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**Assembly of Various Degrees of Interpenetration Co-MOFs Based on Mononuclear or Dinuclear Cluster Units: Magnetic Properties and Gas Adsoption** 

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Three new Co-based MOFs with a nanosized tetradentate pyridine ligand, N,N,N',N'-tetrakis(4-(4-pyridine)-phenyl) biphenyl-4,4'-diamine (TPPBDA) and carboxylate co-ligands,  $[Co(TPPBDA)(NO<sub>3</sub>)<sub>2</sub>]<sub>n</sub>•2H<sub>2</sub>O (1)$ ,  $[Co_{2}(TPPBDA)(bpdc)<sub>2</sub>$  $(H_2O)|_n$ •2DMA (2) and  $[Co(TPPBDA)_0, (hfipbb)(H_2O)]_n$ •3.5H<sub>2</sub>O (3)  $(H_2b)$ <sub>pdc</sub> = biphenyldicarboxylic acid,  $H_2$ hfipbb = 4,4'-(hexafluoroisopropylidene)bis-(benzoic acid), DMA = N,N-dimethylacetamide) have been synthesized under hydrothermal conditions. For complex **1**, a large cavity causes 4-fold interpenetration of the network, which can be classified as type IIIa mode of interpenetration. Complex **2** reveals a non-interpenetrating three-dimensional (3D) framework based on  $[C_{02}(\mu_2-H_2O)(CO_2)_2]$  unit. Complex **3** is also a 2-fold interpenetrating 3D net based on  $[Co_2(CO_2)_2]$  cluster. These mononuclear or dinuclear cluster units are interconnected by TPPBDA and carboxylate co-ligands, resulting in interesting structural diversities and various degrees of interpenetration.

#### **Introduction**

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Over the past decade, considerable efforts have been made toward the construction and investigation of metal-organic frameworks (MOFs), not only because of their exceptional coordination properties and structural diversity but also because of their

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<sup>†</sup> Electronic supplementary information (ESI) available: PXRD, UV-vis, the selected bond lengths and angles. CCDC: 1040560-1040562 for 1- **3**. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/

intriguing potential applications<sup>1</sup>. The interests in crystalline solids with designed structures and predictable properties continue to motivate research in MOFs in chemistry and material science today<sup>2</sup>. It is noteworthy that utilizing infinite rod-shaped or other secondary building units (SBUs) can reduce the interpenetration, which provides MOFs with permanent porosity and rigid architectures<sup>3</sup>. Although interpenetration in porous MOFs may greatly reduce the volume of pores, it is beneficial to maintain the host frameworks<sup>4</sup>. Thus, the influencing factors on the degree of interpenetration are a large concern for synthetic chemists. To the best of our knowledge, many other cases have been reported about the influencing factors on the interpenetration<sup>5</sup>. Nevertheless, the exploration of the degree of interpenetration controlled by adjusting the SBUs has not exploited well.

Recently, we reported a neutral tetradentate nanosized ligand, N,N,N',N'-tetrakis  $(4-(4-pyridine)-phenyl)$  biphenyl-4,4'-diamine (TPPBDA) (Scheme 1).<sup>6</sup> By introduction of different co-ligands, a remarkable range of materials containing various architectures have been prepared. Compared to the reported structures, the nanosized ligand may induce the interpenetration. As continuous exploitation on the study of Co(II)-based MOFs with TPPBDA, three new Co(II)-MOFs,  $[Co(TPPBDA)(NO<sub>3</sub>)<sub>2</sub>]<sub>n</sub>•2H<sub>2</sub>O (1), [Co<sub>2</sub>(TPPBDA)(bpdc)<sub>2</sub> (H<sub>2</sub>O)]<sub>n</sub>•2DMA (2) and$  $[Co(TPPBDA)_{0.5}(hfipbb)(H_2O)]_n$ <sup>•</sup>3.5H<sub>2</sub>O (3), have been successfully obtained, in which  $\text{Co}^{2+}$  cations act as the mononuclear or dinuclear SBUs. These mononuclear or dinuclear units are interconnected by TPPBDA and carboxylate co-ligands, resulting in interesting structural diversities and various degrees of interpenetration.



**Scheme 1**. Neutral tetradentate N-containing ligand and carboxylate co-ligands.

#### **Experimental**

#### **Materials and instrumentation**

All reagents were of analytical grade and used without further purification. The IR absorption spectra of the complexes were recorded in the range of  $400 - 4000 \text{ cm}^{-1}$  by means of a Nicolet (Impact 410) spectrometer with KBr pellets. C, H and N elemental analyses were carried out with a Perkin Elmer 240C elemental analyzer. Powder X-ray diffraction (PXRD) measurements were performed on a Bruker D8 Advance X-ray diffractometer using Cu-K $\alpha$  radiation (1.5418 Å), and the X-ray tube was operated at 40 kV and 40 mA. Temperature dependent magnetic susceptibility data for polycrystalline complex **2** were obtained on a MPMS XL-7 SQUID magnetometer under an applied field of 2000 Oe over the temperature range of 1.8-300 K. The gas sorption isotherms for complex **2** were measured by Micromeritics ASAP 2020 M+C surface area analyzer.

**Syntheses of the complexes 1-3:** A mixture of  $H_2O/DMA/CH_3CN$  containing the TPPBDA (79.6 mg, 0.1 mmol),  $Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O$  (58.2 mg, 0.2 mmol) or/and carboxylate co-ligands (0.1 mmol) was mixed in a Teflon vessel within the autoclave. The vessel was heated at 85 °C for 72 h and then cooled to room temperature. The large amounts of red crystals were obtained. Yields of the reaction were ca. 20%, 70%, 10% based on TPPBDA ligand, respectively. Elemental analysis calcd. for  $C_{56}H_{40}CoN_8O_6(H_2O)$  (1): C, 66.21%; H, 4.37%; N, 11.03%, Found: C, 66.18%; H, 4.35%; N, 11.06%. IR(KBr, cm<sup>-1</sup>): 3403(s), 3038(w), 1595(s), 1484(s), 1398(m), 1310(m), 1288(s), 1224(m), 1171(m), 1100(m), 1027(w), 998(m), 817(s), 748(m), 702(m), 563(w), 510(m). For  $C_{84}H_{58}Co_2N_6O_9(C_4H_9NO)$  (2): C, 69.60%; H, 4.82%; N, 7.06%. Found: C, 69.56%; H, 4.81%; N, 7.07%. IR(KBr, cm<sup>-1</sup>): 3399(s), 3031(w), 1595(s), 1517(s), 1486(s), 1388(m), 1321(s), 1291(m), 1175(m), 1086(m), 1039(w), 1006(w), 852(s), 813(s), 770(m), 671(m), 561(m), 522(w), 435(w). For C45H30CoF6N3O5(H2O)3.5 (**3**): C, 58.20%; H, 4.02%; N, 4.52%. Found: C, 58.36%; H, 4.45%; N, 4.67%. IR(KBr, cm<sup>-1</sup>): 3380(s), 3012(w), 1605(s), 1546(s), 1457(s), 1362(m), 1329(s), 1284(m), 1163(m), 1064(m), 1025(w), 819(s), 751(m), 545(w).

#### **X-ray crystallography**

Crystallographic data of **1**-**3** were collected on a Bruker Apex Smart CCD diffractometer with graphite-monochromated Mo-K<sub>α</sub> radiation ( $\lambda$  = 0.71073 Å) at 293 K using the  $\omega$ -scan technique. The intensity data were integrated by using the SAINT program. An empirical absorption correction was applied using the SADABS program.<sup>7</sup> The structures were solved by direct methods and refined anisotropically using full-matrix least-squares procedures based on  $F^2$  values with the SHELXTL-97 package of crystallographic software.<sup>8</sup> The hydrogen atoms were generated geometrically. Crystallographic data of complexes **1-3** are presented in Table 1, the selected bond lengths and angles are given in Tables S1-S3.

#### **Results and discussion**

#### **Description of structures for 1-3**

**Crystal structure of**  $[Co(TPPBDA)(NO<sub>3</sub>)<sub>2</sub>]<sub>n</sub>•2H<sub>2</sub>O$  **(1). The crystal structure** determination reveals that complex **1** crystallizes in orthorhombic crystal system of

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*Pbcm*. In the asymmetric unit, it contains half of one Co(II) cation, half of one TPPBDA ligand, and two halves of  $NO<sub>3</sub>$  anions, and two squeezed lattice water. The Co (II) ion is six-coordinated by four N atoms from four TPPBDA ligands and two NO<sub>3</sub> anions (Fig. 1a). The tetradentate TPPBDA ligands bridge the Co (II) ions into a 3D network (Fig. 1b). The size of this net is  $17.86 \times 17.90$ Å<sup>2</sup> along the *a* axis. We can consider the Co (II) and the TPPBDA ligands as square-planar and tetrahedral four-connected nodes, respectively, thus giving a short vertex symbol  $4^2.8^4$ , which is that of the pts topology,<sup>9</sup> as shown in Fig. 1c. It is worth noting that there are 3D channels running along the *a-*, *b-*, and *c*-axes in the structure of **1**. The framework **1** occupies 58% of the total crystal volume; the remaining space is occupied by the solvent molecules. Such a large cavity causes 4-fold interpenetration of the networks, which can be best described as two sets of a normal 2-fold net that is a  $[2 + 2]$  mode of interpenetration (Fig. 1d). According to Blatovs' classification which is mainly based on the symmetric relationships between the interpenetrating nets, $10$  the interpenetration of **1** can be classified as type IIIa,  $Z = 4[2 \cdot 2]$  ( $Zt = 2$ ;  $Zn = 2$ ).





**Fig. 1** (a) ORTEP drawing of **1** with 30% ellipsoid probability (hydrogen atoms are omitted for clarity), symmetry codes:  $\#1 = x$ , *y*, 1.5-*z*;  $\#2 = 1+x$ ,  $1 + y$ , *z*;  $\#3 = 1+x$ , 1 + *y*, 1.5-*z*. (b) A perspective of 3D framework **1** along the *a* axis. (c) A view of pts topology. (d) Schematic representation of 4-fold interpenetration.

#### **Crystal structure of**  $[Co_2(TPPBDA)(bpdc)_2 (H_2O)]_n \cdot 2DMA (2)$ **.**

Complex **2** crystallizes in a monoclinic space group *C*2/*c* and features a neutral three-dimensional porous framework. The framework of **2** consists of dinuclear  $[C_{O_2}(\mu_2-H_2O)(CO_2)_2]$  units where each Co center has  $CoO_4N_2$  octahedral coordination geometry and the Co···Co distance is 3.664(4) Å (Fig. 2a). Each  $[Co_2(\mu_2-H_2O)(CO_2)_2]$ unit is doubly bridged by two carboxylate groups and one water molecule, which is further coordinated by four pyridine N atoms and two carboxylate O atoms from two different bpdc<sup>2-</sup> ligands. The Co-O bond lengths are in the range  $2.0508(10)$ -  $2.1629(6)$ Å, and the Co-N bond lengths are  $2.1441(11)$  and  $2.1975(15)$  Å, which are comparable to those reported in the literature<sup>11</sup>. The TPPBDA and bpdc<sup>2-</sup> links  $[C_{O_2}(\mu_2-H_2O)(CO_2)_2]$  units to form a 3D structure (Fig. 2b). To reduce multidimensional structures to simple nodes and connection nets, these  $[C_{O_2}(\mu_2-H_2O)(CO_2)_2]$  units, acting as 6-connected nodes, are inter-linked by the 4-connected TPPBDA ligands and H2bpdc linkers into a 3D open framework with 4,6-connected topology (Fig. 2c). According to a calculation performed using PLATON, compound **2** contains a solvent accessible void space of 26% of the total



**Fig. 2** (a) ORTEP drawing of **2** with 30% ellipsoid probability (hydrogen atoms and lattice water molecules are omitted for clarity), symmetry codes:  $\#1 = 1 - x$ , *y*, 1.5-*z*; #2 = 0.5 + *x*, 0.5 - *y*, -0.5 + *z*. (b) A perspective of 3D framework **2**. (c) Schematic representation of topology.

#### **Crystal structure of**  $[Co(TPPBDA)_{0.5}(hfipbb)(H_2O)]_n \cdot 3.5H_2O(3)$ **.**

Complex **3** crystallizes in the monoclinic crystal system with one Co cation, half of  $TPPBDA$ , one hfipb $b<sup>2</sup>$  ligand, as well as one coordinated water molecule, and three and a half molecules of water in the asymmetric unit. Each Co(II) unit has an octahedral coordination geometry (Fig. 3a). The hfipbb<sup>2-</sup> ligand takes bidentate and monodentate coordination modes, two Co(II) ions are bridged by two carboxylate O atoms forming a four-membered square-shaped  $[Cd<sub>2</sub>(CO<sub>2</sub>)<sub>2</sub>]$  unit. The hfipbb<sup>2-</sup> anions connect to the  $[Cd_2(CO_2)_2]$  units to form 1D chains. Then the chains are linked by the

neutral tetradentate ligand to generate a 3D net (Fig. 3b).

Complex **3** is a three-dimensinal net composed of meso-helical chains, which are labeled as L and R chains (Fig. 3c). Similarly, the two symmetrically related helices coexist in the centrosymmetric solid, in which they show left-handed and right-handed enantiomorphs, respectively. However, the L and R type single meso-helical chains are arranged alternatively, which indicates the achirality feature, further confirmed by the achiral space group *C*2/*m*. The central axis is a 2-fold screw axis.

To better insight into the nature of the intricate framework, topological approach can be applied. As discussed above,  $[Co_2(CO_2)_2]$  cluster units are defined as 6-connected nodes. Likewise, TPPBDA ligands linking with four  $[Co_2(CO_2)_2]$  cluster units can act as 4-connected nodes. On the basis of the simplification principle, the resulting structure is a (4,6)-connected net with point symbol  $\{4^6\}\{4^2.6^8.8^5\}$  (Fig 3d). Compound **3** contains a solvent accessible void space of 42.5% of the total crystal volume. In order to minimize the big void cavities and stabilize the framework, the potential voids formed by a single 3D network show incorporation with one other identical network, thus giving a 2-fold interpenetrating network (Fig. 3e).





**Fig. 3** (a) ORTEP drawing of **3** with 30% ellipsoid probability (hydrogen atoms are omitted for clarity), symmetry codes: #1 = -0.5 + *x*, 0.5 + *y*, 1 + *z*; #2 = 1.5 - *x*, 0.5 - *y*, 2 - *z*; #3 = 0.5 + *x*, 0.5 - *y*, 1 + *z*. (b) A perspective of 3D framework **3**. (c) View of meso-helical chains (marked by L and R) in net **3**. (d) and (e) Schematic representation of topology and 2-fold interpenetration.

#### **Gas uptake properties of 2**

To check the permanent porosities,  $N_2$  sorption isotherms at 77 K were measured for the desolvated samples obtained by soaking **2** in acetone, and then vacuum-drying at 90℃ overnight. As shown in Fig. 4 (left), **2** displays typical type-I adsorption isotherms. 2 adsorbs 157 cm<sup>3</sup> $g^{-1}$  of N<sub>2</sub> at 77 K, and the Brunauer-Emmett-Teller (BET)

and Langmuir surface areas are  $468m^2g^{-1}$  and  $702m^2g^{-1}$ , respectively. Low-pressure CH4 uptakes of desolvated samples of **2** were also determined using volumetric gas adsorption measurements. The adsorption isotherms were fully reversible, and exhibit 14  $\text{cm}^3 \text{g}^{-1}$  CH<sub>4</sub> uptake under the conditions of 273 K and 850 mmHg. However, the adsorption amount of  $CO_2$  is  $61 \text{cm}^3 \text{g}^{-1}$  under the same conditions (Fig. 4, right). At room temperature and 850 mmHg, the  $CO_2$  uptake of 2 is 29 cm<sup>3</sup>g<sup>-1</sup> (Fig. 5, left). The isosteric heats of adsorption  $(Q_{st})$  were calculated from the  $CO<sub>2</sub>$  adsorption isotherms by using the Clausius-Clapeyron equation (Supporting Information), which is similar to values for some other MOF materials (Fig. 5, right).<sup>12</sup> The result shows that  $CO<sub>2</sub>$ and CH<sub>4</sub> isotherms at 273 K reach the maximums of  $61 \text{cm}^3 \text{g}^{-1}$  and  $14 \text{cm}^3 \text{g}^{-1}$ , respectively. When the pressure is increased, the  $CO<sub>2</sub>$  adsorption capacity greatly exceeded that of CH4. The results show that **2** has a potential application in gas separation processes of  $CH<sub>4</sub>/CO<sub>2</sub>$ .



**Fig. 4** Left: Nitrogen isotherms measured at 77 K (black and red represent adsorption and desorption branches, respectively). Right:  $CO<sub>2</sub>$  and  $CH<sub>4</sub>$  sorption isotherms measured at 273 K.



**Fig. 5** Left:  $CO<sub>2</sub>$  gas adsorption isotherms at 273 K and 298K. Right: The isosteric heats of adsorption were calculated from the  $CO<sub>2</sub>$  adsorption isotherms by using the Clausius-Clapeyron equation in complex **2**.

#### **Magnetic properties of 2**

Temperature dependent magnetic susceptibilities of **2** were investigated in the temperature range of 1.8-300 K. The temperature dependence of magnetic susceptibility of 2 in the forms of  $\chi_M T$  and  $\chi_M$  versus T is displayed in Fig 6, the experimental  $\chi_M T$  value equal to 4.81 cm<sup>3</sup> K mol<sup>-1</sup> at 300 K is greater than the expected value of 3.75 cm<sup>3</sup> K mol<sup>-1</sup> for two isolated high-spin Co(II) ions ( $g = 2$  and  $S = 3/2$ ) per formula because of the prominent orbital contribution contribution arising from the  $4T_{1g}$  ground state of Co(II). Upon cooling,  $\chi_M T$  continuously decreases and reaches  $0.73 \text{ cm}^3 \text{ K mol}^{-1}$  at 1.8K. The magnetic susceptibility data were fitted assuming that the carboxylate and oxygen bridges of the Co(II) ions form an isolated spin dimer system. The intermolecular magnetic coupling constant *zj*' was taken into account, the magnetic susceptibility from 7-300K was fitted with equation (1), which is deduced from the spin Hamiltonian,  $H = -2JS_1S_2$ : <sup>13</sup>

$$
\hat{H} = -2J\hat{S}_1\hat{S}_2 = -J(\hat{S}_T^2 - \sum_{i=1}^2 \hat{S}^2)
$$

$$
E(S_T) = -JS_T(S_T+1)
$$

$$
S_T = 0, 1, 2, 3
$$

$$
E(S_T) = 0, -2J, -6J, -12J
$$

$$
\chi_{\rm M} = \frac{Ng^2 \beta^2}{3kT} \times \frac{84e^{12J/kT} + 30e^{6J/kT} + 6e^{2J/kT}}{7e^{12J/kT} + 5e^{6J/kT} + 3e^{2J/kT} + 1}
$$
\n
$$
= \frac{2Ng^2 \beta^2}{kT} \times \frac{14e^{12J/kT} + 5e^{6J/kT} + e^{2J/kT}}{7e^{12J/kT} + 5e^{6J/kT} + 3e^{2J/kT} + 1}
$$
\n(1)

An excellent fit was obtained with  $J = -6.70 \text{ cm}^{-1}$ ,  $zj' = -1.07 \text{ cm}^{-1}$  and  $g = 2.17$  with an agreement factor  $R = 3.7 \times 10^{-7}$ . This result also indicates antiferromagnetic coupling between two  $Co<sup>H</sup>$  centers through two carboxylate bridges and two oxygen bridges.



**Fig. 6** Temperature dependence of  $\chi_M$  and  $\chi_M$ *T* for 2 (open circles and red line represent experimental data and fits).

#### **Conclusion**

In summary, three Co-MOFs have been successfully obtained, in which  $Co<sup>2+</sup>$  cations act as the mononuclear or dinuclear  $SBUs.<sup>14</sup>$  The interpenetration forms were well tuned by  $Co^{2+}$  cations as different mononuclear or dinuclear SBUs: a large cavity causes 4-fold interpenetration for mononuclear  $\text{Co}^{2+}$  cations in complex 1; complex 2 shows a non-interpenetrating 3D framework based on  $[Co_2(\mu_2-H_2O)(CO_2)_2]$  unit; whereas complex **3** reveals a 2-fold interpenetrating 3D framework with dinuclear  $[C<sub>02</sub>(CO<sub>2</sub>)<sub>2</sub>]$  unit. The magneto-chemical analysis of complex 2 indicates that there are dominant antiferromagnetic interactions between  $Co<sup>2+</sup>$  ions in these systems. In addition, complex  $2$  exhibits microporous sorption for  $N_2$ ,  $CO_2$  and  $CH_4$ .

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Complex	$1^{[a]}$	$2^{[a]}$	$3^{[a]}$
Empirical formula	$C_{56}H_{40}CoN_8O_6$	$C_{84}H_{58}Co_2N_6O_9$	$C_{45}H_{30}CoF_6N_3O_5$
Formula weight	979.89	1413.22	865.65
Crystal system	Orthorhombic	Monoclinic	Monoclinic
space group	Pbcm	C2/c	C2/m
$a(\AA)$	20.7677(15)	22.2737(18)	22.8097(18)
$b(\AA)$	11.7553(9)	29.146(2)	36.6950(15)
c(A)	28.509(2)	13.8719(12)	14.4183(11)
$\alpha$ <sup>o</sup> )	90	90	90
$\beta$ <sup>(°)</sup>	90	108.451(2)	106.313(2)
$\gamma$ <sup>(°)</sup>	90	90	90
Volume $(\AA^3)$	6959.9(9)	8542.6(12)	11582.3(14)
Z	$\overline{\mathcal{L}}$	4	8
$D_{c}$ , $g$ cm <sup>-3</sup>	0.935	1.099	0.993
Absorptioncoefficient,mm <sup>1</sup>	0.289	0.442	0.351
F(000)	2028	2920	3536
Reflections collected /	36797 / 7005	21281/7552	42053 / 14579
unique	$[R(int) = 0.0129]$	$[R(int) = 0.0364]$	$[R(int) = 0.0106]$
Data/restraints/paramers	7005 / 0 / 337	7552 / 378 / 511	14579 / 0 / 676
Goodness-of-fit on $F^2$	1.034	1.018	1.003
Final R indices $[I>2\sigma(I)]$	$R1^{[b]} = 0.0484$ ,	$R1^{[b]} = 0.0520,$	$R1^{[b]} = 0.0392,$
	$wR2^{[c]}=0.1235$	$wR2^{[c]} = 0.1448$	$wR2^{[c]} = 0.1383$
Largest diff.and hole $e.\AA^{-3}$	0.240 and -0.334	0.544 and -0.462	$0.291$ and $-0.268$

**Table 1** Crystal data and structure refinements parameters of complexes **1**, **2** and **3** 

[a] The residual electron densities were flattened by using the SQUEEZE option of PLATON.

[b] *R*1 = Σ|*|F*o|-|*F*c||/|Σ|*F*o|.

 $[c] wR2 = {\sum [w(F_0^2 - F_c^2)^2]}/{\sum [w(F_0^2)^2]} \cdot \frac{1}{2}$ ; where  $w = 1/[\sigma^2(F_0^2)+(aP)^2+bP]$ ,  $P = (F_0^2+2F_c^2)/3$ .

## **Assembly of Various Degrees of Interpenetration Co-MOFs Based on Mononuclear or Dinuclear Cluster Units: Magnetic Properties and Gas Adsoption**

Meng-Xi Zheng, Xiang-Jing Gao, Chuan-Lei Zhang, Ling Qin and He-Gen Zheng<sup>∗</sup>



The mononuclear or dinuclear Co cluster units are interconnected by mixed ligands, resulting in interesting structural diversity and various degrees of interpenetration.