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ARTICLE TYPE

CO₂ Reduction at Low Overpotential on Cu Electrodes in the Presence of Impurities at Subsurface

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We report a theoretical analysis of the Cu electrode, modified by O impurities at the subsurface, concerning electrochemical activity of CO₂ reduction reaction and selectivity between CO and HCOOH evolution. On the basis of quantum mechanical simulations, we propose a novel Cu electrode modified by C impurities at the subsurface for CO₂ electroreduction, which has a better catalytic selectivity.

Carbon dioxide (CO₂) emission causes global climate change from an environmental point of view. CO₂ reduction can solve these environmental problems, and at the same time it can produce economic profits due to the fact that the products of CO₂ reduction can serve as fuels. In order to achieve benefits from CO₂ reduction, we need an efficient catalyst to reduce CO₂ with minimal energy costs and maximal outputs. If we are able to achieve efficient electrochemical processes, many renewable electrical energy resources can be used to reduce CO₂, simultaneously solving the storage issues of intermittent energies, e.g., solar, wind, geothermal, etc.¹ Polycrystalline copper (Cu) electrodes have drawn a lot of attentions because they can be used to reduce CO₂ to hydrocarbon species.² However, the main drawback of the Cu electrode is attributed to its very high overpotential in the order of 1 V with respect to the reversible hydrogen electrode (RHE). This is because the hydrogen evolution reaction (HER) is always dominant at low overpotential.³

Recently, Li *et al.*⁴ reported a promising experimental technique to fabricate modified Cu electrodes by annealing Cu foil in air, followed by electrochemical reduction of the resulting Cu₂O layers. With the modified Cu electrodes they can achieve a much better performance in CO₂ reduction to CO and HCOOH with a low overpotential of less than 0.5 V.³ Catalytic selectivity is yet another important issue. In the above experiments of Li *et al.*,⁴ the CO evolution reaction was preferred over the HCOOH evolution at a low overpotential

<0.3 V, while the HCOOH evolution reaction was enhanced at an overpotential >0.3 V.

It is well-known that the Cu(211) surface is the active site, where CO₂ can be electrochemically reduced to a hydrocarbon at a high overpotential of ~1 V.⁵ However, the activity and selectivity of CO₂ reduction on the modified Cu electrodes are quite different. Therefore, many fundamental issues of the modified Cu electrodes are not understood yet.

In the present work, we have simulated the CO₂ reduction at the active sites of modified Cu electrodes to elucidate the underlying chemical mechanism in the experiment.⁴ We have carried out all the calculations using density functional theory (DFT). We have found that the active sites of the modified Cu electrodes are correlated with the Cu(111) surface, decorated by interstitial oxygen impurities at the subsurface. However, the Cu(211) surface is passivated by adsorbed oxygen species at the topmost surface. Moreover, we propose a novel Cu electrode, modified by interstitial carbon at the subsurface, which possesses a similar activity but a better HCOOH selectivity compared to the oxygen modified Cu electrodes.

On one hand, the modified Cu electrodes were prepared by annealing Cu foils in air and the subsequent electrochemical reduction. On the other hand, it has been found that ultrathin oxide films on surfaces of transition metals are active sites for chemical reactions.⁶ Therefore, we employed periodic Cu slab models, modified with interstitial oxygen impurities at the subsurface, to simulate the surface activity of the modified Cu electrodes. The Cu surface models correspond to two crystal planes, that is, Cu(111) and Cu(211), respectively. In addition, the unit cells of Cu(111) and Cu(211) surfaces are constructed based on (2×2) supercells containing five atomic layers, and (3×1) supercells containing nine atomic layers, respectively. The concentrations of interstitial oxygen impurities are in the range from one to three atoms in each unit cell, as shown in **Fig.1**. The Cu surfaces containing interstitial oxygen atoms are abbreviated with Cu_O(111) and Cu_O(211), respectively. In the Cu_O(111) case, the initial sites of oxygen impurities are described using subscripts *t* (tetrahedral), *o* (octahedral), and *b* (below). For the Cu_O(211) case, the subscripts (*a*, *b*, and *c*) point to three different initial positions of oxygen impurities at the subsurface as shown in **Fig.1**. The number of subscripts corresponds to the number of O atoms for all cases.

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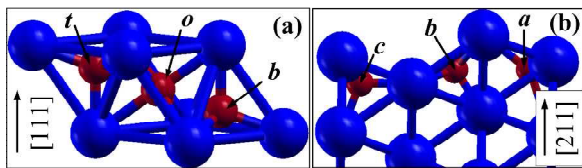


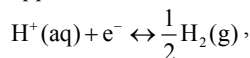
Fig.1 Side views of (a) Cu(111) and (b) Cu(211) surfaces with interstitial oxygen impurities. Cu and O atoms are represented in blue and red balls, respectively. Three cases of interstitial oxygen are denoted with *t*, *b*, and *o* at the Cu(111) subsurface; *a*, *b*, and *c* at the Cu(211) subsurface.

The stability of modified Cu₀(111) and Cu₀(211) surfaces are analyzed using formation energies of interstitial oxygen impurities calculated as follows:

$$E_f = E_{tot} - E_{Cu} - n\mu_O,$$

where E_{tot} , E_{Cu} , n , and μ_O correspond to the total energies of modified Cu surfaces and the pristine Cu surface, the number of interstitial impurities, and the chemical potential of oxygen atom, respectively. The chemical potential μ_O was estimated as half the energy of a O₂ molecule, and added fluctuations in the realistic environment, $\Delta\mu$. A negative formation energy accounts for a more stable surface.

We have calculated the free energy diagrams to analyze the electrochemical reduction processes using the computational hydrogen electrode approach:



the above chemical reaction is defined to be in equilibrium without applied voltage and with H₂ gas at pressure of 1 atm, at any pH values and temperatures. Therefore, the chemical potential of a proton-electron pair, that is, $\mu(\text{H}^+) + \mu(\text{e}^-)$, is equal to $\frac{1}{2}\mu(\text{H}_2)$ without applied external potential ($U = 0$ V). The free energy change, ΔG , of all electrochemical reactions with respect to chemical potential can be estimated using $\Delta G = -eU$, where e is the elementary (positive) charge. In addition, the binding energies of chemical intermediates in electrochemical reductions are referred to the isolated CO₂ molecule.

All-electron Kohn-Sham equations were solved using the scheme of Linear Combination Atomic Orbitals (LCAO), as implemented in CRYSTAL09 code.⁷ Gaussian-type basis sets were used for H,⁸ C,⁹ O,¹⁰ and Cu¹¹. The difference of Cu(111) surface energies is ~ 0.03 eV/Å² compared with another basis set.¹² We have employed the Perdew-Burke-Ernzerhof (PBE) functional for the electronic exchange-correlation potential at the level of the Generalized Gradient Approximation (GGA). The hybrid PBE0 functional was utilized to validate some important energies. Calculated total energies were integrated with k -point grids, resulting in (5×5×1) and (4×4×1) sampling for the Cu₀(111) and Cu₀(211) surfaces, respectively.

In the case of O_o, the formation energy of the modified Cu₀(111) surface is -0.3 eV (see Fig.2). With increasing the concentration, the interstitial oxygen impurities stabilizes the Cu₀(111) surface with a formation energy of -1.3 eV (O_{bt}). As the oxygen content is further increased to O_{bto}, the modified Cu₀(111) surface shows an enhanced roughness (see Fig.2) in comparison with the pristine Cu(111) surface. In contrast to the Cu₀(111) surface, the interstitial oxygen impurities are not

stable at the Cu₀(211) subsurface. Although the calculated formation energies of Cu₀(211) surfaces suggest the Cu(211) surface can be stabilized by incorporated oxygen impurities, the critical point is that the oxygen impurities prefer to be adsorbed on the top of Cu(211) surface, instead of being present at the subsurface (see Fig.2). However, the adsorbed oxygen on the top of Cu₀(211) surface is definitely repulsive to a CO₂ molecule. Therefore, the catalytic activity of CO₂ reduction towards hydrocarbon on the Cu₀(211) surfaces of modified Cu electrodes should be much lower than that on the unmodified polycrystalline Cu electrodes.³ This is indeed consistent with the observations in the experiments of Li *et al.*⁴ Therefore, we conclude that the observed enhanced CO₂ reduction is not due to the contribution from Cu(211) and Cu₀(211) surfaces.

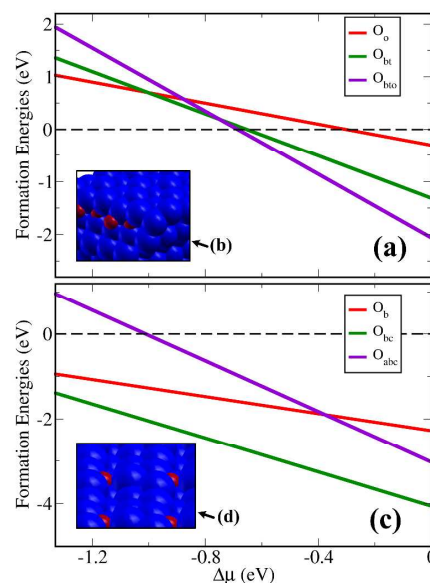


Fig. 2 Calculated formation energies of (a) Cu₀(111) and (c) Cu₀(211) surfaces with a variety of studied concentration of interstitial oxygen impurities at subsurfaces, where the $\Delta\mu$ was referred to a half energy of a O₂ molecule. The black dashed line is the reference (pristine Cu(111) and Cu(211) surfaces). The most stable (b) Cu₀(111) and (d) Cu₀(211) surfaces are shown, where Cu and O atoms are represented in blue and red, respectively.

Calculated d-band states of a pristine Cu(111) surface are shown in Fig.3, which is comparable with the previous calculated results.¹³ However, we have found the modified Cu₀(111) surface has more d-band states in the range from the Fermi level to -2 eV below. In light of the d-band model,¹⁴ a CO₂ molecule should be bound with the modified Cu₀(111) surface more stronger, compared with a pristine Cu(111) surface. This results from a stronger repulsion between d-band states and antibonding states of *C and *O species (hereafter, we will use an asterisk to describe a adsorbed species on a surface, unless otherwise stated). In the case of O_{bto} impurities, the calculated binding energy of CO₂ on the Cu₀(111) surface (-0.41 eV) is much stronger, by ~ 0.5 eV compared with that on a pristine Cu(111) surface. Therefore, we understand that the enhanced activity of CO₂ electroreduction in the experiments of Li *et al.*⁴ was probably

due to a stronger binding strength of CO₂ molecules on the modified Cu electrodes. The active sites of the modified Cu electrodes are contributed from the modified Cu₀(111) surface instead of modified Cu₀(211) surface. As it is well-known, Cu(111) surfaces are the most stable facets on polycrystalline Cu electrodes. Hence, the improved activity on the Cu(111) surface can reasonably explain the observed high efficiency in the experiment.⁴

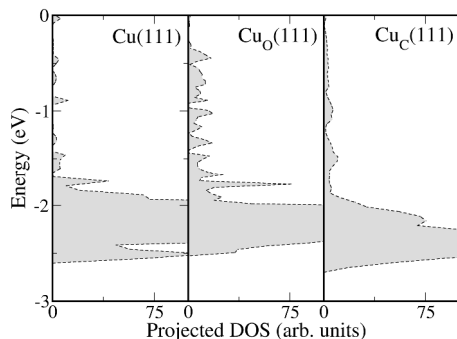


Fig.3 Calculated d-band states of a pristine Cu(111), the modified Cu₀(111) and Cu_C(111) in the O_{bito} and C_{bito} cases, respectively. The Fermi energies are specified to be zero.

Moreover, the HER on the unmodified Cu electrodes is dominant at a low overpotential.⁵ However, the activity of the HER seems to be suppressed on the modified Cu electrodes.⁴ The HER performance on different catalysts can be described in terms of a volcano curve, where the activity of catalysts is a function of the hydrogen binding strength on the surface of catalysts. The activity of the HER on the Pt(111) surface was demonstrated to be close to the optimal activity (the peak of volcano). In the present work, the computed hydrogen binding energy on a Pt(111) surface (-0.08 eV) is in agreement with the reported value.¹⁵ On the pristine Cu(111) surface, the binding strength of hydrogen is slightly weaker (0.10 eV). However, the binding strength of hydrogen on the modified Cu₀(111) surface is too strong, that is, -0.93 eV. The trend is also consistent with calculations using hybrid PBE0 functional (-1.51 eV). This argument can also help to understand why the CO₂ reduction can outcompete the HER at a low overpotential in the experiment.⁴ In other words, the HER activity is strongly suppressed by the process of H₂ desorption on the modified Cu electrodes.

The next important issue is to understand why the CO and HCOOH evolution can take place at a quite low overpotential (-0.5 V vs. RHE). In the work of Hansen *et al.*,¹⁶ they had proposed two important factors for CO₂ reduction to obtain a high activity of CO evolution. In the case of the Au(211) surface, the *COOH species are not stable. The enhanced overpotential is helpful for CO evolution by means of stabilizing the *COOH species on the surface. In contrast to the Au(211) surface, *CO desorption becomes quite difficult from a Pt(211) surface due to the very strong binding energy of *CO species. Hence, they pointed out the two binding energies of *COOH and *CO are essentially responsible for the required overpotential in experiment. They proposed that for the CO evolution reaction the optimal catalysts should be able to stabilize *COOH species and possess facile desorption of resulting *CO species.

Our calculated ΔG of CO₂ reduction reactions are indeed in agreement with the proposed criteria. Our calculated binding energy indicates that the *COOH on the Cu₀(111) surface is sufficiently stable (see Fig.4). The stability of *COOH can be explained in terms of the scaling relation on heterogeneous surfaces.^{17,18} In other words, the stability of the *COOH is directly correlated with the binding strength of *CO₂ molecule on the Cu₀(111) surface. Moreover, the Cu₀(111) surface has facile *CO desorption (0.01 eV) in the presence of adsorbed *OH species, in comparison with that of the Pt(211) surface.

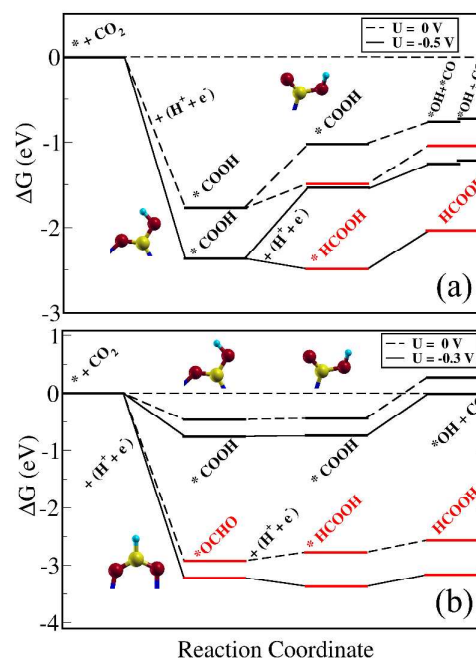


Fig.4 Calculated free energy diagrams of CO₂ electrochemical reductions to CO and HCOOH on (a) Cu₀(111) surface at an overpotential of -0.5 V vs. RHE, and (b) Cu_C(111) surface at an overpotential of -0.3 V vs. RHE.

From the energetic point of view, the HCOOH evolution reaction should be always more favorable in comparison with the CO evolution one (see Fig.4). However, the desorption of *HCOOH is more difficult than that of *CO, therefore, it was observed in the experiments that CO evolution outcompetes HCOOH evolution on the modified Cu electrodes at a low overpotential (<0.3 V). In addition, it was also observed that at higher overpotential (>0.3 V) the CO evolution fails in the competition. This is because the *HCOOH formation becomes exergonic at higher overpotentials. The required overpotential is ~0.28 V in our DFT calculations (PBE0: ~0.32 V), which is consistent with the minimal experimental overpotential of decreasing the amount of CO evolution (~0.3 V).⁴

In principle, it is possible to manipulate the HCOOH and CO evolution by means of tuning the relative stability of some important species, namely, *CO, *COOH, and *HCOOH, as well as the desorption energies of *CO and *HCOOH. At the first step, we simply neglect the structural effects of modified catalysts and uniquely optimize the electronic structures. In light of the d-band model and the fact of weak reactivity on a pristine Cu(111) surface, we hope that the d-band states of the new catalyst are close to those of the Cu₀(111) surface to

improve the CO₂ binding strength (see Fig. 3). Meanwhile, the new catalyst must be stable under specific conditions. We screen a variety of elements close to oxygen in the Periodic Table, that is, C, N, and F elements, using the same atomistic models (see Fig. 1). We also use the same nomenclature in the following discussions. First, calculated formation energies implicate that Cu_N(111) surfaces are not stable. Additionally, in the F_{bto} case of modified Cu_F(111) surfaces, the interstitial F atoms prefer to aggregate in forming a F₂ molecule at the subsurface. Obviously, the F₂ molecule is likely to destroy the surface due to its volume expansion. In addition, we do not want to give rise to potential secondary pollution from our catalysts. Thus, nitrides and fluorides are not good precursors in experiments. Fortunately, we found the Cu_C(111) surfaces are stable with -1.21, -1.27, and -1.14 eV/C in the cases of C₀, C₁₀ and C_{bto}, respectively. Meanwhile, the Cu_C(111) surfaces exhibit enhanced reactivity of CO₂ adsorption, compared with a pristine Cu(111) surface, because the effective d-band center of the Cu_C(111) surface shifts towards the Fermi level (see Fig. 4).

Furthermore, we have performed calculations based on the most stable Cu_C(111) surface in the case of C_{bto} impurities. Calculated free energy diagrams indicate that the pathway of HCOOH evolution is energetically more favorable than the CO evolution (see Fig. 4). Moreover, CO desorption from the Cu_C(111) surface is not preferred due to an adsorption energy of ~0.7 eV (PBE0: ~0.65 eV), which is significantly higher than that of the Cu₀(111) surface. In contrast, the HCOOH desorption energy from the Cu_C(111) surface is only ~0.2 eV (PBE0: ~0.14 eV). Hence, the HCOOH evolution can always outcompete the CO evolution from the desorption point of view. Moreover, the *HCOOH formation is not very difficult compared with the formation of *COOH (PBE: ~0.13 eV, PBE0: ~0.11 eV). In addition, we found a stable intermediate, namely, *OCHO, on the Cu_C(111) surface. This intermediate can make use of another pathway to drive CO₂ reduction to the HCOOH evolution (see Fig. 4), and meanwhile suppress the CO evolution. Hence, one can expect that the efficiency of HCOOH evolution should be much higher than CO evolution. The validations of our theoretical predictions need further experimental efforts.

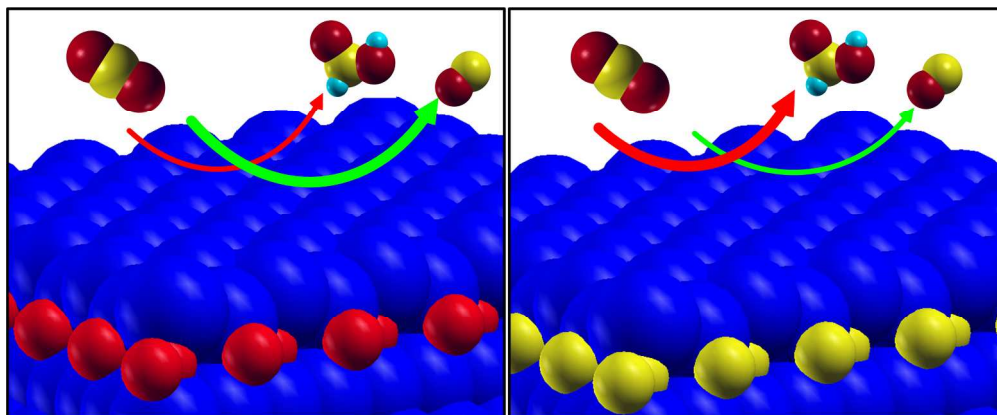
In summary, we have performed Density Functional Theory calculations to elucidate the CO₂ reduction reactions at a low overpotential on the modified Cu electrodes in experiments. We have found the interstitial oxygen impurities can be stable at Cu(111) subsurface, which are able to improve the binding ability of CO₂ molecules on the modified Cu electrodes. Therefore, CO₂ reduction reactions become feasible at a low overpotential. The catalytic selectivity can be attributable to the competitions and compromise between CO desorption and *HCOOH formation. Our calculations are consistent with the observed transition from CO evolution to HCOOH evolution. Based on the better understanding concerning the previous experimental results, we performed further studies to propose a novel catalyst, that is, the carbon modified Cu electrodes. It has a better HCOOH selectivity in CO₂ electroreduction.

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Tune the electrochemical activity and selectivity of Cu electrodes by carbon and oxygen impurities at their subsurface.

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