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Productive encounter: Molecularly imprinted nanoparticles prepared using magnetic templates

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Synthesis of core shell nanoparticles by surface initiated reversible addition fragmentation chain transfer polymerization in presence of a chiral template conjugated to magnetic nanoparticles is reported. The approach leads to imprinted nanoparticles featuring enantioselectivity and ¹⁰ enhanced affinity compared to nanoparticles prepared using free template.

Molecular imprinting is an established technique for producing polymers complementing specific molecules with respect to shape and functionality. ¹⁻⁶ These molecularly imprinted ¹⁵ polymers (MIPs) can now mimic antibodies with respect to both

- ¹⁵ porymers (with s) can now minine antibodies with respect to both binding affinity and size and their ability to function in complex environments has considerably expanded the scope of applications e.g. receptor assays, sensors, affinity separations. The development of new and improved methods for producing ²⁰ these receptors is hence an urgent goal.
- Most examples of high fidelity molecular imprinting have been demonstrated using highly cross-linked macroporous polymers as the imprinting matrix. ⁴⁻⁶ In such amorphous polymers the templated sites are not uniform and hence binding curves do not
- ²⁵ follow a simple 1:1 ligand receptor binding model. ⁷ Template occlusion is another recurring problem in traditional molecular imprinting. Typically a small fraction of the template added to the monomer mixture remains in the polymer matrix which can result in bleeding a process detrimental in trace analysis. ⁸ Moreover,
- ³⁰ template recovery and recycling is complicated and require multiple purification steps. Finally, traditional imprinting techniques are difficult to upscale, a fact seriously compromising commercial exploitation.

Several approaches addressing the aforementioned problems have

- ³⁵ been proposed. Thermodynamically controlled polymerizations^{9,10} or post-treatments such as thermal curing or annealing¹¹ are anticipated to reduce structural heterogeneity whereas the use of immobilized templates instead reduces binding site diversity, presumably due to a preferential
- ⁴⁰ orientation imposed by the interfacial confinement of the template. ¹²⁻¹⁴ On the other hand, binding sites in imprinted nanoparticles experience less diverse microenvironments and, as for proteins, such particles can be affinity purified in order to further boost affinity. ¹⁵⁻¹⁷ An elegant approach in this context is
- ⁴⁵ surface imprinting of nanoparticles by solid phase synthesis.¹⁸⁻²⁰ The nanoparticles are here synthesized in presence of template modified solid supports whereby growing particles adhere to the support surface. Post-synthesis, the particles can be affinity purified in situ leading to high affinity receptors in template free
- ⁵⁰ form. A limitation with the examples demonstrated thus far is related to the low specific surface area of the solid support beads. This translates into low particle yields (< 1 mg/g support beads) and a need for large reactors which essentially limits the technique to serial synthesis protocols.²¹

⁵⁵ Here we introduce a novel scalable process to produce surface imprinted nanoparticles in high yield and in template free form. This is based on the use of nanosized magnetic placeholder templates (Figure 1) in a process tolerating high template concentrations. The increased product yield should make this
⁶⁰ method ideal for both small-scale parallel synthesis and large-scale synthesis by established polymerization techniques. In this report we demonstrate the feasibility of the concept focusing on an extensively studied model system combined with RAFT mediated polymerization²² from nanosized silica cores (Figure 1)



Figure 1. Principle of using magnetic templates for synthesis, affinity enrichment and purification of surface imprinted core shell micro and nanoparticles. (1) Polymerization of monomers in presence of the binary 70 particle suspension, (2) Separating the magnetic template and polymers adhering to the template from the crude reaction mixture. (3) Washing off loosely bound unreacted monomers and oligomers, (4) Gradually releasing the polymer adhering to the magnetic template by physical or chemical means thereby enriching high affinity MIP particles. (5) Reuse 75 of the magnetic template repeating steps 1-4.

Prior to introducing magnetic templates a procedure for producing core-shell imprinted nanoparticles using soluble template was developed. This relied on our previously reported procedure for surface initiated polymerization of methacrylic acid 80 (MAA) and ethyleneglycol dimethacrylate (EGDMA) from

mesoporous silica using L-phenylalanine anilide (L-PA) as template.^{10, 22}

The RAFT agent was coupled to the aminofunctionalized colloidal silica nanoparticles via the R-group according to Li et ⁸⁵ al. ²³ resulting in surface coverages in accordance with our previous investigations for modification of mesoporous silicas (Table S1). ^{10, 22} Each step was accompanied by probing the

colloidal stability in different solvents. Hence, whereas the bare and aminofunctionalized colloidal core particles SiNP and SiNP-⁹⁰ NH₂ formed stable dispersions in polar solvents (e.g. isopropanol, acetone, acetonitrile) the RAFT modified particles were partially aggregated nevertheless being well dispersible in alcohols such as

isopropanol (Figure S2). The particles could be recovered by

either centrifugation at ca 3500rpm or by precipitation in hexane. Dynamic light scattering (DLS) resulting from the particles dispersed in isopropanol (SiNPs) or water (magNPs) showed Zaverage particle sizes in rough agreement with particle and s aggregate diameters estimated from corresponding TEM images (Figure S3) (Table 1).

Table 1. Z-average particle size and polydispersity from DLS of nanoparticles used in the study

Particle ^a	Diameter (nm)	Polydispersity
magNP	548	0.276
magNP@SiO2	605	0.206
magNP-NH ₂	442	0.101
magNP-L-Phe	281	0.216
SiNP	25	0.245
SiNP-NH ₂	32	0.183
SiNP-RAFT	65	0.341
SiNP-MIP2	45	0.255

a) The dispersing solvents were water (magNP) or isopropanol (SiNP).
 ¹⁰ The specific surface areas by BET of magNP@SiO₂ and SiNP were 110 m²/g and 182 m²/g respectively.

Imprinted copolymers of MAA and EGDMA were then grafted from the supports according to Figure S1 and Table S2, i.e. in a

- ¹⁵ 1:5 molar ratio of MAA to EGDMA in presence of ca 5 mol% of L-phenylalanine anilide (L-PA) as chiral template and with the beads dispersed in toluene. Grafting requires an external source of primary radicals which was here provided by a soluble initiator, added in substoichiometric amounts with respect to the
- ²⁰ RAFT agent.²² The quantity of monomer relative to the silica supports were adjusted to result in shells with approximately 4 nm thick shells. After polymerization the beads were isolated by centrifugation and subjected to repetitive washing-centrifugation cycles in order to remove any leachables (e.g. template,
- ²⁵ oligomers, unreacted monomers). Five cycles were sufficient for exhaustive template removal as concluded by HPLC analysis of the washing fractions. The beads were subsequently characterised by FTIR, TEM, DLS, TGA and elemental analysis. The FTIR spectra of the core shell beads shown in Figure S4 display two
- ³⁰ characteristic bands *i.e.* the carbonyl stretching of the polymer matrix at ca 1740 cm⁻¹ and the siloxane vibration of silica core at ca 1120 cm⁻¹. As expected, the ratio of these band intensities scale with the density of grafted polymer in agreement with corresponding data from mesoporous composites²² all in all ³⁵ indicating a successful grafting of the polymer shell.
- In Table S2 the apparent thickness, calculated from the TGA mass loss data and elemental analysis, have been compared with the nominal thickness, estimated assuming the grafted shell to consist of monomers forming a liquid film covering the core
- ⁴⁰ surface. The somewhat lower measured thickness compared to the nominal values agrees with our previous report²² and can be attributed to solution chain growth, nevertheless resulting in an acceptable conversion of monomer to shell polymer. TEM images confirmed the core shell architecture with shells ⁴⁵ appearing brighter due to their lower electron density (Figure
- 2B). The images further revealed separate or smaller aggregates of polydisperse particles.



Figure 2. TEM (A-C) and SEM (D) images of magNP-L-Phe (A), the L-PA imprinted core shell nanoparticles SiNP-MIP1 (B) and SiNP-MIP2 (C ss and D). Scale bar = 20nm (A-C) or 300nm (D).

The particles were subsequently tested for their affinity towards the template L-PA and its optical antipode D-PA in acetonitrile. After incubating the particles with solutions of L-PA or D-PA of known concentrations the free concentration of the solutes were determined by reversed phase HPLC. Binding curves were then constructed (Figure 3) by plotting the specific amount of bound solute against the free concentration of solute.



Figure 3. Binding curves of L-PA (solid symbols) and D-PA (open symbols) on imprinted (circles) and nonimprinted (crosses) core-shell ⁶⁵ particles in acetonitrile. A) L-PA imprinted SiNP-MIP1 and SiNP-NIP1, B) magNP-LPA imprinted SiNP-MIP2.

The curves display a distinct saturation behaviour with a clear preference for the templated L-form. In fact, the binding curve for the D-form coincides with the binding curve for the nonimprinted ⁷⁰ particles indicating the presence of highly discriminative imprinted sites. This also contrasts with binding curves reported for analogously prepared mesoporous materials where binding is weaker and less selective.²²

We then turned to the design of magnetic placeholder templates. ⁷⁵ Magnetite was chosen as a magnetic core, since this material can be obtained conveniently by the co-precipitation of Fe(II)/Fe(III) in aqueous media under base catalysis. A silica shell was then applied by aqueous hydrolysis of tetraethyl orthosilicate (TEOS) in MeOH and the resulting agglomerates (Figure 2A) ⁸⁰ characterised by DLS, FTIR and TEM. The TEM images (Figure S6) revealed core particles in the desired size range of ~ 10 nm, visible crystal lattice planes (Figure S6 c,d) confirming the presence and precisely rendering the borders of magnetite cores and the presence of the silica shell.

- ⁵ The beads were then aminofunctionalized by reaction with APS followed by EDC catalyzed attachment of Fmoc-L-phenylalanine (Figure S1). Piperidine catalyzed deprotection yielded the template L-Phe attached via its carboxyl group to the aminofunctionalized surface as indicated by the appearance of the
- ¹⁰ characteristic amide stretching bands (Figure S7) and by the release of fulvene-piperidine adduct upon Fmoc-Phe deprotection (Table S1). With the template modified magnetic beads at hand our next goal was to use them for affinity capture of the imprinted nanoparticles. Crucial in this context was the development of a
- ¹⁵ colorimetric particle assay allowing facile quantification of soluble nonbound particles (Figure S8). This consisted of first converting the RAFT end groups to thiols through aminolysis followed by colorimetric detection of the thiol decorated colloidal particles by the established Ellman thiol assay.²⁴ Successful
- ²⁰ aminolysis was confirmed by disappearance of the pink color and the UV absorption band at 302 nm characteristic for the dithioester RAFT group (Figure S9). The aminolysed SiNP-MIP1 and SiNP-NIP1 were then incubated with magNP-L-Phe in acetonitrile followed by separation of the magnetic fraction. The
- ²⁵ magnetic fraction was subsequently washed three times with acetonitrile and three times with acidified methanol while collecting the particles with magnet between each washing step. The original incubation solution and each wash fraction were subsequently tested for nonbound particles using the Ellman ³⁰ assay. Figure 4 shows the normalized fraction of soluble
- particles remaining free after incubation with magNP-L-Phe.



Figure 4. Distribution of nanoparticles SiNP-MIP1 (blue bars) and SiNP-NIP1 (red bars) (1mg/mL) in the fractions resulting from washing and elution of particles bound to magNP-L-Phe. SN=fraction of NPs ³⁵ remaining in supernatant after incubation and separation of the magnetic fraction, W1-W3=washes with 500μL acetonitrile, E1-E3=elution with MeOH/water/formic acid (80/5/15, v/v/v). The particle concentration was determined by the Ellman thiol assay and normalized with respect to particle concentration prior to fractionation.

- ⁴⁰ It is clear from the relative abundance of free imprinted versus nonimprinted particles that the L-PA imprinted particles adhere more strongly to magNP-L-Phe, i.e. whereas ca 60% of the adhered imprinted particles are recovered in the elution fractions using the strongly acidified solutions only 40% was found when
- ⁴⁵ the nonimprinted particles were incubated. Hence, binding occurs in spite of the fact that L-Phe is coupled through an aliphatic linker resulting in a ligand only partially matching the structure of

the soluble template L-PA (featuring an aromatic amide substituent). This can be understood given the previously ⁵⁰ reported crossreactivities of L-PA versus L-phenylalanine ethyl amide imprinted polymers²⁵ and therefore the difference is not expected to compromise the affinity of the L-PA imprinted beads for magNP-L-Phe.

Conversely this would indicate that the magnetic ligand can serve

- ⁵⁵ as a template to generate binding sites complementary to L-PA. In order to test this hypothesis, the template modified magnetic particles were introduced to replace the soluble small molecular template in the previously described imprinting protocol. In this approach, polymerization occurs under homogenous conditions
 ⁶⁰ contrasting with the previously introduced approach for solid
- phase synthesis.^{18,26} After magnetic collection, the magnetic fraction was washed followed by elution of the strongly bound particles by acidified methanol. Ca 17mg particles (SiNP-MIP2) were recovered in the 65 elution fraction corresponding to an overall gravimetric yield of ca 9% assuming quantitative conversion of monomer to core shell polymer. Analysis of the different particle fractions by TGA (Figure S10) showed an increasing mass loss in the order magNP-L-Phe (4%), magNP-L-Phe/SiNP-MIP2 after washing (37%),
- ⁷⁰ magNP-L-Phe/SiNP-MIP2 prior to washing (40%) and released SiNP-MIP2 (43%). This order is expected assuming the aggregates to consist of silica core/polymer shell particles loosely and strongly adhered to magnetic FeO/SiO₂ particles. The purified SiNP-MIP2 were then characterised by FTIR, TEM,
- ⁷⁵ SEM and DLS. Collectively the data demonstrate the successful formation of nanoparticles (Figure 2C,D; Figure S11) with a Z-average size of 45 nm and a polydispersity of 0.255. The close agreement between the polydispersity index of SiNP-MIP2 and the silica core offers support for the anticipated core shell
- ⁸⁰ architecture. We next turned to the binding tests which were performed as for the particles prepared using the soluble template. Interestingly the material prepared using the magnetic placeholder template showed somewhat steeper binding curves and higher uptake of L-PA compared to SiNP-MIP1 (Figure 3B).
 ⁸⁵ Hence the immobilized ligand seems rather effective in generating imprinted sites complementary to L-PA. The enantioselectivity and affinity appears particularly striking given that the particles were generated using *ca* 10 times less template compared to the conventional procedure.

The here reported method profit from the high surface to volume ratio of template decorated magnetic nano-particles potentially allowing a much higher product yield of affinity enriched imprinted particles. In addition, the use of immobilized templates ⁹⁵ and RAFT mediated surface initiated polymerization should lead to more accessible and uniform binding sites. RAFT controlled precipitation polymerization is in this context known to create monodiperse nanoparticles with superior recognition properties.²⁷ These aspects in addition to the fact that polymerization take ¹⁰⁰ place in homogenous media hold great promise with respect to method scalability and parallel synthesis. We are currently exploiting these possibilities while applying the concept to other model systems including those of biological significance.

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- 26. The outcome here depends on how the two types of nano-particles collide or aggregate as polymerization progresses. DLS of a solution
- of the reacting nano-particles dispersed in a polymerization mimicking solvent allowed some preliminary predictions in this regard. The results (Figure S12) revealed a tendency for SiNP-RAFT to associate to magNP-L-Phe. Such aggregates could be the precursor for imprinted particles which hence can be magnetically collected
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