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## **ARTICLE TYPE**

### Herringbone to Cofacial Solid State Packing via H-bonding in Diketopyrrolopyrrole (DPP) based Molecular Crystals: Influence on Charge Transport

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The mono alkylation of DPP derivatives leads to cofacial  $\pi$ - $\pi$  stacking via H-bonding unlike their di-alkylated counterpart, which exhibits classical herringbone packing pattern. Single <sup>10</sup> crystal organic field-effect transistor (OFET) measurements reveal significant enhancement of charge carrier mobility for mono-hexyl DPP.

The performance of optoelectronic devices based on organic semiconductors is limited by poor charge transport. The charge 15 carrier mobility ( $\mu$ ) in molecular semiconductors is strongly influenced by a variety of factors such as purity, traps, defects and molecular stability.<sup>1</sup> To surmount these limitations, molecular engineering has taken a centre stage.<sup>2-4</sup> Large numbers

- of new organic functional materials have been synthesized to <sup>20</sup> achieve the best performance for devices such as organic photovoltaics (OPVs) and organic field effect transistors (OFETs).<sup>5-7</sup> In this regard, diketopyrrolopyrrole (DPP) based  $\pi$ conjugated systems have shown promise as a versatile material with excellent charge transport properties.<sup>8, 9</sup> Yun-Hi Kim and co-
- <sup>25</sup> workers have reported one of the highest p-type charge carrier mobilities for DPP-based polymer, surpassing the mobility of amorphous silicon.<sup>10</sup> Our group has also observed one of the best electron mobility for DPP based  $\pi$ -conjugated low-band gap polymer.<sup>11</sup> Inspired by these results, we envisioned to investigate
- <sup>30</sup> the structure-property correlation of DPP monomers to explore the full potential as a molecular material for organic electronics.

The DPP-based monomers were synthesized by pseudo-Stobbe condensation reaction followed by di-alkylation of the amide groups present in the DPP chromophore.<sup>12</sup> While following this <sup>35</sup> procedure a minor product was also separated along with the expected di-alkylated DPP derivatives. The highly crystalline minor product was identified as mono-alkylated DPP by <sup>1</sup>H, <sup>13</sup>C and ESI mass spectroscopy (Electronic Supplementary Information, ESI<sup>†</sup>). To our surprise, single crystal structure <sup>40</sup> determination reveals classical herringbone<sup>12</sup> and cofacial layered structure for di-and mono-hexyl DPP, respectively. This encouraged us to enhance the yield of mono-hexyl DPP by controlling the reaction condition, shown in Scheme 1 and pursue

further with an aim to study the comparative charge transport 45 properties of this important class of material which was ignored previously. Herein, we report the synthesis and characterization of mono-alkylated DPP for the first-time. Detailed studies have unrevealed several unusual features including the role of Hbonding to influence the favourable packing for enhanced charge <sup>50</sup> transport in single crystals of mono-hexyl DPP than its dialkylated counterpart.



Scheme. 1 In top, synthesis of PDPP-MH and TDPP-MH starting from their corresponding DPP molecules has been shown. The chemical structures of four different DPP molecules are shown at the bottom

The optical properties of mono-hexyl DPP molecules were almost similar to their di-alkylated analogues with the appearance of the dual band structure in absorption and emission spectra (Figure 1 (a) and (b)). The low energy band in the absorption spectrum 60 showed vibronic splitting for both mono-hexyl derivatives. But there was minor blue shift (10 nm) of the lowest energy band for TDPP-MH ( $\lambda_{max}$  at 500, 540 nm) as compared to TDPP-DH ( $\lambda_{max}$ at 510, 550 nm) indicating increase in band gap while optical band gap remained almost same for both the phenyl coupled DPP 65 (PDPP) derivatives. Both PDPP-MH and TDPP-MH showed blue-shifted vibronic emissive band in comparison with the dihexyl DPP molecules. The hypsochromic shift was measured to be ~10 nm for TDPP-MH while it was 25 nm for PDPP-MH. As a result, we observe a significant decrease in Stokes shift for 70 PDPP-MH (0.19 eV) as compared to PDPP-DH (0.32 eV), but for TDPP-MH and TDPP-DH the Stokes shift was almost similar (0.4-0.5 eV). The reduction in Stokes shift suggests a decrease in torsion in PDPP-MH. Further, density functional theory (DFT) calculations have corroborated our assumption. The detailed optical properties of mono-hexyl derivatives are given in Table S1 in the ESI<sup>†</sup>. Theoretical studies have shown that in the absence of alkyl chain the repulsive interaction between *ortho*-<sup>5</sup> hydrogen of phenyl group and a nearby hydrogen atom attached to the alkyl group vanishes and the phenyl group appears almost in the same plane with DPP core with torsion of 8° enhancing electronic coupling between donor and acceptor moieties (Figure

1d). On the other hand, the other phenyl ring remains as twisted 10 as in PDPP-DH by 36° due to the above said repulsive interaction. Calculations on TDPP-MH predicts the transorientation of two thiophene units with respect to each other and both the donor groups lie in the same plane with the acceptor having a minimum torsion of less than 1°.



Fig. 1 The normalized absorption (a) and emission (b) spectra and fluorescence excited state decay profile (c) of mono-and di-hexyl DPP derivatives. (d) Energy minimized ground state geometry of PDPP-MH and TDPP-MH obtained by density functional theory (DFT) calculations. 20 The torsion between donor and acceptor units has also been mentioned

The consequence of higher planarity in TDPP-MH is the stronger intramolecular electronic coupling between donor and acceptor moieties resulting in higher molar extinction coefficient for TDPP-MH (20300 L mol<sup>-1</sup> cm<sup>-1</sup>) than the PDPP-MH (15000 L

- <sup>25</sup> mol<sup>-1</sup> cm<sup>-1</sup>) as measured experimentally. Fluorescence quantum yield decreases from 80% in PDPP-MH to 72% in TDPP-MH which is attributed to the heavy atom effect facilitating the intersystem crossing rates.<sup>13</sup> On the similar line, fluorescence excited state lifetime also decreases from PDPP-MH to TDPP-MH as
- <sup>30</sup> shown in Figure 1C. The frontier molecular orbital energies were measured from cyclic voltammetry (CV) experiment. The HOMO and LUMO energies are summarized in Table S1 (Electronic Supplementary Information, ESI†). The HOMO energy level destabilized by 0.1 eV in TDPP-MH from PDPP-MH while
- $_{35}$  LUMO stabilized by  $\sim 0.2$  eV due to the change of phenyl with thiophene in the donor moiety.

The single crystal X-ray diffraction study reveals the molecular arrangements and their interactions in the solid state. Both the molecules crystallize in the monoclinic space group  $P2_1/n$  z=4.

<sup>40</sup> The unique difference between di- and mono- hexyl DPP derivatives lies in their crystal packing arrangements. The free amide group of mono-alkylated DPP molecules plays a pivotal role to turn around the molecular packing in the unit cell. PDPP-

MH exhibits cofacial layered structure while TDPP-MH 45 demonstrates cofacial herringbone packing due to the intermolecular H-bonding between two amide functionalities in DPP chromophore locking the orientation of the molecules. Thus, intermolecular H-bonding influences the packing pattern of DPP-MH deviating from their di-hexyl counterparts which show 50 exclusively classical edge-to-face herringbone packing. Both, PDPP-MH and TDPP-MH showed slip-stacked packing arrangement with donor thiophene/phenyl group appears top of the acceptor DPP chromophore, which is a common feature of D-A-D molecular crystals presumably due to the charge transfer 55 interaction. The shortest  $\pi$ - $\pi$  packing distances were measured to be 3.48 Å and 3.71 Å for PDPP-MH and TDPP-MH, respectively. Similar to our theoretical findings, we see the variation in degree of torsion between the acceptor moiety and two donor groups in both the molecules due to relief from steric 60 repulsion in the absence of an alkyl chain. Torsion varies from 2° to as high as 24° in PDPP-MH while they were 1° and 10° in TDPP-MH. The N-H"O H-bond distances were calculated as 2.04 Å and 1.94 Å in PDPP-MH and TDPP-MH, respectively. Weak C-H"O interaction was also found to operate in both the 65 crystals with the distances of 2.43 Å and 2.27 Å in PDPP-MH and TDPP-MH.



Fig. 2 Cofacial packing via H-bonding in PDP-MH and TDPP-MH



70 Fig. 3 Bright field TEM images (a, c) and corresponding SAED pattern (b, d) of the single crystalline microwire from PDPP-MH (a, b) and TDPP-MH (c, d)

The transmission electron microscopy (TEM) study shows that the nano-to micron sized structures deposited on copper grid are

60

single crystalline in nature (Figure 3). The mono-hexyl DPP derivatives form fiber like single crystal while di-hexyl analogues form small plate like crystallites. We believe that the presence of H-bonding interaction is responsible for variation in <sup>5</sup> morphologies for mono-hexyl DPP derivatives. The ordered bright diffraction spots in selected area electron diffraction (SAED) pattern confirms the single crystallinity of microstructures. The bright field TEM images and corresponding SAED patterns from single crystal di-hexyl derivatives have been <sup>10</sup> provided in the ESI<sup>†</sup>.

The H-bonding interaction was further supported by differential scanning calorimetry (DSC). For TDPP-MH, we have observed the appearance of two endo-and exothermic transitions during heating and cooling cycles, respectively (Figure 4). The first 15 endothermic peak was positioned at 207 °C with an enthalpy

- change of ~2.3 kJ/mole which matches quite well with the Hbonding energy. This compound melted around 250 °C. The change in enthalpy during melting transition was measured to be 21.5 kJ/mole. But dual peaks were not observed for PDPP-MH.
- <sup>20</sup> In both heating and cooling cycles only one peak was observed, though the enthalpy change during melting process around 250 °C was higher i.e. ~26.8 kJ/mole. The phase transitions observed in DSC measurements were further confirmed by optical polarising microscopy. For TDPP-MH crystals, we have observed
- <sup>25</sup> a thermal transition around 208 °C associated with the change in crystal texture followed by an isotropic phase transition at 255 °C. But PDPP-MH crystals turn into an isotropic phase at 247 °C with a single thermal transition.



Fig. 4 DSC thermograms of PDPP-MH and TDPP-MH

30

OFET devices were fabricated using Si/SiO<sub>2</sub> as a gate and dielectric material in bottom-gate bottom-contact (BGBC) device geometry. Photo-lithographically patterned gold (Au) source and drain contacts were developed on hexamethyldisilazane (HMDS) <sup>35</sup> treated 190 nm thick dielectric layer of SiO<sub>2</sub>. The solutions of

respective organic materials were drop-casted on the substrate and crystals were grown by the slow evaporation method in the channel. The mono-hexyl derivatives form thin and sharp needle like crystal whereas their di-hexyl analogues form bigger 40 crystallites in between the source and drain electrodes. The measurements were carried out at ambient condition at room temperature. The device data have been summarized in Table 1.

Significant differences in charge transport characteristics were observed between mono-and di-hexyl DPP derivatives, which are

<sup>45</sup> related to the intrinsic nature of the molecular semiconductors. We have observed higher *p*-type mobilities for mono-hexyl DPPs

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as compared to their di-hexyl counterparts. The average *p*-type mobilities for PDPP-MH and TDPP-MH was recorded to be 1.6 x  $10^{-2}$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> and 0.3 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, respectively. Whereas the <sup>50</sup> average mobility values were two orders less for their di-hexyl analogues. The best mobilities obtained from PDPP-DH and TDPP-MH. The highest mobilities obtained from PDPP-DH and TDPP-DH were 3 x  $10^{-3}$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> and 5 x  $10^{-2}$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, respectively. On/off ratios of these devices were ~  $10^2$ - $10^3$ . The <sup>55</sup> current shows saturation behaviour with increase in source-drain voltage for almost all the devices prepared from DPP-DH. This may be related to the crystal quality or contact resistance at the semiconductor-electrode interface.



Fig. 5 Output characteristics (a, b) and optical images of the corresponding devices (c, d) fabricated from single crystalline PDPP-MH (a, c) and TDPP-MH (b, d) microwires

Table 1 Summary of the OFET de	evice parameters
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Compounds	Structure	p-type mobility ( $\mu_{average}$ ) ( $cm^2 V^{-1} s^{-1}$ )	On/Off
PDPP-MH	Cofacial Layered	1.6 x 10 <sup>-2</sup>	10 <sup>2</sup>
TDPP-MH	Cofacial Herringbone	0.3	10 <sup>2</sup>
PDPP-DH	Classical Herringbone	2 x 10 <sup>-4</sup>	10 <sup>2</sup>
TDPP-DH	Classical Herringbone	4 x 10 <sup>-3</sup>	10 <sup>3</sup>

65 Devices were fabricated in bottom-gate bottom-contact geometry

The cofacial  $\pi$ -stacking improves the transfer integral as compared to edge-to-face herringbone packing according to Marcus theory, hence predicting better carrier transport through the  $\pi$ -stacked molecules if other factors remain the same.<sup>14, 15</sup> <sup>70</sup> Zhenan Bao and co-workers have also reported the same for tetracene derivatives.<sup>16</sup>

To correlate the experimental observation theoretically, the transfer integral (*t*) has been calculated for all the four crystals in the three crystallographic directions (Figure S15, S16, ESI $\dagger$ ).<sup>1</sup>

105

Employing wB97xD function and 6-31g\* basis set, the single point energy of dimeric structure (obtained from crystal structure) was calculated. The half of the energy difference between HOMO and HOMO-1 gives an estimate of transfer integral for hole

- <sup>5</sup> transport while for electron transport, it was the energy difference between LUMO and LUMO+1. PDPP-DH has the t in zdirection, which corresponds to a herringbone arrangement. The tvalues (0.095 eV and 0.10 eV) predict equally good hole and electron mobilities. Along the y-direction smaller t (0.02 eV and
- <sup>10</sup> 0.015 eV) indicates reduced transport. Charge transfer integrals are zero in x-direction. Removing one hexyl group (PDPP-MH), results in the cofacial crystal structure with hydrogen bonding in y-direction. However, very little (0.005 eV and 0.005 eV) *t* value was observed in this direction. Charge transfer integrals are about
- <sup>15</sup> equally large in the other two directions, (0.05 eV and 0.07 eV (x-direction)) and 0.065 and 0.065 eV (z-direction) for hole and electron transport, respectively. But, for both mono-and di-hexyl TDPP, we observe that all the directions can take part in charge transport with relatively higher t value for DPP-MH system. The
- <sup>20</sup> TDPP-DH system has the largest t (0.075 eV and 0.14 eV) along the y-direction predicting good electron mobility. Upon removing one hexyl group hydrogen bonding leads to rearrangement of the crystal structure. For TDPP-MH the best mobility is predicted in the z-direction, where t for electron and hole transport reaches
- 25 0.125 eV. The *t* values of all the four molecules in three crystallographic directions are summarized in Table S2 (ESI<sup>†</sup>). Since, all the FET measurements were carried out in ambient condition, experimentally no electron transport was observed.

The overall result indicates that DPP-DH systems conduct mainly

- $_{30}$  in one direction with high t value, while in DPP-MH systems charge transport occurs in two or three directions with the considerable t values. Therefore, the total charge transport predicted for mono-substituted systems is increased as compared to di-substituted ones. The hydrogen bonding direction does,
- $_{35}$  however, contribute very little to the charge transport. Transport mainly occurs along  $\pi$ -stacks or along herringbone arrangements and appears to be influenced by the exact distance and orientation of the molecules.

#### Conclusions

- <sup>40</sup> In summary, two novel mono-hexyl DPP derivatives with phenyl and thiophene as donor groups have been synthesized and their properties are comprehensively characterized including single crystal X-ray analysis and charge transport measurement. The single crystal structure determination reveals the uniqueness of
- <sup>45</sup> mono-hexyl DPP from their di-hexyl analogues. While di-hexyl PDPP and TDPP exhibit herringbone packing arrangements, the mono-hexyl derivatives show cofacial layered structure. The deviation is due to the intermolecular H-bonding by the free amide group in mono-hexyl DPP molecules. Higher *p*-type
- <sup>50</sup> charge carrier mobilities were observed for PDPP-MH and TDPP-MH as compared to the di-hexyl DPP molecules. Theory agrees well with the experimental observation with respect to charge transport properties in this class of molecular semiconductors.

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- Experimental details, and additional spectra]. See 60 DOI: 10.1039/b000000x/
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