Journal of Materials Chemistry A

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Cite this: DOI: 10.1039/c0xx00000x

ARTICLE TYPE

An Expanded Isoindigo Unit as a New Building Block for Conjugated Polymer Leading to High-Performance Solar Cells

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Received (in XXX, XXX) Xth XXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX

5 DOI: 10.1039/b000000x

Design and synthesis of novel monomers may create new opportunities for high performance semiconducting polymers. The synthetic modification of isoindigo related building block based on rational design has been carried out in this study. An expanded isoindigo unit (IBTI) with 5,5'-divinyl-2,2'-bithiophene as conjugated bridge has been successfully incorporated into donor/acceptor conjugated polymer, **PBDT-IBTI**. The new polymer showed broad absorption and large extinction coefficient in visible region. The optical bandgen HOMO and LUMO lawels of this new polymer are $1.71 \text{ eV}_{\odot} = 5.24 \text{ eV}$ and 2.72 eV remeatively. The new more also also as a structure of the second second

¹⁰ bandgap, HOMO and LUMO levels of this new polymer are 1.71 eV, -5.24 eV and -3.72 eV, respectively. The polymer solar cells based on **PBDT-IBTI** and (6,6)-phenyl-C61 butyric acid methyl ester ($PC_{61}BM$) blends achieved a PCE of 6.41%. This work has demonstrated that the IBTI is a promising building block for organic photovoltaic (OPV) polymers. The finding has also revealed the versatility of oxindole as basic building element for isoindigo like conjugated monomer.

Introduction

- ¹⁵ Conjugated polymers have been extensively studied for applications in optoelectronic devices such as organic photovoltaics (OPVs) and organic field-effect transistors (OFETs).¹ Lactam based acceptors such as diketopyrrolopyrole (DPP),² isoindigo (iI),³ and benzodipyrrolidone⁴ have been
- ²⁰ incorporated into many donor/acceptor polymers for OPV and OFET applications. Isoindigo processes a central carbon-carbon double bond which is conjugated with two oxindole rings to form an extended π -conjugated structure. Its importance as building block for conjugated polymers has been evident from many ²⁵ recent studies of high-performance materials for OPV and OFET
- applications.^{3,5-7}

Some recent studies suggested that the structure of isoindigo was slightly twisted because of steric repulsion between the protons on the phenyl rings and the carbonyl oxygens of the oxindoles ³⁰ (Fig. 1).^{3a,3b,6} This twist could reduce donor-acceptor interactions,

- ³⁰ (Fig. 1).^{32,55,6} This twist could reduce donor-acceptor interactions, effective conjugation and could adversely affect the inter-chain stacking. To avoid the steric repulsion, several groups synthesized thienoisoindigo (Til) as monomer recently (Fig. 1).⁶ The steric repulsion was relieved in Til. Also, the short oxygen-sulfur
- ³⁵ distance in Til can result in favorable Coulombic interactions between the oxygen atoms of carbonyls and sulfur atoms of thiophenes. This no-bonding interaction can enhance the coplanarity of the monomer.^{6c} Many conjugated polymers based on Tils have been synthesized. The highest hole mobility of 0.28
- ⁴⁰ cm²V⁻¹s⁻¹ and the highest electron mobility of 0.03 cm²V⁻¹s⁻¹ were achieved in ambipolar OFET devices based on those polymers.^{6b} The OPV device study on Til based polymer was also reported and the power conversion efficiency (PCE) of 4.27% was achieved.^{6c} So far, the most of Til based conjugated polymers
- ⁴⁵ showed absorption expanded into near infrared region and narrow bandgap.



Fig. 1 Structures of isoindigo (iI), thienoisoindigo (TiI), thienopyrroledione (TPD dimer), and IBTI.

⁵⁰ However, they showed very high HOMO energy level. The high HOMO energy levels resulted in very low open circuit voltage $(V_{\rm oc})$ (about 0.5 V) in bulk heterojunction (BHJ) polymer solar cell devices.^{6c}

Herein, we report a novel polymer (**PBDT-IBTI**) based on an ⁵⁵ expanded isoindigo unit (IBTI) and benzo[1,2-*b*:4,5*b*']dithiophene (BDT) as repeating units. Instead of a double bond as bridge in isoindigo, IBTI unit used 5,5'-divinyl-2,2'bithiophene as conjugated bridge to connect two oxindole rings (Fig. 1). So, the conjugated core was expended. The steric ⁶⁰ repulsion between the protons on the phenyl rings and the carbonyl oxygens of the oxindoles can be avoided in the new monomer. The four carbon atoms, oxygen and sulfur atom positioned in space to form a spatial relation which resembled a six-member ring structure. The short oxygen-sulfur distance in ⁶⁵ IBTI can also result in favorable Coulombic interactions between

the carbonyl oxygen atoms of oxindoles and sulfur atoms of thiophenes. Similar interactions in bisthienopyrroledione (TPDdimer) and DPP were studied by DFT calculation and were confirmed by X-ray crystal structure (Fig. 1).⁸ The non-bonding ⁷⁰ interactions were considered important for co-planarity and conjugation.

Bis-oxindole with 5,5'-divinyl-2,2'-bithiophene as bridge has been studied for small molecule OFET application by Liu.⁹ However, such structure has never been incorporated into donor/acceptor conjugated polymer for OPV study. The compound reported by Liu was functionalized at 5,5' position. Recent study showed that the substitution pattern played a signification role on the photo-physical properties of molecular the state of the photo-physical properties of the photo-physical photo-photo-physical photo-physical photo-physical photo-physical photo-physical photo-photo-physical photo-photo-photo-photo-physical photo-p

- s isoindigo compound.¹⁰ The 6,6' substituted isoindigo derivatives showed more extended π -conjugation than the 5,5' substituted isoindigo derivatives because of the large contribution of electronic transition dipole moment in the axis aligned with the 6,6' carbons.^{3a,10a} We chose to synthesize the bridged bis-oxindole
- ¹⁰ building block (IBTI) with bromides positioned at 6,6° carbon for achieving extended π -conjugation in targeted polymer. The PSC device studies on polymer **PBDT-IBTI** showed a high PCE of 6.41% and a high FF of 71%.

Experimental Section

15 Materials and methods

All the chemicals were purchased from Sigma-Aldrich Chemical Co. and Sinopharm Chemical Reagent Co. Toluene was freshly distilled over sodium wire under a nitrogen atmosphere prior to use. Flash chromatography was carried out on silica gel (200-300

- ²⁰ mesh). Nuclear magnetic resonance (NMR) spectra were recorded on a Mercury plus 400 MHz instrument. Gel permeation chromatography (GPC) analyses were carried out on a Shimadzu SIL-20A liquid chromatography instrument using tetrahydrofuran as an eluent with polystyrenes as standards. Thermogravimetric
- ²⁵ analyses (TGAs) were carried out on a TA instrument Q5000IR at a heating rate of 20 °C min⁻¹ under nitrogen gas flow. Differential scanning calorimetry (DSC) studies were carried out with a Perkin Elmer Pyris 1 under nitrogen flow. Sample (about 5.0 mg in weight) was first heated up to 300°C and held for 2 min
- ³⁰ to remove thermal history, followed by cooling at a rate of 10 °C min⁻¹ to 20 °C and then heating at a rate of 10 °C min⁻¹ to 300 °C. UV-vis spectra were recorded on a Perkin Elmer Lambda 20 UV-vis spectrophotometer. The absorption coefficient of polymer in chloroform solution was calculated according to the Beer-
- ³⁵ Lambert law. Cyclic voltammetry (CV) measurements were conducted on a CHI 600 electrochemical analyzer with a threeelectrode cell under a nitrogen atmosphere in a deoxygenated anhydrous acetonitrile solution of tetra-*n*butylammoniumhexafluorophosphate (0.1 M) at a scan rate of 50
- $_{40}$ mV s⁻¹. A platinum disk electrode was used as a working electrode, a platinum wire was used as a counter electrode, and an Ag/Ag⁺ (0.01 M AgNO₃ in acetonitrile) electrode was used as a reference electrode. The polymer films were coated on the surface of the platinum disk electrode. The CV curves were
- ⁴⁵ calibrated with ferrocene/ferrocenium (Fc/Fc⁺) redox couple as an external standard which was measured under the same conditions before and after the measurement of samples. Atomic force microscopy (AFM) images were obtained using a Multi Mode V microscope. Elemental analyses were performed at Shanghai ⁵⁰ institute of organic chemistry, Chinese academy of sciences.
- institute of organic chemistry, chinese academy of sele

Polymer Solar Cell Fabrication and Testing

PSC devices with the conventional structure of ITO/PEDOT:PSS(40 nm)/polymer:PCBM/LiF/Al were fabricated. Patterned ITO glass substrates were cleaned by ultrasonic

55 treatment in aqueous detergent solution, isopropyl alcohol and acetone sequentially. Organic residue was further removed by treating with UV-ozone for 20 min. A thin film of PEDOT:PSS (about 40 nm) were spin-coated on pre-cleaned ITO substrates and the substrates were dried at 150 °C for 10 min. A blend of 60 Polymer and PC₆₁BM was dissolved in a chosen solvent with or without DIO. The solution was filtered through а poly(tetrafluoroethylene) (PTFE) filter (0.45 µm) and was spincoated at 1200 rpm for 40 s on top of the PEDOT:PSS layer. The samples were transferred into an evaporator. Lithium fluoride (0.6 65 nm) and aluminium (100 nm) layers were thermally deposited under a vacuum of 1.0×10^{-6} Torr through a shadow mask (active area 7.25 mm²). The PSC devices with inverted structure of ITO/ZnO/polymer:PC₆₁BM/MoO₃/Ag were fabricated as follows: a ca. 40 nm zinc oxide was obtained by spin-coating sol-gel zinc 70 oxide (ZnO) precursor solution onto the pre-cleaned indium tin oxide (ITO) substrates. The coated substrates were baked at 150°C for 40 min in air and transferred into a nitrogen filled glove box. A chlorobenzene solution of polymer (8 mg mL⁻¹) and PC₆₁BM at deferent weight ratios with or without 1,8-75 diiodooctane (DIO) was spin coated on top of the zinc oxide layer. The samples were transferred into an evaporator and a MoO₃ layer (7 nm) and an Ag layer (80 nm) were thermally deposited under a vacuum of 1.0×10^{-6} Torr. The device area was restricted

The PCEs were measured in a glove box directly after fabrication. The devices for the EQE measurements were encapsulated in the glove box and measured in air. A Newport 94023A solar simulator equipped with a 450 W xenon lamp and an air mass (AM) 1.5G filter was used to generate a simulated AM 1.5G solar spectrum irradiation source. The irradiation intensity was 100 mW cm⁻² calibrated by a Newport standard silicon solar cell 91150. A Newport monochromator 74125 and power meter 1918 with silicon detector 918D were used in EQE measurements.

Procedure for Synthesis of Monomer and Polymer

to 7.25 mm^2 by a shadow mask.

- ⁹⁰ The synthetic routes for monomers and polymers are shown in Scheme 1. (*E*)-6,6'-Dibromo-1,1'-bis(2-octyldodecyloxy)-[3,3'biindolinylidene]-2,2'-dione (il),^{3b} 2,6-bis(trimethyltin)-4,8-di(2ethylhexyl)-benzo[1,2-*b*:4,5-*b*']-dithiophene (BDT),¹¹ 2,2'bithiophene-5,5'-dicarbaldehyde,¹² and 6-bromooxindole (1)^{3b} ⁹⁵ were synthesized according to literature procedures.
- (3Z,3'Z)-3,3'-(2,2'-Bithiophene-5,5'-diylbis(methan-1-yl-1ylidene))bis(6-bromoindolin-2-one) (2)

To a suspension of 6-bromooxindole (4.40 g, 20.0 mmol) and 2,2'-bithiophene-5,5'-dicarbaldehyde (2.22 g, 10.0 mmol) in absolute alcohol (70 mL), piperidine (1.0 mL) was added. The mixture was refluxed for 24 h and a red precipitate was formed. The reaction mixture was cooled to room temperature and was filtered. The solid material was washed with ethanol and was dried under vacuum to give the titled compound (5.31 g, 87.0% yield). Due to its poor solubility, the product was used for next step directly without further characterization.

(3Z,3'Z)-3,3'-(2,2'-Bithiophene-5,5'-diylbis(methan-1-yl-1-ylidene))bis(6-bromo-1-(2-octyldodecyloxy)indolin-2-one) (IBTI)

Anhydrous potassium carbonate (3.85 g, 27.85 mmol) and 2 (3.40 g, 5.57 mmol) were dissolved in *N*,*N*-dimethylformamide (80 mL). The solution was heated at 120 °C for 1 h under nitrogen. 9-(Iodomethyl)nonadecane (6.83 g, 16.71 mmol) was

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added dropwise and the mixture was heated at 120 °C for 24 h. It was cooled to room temperature and was poured into water (300 mL). The mixture was extracted with dichloromethane. The combined organic layer was collected and was dried with s anhydrous sodium sulfate. Solvent was removed under reduced pressure and residue was purified by flash chromatography on

- silica gel with diethyl ether / hexane (1:10) as eluent to give the titled compound IBTI (2.70 g, 41.3% yield) as a deep-red solid. ¹H NMR (400 MHz, CDCl₃, δ): 7.62 (d, J = 4.2 Hz, 2 H), 7.60 (s,
- ¹⁰ 2 H), 7.43 (d, J = 4.2 Hz, 2 H), 7.33 (dd, J = 8.1, 1.6 Hz, 2 H), 7.16 (dd, J = 8.1, 1.5 Hz, 2 H), 6.94 (d, J = 1.5 Hz, 2 H), 3.64 (d, J = 7.5 Hz, 4 H), 1.91 (m, 2 H), 1.41-1.13 (m, 64 H), 0.94-0.79 (m, 12 H). ¹³C NMR (100 MHz, CDCl₃, δ): 166.58, 145.05, 143.03, 138.58, 137.99, 128.11, 126.03, 125.00, 122.99, 122.13,
- 15 120.32, 119.93, 111.98, 44.77, 36.41, 32.13, 31.78, 30.22, 29.87, 29.81, 29.76, 29.57, 29.50, 26.66, 22.90, 14.34. Anal. calcd for $C_{66}H_{94}Br_2N_2O_2S_2$: C 67.67, H 8.09, N 2.39; found: C 67.52, H 8.44, N 2.32.

Polymer PBDT-IBTI

- $_{20}$ IBTI (0.35 g, 0.30 mmol), BDT (0.23 g, 0.30 mmol), Pd₂(dba)₃ (0.0055 g, 0.0060 mmol), P(o-tol)₃ (0.0073 g, 0.0240 mmol) and degassed toluene (10.0 mL) were added to a Schlenk tube. The solution was subjected to three cycles of evacuation and admission of nitrogen and was stirred at 90 °C for 4 h. After
- ²⁵ cooled to room temperature, the mixture was poured into methanol (100 mL) and was stirred for 2 h. A deep blue precipitate was collected by filtration. It was washed with methanol and hexane in a Soxhlet extractor for 24 h each. It was extracted with hot chloroform in an extractor for 24 h. After
- ³⁰ removing solvent, a deep blue solid with metallic luster was collected (0.31g, 70.1% yield). ¹H NMR (400 MHz, CDCl₃, δ): 8.1-6.5 (br, 14H), 4.4-3.5 (br, 8H), 2.0-0.5 (br, 108H). Anal. calcd for (C₉₂H₁₃₀N₂O₄S₄)n: C 75.88, H 9.00, N 1.92; found: C 75.37, H 9.32, N 1.85. GPC (THF): M_n = 43.5 kDa, M_w = 115.8 kDa, ³⁵ PDI = 2.67.

Results and Discussion

Synthesis and Characterization

The key intermediate **2** was prepared by the Knoevenagel reaction of 6-bromooxindole and 2,2'-bithiophene-5,5-⁴⁰ dicarbaldehyde in the presence of piperidine in absolute ethanol (**Scheme 1**). The stereoselectivity of this condensation reaction has been carefully studied by using several different kinds of NMR techniques and a *Z*,*Z* configuration of the product have been firmly established in a recent paper by Liu.⁹ The crude

- ⁴⁵ products 2 showed poor solubility in common solvents such as dichloromethane, acetone and ethyl ether. It was washed with methanol and was used directly in next step. The large branched 2-octyldodecyl groups were chosen as side chains for improving solubility. The compound 2 was *N*-alkylated to give monomer
- ⁵⁰ IBTI. The polymer **PBDT-IBTI** was synthesized by Stille crosscoupling reaction with monomer IBTI and 2,6-bis(trimethyltin)-4,8-di(2-ethylhexyloxy)benzo[1,2-b:4,5-b']-dithiophene (BDT) at 1:1 ratio in the presence of Pd₂(dba)₃ and P(*o*-tol)₃ as catalyst and ligand (**Scheme 1**). The polymer was purified by precipitation in



Scheme 1 Synthesis of monomer and polymer PBDT-IBTI.

methanol and by washing with methanol and hexane in a Soxhlet extractor for 24 h each. The **PBDT-IBTI** was soluble in common organic solvents such as tetrahydrofuran, chloroform (CF) and ⁶⁰ chlorobenzene (CB) at room temperature. The number-average molecular weights (M_n) and polydispersity indexe (PDI) of the polymers were determined by gel permeation chromatography (GPC) using polystyrenes as standards with tetrahydrofuran as eluent. The polymer showed a relatively high M_n of 43.5 kDa and ⁶⁵ a PDI of 2.67.

Thermal properties

Thermogravimetric analyses of the polymer are exhibited in Fig. S4[†]. The polymer **PBDT-IBTI** revealed a decomposition temperature (T_{d} , at 5% weight loss) of 393 °C. However, it did ⁷⁰ not show any noticeable glass transition in differential scanning calorimetry (DSC) analysis in the temperature range from 40 °C to 300 °C (Fig. S5[†]). The results indicated that **PBDT-IBTI** possesses a good thermal stability which is a prerequisite for device application.

75 Optical Properties

IBTI, The absorption spectra of 6,6'-dibromodi(2octyldodecyl)isoindigo (iI), and polymer PBDT-IBTI in CF solution are shown in Fig. 2a. The IBTI and iI monomers had same branched alkyl chains in this study. The spectra of the 80 PBDT-IBTI thin film and the PBDT-IBTI dilute solution are displayed in Fig. 2b. The solution absorption spectrum of iI showed a high-energy band with a maximum at 402 nm (ε = 3.03×10^4 M⁻¹cm⁻¹) and a much weak low-energy band with a maximum at 503 nm. The solution absorption spectrum of 85 monomer IBTI displayed a single band with a maximum at 522 nm ($\varepsilon = 5.91 \times 10^4 \text{ M}^{-1} \text{ cm}^{-1}$) and a shoulder at 552 nm. Due to extended conjugation in IBTI, the IBTI spectrum was significant red-shifted and had much large extinction coefficient at its peak compared with iI spectrum. The solution absorption spectrum of 90 polymer PBDT-IBTI showed broad absorption from 475 to 700 nm with a maximum at 594 nm ($\varepsilon = 1.10 \times 10^5 \text{ M}^{-1} \text{ cm}^{-1}$) and a shoulder at 641 nm. The absorption spectrum of polymer thinfilm was slightly red shifted (12 nm at maximum) compared with that of solution spectrum. The thin-film spectrum showed a 95 maximum at 606 nm with a shoulder at 657 nm. The optical

bandgap was 1.71eV calculated from the onset absorption (725



Fig. 2 UV-vis absorption spectra, (a) 6,6°-dibromodi(2-octyldodecyl)isoindigo (il), monomer IBTI and polymer **PBDT-IBTI** in chloroform solution; (b) 5 polymer **PBDT-IBTI** in chloroform solution and as thin film.

nm) on the thin-film spectrum. The new polymer showed broad absorption in visible region and large extinction coefficient. These are required properties of conjugated polymer for high performance PSC applications.

10 Electrochemical properties

- The cyclic voltammetry (CV) curve of polymer **PBDT-IBTI** film is shown in Fig. 3a. The potentials were referenced to the ferrocene/ferrocenium redox couple (Fc/Fc⁺). The redox potential of Fc/Fc^+ was assumed an absolute energy level of -4.8 eV to
- ¹⁵ vacuum. The redox potential of Fc/Fc^+ was measured under the same condition as polymer samples and was located at 0.09 V related to the Ag/Ag⁺ electrode. The electrochemical potentials were converted to the corresponding energy levels by the following equations: ¹³
- ²⁰ $E_{\text{HOMO}} = -(4.8 E_{1/2, \text{ Fc,Fc+}} + E_{\text{ox,onset}}) = -(4.71 + E_{\text{ox,onset}}) \text{ eV}$ $E_{\text{LUMO}} = -(4.8 - E_{1/2, \text{ Fc,Fc+}} + E_{\text{red,onset}}) = -(4.71 + E_{\text{red,onset}}) \text{ eV}$ The CV curve showed the oxidation onset ($E_{\text{ox,onset}}$) at 0.53 V and the reduction onset ($E_{\text{red,onset}}$) at -0.99 V. The HOMO and LUMO energy level were at -5.24 and -3.72 eV, respectively. The HOMO
- ²⁵ level is comparable to those of the isoindigo and bis(alkoxy)substituted BDT copolymers (**PBDT-ID**).^{5c} It is deeper than those of Til-based polymers (about -5.0 eV).⁶ The relatively deep HOMO level is necessary for achieving high open-circuit voltage (V_{oc}) in OPV devices and also benefits the stability of the
- ³⁰ polymer by avoiding unintentional p-doping.¹⁴ The energy levels of the new polymer resembled the ideal polymer which was described in the design rules for donor polymers for fullerene



Fig. 3 Cyclic voltammograms (a) of **PBDT-IBTI** film in n-Bu₄NPF₆ solution (0.1 mol/L in acetonitrile) at a scan rate of 50 mV/s. Energy diagrams (b) showing theoretically calculated HOMO/LUMO energy levels as well as their topographical representations.

40 based BHJ solar cell devices.15

In order to gain further insights into the structural and electronic properties of polymers **PBDT-IBTI** and **PBDT-ID**, density functional theory (DFT) analysis was carried out at the B3LYP/6-31G** level of theory on model compounds representing the ⁴⁵ repeat units in the corresponding polymers (Fig. 3b). Their molecular orbital energy levels and orbital electron density plots are reported and compared with the experimental data. The theoretical HOMO/LUMO energies were found to be -5.10/-3.12 eV for **PBDT-IBTI**, -5.21/-3.59 eV for **PBDT-ID**, respectively. ⁵⁰ The HOMO wave function was more evenly delocalized over the whole polymer backbone for **PBDT-IBTI** than that for **PBDT-ID**, and the LUMO wave function was more effectively delocalized on the acceptor unit for **PBDT-IBTI** than that for **PBDT-ID**. The HOMO wave function of **PBDT-IBTI** was favorable for ⁵⁵ transmission of the carrier.

Polymer photovoltaic device characteristics

The photovoltaic properties of the new polymer were investigated in PSC devices with a conventional structure of ITO/PEDOT:PSS/polymer:PC₆₁BM/LiF/AI. The devices were ⁶⁰ optimized by varying the weight ratios of the blend. The optimized weight ratio between the polymer and PC₆₁BM was 1:2 (*w/w*). 1,8-Diiodooctane (DIO) was added as the processing additive to optimize the morphology of the active layer.¹⁶ The current density-voltage (*J-V*) curves of devices are presented in ⁶⁵ Fig. S7†. The device with PBDTT-FID:PC₆₁BM (1:2, *w/w*)

Table 1 The photovoltaic performance of the inverted PSCs based on **PBDT-IBTI**: $PC_{61}BM$ (1:2, w/w) (8 mg/ml) with different ratios of DIO in *CB* (v/v).

DIO $(v/v)^a$	$V_{oc}\left(\mathbf{V}\right)$	J_{sc} (mA cm ⁻²)	FF	PCE (%)
0 % DIO	0.77	6.84	0.66	3.49
1 % DIO	0.76	10.81	0.71	5.89
2 % DIO	0.77	11.71	0.71	6.41
3 % DIO	0.76	11.07	0.72	6.09

^a DIO content in chlorobenzene.

as active layer processed with DIO (1%, ν/ν) from CB showed the best performance. It had a V_{oc} of 0.71 V, a J_{sc} of 9.40 mA cm⁻², a 5 FF of 0.66 and a PCE of 3.0%.

The devices with inverted structures were also studied. The devices with a structure of ITO/ZnO/polymer:PC₆₁BM/MoO₃/Ag were fabricated and were tested under simulated illumination of air mass (AM) 1.5G (100 mWcm⁻²). The weight ratios of PBDT-¹⁰ IBTI to PC₆₁BM were varied from 1:1, 1:2 to 1:3. The electrical

- output characteristics were summarized in Table S1[†]. The best performances were resulted from devices with a blend of PBDT-IBTI and PC₆₁BM at 1: 2 weight ratios. A PCE of 3.49 % and a V_{oc} of 0.77 V can be achieved in device based on such blend. The
- ¹⁵ device performances were limited by low short-circuit current density (J_{sc}). Thermal annealing did not bring much improvement on device performances. DIO was introduced to improve the morphology of the active layer. Addition of a small amount of DIO to the solution of polymer and PC₆₁BM before spin-coating ²⁰ significantly improved the solar cell performances.



Fig. 4 (a) *J-V* curves and (b) EQEs for the conventional PSCs based on **PBDT-IBTI**:PC₆₁BM (1:2, w/w) with different ratios of DIO in *CB* (v/v).



Fig. 5 AFM images of active layers processed from **PBDT-IBTI** : $PC_{61}BM$ (1:2, w/w) blend film: height images (a, d) , phase images (b, e), and topography image (c, f), without DIO (upper row) and with 2% DIO (lower row). All images are $1 \times 1 \ \mu m$.

- ³⁰ The current density-voltage (*J-V*) curves are shown in Fig. 4a and the device performances are summarized in Table 1. The optimum DIO content was 2% (ν/ν). It resulted in a device with a V_{oc} of 0.77V, a J_{sc} of 11.71 mA cm⁻², a FF of 71% and a PCE of 6.41%. The high J_{sc} of the devices resulted from efficient light ³⁵ absorption and optimized morphology of the active layers. To confirm the accuracy of the measurement, the corresponding external quantum efficiency (EQE) of the solar cell device was measured under illumination of monochromatic light (Fig. 4b). The photo-response of the solar cell extended from 300 nm to 40 700 nm with a peak reached near 60% around 600 nm. The EQE
- result was consistent with the UV-vis absorption spectrum of the **PBDT-IBTI**: $PC_{61}BM$ blend film.

The effect of additive on morphology was studied by atomic force microscopy (AFM). The AFM topography and phase image

- ⁴⁵ were measured on films which were spin-coated on top of ITO with a thin layer of zinc oxide from a blend of **PBDT-IBTI**: $PC_{61}BM$ (1:2, w/w) with or without DIO. DIO has been reported as additive for the improvement the morphology of the active layer.¹⁴ The active layer processed with DIO [2% (v/v)] displayed
- ⁵⁰ an improved morphology compared with the one processed without DIO (Fig. 5). After addition of DIO, the big domains were broken into pieces, indicating a good miscibility and large interface area between donor and acceptor. The well-proportioned and interpenetrated morphology was achieved after addition of
- ⁵⁵ DIO. This ensured the efficient exciton separation and charge transport and resulted in the enhancement of photocurrent (from 6.84 to 11.71mA cm⁻²) and PCE.

Conclusions

Synthetic modification of isoindigo related building blocks based on rational design has been carried out in this study. An expanded isoindigo unit with 5,5'-divinyl-2,2'-bithiophene as conjugated bridge has been successfully incorporated into donor/acceptor conjugated polymer for the first time. The steric repulsion has been avoided and favorable oxygen-sulfur non-bonding interaction has been introduced into the building block. The new polymer **PBDT-IBTI** showed broad absorption in visible region, large extinction coefficient and a relatively deep HOMO energy level. The optical bandgap, HOMO and LUMO levels of this 95

polymer are 1.71 eV, -5.24 eV and -3.72 eV, respectively. Polymer solar cells based on **PBDT-IBTI**:PC₆₁BM blends as active layer achieved a PCE of 6.41%. This work has demonstrated that the IBTI is a promising building block for OPV $_{5}$ polymers. Furthermore, this work has shown the versatility of

5 polymers. Furthermore, this work has shown the versatility of oxindole as a basic building element for expanded isoindigo like conjugated monomer.

This work was supported by National Natural Science Foundation ¹⁰ of China (NSFC Grant Nos. 21174084, 21274087), by the National Basic Research Program of China (973 Program) (2012CB932402), and by Doctoral Fund of Ministry of Education of China (Grant No. 20120073110032).

Notes and references

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Graph Abstract

An expanded isoindigo unit (IBTI) has been incorporated into donor/acceptor conjugated polymer for the first time. The PCE of the solar cell device based on the new polymer reached 6.41% with a fill factor of 0.71.

