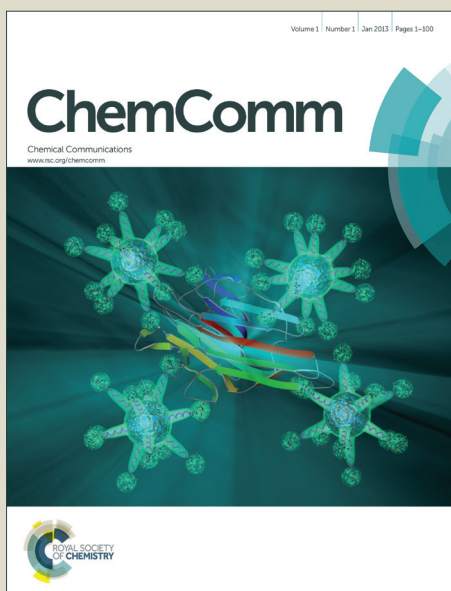


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Oxygen Etching of Thick MoS₂ Films

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Oxygen annealing of thick MoS₂ films results in randomly oriented and controllable triangular etched shapes, forming pits with uniform etching angles. These etching morphologies differ across the sample based on the defect sites situated on the basal plane surface, forming numerous features in different bulk sample thicknesses.

2D Molybdenite (MoS₂) is an atomically thin transition-metal dichalcogenide that has emerged as a new material with great promise for applications in optoelectronics, nanoelectronics and spintronics.^{1,2,3} MoS₂ has attracted a lot of attention with the analogous potential of graphene and its many applications.^{4,5,6,7} Naturally found MoS₂ is a layered semiconductor material with a bandgap that ranges between 1.2-1.8 eV, depending on the number of layers.^{1,2,8,9} Single layer MoS₂ is an excellent 2D material system for studying spintronics due to its broken inversion symmetry, which easily gives rise to coupling of spins and spin splitting in the valence band. MoS₂ has garnered significant attention due to the existence of a direct bandgap, which graphene's lack of has limited its applications.^{1,3,10,11,12} Studies have shown that as the thickness of MoS₂ sheets decreases from bulk to monolayer, the electronic bandgap is converted from indirect to direct bandgap; which can be observed by changes in the photoluminescence (PL) and Raman signals.^{13,14,15,16,9,17,18,19,20} Bulk MoS₂ is composed of van der Waals bonded S-Mo-S sheets. Each sheet is composed of a hexagonal plane of Mo atoms sandwiched in between two hexagonal planes of S atoms, which are held together by covalent bonds.^{10,17,1,21,22}

The ability to etch thick films of MoS₂ to few layers or a single layer allows for control over properties such as the band gap and optical properties. Understanding the etching process of materials is essential to achieving such control and may also aid in understanding their functionalities, optical properties, and electronic properties. Control of the etching process parameters can produce various ordered structures and fascinating displays. It is known that the etching pattern of MoS₂ follows 120° or 60° between etched lines.^{23,24} In this work we report the first observations of constant

angle etched CVD grown MoS₂, which reveals triangular, star, and hybrid etching patterns in a well-oriented manner by simple heating of the sample under oxygen-containing environment. We find that MoS₂ tends to etch triangular pits that follow the same etching angle and the same etching pattern.

MoS₂ films were grown on 300 nm thick SiO₂ by using an in-house thermal chemical vapor deposition (CVD) system. The samples were grown by ramping the system from room temperature to 500°C in 1 hour and held for an additional hour at a pressure of 1-2 torr, with a gas flow mixture of Ar/H₂ (4:1). This reaction converts the ammonium thiomolybdate precursor to MoS₂ in the presence of H₂.²⁵ Hydrogen in the first annealing step also reduces the amount of oxygen present in the tube. This process produced a MoS₂ film with a thickness of 10-20nm, and is standard for growing such a material by CVD.^{20,21,22,25} The MoS₂ sample was exposed to an Ar/O₂ mixture at a flow rate of 400 sccm, which initiates the etching process. After specified etching times, the furnace was cooled to room temperature under ambient conditions. The entire etching process is illustrated schematically in Figure 1. As the etching process begins, triangular and star shaped etched morphologies become apparent and grow in size as exposure time is increased (t₁<t₂<t₃), as shown in Figure 1. This oxidation etching effect can be prevented with the introduction of sulfur into the CVD chamber.

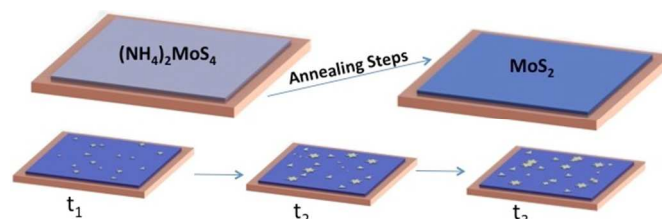


Figure 1. Schematic of the growth to etching process.

The etching morphology was characterized using a scanning electron microscope (SEM) (Leo-supra, 1550). Figure 2d shows that

the MoS₂ etching follows the same pattern of other published works on the growth of MoS₂, which follow triangular and star shaped morphologies.^{21,22,26,27} The etching initiation appears as a small indentation on the oxidized surface of the film as shown by SEM in Figure 2a. The initiation pits and triangular shape of the etched pits tend to come from extended lattice defects, such as edges or grain boundary mismatches between layers.^{12,28,29,30} As the etching process continues, the pits grow and take the form of triangles and other geometric forms, which contain pseudo-hexagonal shaped branched nerves. The tendency for these symmetric patterns to form may be due to etching occurring along the preferential crystalline direction, a conclusion derived from our observation of numerous straight edged triangular shaped features.

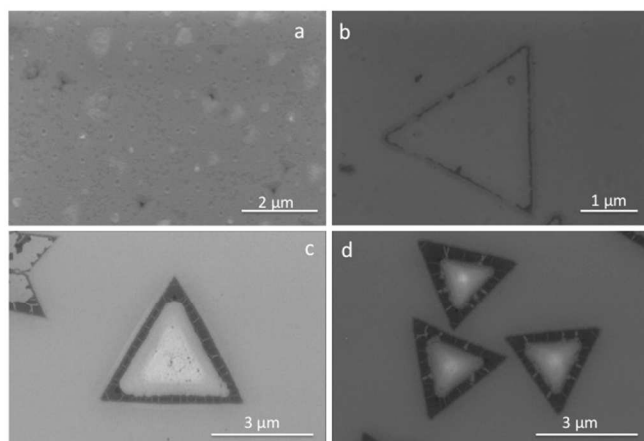


Figure 2. SEM images showing a time ordered evolution of etched patterns from a-d) under the oxygen flow.

As the etching initiates, as shown in Figure 2, we see that the pits in different regions look like straight edge triangles with a contour around it. This can be observed in Figures 3a-3d, where depending on the different defect sites, we obtain various geometric shapes with the triangle being the most commonly seen with an average domain size of 3 μm. As time progresses, we observe MoS₂ etching at an angle forming an almost perfect pyramid structure in the middle, as seen in Figures 2c and 2d. When enough time was allocated for the etching process, the pyramids disappeared, leaving behind a triangular pit. From Figure 3a,^{31,32} it is known that as the layers of MoS₂ increase, the interactions between the substrate and the material also increase, making the first MoS₂ layer more strongly bonded to the substrate.³³ This is a reasonable explanation for the presence of small triangles and the veins left on the bottom of the pits as seen in Figures 3e and 3f. Extensive studies on graphene etching have shown that etching occurs along the preferential crystalline direction but with a high uncertainty due to randomness.^{34,35,36,37,38} Etching veins start from the grain boundaries branching inward as seen in Figures 2c and 2d to a point of origin of the grown crystals as observed in Figures 3a-3c. Depending on the etching rate along the lattice, the etching occurs in the regions that are most energetically favorable, leaving behind a vein-like network. Due to the etching effect, these veins like structures could very well come from residual stress. These reasonings explain the ordered loops formed in Figure 2b and the hexagonally organized hexagonal features in Figure 3e. These two-dimensional quasicrystalline or ‘Penrose tilings’ with six-fold symmetry look ordered but non-periodic. Based on previous research done on etching, it has been concluded that by taking into account the Eley–Rideal and Langmuir–Hinshelwood mechanism, MoS₂ thickness can alter the etching rate and the way oxygen reacts with the material.^{39,40,41,23}

The thickness and grain boundaries in CVD-prepared MoS₂ allow for the observation of these etching shapes, previously not seen in exfoliated samples. As previously discussed, due to different edge directions based on the layer orientation we could have different rates of etching yielding various shapes as seen in Figures 3a-3d. Having a higher than usual defect ratio, these sites of the sample tend to have an agglomeration of shapes forming star-like morphologies. These still follow the same etching angle pattern and similar features are seen as previously observed on the triangular regions.

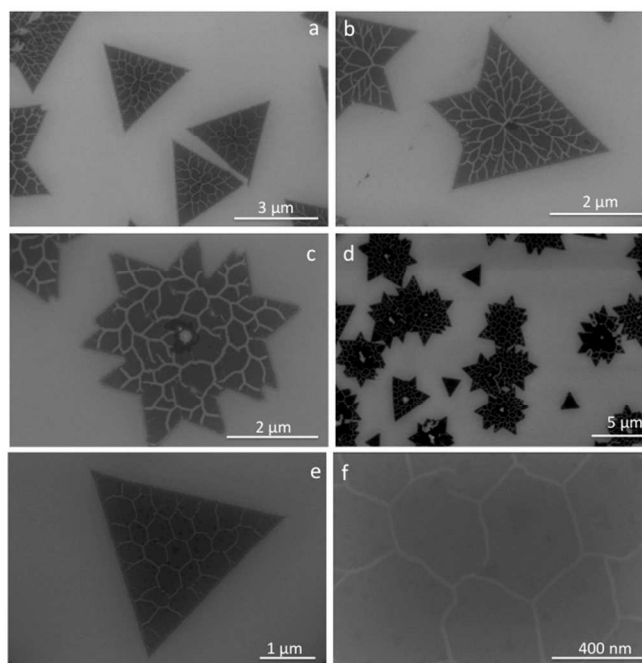


Figure 3. SEM images of complete etched patterns a-d) under the oxygen flow. In e-f) single monolayer regions are observed on the bottom of the triangular pits.

Surface morphology and etching thicknesses were studied using an AFM (Veeco) under tapping mode imaging. As seen in the AFM images, the triangular shape pits resemble the lattice of the MoS₂ basal plane surface with the edge along the zigzag direction having a lattice terminating in either Mo or S. Raman spectroscopy taken using a Horiba LabRAM HR spectrometer with an excitation wavelength of 532 nm was used to determine the number of MoS₂ layers but this can also be easily calculated from the AFM height profile. We can clearly see in Figure 4a that the color scheme changes across the sample, pointing out that the sample has a wavy trend due to the substrate and film preparation.⁴² This is an advantage since we can see what happens to the etching pattern in the film at different levels of thickness. In Figure 4c we observe and validate that the triangles are in fact etching down into the film and the veins observed in the SEM images are illustrated in Figure 4d. Despite the fact that it has previously been reported that it is difficult to obtain AFM height profiles of thick MoS₂ layers²³, we have successfully obtained such measurements with our grown MoS₂ films as seen in Figure 4b. Taking into consideration the variations of film thickness across our samples, we deduce that the film has regions in between 10-20nm, which can clearly be seen in our AFM height profile. It is known that the surface roughness of the under layer substrate affects the thickness of layer.⁴³ We report in this work, for the first time, the etching angle calculated from the height profile measurement. We report the etching angle to be 85 degrees

normal to the wafer and constant across all samples. We can see that for both etched pits at 10 nm and 20 nm it follows the same etching angle all the way down to the substrate. We assume that this angle, just like the shape of the etched pits, comes from the lattice structure out of the material.

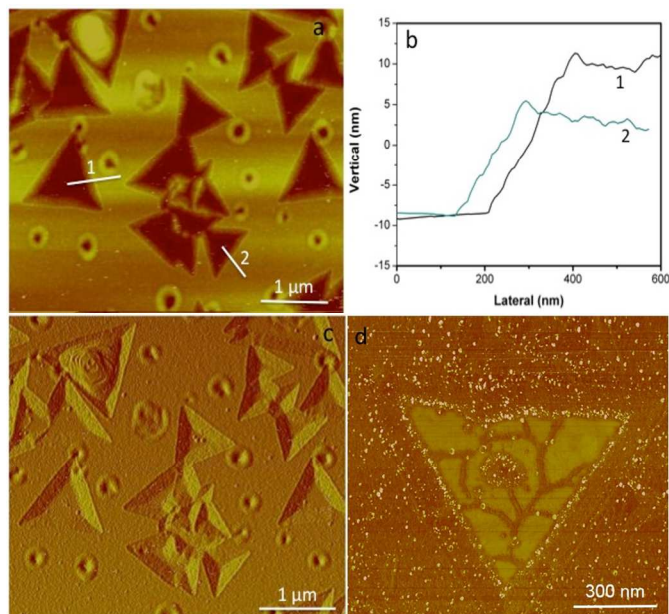


Figure 4. AFM images a, c-d) of etched pits along with height profile in b) of the MoS₂ pits.

Mo atoms are coordinated to six S atoms, and become saturated with S in the bulk form of MoS₂. Density functional theory (DFT) calculations have shown that structures with one S atom per Mo edge atom results in S atoms at the edge.^{28,44,45} In other cases where two S atoms per Mo edge are present, the structure will have a Mo edge.^{28,44,45} Looking at the MoS₂ structure we see a hexagonal layered lattice (Figure 5a, 5b and 5c) just as seen in graphene. Raw TEM image in Figure 5c was taken using an FEI Titan G² 60-300 aberration-corrected TEM operating at 200 keV Low pass and radial image filters were subsequently Transmission Electron Microscopy (TEM) confirms this hexagonal MoS₂ structure as shown in Figure 5c where the higher intensity positions are Mo atoms and the lower intensity positions are S atoms. Furthermore, the image also confirms that the film is 2H-MoS₂ based on the relative intensity ratio of the two atomic positions.^{46,47} MoS₂ is yet unique considering that one layer is composed of three individual atomic layers where Mo layers are sandwiched between two S layers, giving a zigzag three-atom layer rather than a one-atom layer of carbon known as graphene. Taking this in consideration we can now say that the MoS₂ basal plane surface has a triangular shape. The MoS₂ edge structure is known to be represented by either Mo-terminated zigzag (ZZ-Mo) or S₂-terminated zigzag (ZZ-S₂), both illustrated in Figure 5a and 5b.^{28,29,30} By knowing this, the edges of the layer forming the pits are either ZZ-Mo or ZZ-S₂ edge. These microscale pits described earlier can act as seeds across the surface and the zigzag edges will grow when a reaction starts from a surface vacancy.^{36,37} Evidence from other studies suggests that the Mo-edge is in fact more energetically stable and thus, more favored than the S edge.^{48,49} We know that the triangular pit sizes are dependent on the number of layers of MoS₂; the more layers there are, the bigger the pit sizes and the lower the etching rate on the MoS₂ surface. Since our study is based on oxygen etching, it has been previously reported that due to structural defects, etching is likely to happen at those sites. This notion tends to imply

that etching strongly depends on the crystallographic orientation, which results in the zigzag formations. The Mo and S atoms found at the edge are converted to MoO₃ and SO₂ gas molecules at higher temperatures.^{23,50,51} If the etching starts from a ZZ-Mo edge, the oxygen reacts first with Mo and then S, but if the etching starts with a ZZ-S₂, then we have a reaction with S first and then with Mo.^{23,24,48,49} It has been reported in previous studies that ZZ-Mo triangles have sharper and straighter edges than ZZ-S₂ triangles.²¹ Knowing this we concluded that our material edges are in fact ZZ-Mo edge illustrated in Figure 5a.

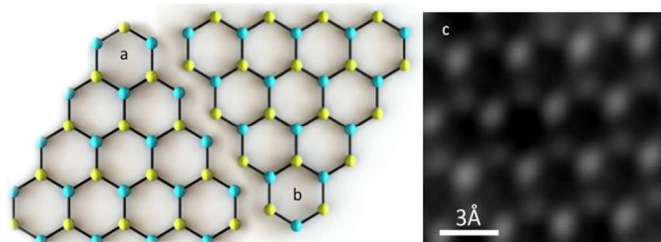


Figure 5. Molecular Schematic of (a) Mo edge and (b) S edge. (c) a filtered high resolution TEM image of an MoS₂ sample.

Raman spectroscopy measurements were carried out using a Renishaw DXR Raman spectroscopy system with a 532 nm laser (8mW excitation power, 100x objective lens) used to characterize the layers, quality, and etching of the MoS₂ film. In Figure 6a the domains clearly demonstrate the characteristic Raman peaks of E_{12g} and A_{2g}, from single and bulk MoS₂ film.^{21,52} Raman E_{2g} mode at 386 cm⁻¹ and A_{1g} mode at 405 cm⁻¹ correspond to a monolayer MoS₂ which is found on the bottom of the pits as seen in the SEM Figure 3e. Raman E_{2g} mode at 384 cm⁻¹ and A_{1g} mode at 414 cm⁻¹ correspond to bulk MoS₂ prior to oxidation due to the film being 10-20 nm thick. PL of these samples was taken and recorded to have a noticeable intensity only inside the wells. The PL spectra seen in Figure 6b was recorded at 670.5 nm translating to about 1.85eV, which has previously been reported in literature as monolayer regions. The high PL intensity, which has been previously reported, is known to come from the decrease of electron density resulting from sulfur-rich defect states after etching.^{20,21,22,23,25,30,53} In both Raman and PL a peak located at 521 cm⁻¹ in the spectrum was observed, corresponding to the Si/SiO₂ substrate. The measurements provide evidence of the MoS₂ etching and the fact that a longer etching time (20–30 min) leads to the complete removal of the MoS₂ film spanning a high intensity Si peak. After etching, the presence of MoO₃ was not observed by Raman spectroscopy, in accordance with previous studies.²³

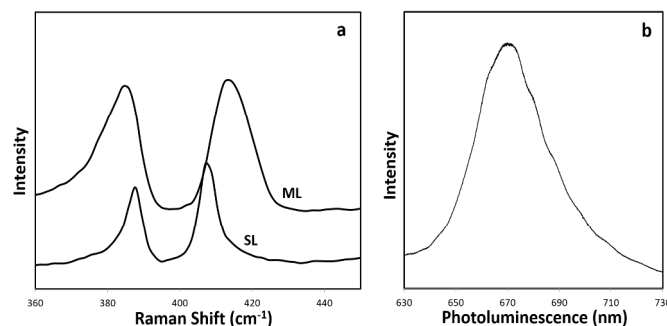


Figure 6. In (a) we see the Raman spectra of multi-layer (ML) MoS₂ and Single layer MoS₂ (SL) along with the Photoluminescence in (b).

In summary, we report on the etching of thick MoS₂ films with well-oriented etching patterned pits of different morphologies. Etching pits were obtained by exposing the MoS₂ film to oxygen flow. Similar etching angles were observed to be constant across the sample along with veins and minuscule triangles features. We concluded that the etching effects initiate on the MoS₂ films rather than substrate, but the substrate plays a role in the etching of the MoS₂ layer in contact with the Si/SiO₂ substrate.

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Notes and references

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- B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti, and A. Kis, *Nat. Nanotechnol.*, 2011, **6**, 147–150.
- Y. Yoon, W. Park, G.-Y. Bae, Y. Kim, H. S. Jang, Y. Hyun, S. K. Lim, Y. H. Kahng, W.-K. Hong, B. H. Lee, and H. C. Ko, *Small*, 2013, n/a–n/a.
- Q. H. Wang, K. Kalantar-Zadeh, A. Kis, J. N. Coleman, and M. S. Strano, *Nat. Nanotechnol.*, 2012, **7**, 699–712.
- J. R. Kyle, A. Guvenc, W. Wang, M. Ghazinejad, J. Lin, S. Guo, C. S. Ozkan, and M. Ozkan, *Small*, 2011, **7**, 2599–2606.
- S. Guo, M. Ghazinejad, X. Qin, H. Sun, W. Wang, F. Zaera, M. Ozkan, and C. S. Ozkan, *Small*, 2012, **8**, 1073–1080.
- W. Wang, S. Guo, M. Penchev, I. Ruiz, K. N. Bozhilov, D. Yan, M. Ozkan, and C. S. Ozkan, *Nano Energy*, 2013, **2**, 294–303.
- Q. Yu, L. A. Jauregui, W. Wu, R. Colby, J. Tian, Z. Su, H. Cao, Z. Liu, D. Pandey, D. Wei, T. F. Chung, P. Peng, N. P. Guisinger, E. A. Stach, J. Bao, S.-S. Pei, and Y. P. Chen, *Nat. Mater.*, 2011, **10**, 443–449.
- S. Lei, L. Ge, Z. Liu, S. Najmaei, G. Shi, G. You, J. Lou, R. Vajtai, and P. M. Ajayan, *Nano Lett.*, 2013, **13**, 2777–2781.
- J. K. Ellis, M. J. Lucero, and G. E. Scuseria, *Appl. Phys. Lett.*, 2011, **99**, 261908.
- K. F. Mak, C. Lee, J. Hone, J. Shan, and T. F. Heinz, *Phys. Rev. Lett.*, 2010, **105**.
- A. K. Geim and K. S. Novoselov, *Nat. Mater.*, 2007, **6**, 183–191.
- A. H. Castro Neto, N. M. R. Peres, K. S. Novoselov, and A. K. Geim, *Rev. Mod. Phys.*, 2009, **81**, 109–162.
- M. Valden, *Science*, 1998, **281**, 1647–1650.
- Y. Guo, *Science*, 2004, **306**, 1915–1917.
- L.-Y. Ma, L. Tang, Z.-L. Guan, K. He, K. An, X.-C. Ma, J.-F. Jia, Q.-K. Xue, Y. Han, S. Huang, and F. Liu, *Phys. Rev. Lett.*, 2006, **97**.
- X. Ma, P. Jiang, Y. Qi, J. Jia, Y. Yang, W. Duan, W.-X. Li, X. Bao, S. B. Zhang, and Q.-K. Xue, *Proc. Natl. Acad. Sci.*, 2007, **104**, 9204–9208.
- A. Splendiani, L. Sun, Y. Zhang, T. Li, J. Kim, C.-Y. Chim, G. Galli, and F. Wang, *Nano Lett.*, 2010, **10**, 1271–1275.
- C. Lee, H. Yan, L. E. Brus, T. F. Heinz, J. Hone, and S. Ryu, *ACS Nano*, 2010, **4**, 2695–2700.
- Y. Wang, C. Cong, C. Qiu, and T. Yu, *Small*, 2013, **9**, 2857–2861.
- Y. Zhan, Z. Liu, S. Najmaei, P. M. Ajayan, and J. Lou, *Small*, 2012, **8**, 966–971.
- A. M. van der Zande, P. Y. Huang, D. A. Chenet, T. C. Berkelbach, Y. You, G.-H. Lee, T. F. Heinz, D. R. Reichman, D. A. Muller, and J. C. Hone, *Nat. Mater.*, 2013, **12**, 554–561.
- S. Najmaei, Z. Liu, W. Zhou, X. Zou, G. Shi, S. Lei, B. I. Yakobson, J.-C. Idrobo, P. M. Ajayan, and J. Lou, *Nat. Mater.*, 2013, **12**, 754–759.
- H. Zhou, F. Yu, Y. Liu, X. Zou, C. Cong, C. Qiu, T. Yu, Z. Yan, X. Shen, L. Sun, B. I. Yakobson, and J. M. Tour, *Nano Res.*, 2013, **6**, 703–711.
- M. Yamamoto, T. L. Einstein, M. S. Fuhrer, and W. G. Cullen, *J. Phys. Chem. C*, 2013, **117**, 25643–25649.
- K.-K. Liu, W. Zhang, Y.-H. Lee, Y.-C. Lin, M.-T. Chang, C.-Y. Su, C.-S. Chang, H. Li, Y. Shi, H. Zhang, C.-S. Lai, and L.-J. Li, *Nano Lett.*, 2012, **12**, 1538–1544.
- S. Wu, C. Huang, G. Aivazian, J. S. Ross, D. H. Cobden, and X. Xu, *ACS Nano*, 2013, **7**, 2768–2772.
- R. Ionescu, W. Wang, Y. Chai, Z. Mutlu, I. Ruiz, Z. Favors, D. Wickramaratne, M. Neupane, L. Zavala, R. Lake, M. Ozkan, and C. Ozkan, *IEEE Trans. Nanotechnol.*, 2014, 1–1.
- S. Helveg, J. Lauritsen, E. Lægsgaard, I. Stensgaard, J. Nørskov, B. Clausen, H. Topsøe, and F. Besenbacher, *Phys. Rev. Lett.*, 2000, **84**, 951–954.
- T. F. Jaramillo, K. P. Jorgensen, J. Bonde, J. H. Nielsen, S. Horch, and I. Chorkendorff, *Science*, 2007, **317**, 100–102.
- H. Schweiger, *J. Catal.*, 2002, **207**, 76–87.
- J. Wu, H. Li, Z. Yin, H. Li, J. Liu, X. Cao, Q. Zhang, and H. Zhang, *Small*, 2013, n/a–n/a.
- A. Castellanos-Gomez, M. Barkelid, A. M. Goossens, V. E. Calado, H. S. J. van der Zant, and G. A. Steele, *Nano Lett.*, 2012, **12**, 3187–3192.
- D. Le, D. Sun, W. Lu, L. Bartels, and T. S. Rahman, *Phys. Rev. B*, 2012, **85**.
- D. Geng, B. Wu, Y. Guo, B. Luo, Y. Xue, J. Chen, G. Yu, and Y. Liu, *J. Am. Chem. Soc.*, 2013, **135**, 6431–6434.
- B. Wu, D. Geng, Z. Xu, Y. Guo, L. Huang, Y. Xue, J. Chen, G. Yu, and Y. Liu, *NPG Asia Mater.*, 2013, **5**, e36.
- R. Yang, L. Zhang, Y. Wang, Z. Shi, D. Shi, H. Gao, E. Wang, and G. Zhang, *Adv. Mater.*, 2010, **22**, 4014–4019.
- Z. Shi, R. Yang, L. Zhang, Y. Wang, D. Liu, D. Shi, E. Wang, and G. Zhang, *Adv. Mater.*, 2011, **23**, 3061–3065.
- G. Xie, Z. Shi, R. Yang, D. Liu, W. Yang, M. Cheng, D. Wang, D. Shi, and G. Zhang, *Nano Lett.*, 2012, **12**, 4642–4646.
- R. T. Yang, *J. Chem. Phys.*, 1981, **75**, 4471.
- H. Chang and A. J. Bard, *J. Am. Chem. Soc.*, 1990, **112**, 4598–4599.
- J. R. Hahn, H. Kang, S. M. Lee, and Y. H. Lee, *J. Phys. Chem. B*, 1999, **103**, 9944–9951.
- J. Brivio, D. T. L. Alexander, and A. Kis, *Nano Lett.*, 2011, **11**, 5148–5153.
- H. R. Gutiérrez, N. Perea-López, A. L. Elías, A. Berkdemir, B. Wang, R. Lv, F. López-Urías, V. H. Crespi, H. Terrones, and M. Terrones, *Nano Lett.*, 2013, **13**, 3447–3454.
- L. Byskov, *J. Catal.*, 1999, **187**, 109–122.
- P. Raybaud, J. Hafner, G. Kresse, and H. Toulhoat, *Surf. Sci.*, 1998, **407**, 237–250.
- R. J. Wu, M. L. Odlyzko, and K. A. Mkhoyan, *Ultramicroscopy*, 2014, **147**, 8–20.
- G. Eda, T. Fujita, H. Yamaguchi, D. Voiry, M. Chen, and M. Chhowalla, *ACS Nano*, 2012, **6**, 7311–7317.
- J. V. Lauritsen, J. Kibsgaard, S. Helveg, H. Topsøe, B. S. Clausen, E. Lægsgaard, and F. Besenbacher, *Nat. Nanotechnol.*, 2007, **2**, 53–58.
- Line S. Byskov, Jens K. Nørskov, Bjerne S. Clausen, and Henrik Topsøe, *Catal. Lett.*, 1999, **64**, 95–99.
- S. Ross and A. Sussman, *J. Phys. Chem.*, 1955, **59**, 889–892.
- Jeffrey R. Lince and Peter P. Frantz, *Tribol. Lett.*, 2000, **9**, 211–218.
- B. Chakraborty, H. S. S. R. Matte, A. K. Sood, and C. N. R. Rao, *J. Raman Spectrosc.*, 2013, **44**, 92–96.
- K. F. Mak, K. He, C. Lee, G. H. Lee, J. Hone, T. F. Heinz, and J. Shan, *Nat. Mater.*, 2012, **12**, 207–211.