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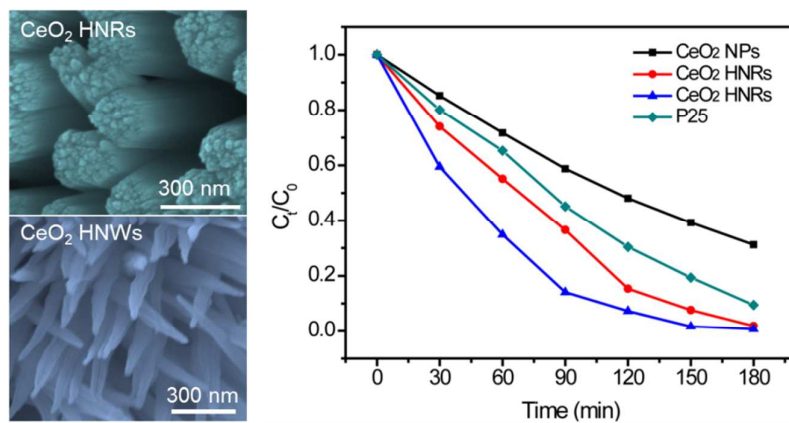
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## Facile Electrochemical Synthesis of CeO<sub>2</sub> Hierarchical Nanorods and Nanowires with Excellent Photocatalytic Activities



A simple, cost-effective and controllable electrochemical method has been developed to synthesize free-standing CeO<sub>2</sub> hierarchical nanorods and nanowires.

## ARTICLE

# Facile Electrochemical Synthesis of CeO<sub>2</sub> Hierarchical Nanorods and Nanowires with Excellent Photocatalytic Activities

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[www.rsc.org/](http://www.rsc.org/)Chao Zhang,<sup>a</sup> Xiyue Zhang,<sup>a</sup> Yichen Wang,<sup>a</sup> Shilei Xie,<sup>a</sup> Yi Liu,<sup>b</sup> Xihong Lu,<sup>a\*</sup> and Yexiang Tong<sup>a</sup>

In this work, we developed a simple, cost-effective and controllable electrochemical method to synthesize free-standing CeO<sub>2</sub> hierarchical nanorods and nanowires. Due to the hierarchical one-dimensional nanostructures and increased surface areas, both the CeO<sub>2</sub> hierarchical nanorods and nanowires exhibit substantially higher photocatalytic performance than the commercial CeO<sub>2</sub> nanoparticles in the degradation of methyl orange.

## Introduction

Rare earth oxides, one of the most important families of rare-earth materials, are a particularly interesting class of materials because of their unique optical, electronic, magnetic, and catalytic properties arising from the confinement of the 4f electrons.<sup>1</sup> These unique properties are crucial for many practical applications, such as optical communication, optical displays, efficient catalysis, UV shielding, and medical diagnostics.<sup>2-6</sup> As one of the most functional rare earth oxides, CeO<sub>2</sub> has attracted numerous attentions due to its excellent physical and chemical properties and wide applications in catalysts, fuel cells, UV blockers, solar cells, oxygen storage capacitors and sensors.<sup>7-11</sup> It is widely known that the photocatalytic, magnetic, electronic, and catalytic properties of CeO<sub>2</sub> are strongly size/shape dependent at the nanometer scale.<sup>9, 12, 13</sup> In this regard, considerable effort has been made to the design and synthesis of CeO<sub>2</sub> nanomaterials with controllable morphology and large surface area.<sup>14-19</sup> Among various nanostructures, one-dimensional (1D) CeO<sub>2</sub> nanostructures, such as nanorods, nanotubes, and nanowires have stimulated intensive interest because of their unique optical, electronic, and mechanical properties and potential applications in the fabrication of nanoscale devices.<sup>12, 20-23</sup> On the other hand, the assembly of 1D nanostructure into an appropriate hierarchical/superstructures has become the focus of research in recent years because they exhibit superior properties than the pristine 1D nanostructure.<sup>24-27</sup> For instance, compared with randomly dispersed nanorods, the highly ordered nanorod superstructures can allow for heterostructures, have larger surface areas, and can form continuous networks.<sup>28</sup> Therefore, once the nanorods as building blocks can be rationally assembled into appropriate three-dimensional (3D) superstructures, they will offer new scientific opportunities for investigating the influence of size and dimensionality with respect to their collective optical, magnetic, and electronic properties.<sup>24, 29</sup> Up to now, some CeO<sub>2</sub> hierarchical nanostructures such as hierarchical nanospheres,<sup>30, 31</sup> nanoflowers<sup>32-34</sup> and urchin-like structure<sup>35</sup> have been reported. Despite these achievements, a simple and efficient method for synthesizing shape-controlled CeO<sub>2</sub> 1D hierarchical nanostructures is still lacking.

Electrochemical deposition technique has been widely recognized as a promising alternating candidate for the preparation of nanomaterials because of its low cost, rapidness, environmentally friendly and potential advantage for large-scale production.<sup>36-38</sup> Moreover, the operational conditions (applied current or potential, composition of the electrolyte) usually allow to modify and control the deposit morphology through changes of mass transport and (electro-)crystallization phenomena.<sup>37, 39-41</sup> In recent years, electrochemical deposition technique has been also employed to the synthesis of CeO<sub>2</sub> nanostructures, and multifarious CeO<sub>2</sub> nanostructures including nanospheres,<sup>31</sup> nanorods,<sup>12</sup> nanowires,<sup>42</sup> nanotubes,<sup>43, 44</sup> and nanobelts<sup>45</sup> have been successfully synthesized by electrochemical deposition technique. Nevertheless, to the best of our knowledge, there are few reports on the preparation of CeO<sub>2</sub> hierarchical nanostructures via the electrochemical deposition technique so far.<sup>17, 19, 32</sup> In this work, we developed a facile and cost-effective electrochemical deposition method to prepare CeO<sub>2</sub> hierarchical nanorods (HNRs) and hierarchical nanowires (HNWs) and demonstrated their implementation as high-efficient photocatalysts for methyl orange (MO) degradation. This present electrochemical method allows us to easily manipulate the morphology and structure of the product, and free-standing CeO<sub>2</sub> HNRs and HNWs could be readily grown on Cu substrates by this electrochemical method. Moreover, both CeO<sub>2</sub> HNRs and HNWs exhibit substantially higher photocatalytic performance than the commercial CeO<sub>2</sub> nanoparticles (NPs) in the degradation of methyl orange (MO).

## Experimental

**Preparation of CeO<sub>2</sub> hierarchical nanorods (HNRs) and nanowires (HNWs):** All reagents used in this work were of analytical grade and were used directly without any purification. All the electrochemical deposition experiments were carried out in a conventional three-electrode glass cell by galvanostatic electrolysis. The working electrode is a Cu foil of 1.5 cm × 3 cm, while the counter electrode and reference electrode is a graphite rod of about 4.0 cm<sup>2</sup> and a saturated Ag/AgCl electrode, respectively. Before

electrodeposition, the Cu foil was cleaned ultrasonically in distilled water, ethanol, and acetone and then rinsed in distilled water again. Electrodeposition of CeO<sub>2</sub> HNRs were performed in solution of 0.01 M Ce(NO<sub>3</sub>)<sub>3</sub> + 0.1 M NH<sub>4</sub>Cl+0.05 M KCl with a current density of 0.44 mA cm<sup>-2</sup> for 2 h at 70 °C, while that of HNWs were carried out in the same electrolyte with a current density of 0.88 mA cm<sup>-2</sup> for 2 h at 70 °C.

**Structural and componential Characterizations:** The morphology and structure of products were characterized by scanning electron microscope (SEM, Quanta 400), X-ray diffraction (XRD, Bruker, D8 ADVANCE) and transmission electron microscopy (TEM, JEM2010-HR). The composition and surface oxidation states of products were studied by X-ray Photoelectron Spectroscopy (XPS, ESCALab250) with 200 W Al KR radiation in twin anode. The optical properties of the products were measured with a UV-Vis-NIR Spectrophotometer (UV, Shimadzu UV-3150). The BET surface area of samples was characterized by nitrogen adsorption/desorption isotherms at 77 K that conducted on an ASAP 2020 V3.03 H instrument. All the samples (powders) were outgassed at 100 °C for 300 min under flowing nitrogen before measurements. For the SEM, XRD and UV-spectra of CeO<sub>2</sub> samples, measurements were directly carried out on the samples without removing the Cu substrates. For the measurements of the TEM, BET and photocatalytic methyl orange (MO), CeO<sub>2</sub> films were exfoliated and ground into powders, then used the powders for these tests.

**Photocatalytic tests:** Photocatalytic properties of the CeO<sub>2</sub> HNRs and HNWs were evaluated using methyl orange (MO) as a demonstration dye under Xe lamp (500 W, Beijing Changtuo, PLSLAX500, wavelength: 200-800 nm) irradiation at room temperature with cooling water. Typically, 20 mg of CeO<sub>2</sub> samples (powders) was suspended in 100 ml of 20 mg/L MO aqueous solution (pH = 3). The suspension was magnetically stirred in the dark for 24 h to ensure establishment of an adsorption-desorption equilibrium before light irradiation. Then, at different light irradiation intervals, 3 ml of the suspensions was collected, filtered through a 0.45 μm membrane, and finally analyzed by a Shimadzu UV-3150a UV-Vis-NIR Spectrophotometer immediately.

## Results and discussion

Electrodeposition of CeO<sub>2</sub> HNRs was performed in solution of 0.01 M Ce(NO<sub>3</sub>)<sub>3</sub> + 0.1 M NH<sub>4</sub>Cl+0.05 M KCl with a current density of 0.44 mA cm<sup>-2</sup> for 2 h at 70 °C. Figure 1 displays the typical scanning electron microscopy (SEM) images of the CeO<sub>2</sub> HNRs. It can be clearly observed that CeO<sub>2</sub> flower-like nanostructures consisting of high density nanorods were successfully grown on Cu substrates. These CeO<sub>2</sub> nanorods are very uniform and have a diameter of around 200-250 nm. More interestingly, the CeO<sub>2</sub> nanorods were constructed by a large number of CeO<sub>2</sub> nanowires with a small size (Figure 1c). To identify the crystalline phase of the product, X-ray diffraction (XRD) spectrum of CeO<sub>2</sub> HNRs was collected and shown in Figure 1d. Besides the Cu peaks coming from the Cu substrates, all of the peaks in the XRD spectrum can be well indexed to a fluorite cubic structure of CeO<sub>2</sub> (JCPDF # 65-2975) with lattice constants  $a = 0.5411$  nm. No any other peaks were detected, indicating the product is highly pure. Additionally, the relative strong intensity of CeO<sub>2</sub> peaks suggests the CeO<sub>2</sub> HNRs possess a good crystallinity.

To investigate the microstructure of CeO<sub>2</sub> HNRs, transmission electron microscopy (TEM) analysis was performed. Figure 2a is a typical TEM image of a single CeO<sub>2</sub> HNR, clearly showing the diameter of the nanorod is about 200 nm, which is in accordance with the SEM result. The enlarged TEM image (inset in Figure 2a) reveals this CeO<sub>2</sub> nanorod was made up of numerous nanowires. The

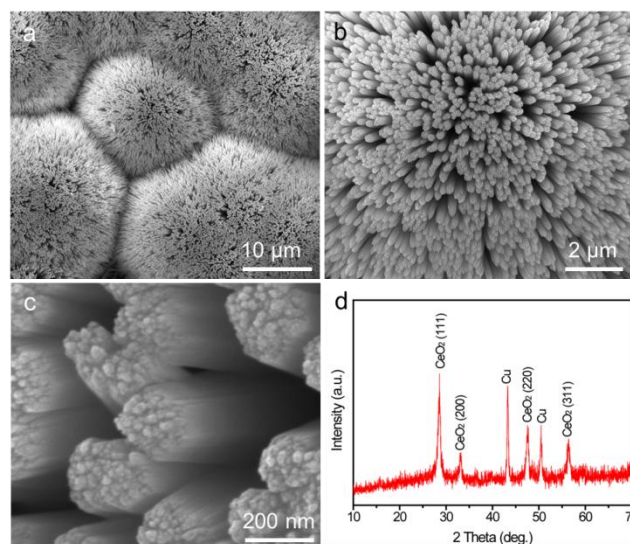


Figure 1. (a-c) SEM images and (d) XRD spectrum of CeO<sub>2</sub> HNRs prepared with a current density of 0.44 mA cm<sup>-2</sup>.

corresponding select area electron diffraction (SAED) pattern of this CeO<sub>2</sub> nanorod is inserted in Figure 2a. The bright diffraction spots confirm the highly crystalline nature of this nanorod, and more evidences can be seen from the HRTEM observation.<sup>15</sup> Figure 2b presents the high-resolution TEM (HRTEM) image of the CeO<sub>2</sub> nanorod. The well-resolved two-dimensional lattice fringes again confirm that the CeO<sub>2</sub> nanorod has a high crystallinity.<sup>15, 18</sup> The lattice fringe with a d-spacing of 0.31 nm matches well with the (111) interplanar spacing of face-centered-cubic CeO<sub>2</sub>.

X-ray photoelectron spectroscopy (XPS) analysis was also conducted to further study the composition of CeO<sub>2</sub> HNRs. The XPS survey spectrum of CeO<sub>2</sub> HNRs is presented in Figure 3a. Only Ce, C and O signals are observed on the surface of the sample, which indicates the high purity of the CeO<sub>2</sub> HNRs because the C single originates from the adventitious carbon. Figure 3b displays the Ce 3d core level XPS spectrum of CeO<sub>2</sub> HNRs, which can be deconvoluted to eight peaks. The peaks labeled as v<sub>1</sub> and u<sub>1</sub> are the characteristic peaks of Ce<sup>3+</sup> states while others are the characteristic peaks of Ce<sup>4+</sup> states.<sup>46</sup> This result indicates that Ce<sup>4+</sup> ions are dominant in the deposit and a small quantity of Ce<sup>3+</sup> is also present in this deposit. Figure 3c shows the O 1s core level XPS spectrum of CeO<sub>2</sub> HNRs, which can be deconvoluted into two peaks. The peak centered at 529.4 eV is assigned to the oxygen bond of Ce–O–Ce, and the peak located at 531.3 eV corresponds to Ce–O–H, respectively.<sup>47</sup> The XPS results mentioned above reveal that the as-deposited product is mainly CeO<sub>2</sub>, which consists with the XRD and TEM analyses.<sup>20, 45</sup>

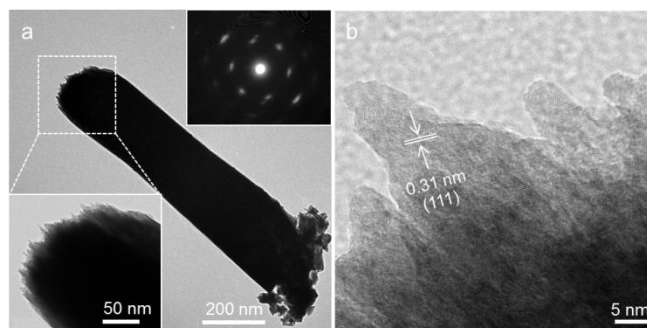


Figure 2. (a) TEM image and (b) HRTEM image of the CeO<sub>2</sub> nanorods. The insets are the enlarged TEM image and corresponding SAED pattern.

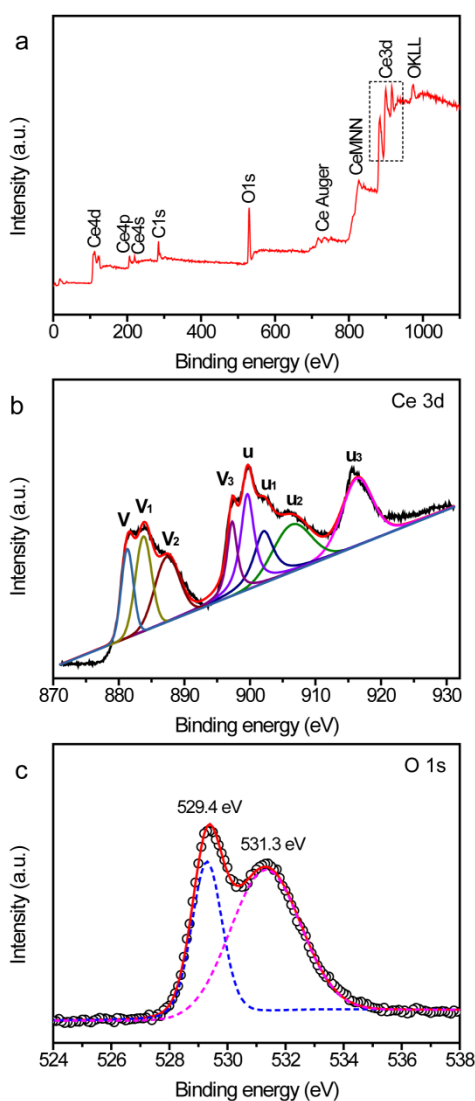


Figure 3. (a) XPS survey spectrum, (b) Ce 3d core level XPS spectrum and (c) O 1s core level XPS spectrum of the as-prepared CeO<sub>2</sub> HNRs.

Our present electrochemical method allows further structural manipulation. As shown in Figure 4a, a new kind of the CeO<sub>2</sub> spear-like hierarchical nanowires (denoted as HNWs) was successfully obtained when simply increased the deposition current density to 0.88 mA cm<sup>-2</sup>. The enlarged SEM images show that these CeO<sub>2</sub> HNWs with a diameter of about 100 nm vertically grew on Cu substrates (Figure 4b and the inset). A typical TEM image of single CeO<sub>2</sub> HNW is shown in Figure 4c. The spear-like morphology of the HNW is distinctly observed, and the diameter of this HNW (measured in middle) is about 80 nm. Figure 4d is the enlarged TEM image taken from the red rectangle part of the HNW (Figure 4c), showing this HNWs was composed of several small size nanowires (indicated by red arrows). Figure 4e displays the HRTEM image of CeO<sub>2</sub> HNW taken from the red circle part. The well-resolved lattice fringes clearly demonstrate that the CeO<sub>2</sub> HNW also has a high degree of crystallinity. The lattice fringe with the spacing of 0.31 nm corresponds to the interplanar spacing of (111) plane spacing of cubic CeO<sub>2</sub>. Additionally, SAED analysis confirms the CeO<sub>2</sub> HNW is highly crystalline (inset in Figure 4e).

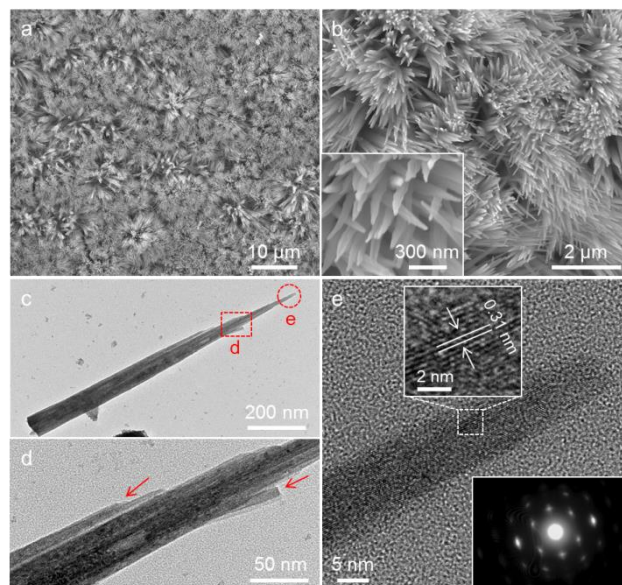


Figure 4. (a-b) SEM images, (c, d) TEM images and (e) HRTEM image of CeO<sub>2</sub> HNWs prepared with a current density of 0.88 mA cm<sup>-2</sup>. The insets in Figure 4e are the enlarge HRTEM image and the corresponding SAED pattern.

Figure 5a shows the nitrogen adsorption–desorption isotherms of the CeO<sub>2</sub> HNRs and HNWs collected at 77 K. Both the CeO<sub>2</sub> HNRs and HNWs exhibited typical Langmuir IV characteristic with an obvious hysteresis loop, suggesting that there are mesopores in CeO<sub>2</sub> HNRs and HNWs. The specific surface areas of CeO<sub>2</sub> HNRs and HNWs calculated by the Brunauer–Emmett–Teller (BET) method are 58.6 and 69.4 m<sup>2</sup> g<sup>-1</sup>, respectively. Figure 5b shows the UV-visible absorption spectra of the CeO<sub>2</sub> HNRs and HNWs. Both the samples exhibited a similar absorption band edge at about 390 nm, which is consistent with the CeO<sub>2</sub> band-gap of 3.2 eV.<sup>7, 12</sup> In comparison to the CeO<sub>2</sub> HNRs, the CeO<sub>2</sub> HNWs displays higher absorption in the wavelength range between 400 and 750 nm, suggesting the CeO<sub>2</sub> HNWs could utilize visible light.

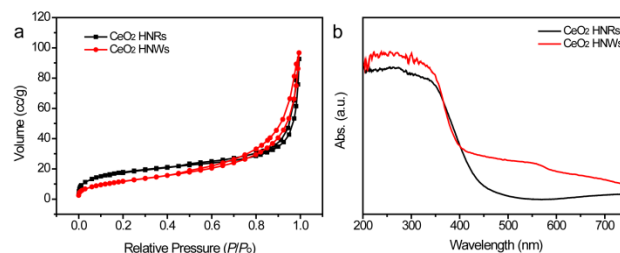


Figure 5. (a) Nitrogen adsorption–desorption isotherms of the CeO<sub>2</sub> HNRs and HNWs at 77 K.

To study the photocatalytic activities of the as-prepared CeO<sub>2</sub> samples, the efficiencies of photocatalytic degradation of MO under light irradiation were measured. For better comparison, commercial CeO<sub>2</sub> nanoparticles (NPs) and TiO<sub>2</sub> nanoparticles (P25, Degussa) were also examined in the same test. Before the photocatalytic tests, the CeO<sub>2</sub> samples were magnetically stirred in the dark for 12 h to reach the adsorption–desorption equilibrium. Figure 6 compares the photodegradation efficiencies of MO as a function of irradiation time for different samples. The y-axis was defined as C<sub>t</sub>/C<sub>0</sub>, where C<sub>t</sub> was the main absorption peak intensity of MO sampled at each irradiated time interval and C<sub>0</sub> was the absorption intensity of the starting concentration. Significantly, both the CeO<sub>2</sub> HNRs and HNWs

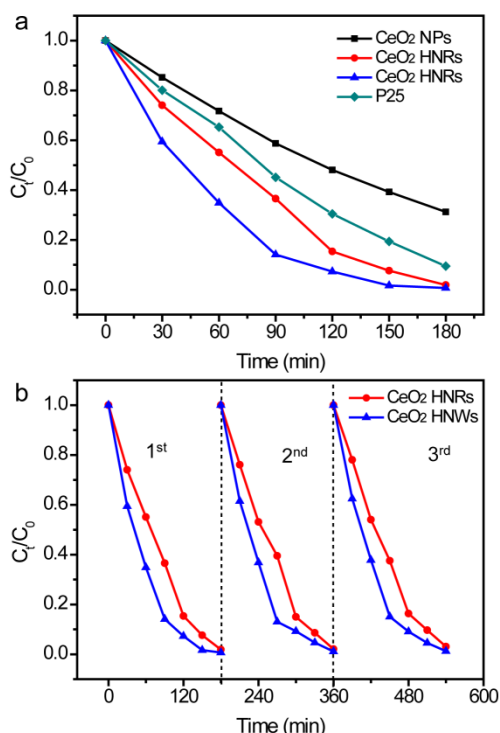


Figure 6. (a) Photodegradation efficiencies of MO for commercial CeO<sub>2</sub> NPs, P25, CeO<sub>2</sub> HNRs and CeO<sub>2</sub> HNWs as a function of light irradiation time. (b) Cycling performance of CeO<sub>2</sub> HNRs and CeO<sub>2</sub> HNWs for MO degradation.

exhibited substantially faster degradation rate of MO than the commercial CeO<sub>2</sub> NPs and P25. After 180 min light irradiation, the photocatalytic degradation efficiencies of MO solution for CeO<sub>2</sub> HNRs and HNWs achieved 98.2% and 99.3%, while the commercial CeO<sub>2</sub> NPs and P25 only reached 68.8% and 89.5% respectively. Moreover, the CeO<sub>2</sub> HNRs and HNWs have good cycling stability. As shown in Figure 6b, both the CeO<sub>2</sub> HNRs and HNWs exhibited no any decay of their photocatalytic activities after three cycles. The BET surface area of CeO<sub>2</sub> HNRs (58.6 m<sup>2</sup> g<sup>-1</sup>) and HNWs (69.4 m<sup>2</sup> g<sup>-1</sup>) is higher than the commercial CeO<sub>2</sub> NPs (50.2 m<sup>2</sup> g<sup>-1</sup>). Additionally, these hierarchical 1D nanorods and nanowires not only can provide large interface areas for photocatalytic reaction, but also shorten the diffusion pathway for minority carriers.<sup>48-50</sup> Therefore, such superior photocatalytic performances of CeO<sub>2</sub> HNRs and HNWs are mainly attributed to their hierarchical 1D nanostructures and higher surface areas. On the other hand, compared to the CeO<sub>2</sub> HNRs, the CeO<sub>2</sub> HNWs show a faster degradation rate of MO. As indicated by Figure 5b, the CeO<sub>2</sub> HNWs have stronger absorption than the CeO<sub>2</sub> HNRs in visible light region, and hence the CeO<sub>2</sub> HNWs may absorb more light and generate more electrons and holes than the CeO<sub>2</sub> HNRs at the same time. The superior photoactivity of CeO<sub>2</sub> HNWs than the HNRs is due to its larger surface area and improved light-harvesting ability.

## Conclusions

In summary, free-standing CeO<sub>2</sub> HNRs and HNWs were successfully synthesized on Cu substrates by a simple, low cost and environmentally friendly electrochemical deposition approach. The morphology and structure of the products can be readily controlled by simply adjusting the current density. Furthermore, both the CeO<sub>2</sub> HNRs and HNWs could remove 98.2% and 99.3% of MO under light irradiation for 180 min, which is substantially higher than the commercial CeO<sub>2</sub> NPs (68.8%) and P25 (89.5%). The superior

photocatalytic performance can be attributed to their increased surface areas and hierarchical 1D nanostructures that provide more active sites for photocatalytic reaction and shorter diffusion length for minority carriers. This work not only demonstrates a facile electrochemical method to synthesize CeO<sub>2</sub> hierarchical 1D nanostructures, but also may open up new opportunities in the design and preparation of high-efficient photocatalysts.

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## Notes and references

<sup>a</sup> MOE of the Key Laboratory of Bioinorganic and Synthetic Chemistry, KLGHEI of Environment and Energy Chemistry, School of Chemistry and Chemical Engineering, Institute of Optoelectronic and Functional Composite Materials, Sun Yat-Sen University, Guangzhou 510275, China. E-mail: luxh6@mail.sysu.edu.cn

<sup>b</sup> School of Chemistry and Chemical Engineering, GuangDong Pharmaceutical University, Guangzhou 510006, China

† Footnotes should appear here. These might include comments relevant to but not central to the matter under discussion, limited experimental and spectral data, and crystallographic data.

Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/b000000x/

- Z.-G. Yan and C.-H. Yan, *J. Mater. Chem.*, 2008, **18**, 5046-5059.
- G.-y. Adachi and N. Imanaka, *Chem. Rev.*, 1998, **98**, 1479-1514.
- R. Li, S. Yabe, M. Yamashita, S. Momose, S. Yoshida, S. Yin and T. Sato, *Mater. Chem. Phys.*, 2002, **75**, 39-44.
- R. Li, S. Yabe, M. Yamashita, S. Momose, S. Yoshida, S. Yin and T. Sato, *Solid State Ionics*, 2002, **151**, 235-241.
- M. V. Ganduglia-Pirovano, A. Hofmann and J. Sauer, *Surf. Sci. Rep.*, 2007, **62**, 219-270.
- S. Yin, Y. Minamidate, S. Tonouchi, T. Goto, Q. Dong, H. Yamane and T. Sato, *RSC Adv.*, 2012, **2**, 5976-5982.
- X. Lu, T. Zhai, H. Cui, J. Shi, S. Xie, Y. Huang, C. Liang and Y. Tong, *J. Mater. Chem.*, 2011, **21**, 5569-5572.
- C. W. Sun, H. Li and L. Q. Chen, *Energy Environ. Sci.*, 2012, **5**, 8475-8505.
- D. Zhang, X. Du, L. Shi and R. Gao, *Dalton Trans.*, 2012, **41**, 14455-14475.
- H. Yu, Y. Bai, X. Zong, F. Tang, G. M. Lu and L. Wang, *Chem. Commun.*, 2012, **48**, 7386-7388.
- X. Zhao, Y. Du, W. Ye, D. Lu, X. Xia and C. Wang, *New J. Chem.*, 2013, **37**, 4045-4051.
- X. H. Lu, D. Z. Zheng, P. Zhang, C. L. Liang, P. Liu and Y. X. Tong, *Chem. Commun.*, 2010, **46**, 7721-7723.
- J. Zhang, H. Kumagai, K. Yamamura, S. Ohara, S. Takami, A. Morikawa, H. Shinjoh, K. Kaneko, T. Adschiri and A. Suda, *Nano Lett.*, 2011, **11**, 361-364.

14. L. González-Rovira, J. M. Sánchez-Amaya, M. López-Haro, E. Del Rio, A. B. Hungria, P. Midgley, J. J. Calvino, S. n. Bernal and F. J. Botana, *Nano Lett.*, 2009, **9**, 1395-1400.
15. X. H. Lu, X. Huang, S. L. Xie, D. Z. Zheng, Z. Q. Liu, C. L. Liang and Y. X. Tong, *Langmuir*, 2010, **26**, 7569-7573.
16. Q. Yuan, H. H. Duan, L. L. Li, Z. X. Li, W. T. Duan, L. S. Zhang, W. G. Song and C. H. Yan, *Adv. Mater.*, 2010, **22**, 1475-1478.
17. Y. Zheng, K. Liu, H. Qiao, Y. Zhang, Y. Song, M. Yang, Y. Huang, N. Guo, Y. Jia and H. You, *Crystengcomm*, 2011, **13**, 1786-1788.
18. Z. Ji, X. Wang, H. Zhang, S. Lin, H. Meng, B. Sun, S. George, T. Xia, A. E. Nel and J. I. Zink, *ACS Nano*, 2012, **6**, 5366-5380.
19. F. Meng, F. Lu, L. Wang, J. Cui and J. Lu, *Sci. Adv. Mater.*, 2012, **4**, 1018-1023.
20. G. R. Li, D. L. Qu, L. Arurault and Y. X. Tong, *J. Phys. Chem. C*, 2009, **113**, 1235-1241.
21. Z.-R. Tang, Y. Zhang and Y.-J. Xu, *RSC Adv.*, 2011, **1**, 1772-1777.
22. P. Pal, S. K. Pahari, A. Sinhamahapatra, M. Jayachandran, G. M. Kiruthika, H. C. Bajaj and A. B. Panda, *RSC Adv.*, 2013, **3**, 10837-10847.
23. R. Pérez-Hernández, G. Mondragon-Galicia, A. Allende and J. Palacios, *Phys. Chem. Chem. Phys.*, 2013, **15**, 12702-12708.
24. J. N. Tiwari, R. N. Tiwari and K. S. Kim, *Prog. Mater. Sci.*, 2012, **57**, 724-803.
25. C. Z. Yao, B. H. Wei, H. X. Ma, H. Li, L. X. Meng, X. S. Zhang and Q. J. Gong, *J. Power Sources*, 2013, **237**, 295-299.
26. B. Liu, P. Soares, C. Checkles, Y. Zhao and G. Yu, *Nano Lett.*, 2013, **13**, 3414-3419.
27. X. Luo, Z. Lou, L. Wang, X. Zheng and T. Zhang, *New J. Chem.*, 2014, **38**, 84-89.
28. S. Xie, X. Lu, T. Zhai, J. Gan, W. Li, M. Xu, M. Yu, Y.-M. Zhang and Y. Tong, *Langmuir*, 2012, **28**, 10558-10564.
29. L. S. Zhong, J. S. Hu, H. P. Liang, A. M. Cao, W. G. Song and L. J. Wan, *Adv. Mater.*, 2006, **18**, 6.
30. H. Y. Xiao, Z. H. Ai and L. Z. Zhang, *J. Phys. Chem. C*, 2009, **113**, 16625-16630.
31. X. W. Ouyang, W. Li, S. L. Xie, T. Zhai, M. H. Yu, J. Y. Gan and X. H. Lu, *New J. Chem.*, 2013, **37**, 585-588.
32. R. Yu, L. Yan, P. Zheng, J. Chen and X. Xing, *J. Phys. Chem. C*, 2008, **112**, 19896-19900.
33. J. J. Wei, Z. J. Yang, H. X. Yang, T. Sun and Y. Z. Yang, *Crystengcomm*, 2011, **13**, 4950-4955.
34. N. S. Arul, D. Mangalaraj and T. W. Kim, *Appl. Phys. Lett.*, 2013, **102**, 223115.
35. J. Wei, Z. Yang and Y. Yang, *Crystengcomm*, 2011, **13**, 2418-2424.
36. X. Lu, D. Zheng, T. Zhai, Z. Liu, Y. Huang, S. Xie and Y. Tong, *Energy Environ. Sci.*, 2011, **4**, 2915-2921.
37. G.-R. Li, H. Xu, X.-F. Lu, J.-X. Feng, Y.-X. Tong and C.-Y. Su, *Nanoscale*, 2013, **5**, 4056-4069.
38. H. Li, C. Yao, L. Meng, H. Sun, J. Huang and Q. Gong, *Electrochim. Acta*, 2013, **108**, 45-50.
39. Y. T. Kim, J. H. Han, B. H. Hong and Y. U. Kwon, *Adv. Mater.*, 2010, **22**, 515-518.
40. X. Lu, X. Huang, S. Xie, T. Zhai, C. Wang, P. Zhang, M. Yu, W. Li, C. Liang and Y. Tong, *J. Mater. Chem.*, 2012, **22**, 13357-13364.
41. H. Li, C. Chen, X. Huang, Y. Leng, M. Hou, X. Xiao, J. Bao, J. You, W. Zhang, Y. Wang, J. Song, Y. Wang, Q. Liu and G. A. Hope, *J. Power Sources*, 2014, **247**, 915-919.
42. X. H. Lu, D. Z. Zheng, J. Y. Gan, Z. Q. Liu, C. L. Liang, P. Liu and Y. X. Tong, *J. Mater. Chem.*, 2010, **20**, 7118-7122.
43. M. Xu, S. L. Xie, X. H. Lu, Z. Q. Liu, Y. Y. Huang, Y. F. Zhao, J. Q. Ye and Y. X. Tong, *J. Electrochem. Soc.*, 2011, **158**, E41-E44.
44. C.-h. Zeng, S. Xie, M. Yu, Y. Yang, X. Lu and Y. Tong, *J. Power Sources*, 2014, **247**, 545-550.
45. G.-R. Li, D.-L. Qu and Y.-X. Tong, *Electrochem. Commun.*, 2008, **10**, 80-84.
46. P. Burroughs, A. Hamnett, A. F. Orchard and G. Thornton, *J. Chem. Soc., Dalton Trans.*, 1976, 1686-1698.
47. H.-I. Chen and H.-Y. Chang, *Solid State Commun.*, 2005, **133**, 593-598.
48. I. Gonzalez-Valls and M. Lira-Cantu, *Energy Environ. Sci.*, 2009, **2**, 19-34.
49. K. Nakata and A. Fujishima, *J. Photochem. Photobiology C: Photochem. Rev.*, 2012, **13**, 169-189.
50. Y. Wang, Q. Wang, X. Zhan, F. Wang, M. Safdar and J. He, *Nanoscale*, 2013, **5**, 8326-8339.