

ChemComm

Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this *Accepted Manuscript* with the edited and formatted *Advance Article* as soon as it is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

COMMUNICATION

An acid-free rechargeable battery based on PbSO₄ and spinel LiMn₂O₄

Cite this: DOI: 10.1039/x0xx00000x

Yu Liu,^{abc} Zubiao Wen,^b Xiongwei Wu,^a Xiaowei Wang,^d Yuping Wu^{a,d*} and Rudolf Holze^{c,*}

Received 00th January 2012,

Accepted 00th January 2012

DOI: 10.1039/x0xx00000x

www.rsc.org/

An acid-free lead rechargeable battery system comprising PbSO₄ as the negative electrode, LiMn₂O₄ as the positive electrode and a neutral Li₂SO₄ aqueous solution as electrolyte is reported and its electrochemical performance is very good. It provides a great promise to reduce the global use of lead by 50% since its cost is also low.

Batteries are currently being further developed to power an increasingly diverse range of applications, from portable electronics, electric vehicles to smart grids.¹ The lead-acid battery is commercially one of the most successful electrochemical systems, and no other battery is yet able to replace it in the field of energy storage. Its market share is still increasing in the last years, albeit batteries based on other chemistries are rapidly catching up.^{2,3} The main reasons for this are its specific advantages including low cost, high reliability, and easy construction. In addition, its raw materials are practically unlimited, about 95 % can be recycled. However, sulphuric acid, which is used as the electrolyte for lead acid battery, causes serious damage to equipment, and Pb leads to serious environmental pollution. As a result, it is urgently needed to reduce the used amount of lead and acid.

Recently, most research in this area has focused on electrodes and electrolytes of batteries.^{4,5} For example a capacitor-battery system based on a PbO₂ positive electrode and activated carbon negative electrode was reported.⁵ However, it still uses acid and the amount of Pb can only be reduced by 25 wt.% at most. In addition, its energy density is below 50 % of that of lead acid battery, lower than 25 Wh kg⁻¹ although its rate capability and cycling life is much improved. In order to increase the specific capacity of lead batteries and reduce the amount of Pb and acid, here we reported an acid-free lead battery based on PbSO₄ as a negative electrode and spinel LiMn₂O₄ as a positive electrode in 0.5 M Li₂SO₄ neutral aqueous electrolyte. Its cycling performance is excellent with 100 % Coulombic efficiency and its energy density can be 68 Wh kg⁻¹.

The preparation of the spinel LiMn₂O₄ is as follows. At first, a nanowire bimesite-MnO₂ (δ-MnO₂) was synthesized by a hydrothermal reaction from a solution of MnSO₄, (NH₄)₂S₂O₈ and (NH₄)₂SO₄ in a molar ratio of 1: 1: 4 at 140 °C as we described previously.^{3a} The prepared δ-MnO₂ nanowires were milled with LiOH·H₂O in ethanol; then the mixture was sonicated for 5 h. After the ethanol was evaporated, the mixture was transferred into a furnace and heat-treated at 700 °C for 8 h to get LiMn₂O₄ nanocubes.

The spinel LiMn₂O₄ nanocubes were analysed by X-ray diffraction on a Bruker D4 X-ray diffractometer with Ni-filtered CuK_α radiation. They were mixed with acetylene black and poly(tetrafluoroethylene) (PTFE) in a weight ratio of 7.5:1.5:1 dispersed in ethanol. After drying, the mixture was pressed into a film with an active mass loading of about 3.75 mg/cm², the film was cut into a

disk of about 2 mg which was pressed onto Ni-grid at a pressure of 10 MPa, and finally dried at 120 °C for 12 h to act as the positive electrode. The cyclic voltammetric (CV) testing of LiMn₂O₄ electrodes was performed in 0.5 M Li₂SO₄ aqueous solution in a three-electrode cell with a nickel grid and a saturated calomel electrode (SCE) as counter and reference electrodes, respectively. The CV data were collected on an electrochemical working station of CHI660C (Chenhua, China).

PbSO₄ powder (>99.9 wt%) was purchased from Second Shanghai Smelting Plant (China) and used as received. The PbSO₄ electrode was prepared in the same way as the LiMn₂O₄ electrode. A two-electrode cell consisting of the above LiMn₂O₄ positive electrode and the PbSO₄ negative electrode with a distance of about 1 cm was used to test the charge-discharge and cycling behavior of the as-prepared LiMn₂O₄ in 0.5 M Li₂SO₄ solution on a Land 2001A cell tester.

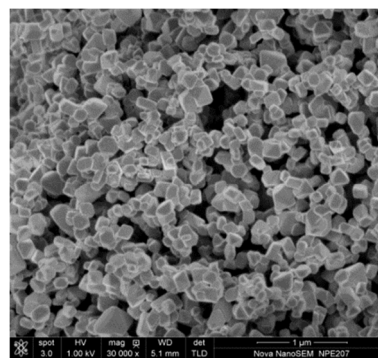


Fig. 1 Field emission scanning electron micrograph (FESEM) of the prepared LiMn₂O₄ nanocubes.

The X-ray diffraction pattern (Fig. S1, ESI†) shows that the as-prepared LiMn₂O₄ is highly crystalline with spinel structure; the characteristic peaks of planes can be clearly identified. This high degree of ordering is caused by the high temperature treatment (700 °C) during the synthesis process. FESEM micrograph of the prepared LiMn₂O₄ (Fig. 1) shows clearly that this material exists in nanocubes. This is different from the prepared Na_xMnO₂ nanorods from the heat-treatment of Na₂CO₃ with the MnO₂ nanowires.^{3a} The

main reason are presumably differences in the crystallization processes.

Cyclic voltammograms (CVs) of PbSO_4 and LiMn_2O_4 electrodes in 0.5 M Li_2SO_4 aqueous electrolyte at the scan rate of 5 mV s^{-1} are shown in Fig. 2. In the case of PbSO_4 , there is one set of redox peaks located at $-0.6/-0.36 \text{ V}$ (vs. SCE), which is due to the redox reaction of PbSO_4/Pb . This implies that PbSO_4 is stable in neutral aqueous solution and its redox reactions can reversibly take place. Two sets of redox peaks related to LiMn_2O_4 are situated at $0.62/0.86 \text{ V}$ and $0.78/1.0 \text{ V}$ (vs. SCE), respectively, which correspond to the intercalation/de-intercalation of Li^+ ions into/from LiMn_2O_4 in the aqueous electrolyte solution. This is similar to our former reports.⁶ Since there is a potential difference between PbSO_4 and LiMn_2O_4 , they can be assembled into a battery system whose redox reactions during charge/discharge processes can be schematically shown in Scheme 1.

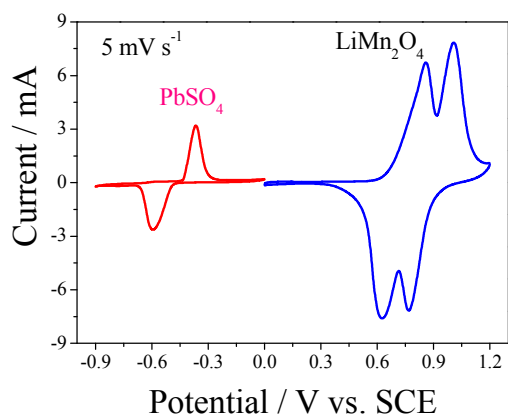
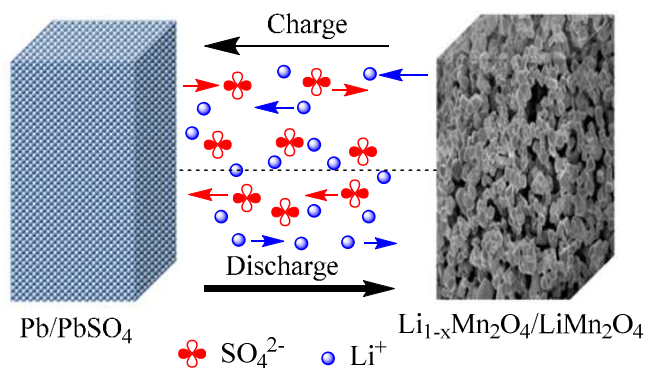


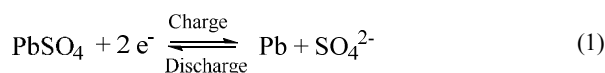
Fig. 2 Cyclic voltammograms (CVs) of PbSO_4 and LiMn_2O_4 electrodes in 0.5 M Li_2SO_4 aqueous electrolyte at the scan rate of 5 mV s^{-1} , which was tested by using nickel-grid and SCE as the counter and reference electrodes, respectively, and each electrode is about 2 mg.



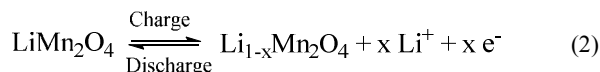
Scheme 1 Schematic illustration of the redox reactions for acid-free $\text{PbSO}_4//\text{LiMn}_2\text{O}_4$ aqueous rechargeable battery during the charge/discharge processes.

Their electrode and cell reactions are shown simplified as the following:

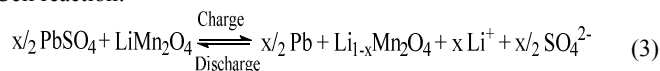
Negative electrode:



Positive electrode:



Cell reaction:



During charging PbSO_4 is turned into Pb by releasing sulphate anions into the neutral aqueous electrolyte. In the meanwhile, LiMn_2O_4 is turned into $\text{Li}_{1-x}\text{Mn}_2\text{O}_4$ by releasing Li^+ cations into the electrolyte. As a result, the released sulphate anions and lithium cations keep the electrolyte always electrically neutral. During the discharge process, Pb gets SO_4^{2-} anion from the neutral electrolyte to turn into PbSO_4 together with the gain of Li^+ ion from the electrolyte into $\text{Li}_{1-x}\text{Mn}_2\text{O}_4$ to become LiMn_2O_4 . This means that this battery does not need acid solution since the positive electrode is LiMn_2O_4 instead of PbO_2 . The latter needs an acid media to realize its fast redox reaction to PbSO_4 .

According to the definition of aqueous rechargeable lithium batteries (ARLBs),⁷ it can be assigned as a 1st generation ARLBs type. During the charge or discharge process, both electrodes exist as solid phases, and the total volume change is less than those for $\text{Zn}/\text{LiMn}_2\text{O}_4$, $\text{Zn}/\text{LiFePO}_4$ or $\text{Zn}/\text{Na}_x\text{MnO}_2$ ³ since Pb or PbSO_4 could not dissolve into the neutral solution as Zn .

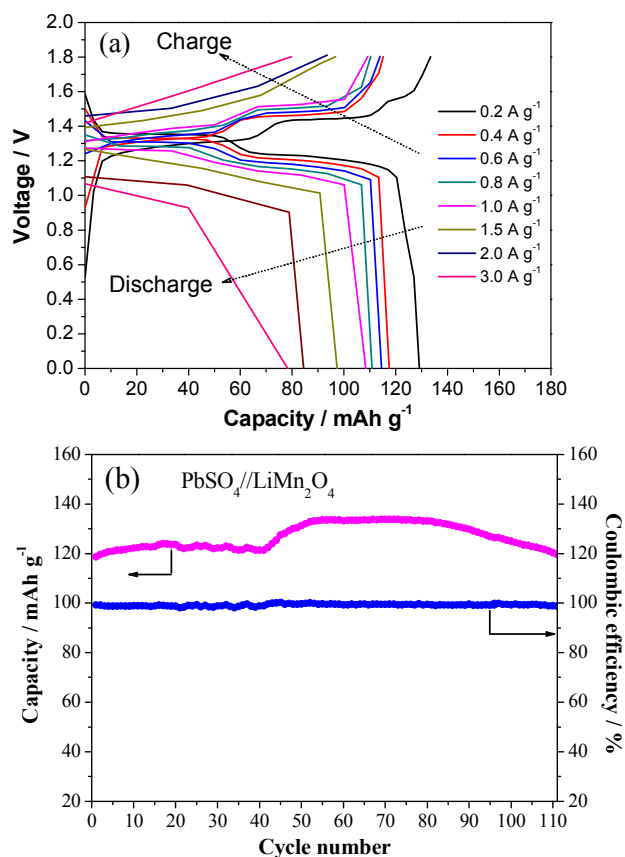


Fig. 3 (a) Charge and discharge curves of the $\text{PbSO}_4/\text{Li}_2\text{SO}_4/\text{LiMn}_2\text{O}_4$ battery at different current densities based on the spinel positive electrode, and (b) cycling performance of the battery at 400 mA g^{-1} based on LiMn_2O_4 .

The electrochemical performance of the $\text{PbSO}_4//\text{LiMn}_2\text{O}_4$ battery at different current densities between 0–1.8 V is shown in Fig. 3. At the lower current densities two discharge/charge voltage platforms at about 1.27/1.35 V and 1.44/1.23 V, respectively, can be identified which are consistent with the CV results. This is due to

the intercalation/deintercalation of Li^+ ions into/from different sites of the spinel host. Its average discharge voltage is 1.30 V. The reversible capacity of the battery is 128 mAh g^{-1} at 200 mA g^{-1} based on the LiMn_2O_4 positive electrode, which is similar to the reported value with aqueous electrolytes.⁶ When the charge/discharge current increases from 0.4, 0.6, 0.8, 1.0, 1.5, 2.0 to 3.0 A g^{-1} , the discharge capacity decreases from 117.5, 114.6, 110.9, 108.8, 97.5, 94.6, to 78.4 mAh g^{-1} , respectively. This suggests that the acid-free lead battery can be charged and discharged fast, and can meet the rapid change of power supply and demand from the grid (Fig. S2, ESI†). However, its rate capability cannot be compared with those of $\text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_4$ and $\text{Li}[\text{Ni}_{1/3}\text{Co}_{1/3}\text{Mn}_{1/3}]\text{O}_2$.⁸ The main reason is the slow redox kinetics of Pd/PbSO_4 , this is also a key problem for the lead acid rechargeable batteries.⁹

In the case of the lead acid battery, about half of the theoretical capacity of the negative electrode can be obtained practically. Applying this to the present battery system the practical capacity of PbSO_4 would be about 88 mAh g^{-1} . Combining the practical capacity of LiMn_2O_4 (128 mAh g^{-1}) and the average discharge voltage (1.3 V), the calculated energy density based on the two electrodes will be 68 Wh kg^{-1} . This value is comparable with that of lead acid batteries.

The cycling performance of the $\text{PbSO}_4/\text{LiMn}_2\text{O}_4$ at the full discharge capacity of 128 mAh g^{-1} based on LiMn_2O_4 at the current density of 400 mA g^{-1} (Fig. 3b) shows that the Coulombic efficiency is almost 100 % except in the initial cycles, this again is similar to lithium ion batteries. This means that the acid-free lead battery shows good charge and discharge reversibility. In terms of the spinel LiMn_2O_4 , it is very stable, its redox potential is below that of oxygen evolution and it can be cycled up to 10 000 full cycles.⁶ For the redox couple Pb/PbSO_4 , Fig. 2 shows clearly that its redox potential is above that for hydrogen evolution due to the large hydrogen overpotential of Pb. In the case of LiMn_2O_4 , its redox potentials are also below that of oxygen evolution, which has been well illustrated in ARLBs.^{1d, 6a, 6b, 7} As a result, there are basically no side reactions (gassing) resulting in 100 % Coulombic efficiency which is evidently higher than that of lead acid battery. Another reason is that the charge cut-off voltage is 1.8 V, much lower than 2.4 V for lead acid battery. If it is used for energy storage, this means that it is superior to lead acid battery since its self-discharge rate will also be lower due to lower working voltage and less irreversible reactions. After 110 full cycles, the capacity loss based on LiMn_2O_4 is less than 1 % compared with the initial cycles. In the case of the Pb/PbSO_4 electrode, it is known that there is no phase change and dendrite problem like Zn, it will be stable during cycling. As a result, the cycling behavior of this battery system will be superior to the reported $\text{Zn}/\text{LiMn}_2\text{O}_4$ and $\text{Zn}/\text{Na}_x\text{MnO}_2$,³ which can be above 1000 cycles. Of course, further research on the extended cycling is needed.

In conclusion, an acid-free lead battery has been assembled using PbSO_4 as the negative electrode, spinel LiMn_2O_4 nanocubes as the positive electrode, and an aqueous solution of 0.5 M Li_2SO_4 as the electrolyte. Its average discharge voltage is 1.3 V and the specific capacity based on the spinel is stable at 128 mAh g^{-1} . The calculated energy density based on the practical capacities of the two electrodes is 68 Wh kg^{-1} . The battery system shows less than 1 % capacity loss after 110 full cycles at the current density of 400 mA g^{-1} between 0-1.8 V and 100% Coulombic efficiency except the initial several cycles. The above results show clearly that the positive electrode of lead acid battery (PbO_2) can be completely substituted by the environmentally friendly and cheap LiMn_2O_4 , which means that 50 % of Pb can be saved. In addition, sulphuric acid is not needed. It will be more friendly to the operators and

during use. Our battery system can be a good candidate as a power source for electric bicycles or energy storage on a large scale.

Financial support from Natural Foundation for Distinguished Young Scholars of China (NSFC) is greatly appreciated.

Notes and references

^a College of Science, Hunan Agriculture University, Changsha, Hunan 410128, China

^b College of Chemistry and Chemical Engineering, Jiangxi Normal University, Nanchang 330022, China

^c Technische Universität Chemnitz, Institut für Chemie, AG Elektrochemie, D-09107 Chemnitz, Germany. E-mail: rudolf.holze@chemie.tu-chemnitz.de

^d New Energy and Materials Laboratory (NEML), Department of Chemistry & Shanghai Key Laboratory of Molecular Catalysis and Innovative Materials, Fudan University, Shanghai 200433, China. E-mail: wuyup@fudan.edu.cn; Fax: +86-21-55664223

Electronic Supplementary Information (ESI) available: X-ray diffraction pattern of the as-prepared LiMn_2O_4 and change of capacity of the battery at different current densities. See DOI: 10.1039/c000000x/

- (a) J. Gao, M.A. Lowe, Y. Kiya, H.D. Abruna, *J. Phys. Chem. C* 2011, **115**, 25132; (b) Z.Q. Peng, S.A. Freunberger, Y.H. Chen, P.G. Bruce, *Science*, 2012, **337**, 563; (c) Y. Yang, G.Y. Zheng, Y. Cui, *Energy Environ. Sci.* 2013, **6**, 1552; (d) X.J. Wang, Y.Y. Hou, Y.S. Zhu, Y.P. Wu, R. Holze, *Sci. Rep.*, 2013, **3**, 1401; (e) X.J. Wang, Q.T. Qu, Y.Y. Hou, F.X. Wang, Y.P. Wu, *Chem. Commun.*, 2013, **49**, 6179; (f) X.W. Wu, X.J. Xiang, J. Liu, J.P. Hu, Y.P. Wu, *Pure Appl. Chem.* 2014, **86**, 661.
- (a) M. Armand and J. M. Tarascon, *Nature*, 2008, 451, 652; (b) B. Hong, L. X. Jiang, K. T. Hao, F. Y. Liu, X. Y. Yu, H. T. Xue, J. Li and Y. X. Liu, *J. Power Sources*, 2014, **256**, 294; (c) J. Garche, *Phys. Chem. Chem. Phys.*, 2001, **3**, 356; (d) M. Saravanan, M. Ganesan and S. Ambalavanan, *J. Power Sources*, 2014, **251**, 20.
- (a) B. H. Zhang, Y. Liu, X. W. Wu, Y. Q. Yang, Z. Chang, Z. B. Wen and Y. P. Wu, *Chem. Commun.*, 2014, **50**, 1209; (b) H.P. Zhang, X. Wu, T. Yang, S.S. Liang, X.J. Yang, *Chem. Commun.* 2013, **49**, 9977; (c) J. Yan, J. Wang, H. Liu, Z. Bakenov, D. Gosselink, P. Chen, *J. Power Sources*, 2012, **216**, 222.
- K. Maeda, N. Arakawa, K. Fukui and H. Kuramochi, *Phys. Chem. Chem. Phys.*, 2014, **16**, 4911.
- A. Banerjee, M.K. Ravikumar, A. Jalajakshi, S.A. Gaffoor, A.K. Shukla, in *Lead-Acid Batteries And Capacitors, New Designs, And New Applications*, Eds. K. Bullock, P.T. Moseley, B.Y. Liaw, *ECSTransactions*, 2012, 41(13), pp. 101-113.
- (a) Q. T. Qu, L. J. Fu, X. Y. Zhan, D. Samuelis, J. Maier, L. Li, S. Tian, Z. H. Li and Y. P. Wu, *Energy Environ. Sci.*, 2011, **4**, 3985; (b) W. Tang, Y.Y. Hou, F.X. Wang, L.L. Liu, Y.P. Wu, K. Zhu, *Nano Lett.*, 2013, **13**, 2036; (c) W. Tang, L.L. Liu, S. Tian, L. Li, L.L. Li, Y.B. Yue, Y. Bai, Y.P. Wu, K. Zhu, *Electrochem. Commun.*, 2011, **13**, 1159; (d) P. R. Gao, W. X. Lv, R. Zhang, Y. Liu, G. H. Li, X. F. Bu and L. X. Lei, *J. Power Sources*, 2014, **248**, 363.
- (a) W. Tang, Y.S. Zhu, Y.Y. Hou, L.L. Liu, Y.P. Wu, K.P. Loh, H.P. Zhang and K. Zhu, *Energy Environ. Sci.*, 2013, **6**, 2093; (b) Z. Chang, Y.Q. Yang, M.X. Li, X.W. Wang, Y.P. Wu, *J. Mater. Chem. A*, 2014, **2**, 10739.
- (a) L. Zhou, D. Y. Zhao and X. W. Lou, *Angew. Chem. Int. Ed.*, 2012, **51**, 239; (b) F. X. Wang, S. Y. Xiao, Z. Chang, Y. Q. Yang and Y. P. Wu, *Chem. Commun.*, 2013, **49**, 9209.
- M.P. Vinod, K.; Vijayamohan, *J. Power Sources*, 1994, **50**, 67.

Table of Content

An acid-free rechargeable battery based on PbSO₄ and spinel LiMn₂O₄

Yu Liu, Zubiao Wen, Xiongwei Wu, Xiaowei Wang, Yuping Wu and Rudolf Holze

A low cost and better friendly rechargeable battery with good electrochemical performance, which is of great promise to reduce the use amount of Pb by 50%.

