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ARTICLE TYPE

In situ preparation of 3D graphene aerogels@hierarchical Fe_3O_4 nanoclustersas high rate and long cycle anode materials for lithium ion batteries

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We describe a novel strategy for in situ fabrication of hierarchical Fe₃O₄ nanoclusters/GAs. Fe₃O₄ NCs/GAs deliver an excellent rate capability (the reversible capacities obtained ¹⁰ were 1442, 392 and 118 mAhg⁻¹ at 0.1C, 12C and 35C rates), and a high reversible capacity of 577 mAh g⁻¹over 300 cycles at the current density of 5.2 A/g (6C).

Rechargeable lithium-ion batteries (LIB) are key device for mobile communication devices, portable electronic devices, and ¹⁵ electrical/hybrid vehicles (EVs/EHVs). The main challenges in this field are achieving high capacity, excellent cycling performance and rate capability for both anode and cathode materials to meet the increasing power supply requirements for a variety of applications.^[1] However, the limited theoretical ²⁰ capacity of 372 mAhg⁻¹ of the commercial anode material of

- ²⁰ capacity of 572 mixing for the commercial anode material of graphite cannot fully meet the requirement.^[2] For the purpose of improving the electrochemical performance of LIBs, considerable efforts have been made recently in searching new electrode materials, which always have higher energy density compared ²⁵ with the existing system.^[3] Fe₃O₄ is considered as a promising
- ²⁵ with the existing system.²⁵ Fe₃O₄ is considered as a promising anode material candidate because of its high theoretical capacity, natural abundance, nontoxic nature, and low cost.^[3d, 4] Nevertheless, its application in practical LIB is hindered due to its low rate performance arising from kinetic limitations due to
- ³⁰ low intrinsic electric conductivity and poor cycling stability resulting from large volume expansion occurring during cycling.^[5] For these challenges, some strategies have been made to resolve these issues significantly. The first strategy is to introduce Fe₃O₄ nanostructure resulting in shorter path lengths for
- $_{35}$ the transport of electrons and lithium ions. $^{[3b,6]}$ The other promising strategy is to construct hybrid electrodes composed of Fe₃O₄ and conductive additive providing rapid electron channel. $^{[3d,4a,4c,5a,7]}$
- More recently, compared with 2D graphene, 3D graphene ⁴⁰ aerogels (GAs) have drawn much more attention due to its strong mechanical strengths, fast mass and electron transport rate benefiting from the 3D interconnected framework and the intriguing properties of graphene. Therefore, constructing GAs hybrid 3D hierarchical Fe₃O₄ will be an effective way to achieve ⁴⁵ both longer cycling life and higher rate performance for LIBs.

In this work, we describe a novel strategy which is both facile and cost effective for in situ fabrication of 3D hierarchical Fe_3O_4 nanoclusters /GAs (Fe₃O₄ NCs/GAs). Herein, hierarchical Fe₃O₄ NCs can effectively bring down the mean diffusion time of the ⁵⁰ lithium ions by minimizing the diffusion length, and the GAs provide protection against the volume changes of Fe₃O₄ NCs in 3D space during electrochemical process. Meanwhile, the

interconnected 3D networks and channel abundance can enhance

the electrical conductivity and ion diffusion. The fabrication process of 3D Fe₃O₄ NCs/GAs is illustrated in Figure 1a (the detail in ESI). The morphology and microstructure of Fe₃O₄ NCs and Fe₃O₄ NCs/GAs were examined by scanning electron microscopy (SEM), transmission electron microscope (TEM), and high resolution TEM (HRTEM) measurement. SEM and TEM images confirm that Fe₃O₄ NCs shown uniformly nanocluster structure (Figure S1) and the Fe₃O₄NCs are well wrapped by the graphene (Figure 1b, c). This feature can be characterized by EDX mapping of the elements C, Fe, and O.



Figure 1. a) Schematic diagram of the fabrication of 3D Fe₃O₄ NCs/GAs. b, c, d) SEM and TEM image of the Fe₃O₄NCs/GAs. e, f) The

corresponding elemental mapping (C, Fe and O) and HRTEM images of Fe_3O_4 NCs/GAs, inset: SAED pattern.

nanoparticles (5nm) .The SAED pattern of a single Fe_3O_4 NCs shows a very light ring-pattern in Figure 1e. The diffraction rings s can be indexed to (200), (311), (400), (422), (440) and (511) planes of a face-centered cubic magnetite phase.

The pattern confirmed the formation of Fe₃O₄ (JCPDS no.19-0629) in the Fe₃O₄ NCs/GAs (Figure 2a).^[8] From the TGA, it was estimated that the amount of graphene in the composites was

- ¹⁰ about ~33 wt% (Figure S2). Brunauer-Emmett-Teller analysis shows a specific surface area of 118 m²/g for Fe₃O₄ NCs/GAs (Figure 2b), which is much higher than the previously reported for Fe₃O₄/graphene composites.^[4c,9] The observed type hysteresis loop confirms the mesoporous structure. Based on the Barrett-
- ¹⁵ Joyner-Halenda model, a well-defined mesopore (5.6nm) was obtained (Figure S3), which contributed to the pores existing between the Fe_3O_4 NPs. The mesoporous configuration benefits electrolyte diffusion to active sites. The macropores of 50-250 nm attribute to the pores existing between nanocluster and 3D
- ²⁰ graphene, which is benefit to transport of the electrolyte and enhancing the rate performance.^[10] The Fe 2p XPS spectrum of the Fe₃O₄ NCs/GAs (Figure S4) exhibits two peaks at 710.7 and 724.8 eV, corresponding to the Fe $2p_{3/2}$ and Fe $2p_{1/2}$ of Fe₃O₄.^[8a,9a,9b,11] The presence of Fe₃O₄ can be further confirmed ²⁵ by the O1s predominant peak at 533.0 eV, which corresponds to

the oxygen species in the Fe₃O₄ phase.^[11]



Figure 2. a) XRD profiles of Fe₃O₄ NCs/GAs composite. b) N₂ adsorption/desorption isotherms of Fe₃O₄ NCs/GAs. c,d) XPS ³⁰ spectrum of the Fe₃O₄ NCs/GAs and Fe₃O₄, and the high resolution spectra of Fe 2p and O 1s.

The insertion/extraction of Fe₃O₄ NCs/GAs anodes was characterized using cyclic voltammetry (CV). As shown in Figure 3a, three peaks, located at 1.58, 0.90 and 0.75 V, were observed ³⁵ in the first insertion process. The two insertion peaks at higher voltage disappeared in the subsequent cycles, and they were possibly attributed to lithium intercalation to formation $Li_xFe_3O_4^{[12]}$ and the formation of a solid electrolyte interphase (SEI) film caused by the decomposition of electrolyte, which ⁴⁰ contributes to the irreversible capacity, as well as the three steps of the lithiation reaction of Fe₃O₄.^[8b,13] The sharp reduction peak at about 0.75 V is observed in the first cathodic scan for the Fe₃O₄ NCs/GAs, which can be attributed to the reduction of Fe³⁺/Fe²⁺ to Fe⁰. The extraction process presents two well-defined





Figure3. a) CV curve of Fe₃O₄ NCs/GAs nanocomposites in the potential window 0.01V-3V with 0.01mV/s scanning rate. b) Rate performance of the Fe₃O₄ NCs/GAs nanocomposites at different ⁵⁰ current densities. c) Cycling performance of the Fe₃O₄ NCs/GAs nanocomposites at a current density of 6C for 300 cycles.

nanceomposites at a current density of de for 500 cycles. oxidation of Fe^0 to $Fe^{2+}/Fe^{3+}[13b,14]$ From the second cycle, both cathodic and anodic peaks are positively shifted in the subsequent cycles which attribute to the structural modification after the first so cycle due to lithium insertion and extraction,^[12a-b,13a] which is easy to insertion. It also worth noted that the peaks at 0.8 V during discharge and at 1.61 and 1.81 V during charge (Fe₃O₄ \leftrightarrow Fe) for the second or third cycles of the Fe₃O₄ NCs/GAs electrode are almost overlapped. This can be indexed to the high or reversibility of Fe₃O₄ NCs/GAs, indicating that the unique structure can improve the sufficient oxidation reaction process during the anodic cycling and insure excellent reversibility of the Fe₃O₄ NCs/GAs anode.

Galvanostatic discharge/charge at 0.1C of the cell exhibited a 65 first cycle reversible capacity of 1442 mAhg⁻¹ (Figure S5).The reversible capacity observed is higher than the theoretical capacity for Fe₃O₄based anodes, which is 926 mAhg⁻¹ following an 8e⁻ transfer reaction per formula weight.^[8a,9a-b,10,13c,14] However, such higher-than-theoretical capacity values have been 70 observed before for transition metal oxides anodes, which is attributed to the formation of a pseudo-capacitive gel like film.^[4d,13c,14] The reversible capacities obtained were 1442, 1221, 1020 and 845 mAhg-1 for 0.1, 0.2, 0.5 and 1.2 C rates (considering 865 mAg⁻¹ as 1C for simplicity Figure 3c, S6), 75 respectively. The cell was further cycled at an extremely high current rate of 12 C and 35 C for 10 cycles. The reversible specific capacity was 392 and 118 mAhg⁻¹. Additionally, the capacity was exceptionally stable at such a high current rate. More importantly, when the current rate was set back to 0.1C ⁸⁰ after such high rate cycling, the capacity of the cell recovered to 1200 mAhg⁻¹, suggesting that the hierarchical nanostructure was not dramatically affected by the high rate charge/discharge cycles.

The cyclic stability of Fe_3O_4+GAs and $Fe_3O_4 NCs/GAs$ was ⁸⁵ measured. The capacity and cyclic stability of the Fe_3O_4 NCs/GAs composite are higher than the Fe_3O_4+GAs at a 1C (Figure S7), which attribute to the strong anchoring of Fe_3O_4 NCs on GAs empowers fast electron transport through graphene layers to Fe_3O_4 NCs to enhance the electrochemical performance. A cell was cycled at 5200 mAg⁻¹ (i.e. 6C) for 300 cycles (Figure 3c). Fe_3O_4 NCs/GAs composite presents excellent cycling stability with the initial discharge capacity of 1000 mAh g⁻¹ and the ^s discharge capacity of 577 mAhg⁻¹ until 300 cycles. To our knowledge, the 3D hierarchical Fe₃O₄ NCs/GAs composite shows

- excellent C-rate retention and cycling capability than the anodes of Fe_3O_4 or Fe_3O_4 graphene hybrids/composition reported in recent literature, ^[4-7,8,9a-b, 13c,14] which good confinement of active
- ¹⁰ materials by the graphene 3D network and Fe_3O_4 NCs further compensated for the volume due to ultrasmall nanoparticles of Fe_3O_4 NCs and the effective nanopores each other (Figure S8,9).
- To verify the good performance of the Fe₃O₄ NCs/GAs, ac impedance measurements were also conducted as shown in ¹⁵ Figure S10, which indicates that the charge-transfer resistances of
- the Fe₃O₄ NCs/GAs electrodes are smaller than Fe₃O₄ electrode. In addition, we also found that the diameter of the semicircle was decrease after 300 cycles for Fe₃O₄ NCs/GAs. Therefore, it can be assumed that the Fe₃O₄ inter-particle resistance of the
- ²⁰ electrode was suppressed by the *in situ* self-assembly GAs. This result demonstrated that the GAs serves as a conductive network, which improves the local conductivity, and then leads to a high electrochemical performance for the Fe₃O₄ NCs/GAs electrodes as anode materials for LIBs.

25 Conclusions

In conclusion, novel hierarchical Fe₃O₄ NCs encapsulated in GAs (Fe₃O₄ NCs/GAs) have been successfully fabricated by a facile and scalable *in situ* synthesis method. In this architecture, hierarchical Fe₃O₄ NCs can suppress the stacking of graphene

- $_{30}$ layers, and effectively bring down the mean diffusion time of the lithium ions. The GAs can effectively accommodate the mechanical stress induced by the volume change of embedded $\rm Fe_3O_4$ and provide excellent electron transfer and ion diffusion during the charge and discharge processes. Due to the synergistic
- ³⁵ effects, the Fe₃O₄ NCs/GAs exhibits an extremely excellent rate capability (The reversible capacities obtained were 392 and 118 mAhg⁻¹ for 12 C and 35 C rates), and a long cycle performance (reversible capacity of 577 mAh g⁻¹ over 300 cycles at the current density of 5.2 A/g).
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Notes and references

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- 1 a) M.R. Palacin, Chemical Society Reviews2009, 38, 2565. b) M.
- Armand, J.M. Tarascon, *Nature* 2008, 451, 652. c) B. Dunn, H. Kamath, *Science* 2011, 334, 928. d) J.M. Tarascon, M. Armand, *Nature* 2001, 414, 359. e) J. B. Goodenough, Y. Kim, *Chem. Mater.* 2010,22, 587.

- 2 a) A. Manthiram, J. Phys. Chem. Lett. 2011, 2, 176. b) J. Cabana, L. Manaanduit, D. Larahar, M.P. Palaain, Adv. Mater. 2010, 22, E170.
- Monconduit, D. Larcher, M.R. Palacin, Adv.Mater. 2010, 22, E170.
 a) Y. Idota, T. Kubota, A. Matsufuji, Y. Maekawa, T. Miyasaka, Science 1997, 276, 1395. b) P. Poizot, S. Laruelle, S.Grugeon, L. Dupont, J.M. Tarascon, Nature 2000, 407, 496. c) C.K. Chan, H. Peng, G. Liu, K.M. Wrath, X.F. Zhang, R.A. Huggins, Y. Cui, Nat. Nanotechnol. 2008, 3, 31. d) P.L. Taberna, S. Mitra, P. Poizot, P. Simon, J.M. Tarascon, Nat.Mater. 2006, 5, 567. e) S.M. Xu, C.M. Hessel, H. Ren, R.B. Yu, Q. Jin, M. Yang, H.J. Zhao, D. Wang, Energy. Environ. Sci. 2014, 7, 632. f) J.Y. Wang, N.L. Yang, H.J. Tang, Z.H. Dong, Q. Jin, M. Yang, D. Kisailus, H.J. Zhao, Z.Y. Tang, D. Wang, Angew. Chem. Int. Ed. 2013, 52, 6417, g) X.L. Jia,
 - Tang, D. Wang, Angew. Chem. Int. Ed. 2013, 52, 6417. g) X.L. Jia, Z. Chen, X. Cui, Y.T. Peng, X.L. Wang, G. Wang, F. Wei, Y.F. Lu, ACS Nano, 2012, 6, 9911. h) X.J. Zhu, Y.W. Zhu, S. Murali, M.D. Stoller, R.S. Ruoff, ACS Nano, 2011, 5, 3333.
- a) Z.S. Wu, W.C. Ren, L. Wen, L.B. Gao, J.P. Zhao, Z.P. Chen, G.M.
 Zhou, F. Li, H.M. Cheng, ACS Nano 2010, 4, 3187. b) Z.M. Cui,
 L.Y. Hang,W.G. Song, Y.G. Guo, Chem. Mater. 2009, 21, 1162. c)
 G.M. Zhou, D.W. Wang, F. Li, L.L. Zhang, N. Li, Z.S. Wu, L. Wen,
 G.Q. Lu, H.M. Cheng, Chem. Mater. 2010, 22, 5306. d) Y. He, L.
 Huang, J.S. Cai, X.M. Zheng, S.G. Sun, Electrochim.Acta 2010, 55,
 1140.
- a) T. Muraliganth, A.V. Murugan, A. Manthiram, *Chem. Commun.* 2009, 47, 7360. b) N. Kang, J.H. Park, J. Choi, J. Jin, J. Chun, I.G. Jung, J. Jeong, J.G. Park, S.M. Lee, H.J. Kim, S.U. Son, *Angew. Chem. Int. Ed.* 2012, 51, 6626. c) L.W. Ji, Z. Lin, M. Alcoutlabi, X.W. Zhang, *Energy. Environ. Sci.* 2011, 4, 2682. d) J.S. Chen, Y.M. Zhang, (David) X.W. Lou, *ACS Appl. Mater. Interfaces* 2011, 3, 3276.
 - a) J. Maier, Nat. Mater. 2005, 4, 805, b) B. Koo, H. Xiong, M.D. Slater, V.B. Prakapenka, M. Balasubramanian, P. Podsiadlo, C.S.
- Johnson, T. Rajh, E.V. Shevchenko, *Nano Lett.* 2012, 12, 2429. c) S. Wang, J. Zhang, C. Chen, *J. Power Sources* 2010, 195, 5379. d) H. Duan, J. Gnanaraj, X. Chen, B. Li, J. Liang, *J. Power Sources* 2008, 185, 512. e) F.S. Ke, L. Huang, B. Zhang, G.Z. Wei, L.J. Xue, J.T. Li, S.G. Sun, *Electrochim. Acta* 2012, 78, 585. f) S.K. Behera, *J. Power Sources* 2011, 196, 8669. g) M. Sasidharan, N. Gunawardhana, M. Yoshio, K. Nakashima, *Ionics*2013, 19, 25. h) H.-S. Lim, B.-Y. Jung, Y.-K. Sun, K.-D. Suh, *Electrochem. Acta* 2012, 75, 123. i) D.W. Su, H.-J. Ahn, G.X. Wang, *J. Power Sources* 2013, 244, 742. j) 6J.S. Xu, Y.-J. Zhu, F. Chen, *J. Solid State Chemistry* 2013, 199, 204.
- a) W.M. Zhang, X.L. Wu, J.S. Hu, Y.G. Guo, L.J. Wan, Adv. Funct. Mater. 2008, 18, 3941. b) C.M. Ban, Z.C. Wu, D.T. Gillaspie, L. Chen, Y.F. Yan, J.L. Blackburn, A.C. Dillon, Adv. Mater. 2010, 22, E145. c) W.F. Chen, S.R. Li, C.H. Chen, L.F. Yan, Adv. Mater. 2011, 23, 5679. d) E. Kang, Y.S. Jung, A.S. Cavanagh, G.H. Kim, S.M. George, A.C. Dillon, J.K. Kim, J. Lee, Adv. Funct. Mater. 2011, 21, 2430. e) T. Zhu, J.S. Chen, (David) X.W. Lou, J Phys. Chem. C 2011, 115, 9814.
- 8 a) W. Wei, S.B. Yang, H.X. Zhou, I. Lieberwirth, X.L. Feng, K.
 Müllen, Adv. Mater. 2013, 25, 2909. b) W.-L. Chiang, C.-J. Ke, Z.-X. Liao, S.-Y. Chen, F.-R. Chen, C.-Y. Tsai, Y.N. Xia, H.-W. Sung, Small 2012, 8, 3584. c) J. Liu, Z.K. Sun, Y.H. Deng, Y. Zou, C.Y. Li, X.H. Guo, L.Q. Xiong, Y. Gao, F.Y. Li, D.Y. Zhao, Angew. Chem. 2009, 121, 5989.
- a) B.J. Li, H.Q. Cao, J. Shao, M.Z. Qu, *Chem. Commun.* 2011, 47, 10374. b) M. Zhang, M.Q. Jia, *J. Alloys and Compounds* 2013, 551, 53. c) Y. Chen, B.H. Song, X.S. Tang, L. Lu, J.M. Xue, *J. Mater. Chem.* 2012, 22, 17656.
- 10 F. Zhang, T.F. Zhang, X. Yang, L. Zhang, K. Leng, Y. Huang, Y.S. Chen, *Energy. Environ. Sci.* **2013**, 6, 1623.
 - 11 Y.C. Dong, R.G. Ma, M.J. Hu, H. Cheng, C.K. Tsang, Q.D. Yang, Y.Y. Li, J.A. Zapien, *J. Solid State Chemistry* **2013**, 201, 330.
 - 12 a) D.W. Su, H.S. Kim, W.S. Kim, G.X. Wang, *Microporous and Mesoporous Materials* **2012**, 149, 152. b) J. Chen, L. Xu, W. Li, X. Gou, *Adv. Mater.* **2005**, 17, 582.

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13 a) Z.C. Yang, J.G. Shen, L.A. Archer, J. Mater. Chem. 2011, 21, 11092. b) Y.Z. Piao, H.S. Kim, Y.E. Sung, T. Hyeon, Chem. Commun. 2010, 46, 118. c) S. Grugeon, S. Laruelle, L. Dupont, J.M. Tarascon, Solid Stae Sci. 2003, 5, 895. 14 S. Laruelle, S. Grugeon, P. Poizot, M. Dolle, L. Dupont, J.M. Tarascon, J. Electrochem. Soc. 2002, 149, A627.