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# Electronic and Transport Properties of Porous Graphenes: Two-Dimensional Benzo- and Aza-fused π-Conjugated-Microporous-Polymers Sheets and Boron-Nitrogen Co-doped Derivatives<sup>†</sup>

Zhao-Di Yang <sup>a,c</sup> Wenzhi Wu <sup>b,c</sup> and Xiao Cheng Zeng<sup>c</sup>\*

<sup>a</sup>Key Laboratory of Green Chemical Engineering and Technology of College of Heilongjiang Province, College of Chemical and Environmental Engineering, Harbin University of Science and Technology, Harbin 150040, China

<sup>b</sup> State Key Laboratory of Mechanics and Control of Mechanical Structures and Key Laboratory for Intelligent Nano Materials and Devices of Ministry of Education University of Aeronautics and Astronautics, Nanjing 210016, China

<sup>c</sup>Department of Chemistry, University of Nebraska-Lincoln, Lincoln, Nebraska, 68588, USA \*E-mail: xzeng1@unl.edu

† Electronic Supplementary Information (ESI) available. Computed transmission spectra and the corresponding DOS of left electrode and right electrode at different bias for (a) Benzo-CMP, (b) Aza-CMP, (c) *cis*-BN-Benzo-CMP, (d) *trans*-BN-Benzo-CMP, and (e) *cis*-BN-Aza-CMP are provided.

**Abstract** Inspired by recent successful synthesis of porous frameworks based on aza-fused conjugated-microporous-polymer (*Angew. Chem. Int. Ed.*, 2011, **50**, 8753-8757), we investigate electronic and electron transport properties of two-dimensional (2D) Benzo-conjugated-microporous-polymer (Benzo-CMP) and Aza-CMP sheets and their boron-nitrogen (BN) co-doped derivatives using density-functional theory calculations and the non-equilibrium Green's function (NEGF) method. Both Benzo-CMP and Aza-CMP possess pore diameters about 12 Å and both are semiconductors with a direct bandgap of 0.92 and 1.07 eV, respectively. The BN co-doped derivatives possess smaller band gaps (0.47 to 0.6 eV). Computed current-voltage (*I-V<sub>b</sub>*) curves are consistent with the semiconducting properties for all 2D systems. The tunable

electronic and electron transport properties via BN co-doping offered by Benzo-CMP and Aza-CMP may be exploited for applications in low-dimensional electronics.

# Introduction

Two-dimensional (2D) materials such as graphene monolayer<sup>1-5</sup> have attracted tremendous interests due to their unique electronic properties and potential applications in future electronic devices. However, for graphene, the zero bandgap is a major issue that hampers its practical application in nanoelectronics.<sup>6</sup> One way to open the bandgap is to tailor the 2D graphene to 1D graphene nanoribbons (GNRs). As such, one can further modify GNRs' band structures, transport and optical properties through either doping or introducing defect and chemical functionalization.<sup>7-10</sup> Another way to open the bandgap is to utilize the *porous graphene*. Bieri *et al.* have successfully synthesized 2D polyphenylene, <sup>11</sup> a prototype of porous graphene with a direct bandgap in the range of 2.3-3.2 eV. <sup>12-15</sup> In addition, the  $\pi$ -conjugated microporous polymers (CMPs) are a class of porous frameworks consisting of an extended  $\pi$ -conjugated system and inherent nanopores. The amorphous powders of CMPs have been investigated for possible applications as gas-storage agents, light-harvesting materials, catalysts, capacitors, and batteries.<sup>16-20</sup> A 2D sheet of the CMPs can be categorized as "porous graphenes".<sup>21</sup> Recently Jiang *et al.* synthesized porous frameworks based on aza-fused CMPs with pore size of 1.4 nm. They demonstrated novel functions of aza-fused CMPs for potential applications in supercapacitor energy storage and electric power supply.<sup>22</sup> The structure of the 2D sheet of aza-fused CMPs, denoted as Aza-CMP in this study, is essentially a honeycomb network which can be also viewed as an N-doped porous graphene (Fig. 1(b)). If all of the N atoms in Aza-CMP are replaced by C atoms, it would resemble the structure of a graphene but with missing some

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phenyl rings in a periodic fashion, named as Benzo-CMP in this study (Fig. 1(a)). This structure can be also viewed as five-fused-benzenes replacing each double bond in the graphene. Our *ab initio* calculations show that like 2D polyphenylene, these two newly designed porous graphenes can possess sizable bandgap and nanometer-scale pore size.

Note that the hexagonal boron nitride (*h*-BN) monolayer has a 2D structure similar to the graphene and their lattice constant difference is only 2%. The *h*-BN monolayer is an insulator with a wide bandgap of ~5.9 eV. Recently, planar graphene/*h*-BN heterostructures or hybrid graphene/*h*-BN domains have been fabricated as atomically thin integrated circuitry for electronic and optical applications.<sup>23-25</sup> Our previous theoretical studies suggest that hybrid graphene/boron nitride nanoribbons can exhibit half-metallicity.<sup>26,27</sup> Therefore, either BN co-doping or graphene/*h*-BN hybridization can be utilized to tune the electronic and transport properties of pure graphene and GNRs. <sup>23-27</sup> In this article, we present computed electronic and transport properties of several porous graphene heterostructures, including 2D Benzo-CMP and Aza-CMP sheets as well as their BN co-doped derivatives (i.e., replacing C<sub>1</sub> atoms in hexagonal rings by hexagonal BN rings; see Fig. 1).

### **Computational Details**

Geometry optimizations and electronic structure calculations are based on density functional theory (DFT) methods implemented in the SIESTA package.<sup>28</sup> The generalized gradient approximation (GGA) in the Perdue-Burke-Ernzerhof (PBE) form is used to treat the exchange-correlation potential <sup>29</sup> and the double- $\zeta$  polarized numerical atomic-orbital basis sets are selected for the DFT calculations. A real-space grid with an equivalent energy cutoff of 300 Ry is

adopted to expand the electron density for numerical integration. The conjugate-gradient method is used to optimize the geometry and all the atoms in the unit cell are fully relaxed until the force on each atom is less than 0.01 eV/Å. A vacuum layer larger than 10 Å is used to avoid interaction among periodic images. The Brillouin zone is sampled at  $50 \times 50 \times 1$  Monkhorst meshes. The electron transport properties are computed using the non-equilibrium Green's function (NEGF) method based on the Keldysh formalism, as implemented in the TRANSIESTA program. <sup>30</sup> The left and right electrodes are chosen to be the same as the scattering region. The current through the system is calculated according to the Landauer–Bu $\square$ ttiker formula<sup>31</sup>

$$I = G_0 \int T(E, V_b) [f_L(E) - f_R(E)] dE$$

where  $G_0 = 2e^2/h$  is the unit of quantum conductance,  $T(E, V_b)$  is the transmission coefficient at energy *E* and the bias voltage  $V_b$ , and  $f_{R(L)}(E)$  are the Fermi distribution functions at right (left) electrode. The single- $\zeta$  plus polarization (SZP) basis set is selected to lower high computing cost associated with the computation of transport properties. Previously, Brandbyge *et. al* showed that by changing from SZP to DZP no significant improvement of the transmission function was seen.<sup>30</sup> Moreover, a comparison of a plane-wave calculation of a transmission function with an SZP calculation shows no difference even in the tunneling regime within the spatial extension of the SZP functions.<sup>32</sup> In addition, we have examined the transmission function using double- $\zeta$  plus polarization (DZP) basis set calculations and found nearly the same results for electron transmission.

## **Results and Discussions**

Geometrical Structures and Electronic Properties of Benzo-CMP and Aza-CMP

We first examine the atomic structures of 2D Benzo-CMP and Aza-CMP, as shown in Fig. 1(a) and 1(b). Compared with the graphene whose unit cell contains two carbon atoms, the unit cell of 2D Benzo-CMP can be constructed by replacing each double bond in the graphene with five fused-benzenes to form planar graphene-like honeycomb structure, resulting in a wider pore diameter (12.22Å) compared to that of the 2D polyphenylene (2.48Å). <sup>14</sup> The optimized lattice parameter is 17.27 Å. As shown in Fig. 1(a), there are two different carbon sites in Benzo-CMP: each C<sub>1</sub> atom is bonded with three carbon atoms and each C<sub>2</sub> atom is bonded with two carbon atoms and one hydrogen atom. The bond length of C<sub>1</sub>-C<sub>1</sub>(1.470Å) is longer than that of C<sub>1</sub>-C<sub>2</sub> (1.400Å). Obviously, Aza-CMP can be constructed by substituting some C<sub>2</sub> by N atoms to form N-doped porous graphene, as shown in Fig. 1(b). The optimized lattice constant is 16.59 Å, which is smaller than that of Benzo-CMP. The calculated diameter of the pores is 11.54 Å, within the range of experimental values (0.9 – 1.65 nm).<sup>22</sup> Replacing C<sub>2</sub> by N makes C<sub>1</sub>-N (1.327Å) bond shorter than C<sub>1</sub>-C<sub>2</sub> bond in Benzo-CMP. So the pore diameter of Aza-CMP is slightly smaller.

Next, we compute the electronic properties of Benzo-CMP and Aza-CMP. The electronic band structure of Benzo-CMP is plotted in Fig. 2(a), together with the density of states (DOS). Contrary to the graphene, Benzo-CMP is a semiconductor with a direct bandgap of 0.92 eV at the  $\Gamma$  point. This bandgap is less than the bandgap of polyphenylene network (~2.35 eV).<sup>13-15</sup> Note however that since the PBE/DFT calculation tends to underestimate the energy gap of semiconductors, the realistic bandgap of Benzo-CMP may be somewhat larger than 0.92 eV. A partial density of states (PDOS) analysis shows that both C<sub>1</sub> and C<sub>2</sub> atoms in Benzo-CMP contribute to the valence band maximum (VBM) and the conduction band minimum (CBM), and the contribution from C<sub>2</sub> atoms is more pronounced than that from C<sub>1</sub> atoms. The wave functions

corresponding to the VBM and CBM states are plotted in the insets of Fig. 2(a). Blue and vellow iso-surfaces represent positive and negative signs of wave function, respectively. It can be seen that the VBM of Benzo-CMP is mainly contributed by the  $p_z$  atomic orbitals of C<sub>2</sub> atoms and the  $sp^2$  orbitals of C<sub>1</sub>–C<sub>2</sub> bonds, whereas the CBM of Benzo-CMP is mainly contributed by the  $p_z$ atomic orbitals of  $C_2$  atoms and the  $\pi$  orbitals of  $C_1$  hexagonal rings. Such highly localized states for Benzo-CMP are very different from the delocalized  $\pi$  and  $\pi^*$  states for the graphene sheet, which lead to semiconducting characteristics for Benzo-CMP. For comparison, the electronic band structures and DOS of Aza-CMP are shown in Fig. 2(b). Clearly, the Aza-CMP is also a semiconductor with a direct bandgap of 1.07 eV, which is slightly larger than that of Benzo-CMP (0.92 eV). Because the number of valence electrons of the nitrogen atom is one more than that of the carbon atom, the N substitution changes the original band filling, shifts the Fermi level upward to the valence band, and induces new electronic states within the valence bands (indicated by red arrows in Fig. 2(b)). Again, to obtain more detailed information about the characteristic semiconducting bands, the amplitude of the wave functions of VBM and CBM are plotted in Fig. 2(b). Clearly, the states corresponding to CBM of Aza-CMP are mainly contributed by the  $p_z$  atomic orbitals of C<sub>2</sub> and N atoms and the  $\pi$  orbitals of C<sub>1</sub> hexagonal rings, whereas the states of VBM of Aza-CMP are mainly contributed by the  $sp^2$  orbitals of C<sub>1</sub>-N and C<sub>1</sub>-C<sub>2</sub> bonds, suggesting that the impurity states around the Fermi level due to N atoms are strongly hybridized with the states of C atoms.

#### Geometrical Structures and Electronic Properties of BN co-doped derivatives

To explore the effect of BN doping on the structures and electronic properties of both Benzo-CMP and Aza-CMP, we examine several BN co-doped derivatives of Benzo-CMP and Aza-CMP. Here we use two hexagonal BN rings to replace  $C_1$  hexagonal rings in the unit cell. Due to Journal of Materials Chemistry C Accepted Manuscript

different positions for B and N atoms on the two *h*-BN rings, four derivatives of the Benzo-CMP and Aza-CMP, namely, *cis*-BN-Benzo-CMP, *cis*-BN-Aza-CMP, *trans*-BN-Benzo-CMP and *trans*-BN-Aza-CMP, can be constructed, as shown in Fig. 1(c)-(f), respectively. Geometry optimizations show that their unit-cell lattice constant is 17.33 Å, 16.68 Å, 17.36 Å and 16.70 Å, respectively, slightly larger than the corresponding lattice constant of Benzo-CMP (17.27Å) or Aza-CMP (16.59Å) because B-N bond length (1.499 or 1.481Å) in the *h*-BN rings is longer than  $C_1$ - $C_1$  bond (1.470Å) in  $C_1$  hexagonal rings. Comparing to the *cis*-BN co-doped derivatives with the *trans*-BN co-doped derivatives, hexagonal pores of the *cis*-BN-Benzo-CMP and *cis*-BN-Aza-CMP appear to be distorted. Electronegativity of C is lower than that of N and greater than B, so the bond length of B- $C_2$  is longer than that of N- $C_2$ . Indeed, the bond length of B- $C_2$  and N- $C_2$  in *cis*-BN-Benzo-CMP is 1.492 and 1.352Å, respectively. When two *h*-BN rings in the unit cell are arranged according to *cis*-position, the difference in bond length results in distorted hexagonal pores. The same situation occurs in *cis*-BN-Aza-CMP.

The computed electronic band structures of the four different BN co-doped derivatives are shown in Fig. 3(a)-(d), respectively. It is obvious that BN co-doping plays an important role in tuning the bandgap of Benzo-CMP and Aza-CMP. They all display distinct semiconducting behaviors. Compared to Benzo-CMP and Aza-CMP, all four derivatives possess smaller bandgaps. In details, *trans*-BN-Benzo-CMP has a direct bandgap of 0.57 eV, nearly the same as that of *cis*-BN-Benzo-CMP (0.58 eV), while *trans*-BN-Aza-CMP has the smallest bandgap with the value of 0.47 eV, notably less than that of *cis*-BN-Aza-CMP (0.6 eV).

To clarify the influence of BN co-doping, the total density of states (TDOS) and partial density of states (PDOS) of the four derivatives are shown in Fig. 3(a)-(d), respectively. For *trans*-BN-Benzo-CMP and *cis*-BN-Benzo-CMP, there is a peak at the bottom of conduction band in PDOS

of N and C atoms and a peak at the top of valence band in PDOS of B and C atoms, which means that the N atoms mainly contribute to the bottom of conduction band and B atoms mainly contribute to the top of valence band, consistent with the wave function distribution of VBM and CBM shown in the insets of Fig. 3(a) and (b), respectively. Here one can see that the VBM is mainly contributed by the  $p_z$  atomic orbitals of C<sub>2</sub> atoms and the  $sp^2$  orbitals of B-C<sub>2</sub> bonds, whereas the CBM is mainly contributed by the  $p_z$  atomic orbitals of  $C_2$  atoms and the delocalized  $\pi$  orbitals of N atoms. It is known that the 2s- and the 2p-electronic orbital energies of B atom are higher than those of C atom but lower than those of N atoms. Therefore, the doping of B atoms will move the corresponding energy bands upward and the doping of N atoms will move the energy bands downward. As the VBM is mainly contributed by B atoms and the CBM is contributed by N atoms for *trans*-BN-Benzo-CMP and *cis*-BN-Benzo-CMP, as a result, the bottom of conduction band moves downward and the top of valence band moves upward, resulting in reduced bandgap in comparison with Benzo-CMP. h-BN co-doping shows similar influence for trans-BN-Aza-CMP and cis-BN-Aza-CMP. As shown in the total DOS and PDOS in Fig. 3(c) and (d), N atoms mainly contribute to the bottom of conduction band and both B and N atoms contribute to the top of valence band, consistent with the wave function distribution of VBM and CBM shown in the insets of Fig. 3(c) and (d). From the wave function distribution of VBM and CBM one can find that B atom contributions on h-BN rings to the top of valence band are larger than the corresponding N atoms contributions. Thus the influence of the B atoms shifts up the VBM and N atoms shifts down the CBM, which results in reduced bandgap of *trans*-BN-Aza-CMP and *cis*-BN-Aza-CMP, compared to Aza-CMP.

#### **Transport Properties**

In addition to the electronic structures, we compute the electron transport properties and currentvoltage ( $I-V_b$ ) characteristics of the Benzo-CMP and Aza-CMP and their BN co-doped derivatives. A two-probe system is employed to compute the  $I-V_b$  curve. Here, we treat the *trans*-BN-Aza-CMP as a prototype model for the discussion. As shown in Fig. 4(a), the semiinfinite left- and right-electrodes are in contact with a finite central scattering region with the same width. Both electrodes and the central region are made of the 2D *trans*-BN-Aza-CMP sheet. The transmissions are in the *z* axis direction. Periodic boundary conditions are imposed on the plane normal to the *z* direction. The computed zero-bias transmission spectrum is shown in Fig. 4(b), which apparently mimics the band structure of the *trans*-BN-Aza-CMP sheet. There is a region of zero transmission with a width of 0.49 eV located around the Fermi level, coinciding with the computed band gap (0.47 eV) of the *trans*-BN-Aza-CMP sheet. The current versus bias voltage curve is shown in Fig. 4(c), which displays typical semiconducting behavior, consistent with the zero-bias transmission spectrum shown in Fig. 4(b). The system exhibits a threshold voltage  $V_{th} = 0.6$  V below which the current is nearly to be zero.

To further understand the semiconducting  $I-V_b$  characteristic of the 2D *trans*-BN-Aza-CMP sheet, the transmission spectra at two typical bias voltages are presented in Fig. 5(a). It is known that the current is mainly attributed to the transmission coefficient around the Fermi level within the bias window. At zero bias, there is a transmission gap of 0.49 eV due to the bandgap of *trans*-BN-Aza-CMP. When the bias is increased to  $V_{th} = 0.6$  V, the valence band of the left electrode matches well with the bottom of the conduction band of the right electrode, as shown in Fig. 5(b). Consequently, the electrons can tunnel from the valence band of the left electrode to the conduction band of the right one, resulting in a transmission peak around the Fermi level

within the bias window. Based on the Landauer–Bu $\Box$ ttiker formula, the current should flow. Compared with the case of 0.6 V bias, when the bias window widens from [-0.3, 0.3] V to [-0.5, 0.5] V under the bias of 1.0 V, the transmission coefficient becomes larger, leading to a notable increase in the current.

For comparison, the I-V<sub>b</sub> curves of Benzo-CMP, Aza-CMP, cis-BN-Benzo-CMP, trans-BN-Benzo-CMP and *cis*-BN-Aza-CMP at zero bias are presented in Fig. 4(c). As mentioned above, the Benzo-CMP and Aza-CMP are semiconductors with a direct bandgap of 0.92 and 1.07 eV, respectively. So, the current cannot be observed until the bias voltage exceeds 1.0 V for Benzo-CMP. But the current does not arise even when the bias voltage reaches 1.2 V for Aza-CMP. On the other hand, due to the smaller bandgaps for all the BN co-doped derivatives, smaller transmission gaps at zero bias voltage, i.e., 0.58, 0.58, 0.63 and 0.49 eV, are found for *cis*-BN-Benzo-CMP, trans-BN-Benzo-CMP, cis-BN-Aza-CMP and trans-BN-Aza-CMP, respectively, consistent with their corresponding band structures. When the bias is greater than  $V_{th}$ , the current arises. More specifically, obvious current is seen when the bias is greater than 0.6 V for *trans*-BN-Benzo-CMP. Moreover, the transmission coefficient increases with increasing the bias voltage in the range of 0.6-1.0 V, which leads to the monotonic current increase in this bias window, similar to trans-BN-Aza-CMP. For cis-BN-Benzo-CMP and cis-BN-Aza-CMP, the current also starts to arise when the bias exceeds 0.6 V. However, the current becomes constant (~1.0  $\mu$ A) from 0.6 to 1.0 V, consistent with the transmission spectra shown in Fig. S1 in the Electronic Supplementary Information (ESI).

## Conclusion

In conclusion, we have shown that 2D Benzo-CMP and Aza-CMP sheets and their BN codoped derivatives possess distinct structural, electronic and transport properties that are notably different from those of the graphene. Our DFT computations show that the Benzo-CMP and Aza-CMP, both being porous graphenes with relatively large pore size, are semiconductors with a direct bandgap of 0.92 and 1.07 eV, respectively. The computed  $I-V_b$  curves are consistent with of the semiconducting characteristics. BN co-doped derivatives possess relatively small bandgaps (0.47 - 0.6 eV). The tunable electronic and transport properties of Benzo-CMP and Aza-CMP through BN co-doped derivatives may be exploited for applications in lowdimensional nanoelectronic devices.

#### Acknowledgements

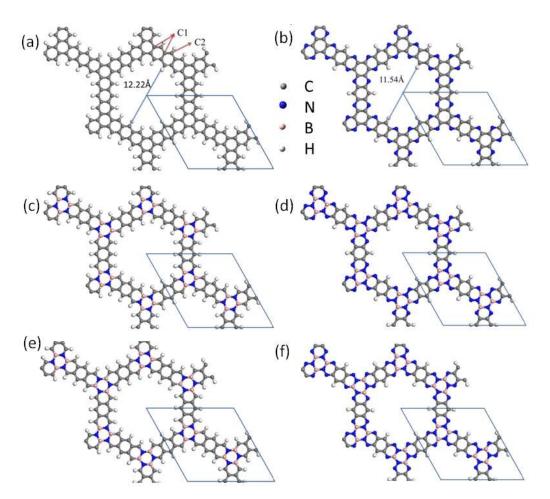
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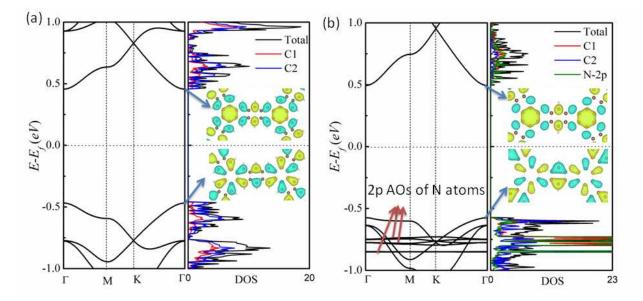
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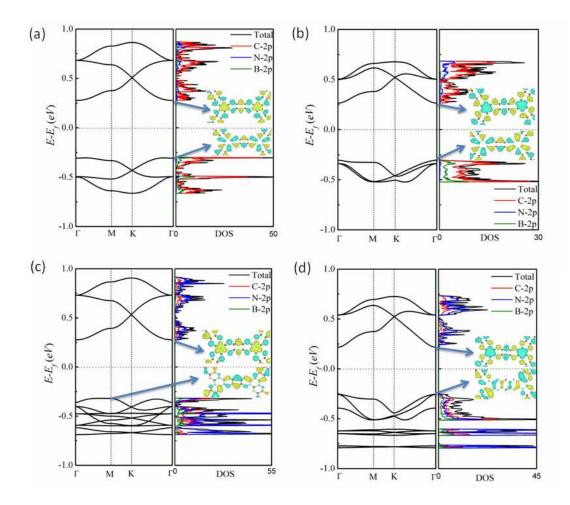
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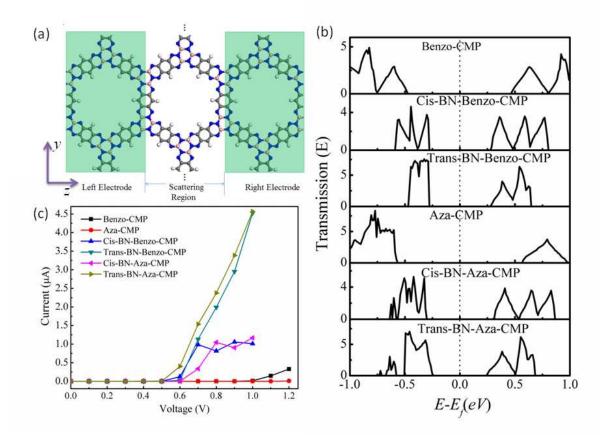
**Figure 1** Atomic structures of 2D (a) Benzo-CMP, (b) Aza-CMP, (c) *cis*-BN-Benzo-CMP, (d) *cis*-BN-Aza-CMP, (e) *trans*-BN-Benzo-CMP and (f) *trans*-BN-Aza-CMP sheets in  $2 \times 2$  supercell. The rhombus in (a) – (f) represents the unit cell. The gray, white, blue and pink balls denote C, H, N and B atoms, respectively.



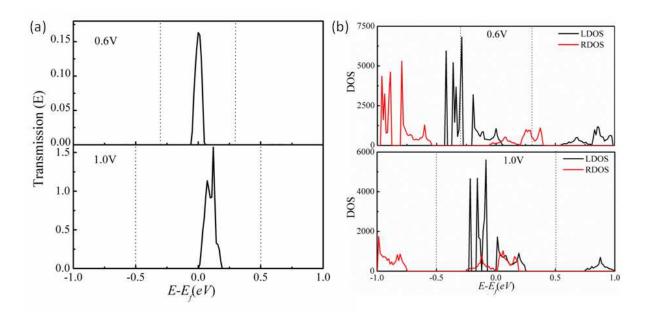
**Figure 2** Computed band structures (left panel) and total density of states and partial density of states (right panel) of (a) Benzo-CMP and (b) Aza-CMP in their primitive unit cell. The wave functions of the lowest conduction band and the highest valence band are plotted in the insets, where the isosurface value is  $0.05 \text{ e} \text{ Å}^{-3}$ .



**Figure 3** Computed band structures and total density of states and partial density of states of BN co-doped derivatives: (a) *cis*-BN-Benzo-CMP, (b) *trans*-BN-Benzo-CMP, (c) *cis*-BN-Aza-CMP and (d) *trans*-BN-Aza-CMP. The wave functions of the lowest conduction band and the highest valence band are plotted in the insets, where the isosurface value is  $0.05 \text{ e} \text{ Å}^{-3}$ .



**Figure 4.** (a) Schematic view of the two-probe *trans*-BN-Aza-CMP device. (b) Computed transmission spectra of Benzo-CMP, Aza-CMP, *cis*-BN-Benzo-CMP, *trans*-BN-Benzo-CMP, *cis*-BN-Aza-CMP and *trans*-BN-Aza-CMP (from bottom to top) under zero bias voltage, respectively. (c) Variation of current as a function of the bias voltage ( $I-V_b$  curves).



**Figure 5** (a) Computed transmission spectra of *trans*-BN-Aza-CMP for 0.6 and 1.0 V bias voltages (from bottom to top), respectively. The dashed lines indicate the bias windows. (b) Density of states of the left (LDOS) and right (RDOS) electrodes under the 0.6 and 1.0 V bias voltages for the two-probe *trans*-BN-Aza-CMP device.

# **TOC graphics**

