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ARTICLE

Plasmonic Enhancement of the Optical Absorption and Catalytic Efficiency of BiVO₄ Photoanodes Decorated with Ag@SiO₂ Core-Shell Nanoparticles

Cite this: DOI: 10.1039/x0xx00000x

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Received 00th January 2014,
Accepted 00th January 2014

DOI: 10.1039/x0xx00000x

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Recent progress in the development of bismuth vanadate (BiVO₄) photoanodes has firmly established it as a promising material for solar water splitting applications. Performance limitations due to intrinsically poor catalytic activity and slow electron transport have been successfully addressed through the application of water oxidation co-catalysts and novel doping strategies. The next bottleneck to tackle is the modest optical absorption in BiVO₄, particularly close to its absorption edge of 2.4 eV. Here, we explore the modification of the BiVO₄ surface with Ag@SiO₂ core-shell plasmonic nanoparticles. A photocurrent enhancement by a factor of ~2.5 is found under 1 sun illumination (AM1.5). We show that this enhancement consists of two contributions: optical absorption and catalysis. The optical absorption enhancement is induced by the excitation of localized surface plasmon resonances in the Ag nanoparticles, and agrees well with our full-field electromagnetic simulations. Far-field effects (scattering) are found to be dominant, with a smaller contribution from near-field plasmonic enhancement. In addition, a significant catalytic enhancement is observed, which is tentatively attributed to the electrocatalytic activity of the Ag@SiO₂ nanoparticles.

In the field of solar water splitting, bismuth vanadate (BiVO₄) has emerged as one of the most promising photoanode materials. The monoclinic scheelite phase (clinobisvanite) is the most photo-active form of BiVO₄. The material shows n-type conductivity and has a bandgap of ~2.4 eV.¹⁻³ An attractive aspect of BiVO₄ is that its band edge positions are more favorably located than other well-known metal oxides for solar water splitting, such as hematite or tungsten trioxide.⁴⁻⁶ BiVO₄ photoanodes profit from a good stability in aqueous solutions with pH values between 3 and 11.⁷ However, the performance of the material is hampered by two main limitations: (i) slow hole transfer across the semiconductor/electrolyte interface (i.e., low catalytic activity for water oxidation) and (ii) slow electron transport, which results in sub-optimal carrier separation. In recent years, these issues have successfully been addressed. The first issue has been largely solved by attaching co-catalysts to the surface of BiVO₄. Oxygen evolution

catalysts such as RhO₂,⁸ cobalt phosphate (Co-Pi),⁹⁻¹¹ cobalt borate (Co-Bi),¹² nickel borate (Ni-Bi),¹³ and iron oxyhydroxide (FeOOH)¹⁴ have been successfully used. At sufficiently high bias potentials, catalytic activities of up to 100% have been shown,^{9, 11, 15} demonstrating that the problem of slow water oxidation catalysis has been solved for BiVO₄. The second limitation (slow electron transport) can be solved by introducing a donor-type dopant, such as W and Mo,^{9, 11, 15-18} thereby increasing the carrier concentration. Recently, we have shown that this can be further improved by introducing a gradient in the dopant concentration in BiVO₄. The gradient dopant profile introduces a small but effective electrical potential gradient in the film, which improves the carrier separation efficiency to values as high as 80%.¹⁹ With these modifications an AM1.5 photocurrent of 3.6 mA/cm² has been achieved at 1.23 V vs the reversible hydrogen electrode (RHE),

which firmly establishes BiVO₄ as the currently highest performing metal oxide photoanode.¹⁹

With the catalysis and charge transport issues solved, modest light absorption—especially for photons with energies close to the bandgap—becomes the next main bottleneck. To illustrate this, if we assume that all the light absorbed in our best-performing W:BiVO₄ samples contributes to photocurrent, we would still *only* have ~5 mA/cm².¹⁹ For comparison, the maximum theoretical photocurrent for BiVO₄ is ~7.5 mA/cm² (assuming that all photons with an energy higher than the bandgap are absorbed and collected). This means that a significant increase of the photocurrent (2.5 mA/cm²) can be achieved by improving light absorption. Unfortunately, simply increasing the thickness of the film to absorb more light will not work, since the carrier diffusion length in BiVO₄ is ~100 nm.^{20, 21} Considering that the best-performing BiVO₄ already has a thickness of ~200 nm, other strategies to improve light absorption are therefore needed.

Recently, surface plasmon resonances of metallic nanostructures, such as Au, Ag, and Al, have been extensively used to improve the absorption—and the efficiency—in various photocatalytic systems and solar cells.^{22–26} The same approach has also been pursued for solar water splitting applications.^{27–33} For example, Thimsen et al. reported a significant improvement of the optical absorption in hematite when Au plasmonic nanoparticles are introduced on the hematite surface or at the interface between the substrate and hematite film.²⁸ However, this improvement is not accompanied by the same improvement in the overall efficiency; the photocurrent was found to decrease instead. This inconsistency is attributed to the introduction of traps at the interface between the Au and hematite. One way to circumvent this problem is by introducing a thin SiO₂ insulating layer on the surface of the Au nanoparticles. Thomann et al. demonstrated that this effectively prevents nanoparticle-induced charge recombination.²⁹

In this paper, we explore the application of core-shell nanoparticles on the surface of 100 nm-thick spray-deposited BiVO₄ films as a route to enhance the optical absorption in the films. Similar to the approach of Thomann et al.,²⁹ SiO₂ is used as the insulating layer around the core metal. However, here, Ag is used as the core metal because: (i) Ag is less expensive than Au, and (ii) the plasmon frequency of Ag can be tailored between 400 to 600 nm,^{34, 35} providing a better match to the absorption spectrum of BiVO₄. We predict and show the beneficial effect of the Ag@SiO₂ core-shell nanoparticles on the performance of BiVO₄ photoanodes. Through a detailed photoelectrochemical analysis, we are able to distinguish between absorption and catalytic effects that contribute to the observed improvements. To the best of our knowledge, this is the first report on the application of core-shell nanoparticles to improve the absorption and efficiency of BiVO₄ photoanodes.

We first perform full-field electromagnetic simulations to assess the performance enhancements of Ag@SiO₂ nanoparticles on BiVO₄ films. In these simulations, we numerically solve frequency-domain Maxwell equations using a finite element method, following a previously reported

approach.^{29, 36} We consider two different types of BiVO₄ samples: (i) bare BiVO₄ and (ii) BiVO₄ decorated with Ag@SiO₂ core-shell nanoparticles (Ag@SiO₂/BiVO₄), both are shown in Figure 1. The diameter of the Ag core is 50 nm, and the thickness of the SiO₂ shell is 10 nm, based on TEM analysis of the nanoparticles (see ESI, Figure S1†). To make our calculations as realistic as possible, we explicitly take into account the surface roughness of the films in relation to the Ag@SiO₂ particle size. The surface roughness was measured with atomic force microscopy (see ESI, Figure S2†).

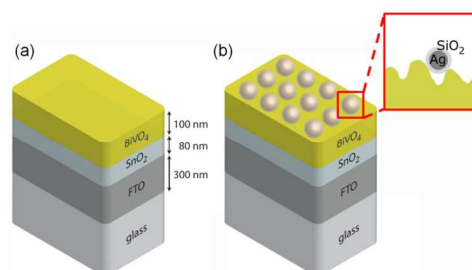


Figure 1. Schematic illustration of the (a) bare and (b) Ag@SiO₂ decorated BiVO₄ (Ag@SiO₂/BiVO₄) samples.

The direct results of the electromagnetic calculations are the amplitudes of the electric and magnetic fields (E , H). Based on these, we calculate the rate of light absorption at every point inside the light absorbing layer, using the following equation:

$$P_d = \frac{1}{2} \omega \varepsilon''(\omega) |\overline{E}(r, \omega)|^2 \quad (1)$$

Here, ω is the angular frequency of the incident light, $\varepsilon''(\omega)$ is the imaginary part of the permittivity of the material, and $|\overline{E}(r, \omega)|$ describes the amplitude of the electric field at position r and angular frequency ω . Inside the BiVO₄, the power-loss density (P_d) is the rate of electron-hole pair generation at every point (in W/m³), while in the Ag nanoparticles, it describes the absorbed photons that dissipate their energy as phonons.

Figure 2 shows the distribution of P_d at the cross-section of the different samples under illumination with 465 nm light, given an incident power of 1 W. The photoanode is illuminated from the substrate side (back-side illumination). Non-zero optical absorption takes place only within Ag and BiVO₄ since other layers in the structure do not absorb 465 nm light. Note that the absorption profile over the bare BiVO₄ (Figure 2a) is not uniform, despite the planar wave-front of the excitation light. This is caused by the wavelength-scale dimensions of the superficial features of BiVO₄, which affect the propagation of light.³⁷ Such wavelength-scale features can have Mie resonances and, if properly engineered, can be used to reduce the reflectance from the layer.³⁷ Figure 2b and c shows a BiVO₄ film segment with a single Ag@SiO₂ nanoparticle, which corresponds to a 25% surface coverage. Results from the two conditions were averaged to take the random position of Ag@SiO₂ into consideration. The absorption in the BiVO₄ is enhanced, especially in the area underneath the nanoparticle.

Further increase of the surface coverage to up to 50% (Figure 2d) results in an even stronger absorption enhancement.

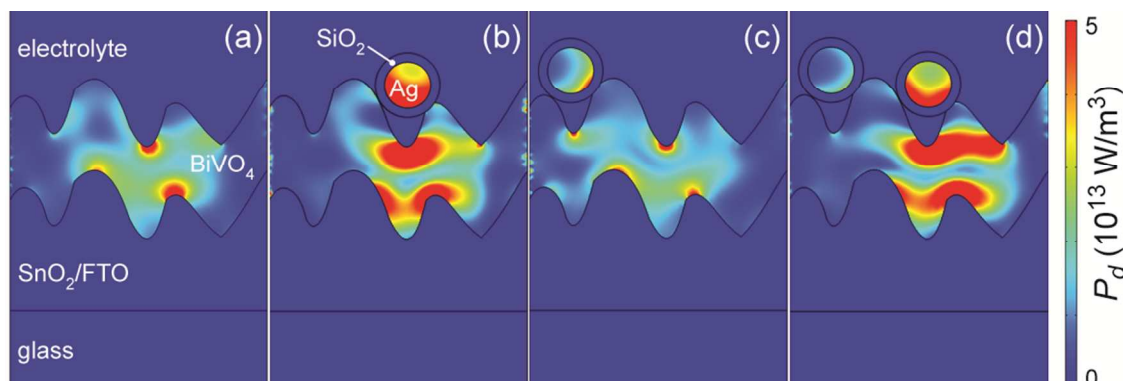


Figure 2. Color plots of the amplitude of power-loss density, P_d , in the samples with an incident power of 1 W and wavelength of 465 nm. The illumination direction is from the bottom (back-side of the samples). Bare BiVO_4 is shown in (a). BiVO_4 with 25% surface coverage of Ag@SiO_2 are shown in (b) and (c), which are averaged to take the position of the nanoparticle into consideration. Finally, BiVO_4 with 50% surface coverage of Ag@SiO_2 nanoparticles is shown in (d).

The total absorbance in the BiVO_4 film is calculated by integrating P_d over the entire volume of BiVO_4 , according to the following equation:

$$A(\lambda) = \frac{\int_{\text{BiVO}_4 \text{ volume}} P_d(\vec{r}, \lambda) d^3 r}{P_{inc}} \quad (2)$$

where P_{inc} is the power of the incident light. Figure 3a shows the calculated absorption spectra of BiVO_4 films with 0%, 25% and 50% surface coverage of Ag@SiO_2 nanoparticles. A near-field enhancement effect can be identified by the presence of peaks in the absorption spectrum, corresponding to the plasmonic resonance(s) of the nanoparticles.^{22, 29} Our simulation based on a small area in BiVO_4 adjacent to the nanoparticle shows that we would expect a sharp peak between 410 and 420 nm (see ESI, Figure S3†). However, we observe a broadband enhancement in Figure 3a, i.e., it is not specific to a certain wavelength range. We therefore conclude that effects other than near-field enhancement must also play a role. One possible explanation is the occurrence of light scattering from the Ag@SiO_2 nanoparticles and/or from the surface features of the

BiVO_4 layer. Optical scattering, which can be classified as a far-field effect, is indeed known to enhance the absorption under the right conditions.³⁸ Indeed, the scattering cross-section of the Ag@SiO_2 nanoparticle used here is much higher than its absorption cross-section (see ESI, Figure S4†).

Figure 3b shows the calculated absorption enhancement—integrated over the AM1.5 spectrum—as a function of the surface coverage. A maximum is found between 40-60%. At higher coverage, the optical dissipation in Ag becomes larger than the plasmonic induced optical absorption in BiVO_4 ; hence, smaller absorption enhancement is observed. Based on this, we covered ~50% of the surface of our 100 nm-thick, spray pyrolysed BiVO_4 film with the Ag@SiO_2 nanoparticles. To obtain this coverage we used ~2.5 $\mu\text{L}/\text{cm}^2$ of the solution of Ag@SiO_2 nanoparticles. The optical density and the particle size distribution of the solution are shown in the ESI, Figure S5†. Detailed sample preparation and characterization procedures can be found in the ESI†. Our SEM analysis shows that the coverage is not perfectly uniform (see ESI, Figure S6†). Nevertheless, the overall coverage is estimated to be 35-45%, which is close to the desired coverage.

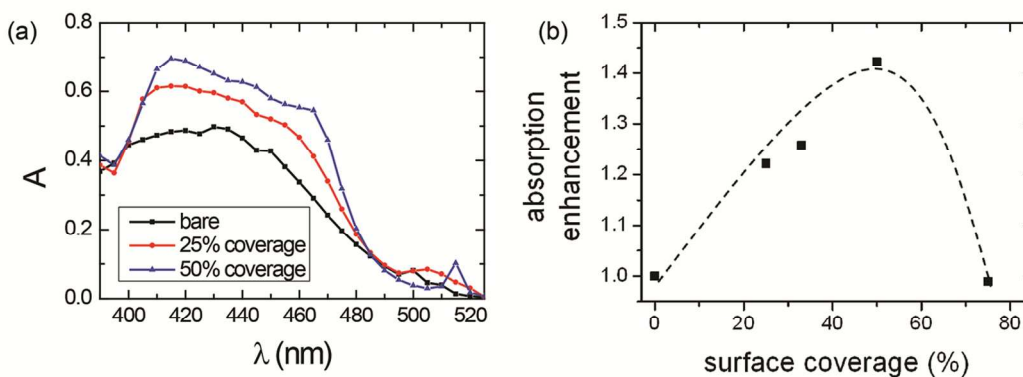


Figure 3. (a) Simulated absorption of BiVO_4 with 0% (bare), 25% and 50% surface coverage of Ag@SiO_2 nanoparticles. (b) The calculated absorption enhancement as a function of the BiVO_4 surface coverage with Ag@SiO_2 nanoparticles.

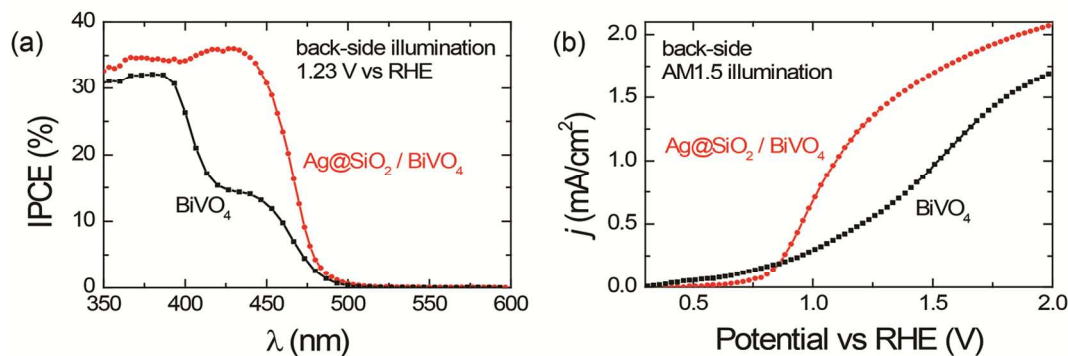


Figure 4. (a) Incident photon-to-current efficiencies (IPCE) and (b) AM1.5 photocurrents of bare and Ag@SiO₂/BiVO₄. The illumination direction is from the substrate side of the samples. The IPCE values were measured at 1.23 V vs RHE.

Figure 4a shows the IPCE of the bare BiVO₄ and the Ag@SiO₂/BiVO₄ at 1.23 V vs RHE under back-side illumination. Upon decoration with Ag@SiO₂, the IPCE of the BiVO₄ photoanode is greatly improved at wavelengths larger than 400 nm. This agrees well with our simulations, and is consistent with optical absorption measurements on these same samples (see ESI, Figure S7†). The same broadband enhancement between 400 and 500 nm is also observed, which confirms that both near- and far-field effects play important roles in the enhancement mechanism.

The increase in IPCE is also reflected in the AM1.5 photocurrent, shown in Figure 4b. A photocurrent of ~1.4 mA/cm² at 1.23 V vs RHE is achieved for Ag@SiO₂/BiVO₄. This is a ~2.5-fold improvement compared to bare BiVO₄. To the best of our knowledge, this is one of the highest AM1.5 photocurrent enhancements reported for a photoanode decorated with metal plasmonic nanoparticles.^{28–33}

To ensure that the observed photocurrent is not a result of photocorrosion or oxidation of the sample, we investigated the stability of Ag@SiO₂/BiVO₄ under illumination. Figure 5 shows the chronoamperometry measurement of the Ag@SiO₂/BiVO₄ sample, at 1.23 V vs RHE under back-side chopped AM1.5 illumination. The stability of the sample is excellent, with no noticeable degradation in 25 minutes. Integration of the photocurrent gives a total charge of 1125 mC/cm², which is ~10 times larger than the charge that would be needed to photo-corrode the nanoparticles. This proves that the photocurrent enhancement can indeed be attributed to improved performance of the Ag@SiO₂-modified photoelectrode.

The 2.5-fold photocurrent improvement we observed is, however, much higher than the ~40% enhancement (Fig. 3b) that is predicted by integrating the calculated absorbance over the AM1.5 irradiance.³⁶ To reconcile this, a thorough photoelectrochemical analysis of the modified BiVO₄ films was performed. The total enhancement factor (f) that is induced by the Ag@SiO₂ nanoparticles can be written as:

$$f = \frac{J_{\text{Ag@SiO}_2/\text{BiVO}_4}}{J_{\text{BiVO}_4}} \quad (3)$$

$J_{\text{Ag@SiO}_2/\text{BiVO}_4}$ and J_{BiVO_4} are the photocurrents of Ag@SiO₂/BiVO₄ and bare BiVO₄, respectively. We now assume that the enhancement can be caused by the increased absorption in BiVO₄ (as a result of plasmonic resonances and scattering) as well as surface catalytic effects. To isolate the absorption enhancement, H₂O₂ was added to the electrolyte to serve as a very effective hole scavenger. Recent studies have shown that the efficiency of hole injection into the electrolyte is ~100% in the presence of H₂O₂.^{11, 39} Under these conditions, it is reasonable to assume that catalysis and surface recombination no longer play a significant role, and that the photocurrent enhancement is entirely due to absorption effects:

$$f_{\text{abs}} = \frac{J_{\text{Ag@SiO}_2/\text{BiVO}_4 + \text{H}_2\text{O}_2}}{J_{\text{BiVO}_4 + \text{H}_2\text{O}_2}} \quad (4)$$

where $J_{\text{Ag@SiO}_2/\text{BiVO}_4 + \text{H}_2\text{O}_2}$ and $J_{\text{BiVO}_4 + \text{H}_2\text{O}_2}$ are the photocurrents of Ag@SiO₂/BiVO₄ and bare BiVO₄ in the presence of H₂O₂, respectively. The catalytic enhancement is obtained by dividing the total enhancement by the absorption enhancement:

$$f_{\text{cat}} = \frac{f}{f_{\text{abs}}} \quad (5)$$

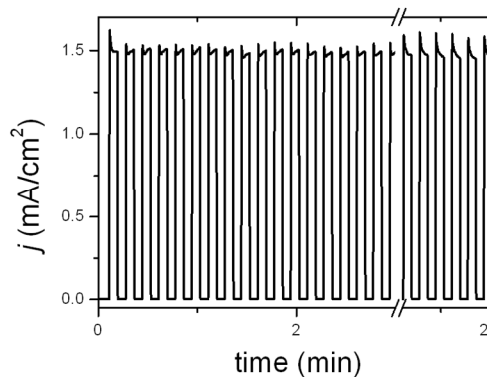


Figure 5. Chronoamperometry plot of Ag@SiO₂/BiVO₄ under back-side chopped AM1.5 illumination at 1.23 V vs RHE.

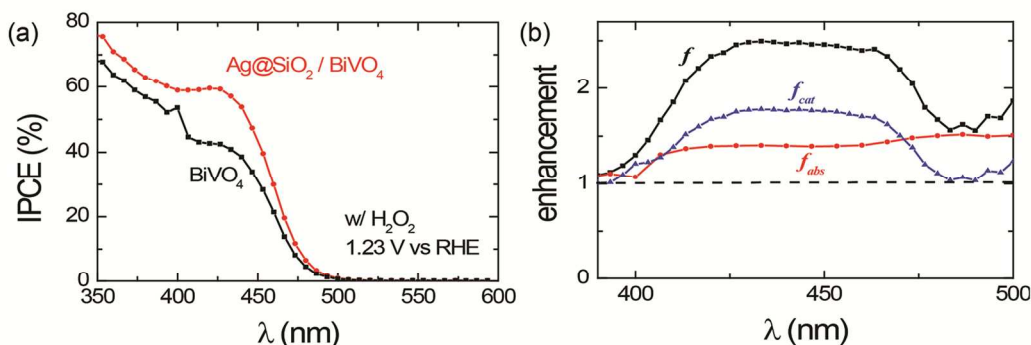


Figure 6. (a) IPCE of bare BiVO_4 and $\text{Ag@SiO}_2/\text{BiVO}_4$ at 1.23 V vs RHE in the presence of 0.5 M H_2O_2 in the electrolyte. (b) Absorption (f_{abs}), catalytic (f_{cat}) and total enhancements (f) in $\text{Ag@SiO}_2/\text{BiVO}_4$ as a function of excitation wavelength.

Figure 6a shows the IPCE of the bare and $\text{Ag@SiO}_2/\text{BiVO}_4$ in the presence of H_2O_2 at 1.23 V vs RHE. The enhancement occurs in the same spectral region as in Figure 4a, but it is significantly smaller. From the data in Figs. 4a and 6a, the absorption, catalytic and total enhancements are calculated from Eqs. 3-5 and shown in Figure 6b. Integration of the absorption enhancement over the AM1.5 spectrum gives an overall enhancement of $\sim 33\%$, which closely matches the predicted enhancement from our full-field electromagnetic simulations. A similar enhancement ($\sim 28\%$ at 1.23 V vs RHE) is observed when we compare the photocurrents of bare BiVO_4 and $\text{Ag@SiO}_2/\text{BiVO}_4$ in the presence of H_2O_2 under actual AM1.5 illumination (see ESI, Figure S8†).

Figure 6b clearly shows that the catalytic enhancement is significantly larger than the absorption enhancement. In order to shed light on the cause of this catalytic enhancement, we determined the catalytic and carrier separation efficiencies of the bare BiVO_4 and $\text{Ag@SiO}_2/\text{BiVO}_4$ (see ESI, Figure S9†).¹¹ The carrier separation efficiencies of both samples are found to be identical, which proves that the enhancement cannot be attributed to the formation of a rectifying junction between the Ag@SiO_2 and the BiVO_4 . In contrast, the catalytic efficiency for the $\text{Ag@SiO}_2/\text{BiVO}_4$ sample is $\sim 85\%$ larger than for the bare BiVO_4 sample at 1.23 V vs. RHE. We tentatively attribute this difference to the electrocatalytic activity of Ag@SiO_2 nanoparticles on the surface of BiVO_4 . At low potentials (< 0.9 vs RHE), the catalytic efficiency of $\text{Ag@SiO}_2/\text{BiVO}_4$ is lower than that of the bare sample. In this case, Ag@SiO_2 nanoparticles are inactive as catalyst; they just block part of the BiVO_4 surface, which explains the lower catalytic efficiency (Fig. S9). This is also evident in Fig. 4b, where the photocurrent of the $\text{Ag@SiO}_2/\text{BiVO}_4$ is lower than that of the bare BiVO_4 . The situation is, however, reversed at higher potentials where the catalytic efficiency of $\text{Ag@SiO}_2/\text{BiVO}_4$ exceeds that of bare BiVO_4 . It appears that at these potentials the Ag@SiO_2 nanoparticles no longer block charge transfer and, instead, catalyze the water oxidation reaction. While there have been no reports of their catalytic activity for water oxidation, redox catalysis using Ag@SiO_2 and Ag-SiO_2

composites have been reported.^{40, 41} To confirm this, we deposited the Ag@SiO_2 core-shell nanoparticles on the surface of an FTO substrate. We then compare the dark current of this $\text{Ag@SiO}_2/\text{FTO}$ and a bare FTO substrate as a function of potential (see ESI, Figure S10†), and observe a ~ 200 mV cathodic shift of the onset potential for the $\text{Ag@SiO}_2/\text{FTO}$. This shows that Ag@SiO_2 indeed behaves as a (dark) catalyst for water oxidation.

One surprising and counter-intuitive observation from Fig. 6b is that the catalytic enhancement shows a pronounced dependence on wavelength. Although the exact reason for this is unclear at the moment, we note that the center of the spectral region where catalytic enhancement occurs overlaps exactly with the optical absorbance of the Ag@SiO_2 nanoparticles (Fig. S5a). We therefore speculate that the electric field that propagates from the Ag@SiO_2 nanoparticles aids the water oxidation reaction, for example by enhancing the carrier separation efficiency near the surface,⁴² or through plasmon-induced resonant energy transfer (PIRET).⁴³ Further investigation is needed to elucidate this, which is beyond the scope of the current work.

Conclusions

In summary, we have explored the application of Ag@SiO_2 core-shell nanoparticles on the surface of BiVO_4 to improve the absorption in BiVO_4 . By performing a thorough photoelectrochemical analysis with an efficient hole scavenger, we can distinguish absorption and catalytic contributions to the observed enhancement. The absorption enhancement ($\sim 33\%$) is mostly due to far-field effects (scattering), with a smaller contribution due to near-field localized surface plasmon resonances. This is markedly different from the mechanism in smaller (< 20 nm) metal plasmonic nanoparticles, where near-field effects dominate.⁴³ The experimental results agree well with the full-field electromagnetic simulations. The catalytic enhancement is twice as large as the absorption enhancement, and is tentatively assigned to the improvement of water oxidation kinetics due to the catalytic activity of Ag@SiO_2

nanoparticles. The effects combine to give a ~2.5-fold improvement of AM1.5 photocurrent at 1.23 V vs RHE, which is one of the highest photocurrent enhancements reported for a photoanode decorated with capped metal plasmonic nanoparticles. Such a large improvement is crucial for the development of BiVO₄ photoanodes, since modest absorption—especially for wavelengths close to the band edge—has been identified as one of the main performance-limiting factors.

Acknowledgements

We gratefully acknowledge the European Commission's Framework Project 7 (NanoPEC, Project 227179) for financial support of this work. We thank Ulrike Bloeck for the TEM imaging. F.F.A. thanks Dr. Andrea Baldi for valuable discussions.

Notes and references

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† Electronic Supplementary Information (ESI) available: experimental procedures, Figures S1-10. See DOI: 10.1039/b000000x/

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