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PAPER

Low-temperature solution-processable Ni(OH)₂ ultrathin nanosheet/Ngraphene nanohybrids for high-performance supercapacitor electrodes[†]

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A novel and facile strategy is developed to fabricate highly nitrogen-doped graphene (N-graphene) based nanohybrids with ultrathin nanosheet nanocrystals using a low-temperature, solution processing method

- ¹⁰ for high-performance supercapacitor electrodes. High N doping can be achieved together with one of the lowest oxygenous contents in chemically-reduced graphene and related nanohybrids at low temperature inspired by the large scale residue defects of chemically-reduced graphene. The high-quality ultrathin Ni(OH)₂ nanosheet nanocrystal/N-graphene layered nanohybrids can be applied as supercapacitor electrodes with ultrahigh capacitances of ~1551 F/g, excellent rate performance in the scan rate from 2 to
- ¹⁵ 100 mV/s and in the discharge current from 1.5 A/g to 30 A/g and good cycling stability. Moreover, the capacitance of Ni(OH)₂ nanosheet/N-graphene nanohybrids is two and one orders of magnitude higher than that for pure nanocrystals and that for the physical mixture of nanocrystal/N-graphene, respectively. Electron transfer in supercapacitor electrodes based on nanohybrids is over 100 times faster than that in electrodes from pure nanocrystals and several tens of times faster than that in electrodes from the term of the physical mixture of the term of term of the term of term

20 nanocrystal/N-graphene mixture. The results provide a low-cost, solution-processable and easily scalable route to high-performance graphene nanohybrid electrodes for energy applications.

1. Introduction

Graphene has great potential in various applications for its unique electronic properties.¹⁻⁵ Much attention has been paid to ²⁵ manipulate the electronic properties of graphene by size effect, defect, interface and hybrid.⁵⁻¹⁵ Atomic doping can alter the electronic properties of graphene intrinsically and modify the local composition and electronic structure by including foreign atoms in matrix. Recently, nitrogen (N) doped graphene has ³⁰ attracted a lot of endeavors for its excellent performance in

- electronic devices.¹⁶⁻²⁵ There are usually three routes to achieve N doping in graphene, including doping during chemical vapor deposition (CVD) growth,¹⁶ high temperature thermal annealing or decomposition with N containing species,¹⁷ and NH₃/N₂
- ³⁵ plasma.^{18,19} These methods are not solution-processable and need high-temperature processing like thermal annealing/deposition or expensive equipments, limiting the cost and large scale applications. Despite many efforts, however, direct lowtemperature, solution-processable N doping in graphene or ⁴⁰ reduced graphene is still a challenge.

Meanwhile, due to the low capacitance of commercial carbonbased electrical double layer capacitor, metal oxide-based Faradaic pseudocapacitive electrodes have been recognized as one of the most potential alternative supercapacitor electrodes.²⁶⁻ ³⁶ Metal oxides usually have low conductivity and their hybrids with more conductive scaffolds are expected to improve the electron transfer in electrodes and capacitive performance. For example, nanoporous gold and carbon nanotubes (CNTs) are applied as conductive scaffolds for hybrid supercapacitor
 ⁵⁰ electrodes to improve the capacitive performance of metal oxides.^{26,27,37-40} However, these kinds of scaffolds are expensive and low-cost high-quality scaffolds for metal oxide hybrids with efficient electron transfer in electrodes are highly expected. N-graphene, due to its improved electronic conductivity with N
 ⁵⁵ doping and two-dimensional structures, is a very potential alternative to nanoporous metal and CNTs in hybrid supercapacitor electrodes.

Chemically-reduced graphene from exfoliated graphite oxide is one of most available graphene and the only type of graphene to ⁶⁰ date which can be produced by the ton.⁴¹ Here, inspired by the large scale residue defects in chemically-reduced graphene, a novel and facile method was developed to prepare solutionprocessable N-doped graphene (N-graphene) and related Ngraphene nanohybrids with Ni(OH)₂ nanosheet nanocrystals at ⁶⁵ low-temperature based on chemically-reduced graphene. It was found for the first time that high N doping can be achieved even in chemically-reduced graphene by low temperature, solutionprocessed way due to the large scale residue defects. Moreover,

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simultaneous *in situ* growth of Ni(OH)₂ nanosheet nanocrystals was achieved with a close contact with N-graphene by applying a negatively-charged surfactant as the growth medium and the resulted layered nanohybrids demonstrate an ultrahigh s capacitance of ~1551 F/g for graphene hybrid electrodes with

- excellent rate performance (from 2 to 100 mV/s and from 1.5 A/g to 30 A/g) and good cycling stability. The N-graphene hybrid electrodes also show over 100 times higher electron transfer rate than pure nanocrystals and several times higher than
- ¹⁰ nanocrystals/N-graphene mixture, leading to at least one order of magnitude higher capacitance in hybrid electrodes than other two types of electrodes. This low-temperature method may provide a novel route to large scale N-graphene nanohybrids for high performance electrodes in energy storage and conversion.

2. Results and discussion

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The two-step method was developed for preparing highly N doped graphene at low temperature, starting from graphene oxide modified by negatively-charged surfactant (**Figure 1**). The ²⁰ surfactant-modified graphene oxide (S-graphene oxide) was first reduced by normal chemical method, e.g., using hydrazine in the presence of surfactant. Then obtained chemically-reduced graphene was further processed under 150 °C with hydrothermal treatments in the presence of NH₃·H₂O. This strategy can be ²⁵ further used to prepare various N-doped graphene/nanosheet nanocrystal hybrids, where metal salt was added to the chemically-reduced graphene before hydrothermal reactions (**Figure 1**). Facile *in situ* nanocrystal growth on N-graphene achieved high-quality N-graphene based hybrid easily.



Figure 1. Scheme for low-temperature, solution-processable N doping in chemically-reduced graphene and nanosheet nanocrystal/N-graphene hybrids. S-graphene oxide: surfactant-modified graphene oxide. Solution concentration for chemically-³⁵ reduced graphene and nitrogen-doped graphene (N-graphene): 1 mg/mL.

The obtained N-graphene is highly solution processable in aqueous solutions (e.g., 1 mg/mL, **Figure 1**). N-graphene had ⁴⁰ high quality with strong six-fold symmetry in the diffractions. The intensity of {1-210} diffractions were lower than {0-110} diffractions, indicating single layer sheet or random stacks of single layer sheets for the sample.⁴² TEM images showed the N-graphene was very flexible with tiny wrinkles on the surfaces

⁴⁵ (Figure 2). Raman spectra were also applied to characterize the graphene oxide, chemically-reduced graphene and N-graphene. G peak of Raman spectra shifted from ~1602 cm⁻¹ for graphene oxide to ~1589 cm⁻¹ for chemically-reduced graphene (Figure)

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3).^{5a,5b} The G peak for chemically-reduced graphene shifted ⁵⁰ slightly to ~1592 cm⁻¹ for N-graphene (**Figure 3**). The ratio of I_D/I_G is even larger in chemically-reduced graphene than that in graphene oxide, indicating a large scale of residue defects after chemical reduction. The large scale residue defects enhanced the feasibility of direct doping nitrogen to chemically-reduced ⁵⁵ graphene even at low temperature.



Figure 2. TEM image and diffraction for N-graphene. (a) TEM image. (b) Electron diffraction pattern. c, Intensity profile of the cross line in b.



Figure 3. Raman characterizations of graphene oxide, graphene and N-graphene.

X-ray photoelectron spectra (XPS) measurements were then ⁶⁵ conducted to study the atomic species and chemical bonds in the N-graphene (**Figure 4**). Obvious N doping was observed in N doped graphene with N/C ratio of up to ~4.5% (**Figure 4a**). The N doping in graphene usually has several type of N dopants. The peak-fitting analysis of N 1s XPS spectra indicated that there ⁷⁰ were three fitting peaks in N 1s spectra and was consistent with previous studies (**Figure 4b**).¹⁸ N dopants include pyridinic N (~398.7 eV), pyrrolic N (~399.7 eV) and graphitic N (~401.4 eV) types with percentage of ~6.9%, 27.2% and 65.9%, respectively. The large part of graphitic N doping is induced by the large scale ⁷⁵ residue defects in the chemically-reduced graphene. Analysis of C 1s XPS spectra also showed the significant decrease of oxygenous groups both in chemically-reduced graphene and N doped graphene (**Figure 4c**), comparing with that in graphene oxide. The O/C atomic ratio decreased from ~50% in graphene oxide to 12% in chemically-reduced graphene, and to as low as ~7% in N-graphene. The present N-graphene has one of the ⁵ lowest oxygenous contents in low-temperature processed chemically-reduced graphene, which are normally achieved only in ultrahigh temperature processing under inert protected gas. Importantly, the N doping in graphene is stable in the solution and no obvious lose of N doping is observed even after over 6 ¹⁰ months storage. These excellent chemical characteristics in two-

step N-graphene are expected to improve the electronic transport in N-graphene by removing most oxygenous electron scattering and trapping centers.



- ¹⁵ Figure 4. XPS characterizations of N-graphene. (a) Survey scan of N-graphene. (b) Detail scan and multi-peak fitting of N 1s in N-graphene, (c) Comparison of C1s spectra of graphene oxide (I), graphene (II), and N-graphene (III).
- 20 The strategy can be further modified to prepare metal hydroxide ultrathin nanosheet nanocrystal/N-graphene hybrids for high performance supercapacitor electrodes. By introducing metal salt with NH₃•H₂O in the chemically-reduced graphene, metal hydroxide ultrathin nanosheet nanocrystal/N-graphene
- ²⁵ layered nanohybrids or heterostructures can be simultaneously achieved during the transformation of chemically-reduced graphene to N-graphene (Figure 1). As a demonstration, Ni(OH)₂/N-graphene nanohybrids were developed using this

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strategy. X-ray diffraction (XRD) results showed the nickel ³⁰ hydroxide nanocrystals were successfully included in the hybrid with N-graphene (**Figure 5a**). The XRD patterns are analyzed and applied to identify the resulted nickel compounds. The peaks at ~18° and ~39° indicate (001) and (002) diffraction planes of β phase Ni(OH)₂, respectively, while the peaks at ~12° and ~21° ³⁵ indicate the (003) and (006) diffraction planes of α phase Ni(OH)₂, consistent with previous reports on Ni(OH)₂.^{32,43} The XRD patterns in Ni(OH)₂ and Ni(OH)₂/N-graphene nanohybrids imply the resulted Ni(OH)₂ nanocrystals were a mixture of α and β phases. XPS spectra further confirm the existences of N doping ⁴⁰ in Ni(OH)₂/N-graphene nanohybrids with a N/C ratio of ~4% (**Figure 5b**). Usually the weight ratio of Ni(OH)₂ nanocrystals in



the Ni(OH)₂/N-graphene hybrids was ~70%.





Figure 6. TEM images of Ni(OH)₂ nanosheet nanocrystals (a, b) ⁵⁰ and Ni(OH)₂/N-graphene hybrid (c, d). Inset in a and c, electron diffraction patterns of Ni(OH)₂ nanocrystal (a) and Ni(OH)₂/Ngraphene hybrid (c).

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The morphologies of Ni(OH)₂ ultrathin nanosheet nanocrystals and Ni(OH)₂/N-graphene nanohybrids were further studied by transmission electron microscopy (TEM) (**Figure 6**). Ultrathin 5 Ni(OH)₂ nanocrystals were usually formed with a size from a



Figure 7. (a) Cyclic voltammetry curves of electrodes based on Ni(OH)₂, Ni(OH)₂/N-graphene mixture and Ni(OH)₂/N-graphene hybrid. (b) Cyclic voltammetry curve of electrodes based on N-¹⁰ graphene. Scan rates: 50 mV/s.



Figure 8. Electrochemical capacitive performance of Ni(OH)₂/Ngraphene hybrid electrodes from cyclic voltammetry measurements. (a) Cyclic voltammetry curves at different scan 15 rates. (b) Calculated specific capacitances at different scan rates.

few tens to a few hundreds nanometers with a thickness of a few nanometers (**Figure 6a and 6b**). In the presence of N-graphene, ultrathin $Ni(OH)_2$ nanocrystals were grown in situ on N-graphene

²⁰ and formed close contacts with N-graphene whose surfaces were mostly covered by Ni(OH)₂ nanosheet nanocrystals (Figure 6c and 6d). High-resolution TEM images demonstrate no obvious differences in atomic structure for pure Ni(OH)₂ nanocrystals and Ni(OH)₂ nanocrystals in Ni(OH)₂/N-graphene hybrids (Figure 6b ²⁵ and 6d), which is consistent with the XRD spectra (Figure 5a). The in situ growth of Ni(OH)₂ nanocrystal on N-graphene was further confirmed by the electron diffractions in pure Ni(OH)₂ and Ni(OH)₂/N-graphene hybrids (Inset in Figure 6a and 6c). Most importantly, this close contact will help the electronic ³⁰ transfer between nanosheet nanocrystals and N-graphene when forming supercapacitor electrodes. No G peak shift in Raman spectra was observed (Figure S1), implying no obvious damages of N-graphene were induced by the *in situ* growth and close contacts of Ni(OH)₂ nanocrystals.



Figure 9. Electrochemical capacitive performance of Ni(OH)₂/Ngraphene hybrid electrodes from discharge measurements. (a) Discharge curves at different discharge current of Ni(OH)₂/N-⁴⁰ graphene hybrid electrodes. (b) Three discharge cycles for Ni(OH)₂/N-graphene hybrid electrodes at discharge current of 1.5 A/g. (c) Calculated specific capacitance for electrodes based on Ni(OH)₂/N-graphene hybrid, pure Ni(OH)₂ nanocrystals, and Ni(OH)₂/N-graphene mixture at discharge current of 1.5 A/g. (d) ⁴⁵ Capacitance comparison of electrodes based on Ni(OH)₂/Ngraphene hybrid, pure Ni(OH)₂, and Ni(OH)₂/N-graphene mixture at different discharge currents.

The electrochemical performance of Ni(OH)2/N-graphene 50 hybrid electrodes were measured using cyclovometry (CV) and charge-discharge techniques. The CV measurements show Ni(OH)₂/N-graphene hybrids have a much higher current density than pure Ni(OH)₂ nanocrystals and Ni(OH)₂/N-graphene mixtures with the same amount of $Ni(OH)_2$ (Figure 7a). The 55 current density of electrodes based on pure N-graphene is two magnitudes of order lower than Ni(OH)2/N-graphene hybrids with neglectable capacitance (Figure 7b). The comparisons indicate the N-graphene play a very significant role in the hybrid Importantly, the Ni(OH)₂/N-graphene hybrid electrodes. 60 electrodes showed an over one order of magnitude higher current density than that in Ni(OH)₂/N-graphene mixture, implying that the close contact and connection between N-graphene and Ni(OH)2 nanosheet nanocrystals induced by in situ growth mediated by negatively-charged surfactant were very important

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for the performance of the hybrid electrodes. The specific capacitance of the hybrid supercapacitor electrodes can reach as high as ~1551 F/g at 2 mV/s (based on the mass of Ni(OH)₂, ~1086 F/g based on the mass of Ni(OH)₂/N-graphene hybrids),

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- s and ~1413 F/g at 5 mV/s. Even at a high san rate of 50 mV/s, the specific capacitance keeps at ~1046 F/g, ~70% of the original capacitance (**Figure 8**). Further increasing the scan rate up to 100 mV/s, the capacitance is still over 60% of the original capacitance. The control measurements of pure glassy carbon electrodes
- ¹⁰ demonstrate a neglectable tiny electrochemical current and capacitance (Figure S2). The galvanostatic charge-discharge measurements show a similar trend with the capacitance of ~1536 F/g at 1.5 A/g (based on the mass of Ni(OH)₂, ~1075 F/g based on the mass of Ni(OH)₂/N-graphene hybrids), ~1424 F/g at 3 A/g,
- ¹⁵ and 1125 F/g at 12 A/g, and kept over 60% of original capacitance even at 30 A/g (**Figure 9**). Moreover, the specific capacitance of the Ni(OH)₂/N-graphene hybrid electrodes is over 7 times higher than that in Ni(OH)₂/N-graphene mixture and over 60 times higher than that in pure Ni(OH)₂ nanocrystals (**Figure**
- ²⁰ 9b and 9c). The capacitance of the N-graphene/nanocrystal hybrid supercapacitor electrodes is higher than that of hybrid electrodes based on chemically-reduced graphene (~1267 F/g at 5 mV/s)³² and CVD-grown graphene foam (816 F/g at 5 mV/s).³³ More importantly, the Ni(OH)₂/N-graphene hybrid supercapacitor
- ²⁵ electrodes have excellent rate performance with ~70% of the original capacitance at a discharge current of 22 A/g while only ~14% at 22 A/g for Ni(OH)₂/N-graphene mixture and ~25% at 8 A/g for pure Ni(OH)₂ nanocrystals are observed (Figure 9d). The Ni(OH)₂/N-graphene hybrid supercapacitor electrodes also
 ³⁰ demonstrate good performance in cycling stability. The
- capacitance for electrodes increases progressively due to activation process in the beginning,^{26c} and has no obvious degradation for 2500 cycles (**Figure 10**).



³⁵ **Figure 10.** Cycling electrochemical stability for Ni(OH)₂/Ngraphene hybrid electrodes.

The ultrahigh capacitance and excellent rate performance and good cycling stability are partially attributed to the high quality of ⁴⁰ N-graphene and smooth electron transfer between in situ grown nanosheet nanocrystals and N-graphene. The electron transfer properties in the Ni(OH)₂/N-graphene hybrid electrodes were investigated by the electrochemical impedance spectra (**Figure 11**). The Ni(OH)₂/N-graphene hybrid has much smaller electron

⁴⁵ transfer resistance than that in pure Ni(OH)₂ nanocrystals and Ni(OH)₂/N-graphene mixture (**Figure 11**). The electron transfer resistance (R_{et}) is less than 200 ohm for Ni(OH)₂/N-graphene

hybrids while at the level of ~15000 ohm for Ni(OH)₂/Ngraphene mixtures and at the level of ~80000 ohm for pure ⁵⁰ Ni(OH)₂ nanocrystals. In general, electron transfer in Ni(OH)₂/Ngraphene hybrid electrodes is over 100 times faster than that in pure Ni(OH)₂ nanocrystals, and several tens of times faster than that in Ni(OH)₂/N-graphene mixture. The smooth electron transfer between nanosheet nanocrystals and N-graphene will ⁵⁵ improve the capacitance, high rate performance and stability in the electrodes as mentioned above. All these electrochemical measurements indicated that the solution-processable N-graphene nanohybrids could be an excellent platform for applications in supercapacitor electrodes, considering its low cost because of wet ⁶⁰ chemical synthesis and solution processability of N-graphene.



Figure 11. Electrochemical impedance spectra of Ni(OH)₂, Ni(OH)₂/N-graphene mixture and Ni(OH)₂/N-graphene hybrid electrodes. (a) Electrochemical impedance spectra of three ⁶⁵ electrodes. (b) Enlarge view of electrochemical impedance spectra of Ni(OH)₂/N-graphene hybrid electrode. Inset in b, analytical model for impedance spectra where Rs is resistance of electrolyte, Ret electron transfer resistance, C capacitance and Zw Warburg impedance. Frequency: 100 kHz-0.1 Hz.

3. Conclusion

A novel and facile two-step strategy was developed for preparing solution-processable, N-doped graphene and related layered nanohybrids with *in situ* grown Ni(OH)₂ ultrathin ⁷⁵ nanosheet nanocrystals at low temperature for supercapacitor electrodes. The obtained N-graphene has high N doping and lowest oxygenous content by low-temperature processing methods. N-graphene nanohybrids with Ni(OH)₂ nanosheet nanocrystals demonstrate ultrahigh capacitances of ~1551 F/g, ⁸⁰ excellent rate performance from 2 to 100 mV/s and from 1.5 A/g to 30 A/g and good cycling stability. The high electrochemical performance of N-graphene hybrid electrodes results from the unique electronic properties of N–graphene and ultrahigh enhancement of electron transfer between ultrathin nanosheet nanocrystals and N-graphene in the hybrid electrodes. The strategy is also compatible with other various nanocrystals such as transition metal and metal dichalcogenides nanocrystals. Thereby the low temperature, solution-processable, high quality

5 N-graphene and related nanohybrids may facilitate the large scale applications of N-graphene hybrids in energy storage and conversion.

4. Experimental

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¹⁰ Fabrication of N-graphene and Ni(OH)₂ nanosheet/N-graphene nanohybrids:

Graphite oxide was fabricated following the previous reports.⁵ The graphite oxide was dissolved in deionized (DI) water and exfoliated in the presence of sodium dodecyl benzene sulfonate

- (SDBS) using ultrasonic processing. The resulted S-graphene oxide was firstly chemically reduced by hydrazine at 80 °C under continuous stirring to achieve high quality chemically-reduced graphene. To prepare N-graphene, 1 mg/mL chemically-reduced graphene was dispersed in DI water then mixed with ammonia by
- ²⁰ a certain mixing weight ratio (ammonia/graphene=18) and transferred to a high pressure autoclave to keep the hydrothermal reactions at 150 °C for 3 hours. The obtained N-graphene was washed by deionized water and ethanol thoroughly. To prepare the Ni(OH)₂ nanosheet/N-graphene nanohybrids, 1 mg/mL
- ²⁵ chemically-reduced graphene was first mixed with nickel salt (Ni(NO₃)₂ or Ni(Ac)₂) for 30 min and then mixed with ammonia with a certain ratio (1 mg graphene for 18 mg ammonia and 0.034 mmol nickel salts) in DI water under continuous stirring. The resulted mixture was transferred to a high pressure autoclave for
- ³⁰ hydrothermal reactions at 150 °C for 3 hours. The mass of graphene before the Ni(OH)₂ nanocrystal growth and graphene nanohybrids after nanocrystal were measured by a highsensitivity electronic balance (0.01 mg sensitivity, Mettler Toledo, Model AB135-S) to determine the loading amount of
- as nanocrystals. The weight ratio of Ni(OH)₂ nanocrystals in Ni(OH)₂ nanosheet/N-graphene nanohybrids was \sim 70 wt.%. As a control, nickel salts and ammonia were mixed in the same ratio but without chemically-reduced graphene to prepare pure Ni(OH)₂ nanocrystals.

Characterizations and electrochemical measurements:

The TEM imaging was conducted by JEOL JEM-2010F transmission electron microscope with 200 kV accelerated voltage. The XPS spectra were collected from AXIS ultra DLD (Ching the Kertur) with the Verture of the transmission of transmission of the transmission of transmissi

- ⁴⁵ (Shimadzu Kratos) with a X-ray source of monochromatic Al-K α under 15kV and 10mA. XRD measurements were done by a Bruker D8 Advance diffractometer with a Cu-K α radiation source. Raman spectra were measured from Laser Raman Microscope Raman-11 with a 540 nm laser (Nanophoton Co., Japan). To
- ⁵⁰ prepare electrodes for electrochemical measurements, Ni(OH)₂/N-graphene hybrids, pure Ni(OH)₂ nanocrystals or Ni(OH)₂/N-graphene mixture was mixed with Nafion (Sigma-Aldrich, 5% solution in a mixture of water and lower aliphatic alcohols-IPA) followed by drop-drying on freshly polished glassy
- ⁵⁵ carbon. The CV and charge-discharge measurements were done in 30 wt.% KOH solution by an electrochemical station (Ivium Technologies) using Ag/AgCl as reference electrode and Pt wire as counter electrode. The Ni(OH)₂/N-graphene hybrid electrodes

were stabilized and activated by a number of electrochemical scan cycles before the measurements. The specific capacitance (C) is calculated by C = S/(2VUm) in CV measurements where S is area of CV curves, V scan rate, U potential window and m weight of Ni(OH)₂ in electrodes, or by C = I/(-dU/dt)m in discharge measurements where I is discharge current density, U potential ⁶⁵ window and t discharge time.^{31,32} Electrochemical impedance

spectra were collected by the same electrochemical station with a frequency from 100 kHz to 0.1 Hz.

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‡ Notes and references

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Highly nitrogen-doped graphene (N-graphene) based layered, quasi-two dimensional nanohybrids with ultrathin nanosheet nanocrystals has high performance in supercapacitor electrodes.