

Analytical Methods

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Enhanced gas sensing performance of one-pot-synthesized Pt/CdIn₂O₄ composites with controlled morphologies

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4 **Abstract:** In this paper, Pt/CdIn₂O₄ composites with different morphologies and
5
6 various Pt contents were prepared through a facile one-pot hydrothermal method. The
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8 crystallographic surface of CdIn₂O₄ crystals in different shapes, including octahedrons,
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10 truncated octahedrons, 14-facets polyhedrons and cubes were successfully controlled
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12 with different Pt contents. Inspired by the unique function of Pt and CdIn₂O₄, the
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14 Pt/CdIn₂O₄ composites were applied as gas-sensing materials. The tests of
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16 gas-sensing performance demonstrated that not only the Pt dopant but also the surface
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18 structure of CdIn₂O₄ crystals affected the gas-sensing properties of the samples
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20 obtained. CdIn₂O₄ with 0.98 at.% Pt exhibited the superior response to acetone,
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22 indicating a great potential application in gas sensors.
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29 **Keywords:** CdIn₂O₄; Pt modification; Gas Sensor; Enhanced Performance
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1. Introduction

The application of semiconductor metal oxides in the production of gas sensors has triggered much research activity because of their high response for different gaseous species [1-5]. The gas-sensing process of metal oxide sensors generally involves a catalytic reaction between the gas to be monitored and the adsorbed oxygen species on the surface of semiconductor metal oxides [6-10]. The particle size, defects, surface properties of metal oxides directly affect the state and the amount of adsorbed oxygen species and consequently the performance of the gas sensors. Different crystallographic facets corresponding to different shapes of metal oxides have distinctive surface atom structures and surface energies, which have been demonstrated in chemical reactivity, sensing property of nano- and micro-sized crystals [11-16]. Therefore, it is of great importance to predict and control the exposed facets and to explore the facet-dependent properties of metal oxides.

Besides the control over the morphologies of the metal oxides, surface modification by metals can also provide an alternative method to improve the sensor performance [17-27]. Nowadays, it is believed that the presence of catalytic noble metal elements, such as Au, Pt and Pd, on the surface of the metal oxides can enhance the sensing performance due to the good charge storage capability and the well-known spillover effect. For instance, it was found that in situ vapor deposition of Pd and Ag particles onto SnO₂ nanowires and nanobelts could enhance the gas sensing performance of SnO₂ materials [22, 23]. Especially, the metal oxide/metal hybrid

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4 nanomaterials such as Au/ZnO nanopyramids [24], Ag/TiO₂ microspheres [25],
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6 Pd/ZnO nanowires [26], Pt/ZnO nanoflowers [27] have demonstrated great potential
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8 for application in gas sensors.
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11 As a typical n-type semiconducting materials, cadmium indate (CdIn₂O₄) has
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13 received much attention over the past decade due to its unique optical and electronic
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15 properties [28-30]. Containing CdO and In₂O₃, that are the two good base materials of
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17 transparent conducting oxides [31], CdIn₂O₄ is characterized by its large energy
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19 separation between the first and second conduction bands and excellent electrical
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21 properties with a carrier concentration of about $1 \times 10^{20} \text{ cm}^{-3}$ without additional doping
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23 [32] and is expected to be an excellent gas sensing materials. Although the gas
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25 sensing properties of both CdIn₂O₄ thin films and CdIn₂O₄ nanoparticles have been
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27 studied [33, 34], the systematic investigation on the shape control of CdIn₂O₄ and the
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29 effect of Pt particles on the gas sensing properties of CdIn₂O₄ have not yet been
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31 accomplished.
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39 In this study, CdIn₂O₄/Pt composites with different morphologies including
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41 octahedrons, truncated octahedrons, 14-facets polyhedrons, and cubes were fabricated
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43 via a fast one-step hydrothermal method. Furthermore, the sizes gradually decreased
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45 with the shape evolution of CdIn₂O₄ from octahedron to cube. Especially, the
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47 influences of the noble metal, Pt, on the gas-sensing properties of CdIn₂O₄ have been
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55 2. Experimental section

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2.1 Materials synthesis

All of the reagents were analytical purity and were used without further purification. Cadmium nitrate ($\text{Cd}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$), indium nitrate ($\text{In}(\text{NO}_3)_3 \cdot 4.5\text{H}_2\text{O}$) and potassium hydroxide (KOH) were purchased from Beijing Chemical Reagent Company Limited of China. Hexachloroplatinic acid ($\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$) was obtained from Sinopharm Chemical Reagent Company Limited of China.

Typically, a aqueous solution of $\text{Cd}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$ was added to a aqueous solution of $\text{In}(\text{NO}_3)_3 \cdot 4.5\text{H}_2\text{O}$ at room temperature to form a clear solution with Cd/In molar ratio 1:1.8. A solution of KOH was added to the mixture with the molar ratios of the $\text{Cd}^{2+} : \text{In}^{3+} : \text{OH}^- = 1:1.8:8$ to form a white emulsion under stirring. The influence of molar ratio on the purity of CdIn_2O_4 has been discussed in our previous paper [35]. The emulsion was stirred for another 2 h. Then a suitable amount of $\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$ aqueous solution (0.02 g mL^{-1}) was added into the mixture. After further stirring for 1 h, the stock mixture was transferred into a 100 mL Teflon-lined stainless steel autoclave that was subsequently sealed and maintained at $260 \text{ }^\circ\text{C}$ for 20 h. After the reaction mixture had been cooled to room temperature naturally, the resulting precipitate was collected by centrifugation and washed several times with ethanol and deionized water alternately and dried at $70 \text{ }^\circ\text{C}$ in air for further characterization.

To evaluate the effect of Pt amount on the gas sensing properties of the base material, we have fabricated several types of devices with the increasing weight percent of Pt. The loading amount of Pt was denoted as $x \text{ at.}\%$, where x indicates the

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4 percentage of Pt in the final products measured by an energy dispersive X-ray
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6 spectrometer (EDX).
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8 9 **2.2 Materials characterizations**

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11 The phases of the samples were characterized using X-ray diffraction (XRD;
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13 D/Max-2000 Rigaku) with Cu K α radiation ($\lambda = 0.15418$ nm, 40 kV, 100 mA) at a
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15 scan rate of 8 °/min. The structural properties and morphology of the products were
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17 studied by field emission scanning electron microscopy (FESEM, Hitachi S-4800)
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19 with an accelerating voltage of 10 kV. energy dispersive X-ray (EDX) spectra were
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21 determined using the Hitachi S-4800 EDX spectrometer attached to field emission
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23 scanning electron microscopy. Transmission electron microscopy (TEM) images of
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25 the samples were taken at an accelerating voltage of 200 kV with a Tecnai F20 TEM.
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31 32 **2.3 Fabrication and measurement of gas sensor**

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34 The gas sensor structure and the testing method were similar to that of our
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36 previous reports [36, 37]. The gas sensor was fabricated by coating aqueous slurry of
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38 the synthesized samples onto the ceramic tube without an additional annealing
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40 process. Then a Ni-Cr alloy coil through the tube was employed as a heater to control
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42 the operating temperature, and the electrical contact was made through connecting the
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44 four platinum wires with the instrument base by gold paste. The gas sensing
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46 properties (sensor response and response/recovery curves) were performed on a
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48 computer controlled WS-30A (Weisheng Electronics Co. Ltd., China) system.
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51 Typically, the gas sensors were placed in a chamber and the gas sensing experiments
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54 were conducted under a pressure of 1 standard atmosphere and at relatively humidity
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4 of 20%. Detecting gases were injected into the test chamber by a microinjector and
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6 were mixed with air by a little electric fan. The sensor was exposed to air again by
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8 removing the chamber. For reducing gases such as acetone, the gas response is
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10 determined as the ratio R_a/R_g , where R_a is the resistance in ambient air and R_g is the
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12 resistance in tested gas, respectively.
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15 16 17 18 **3. Results and discussion**

19 20 21 **3.1 Crystalline structure and morphology.**

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23 Fig. 1 (a-e) shows the typical morphologies of the sample obtained with different
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25 Pt amounts, from which we can see that all the CdIn_2O_4 particles are polyhedrons.
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27 The morphologies and sizes of CdIn_2O_4 change with the increase of Pt contents. The
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29 morphologies of CdIn_2O_4 vary from octahedron, corresponding to Pt = 0 at.% in Fig.
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31 1a, to truncated octahedrons corresponding to Pt = 0.26 at.% in Fig. 1b, to 14-facets
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33 polyhedrons corresponding to Pt = 0.40 at.% in Fig. 1c, to microcubes corresponding
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35 to Pt = 0.98 at.% in Fig. 1d, and microcubes with smaller size corresponding to Pt =
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37 1.75 at.% in Fig. 1e. From the images in Fig. 1(a-e), it is also obvious that the size is
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39 smaller and smaller with the increment of Pt amount. Corresponding EDX spectra,
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41 shown in Fig. 1(a'-e'), confirm that the samples consist primarily of Cd, In, O for
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43 pristine CdIn_2O_4 and Cd, In, O, Pt for Pt/ CdIn_2O_4 , respectively.
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52 The crystal phase of Pt/ CdIn_2O_4 with different loading amounts of Pt was
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54 characterized by X-ray powder diffraction. As shown in Fig. 2a, the sharp diffraction
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56 peaks indicate the good crystalline quality of CdIn_2O_4 when the weight percent of Pt
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4 was 0 at.%. The XRD pattern shows peaks at 16.7°, 27.5°, 32.3°, 33.8°, 39.2°, 48.6°,
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6 51.8°, 56.8°, 59.6°, 64.2°, 66.9°, 67.7°, 71.2°, and 77.9° indexed to (111), (220), (311),
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8 (222), (400), (422), (511), (440), (531), (620), (533), (622), (444), and (642) planes of
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10 the CdIn₂O₄ crystal given by the standard data file (JCPDS No. 29-0258). CdIn₂O₄
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12 crystals are of cubic spinel structure with lattice constant a = 9.166 nm. There are no
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14 other clear peaks coincident with those peaks of impurities. By comparison, it is
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16 observed that the XRD patterns of Pt/CdIn₂O₄, Fig. 2(b-e), are very similar to that of
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18 the pure one, indicating that the formation of Pt in the hydrothermal reaction process
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20 has no influence on the crystal structure of CdIn₂O₄. There are almost no diffraction
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22 peaks of Pt in the XRD pattern when the Pt amount is as low as 0.26 at.% (Fig. 2b),
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24 which can be explained by consider that the quantity of Pt is much smaller than that of
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26 CdIn₂O₄. With increasing weight percent of Pt, 0.40 at.%, 0.98 at.% and 1.75 at.%, the
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28 peaks indexed to Pt at 39.8°, 46.2°, which can be ascribed to the (111), (200) planes of
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30 face-centered cubic (fcc) Pt (JCPDS No. 04-0802), also appear and the intensities of
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32 those peaks increase gradually, indicating more and more Pt particles are attached on
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34 the CdIn₂O₄ (Fig. 2c, 2d, 2e).
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44 Fig. 3a is a TEM image of CdIn₂O₄ particles loaded with 0.98 at.% Pt, clearly
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46 showing their microcube morphology with sizes of about 120 nm. CdIn₂O₄ has an
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48 FCC cubic structure and the exposed surfaces of the cubes are made of six {100}
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50 facets. As shown in Fig 3b, two well-faceted cubic particles under an electron beam
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52 perpendicular to their top surface resemble squares. Fig. 3c is a high-resolution TEM
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54 (HR-TEM) image showing the interface between Pt and CdIn₂O₄. The contrast
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4 difference between Pt and CdIn₂O₄ is apparent. HRTEM image reveals the clear
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6 parallel fringes with d-spacing of 0.324 nm, which corresponds to {220} crystal
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8 planes of CdIn₂O₄. The interplanar distance of 0.227 nm corresponds to Pt (111)
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10 planes. Fig. 3d contains the corresponding electron diffraction patterns of Pt-CdIn₂O₄
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12 in TEM apparatus, showing two different natures of crystals. One set of diffraction
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14 patterns with a square symmetry, which is the typical diffraction pattern of the [001]
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16 zone axis, can be indexed to single-crystal CdIn₂O₄, and the other irregular diffraction
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18 spots (marked with white arrows) arise from the small-sized Pt nanoparticles.
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23 24 **3.2 Formation mechanism of the morphology**

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26 SEM observation of the samples reveals that the shape and size of the
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28 as-prepared CdIn₂O₄ polyhedral particles depend closely on the addition of Pt. As the
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30 amount of Pt increases, the morphologies of the samples change from octahedron to
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32 cube with the decreasing size. Pt particles was derived from the reduction of precursor,
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34 H₂PtCl₆·6H₂O. Hexachloroplatinic acid is a weak acid and it has acidic character to
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36 affect the pH value of the whole reaction. In fact, the growth of CdIn₂O₄ is related to
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38 the specific pH value of the system. Previous investigations confirmed that OH⁻ was a
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40 successful capping agent for different crystals [38-40]. As for the samples
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42 corresponding to different amount of Pt in this work, measurements of the pH value
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44 of the reaction system before and after the hydrothermal experiments showed the
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46 changed acidity and detailed pH information is summarized in Table 1. As we can see,
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48 before the addition of H₂PtCl₆·6H₂O aqueous solution, the original pH value of the
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50 system was 12.46. There was a chemical reaction of the OH⁻ with the metal cations
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4 during the hydrothermal process which could generate insoluble final product,
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6 resulting in decrease of pH value, 9.23. With the increase of the Pt precursor, pH
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8 values appeared decreasing gradually. Therefore, it could be concluded that the
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10 modification of Pt with various content might modulate the OH⁻ content and further
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12 modulate the morphology of the final products.
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15 16 **3.3 Gas sensing properties**

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18 It is well known that morphology, size, and surface structure of metal oxide
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20 semiconductor can dramatically change its gas-sensing performance [41, 42]. The
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22 deposition of noble metal particles onto the metal oxide has also been demonstrated to
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24 improve the gas-sensing properties [26, 27]. So, it could be expected that such a
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26 CdIn₂O₄ modified by Pt particles might be a potential candidate for gas sensing
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28 applications. First, the sensors based on CdIn₂O₄ modified by different amount of Pt
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30 were performed at different operating temperatures to 100 ppm of acetone. The
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32 optimal operating temperature seems to correlate with the activation energy barrier
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34 related to testing gases [43]. The sensor response as a function of the operating
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36 temperatures from 150 to 420 °C is shown in Fig. 4, which indicates that the response
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38 of the sensors increase and reach maximum at 270 °C and then decrease rapidly with
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40 further increase of operating temperature. From Fig. 4, other important information
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42 could also be obtained. Compared with the pristine CdIn₂O₄, the gas sensing
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44 performance of Pt/CdIn₂O₄ to acetone increases with the increment of Pt amount, up
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46 to 0.98 at.%, and decreases with further increment of Pt amount. CdIn₂O₄ modified by
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48 1.75 at.% Pt leads to poor responses maybe because too much Pt particles blott out the
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4 active sites of CdIn_2O_4 and cause a decrease of the adsorbed oxygen and finally a
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6 poor response of acetone.
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9 A comparison of the response/recovery performance of the sensors based on
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11 CdIn_2O_4 and $\text{Pt/CdIn}_2\text{O}_4$ as a function of acetone concentrations at the working
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13 temperature of 270 °C is presented in Fig. 5a. It can be seen that the response of
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15 CdIn_2O_4 based sensor gradually increases as the acetone concentrations increase from
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17 10 to 200 ppm, which demonstrates that CdIn_2O_4 based sensor has a wide detection
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19 limit for acetone. The sensor based on CdIn_2O_4 modified by Pt shows much higher
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21 response than the pristine CdIn_2O_4 sensor. The relations between sensor response and
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23 gas concentration are shown in Fig.5b. The linear relationships between the responses
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25 and the gas concentrations suggest that the sensor response exhibits a good
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27 dependence on the gas concentration.
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34 Such good performances can be attributed to the contribution of the gas-sensing
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36 reactivity on crystal facets in different orientations of CdIn_2O_4 and the interface
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38 between CdIn_2O_4 and Pt particles. In what follows, the possible contribution is
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40 discussed in two sections. The first is dealing with the adsorption/desorption
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42 phenomena and reactions at the surface. The second describes the possible chemical
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44 influence of the catalytic activity of Pt particles in the region close to the interface.
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49 CdIn_2O_4 is an n-type semiconductor. According to Wolkenstein's model for
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51 semiconductor [44, 45], exposure of the CdIn_2O_4 sensor to air results in O_2 adsorption
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53 on the CdIn_2O_4 surface, which traps electrons from the conduction band of
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55 semiconductor to form oxygen species (O_2^- , O^- , O^{2-}). When a kind of reductive gas,
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4 for example, acetone, is introduced, the reductive gas would react with the adsorbed
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6 oxygen species, which leads to a relatively strong activation on the surface of the
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8 CdIn₂O₄ sensor. Therefore, surface accessibility is crucial to maintain the high
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10 response of the crystals. From a crystallographic point of view, CdIn₂O₄ is a structure
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12 of spinel oxides, forming face-centered-cubic (FCC) closed packing. As an FCC
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14 crystal, a general sequence of surface energies may hold, $\gamma\{111\} < \gamma\{100\} < \gamma\{110\}$
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16 [46]. The cubic CdIn₂O₄ may have a larger active surface area of {100} facets than
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18 the octahedral, which can provide more active space for the interaction between
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20 CdIn₂O₄ and the detected gases and thus demonstrates a higher response.
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26 The chemical effects of the Pt/CdIn₂O₄ interface are related to the catalytic
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28 nature of Pt. By the assistance of Pt, oxygen molecules can be more easily adsorbed
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30 on the surface of CdIn₂O₄. Because Pt is a far better oxygen dissociation catalyst than
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32 CdIn₂O₄, adsorbed oxygen can diffuse faster to surface vacancies and capture
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34 electrons from the conduction band of CdIn₂O₄ to become oxygen ions. This process
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36 increases both the quantity of adsorbed oxygen and the molecule-ion conversion rate.
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38 Furthermore, because Pt can break hydrocarbons into more active radicals due to the
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40 catalysis, the production of the reaction between surface adsorbed oxygen ions and
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42 reducing gases is increased. At the regions close to the Pt/CdIn₂O₄ interface, the
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44 electrons are released more easily from the surface reaction back into the conduction
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46 band of CdIn₂O₄, which greatly increases the conductivity of Pt/CdIn₂O₄ in reducing
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48 gas ambience. Thus, these regions close to the Pt/CdIn₂O₄ interface make the sensors
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50 more active in gas detection.
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4. Conclusion

In summary, CdIn_2O_4 with different polyhedral shape modified by Pt particles was synthesized in one step by a convenient and low-temperature hydrothermal process without further heat treatment. The changing trend of CdIn_2O_4 shape changed from regular octahedral to truncated octahedral, 14-faceted polyhedral, and then regular cubic. In addition, gas sensors based on polyhedral CdIn_2O_4 exhibited high response to acetone. The cubic CdIn_2O_4 showed distinctly greater response than the other because of their exposed active facets and the catalytic activity of Pt particles.

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Figure captions

Fig. 1 SEM images and corresponding EDS spectra of CdIn₂O₄ with different weight percent of Pt: (a, a') 0 at.%; (b, b') 0.26 at.%; (c, c') 0.40 at.%; (d, d') 0.98 at.%; (e, e') 1.75 at.%.

Fig. 2 XRD patterns of the CdIn₂O₄ and Pt/CdIn₂O₄ with different weight percent of Pt: (a) 0 at.%; (b) 0.26 at.%; (c) 0.40 at.%; (d) 0.98 at.% (e) 1.75 at.%.

Fig. 3 TEM images (a, b), HRTEM image (c), corresponding electron diffraction patterns (d) of CdIn₂O₄ modified by 0.98 at.% Pt.

Fig.4 The responses of Pt/CdIn₂O₄ with various Pt contents to 100 ppm of acetone at different operating temperature.

Fig.5 (a) Response and recovery curves of Pt/CdIn₂O₄ with various Pt contents to acetone with increasing concentrations at operating temperature of 270 °C; (b) Calibration curves of the sensor to acetone.

Table 1 The pH values of the system with different amount of Pt precursor before and after the hydrothermal experiments.

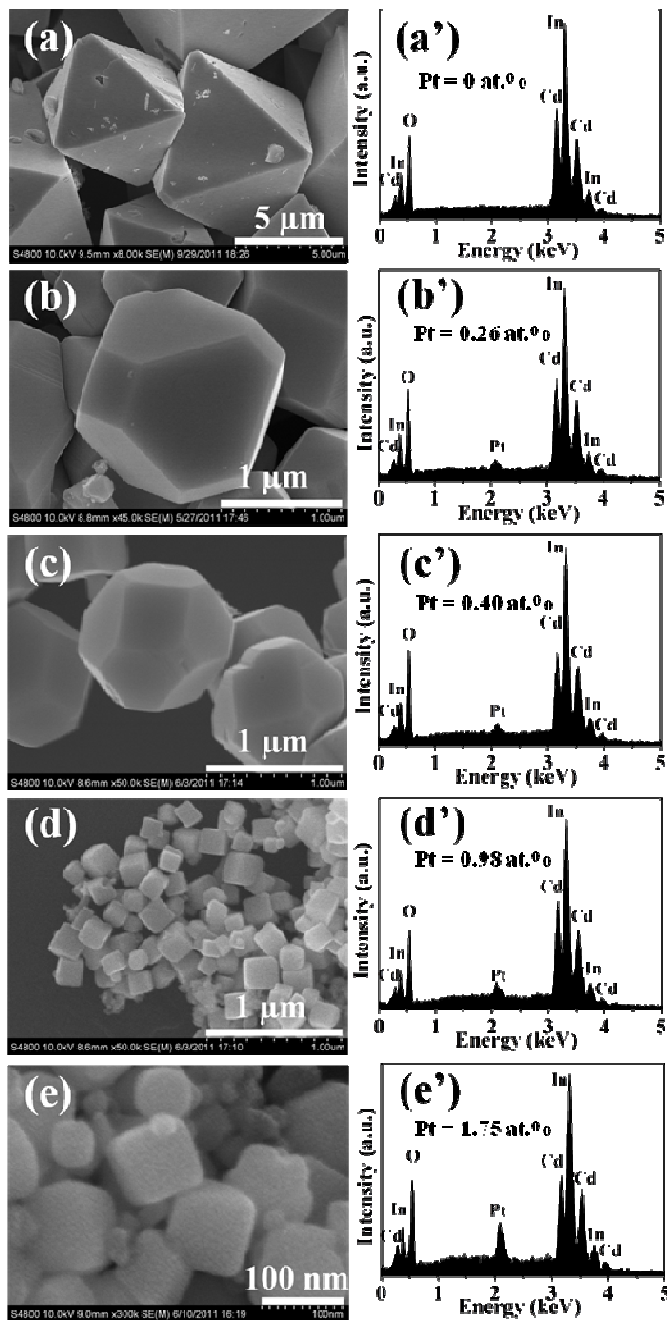


Fig. 1

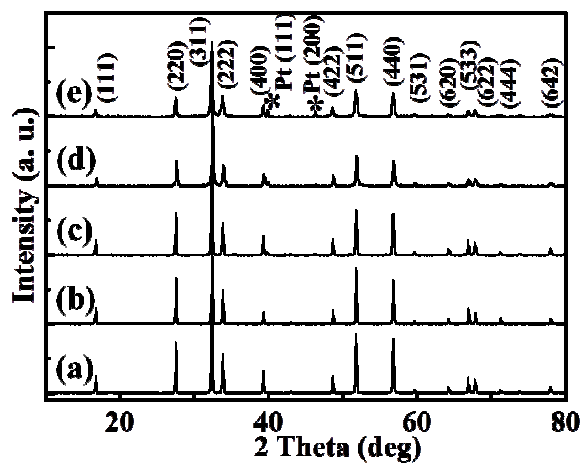


Fig. 2

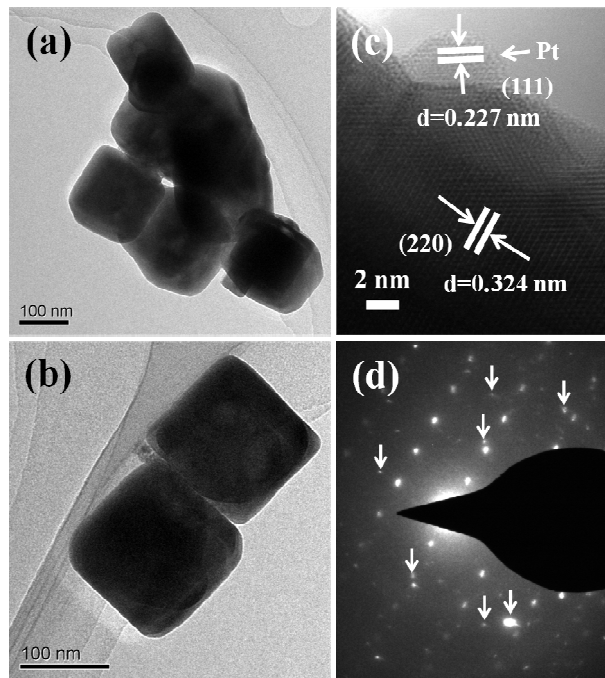


Fig. 3

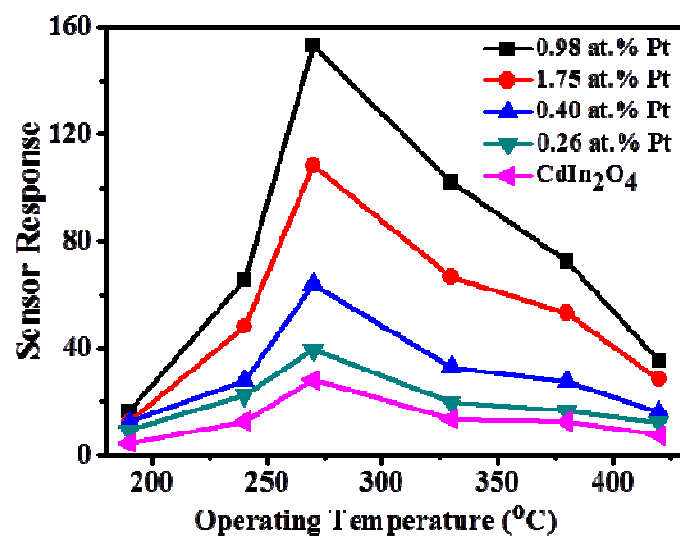


Fig. 4

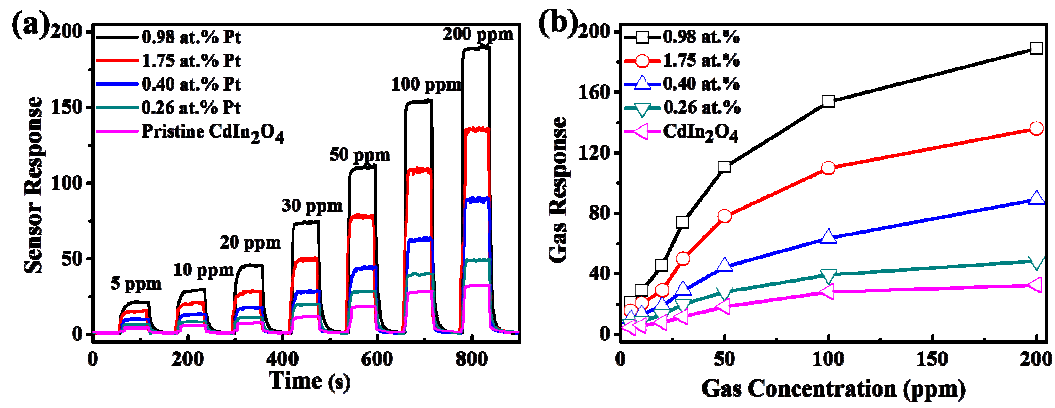


Fig. 5

Table 1

Pt Contents	pH Value of the System Before the Hydrothermal Experiments	pH Value of the System after the Hydrothermal Experiments
0 at. %	12.46	9.23
0.26 at. %	11.76	8.36
0.40 at. %	10.35	8.18
0.98 at. %	9.63	7.36
1.75 at. %	9.21	7.14