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1 **Electrochemical Activation of Commercial Polyacrylonitrile-based Carbon Fiber for Oxygen**  
2 **Reduction Reaction†**

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9 **Abstract**

10 Nitrogen (N)-doped carbon and its non-noble metal composite replacing platinum-based oxygen reduction  
11 reaction (ORR) electrocatalysts still have some fundamental problems remained. Here the micron-sized  
12 commercial polyacrylonitrile-based carbon fiber (PAN-CF) electrode was modified by electrochemical  
13 method, converting its inherent pyridinic-N into 2-pyridone (or 2-hydroxyl pyridine) functional group  
14 existing in three-dimensional active layers with remarkable ORR catalytic activity and stability. Carbon  
15 atom adjacent to nitrogen and oxygen atom is prone to acting as an active site to efficiently catalyze a two-  
16 electron ORR process. However, after coordinating pyridone to Cu<sup>2+</sup> ion, together with the electrochemical  
17 reaction, the chemical redox between Cu<sup>+</sup> and ORR intermediates synergistically tends to a four-electron  
18 pathway in alkaline solution. In different medium, the complexation and dissociation can induce the charge  
19 transfer and reconstruction among proton, metal ion and pyridone functionalities, eventually leading to the  
20 changes of ORR performance.

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26 **Broader context**

27 Oxygen reduction reaction (ORR) is widely applied in both energy conversion devices (e.g., fuel cells,  
28 metal-air batteries) and environmental pollution control, which are serious challenges humanity will face  
29 for a long time. Four-electron ORR is mainly used as cathode reaction for energy storage devices (fuel  
30 cells, metal-air batteries, *etc*), while two-electron ORR is mainly for continuous in-situ generation of  $H_2O_2$   
31 in electro-Fenton process to degrade refractory pollutants. It is of great significance to investigate and  
32 control the selectivity of cathode materials for specific application. Here, we report for the first time that  
33 polyacrylonitrile-based carbon fiber (PAN-CF) was electrochemically activated, obtaining obviously  
34 enhanced ORR activity. We reveal that the ORR selectivity can be regulated from  $2e^-$  to  $4e^-$  pathway by  
35 forming one or two adjacent active site(s) in different medium. Thus, the systematical ORR mechanisms  
36 are useful for future design and preparation of catalyst and electrode.

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## 50 Introduction

51 High efficient oxygen reduction reaction (ORR) of four- or two-electron ( $4e^-$  or  $2e^-$ ) pathway is of great  
52 value in a variety of applications such as the clean power generation in proton exchange membrane fuel cell  
53 (PEMFC),<sup>1-3</sup> metal-air batteries,<sup>4,5</sup> the great reduction of energy consumption in chlor-alkali industry *via*  
54 replacing hydrogen-evolving cathode with an air cathode,<sup>6,7</sup> the yield of  $H_2O_2$  on line<sup>8</sup> and efficient  
55 degradation of persistent organic pollutants (POPs) *via* electro-Fenton process,<sup>9</sup> *etc.*

56 Platinum (Pt) has been considered as the most efficient electrocatalyst for  $4e^-$  ORR in fuel cells but  
57 with very high cost. Therefore, besides a large amount of research on improving Pt-based electrocatalysts,  
58<sup>10,11</sup> various non-noble metal alternative catalysts like metal- $N_4$  complexes (metal porphyrins, metal  
59 phthalocyanine), transition metal oxides and chalcogenides, N-doped carbon (carbon nanotube, graphene,  
60 fiber, nanocage, *etc*) have been developed.<sup>2, 12-15</sup> Among them, N-doped carbon and its non-noble metal  
61 composite are the most promising.<sup>16, 17</sup> However, the stability, durability, activity and cost of these  
62 materials for ORR are still inferior compared to Pt-based catalysts and unable to meet the demand of  
63 practical applications.<sup>2, 18</sup> Besides, these researches mainly focus on nano-scale materials *via* complex  
64 preparation procedure, and many controversies over the mechanism still remain, such as the categories of  
65 nitrogen groups, the roles of the medium and non-noble metal, and the correlation of N-doping structure  
66 with ORR performance, the corresponding research on micro-scale ones is far less enough yet.<sup>19-25</sup>

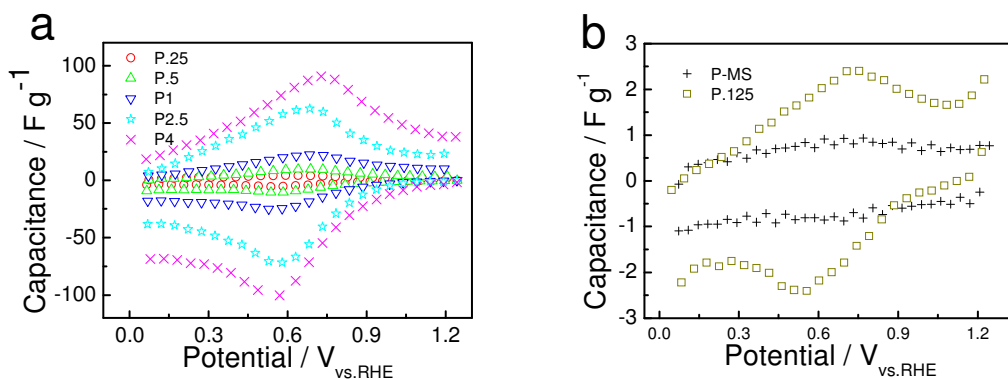
67 Commercially available polyacrylonitrile-based carbon fiber (PAN-CF) is a micro-sized N-containing  
68 carbon material (N content: 2~6 wt.%) *via* high-temperature carbonization (1200~1600 °C),<sup>26</sup> its low price  
69 and high electrical conductivity and structural stability have attracted much interest in flow batteries,<sup>27</sup> the  
70 dissolved oxygen seawater battery (SWB),<sup>28, 29</sup> *etc.* However, the poor ORR performance of PAN-CF  
71 cathode resulting in a low volume specific power of SWB ( $\sim 2.7 \text{ mW L}^{-1}$ ) has hampered its widespread use.  
72 Therefore, in this study, we intend to enhance the ORR activities of PAN-CF electrode *via* electrochemical  
73 modification, and reveal the mechanism for future design and preparation of catalyst and electrode.

## 74 Results and discussion

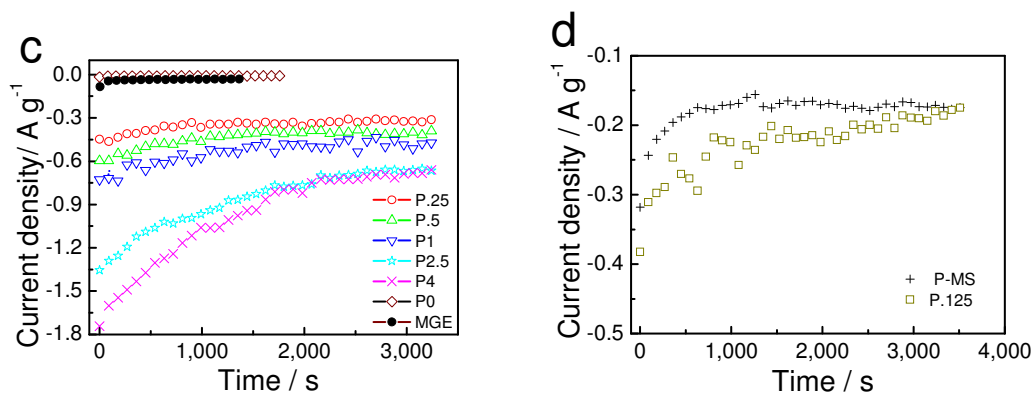
75 The commercial PAN-CF was activated by recurrent galvanic pulses in  $H_2SO_4$ , and similar to the  
76 modified graphite electrode (MGE),<sup>30, 31</sup> to obtain the modified PAN-CFs (MPAN-CFs, denoted as P.125,

77 P.25, *etc.*, in terms of activation current density of 0.125, 0.25 A g<sup>-1</sup>, *etc.* (Fig. S1). It is worth noting that  
78 alike MGE, pseudo-capacitive characteristic attributed to the surface O-containing groups could also be  
79 observed on MPAN-CFs (Fig. 1a). Comparably, the ORR current density (*i.e.* plateau value in Fig. 1c) of  
80 N-free MGE was far lower than those of MPAN-CFs, while the ORR current density and pseudo-  
81 capacitance of MPAN-CFs increased with increasing activation strength (P2.5 had nearly the same ORR  
82 current density as P4 and the fiber wire was found easily break or lose electrochemical performance beyond  
83 5 A g<sup>-1</sup> of activation current density); and the worst of all, PAN-CF showed neither capacitive character nor  
84 ORR activity. Additionally, we modified PAN-CF by chemical oxidation in molten sodium nitrate (NaNO<sub>3</sub>)  
85 at 400 °C (denoted as P-MS), it exhibited small double-layer capacitance (Fig. 1b) and however notable  
86 ORR current density (Fig. 1d). As a consequence, the results indicate that O-containing groups alone have  
87 poor ORR activity and the inherent N in PAN-CF is converted into N-containing groups with high ORR  
88 activity via chemical or electrochemical activation.

89 Although the smooth surface and apparent diameter of PAN-CF changed little under scanning electron  
90 microscope (SEM), the original turbostratic graphite structure was indeed further damaged by  
91 electrochemical activation, inferring from the broadening and weakening of the diffraction peak at 2θ =  
92 25.6° (002) in X-ray diffraction (XRD) and the increasing intensity ratio of D to G peak (I<sub>D</sub>/I<sub>G</sub>) in Raman  
93 spectroscopy with the increase of activation intensity (Fig. S2). Besides, the Brunauer-Emmett-Teller  
94 (BET) specific surface area doubled only from 0.537 to 1.058 m<sup>2</sup> g<sup>-1</sup>. All indicate the electrochemical  
95 activation has an advantage of mildly adding surface defects but bringing no obvious damage to surface  
96 morphology.



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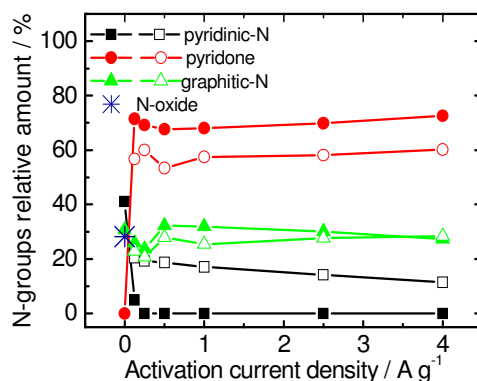
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99 **Fig. 1** CV capacitance vs. potential curves for (a) MPAN-CFs (P.25, P.5, P1, P2.5 and P4) , (b) P-MS and  
 100 MPAN-CF (P.125) in deaerated 2 mol L<sup>-1</sup> H<sub>2</sub>SO<sub>4</sub>. Current density vs. time curves for (c) MGE, PAN-CF  
 101 (P0), P.25, P.5, P1, P2.5 and P4, (d) P-MS and P.125 at 0.33 V under magnetic stirring of 600 r.p.m. in  
 102 naturally aerated pH8.2 seawater. P-MS exhibited nearly the same ORR current density but a far less  
 103 capacitance as P.125.

104 Surface chemical compositions of PAN-CF before and after activation were analyzed, and they were  
 105 subsequently ion-sputtered for 100 s to conduct an in-depth profile analysis by X-ray photoelectron  
 106 spectroscopy (XPS). Compared with PAN-CF surface, O and N content evidently increased both on surface  
 107 and in ion-sputtered profile of MPAN-CFs (Table S1), and the rise of N content definitely comes from the  
 108 surface enrichment of inherent N by electrochemical/chemical oxidization and etching action. However, the  
 109 contents of O and N seemed no direct correlation with ORR activity and pseudo-capacitance shown in Fig.  
 110 1. Fitting C1s spectra (Table S2), 284.3 and 285.2 eV represent the functionalized sp<sup>2</sup> and sp<sup>3</sup> C-C bond on  
 111 defect sites, respectively, and the ratio of their total sum to the content of non-functionalized sp<sup>2</sup> C-C bond  
 112 at 284.6 eV reflects the degree of defects and the proportion of disordered carbon.<sup>32</sup> As a result, the much  
 113 higher surface ratios (>4) and in-depth ratios (>3) of MPAN-CFs than PAN-CF surface ratio (1.8) indicate  
 114 a remarkable increase of defects from surface to deep profile after activation. Furthermore, these in-depth  
 115 ratios depended on the activation intensity, the higher the activation current density the higher the in-depth  
 116 ratios, indicative of a thicker three-dimensional active layer.

117 Fitting N1s spectra (Fig. S3), pyridinic-N, graphitic-N and pyridine-N-oxide existed on PAN-CF  
 118 surface, consistent with literatures.<sup>33, 34</sup> After activation, unstable pyridine-N-oxide completely disappeared  
 119 and surprisingly, pyridinic-N only remained in ion-sputtered profile; simultaneously, a pair of intense peaks

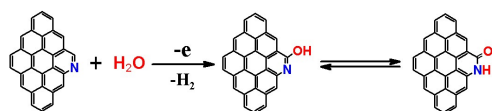
120 at 399.4 and 400.2 eV appeared, which were ever ascribed to pyrrolic-N/pyridone, <sup>32, 35</sup> but in our work  
 121 they solely correspond to pyridone due to the absence of pyrrolic-N in PAN-CF. <sup>33, 34</sup> And more  
 122 importantly, a decrease in pyridinic-N was markedly associated with an increase in pyridone, a N- and O-  
 123 doped functional group (Fig. 2). Considering the little change of stable graphitic-N and the 1/2~2/3 more  
 124 pyridone in total N, thereby newly generated pyridone is taken for granted as the fundamental reason for  
 125 the enhancement of ORR activity.



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127 **Fig. 2** Changes in relative amount of different N-groups on PAN-CF and MPAN-CFs surface (filled  
 128 symbols) and in their 100s-ion-sputtered profile (unfilled symbols) with the activation current density.

129 Accordingly, an ortho-selectively nucleophilic substitution reaction probably occurs during  
 130 electrochemical oxidation shown in eq. (1), *i.e.* pyridinic-N is changed into 2-pyridone or 2-hydroxyl  
 131 pyridine, a pair of tautomeric compound corresponding to pyridone ring of amide-type at 399.4 eV or imide  
 132 alcohol-type at 400.2 eV. Recently, Achour *et al* also obtained pyridone from pyridinic-N by  
 133 electrochemical anodic oxidation of N-doped carbon nano-wall films. <sup>36</sup> Silva R. *et al* observed the high  
 134 ORR activity from N- and O-doped meso-porous carbons. <sup>37</sup>



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(1)

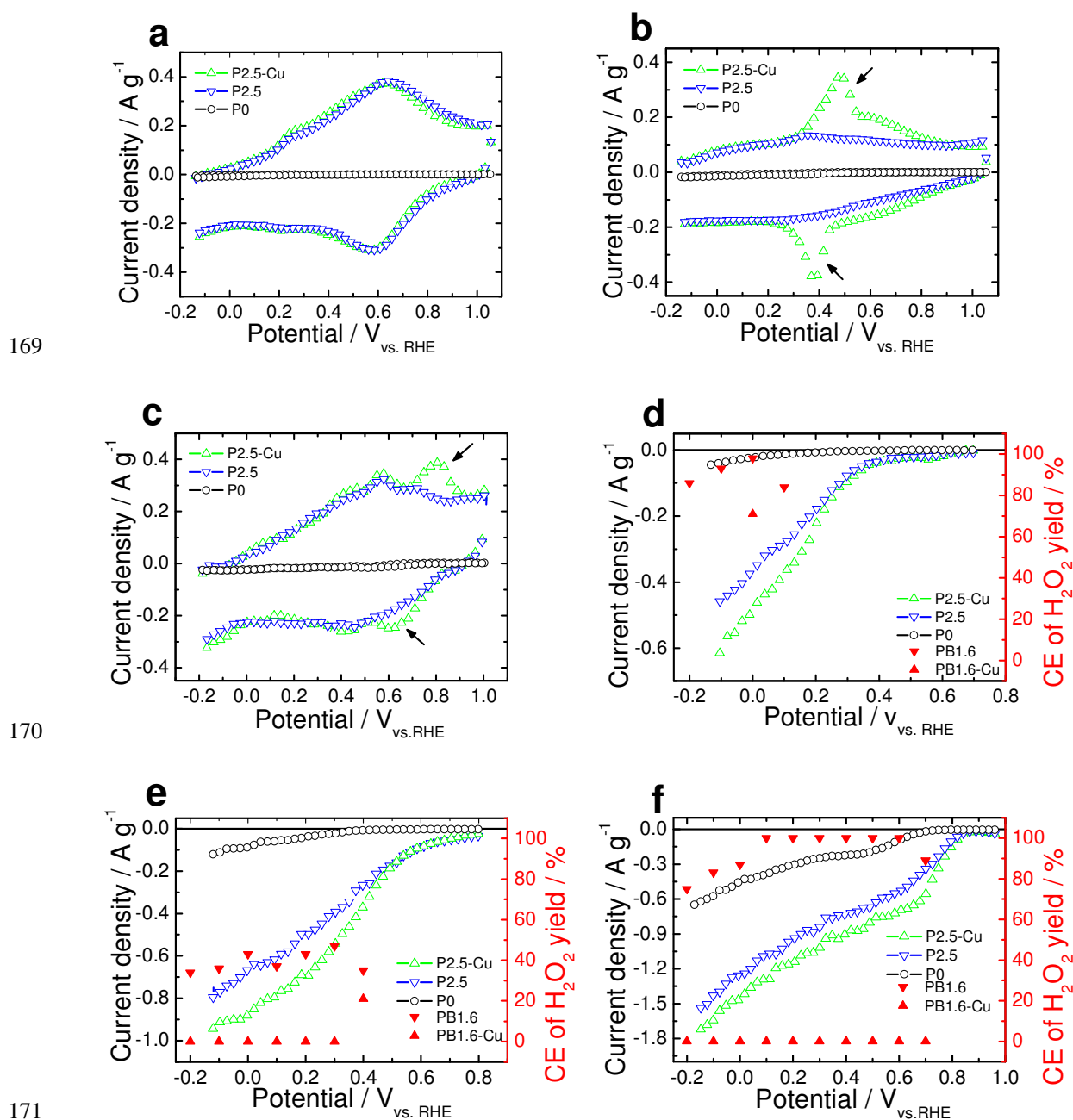
136 Subsequently, the influence of the electrolyte medium and metal ion on ORR properties of MPAN-CFs  
 137 is investigated in detail. As a whole, ORR onset potential (the separate points of two potential~time curves  
 138 under nitrogen- and oxygen-saturated condition) moved positively with the increase of medium pH (Fig.

139 S4), indicative of a better activity in alkaline solution. Two plateau onset potentials at 0.91 V in alkaline  
140 solution as well as 0.61 V in acidic solution arose at both ends; and compared with the reported Pt/C,<sup>38</sup> a  
141 small potential difference of *ca.* 0.1 V also confirms an excellent ORR activity of MPAN-CF in alkaline  
142 solution. Next, a MPAN-CF was immersed into the saturated CuCl<sub>2</sub> solution to prepare a composite  
143 electrode (denoted as MPAN-CF-Cu(II)) through Lewis acid-base reaction, and its CV characteristics were  
144 compared with those of PAN-CF and MPAN-CF in H<sub>2</sub>SO<sub>4</sub>, Na<sub>2</sub>SO<sub>4</sub> and KOH (Fig. 3a-c). In fact, aside  
145 from the wide pseudo-capacitive redox peaks in all electrolytes, MPAN-CF-Cu(II) newly showed a pair of  
146 Cu(II)/Cu(I) redox peaks whose reduction peak potential is at 0.7 V in alkaline and 0.4 V in neutral solution  
147 (arrows in Fig. 3b and c). At the same time, on polarization curves (Fig. 3d-f), MPAN-CF-Cu(II) had the  
148 highest polarization current density and the most positive onset potential, *i.e.* the best activity in all  
149 conditions, which may be attributed to the complexation of pyridone with Cu<sup>2+</sup> that changes the ORR  
150 selectivity to reduce the molecular O<sub>2</sub> more completely and to enhance the ORR activity subsequently. In a  
151 word, as for the ORR activity, MPAN-CF-Cu(II) is the best in terms of electrode, and the alkaline solution  
152 is the best in terms of medium.

153 Additionally, ORR selectivity of MPAN-CF and MPAN-CF-Cu(II) was evaluated by the current  
154 efficiency (CE) of H<sub>2</sub>O<sub>2</sub> yield. H<sub>2</sub>O<sub>2</sub> was detected by titanium potassium oxalate (C<sub>4</sub>K<sub>2</sub>O<sub>9</sub>Ti)  
155 spectrophotometry method instead of conventional rotating ring disk electrode (RRDE) used by nano-sized  
156 materials.<sup>39</sup> To meet the requirements of detection limit and stability, the CE of H<sub>2</sub>O<sub>2</sub> yield had to be  
157 measured at high cathodic overpotential in short time (the red filled triangles in Fig. 3d-f). In acidic  
158 solution, both MPAN-CF and MPAN-CF-Cu(II) (the least point for poor stability) exhibited a predominant  
159 2e<sup>-</sup> pathway. In alkaline solution, MPAN-CF was dominated by 2e<sup>-</sup> process whereas MPAN-CF-Cu(II)  
160 revealed a 4e<sup>-</sup> one at potentials negative than 0.7 V with the ORR current up to 0.54 A m<sup>-2</sup> (relative to the  
161 BET specific surface area) at 0.7 V. Moreover, in neutral solution, the mixed 2e<sup>-</sup> and 4e<sup>-</sup> reduction process  
162 for MPAN-CF and the dominant 4e<sup>-</sup> process for MPAN-CF-Cu(II) occurred at potentials negative than 0.4  
163 V. Similarly, the MPAN-CF-Fe(III) by coordinating pyridone to Fe<sup>3+</sup> also confirmed that the complexation  
164 and dissociation of pyridone with proton (acidic solution) or metal ion in different medium changed the  
165 ORR activity and selectivity (Fig. S5). Finally, a 650-hour performance test of MPAN-CF-Cu(II) at 0.33 V  
166 in pH8.2 seawater was investigated. After an expectedly gradual dissociation of Cu<sup>2+</sup>, the ORR current



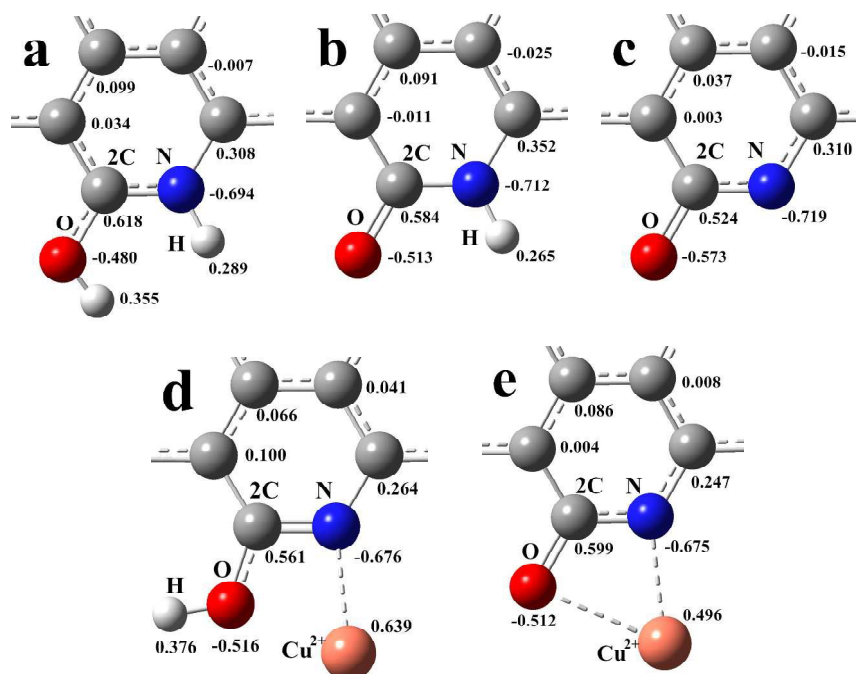
167 density kept constant at *ca.*  $0.67 \text{ A g}^{-1}$  after 36 h, signaling the excellent long-term stability of MPAN-CF  
 168 itself (Fig. S6).



172 **Fig. 3** (a-c) Cyclic voltammograms under deaerated conditions; (d-f) polarization curves under  
 173 magnetically stirring of 600 r.p.m. in naturally aerated solutions for PAN-CF (P0), MPAN-CF (P2.5)  
 174 and MPAN-CF-Cu(II) (P2.5-Cu) with scan rate of  $5 \text{ mVs}^{-1}$ , and current efficiency (CE) of  $\text{H}_2\text{O}_2$  yield  
 175 (the red filled triangles in d-f) at different applied potentials under air-eration condition for MPAN-

176 CF (PB1.6) and MPAN-CF-Cu(II) (PB1.6-Cu) in  $0.5 \text{ mol L}^{-1} \text{H}_2\text{SO}_4$  (a and d),  $0.5 \text{ mol L}^{-1} \text{Na}_2\text{SO}_4$  (b  
177 and e) and  $0.1 \text{ mol L}^{-1} \text{KOH}$  (c and f).

178 To associate ORR performance with N-doping structure, density functional theory (DFT) calculation  
179 was carried out to simulate the optimal structure and surface charge distribution (Fig. 4, Fig. S7). N-  
180 containing groups are believed to randomly disperse on edge plane,<sup>40</sup> here the positive charge density on  
181 2C of pyridone (carbon atom adjacent to N and O) is evidently much higher than that of either pyridinic- or  
182 graphitic-N (carbon atom adjacent to N), thus it is identified as the active site derived from the strong  
183 electron affinity of both N and O.<sup>19, 38</sup> After the complexation of protonated pyridone with  $\text{Cu}^{2+}$  in acidic  
184 solution, a large difference of bond length between  $\text{Cu}^{2+}$  with N and O (1.971 and 2.609 Å) exists, and the  
185 positive charge density on  $\text{Cu}^{2+}$  was 0.639, while those in neutral/alkaline solution were 2.022 Å, 2.177 Å  
186 and 0.495 respectively (Fig. 4d and e), meaning apparently different bond number, bond strength and  
187 charge density. The different complexation strength may change the reduction capability of metal ion, the  
188 weak complexation from one bond makes it reduced at more negative potential while the strong  
189 complexation from two bonds favors losing or receiving the electron and therefore makes metal ion  
190 reduced at more positive potential. That is to say, one bond between  $\text{Cu}^{2+}$  and N in acidic solution and the  
191 two bonds between  $\text{Cu}^{2+}$  with N and O in neutral/alkaline solution resulted in the different charge density,  
192 which probably changed the redox potential of Cu(II)/Cu(I) complex in Fig. 3a-c and the ORR selectivity  
193 in Fig. 3d-f.

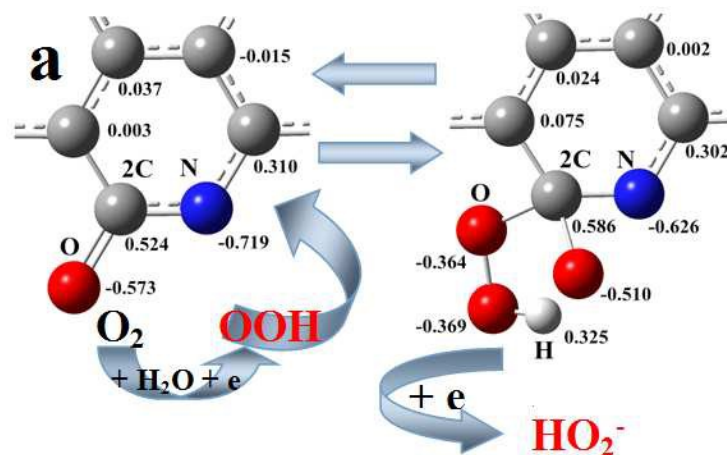


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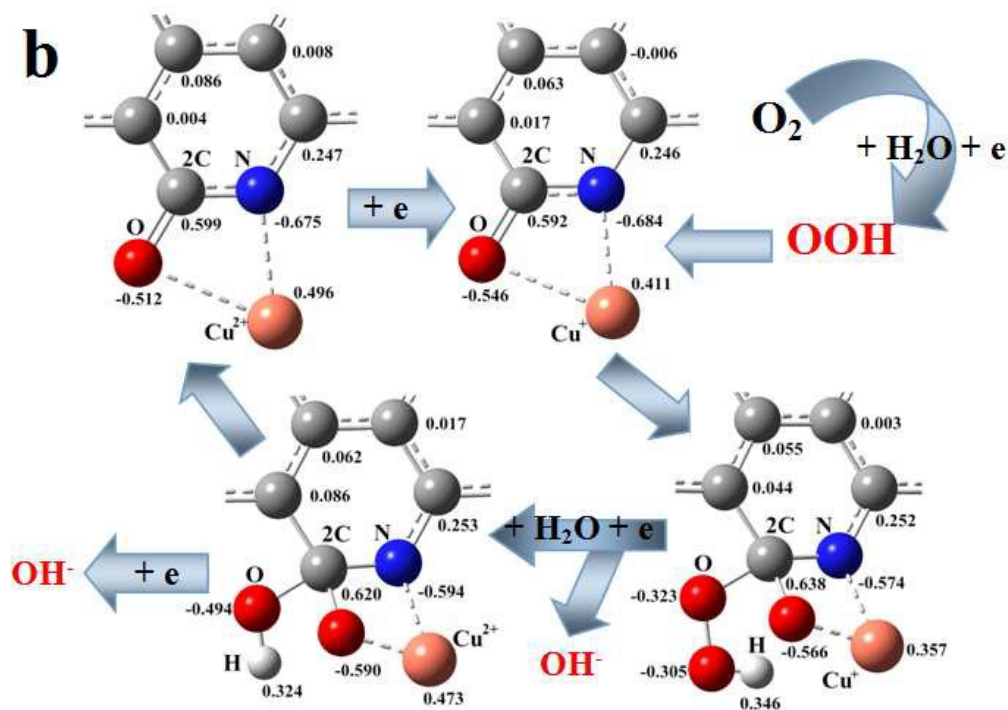
195 **Fig. 4** Surface charge distribution on (a) protonated pyridone and (d) protonated MPAN-CF-Cu(II) in  
 196 acidic solution, (b) pyridone in neutral solution, (c) deprotonated pyridone in alkaline solution and (e)  
 197 MPAN-CF-Cu(II) in neutral and alkaline solution.

198 As for the effect of surface charge transfer and reconstruction on ORR selectivity, take MPAN-CF and  
 199 MPAN-CF-Cu(II) in alkaline solution as an example. In Fig. 5a, molecular O<sub>2</sub> is polarized in outer  
 200 Helmholtz layer over a long range and generated HOO after gaining an electron, <sup>41</sup> HOO is preferentially  
 201 adsorbed on 2C atom to form HOO\*. Considering the bond length of C-O\* (1.469 Å) is longer than that of  
 202 O-O\* (1.459 Å), C-O\* bond breaks more easily to form HO<sub>2</sub><sup>-</sup> in the reduction process of C-O\*OH,  
 203 resulting in a typical 2e<sup>-</sup> pathway. In Fig. 5b, MPAN-CF-Cu(II) complex easily gains an electron to form  
 204 MPAN-CF-Cu(I) complex according to the distinct redox peaks in Fig. 3c, and HOO by reduction of O<sub>2</sub> is  
 205 similarly adsorbed on 2C to form HOO\*. It is notable for Cu<sup>+</sup>, a small interatomic distance of 2.487 Å  
 206 between it and O in HOO\* and the decline of charge density from 0.411 to 0.357 will probably facilitate  
 207 the intra-molecular chemical redox reaction: *i.e.* Cu<sup>+</sup> is oxidized into Cu<sup>2+</sup> with a reasonable rise of charge  
 208 density to 0.473 again, and the O-O\* bond simultaneously breaks to form OH<sup>-</sup> and C-O\*H which is  
 209 subsequently electrochemically reduced into OH<sup>-</sup>, too. Namely, Cu<sup>2+</sup> and 2C in MPAN-CF-Cu(II) complex  
 210 are considered the two adjacent active sites with a tendency of 4e<sup>-</sup> selectivity, and an

211 electrochemical/chemical synergetic effect between metal ion and oxygen reduction intermediate favors the  
 212  $4e^-$  pathway, similar to the ORR acceleration by a coupled electrochemical/chemical process on copper in  
 213  $H_2SO_4$ .<sup>42</sup> In brief, a single active site from 2C atom in pyridone favors a  $2e^-$  pathway and the two adjacent  
 214 active sites from 2C and metal ion facilitate a  $4e^-$  pathway and the best activity.



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216

217 **Fig. 5** Schematic diagram of (a)  $2e^-$  pathway of MPAN-CF and (b)  $4e^-$  pathway of MPAN-CF-Cu(II) in  
 218 alkaline solution.

219 Apart from the predominant active site of 2C, the 6C atom in pyridone (connecting with N) can  
220 presumably act as another adsorption active site, whose net positive charge density of 0.352 in neutral  
221 solution is the highest compared with others (0.303 in acidic and 0.310 in alkaline solution) in Fig. 4, that  
222 may explain why MPAN-CF displayed the mixed 2e<sup>-</sup> and 4e<sup>-</sup> selectivity in neutral solution (Fig. 3e and Fig.  
223 S7).

## 224 Conclusions

225 A low-cost MPAN-CF electrode was obtained by electrochemically modifying micro-scale PAN-CF.  
226 The degree of defects and thickness of active layer in MPAN-CF increased dependent on the activation  
227 intensity, and the newly generated pyridone in three-dimensional active layers was considered responsible  
228 for high ORR activity and stability. Moreover, ORR activity and selectivity of MPAN-CF were influenced  
229 by the medium and metal ion. In alkaline solution, two adjacent active sites from metal ion and carbon  
230 atom connecting with N and O facilitated the 4e<sup>-</sup> selectivity as well as the best activity, and an  
231 electrochemical/chemical synergetic mechanism between metal ion and oxygen reduction intermediate was  
232 proposed to correlate the structure of composite with ORR performance.

## 233 Acknowledgments

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235 Natural Science Foundation of China (41427803), Shandong Province Science and Technology  
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## 237 Notes and references

238 †Electronic Supplementary Information (ESI) available: [Experimental details, supplementary Figs. S1-S7  
239 and Tables S1-S2, including electrode and system arrangement; XRD patterns and Raman spectra of PAN-  
240 CF and MPAN-CFs; N-containing groups analysis by XPS; ORR activity influenced by medium pH; ORR  
241 activity and selectivity for MPAN-CF-Fe(III); durability test; quantum mechanics calculation; element  
242 analysis of C,O and N; carbon species analysis by XPS]. See DOI: 10.1039/c000000000a.

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- 244 1. Y. Bing, H. Liu, L. Zhang, D. Ghosh and J. Zhang, *Chem. Soc. Rev.*, 2010, **39**, 2184.
- 245 2. Z. Chen, D. Higgins, A. Yu, L. Zhang and J. Zhang, *Energy Environ. Sci.*, 2011, **4**, 3167.

- 246 3. E. Negro and V. D. Noto, *J. Power Sources*, 2008, **178**, 634.
- 247 4. D.-W. Wang and D. Su, *Energy Environ. Sci.*, 2014, **7**, 576.
- 248 5. J. W. D. Ng, M. Tang and T. F. Jaramillo, *Energy Environ. Sci.*, 2014, **7**, 2017.
- 249 6. R. S. Figueiredo, R. Bertazzoli and C. A. Rodrigues, *Ind. Eng. Chem. Res.*, 2013, **52**, 5611.
- 250 7. H. T. Chung, J. H. Won and P. Zelenay, *Nat Commun*, 2013, **4**, 1922.
- 251 8. S. Siahrostami, A. Verdaguer-Casadevall, M. Karamad, D. Deiana, P. Malacrida, B. Wickman, M.  
252 Escudero-Escribano, E. A. Paoli, R. Frydendal, T. W. Hansen, I. Chorkendorff, I. E. L. Stephens  
253 and J. Rossmeisl, *Nat. Mater.*, 2013, **12**, 1137.
- 254 9. E. Brillas, I. Sires and M. A. Oturan, *Chem. Rev.*, 2009, **109**, 6570.
- 255 10. M. Nie, P.-K. Shen, M. Wu, Z.-D. Wei and H. Meng, *J. Power Source*, 2006, **162**, 173.
- 256 11. V. D. Noto, E. Negro, S. Polizzi, F. Agresti and G. A. Giffin, *ChemSusChem*, 2012, **5**, 2451.
- 257 12. A. Morozan, B. Josselme and S. Palacin, *Energy Environ. Sci.*, 2011, **4**, 1238.
- 258 13. P. Chen, L.-K. Wang, G. Wang, M.-R. Gao, J. Ge, W.-J. Yuan, Y.-H. Shen, A.-J. Xie and S.-H.  
259 Yu, *Energy Environ. Sci.*, 2014, **7**, 4095.
- 260 14. W. He, C. Jiang, J. Wang and L. Lu, *Angew. Chem., Int. Ed.*, 2014, **53**, 9503.
- 261 15. P. Chen, T. Y. Xiao, Y. H. Qian, S. S. Li and S. H. Yu, *Adv. Mater.*, 2013, **25**, 3192.
- 262 16. H. A. Gasteiger and N. M. Markovic, *Science*, 2009, **324**, 48.
- 263 17. Y. Zheng, Y. Jiao, M. Jaroniec, Y. Jin and S. Z. Qiao, *Small*, 2012, **8**, 3550.
- 264 18. L. Feng, Y. Yan, Y. Chen and L. Wang, *Energy Environ. Sci.*, 2011, **4**, 1892.
- 265 19. K. Gong, F. Du, Z. Xia, M. Durstock and L. Dai, *Science*, 2009, **323**, 760.
- 266 20. L. Qu, Y. Liu, J.-B. Baek and L. Dai, *ACS nano*, 2010, **4**, 1321.
- 267 21. R. Bashyam and P. Zelenay, *Nature*, 2006, **443**, 63.
- 268 22. M. Lefevre, E. Proietti, F. Jaouen and J. P. Dodelet, *Science*, 2009, **324**, 71.
- 269 23. G. Wu, K. L. More, C. M. Johnston and P. Zelenay, *Science*, 2011, **332**, 443.
- 270 24. P. Chen, T.-Y. Xiao, Y.-H. Qian, S.-S. Li and S.-H. Yu, *Adv. Mater.*, 2013, **25**, 3192.
- 271 25. Q. Li, B. W. Noffke, Y. Wang, B. Menezes, D. G. Peters, K. Raghavachari and L.-s. Li, *J. Am.*  
272 *Chem. Soc.*, 2014, **136**, 3358.
- 273 26. M. S. A. Rahaman, A. F. Ismail and A. Mustafa, *Polym. Degrad. Stab.*, 2007, **92**, 1421.
- 274 27. B. Huskinson, M. P. Marshak, C. Suh, S. Er, M. R. Gerhardt, C. J. Galvin, X. D. Chen, A. Aspuru-  
275 Guzik, R. G. Gordon and M. J. Aziz, *Nature*, 2014, **505**, 195.
- 276 28. M. Shinohara, E. Araki, M. Mochizuki, T. Kanazawa and K. Suyehiro, *J. Power Sources*, 2009,  
277 **187**, 253.
- 278 29. Ø. Hasvold, T. Lian, E. Haakaas, N. Størkersen, O. Perelman and S. Cordier, *J. Power Sources*,  
279 2004, **136**, 232.
- 280 30. X. Z. Fan, Y. H. Lu, H. B. Xu, X. F. Kong and J. Wang, *J. Mater. Chem.*, 2011, **21**, 18753.
- 281 31. H. B. Xu, X. Z. Fan, Y. H. Lu, L. A. Zhong, X. F. Kong and J. Wang, *Carbon*, 2010, **48**, 3300.
- 282 32. S. Pylypenko, A. Queen, T. S. Olson, A. Dameron, K. O'Neill, K. C. Neyerlin, B. Pivovar, H. N.  
283 Dinh, D. S. Ginley, T. Gennett and R. O'Hayre, *J. Phys. Chem. C*, 2011, **115**, 13667.
- 284 33. J. R. Pels, F. Kapteijn, J. A. Moulijn, Q. Zhu and K. M. Thomas, *Carbon*, 1995, **33**, 1641.

- 285 34. L. Laffont, M. Monthieux, V. Serin, R. B. Mathur, C. Guimon and M. F. Guimon, *Carbon*, 2004,  
286 **42**, 2485.
- 287 35. W. Shen, *Recent Patents on Chemical Engineering*, 2008, **1**, 27.
- 288 36. A. Achour, S. Vizireanu, G. Dinescu, L. Le Brizoual, M. A. Djouadi and M. Boujtita, *Appl. Surf. Sci.*, 2013, **273**, 49.
- 290 37. R. Silva, D. Voiry, M. Chhowalla and T. Asefa, *J. Am. Chem. Soc.*, 2013, **135**, 7823.
- 291 38. B. Cao, G. M. Veith, R. E. Diaz, J. Liu, E. A. Stach, R. R. Adzic and P. G. Khalifah, *Angew. Chem. Int. Ed.*, 2013, **52**, 10753.
- 293 39. R. M. Sellers, *The Analyst*, 1980, **105**, 950.
- 294 40. P. H. Matter, L. Zhang and U. S. Ozkan, *J. Catal.*, 2006, **239**, 83-96.
- 295 41. C. H. Choi, H.-K. Lim, M. W. Chung, J. C. Park, H. Shin, H. Kim and S. I. Woo, *J. Am. Chem. Soc.*, 2014, **136**, 9070.
- 297 42. Y. Lu, H. Xu, J. Wang and X. Kong, *Electrochimica Acta*, 2009, **54**, 3972.

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