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Enhanced Photovoltaic Performance of Perovskite CH$_3$NH$_3$PbI$_3$ Solar Cells with Free Standing TiO$_2$ Nanotube Array Films

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Free standing TiO$_2$ nanotube array films are fabricated and first applied as electrodes in perovskite CH$_3$NH$_3$PbI$_3$ sensitized solar cells. The device demonstrates improved light absorption with more than 90% of light absorbed in the whole visible range and reduced charge recombination rate, leading to a significant improvement of the photocurrent and efficiency. This study suggests a promising way of improving the conversion efficiency of perovskite solar cells through novel electrode.

The development of organometallic halide perovskite sensitizers has attracted enormous attention on solar cell applications due to their excellent light absorbing characteristics. 1 Perovskite materials were first used as sensitizers to replace traditional organic dye molecules in dye sensitized TiO$_2$ nanocrystalline solar cells with iodide based electrolyte. These cells initially obtained an efficiency in the range of 3.7-6.5%. 2 In 2012, the application of CH$_3$NH$_3$PbI$_3$ leads to a breakthrough on the solid-state dye sensitized solar cell (DSSC), with a power conversion efficiency of 9.7% obtained using submicrometer thick mesoporous TiO$_2$ film electrode sensitized by perovskite CH$_3$NH$_3$PbI$_3$ nanocrystals with spiro-MeOTAD as hole conductors. 3 At the same time, extensive research has been conducted on the development of high efficiency solid state solar cells, with a high efficiency of 10%-15% achieved. 4

The primary advantage of the perovskite absorbers is their direct bandgap with large absorption coefficients over a broad range, which enables efficient light absorption in ultra-thin films. However, when only considering the light harvesting efficiency, perovskite based devices still has large potential to be improved in terms of light management. Simulated results indicate that solar cells can obtain a photocurrent of 27.2 mA/cm$^2$ if the total photons in 280-800 nm could be used to generate electricity, 5 while current perovskite solar cells only obtain photocurrent around 20 mA/cm$^2$ on thin film solid state solar cells 6 and no more than 16 mA/cm$^2$ on iodide electrolyte based solar cells. 2 The light absorption spectrum indicates that a CH$_3$NH$_3$PbI$_3$ sensitized electrode appears to be more efficient to absorb light below 500nm. The absorbance decreases gradually from 500 to 800 nm when applied on nanoparticle electrodes. 6, 7 Proper material selection and engineering design for light management both need to be optimized to increase the photocurrent and the conversion efficiency.

In this paper, CH$_3$NH$_3$PbI$_3$ sensitized perovskite solar cells are fabricated using free standing TiO$_2$ nanotube array electrode. The perovskite absorber was infiltrated into the TiO$_2$ nanotube successfully. With the new TiO$_2$ nanotube electrode, a significant improvement of light absorption was obtained, showing over 90% light absorption in the whole visible range. Combining the effect from a reduced recombination rate, the TiO$_2$ nanotube based perovskite solar cells show a significant increase of the power conversion efficiency over the conventional nanoparticle based solar cells. In this study, the dependence of light absorption and photovoltaic performance with iodide liquid electrolytes on the nanotube length was also investigated, while the solid state hole conductors were not employed in order to avoid variations in the pore-filling fraction, which will otherwise complicate the electron recombination analysis. 8 The device obtained a high photocurrent density of 17.9 mAcm$^2$ using iodide liquid electrolyte and a conversion efficiency of 6.52%. The results indicate that TiO$_2$ nanotubes could be a promising perovskite host for enhancing the light absorption near its energy band edge, and can potentially pave a way for novel perovskite solid state solar cell designs and fabrications.

Here freestanding TiO$_2$ nanotube (TNT) arrays were prepared by a two-step anodization process and then detached from the substrate with in-situ field-assisted chemical dissolusion. 9 After transferring the free standing nanotube arrays to FTO which was covered by a spin-coated TiO$_2$ blocking layer, a sequential deposition approach was used to deposit the perovskite absorber into the nanotubes. PbI$_2$ crystals were deposited into the nanotubes by spin coating of 1M PbI$_2$ solution in N,N-dimethylformamide. For perovskite formation, the TNT/PbI$_2$ film was immersed into a solution of CH$_3$NH$_3$I in 2-propanol. In an attempt to increase the deposition of the perovskite absorber on the high aspect ratio TiO$_2$ structure, a relatively high concentration (30 mg ml$^{-1}$) of CH$_3$NH$_3$I was used, which tends to form CH$_3$NH$_3$PbI$_3$ with small particle size and fits the pore size of nanotube arrays. It was observed that a low concentration of CH$_3$NH$_3$I will result in formation of crystals with sizes typically larger than 500 nm, which may block the pores and hamper the transformation of PbI$_2$ inside the nanotubes. In this case, the formed perovskite crystals settle on the surface of the nanotube arrays and
an inefficient sensitized electrode is produced. Further experimental details are provided in Electronic Supplementary Information.

The formation of perovskite in the nanotube arrays was tracked by X-ray diffraction (XRD) spectroscopy (Figure 1). When the PbI₂ crystals were deposited into the nanotube, two additional peaks appear at approximately 12.72 and 39.52 degrees, respectively, corresponding to the reflection from (101) and (110) lattice planes of hexagonal polytype (JCPDS 07-0235). After the transformation process from PbI₂ to CH₃NH₃PbI₃, a series of new diffraction peaks related to a tetragonal perovskite structure were observed. The strong peaks located at 14.20, 19.78, 28.23, 28.52, 29.80, 31.95, 40.61 and 43.20 degrees correspond to hexagonal polytype (JCPDS 07-0235). The strong peaks located at 14.20, 19.78, 28.23, 28.52, 29.80, 31.95, 40.61 and 43.20 degrees correspond to the (101) and (110) lattice planes of the tetragonal perovskite structure. No impurity peaks appear in both steps, suggesting a complete formation of PbI₂ and its controlled transformation to CH₃NH₃PbI₃. The final perovskite absorber is well crystallized and the phase is pure.

**Figure 1.** XRD spectra of pristine TiO₂ nanotube electrode (a), PbI₂/TiO₂ nanotube electrode (b) and CH₃NH₃PbI/TiO₂ nanotube electrode (c). PbI₂ and CH₃NH₃PbI peaks are marked with star and circle respectively.

Figure 2 compares the morphology of TiO₂ nanotube electrode before and after the perovskite dye deposition using the scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images. Figure 2a shows a typical surface morphology of the as prepared TiO₂ nanotube arrays. After the two step anodization, a porous thin layer was formed on top of the nanotube array, which protects the nanotube arrays from top end bundling and avoids the cracks during the detachment and film transfer process (Figure 2a). This provides a safe way to get ultrathin free standing TiO₂ nanotube array film. It is worth noting that the surface porous layer is connected with the underlayer tube opening (Figure S1), which would not affect the following perovskite deposition process. To investigate the dependence of the photovoltaic performance on the film thickness, three types of TiO₂ nanotube array films were prepared with the tube length controlled at 2.3µm, 4.8µm, and 9.4µm during the anodization process (Figure S2). Figure 1b shows the cross section image of the nanotube arrays, indicating a well-controlled nanotube structure formed. After the perovskite absorber deposition, the nanotube structure remains intact. It is shown that some nanoparticles attached on the edge of the nanotube, and the formation of the perovskite absorber did not block the pores (Figure 2d). Cross-sectional SEM images and TEM images (Figure 2e-f) indicate a successful infiltration of the perovskite absorber into the nanotubes. The particle sizes of CH₃NH₃PbI were limited to less than 100nm as controlled by the pore size of the nanotube. Cross-sectional EDS mapping (Figure S3) indicates Pb and I are well distributed along the nanotubes array film. Weight percentages for Pb and I from EDS elemental analysis are found at 5.22% and 10.11%, respectively, which corresponds to a 0.78% and 2.47% atomic ratio and indicates a good stoichiometric ratio of Pb to I in the obtained perovskite CH₃NH₃PbI₃.

Figure 3a shows typical ultraviolet-visible (UV-Vis) absorption spectrum of CH₃NH₃PbI sensitized 4.8µm TiO₂ nanotube array electrode. For comparison, a spectrum of photoelectrode based on TiO₂ nanoparticle film with the same thickness was presented. When compared, the nanotube based electrodes have significantly larger absorbance of visible light over the whole spectrum than nanoparticle based electrodes. The nanoparticle based electrodes appear to be efficient to absorb light below 500nm. The absorbance decreases gradually from 500 to 800 nm, which is consistent with previously reported results. With TiO₂ nanotubes as the perovskite host, the absorbance is clearly improved, especially in the long wavelength region, which could contribute to the enhanced light trapping ability of the nanotube array. The enhanced light absorption will help generate the photo excited electrons and eventually improve the power conversion efficiency. Note that the scale in the figure is optical density, where an absorbance of ~1 corresponds to ~90% light absorption. Figure 3b presents corresponding absorption characteristics of both nanoparticle and nanotube electrodes, where nanotube electrodes show over 90% of light absorption in almost the full visible light region. Figure 3b also presents the dependence of the light absorption on the nanotube length. It is interesting to observe that when the TiO₂ nanotube length increases from 2.3µm to 9.4µm, the absorption is only slightly increased in the 600-800 nm region, which could be related to the overall strong absorption of CH₃NH₃PbI sensitized TiO₂ nanotube electrodes. It is worth noting that even with 2µm nanotube arrays, the photoelectrode shows a very high light absorption. The results indicate that TiO₂ nanotubes could be a promising host for CH₃NH₃PbI solar cells with significant light absorption.
enhancement effects, which is important in design of high efficiency solar devices and provides a new route to improve current perovskite solar cell technology.

Figure 4a shows the current-voltage curves of perovskite solar cell based on TiO$_2$ nanoparticle and TiO$_2$ nanotube array films under AM1.5 illumination. A photoconversion efficiency of 4.46% is obtained on the 4.8 µm TiO$_2$ nanotube electrode, resulting from a photocurrent density (J$_{SC}$) of 13.1 mA/cm$^2$, open-circuit voltage (V$_{OC}$) of 0.60V and fill factor of 0.568. Notice that the J-V curve is obtained under reverse scan with a slow scan speed, which can minimize the efficiency deviation under different scan directions$^{11}$ (Figure S4 and Table S1). In comparison, on the nanoparticle electrode with the same thickness, the cell efficiency is only 2.99% with a photocurrent density (J$_{SC}$) of 9.5 mA/cm$^2$, open-circuit voltage (V$_{OC}$) of 0.60V and fill factor of 0.525. In our tests, TiO$_2$ nanotube based device also presents higher incident photo to current efficiency (IPCE) at all effective photo response region (inset of Figure 4a). It indicates that using TiO$_2$ nanotubes as the electrode host results in a significant photocurrent increase and thus an obvious photo conversion efficiency improvement.

To understand the effect of TiO$_2$ nanotube array on the improved photovoltaic performance, photovoltage decay measurements were performed to investigate the charge transport and recombination properties of the TiO$_2$ nanotube based perovskite photoelectrode (Figure 4b). Briefly, the electron lifetimes ($\tau_e$) could be derived from the photovoltage decay curve according to,$^{12}$

$$\tau_e = \frac{k_B T}{q} \left( \frac{dV_{OC}}{dt} \right)^{-1}$$

Where $k_B$ is the Boltzmann constant, $T$ is the absolute temperature, and $q$ is the positive elementary charge. Figure 4b shows the photovoltage decay plots and the electron lifetimes as a function of photovoltage (inset). It is observed that the photovoltage of the TiO$_2$ nanotube array electrode decays slower than that of the TiO$_2$ nanoparticle electrode, which also presents a longer electron life time at equal potentials. The decreased electron life time of the TiO$_2$ nanotube electrode means a higher charge recombination rate and a lower charge collection efficiency than those of the TiO$_2$ nanoparticle electrode.$^{13}$ These results indicate that the enhancement of TiO$_2$ nanotube based perovskite solar cells is attributed to both the enhanced light absorption and the reduced charge recombination.

To investigate the potential of TiO$_2$ nanotubes as a host for perovskite solar cells, the dependence of photovoltaic performance on the nanotube film thickness was also studied (Table 1). It is found that the device performances show strong dependence on nanotube length. A photo conversion efficiency of 6.52% is obtained, resulting from a 2.3 µm nanotube array film with a photocurrent density (J$_{SC}$) of 17.9 mA/cm$^2$, open-circuit voltage (V$_{OC}$) of 0.63V and fill factor of 0.578. The device efficiency is among the best results reported on perovskite solar cells using a liquid electrolyte.$^{2, 7}$ Considering that additional surface treatment was not performed and the device parameters were not fully optimized, this result is very encouraging.

<table>
<thead>
<tr>
<th>$V_{OC}$ [V]</th>
<th>J$_{SC}$ [mA/cm$^2$]</th>
<th>FF</th>
<th>η [%]</th>
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<tr>
<td>2.3 µm</td>
<td>0.63</td>
<td>17.9</td>
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<td>4.8 µm</td>
<td>0.60</td>
<td>13.1</td>
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<tr>
<td>9.4 µm</td>
<td>0.54</td>
<td>9.7</td>
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Considering the thickness of the TiO$_2$ nanotube film only has slight effect on the light absorption due to the overall high absorption ability, it is concluded that charge collection efficiency governs the performance of the TiO$_2$ nanotube based perovskite solar cells. Photovoltage decay measurements (Figure S5) show that the photovoltage of the short nanotube array film electrode decays slower, which also presents a longer electron life time at equal potentials. The decreased electron life time of the long nanotube film suggests a high charge recombination rate and relatively low charge collection efficiency, which consequently limits the photo conversion efficiency of the devices. IPCEs (Figure S6) of shorter nanotube based devices show significant higher quantum efficiency resulting from the higher charge collection efficiency, which is consistent with photovoltage decay measurement. Since the electron collection efficiency is close to 100% under short circuit conditions even for 20 µm thick nanotube arrays,$^{14}$ it is therefore reasonable to predict that a higher recombination rate may be due to the restrained regeneration of the oxidized dyes, which is related to excited hole extraction. To get a stable performance, low concentration I$^-$/electrolyte was used here to restrain the dye bleaching, which will also result in a less effective regeneration of the oxidized dyes and lead to an increased electron recombination.$^{15}$ When the length of TiO$_2$ nanotube increases, the I$^-$ diffusion pathway is prolonged, resulting in an enhanced I$^-$ depletion level inside the nanotubes, and subsequently an acceleration of the recombination.$^{15}$ It should be noted that the long term stability of nanotube based device is still not good because of the chemical instability of perovskite in the iodide electrolyte. Similar to other liquid perovskite solar cells, the degradation of the cell performance is accompanied by perovskite bleaching in iodide electrolyte. To improve their long term stability and fully utilize the advantage of nanotube structure, development of TiO$_2$ nanotube based solid state perovskite solar cells is currently in progress.

In summary, perovskite solar cells based on free standing TiO$_2$ nanotube arrays are developed, which presents an improved photovoltaic performance with enhanced photocurrent and increased power conversion efficiency. The improvement is not only applicable to the boosting of light absorption, but also contributes to the reduced charge combination in nanotube electrodes, which suggests TiO$_2$ nanotube arrays could be a promising perovskite host to improve the efficiency further. It is also found the photovoltaic performance increases with the decreasing of the TiO$_2$ nanotube film thickness, attributing to the different photoexcited charge extraction and collection efficiencies. An encouraging photovoltaic
performance with 17.9 mA/cm² photocurrent and 6.52% efficiency was shown by the solar cell with 2.3 µm TiO₂ nanotubes. Considering the feasibility of short nanotubes to provide uniform hole transport material deposition without losing light absorption enhancement features, the well aligned TiO₂ nanotube array films could be a promising electrode for high efficiency solid state perovskite solar cells fabrication.

Notes and references

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Electronic Supplementary Information (ESI) available: Experimental details, Cross section images of free standing TiO₂ nanotube array, EDS mapping of TiO₂/perovskite film and photovoltage decay measurement and IPCE of TiO₂ nanotube devices with varied thickness are available. See DOI: 10.1039/c000000x/


