was further verified by the SEM characterization in Fig. 2d, showing that all the NWs have been transferred from the substrate. Smaller ratio of Zonyl had similarly shown increased transfer efficiency compared to pure PDMS while ~11 wt% Zonyl was demonstrated to offer the best transfer efficiency in our study. Continue to increase the Zonyl ratio may not increase the transfer efficiency. On the contrary, excess Zonyl will affect the mechanical properties of the PDMS matrix and lead to an inferior surface roughness of the polymer. The effects of different Zonyl ratios were further discussed in the supporting information (Figure S1). The chemical structures of PDMS and Zonyl as well as their distinct interactions with the AgNWs are presented in Fig. 2e.

Figure 3a shows the transmission spectra of the AgNW elastic conductor. The highly conductive AgNWs/PDMS composite, with a sheet resistivity of 4.5 Ω/sq, has a transmittance of 80% at 550 nm which is comparable to



**Fig. 2** (a) SEM image of the surface of AgNWs in pure PDMS after peeled off from glass substrate. (b) Top surface image of the glass substrate after peeling off the AgNWs/PDMS without Zonyl. (c) SEM image of the surface of AgNWs/PDMS elastic conductor with ~11 wt% Zonyl after peeled off from glass. (d) Top surface image of the glass substrate after peeling off the AgNWs/PDMS elastic conductor with ~11 wt% Zonyl. (e) Chemical structures of PDMS and Zonyl. The schematic on the right illustrates their distinct interactions with AgNWs.

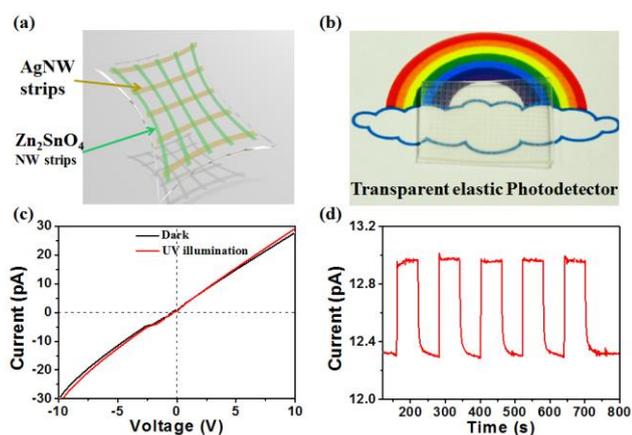


**Fig. 3** (a) Transmittance spectra of the AgNWs/PDMS stretchable conductor. (b) Resistance change of the AgNWs/PDMS conductor as a function of tensile strain. (c) Resistance change in the stretchable conductor as a function of tensile strain at the first, twenty-five and fifty stretching cycles. (d) Photographs of a powered light-emitting diode integrated with the elastic conductor under tensile strain of 0% and 80%.

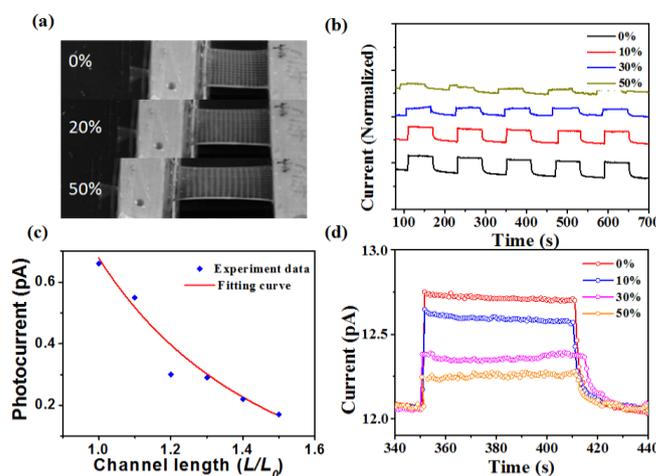
commercial ITO/glass ( $10 \text{ } \Omega/\text{sq}$ ,  $\sim 85\%$  transmittance). Transparency of the elastic conductor can be further improved by using AgNWs with higher aspect ratio.<sup>26</sup> The elastic conductor was stretched up to 100% and resistance changes were measured. Fig. 3b shows the resistance variation  $\Delta R/R_0$  ( $\Delta R = R - R_0$ ,  $R$  is the measured resistance under different strains,  $R_0$  is the initial resistance before stretching) upon stretching. The conductor could maintain good conductivity even at 100% tensile strain. The resistance first increased slowly before 50% strain, which was mainly due to the increasing length of the conducting path. Beyond the stretching strain of 50%, the

resistance change became more significant with  $\Delta R/R_0$  reaching 32.8 at 100% strain. The resistance change followed similar trend when the conductor was released. More stretching test was performed on a new sample in the strain range of 0-50%, as presented in Fig. 3c. Though there was a slight increase in the resistivity as the stretching cycles increased, the conductor could still maintain the conductive path under reversible stretching. Current was passed through the elastic conductor to power a LED, as shown in Fig. 3d. The LED remained lit when the conductor was stretched to 80% strain. Brightness of the LED reduced due to increased resistivity of the conductor under the stretched state. Performance of the stretchable conductor is much better comparable to previous reported AgNW films on PDMS substrates.<sup>16, 17</sup> The performance is also comparable to the distinguished works in the literatures which use other stretchable polymer matrix for the embedding process.<sup>2, 12</sup>

The surfactant-assisted transfer method is shown to be effective to fabricate PDMS-based transparent elastic conductor. Functionality of the NW films can be maintained, while stretchability can be imparted simultaneously by the PDMS matrix. Stretchable and transparent photodetector arrays based on conducting AgNW film and semiconducting  $\text{Zn}_2\text{SnO}_4$  NW films were successfully demonstrated with the developed method. AgNWs were first spray-coated onto the glass substrate through a shadow mask. Subsequently, the shadow mask was rotated by an angle of  $90^\circ$  and used to coat the light sensing materials. Other device configurations can also be realized by using different shadow masks.  $\text{Zn}_2\text{SnO}_4$  NWs were used for the sensing material as they have high sensitivity and selectivity to the Ultraviolet (UV) light.<sup>27</sup> Fig. 4a presents the schematic of the device. The  $\text{Zn}_2\text{SnO}_4$  NWs active channel, with length of 0.5 mm, was formed between the two parallel AgNW lines. Fig. 4b shows a stretchable photodetector device with high transparency. Transmittance spectrum of the device



**Fig. 4** (a) A schematic of the stretchable transparent UV photodetector. (b) A photograph of the stretchable elastic photodetector. (c)  $I$ - $V$  curves of the unstrained photodetector under dark and UV illumination conditions. (d) Dynamic current-time measurement of the device with periodical on/off UV light.



**Fig. 5** (a) Digital photographs of the stretchable photodetector under stretching test. (b) Response of the device at tensile strains of 0%, 10%, 30%, and 50%. (c) Fitting of the photocurrent at different tensile strains. (d) Magnification of the photocurrent responds with an enlarged on/off cycle at tensile strains of 0%, 10%, 30%, and 50%.

can be found in the Supporting Information Fig. S4. *I-V* curve of the device in Fig. 4c indicates good ohmic contact between the AgNWs and the Zn<sub>2</sub>SnO<sub>4</sub> NWs. The device current under dark environment and UV light illustration are also shown in Fig. 4c. The large dark current of the photodetector, 12.3 pA at the bias of 5V, can be contributed to the high carrier density, good mobility in the Zn<sub>2</sub>SnO<sub>4</sub> NWs or the large device area. Dynamic performance of the device, biased with 5 V, is shown in Fig. 4d. The UV light was turned on and off both for 60 s periodically. The device shows fast response to the UV light and good cyclic stability.

Performance of the photodetector was examined when the device was uniaxially stretched along the direction of the channel length. Fig. 5a shows images of the photodetector under 0%, 20% and 50% strains respectively. Photoresponse of the device is shown in Fig. 5b. The device demonstrates stable response up to the tensile strain of 50%, while the photocurrent decreases as the strain value increases. Photoresponse behavior of the device can be understood through the electron-hole separation upon excitation by the incident photons. These electrons are either collected by the electrodes (contribute to the photocurrent) or recombine with holes in the channel (do not contribute to the photocurrent). Consequently, lifetime of the photo-generated carriers  $\tau$  and the transit time of the carriers  $\tau_t$  are two critical parameters for the photodetector behavior. The photoconductive gain  $G$  can be evaluated by the following equation:<sup>28,29</sup>

$$G = \frac{\tau}{\tau_t} = \frac{\mu V \tau}{l^2} \quad (1)$$

Where  $\tau_t = l^2 / \mu V$  ( $l$  is the channel length of the device,  $\mu$  is the carrier mobility and  $V$  is the applied voltage). The photocurrent gain is proportional to the photocurrent  $I_{ph}$  ( $I_{ph} = I_{illumination} - I_{dark}$ ):<sup>30,31</sup>

$$G = \frac{I_{ph} / q}{P_{opt} / h\nu} \quad (2)$$

Where  $P_{opt}$  is the absorbed optical power,  $q$  is the elementary charge,  $h$  is the Planck constant, and  $\nu$  is the frequency of the absorbed photon. By combining Equation 1 and 2, one can obtain:

$$I_{ph} = \frac{P_{opt} q \mu V \tau}{h\nu l^2} \quad (3)$$

From equation 3, it can be concluded that the photocurrent is reciprocal to the square of the channel length with constant incident light power and applied voltage. Fig. 5c plots the photocurrent changes against channel length. The line fitting, red line in Fig. 5c, is performed on the experiment data, showing good agreement with the relationship between the photocurrent and channel length as deduced above. The

magnified photocurrent in a single on/off cycle shows fast respond time of the elastic photodetector. Within 0.8 s, which is limited by the measuring time of our equipment, the photocurrent of the device jumps to the maximum value after the detection light is on. The reset time is defined as the time required for the photocurrent to decrease to  $1/e$  (37%) of the maximum value, which is  $\sim 3$  s for the device. No clear dependence of the time response on the tensile strains was observed. The photoresponse behavior is faster compared to reported stretchable NW photodetector,<sup>9</sup> which has a response time of  $\sim 30$  s and reset time of  $\sim 6$  s at 0% strain. The response/reset time will further increase with increased strains. Switching speed of this device is hindered by the oxygen deficient environment with the device structures fully embedded in PDMS matrix. When the device is stretched, the contact area between the NWs and air will be further decreased (a more oxygen deficient environment), leading to an increased response/reset time. The transparent elastic photodetector reported here has a chemically bonded device structure with more NW surface exposed to air. The contact area changes between the functional NWs and air are negligible upon stretching. Consequently, a fast and stable photoresponse time was observed. Its switching speed is close to reported Zn<sub>2</sub>SnO<sub>4</sub> NW photodetector fabricated on SiO<sub>2</sub>/Si substrate (response time/reset time: 0.46 s/ 0.42 s).<sup>27</sup> Stretching the device along the channel width up to 50% shows almost no change in the photocurrent and photoresponse time, corroborating the mechanism illustrated. Cycling stability test of the device stretched along the channel length direction was also carried out. Detail result and discussion can be found in the supporting information (Figure S5).

## Conclusions

In summary, we have demonstrated an effective method to fabricate transparent and stretchable photodetectors. The elastic photodetector is based on embedded device structures in the PDMS matrix to render stretchable capability to the rigid inorganic materials while the functionality of the device is preserved. To reduce damage to the device structures when peeled off from the supporting substrates, a Zonyl fluorosurfactant was added into the PDMS liquid. The introduced ethylene glycol group from Zonyl greatly increases the bonding strength between the rubber matrix and the NW surface. A stretchable and transparent conductor was fabricated with sheet resistivity of 4.5  $\Omega$ /sq and transmittance of 80% at 550 nm. Following similar procedure, a transparent elastic UV photodetector was also demonstrated. The photodetector has stable performance with tensile strain up to 50%. The photocurrent is reciprocal to the square of the channel length,  $I_{ph} \sim 1/l^2$ . The photodetector has a prompt respond time below 0.8 s and a fast reset time of  $\sim 3$  s. Combining its good transparency and stretchability, the elastic photodetector may find broad applications when integrated with clothes, skins and

organs for wearable and implantable electronics which are unreachable with conventional rigid technologies.

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## Notes

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