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# Graphene Oxide-assisted Synthesis and Photocatalytic Hydrogen Production of Mix-phase Titanium Dioxide (TiO<sub>2</sub>) Nanosheets

Fanpeng Cai,<sup>a, b</sup> Yubin Tang,<sup>b</sup> Hao Shen,<sup>c</sup> Chao Wang,<sup>a</sup> Ao Ren,<sup>a</sup> Lisong Xiao,<sup>a</sup> Wei Gu,<sup>a</sup> Weidong Shi<sup>a,\*</sup>

Novel mix-phase TiO<sub>2</sub> nanosheets were successfully synthesized by a facile Sol-gel method via using graphene oxide (GO) as the sacrifice template for the first time. The SEM, TEM and AFM images revealed that the thickness of mix-phase TiO<sub>2</sub> nanosheets was ca. 66 nm, and the TiO<sub>2</sub> nanosheets were in a polycrystalline phase and made up of nanoparticles with the diameters of 15-25 nm. In addition, we observed that the proportion of rutile and anatase changed with the amount of tetrabutyltitanate (TBT). The results showed that the sample with 4 mL tetrabutyltitanate (TBT-4) exhibited the highest hydrogen production rate (339 µmol/h) in all samples, which was 47% higher than the bulk sample. It was mainly attributed to the surface phase junction of anatase and rutile and the special morphology of the samples.

**Keywords**: mix-phase  $TiO_2$  nanosheets, Sol-gel method, hydrogen production, graphene oxide as the template

### **1** Introduction

Energy and environmental issues at a global level caused by the burning of fossil fuels are important topics. It is urgent to construct new energy systems in order to solve these issues. Hydrogen as an ultimately clean energy provides an ideal alternative to replace fossil fuels. Photocatalytic water splitting is a promising approach to directly converting solar energy into hydrogen energy by producing  $H_2$  from water<sup>1</sup>. Since water and sunlight are naturally abundant, the photocatalytic water splitting is considered to be economically viable to produce hydrogen energy.

Among semiconductor catalysts with the ability of assist  $H_2$  production from water splitting, TiO<sub>2</sub> is the most promising candidate due to its super-high activity, chemical stability, non-toxicity and low cost <sup>2-5</sup>. To the best of our knowledge, the surface phase junction plays an important role in photocatalytic hydrogen production rate <sup>6</sup>. There have been many researchers and review articles showing that the photocatalytic hydrogen production rate of mix-phase TiO<sub>2</sub> is much higher than that of pure phase <sup>7-12</sup>. There were also researchers insisted that there was a synergetic effect exists in mix-phase TiO<sub>2</sub> <sup>13, 14</sup>. Bechstein et al. found that the mix-phase TiO<sub>2</sub> film with 60% anatase and 40% rutile exhibited optimal performance at a 50% enhanced activity compared with pure anatase <sup>15</sup>. Xu et al. proposed a hypothesis about the "intrinsic" photoactivity of mix-phase TiO<sub>2</sub> transfer from anatase phase to rutile <sup>16</sup>. All these researches have demonstrated that the surface phase junction greatly enhanced the photocatalytic performance of mix-phase TiO<sub>2</sub>.

Furthermore, the photocatalytic activity of  $TiO_2$ -based semiconductor is greatly affected by their morphology, including shape, size, porosity and surface feature. For example, Xu et al. reported the morphology had an important influence on photocatalytic activity <sup>17</sup>. Their studies showed the photocatalytic activities of anatase  $TiO_2$  nanocrystals change with facets exposed ratios. It is well reported that the 2D nanosheets have many unique advantages over other structures of photocatalysis such as the small weight, large surface area and the possession of more effective surface <sup>18-23</sup>.

Up to now, the TiO<sub>2</sub> sheets were prepared by various methods. Han et al. synthesized anatase TiO<sub>2</sub> nanosheets with a high percentage of exposed (001) facets via a simple hydrothermal route <sup>24</sup>. And Gu et al. synthesized anatase TiO<sub>2</sub> nanosheets supported on reduced graphene oxide via a solvothermal method <sup>25</sup>. However, the fabrication of mix-phase TiO<sub>2</sub> nanosheets was limited by synthetic methods and the nature of materials. Therefore, it is not a surprise that no papers on the synthesis of mix-phase TiO<sub>2</sub> nanosheets have been reported to date. That makes the design of novel mix-phase TiO<sub>2</sub> nanosheets for photocatalytic hydrogen generation is still a challenge to meet the requirement in research.

Herein, we for the first-time report the synthesis of mix-phase  $TiO_2$  nanosheets with controllable anatase and rutile contents by using GO as the template. The experimental result showed that the as-obtained mix-phase  $TiO_2$  nanosheets exhibited excellent H2 production activities, indicating that the synergetic effect of mix-phase  $TiO_2$  nanosheets plays an important role in the photocatalytic H<sub>2</sub> production rate.

### 2 Experimental

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### 2.1. Chemicals

Flake graphite,  $H_2SO_4$ , HCl,  $P_2O_5$  and KMnO<sub>4</sub> were purchased from Sinopharm (Beijing, China).  $H_2O_2$ , HAc, tetrabutyltitanate (TBT) and ethanol were purchased from Aladdin (Shanghai, China). All reagents were of analytical grade without further purification, and the deionized water was used in all experiments.

### 2.2. Synthesis of samples

Synthesis of GO: GO was synthesized from graphite by a modified Hummers method <sup>26-29</sup>. Firstly, a mixture of graphite (1 g), concentrated H<sub>2</sub>SO<sub>4</sub> (6 ml, 98%), K<sub>2</sub>S<sub>2</sub>O<sub>8</sub> (2 g) and P<sub>2</sub>O<sub>5</sub> (2 g) was heated at 80 °C for 5 h on a hotplate. The mixture was then cooled to room temperature and diluted with 200 ml water. After removing the acid by filtration and washing with water, this pre-oxidized graphite was dried in vacuum oven at 60 °C. The pre-oxidized graphite and 60 ml H<sub>2</sub>SO<sub>4</sub> (98%) was put into a 500-ml flask in an ice-water bath. Then, 8 g KMnO<sub>4</sub> was added into the flask and the temperature of the water bath was increased to 35 °C and allowed to stand for 2 h. After the mixture was diluted with 250 ml water, 20 ml H<sub>2</sub>O<sub>2</sub> was added into the flask. The solution turned yellow. The resulting graphite oxides were washed with HCl aqueous solution (1:10) to remove the acid and metal ions. Finally, the GO sheets were obtained by sonicating the graphite oxide.

Synthesis of GO@TiO<sub>2</sub> nanosheets: GO@TiO<sub>2</sub> nanosheets were synthesized according to Sol-gel process. 30 ml Acetic acid (HAc), tetrabutyltitanate (TBT) were added into a 250-ml beaker dropwise under vigorous stirring. The amount of TBT changes from 1 ml to 5 ml. After vigorous stirring 30 min in the room temperature, the GOs solutions (GOs dispersed in ethanol) were added into the beaker. After 6 h, the composites were washed with ethanol by repeated centrifugation and dried in a vacuum oven.

Synthesis of TiO<sub>2</sub> nanosheets: the GO@TiO<sub>2</sub> nanosheets were calcined at 600 °C in air for 6 h. The powder tuned to totally white from black after the calcination, indicating the removing of the GOs. The Schematic illustrations of the synthetic process of TiO<sub>2</sub> nanosheets were shown in Scheme. 1. The bulk sample was synthesized without GOs. In a typical procedure, 5 mL TBT droped wise in 30mL HAc to form a mix-solution, after stirring for 30 min, add 40 mL ethanol into the previous mix-solution and stirring for another 6h. Then the composites were washed with ethanol by repeated centrifugation and dried in a vacuum oven. After that, the composites were calcined at 500 °C in air for 6 h to obtain the high crystallized anatase TiO<sub>2</sub> particles.

Scheme. 1. Schematic illustrations of the synthetic process of  $TiO_2$  nanosheets.



### 2.3. Characterization

The products were characterized by X-ray diffraction measurements carried out on X-ray diffractometer (XRD, Bruker D8 Advance diffractometer) with Cu-K $\alpha$  radiation in the range of 5-80° at a scanning rate of 7°/min. Scanning electron microscopy (SEM) images were collected on an S-4800 field emission SEM (FESEM, Hitachi, Japan). Transmission electron



### 2.4. Photocatalytic hydrogen production

The photocatalytic hydrogen production experiments were carried out in a Lab-H<sub>2</sub> photocatalytic hydrogen production system. A 300W Xenon arc lamp was used as a light source and was positioned 20 cm away from the reactor in the system. In a typical photocatalytic experiment, 50 mg of the prepared TiO<sub>2</sub> photocatalyst was dispersed with constant stirring in 200 mL 20% methanol aqueous solution. Prior to irradiation, the system was vacuumized to remove the dissolved oxygen. During the whole reaction process, vigorous agitation was performed to ensure the uniform irradiation of the TiO<sub>2</sub> photocatalysts suspension. A certain amount of H<sub>2</sub>PtCl<sub>6</sub>•6H<sub>2</sub>O aqueous solution was dripped into the system to load 0.5 wt% Pt onto the surface of the photocatalyst by a photochemical reduction deposition method. The generated gas was collected intermittently through the septum, and hydrogen content was analyzed by the gas chromatograph (GC-SP7800, Beijing Jing Ke Ruida, China, TCD, nitrogen as a carrier gas and 5 Å molecular sieve column). All glassware was rigorously cleaned and carefully rinsed with distilled water prior to use.

Japan) photoluminescence detector. A Netzsch model STA 449

C thermal analyzer was used in the present investigation.

### 3. Results and discussion

### 3.1. Structure and morphology





A series of samples was synthesized by controlling different quantity of TBT with 1 mL, 2 mL, 3 mL, 4 mL and 5 mL, named as TBT-1, TBT-2, TBT-3, TBT-4 and TBT-5, respectively. The crystallographic structure and phase purity of all samples were examined by X-ray powder diffraction (XRD) studies. As shown in Fig. 1a, the XRD patterns of all samples can be indexed into mix-phase TiO<sub>2</sub>. Particularly, as shown in the magnified XRD patterns in the range from  $2\theta=24^{\circ}$  to  $29^{\circ}$ (Fig. 1b), the percentage of rutile phase decreased as the amount of TBT increased.

To be more specific, a commonly used method to determine the weight percentage of the rutile  $(W_R)$  portion in the mixture

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of rutile and anatase was applied as shown in the following equation  $^{30-32}$ :

$$W_R = \frac{A_R}{0.884A_A + A_R}$$

Here, the  $A_R$  stands for the integrated intensity of rutile (110) peak, and the  $A_A$  represents the integrated intensity of anatase (101) peak. When 1 mL TBT was added, the sample TBT-1 contained 47.8% anatase, and rutile 52.2%. While increased the TBT to 5 mL, the percentage of anatase increased to 81.1%, with only 18.9% of rutile remaining in the sample TBT-5. The detailed comparison for phase composition is summarized in Table 1. The XRD patterns of the sample before calcined was inserted in Fig. S2, which can be indexed into anatase phase with poor crystallinity.

Table 1 Summary of properties of samples synthesizedfrom different amount TBT.					
Sample ID	amount TBT (mL)	of	Anatase %	Rutile %	Surface area (m <sup>2</sup> /g)
TBT-1	1 mL		47.8	52.2	89
TBT-2	2 mL		49.9	51.1	91
TBT-3	3 mL		51	49	53
TBT-4	4 mL		69.4	31.6	59
TBT-5	5 mL		81.1	18.9	41

No GO peak was observed at  $2^{\circ}-10^{\circ}$  in XRD patterns of all samples, indicating that GO has been removed in all samples. The GO's characteristic peaks in Raman spectra are D-band (1350 cm<sup>-1</sup>) and G-band (1588 cm<sup>-1</sup>), and they were not detected in the calcined samples (Fig. 2). The result is consistent with the removal of the GO in the process of calcination.



Fig. 2. Raman spectra of GO (red) and TBT-5 (black).



**Fig. 3.** a) the SEM image, b and c) the TEM image, d) the HRTEM image for TBT-4, respectively. The insert is SAED image for TBT-4.

The morphology and size of as-synthesized TiO<sub>2</sub> nanosheets were studied through the scanning electron microscope (SEM), transmission electron microscope (TEM) and high-resolution transmission electron microscope (HRTEM). As shown in the SEM image (Fig. 3a), TBT-4 sample is composed of numerous sheet structures with thickness of about 66 nm. The TEM image (fig. 3b) further reveals the sheet structure of TBT-4 sample with near transparent appearance and wrinkles on the surface. As can be seen in the higher magnification (Fig. 3c), the asprepared TiO<sub>2</sub> nanosheets presented in a polycrystalline phase and assembled by well-dispersed 15-25 nm nano-particles. The selected area electron diffraction (SAED) pattern (inset of Fig. 3c) can be indexed to the anatase  $TiO_2$  (101) and rutile  $TiO_2$ (110) diffractional rings, which reveals the polycrystalline nature of nanosheets and confirms that the nanosheets contain anatase and rutile TiO<sub>2</sub>. The corresponding HRTEM image (Fig. 3d) showed clear lattice fringes on the edge, which correspond to the (101) plane of anatase  $TiO_2$  phase with interplanar spacing d = 0.35 nm and the (110) plane of rutile TiO<sub>2</sub> phase with interplanar spacing d = 0.32 nm, respectively. This result is in good agreement with the XRD analysis and selected area electron diffraction (SAED) pattern. Low magnification SEM and TEM images of TBT-4 sample (Fig. S3) were also provided to demonstrate the sheet structure and graphene-like morphology of our sample.

In our present work, we obtain  $TiO_2$  nanosheets by a facile Solgel method with GO as the sacrifice template. Because of the distribution of carboxyl groups on the surface of GO, the  $TiO_2$ precursor dispersed on the carbon support and were eager to accumulate along the carboxyl groups. After Sol-gel process, layered  $TiO_2$  precursor covered on the surface of GO. This process was shown in Fig.S1. As a result, change the dosage of TBT may obtain  $TiO_2$  nanosheets with different thickness. In order to confirm the fact, we provide TEM images (Fig.S4) of all other samples in the supporting information. No obvious morphological difference among these samples with different TBT amounts were observed.



**Fig. 4.** a) AFM images of TBT-4 nanosheets, with the inset below showing the line profiles along the red line. b) The AFM 3D image of TBT-4.

In order to investigate the morphology and the thickness of the  $TiO_2$  nanosheets, we study the materials using an atomic force microscope (AFM). Fig. 4a showed the 2D AFM images of  $TiO_2$  nanosheets and its' line profiles along the red line. The red marker corresponds to  $TiO_2$  nanosheets with a height of 87 nm. Fig. 4b shows 3D AFM micrographs of the  $TiO_2$  nanosheets. Clearly the individual  $TiO_2$  presented sheet's structure. And we can observe wrinkle-like features as graphene on the  $TiO_2$  nanosheets. Basically, we can find the surface of  $TiO_2$  nanosheets is very smooth.

It is well acknowledged that surface area is very important to the photocatalytic activity of  $TiO_2$ . The material templated by graphene oxide has an opportunity to possess huge specific surface area values. N<sub>2</sub> adsorption–desorption isotherms were recorded on ASAP 2020 instrument, from which the surface areas were calculated by applying the Brunauer-Emmett-Teller (BET) method. We listed the values of specific surface area of all samples in table 1. TBT-2 has the largest values of specific surface area (ca. 91 m<sup>2</sup>/g), and TBT-5 has the smallest values of specific surface area (ca. 41 m<sup>2</sup>/g). The relative large values of specific surface area may have a great impact on photocatalytic activity of as-prepared TiO<sub>2</sub>.



Fig. 5. The UV-vis absorption spectra of all samples.



**Fig. 6.** Comparison of  $H_2$  evolution rates of all samples using 20% methanol aqueous solution and 0.5 wt% Pt as a cocatalyst; a 300W xenon arc lamp was used as the light source.

### 3.2. Photocatalytic activity

Fig. 5 shows the UV-vis absorption spectrum of all samples. There was no significant increase of the absorption in the light absorption range. As shown in Fig. 5, with increasing the amount of TBT in the system, the absorption edge of the samples had no obviously shifts, reflecting that the band gap values of TiO<sub>2</sub> obtained were close. We have conducted the hydrogen production experiment, and the results were shown in Fig. 6. The TBT-4 (339  $\mu$ mol/h) possess the highest hydrogen production rate whereas the TBT-5 (142  $\mu$ mol/h) displays the lowest hydrogen production rate of bulk sample and P25 were 230  $\mu$ mol/h and 446  $\mu$ mol/h, respectively.

In generally, there are two major factors affecting the photocatalytic activity of TiO<sub>2</sub>, including the A/R ratio and the surface area. On the one hand, Gang et al. reported that the work function of rutile is 0.2 eV lower than that of anatase, the electron affinity of anatase is higher than rutile<sup>33, 37</sup>. This would favor the transfer of photoexited electrons from rutile to anatase, and the transfer of holes from anatase to rutile at a clean interface, thereby effectively suppressing the recombination of photoexited electrons and holes <sup>34, 35</sup>. The reaction mechanism was thus briefly described in Scheme 2. Due to the effective charge separation, the synergetic effect between anatase and rutile leads to the high photocatalytic activity of mix-phase TiO<sub>2</sub> nanosheets <sup>36, 37</sup>. On the other hand, many studies have reported that the surface area of photocatalyst has a very strong effect on the photochemical reactive sites. In other words, the

Scheme. 2. The reasonable photo-generated electron-hole conduction mechanism under a 300W Xenon arc lamp irradiation on  $TiO_2$  surface phase junction.



However, the surface area in our case was not the key factor affecting the photocatalytic hydrogen production rate. Specifically, from TBT-1 to TBT-4, the photocatalytic hydrogen production rate increased with the amount of anatase phases. This phenomenon implied that the A/R ratio played an important role in the photocatalytic hydrogen production of the mix-phase TiO<sub>2</sub> nanosheets rather than the surface area. Obviously, from TBT-1 to TBT-4, the synergetic effect between anatase and rutile to be responsible for the pretty high photocatalytic activity. For TBT-4, the amount of rutile was exactly suitable to assist electron-hole pair separation. This result could also be verified from photoluminescence emission (Fig. 7). In other words, too many rutile made the electron-hole recombination rate higher than the electron conductivity, thus, the photocatalytic hydrogen production rate was decreased with an increased in a rutile amount. That's why the photocatalytic hydrogen production rate of TBT-4 is higher than that of TBT-1, TBT-2 and TBT-3. It's also not surprise that the photocatalytic hydrogen production rate of TBT-4 is higher than that of pure anatase phase of bulk TiO2. Nevertheless, for TBT-5, the photocatalytic hydrogen production rate had a dramatic decreased, which was attributed to the smallest specific surface area and the lowest rutile phase composition. That was to say the amount of rutile is insufficient to assist electron-hole pair separation. In addition, the 2D TiO<sub>2</sub> nanosheets structures also had a contribution to the photocatalytic hydrogen production rate, because sheet structures have more active sites than 0D and 1D structures and exhibit more effective surface than other structures <sup>18, 19</sup>.



Fig. 7. The photoluminescence spectra of all samples (10mg were well-dispersed into 5 mL of ethanol) with excitation wavelength of 280 nm.

### 3.3. Photoluminescence spectra

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The PL emission intensity is related to the recombination of 7 excited electrons and holes, and thus, the lower emission intensity is indicative of a decrease in recombination rate and an increase the photocatalytic activity <sup>41-43</sup>. Photoluminescence (PL) emission spectra of all samples showed in Fig. 7. The 9 measurement was conducted at the excitation wavelength of 280 nm with a photomultiplier tube voltage of 500 V. The PL spectra of all samples showed a broadband with a maximum around 536 nm. According to the PL results, the increase in the TBT amount caused a decrease in photoluminescence emission, although the band-gap energy did not change significantly (Fig. 5). This indicates the electron-hole recombination rate of TBT-4 was much lower than those of other samples. The result is consistent with the photocatalytic hydrogen production rate of all samples.

### Conclusions

In summary, the novel  $TiO_2$  nanosheets were successfully synthesized by a facile Sol-gel method with graphene oxide as the sacrifice template. The as-prepared TiO<sub>2</sub> nanosheets were in a polycrystalline phase which made up of nano-particles, and the nano-particles were dispersed with diameters of 15-25 nm. The results show that TBT-4 sample exhibits the highest hydrogen production rate (339 µmol/h) among all samples, which is 47% higher than the bulk sample. This is because the electron-hole recombination rate of TBT-4 was much lower than those of other samples, thus makes the photocatalytic hydrogen production rate much higher.

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### Notes and references

<sup>a</sup> School of Chemistry and Chemical Engineering, Jiangsu University, Xuefu Road 301, Zhenjiang, 212013, P. R. China.

<sup>b</sup> School of Biology and Chemical Engineering, Jiangsu University of Science and Technology, Zhenjiang, 212003, P. R. China.

<sup>c</sup> Institute of Inorganic Chemistry, University of Cologne, Greinstrasse 6, D-50939 Cologne, Germany.

\*Corresponding author: Tel.: +86 511 8879 0187 Fax. : +86 511 8879 33 1108

E-mail address: swd1978@ujs.edu.cn(W. Shi)

- 1 A. Kudo and Y. Miseki, Chemical Society reviews, 2009, 38, 253-278
- 2 X. Chen and S. S. Mao, Chemical reviews, 2007, 107, 2891-2959.
- 3 M. Addamo, V. Augugliaro, M. Bellardita, A. Paola, V. Loddo, G. Palmisano, L. Palmisano and S. Yurdakal, Catalysis Letters, 2008,
- 37 126, 58-62 4 R. Daghrir, P. Drogui and D. Robert, Industrial & Engineering
- Chemistry Research, 2013, 130226090752004.
- 5 H. B. Wu, H. H. Hng and X. W. Lou, Advanced materials, 2012, 38 24, 2567-2571.
- 6 X. Wang, Q. Xu, M. Li, S. Shen, X. Wang, Y. Wang, Z. Feng, J. 39 Shi, H. Han and C. Li, Angewandte Chemie, 2012, 51, 13089-40 13092.

- A. Zachariah, K. V. Baiju, S. Shukla, K. S. Deepa, J. James and K. G. K. Warrier, Journal of Physical Chemistry C, 2008, 112, 11345-11356.
- R. G. Nair, S. Paul and S. K. Samdarshi, Solar Energy Materials and Solar Cells, 2011, 95, 1901-1907.
- D. A. H. Hanaor, I. Chironi, I. Karatchevtseva, G. Triani and C. C. Sorrell, Advances in Applied Ceramics, 2012, 111, 149-158.
- H. Li, B. Xu and Y. Fan, Chemical Physics Letters, 2013, 558, 66-71.
- H. Tian, G. Zhao, Y.-n. Zhang, Y. Wang and T. Cao, Electrochimica Acta, 2013, 96, 199-205.
- M. Niu, D. Cheng and D. Cao, The Journal of Physical Chemistry *C*, 2014, 140305155820004.
- T. Ohno, K. Sarukawa, K. Tokieda and M. Matsumura, Journal of Catalysis, 2001, 203, 82-86.
- 14 D. C. Hurum, A. G. Agrios, K. A. Gray, T. Rajh and M. C. Thurnauer, The Journal of Physical Chemistry B, 2003, 107, 4545-4549.
- 15 R. Su, R. Bechstein, L. Sø, R. T. Vang, M. Sillassen, B. Esbjörnsson, A. Palmqvist and F. Besenbacher, The Journal of Physical Chemistry C, 2011, 115, 24287-24292.
  - S. Cong and Y. Xu, The Journal of Physical Chemistry C, 2011, 115, 21161-21168.
  - J. Li, Y. Yu, Q. Chen, J. Li and D. Xu, Crystal Growth & Design, 2010, 10, 2111-2115.
  - J. Liu and X. W. Liu, Advanced materials, 2012, 24, 4097-4111.
  - E. L. Tae, K. E. Lee, J. S. Jeong and K. B. Yoon, Journal of the American Chemical Society, 2008, 130, 6534-6543.
  - T. Huang and D. Qiu, Langmuir : the ACS journal of surfaces and colloids, 2014, 30, 35-40.
  - Y. Li, D. Cao, Y. Liu, R. Liu, F. Yang, J. Yin and G. Wang, International Journal of Hydrogen Energy, 2012, 37, 13611-13615.
  - L. Zhang, W. Wang, Z. Chen, L. Zhou, H. Xu and W. Zhu, Journal of Materials Chemistry, 2007, 17, 2526.
  - X. Hu, X. Shen, R. Huang, Y. Masuda, T. Ohji and K. Kato, Journal of Alloys and Compounds, 2013, 580, 373-376.
  - X. Han, Q. Kuang, M. Jin, Z. Xie and L. Zheng, Journal of the American Chemical Society, 2009, 131, 3152-3153.
  - L. Gu, J. Wang, H. Cheng, Y. Zhao, L. Liu and X. Han, ACS applied materials & interfaces, 2013, 5, 3085-3093.
  - Z. Lu, J. Zhu, D. Sim, W. Zhou, W. Shi, H. H. Hng and Q. Yan, Chemistry of Materials, 2011, 23, 5293-5295.
  - C. Bao, L. Song, W. Xing, B. Yuan, C. A. Wilkie, J. Huang, Y. Guo and Y. Hu, Journal of Materials Chemistry, 2012, 22, 6088.
  - W. Tu, Y. Zhou and Z. Zou, Advanced Functional Materials, 2013, 23, 4996-5008.
  - E. V. Iski, E. N. Yitamben, L. Gao and N. P. Guisinger, Advanced Functional Materials, 2013, 23, 2554-2564.
  - H. Zhang and J. F. Banfield, The Journal of Physical Chemistry B, 2000, 104, 3481-3487.
  - F. Zuo, L. Wang and P. Feng, International Journal of Hydrogen Energy, 2014, 39, 711-717.
  - J. G. Yu, J. C. Yu, M. K. P. Leung, W. K. Ho, B. Cheng, X. J. Zhao and J. C. Zhao, Journal of Catalysis, 2003, 217, 69-78.
  - G. Xiong, R. Shao, T. C. Droubay, A. G. Joly, K. M. Beck, S. A. Chambers and W. P. Hess, Advanced Functional Materials, 2007, 17, 2133-2138.
  - D. C. Hurum, K. A. Gray, T. Rajh and M. C. Thurnauer, The journal of physical chemistry. B, 2005, 109, 977-980.
  - J. Zhang, Q. Xu, Z. Feng, M. Li and C. Li, Angewandte Chemie, 2008, 47, 1766-1769.
  - P. Deák, B. l. Aradi and T. Frauenheim, The Journal of Physical Chemistry C, 2011, 115, 3443-3446.
  - D. O. Scanlon, C. W. Dunnill, J. Buckeridge, S. A. Shevlin, A. J. Logsdail, S. M. Woodley, C. R. Catlow, M. J. Powell, R. G. Palgrave, I. P. Parkin, G. W. Watson, T. W. Keal, P. Sherwood, A. Walsh and A. A. Sokol, Nature materials, 2013, 12, 798-801.
    - A. G. Prado and L. L. Costa, Journal of hazardous materials, 2009, 169, 297-301.
  - A. T. Bell, Science, 2003, 299, 1688-1691.
  - J. Wang, Z. Bian, J. Zhu and H. Li, Journal of Materials Chemistry A, 2013, 1, 1296.

- S. K. Choi, S. Kim, S. K. Lim and H. Park, *The Journal of Physical Chemistry C*, 2010, **114**, 16475-16480. N. Murakami, Y. Kurihara, T. Tsubota and T. Ohno, *The Journal of Physical Chemistry C*, 2009, **113**, 3062-3069. A. A. Ismail and D. W. Bahnemann, *Journal of Materials* 41
- 42
- 43 Chemistry, 2011, 21, 11686.



Fig. a) the SEM image, b and c) the TEM image, d) the HRTEM image for TBT-4, respectively. The insert is SAED image for TBT-4.