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Rheological properties and shear-induced structures of ferroelectric nematic liquid crystals

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Recently discovered ferroelectric nematic (N_F) liquid crystals are fluids with a polar orientational order. The electric polarization vector can be aligned by an electric field and by surface anchoring. Here, we explore how the polarization field and effective viscosity of the N_F materials are affected by shear flows. We explore three N_F materials, abbreviated RM734, DIO, and a room-temperature FNLC919, all of which exhibit a paraelectric nematic (N) and an N_F phase. All materials show an increase in the effective viscosity upon cooling, with Arrhenius behavior in broad temperature ranges except near the phase transitions. In DIO and FNLC919, the antiferroelectric SmZ_A phase separating the N and N_F phases shows a strong dependence of the effective viscosity on the shear rate: this viscosity is lower than the viscosity of the N and N_F phases at high shear rates ($\dot{\gamma} = 500 \text{ s}^{-1}$) but is much higher when the shear rate is low, $\dot{\gamma} = 2.5 \text{ s}^{-1}$. The behavior is associated with the layered structure of the SmZ_A phase. All mesophases in all three materials exhibit shear-thinning behavior at low shear rates ($< 100 \text{ s}^{-1}$) and a nearly Newtonian behavior at higher shear rates. In terms of alignment, we observe three regimes in the N and N_F phases: flow-alignment at low shear rates, $\dot{\gamma} < 10^2 \text{ s}^{-1}$, a log-rolling regime with the director and polarization along the vorticity axis at $\dot{\gamma} > 10^3 \text{ s}^{-1}$, and polydomain structures at intermediate rates. In the flow-aligning regime, the N_F polarization does not tilt away from the shear direction, which is in sharp contrast to the flow-induced tilt of the N director. The effect is attributed to the avoidance of splay deformations and associated space charge in the flowing N_F . The temperature and shear rate dependencies of the viscosity and the uncovered shear-induced structural effects of N_F advance our understanding of these materials and potentially facilitate their applications.

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1. Introduction

A nematic (N) liquid crystal (LC) formed by achiral rod-like molecules shows a uniaxial long-range orientational order but a lack of positional order. The molecules align along a common direction described by a unit vector \hat{n} , called the director, with the property $\hat{n} \equiv -\hat{n}$, which makes the material paraelectric. Recent synthesis and characterization of liquid crystals established the existence of the ferroelectric nematic (N_F) phase, in which rod-like molecules with large longitudinal electric dipoles align in a polar fashion along the director, resulting in spontaneous macroscopic electric polarization \mathbf{P} .^{1–4} In the N_F phase, the polarization is strong, $P \approx 6 \times 10^{-2} \text{ C m}^{-2}$, enabling electro-optic response to electric fields as small as $\sim 10^2 \text{ V m}^{-1}$, a thousand times smaller than those used to reorient nonpolar N.⁴

Orientation of a paraelectric N is strongly affected by flows. Over the last few decades, significant progress has been achieved in understanding the shear-induced structures and rheology of the paraelectric N.^{5–9} These materials exhibit different modes of response to the applied shear, such as flow-alignment,^{5,6} tumbling,¹⁰ log-rolling,^{11,12} and kayaking.^{7,8} For example, well-studied nematics MBBA¹³ and 5CB¹⁴ show a flow alignment, *i.e.*, \hat{n} aligns parallel to the shear plane formed by the velocity and its gradient. In contrast, 8CB, formed by molecules with a slightly longer aliphatic tail as compared to 5CB and exhibiting a smectic A (SmA) phase in addition to the N phase, shows a tumbling behavior, with \hat{n} rotating in the shear plane and realigning along the vorticity direction perpendicular to the shear plane.¹⁵

In contrast to the paraelectric N, very little is known about the rheological behavior of the N_F . One should expect a rich plethora of flow phenomena, thanks to the presence of spontaneous electric polarization, its spatial variations, intrinsic to the fluid nature of the N_F , and strong coupling to the electric fields.^{4,16} For example, Dhara *et al.* reported an increase in the effective viscosity of the N_F material, abbreviated RM734, in the

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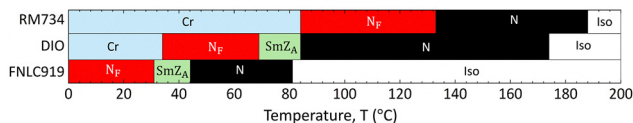


Fig. 1 Phase sequences of RM734, DIO, and FNLC919.

presence of an electric field.¹⁷ Even stronger electroviscous response was reported recently by Nishikawa *et al.* for another N_F material, abbreviated DIO.¹⁸ However, the structural response of N_F to shear, its flow regimes, dependence of the effective viscosities on shear rate and temperature, proximity of phase transitions, *etc.* remain underexplored.

In this work, we perform comparative rheological studies of the N and N_F phases, as well as the intermediate phase separating the N and N_F phases, in three different materials, RM734,¹ DIO,² and FNLC919.¹⁹ Namely, we measure the effective viscosity as a function of temperature and shear rate and determine the structural response to shear flows using a plate-plate rheometer equipped with an *in situ* polarizing optical microscope (POM). Effective viscosity increases as the temperature is lowered and the material transitions from the N to the N_F phase. Far away from the phase transition temperatures, the viscosity follows the Arrhenius behavior but increases sharply near the transition points. As a function of the shear rate, both the N and N_F phases show strong shear-thinning at shear rates below 100 s^{-1} and a nearly Newtonian behavior at higher shear rates. The structural response of the N and N_F phases in all materials reveals three regimes as a function of shear rate: flow alignment at low rates, $\dot{\gamma} < 10^2 \text{ s}^{-1}$, log-rolling at $\dot{\gamma} > 10^3 \text{ s}^{-1}$, and polydomain structures at intermediate rates. In the flow-aligning regime, the N director tilts away from the shear direction, while in the N_F phase, such a tilt is absent. This important difference is caused by the avoidance of splay deformations and associated space charge in the N_F phase.

2. Experimental methods

2.1 Materials

We explore three different N_F materials, abbreviated RM734,¹ DIO,² and a room-temperature FNLC919.^{19,20} The phase sequences for these materials are presented in Fig. 1. RM734 was purchased from Instec, Inc. (purity better than 99 wt%). DIO was synthesized as described previously by Nishikawa *et al.*² The room-temperature N_F material FNLC919 is provided by Merck KGaA (Darmstadt, Germany).

The N and N_F phases in all three materials are separated by an intermediate phase. Its nature is still debated, prompting us to present a brief overview. It was first observed in DIO,^{2,21–23} then in RM734,²⁴ and in other materials.^{23,25–28} This phase was labeled as M_2 ,² N_x ,²² SmZ_A ,²⁹ N_s ,³⁰ and M_{AF} ,³¹ the abbreviations reflect the perceived structure of this intermediate phase. The first report by Nishikawa *et al.* on DIO² pointed out that the transition upon cooling from the N to the new phase, known nowadays as the N_F phase, occurs in two steps. Brown *et al.*²¹ and Erkoreka *et al.* suggested that the phase is antiferroelectric.²²

Subsequent synchrotron-based small and wide-angle X-ray scattering (SAXS/WAXS) combined with polarizing optical microscopy observations by Chen *et al.* demonstrated that the intermediate phase of DIO exhibits a lamellar type of order with a periodicity of 17.5 nm .³² Within each period, there are two sublayers of thickness $w = 8.8 \text{ nm}$ each, manifested by an equilibrium sinusoidal electron-density modulation observed in non-resonant SAXS. The molecules are tangential to the layers and polarly ordered; the direction of polarity alternates from one subdomain to the next. The density modulation observed in the intermediate phase of DIO by Chen *et al.*³² justifies the abbreviation “Sm”, since it traditionally stands for “smectic”. “Z” reflects the fact that the average molecular orientation is parallel to the layers rather than perpendicular to them, as in SmA. The subscript “A” stands for antiferroelectric. A different model was suggested recently by Rupnik *et al.*³³ on the basis of experimental studies of the intermediate phase in RM734, which in a pure material extends over a very narrow temperature range of about $1 \text{ }^\circ\text{C}$. The N phase of RM734 shows a dramatic decrease in the splay elastic constant K_1 upon the approach to the N_F phase.³⁴ When an ionic fluid is added to RM734, the intermediate phase range increases dramatically and shows periodic domains of a period up to $10 \text{ }\mu\text{m}$. The observations are explained by the splay-modulated phase, abbreviated N_s , in which a reversal of the splay $\text{div } \mathbf{P}$ and of the average polarization \mathbf{P} decreases the overall space charge and allows the structure to fill the space efficiently. In another study, a doubly splay-modulated antiferroelectric phase has been observed in RM734 in cells with ionic polymers as aligning layers.³⁵

In what follows, we keep the abbreviation SmZ_A for DIO since the material shows one-dimensional periodic density modulation, a defining property of a smectic order, directly revealed in SAXS³² and supported by light scattering experiments,³⁶ response to shear flow¹⁸ and by the behavior of dislocations in DIO with chiral additives.³⁷ We use the same abbreviation for the intermediate antiferroelectric phase of FNLC919 since a similar density variation with a period of 6.7 nm has been documented in SAXS experiments by Paul *et al.*³⁸

2.2 Rheometry

Rheological measurements are performed using a strain-controlled rheometer Anton Paar, MCR 302 with a parallel-plate measuring system having a plate diameter of 25 mm . The zero values of the normal force and torque were set for each test before introducing the sample. The sample was confined between the plates, and the excess material was removed. The plate gap was fixed to be $200 \text{ }\mu\text{m}$ for RM734, $150 \text{ }\mu\text{m}$ for DIO, and $100 \text{ }\mu\text{m}$ for FNLC919; at these distances, the normal force was zero after the assembly of the test cells. No alignment layers were used. The top plate rotates at different shear rates while the bottom plate is fixed. A Peltier element was attached to the bottom plate to control the temperature with an accuracy of $0.1 \text{ }^\circ\text{C}$. The rheometer was placed under a hood to maintain temperature uniformity. The temperature dependence of effective shear viscosity was measured at a low shear rate of $\dot{\gamma} = 2.5 \text{ s}^{-1}$ and a high shear rate of $\dot{\gamma} = 500 \text{ s}^{-1}$ upon cooling and heating. The cooling and heating rates are $2 \text{ }^\circ\text{ min}^{-1}$ in the



range of 160–110 °C for RM734, 110–58 °C for DIO, and 85–18 °C for FNLC919, respectively.

2.3 *In situ* polarizing optical microscopy

To characterize the shear-induced structural changes, we use the Linkam Optical Shearing System CSS450 in parallel disk geometry. The shear device features a window for observing textures using a polarizing optical microscope (POM) in the transmission mode. The bottom plate rotates relative to the top plate with a controllable angular velocity in the range 0.001–10 rad s⁻¹. Both plates contain quartz windows of 2.5 mm in diameter, centered at 6.25 mm from the axis of rotation. A separately controlled heater maintains the temperature of the sample fixed within 0.1 °C. No alignment layer is used so that the substrates of CSS450 yield degenerate tangential alignment of the director \hat{n} . To reconstruct the structural response under shear flow, we use a POM Olympus BX51 equipped with a video camera Baslar asA1920-155um (10–40 frames per second) with a full-wave-plate (FWP) 530 nm optical compensator. The slow axis λ_g of the optical compensator is oriented at 45° to the crossed polarizers.

2.4 Polychromatic polarizing microscopy

A polychromatic polarizing microscope (PPM) generates a map of director orientation in a single shot of a camera.³⁹ The PPM, invented by Shribak in 2009,⁴⁰ is equipped with a special polychromatic polarization state generator (PSG) and an achromatic circular analyzer. The PPM produces a full hue-saturation-brightness (HSB) color spectrum in birefringent materials. The HSB hues in PPM depend on the slow axis orientation of the birefringent specimen with respect to a preset 'zero' direction (usually oriented along the east-west axis of a microscope's stage). This implies that the color changes occur continuously while the stage rotates, repeating every 180°, and the state of extinction is never observed. The PPM allows one to capture an image of fast-moving and low-birefringent structures in real-time, limited only by the acquisition time of the camera. The PPM approach is different from

that of a conventional polarizing microscope, in which the retardance of the specimen determines the interference color in the Michel-Levy chart.

The optical scheme of a PPM, which consists of a polychromatic PSG and an achromatic circular analyzer, with the N compensating cell, is shown in Fig. 2. The PSG produces polarized light with the polarization ellipse orientation that depends on the wavelength. The PSG comprises a rotatable polarizer, an achromatic quarter-wave plate (AQWP), and an optically active waveplate (OAWP). The polarizer and AQWP produce a polarization ellipse with a major axis parallel to the slow axis of the AQWP. When the elliptically polarized light travels along the optical axis of the waveplate OAWP, the polarization ellipse rotates by some angle. The rotation angle depends on the thickness of the OAWP and the wavelength of light. An achromatic quarter-wave plate (AQWP) with a slow axis oriented at 45° and an analyzer are combined to form the achromatic left circular analyzer. The achromatic quarter-wave plate minimizes the variation in retardance across a broad spectral range.^{41,42}

A limitation of an original PPM device is that the recorded retardance of the birefringent specimen should be less than 250 nm. To overcome the limitation, we add an anisotropic optical compensator, representing a planar LC cell of the N material E7 (Jiangsu Hecheng Display Technology Co., Ltd (HCCH), Jiangsu, China) with a slow axis (director) λ_N . The N compensating cell introduces an optical retardance of an opposite sign to that of the sample, in order to reduce it below 250 nm. The N compensating cell is inserted between the sheared specimen and the achromatic left circular analyzer, with λ_N being perpendicular to the optic axis of the sheared material, so that the retardance of this cell reduces the measured retardance of the system. The compensating cell is assembled from two flat glass substrates, spin-coated with a polyimide PI2555 (Nissan Chemicals, Ltd), and rubbed unidirectionally to achieve a uniaxial planar alignment of λ_N . The optical birefringence $\Delta n = 0.22$ of E7⁴³ (reported at a wavelength of 600 nm, room temperature $T = 25$ °C) is close to the

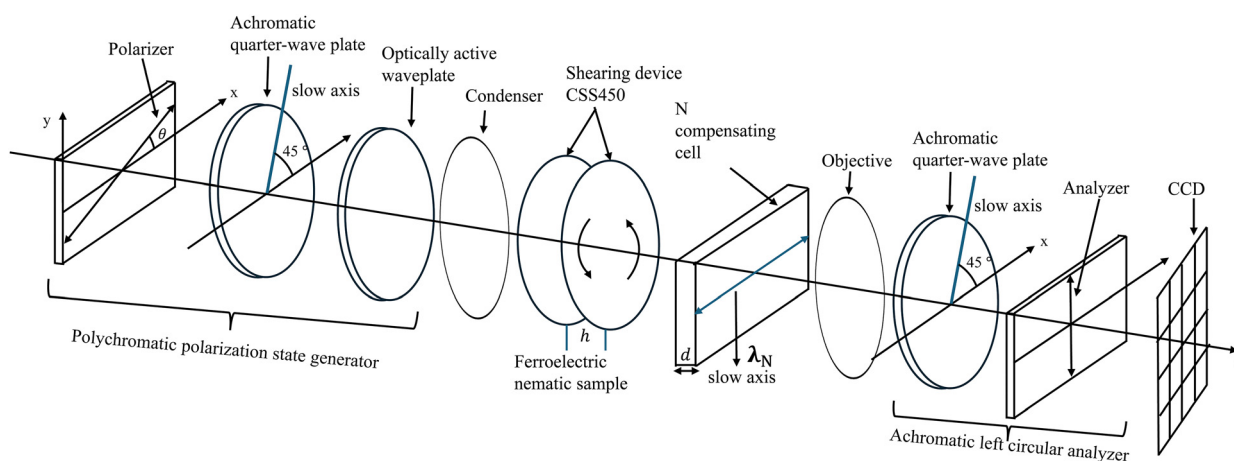


Fig. 2 Schematic of the experimental setup to characterize the structural changes of the ferroelectric nematic liquid crystal caused by shear flow using a PPM with the N compensating cell. λ_N is the slow axis of the N compensating cell.



birefringence of the explored materials; thus, the thickness of the compensating cell was selected to be close to the thickness of the sheared samples. To generate the map of director orientations of the specimen from the HSB hue image of the PPM, spline interpolation of hue data has been performed using the Mathematica code.³⁹

2.5 Optical retardance and PolScope

To measure the optical retardance Γ of the N_F materials as a function of shear rate, we use the LC PolScope approach, invented by Oldenburg,^{44,45} the applications of which to liquid crystals have been described in ref. 46 and 47. The LC PolScope represents a polarizing optical microscope with a variable optical compensator(s), which might be an N cell controlled by an applied electric field. Once the image of a sample is taken for a few different settings of the compensator, numerical analysis allows the unit to map the optical retardance Γ of the sample and reconstruct the in-plane orientation of the optical axis \hat{n} . The PolScope observations in this study are performed using the Exicor Microimager (Hinds Instruments), operating at four wavelengths: 475 nm, 535 nm, 615 nm, and 655 nm, which allows for the characterization of samples with optical retardance up to 3500 nm.

3. Results and discussion

3.1 Shear viscosity vs. temperature

The effective shear viscosity η of all three materials in the N, N_F and intermediate phases generally decreases with temperature, Fig. 3a–c, a behavior typical for many N materials, including MBBA, 5CB and 8CB.^{48–50} Notable features are pretransitional increases of η and discontinuous changes at the phase transitions. In RM734, η drops from 0.10 Pa s in the N_F phase at 115 °C to 0.03 Pa s in the N phase at 160 °C; these values are similar to the previous results obtained by Dhara *et al.*¹⁷

In the middle of the N_F temperature range of RM734, the shear viscosity is $\eta = 0.13$ Pa s, Fig. 3a. Interestingly, this value is close to the rotational viscosity of RM734 $\gamma = 0.15$ Pa s measured at the same temperature by rotating the polarization vector \mathbf{P} by an electric field \mathbf{E} applied perpendicularly to \mathbf{P} , which maximizes the realigning torque $\boldsymbol{\tau} = \mathbf{P} \times \mathbf{E}$.⁵¹ Furthermore, the data for RM734, Fig. 3a, and DIO, Fig. 3b, can be compared to the effective viscosity $\bar{\eta}$, called the “polarization reversal dissipation coefficient” and measured in the electro-optical responses of the N_F phase to the field \mathbf{E} that is antiparallel to \mathbf{P} .⁵² Chen *et al.*⁵² found this coefficient to be $\bar{\eta} = 0.05$ Pa s at the highest temperature of the N_F phase in both RM734 and DIO (and in all their binary mixtures), which is remarkably close to the values of shear viscosity $\eta = 0.06$ Pa s in Fig. 3a and b at the highest temperatures of the N_F phase. The polarization reversal dissipation coefficient $\bar{\eta}$ in ref. 52 shows an Arrhenius-like temperature dependency, which is again close to the behavior of the shear viscosity η , as discussed below.

The temperature dependencies of viscosity away from the transition points can be fitted by an Arrhenius law $\eta = \eta_0 e^{\frac{E_a}{k_B t}}$,

Fig. 3d–f, with the activation energies E_a listed in Table 1. Here, k_B is the Boltzmann constant and t is the absolute temperature. The activation energy E_a in the N_F phase is more than twice that in the N phase in all three materials. The viscosity of DIO and FNLC919 is almost the same at a low shear rate of $\dot{\gamma} = 2.5 \text{ s}^{-1}$ and at a high shear rate of $\dot{\gamma} = 500 \text{ s}^{-1}$ in the N and N_F phases.

In contrast, the viscosity of the SmZ_A in DIO and FNLC919 is significantly higher at a low shear rate of $\dot{\gamma} = 2.5 \text{ s}^{-1}$ and lower at a high shear rate of $\dot{\gamma} = 500 \text{ s}^{-1}$ than the viscosities of the neighboring regions of the N and N_F phases. A similar behavior was reported for the SmZ_A phase by Nishikawa *et al.*¹⁸

Smectic layers of the SmZ_A between two parallel plates are oriented randomly at a low shear rate, with some of them being orthogonal to the flow, which would yield a high viscosity. At a high $\dot{\gamma}$, the layers are mostly parallel to the plates of the rheometer and slide over one another while keeping the average \hat{n} along the shear, which results in a lower viscosity. Note that this behavior is different from the behavior of the SmA phase in 8CB, in which \hat{n} is perpendicular to the layers and η is higher than that of the N phase.^{50,53} The most viscous is the mixture FNLC919, while DIO is the least viscous, despite the fact that the temperature range of the DIO mesophases is lower than that of RM734.

3.2 Shear viscosity vs. shear rate

The variation of effective shear viscosity η of all three materials with the shear rate in the range $0.1 \text{ s}^{-1} \leq \dot{\gamma} \leq 1000 \text{ s}^{-1}$ has been explored in the N, N_F and intermediate phases, Fig. 4. All three materials and all three phases show a pronounced shear-thinning at low $\dot{\gamma} < 1 \text{ s}^{-1}$ and almost Newtonian behavior at high $\dot{\gamma} > 100 \text{ s}^{-1}$. The intermediate range $1 \text{ s}^{-1} < \dot{\gamma} < 100 \text{ s}^{-1}$ exhibits moderate shear-thinning. The shear rate dependencies can be presented by a power law $\eta \propto \dot{\gamma}^{n-1}$, in which $n < 1$ describes shear-thinning and $n = 1$ corresponds to the Newtonian flow behavior.

The measured viscosities of DIO in the N_F phase are higher than those reported previously for a similar range of shear rates in ref. 18. For example, at 60 °C, $\eta = (0.07\text{--}0.09)$ Pa s in Fig. 3b and 4b for $\dot{\gamma} = (2.5\text{--}1000) \text{ s}^{-1}$, while $\eta = 0.04$ Pa s in Fig. 4d of ref. 18, measured at $\dot{\gamma} = 2.6 \text{ s}^{-1}$ and 5000 s^{-1} . Another notable difference is that all three phases of DIO in Fig. 4b show a shear-thinning regime and almost a Newtonian behavior at high shear rates, while Fig. 3 in ref. 18 reports a change from the shear-thinning to the shear-thickening regime in the N_F phase of DIO at $\dot{\gamma} > 20 \text{ s}^{-1}$, although with an exponent $n = 1.02$ close to 1. A potential reason for these discrepancies is the different thickness of the shear cell, 150 μm in our case and 80 μm in ref. 18.

3.3 First-normal stress difference vs. shear rate

The variation of the first-normal stress difference N_1 of all three materials as a function of shear rate in the range of $0.1 \text{ s}^{-1} \leq \dot{\gamma} \leq 1000 \text{ s}^{-1}$ has been observed in the N, N_F and intermediate phases, Fig. 5. N_1 is the difference between the normal stress along the flow and the normal stress in the orthogonal direction of the velocity gradient. N_1 is a measure of the



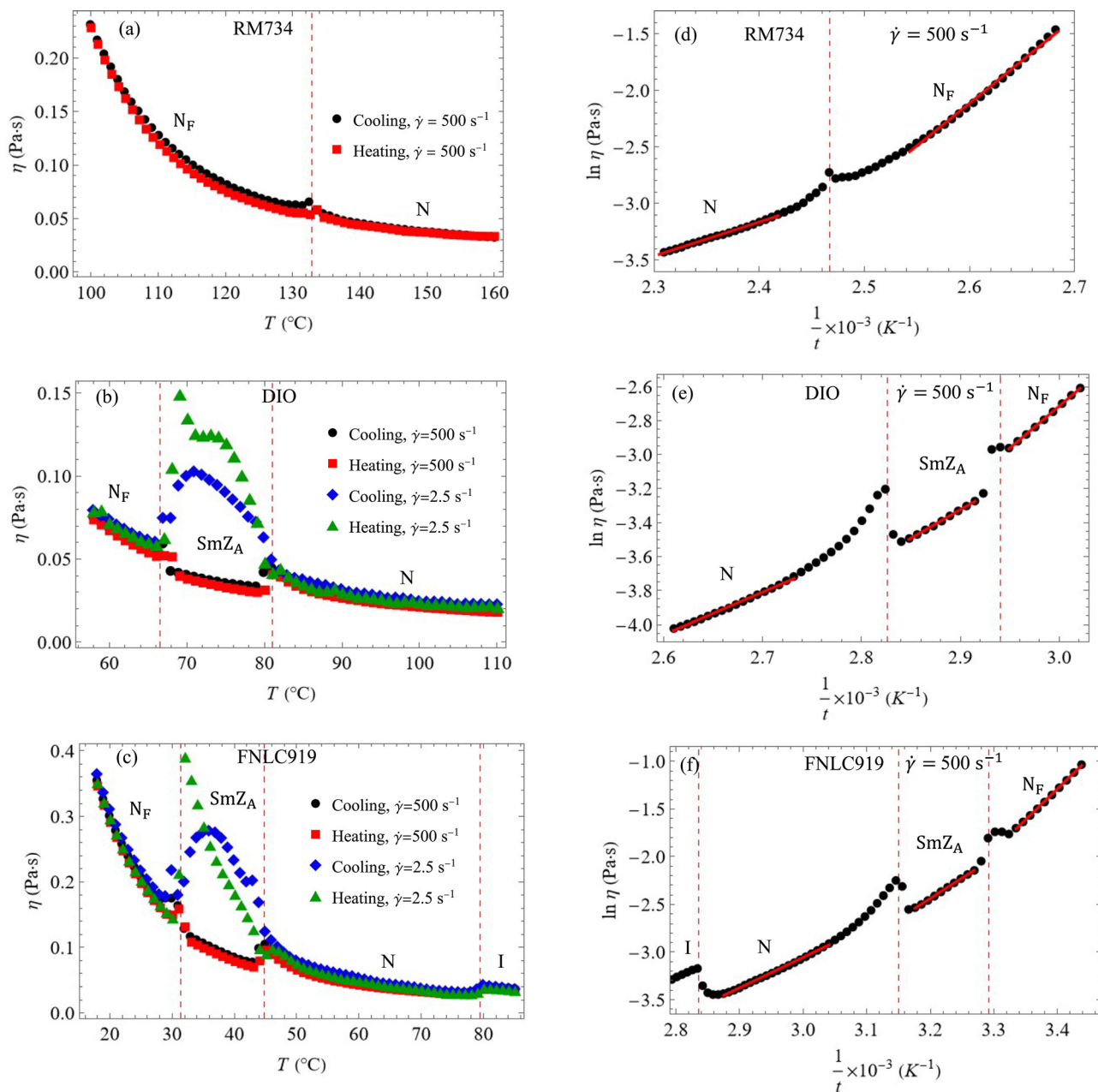


Fig. 3 The temperature dependence of the effective shear viscosity η of RM734 (a) at a shear rate of $\dot{\gamma} = 500 \text{ s}^{-1}$, DIO (b), and FNLC919 (c) at a shear rate of $\dot{\gamma} = 2.5 \text{ s}^{-1}$ and $\dot{\gamma} = 500 \text{ s}^{-1}$. The Arrhenius plot of the effective shear viscosity η of RM734 (d), DIO (e), and FNLC919 (f) at a shear rate of $\dot{\gamma} = 500 \text{ s}^{-1}$.

Table 1 Activation energy of RM734, DIO, and FNLC919 in the N, N_F , and intermediate phases

| RM734 | | DIO | | FNLC919 | |
|------------------------------------|--------------------------------|---|--------------------------------|---|--------------------------------|
| Phase, T ($^{\circ}\text{C}$) | E_a (kJ mol^{-1}) | Phase, T ($^{\circ}\text{C}$) | E_a (kJ mol^{-1}) | Phase, T ($^{\circ}\text{C}$) | E_a (kJ mol^{-1}) |
| N, 160–140 $^{\circ}\text{C}$ | 25.7 ± 0.2 | N, 110–93 $^{\circ}\text{C}$ | 20.4 ± 0.3 | N, 75–55 $^{\circ}\text{C}$ | 25.3 ± 0.3 |
| N_F , 120–100 $^{\circ}\text{C}$ | 60.8 ± 0.5 | SmZ_A , 79–70 $^{\circ}\text{C}$ | 27.8 ± 0.2 | SmZ_A , 42–33 $^{\circ}\text{C}$ | 34.9 ± 0.2 |
| | | N_F , 66–58 $^{\circ}\text{C}$ | 41.1 ± 0.3 | N_F , 28–18 $^{\circ}\text{C}$ | 53.6 ± 0.5 |

non-Newtonian and elastic behavior of fluids under shear; in Newtonian isotropic fluids, $N_1 = 0$. As a rule, isotropic fluids such as polymer solutions show $N_1 > 0$. The reason is that the polymer coils extend along the flow, which creates a restoring

force that tends to return the coil to the initial isotropic shape, thus acting to push the plates of the shear device apart, hence $N_1 > 0$. It thus came as a surprise that some polymers with orientational order show $N_1 < 0$.^{54–57} The qualitative



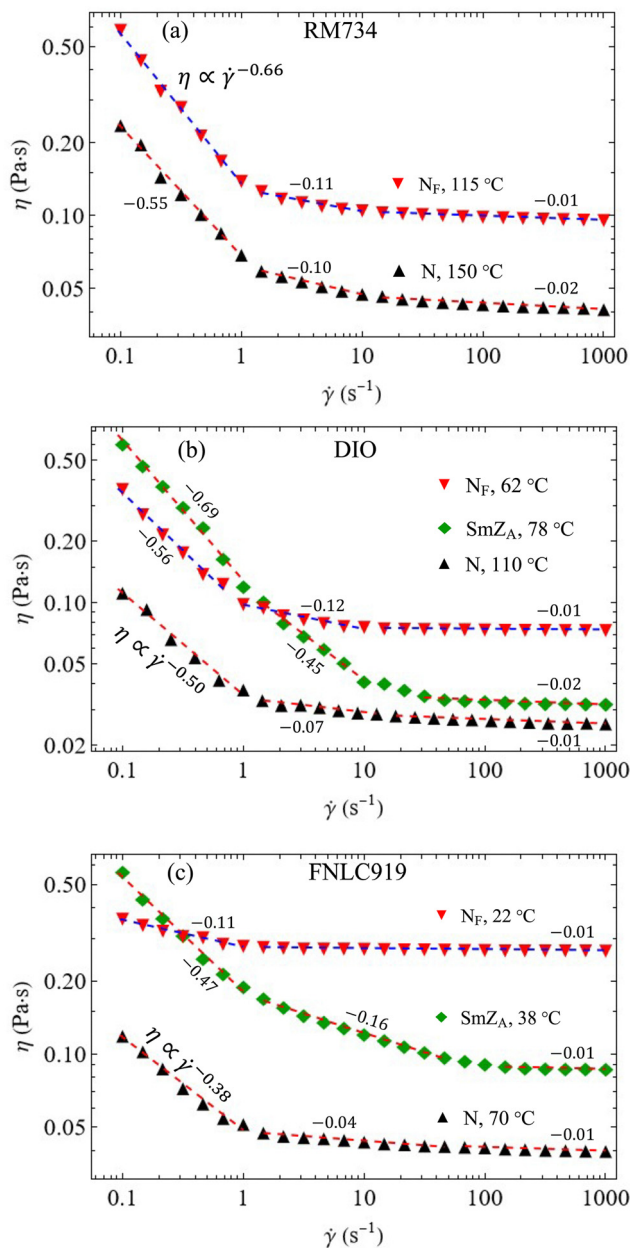


Fig. 4 Variation of effective shear viscosity η of RM734 (a), DIO (b), and FNLC919 (c) with different shear rates $\dot{\gamma}$ in the N, N_F and intermediate phases. The red and blue dashed lines are the best fit to the curve, which is done by using a power law for effective shear viscosity.

explanation is that in these polymers, a moderate shear causes tumbling of the director \hat{n} , which results in a less ordered structure than that in the absence of shear. All three explored materials show $N_1 < 0$ in the N and intermediate phases. The N_F phase in all materials exhibits a small negative N_1 at low shear rates $\dot{\gamma} < 10 \text{ s}^{-1}$ and a positive N_1 at $\dot{\gamma} > 100 \text{ s}^{-1}$. The experiment uncovers a dramatic difference in the behavior of N_1 between the paraelectric, antiferroelectric and ferroelectric phases. The observed behavior does not fully fit the previously developed models. The available models that explain $N_1 < 0$ in the N phase assume that \hat{n} is in the shear plane; as will

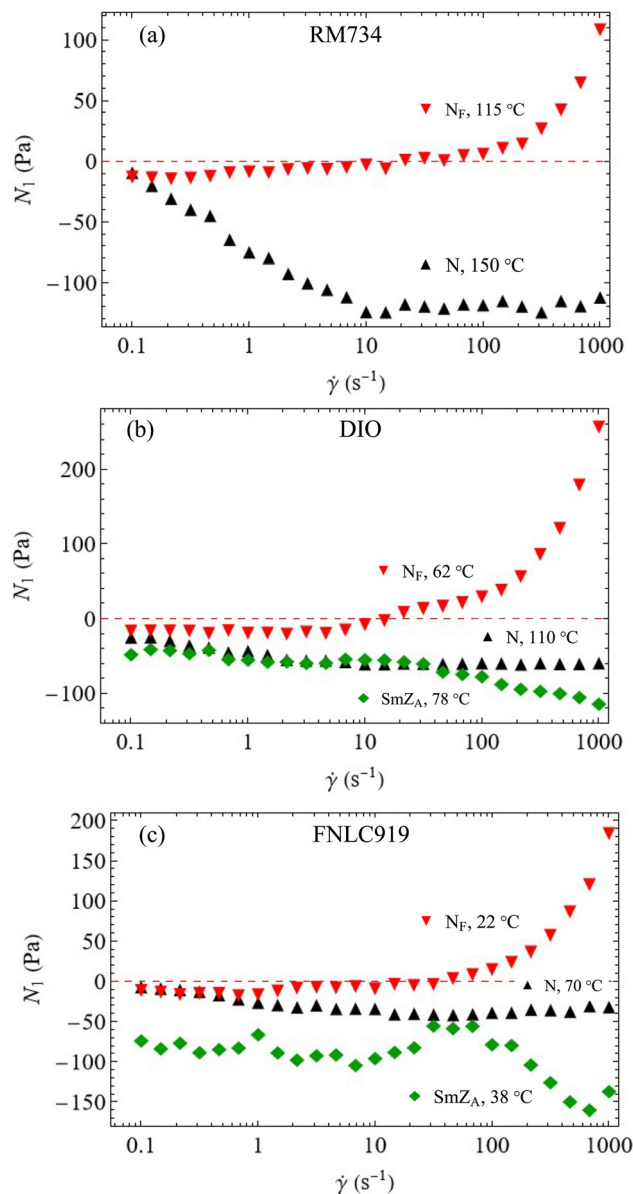


Fig. 5 First normal stress difference N_1 of RM734 (a), DIO (b), and FNLC919 (c) as a function of shear rate $\dot{\gamma}$ in the N, N_F and intermediate phases.

be clear in the next section, this assumption is valid only at $\dot{\gamma} \leq 100 \text{ s}^{-1}$; at higher rates, polydomain structures form and at $\dot{\gamma} \geq 1000 \text{ s}^{-1}$, \hat{n} realigns along the vorticity axis. When \hat{n} deviates from the shear plane, forming twisted structures,⁶⁰ the available models predict $N_1 > 0$.^{61,62}

For the case of the N_F phase, a small negative N_1 at low $\dot{\gamma}$ can be tentatively attributed to misalignments in the shear plane, but the optical retardance discussed below does not support this idea. At high rates, a positive N_1 can be associated with a better alignment of the director as compared to the shear-free case; however, there is no significant difference in the optical retardance of the samples in these two regimes. The intriguing behavior of N_1 deserves further study.



3.4 Realignment regimes of the N phase under shear

3.4.1 RM734. The response of the N phase at $T = 150\text{ }^{\circ}\text{C}$ to the shear exhibits three regimes, depending on the shear rate. As $\dot{\gamma}$ increases from 0.75 s^{-1} to 80 s^{-1} , initially misaligned \hat{n} progressively realigns toward the shear plane, as evidenced by the PPM, Fig. 6a, and by observations with the optical compensator, Fig. S1. The realignment of \hat{n} towards the shear plane indicates the flow-aligning character of RM734 in the N phase at low $\dot{\gamma}$. Similar behavior is observed in conventional N materials, such as MBBA, 5CB, E7, and MLC 7026.^{13,14,63,64} As \hat{n} remains in the shear plane, the elastic deformations are predominantly of the splay-bend type. The relative importance of the viscous and elastic torques is expressed by the Ericksen number $Er = \eta\dot{\gamma}h^2/K_1$, where K_1 is the splay elastic constant. Using typical values $K_1 \approx 2\text{ pN}$ at $150\text{ }^{\circ}\text{C}$,³⁴ cell thickness $h = 10^{-5}\text{ m}$, and effective viscosity $\eta = 0.04\text{ Pa s}$ at $150\text{ }^{\circ}\text{C}$, Fig. 3a, one estimates $Er \approx [2\text{ s}]\dot{\gamma}$. In other words, the viscous torques prevail over the elastic ones at any $\dot{\gamma} > 0.5\text{ s}^{-1}$. As the shear rate increases to $\dot{\gamma} = 120\text{ s}^{-1}$, flow starts to produce disclination loops that multiply with the further growth of $\dot{\gamma}$, Fig. 6b and Fig. S2a. In the range $120\text{ s}^{-1} \leq \dot{\gamma} \leq 1000\text{ s}^{-1}$, the director field is highly distorted, forming disclination-infused polydomains which become more elongated as $\dot{\gamma}$ increases, Fig. 6b and Fig. S2a. The director adopts many different orientations in the plane of the sample. At still higher shear rates, $1000\text{ s}^{-1} \leq \dot{\gamma} \leq 5000\text{ s}^{-1}$, the polydomain textures become progressively

homogeneous with the director \hat{n} gradually realigning towards the vorticity x -axis, as evidenced by the predominance of yellow colors in textural observation with an optical compensator in Fig. 6c and blue colors in Fig. S2b. This regime can be called log-rolling.

3.4.2 DIO and FNLC919. The N phase of DIO and FNLC919 exhibits three similar flow regimes. The director, which is initially misaligned, realigns along the shear plane at low shear rates, $0.75\text{ s}^{-1} \leq \dot{\gamma} \leq 80\text{ s}^{-1}$ in DIO and $0.75\text{ s}^{-1} \leq \dot{\gamma} \leq 50\text{ s}^{-1}$ in FNLC919, Fig. 7a and Fig. S3a, S4 and S5. Intermediate rates bring about polydomain structures, Fig. 7b and Fig. S3b. The highest shear rates, $600\text{ s}^{-1} \leq \dot{\gamma} \leq 1000\text{ s}^{-1}$, reorient \hat{n} along the vorticity x -axis, Fig. 7c and Fig. S3c, d, producing a log-rolling regime.

The observed realignment of the N molecules along the vorticity direction is at odds with the previously reported flow-alignment of the DIO N phase in the shear plane at $\dot{\gamma} > 10^2\text{ s}^{-1}$.¹⁸ Note that the structural analysis in ref. 18 was performed by observations between crossed polarizers; in such a setting, it is difficult to distinguish between two orthogonal directions of the optic axis. Such an ambiguity is removed when the observations are performed with optical compensators, Fig. 6b and c, and in the PPM mode, Fig. 7.

3.5 Realignment regimes of the N_F phase under shear

3.5.1 RM734. The N_F realignment under shear is qualitatively similar to that of the N phase, Fig. 8. At low shear rates

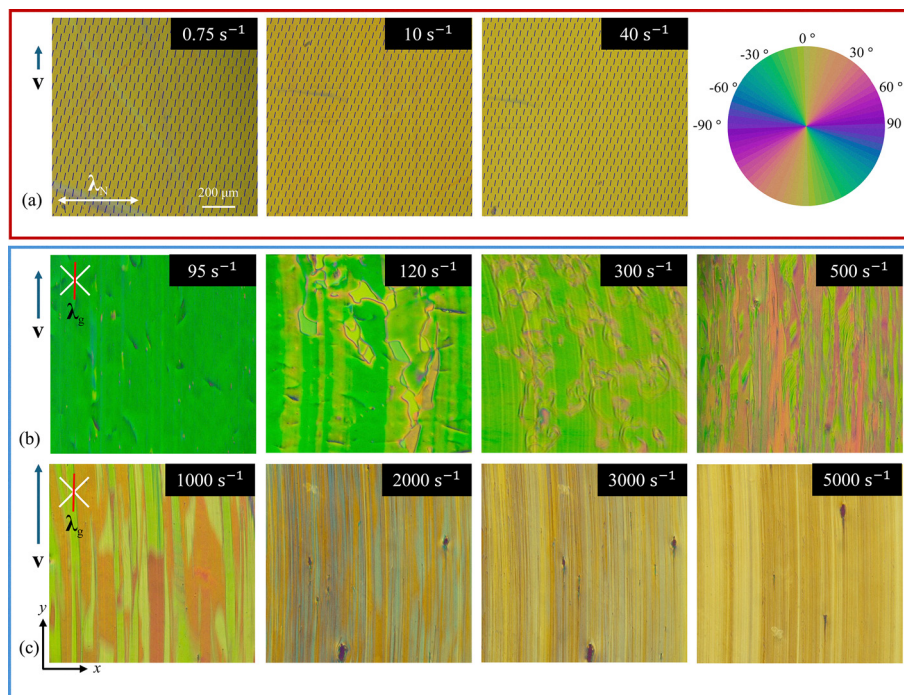


Fig. 6 Structural response of the N phase of RM734 to shear at $T = 150\text{ }^{\circ}\text{C}$ in a $10\text{ }\mu\text{m}$ thick cell revealed in the PPM (a) and POM (b) and (c) modes of observation; the shear rate $\dot{\gamma}$ is indicated on the textures. (a) Flow-alignment at low $\dot{\gamma}$ as revealed by the PPM with the N compensating cell of thickness $9.6\text{ }\mu\text{m}$. The color wheel describes the orientation of the optic axis (director) and the ticks in textures show the local orientation of the director. (b) Polydomain textures at the intermediate $\dot{\gamma}$ as revealed by a POM with crossed polarizers and a FWP compensator. The slow axis λ_g of the FWP shown by a red line is parallel to the shear direction. (c) Progressive realignment of the director towards the vorticity direction at high $\dot{\gamma}$. The slow axis λ_g of the FWP is parallel to the shear direction.



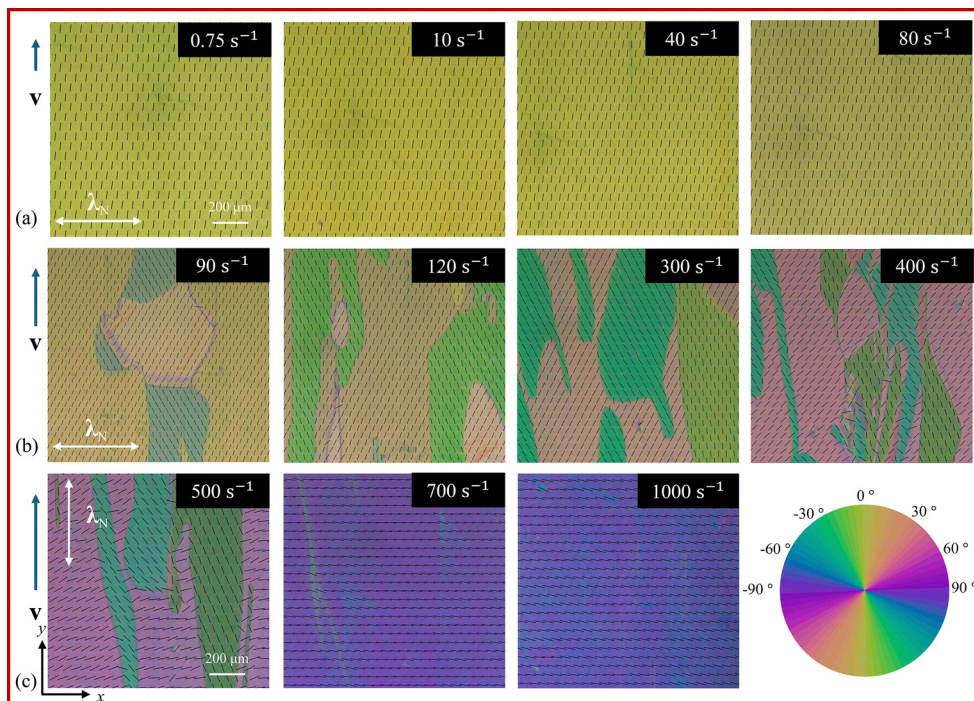


Fig. 7 Structural response of the N phase of DIO to shear at $T = 110\text{ }^{\circ}\text{C}$ in a $10\text{ }\mu\text{m}$ thick cell revealed in the PPM modes of observation; the shear rate $\dot{\gamma}$ is indicated on the textures. (a) Flow-alignment at low $\dot{\gamma}$ as revealed by the PPM with the N compensating cell of thickness $6.8\text{ }\mu\text{m}$. The color wheel describes the orientation of the optic axis (director) and the ticks in textures show the local orientation of the director. (b) Polydomain textures at the intermediate $\dot{\gamma}$. (c) Progressive realignment of the director towards the vorticity direction at high $\dot{\gamma}$.

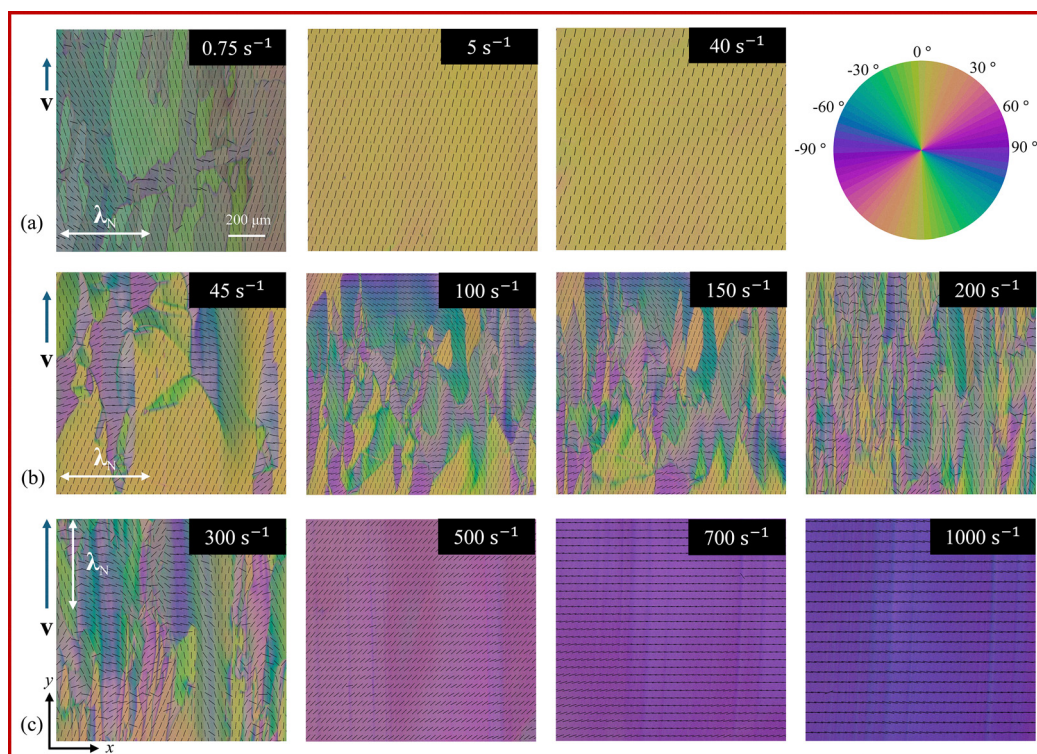


Fig. 8 Structural response of the N_F phase of RM734 to shear at $T = 125\text{ }^{\circ}\text{C}$ in a $10\text{ }\mu\text{m}$ thick cell revealed in the PPM modes of observation; the shear rate $\dot{\gamma}$ is indicated on the textures. (a) Flow-alignment at low $\dot{\gamma}$ as revealed by the PPM with the N compensating cell of thickness $11.1\text{ }\mu\text{m}$. The color wheel describes the orientation of the optic axis (director) and the ticks in textures show the local orientation of the director. (b) Polydomain textures at the intermediate $\dot{\gamma}$. (c) Progressive realignment of the director towards the vorticity direction at high $\dot{\gamma}$.



$\dot{\gamma} \leq 0.75 \text{ s}^{-1}$, the flows are not strong enough to streamline polydomain textures. When $\dot{\gamma}$ increases from 0.75 s^{-1} to 5 s^{-1} , the polydomain texture slowly (within ~ 10 min) transforms into a homogeneous texture with $\hat{\mathbf{n}}$ approaching the shear plane, Fig. 8a. The N_F behaves as a flow-aligning material in the range of $5 \text{ s}^{-1} \leq \dot{\gamma} \leq 40 \text{ s}^{-1}$. Above this range, $45 \text{ s}^{-1} \leq \dot{\gamma} \leq 300 \text{ s}^{-1}$, the flow creates a polydomain structure, Fig. 8b and c. At $\dot{\gamma} > 300 \text{ s}^{-1}$, the shear progressively realigns the optic axis towards the vorticity x -axis, Fig. 8c. The regime is log-rolling when $700 \text{ s}^{-1} \leq \dot{\gamma} \leq 1000 \text{ s}^{-1}$, Fig. 8c.

3.5.2 DIO and FNLC919. The flow aligns $\hat{\mathbf{n}}$ of the N_F in the shear plane at $3 \text{ s}^{-1} \leq \dot{\gamma} \leq 30 \text{ s}^{-1}$, Fig. 9a and Fig. S6a. The intermediate range, $35 \text{ s}^{-1} \leq \dot{\gamma} \leq 350 \text{ s}^{-1}$ in DIO and $35 \text{ s}^{-1} \leq \dot{\gamma} \leq 260 \text{ s}^{-1}$ in FNLC919, produces polydomain textures; the number of domains increases with $\dot{\gamma}$, Fig. 9b and Fig. S6b. The director progressively turns towards the vorticity direction at $\dot{\gamma} > 350 \text{ s}^{-1}$ in DIO and $\dot{\gamma} > 260 \text{ s}^{-1}$ in FNLC919, Fig. 9c and Fig. S6c. At high shear rates $500 \text{ s}^{-1} \leq \dot{\gamma} \leq 1000 \text{ s}^{-1}$, $\hat{\mathbf{n}}$ is along the vorticity x -axis, Fig. 9c and Fig. S6c.

The observed realignment of the N_F molecules along the vorticity direction contradicts the previously reported flow-alignment of the DIO N_F phase in the shear plane at $\dot{\gamma} > 10^2 \text{ s}^{-1}$.¹⁸ As mentioned before, the textures in Fig. 8 and 9, recorded in the PPM mode, allow one to unambiguously determine the direction of the optic axis and polarization along the vorticity direction.

3.6 Director orientation under shear

To quantify the effect of flows on the director structures, we measure the optical retardance Γ as a function of shear rate in the N and N_F phases, Fig. 10a–c. In a separate experiment, we measured the retardance $\Gamma_{\text{max}} = \Delta n \times h$ of planar cells of the same thickness $h = 10 \mu\text{m}$ as the rheometer gap, in the absence of shear; $\Delta n = n_e - n_o$ is the birefringence of the material and n_e and n_o are the extraordinary and ordinary refractive indices, respectively. The Γ_{max} values are shown as dashed lines in Fig. 10a–c. In the N phase of RM734 at $150 \text{ }^\circ\text{C}$, $\Delta n = 0.195$ (measured at $\lambda = 655 \text{ nm}$)⁶⁵ and $\Gamma_{\text{max}} = 1950 \text{ nm}$. The N phase of DIO and FNLC919 shows $\Gamma_{\text{max}} = 1540 \text{ nm}$ at $110 \text{ }^\circ\text{C}$ and 1500 nm at $65 \text{ }^\circ\text{C}$, respectively ($\lambda = 655 \text{ nm}$), Fig. 11a and b. These values are assumed as corresponding to the director parallel to the bounding plates; the effects of a small (on the order of 1°) pretilt angle are neglected.

3.6.1 N phase. In the N phase of all materials, at low shear rates, $\dot{\gamma} \leq 100 \text{ s}^{-1}$, $\Gamma < \Gamma_{\text{max}}$, Fig. 10a–c. The result is natural for a flow-aligning behavior since $\hat{\mathbf{n}}$ in the shear plane tilts away from the horizontal flow direction by some angle θ . For example, in the flow-aligning nematics MBBA and 5CB, $\theta = (7\text{--}15)^\circ$.^{66–68}

When $\hat{\mathbf{n}}$ makes an angle $\theta(z)$ with the y -axis in the flow-align-

$$\text{ment regime, } \Gamma = \int_{z=0}^{z=h} \left(\frac{n_e n_o}{\sqrt{n_e^2 \sin^2 \theta(z) + n_o^2 \cos^2 \theta(z)}} - n_o \right) dz.$$

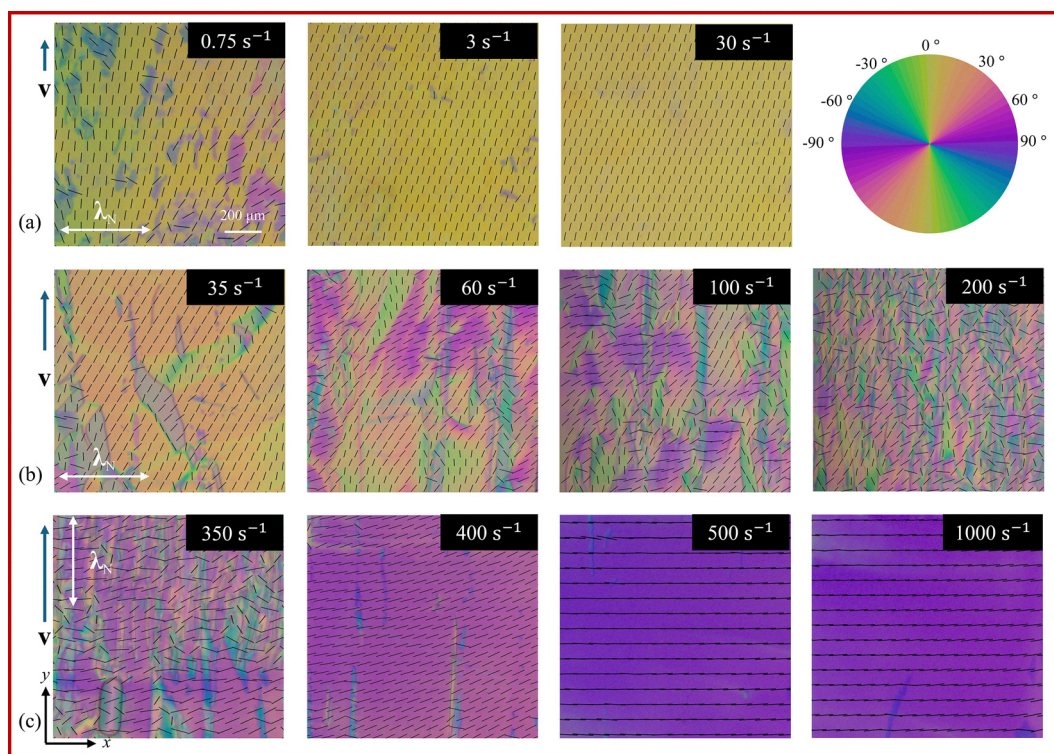


Fig. 9 Structural response of the N_F phase of DIO to shear at $T = 65 \text{ }^\circ\text{C}$ in a $10 \mu\text{m}$ thick cell revealed in the PPM modes of observation; the shear rate $\dot{\gamma}$ is indicated on the textures. (a) Flow-alignment at low $\dot{\gamma}$ as revealed by the PPM with the N compensating cell of thickness $8.4 \mu\text{m}$. The color wheel describes the orientation of the optic axis (director) and the ticks in textures show the local orientation of the director. (b) Polydomain textures at the intermediate $\dot{\gamma}$. (c) Progressive realignment of the director towards the vorticity direction at high $\dot{\gamma}$.



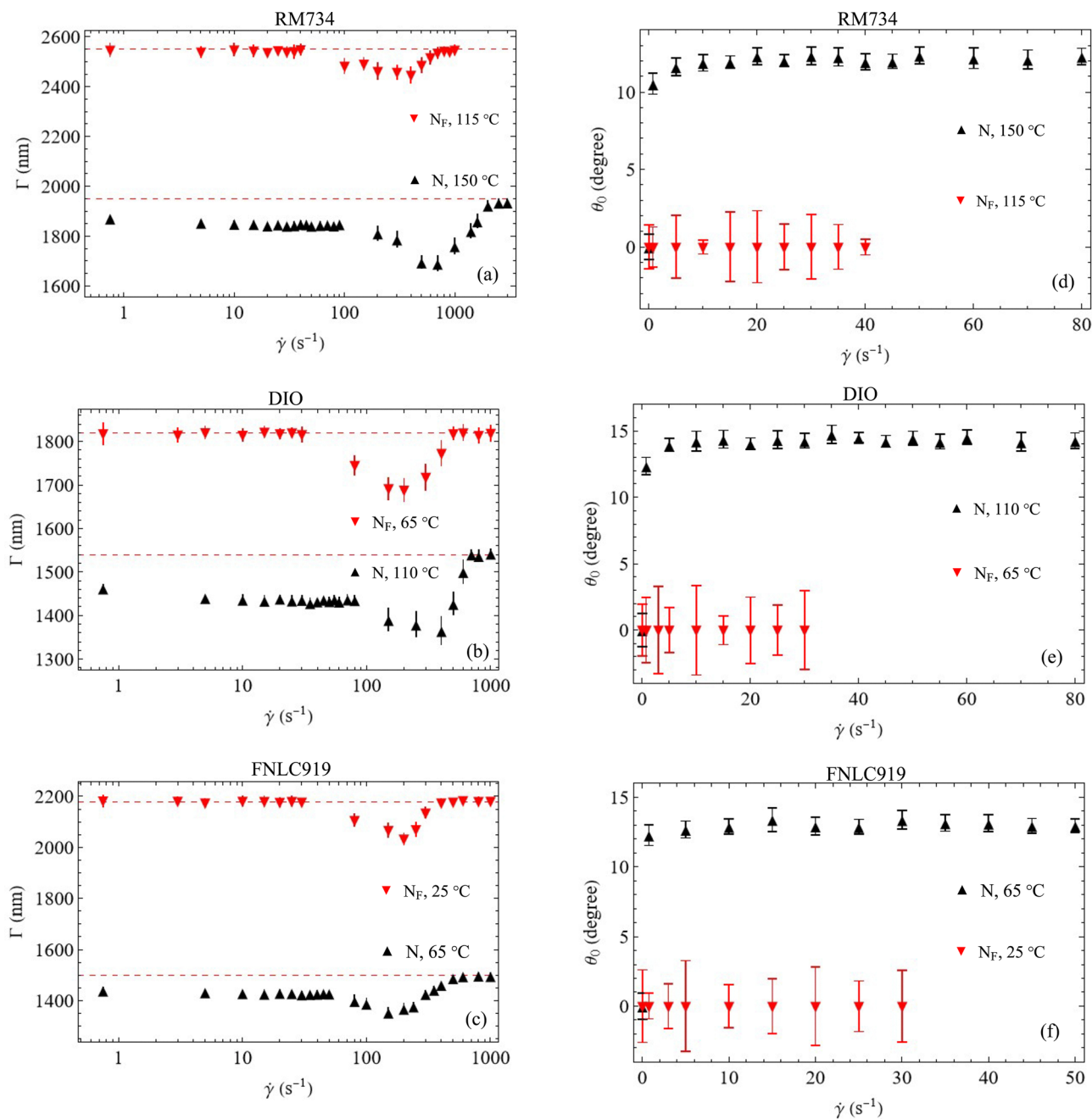


Fig. 10 Shear rate dependence of effective retardance Γ measured using a PolScope Microimager for RM734 (a), DIO (b), and FNLC919 (c) and flow-alignment angle θ_0 of RM734 (d), DIO (e), and FNLC919 (f) in the N and N_F phases; error bars are standard deviations. The dashed red lines indicate the maximum possible retardance Γ_{\max} of a planar cell of $10 \mu\text{m}$ thickness in the absence of shear. Γ of RM734 is measured at $\lambda = 655 \text{ nm}$ in the N phase and $\lambda = 535 \text{ nm}$ in the N_F phase. Γ values of DIO and FNLC919 are measured at $\lambda = 655 \text{ nm}$ in the N and N_F phases.

In a flow-aligning N, it is safe to assume that $\theta(z) = \theta_0$ everywhere except in the thin subsurface layers where the anchoring-imposed orientation persists.^{66–70} We obtained n_e and n_o for RM734 from ref. 16 and measured the values for DIO and FNLC919, Fig. 11c and d, using a wedge-cell interference technique.⁷¹ Using the values of h , n_e and n_o , we determine θ_0 to be in the range $(10\text{--}15)^\circ$ in the flow-aligning regime, Fig. 10d–f. These results are close to what was previously obtained for flow aligning MBBA and 5CB.^{66,67,69}

In the range $100 \text{ s}^{-1} \leq \dot{\gamma} \leq 2000 \text{ s}^{-1}$ for RM734 and $100 \text{ s}^{-1} \leq \dot{\gamma} \leq 500 \text{ s}^{-1}$ for DIO and FNLC919, the polydomain texture yields only some effective Γ since the director field is strongly distorted; twist and light scattering at the defects, Fig. 6b and Fig. S2a, also diminish reliability of the Γ data. At very high shear rates, $\dot{\gamma} > 2000 \text{ s}^{-1}$ for RM734 and $\dot{\gamma} > 500 \text{ s}^{-1}$ for DIO and FNLC919, when \hat{n} is along the vorticity axis, Γ becomes practically equal to Γ_{\max} , Fig. 10a–c. There is thus no significant departure of \hat{n} from the planar state in the log-rolling regime.



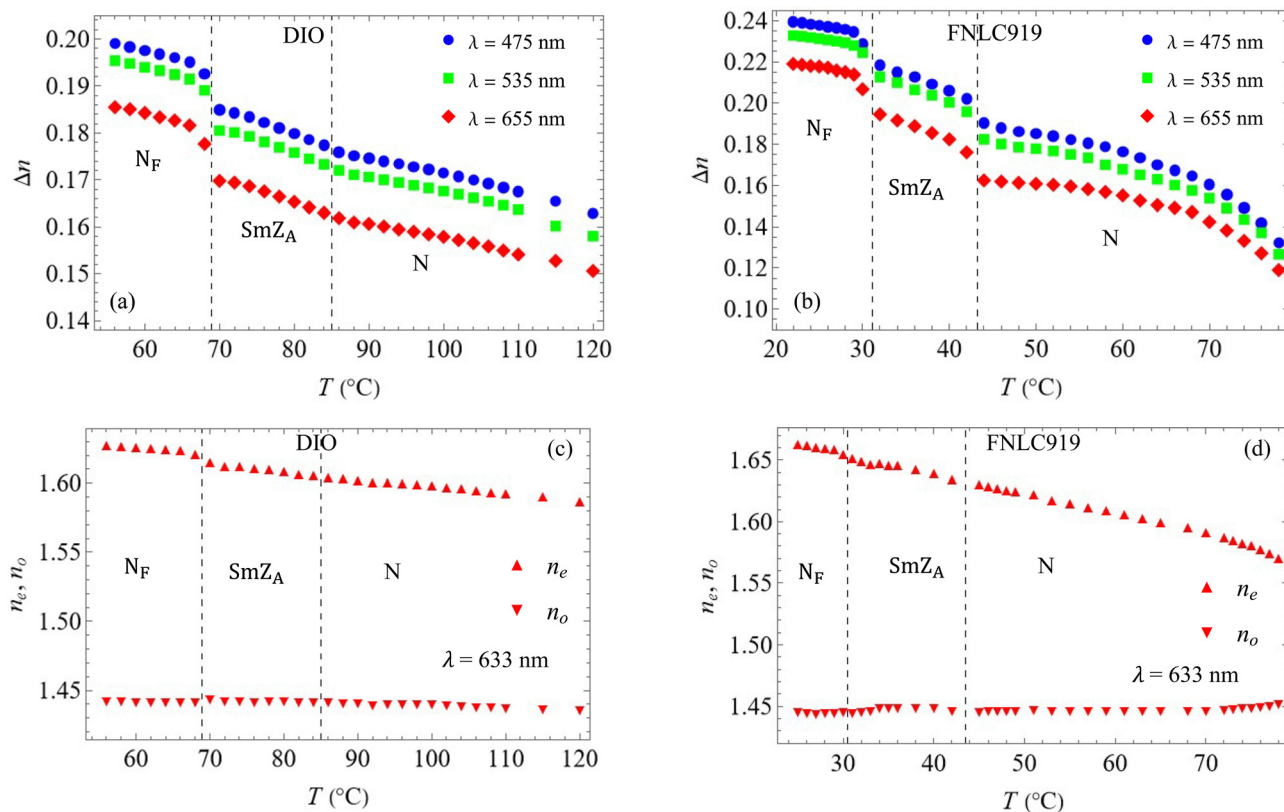


Fig. 11 Temperature dependence of birefringence Δn of DIO (a) and FNLC919 (b) measured using a PolScope Microimager at wavelengths 475 nm, 535 nm, and 655 nm and temperature dependence of the refractive indices of DIO (c) and FNLC919 (d) measured using an interference technique with a wedge cell at 633 nm wavelength.

3.6.2 N_F phase. At low shear rates, in the flow-aligning regime, all three materials in the N_F phase show $\Gamma = \Gamma_{\max}$, Fig. 10a–c. For example, for RM734, $\Gamma = \Gamma_{\max} \approx 2550$ nm (at $\lambda = 535$ nm) for a 10 μm thick homogeneous planar sample at 125 $^\circ\text{C}$.⁶⁵ It means that the optical axis (and thus the polarization \mathbf{P}) does not deviate from the flow direction, in stark contrast to the behavior in the N. The tilt of \mathbf{P} at the bounding surfaces and splay of \mathbf{P} in the bulk create bound charges and increase the electrostatic energy.^{4,72,73} This electrostatic mechanism explains the difference in the flow-alignment of the N_F and N. At very high shear rates, in the log-rolling regime, one also observes $\Gamma = \Gamma_{\max}$, which means that \mathbf{P} avoids tilts and splay but is now oriented along the vorticity x -axis. At intermediate shear rates, the retardance of the N_F phase is much lower than Γ_{\max} , Fig. 10a–c. The reason is the strong director distortions, with a prominent presence of twist^{72–74} along the velocity gradient z -axis. The twist is evident in POM observations with decrossed polarizers, Fig. 12: regions 1, 2, and 3 show different colors when the analyzer is rotated clockwise, Fig. 12a, and then quickly (0.5 s) rotated counterclockwise by the same angle, Fig. 12c.

4. Conclusions

We performed comparative analysis of the paraelectric N, ferroelectric N_F , and antiferroelectric intermediate phases in three

liquid crystal materials, RM734, DIO and FNLC919. As expected, the effective shear viscosity in all phases increases as the temperature decreases, demonstrating the Arrhenius behavior except near the phase transition temperature. All three materials exhibit a strong shear-thinning behavior at low shear rates, $0.1 \text{ s}^{-1} \leq \dot{\gamma} < 10 \text{ s}^{-1}$ and a nearly Newtonian flow behavior at $\dot{\gamma} > 100 \text{ s}^{-1}$. Shear-thinning is especially pronounced in the antiferroelectric phase, which is caused by its layered structure and progressive alignment of the layers under the shear. The same feature produces a dramatic difference in the effective viscosity measured at a constant temperature for two different shear rates: a low shear rate, 2.5 s^{-1} , does not align the layers well and the effective viscosity is very high, while a high shear rate of 500 s^{-1} aligns the layers well and the effective viscosity becomes even lower than that of the N and N_F phases, Fig. 3b and c.

The first-normal stress difference N_1 shows an intriguing behavior, being negative in the N and intermediate phases, but changing from a small negative to a large positive value in the N_F phase as $\dot{\gamma}$ increases. This behavior does not fit the available models developed for the N phase.

The structural response to the shear in both N and N_F shows three distinct regimes. (I) Flow-alignment at low shear rates $\dot{\gamma} < 10^2 \text{ s}^{-1}$, with the director in the shear plane, making an angle $(10\text{--}15)^\circ$ with the flow direction in the N and 0° in the N_F . (II) Polydomain textures with strong director deformations, including twists, at intermediate shear rates. (III) Log-rolling



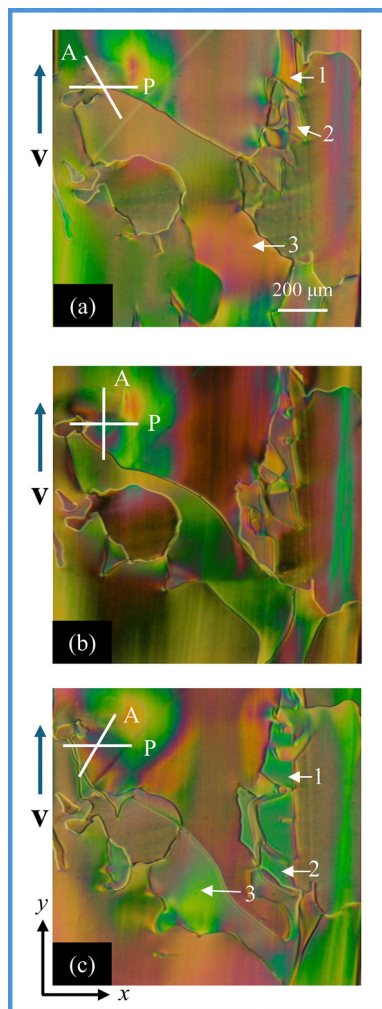


Fig. 12 Twisted domains of the sheared sample RM734 in the N_F phase at $\dot{\gamma} = 100 \text{ s}^{-1}$. [(a)–(c)] POM textures with the polarizer and analyzer making angles of 120° , 90° , and 60° , respectively. $10 \mu\text{m}$ thick sample and $T = 125^\circ\text{C}$.

at high shear rates $\dot{\gamma} > 10^3 \text{ s}^{-1}$, in which the director in the N and the polarization \mathbf{P} in the N_F are parallel to the vorticity direction. The absence of tilts and splay deformations in the flow-aligning and log-rolling regimes is rooted in the electrostatic properties of the N_F phase, which avoids creation of surface and bulk space charges. The uncovered rheological properties and structural dynamics under shear would be useful for a better understanding of the N_F materials and their potential applications in microfluidic devices.

Author contributions

A. C. D. performed the experiments, analyzed the data, and wrote the manuscript. S. P. performed the refractive index measurements, analyzed the results, and contributed to writing that section. O. D. L. conceived the idea, supervised the overall research, and contributed to writing the manuscript.

Conflicts of interest

There are no conflicts to declare.

Data availability

All data needed to evaluate the conclusions in the paper are present in the article and in the supplementary figures. The datasets generated during and/or analyzed during this study are available from the corresponding author on request.

Supplementary information (SI) is available. See DOI: <https://doi.org/10.1039/d5sm01207d>.

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References

- 1 R. J. Mandle, S. J. Cowling and J. W. Goodby, *Phys. Chem. Chem. Phys.*, 2017, **19**, 11429–11435.
- 2 H. Nishikawa, K. Shiroshita, H. Higuchi, Y. Okumura, Y. Haseba, S. I. Yamamoto, K. Sago and H. Kikuchi, *Adv. Mater.*, 2017, **29**, 1702354.
- 3 N. Sebastián, L. Cmok, R. J. Mandle, M. R. de la Fuente, I. D. Olenik, M. Copic and A. Mertelj, *Phys. Rev. Lett.*, 2020, **124**, 037801.
- 4 X. Chen, E. Korblova, D. P. Dong, X. Y. Wei, R. F. Shao, L. Radzihovsky, M. A. Glaser, J. E. MacLennan, D. Bedrov, D. M. Walba and N. A. Clark, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 14021–14031.
- 5 P. G. de Gennes and J. Prost, *The Physics of Liquid Crystals*, Clarendon Press, Oxford, 1993.
- 6 M. Kleman and O. D. Lavrentovich, *Soft matter physics: an introduction*, Springer, 2003.
- 7 A. N. Beris and B. J. Edwards, *Thermodynamics of flowing systems: with internal microstructure*, Oxford University Press, New York, 1994.
- 8 R. G. Larson, *The structure and rheology of complex fluids*, Oxford University Press, New York, 1999.
- 9 P. Oswald and P. Pieranski, *Nematic and cholesteric liquid crystals: concepts and physical properties illustrated by experiments*, Taylor & Francis, Boca Raton, 2005.
- 10 T. Carlsson and K. Skarp, *Liq. Cryst.*, 1986, **1**, 455–471.
- 11 P. Pieransk and E. Guyon, *Phys. Rev. A: At., Mol., Opt. Phys.*, 1974, **9**, 404–417.
- 12 I. Zuniga and F. M. Leslie, *Liq. Cryst.*, 1989, **5**, 725–734.
- 13 D. J. Graziano and M. R. Mackley, *Mol. Cryst. Liq. Cryst.*, 1984, **106**, 103–119.
- 14 P. Mather, D. Pearson and R. Larson, *Liq. Cryst.*, 1996, **20**, 527–538.



- 15 P. T. Mather, D. S. Pearson and R. G. Larson, *Liq. Cryst.*, 1996, **20**, 539–546.
- 16 B. Basnet, S. Paladugu, O. Kurochkin, O. Buluy, N. Aryasova, V. G. Nazarenko, S. V. Shiyanovskii and O. D. Lavrentovich, *Nat. Commun.*, 2025, **16**, 1444.
- 17 M. P. Kumar, J. Karcz, P. Kula, S. Karmakar and S. Dhara, *Phys. Rev. Appl.*, 2023, **19**, 044082.
- 18 H. Nishikawa, P. Salamon, M. T. Máthé, A. Jákli and F. Araoka, *Giant*, 2025, **22**, 100356.
- 19 J. S. Yu, J. H. Lee, J. Y. Lee and J. H. Kim, *Soft Matter*, 2023, **19**, 2446–2453.
- 20 A. Manabe, M. Bremer and M. Kraska, *Liq. Cryst.*, 2021, **48**, 1079–1086.
- 21 S. Brown, E. Cruickshank, J. M. D. Storey, C. T. Imrie, D. Pocięcha, M. Majewska, A. Makal and E. Gorecka, *Chem-PhysChem*, 2021, **22**, 2506–2510.
- 22 A. Erkoreka, A. Mertelj, M. Huang, S. Aya, N. Sebastián and J. Martínez-Perdiguero, *J. Chem. Phys.*, 2023, **159**, 184502.
- 23 P. Nacke, A. Manabe, M. Klasen-Memmer, X. Chen, V. Martínez, G. Freychet, M. Zhernenkov, J. E. Maclennan, N. A. Clark, M. Bremer and F. Giesselmann, *Sci. Rep.*, 2024, **14**, 4473.
- 24 J. Thoen, G. Cordoyiannis, E. Korblova, D. M. Walba, N. A. Clark, W. Jiang, G. H. Mehl and C. Glorieux, *Phys. Rev. E*, 2024, **110**, 014703.
- 25 Y. H. Song, J. X. Li, R. L. Xia, H. Xu, X. X. Zhang, H. Y. Lei, W. F. Peng, S. Q. Dai, S. Aya and M. J. Huang, *Phys. Chem. Chem. Phys.*, 2022, **24**, 11536–11543.
- 26 J. Karcz, N. Rychlowicz, M. Czarnecka, A. Kocot, J. Herman and P. Kula, *Chem. Commun.*, 2023, **59**, 14807–14810.
- 27 E. Cruickshank, P. Rybak, M. M. Majewska, S. Ramsay, C. Wang, C. H. Zhu, R. Walker, J. M. D. Storey, C. T. Imrie, E. Gorecka and D. Pocięcha, *ACS Omega*, 2023, **8**, 36562–36568.
- 28 C. J. Gibb, J. Hobbs, D. I. Nikolova, T. Raistrick, S. R. Berrow, A. Mertelj, N. Osterman, N. Sebastian, H. F. Gleeson and R. J. Mandle, *Nat. Commun.*, 2024, **15**, 5845.
- 29 P. Nacke, R. Tuffin, M. Klasen-Memmer, P. Rudquist and F. Giesselmann, *Sci. Rep.*, 2024, **14**, 15018.
- 30 N. Sebastián, M. Copic and A. Mertelj, *Phys. Rev. E*, 2022, **106**, 021001.
- 31 H. Nishikawa, Y. Okumura, D. Kwaria, A. Nihonyanagi and F. Araoka, *Adv. Mater.*, 2025, **37**, 2501946.
- 32 X. Chen, V. Martínez, E. Korblova, G. Freychet, M. Zhernenkov, M. A. Glaser, C. Wang, C. H. Zhu, L. Radzihovsky, J. E. Maclennan, D. M. Walba and N. A. Clark, *Proc. Natl. Acad. Sci. U. S. A.*, 2023, **120**, e2217150120.
- 33 P. M. Rupnik, E. Hanzel, M. Lovsin, N. Osterman, C. J. Gibb, R. J. Mandle, N. Sebastián and A. Mertelj, *Adv. Sci.*, 2025, **12**, 2414818.
- 34 A. Mertelj, L. Cmok, N. Sebastián, R. J. Mandle, R. R. Parker, A. C. Whitwood, J. W. Goodby and M. Copic, *Phys. Rev. X*, 2018, **8**, 041025.
- 35 Z. J. Ma, M. Jiang, A. L. Sun, S. Z. Yi, J. D. Yang, M. J. Huang, S. Aya and Q. H. Wei, *Phys. Rev. Lett.*, 2025, **134**, 238101.
- 36 A. Ghimire, B. Basnet, H. Wang, P. Guragain, A. Baldwin, R. Twieg, O. D. Lavrentovich, J. Gleeson, A. Jákli and S. Sprunt, *Soft Matter*, 2025, **21**, 8510–8522.
- 37 K. Thapa, O. S. Iadlovská, B. Basnet, H. Wang, A. Paul, J. T. Gleeson and O. D. Lavrentovich, *Phys. Rev. E*, 2024, **109**, 054702.
- 38 A. Paul, M. Paul, M. Badu, A. Ghimire, N. P. Dhakal, S. Sprunt, A. Jákli and J. T. Gleeson, *Materials*, 2025, **18**, 5496.
- 39 M. Rajabi, O. Lavrentovich and M. Shribak, *Liq. Cryst.*, 2023, **50**, 181–190.
- 40 M. Shribak, *Sci. Rep.*, 2015, **5**, 17340.
- 41 J. M. Beckers, *Appl. Opt.*, 1971, **10**, 973–975.
- 42 A. B. Mahler, S. McClain and R. Chipman, *Appl. Opt.*, 2011, **50**, 755–765.
- 43 M. Ma, S. G. Li, X. L. Jing and H. L. Chen, *Opt. Eng.*, 2017, **56**, 117109.
- 44 R. Oldenbourg, *Nature*, 1996, **381**, 811–812.
- 45 M. Shribak and R. Oldenbourg, *Appl. Opt.*, 2003, **42**, 3009–3017.
- 46 O. D. Lavrentovich, *Contemp. Math.*, 2012, **577**, 25–46.
- 47 Y. K. Kim, G. Cukrov, J. Xiang, S. T. Shin and O. D. Lavrentovich, *Soft Matter*, 2015, **11**, 3963–3970.
- 48 K. Negita, *Chem. Phys. Lett.*, 1995, **246**, 353–357.
- 49 K. Negita, *J. Chem. Phys.*, 1996, **105**, 7837–7841.
- 50 K. Negita and S. Uchino, *Mol. Cryst. Liq. Cryst.*, 2002, **378**, 103–112.
- 51 X. Chen, E. Korblova, M. A. Glaser, J. E. Maclennan, D. M. Walba and N. A. Clark, *Proc. Natl. Acad. Sci. U. S. A.*, 2021, **118**, e2104092118.
- 52 X. Chen, Z. C. Zhu, M. J. Magrini, E. Korblova, C. S. Park, M. A. Glaser, J. E. Maclennan, D. M. Walba and N. A. Clark, *Liq. Cryst.*, 2022, **49**, 1531–1544.
- 53 K. Negita, M. Inoue and S. Kondo, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2006, **74**, 051708.
- 54 G. Kiss and R. S. Porter, *J. Polym. Sci., Part C*, 1978, **65**, 193–211.
- 55 G. Kiss and R. S. Porter, *Mol. Cryst. Liq. Cryst.*, 1980, **60**, 267–280.
- 56 A. D. Gotsis and D. G. Baird, *Rheol. Acta*, 1986, **25**, 275–286.
- 57 S. G. Baek, J. J. Magda and S. Cementwala, *J. Rheol.*, 1993, **37**, 935–945.
- 58 G. Marrucci and P. L. Maffettone, *Macromolecules*, 1989, **22**, 4076–4082.
- 59 P. K. Currie, *Mol. Cryst. Liq. Cryst.*, 1981, **73**, 1–6.
- 60 T. Tsuji and A. D. Rey, *Phys. Rev. E: Stat. Phys., Plasmas, Fluids, Relat. Interdiscip. Top.*, 1998, **57**, 5609–5625.
- 61 W. H. Han and A. D. Rey, *J. Non-Newtonian Fluid Mech.*, 1993, **48**, 181–210.
- 62 W. H. Han and A. D. Rey, *Phys. Rev. E: Stat. Phys., Plasmas, Fluids, Relat. Interdiscip. Top.*, 1994, **49**, 597–613.
- 63 P. J. M. Vanbrabant, J. Beeckman, K. Neyts, R. James and F. A. Fernandez, *Appl. Phys. Lett.*, 2009, **95**, 151108.
- 64 H. Y. Wang, T. X. Wu, S. Gauza, J. R. Wu and S. T. Wu, *Liq. Cryst.*, 2006, **33**, 91–98.
- 65 P. Kumari, B. Basnet, H. Wang and O. D. Lavrentovich, *Nat. Commun.*, 2023, **14**, 748.



- 66 C. Gahwiller, *Phys. Rev. Lett.*, 1972, **28**, 1554.
- 67 K. Sarp, S. T. Lagerwall, B. Stebler and D. McQueen, *Phys. Scr.*, 1979, **19**, 339–342.
- 68 W. W. Beens and W. H. Dejeu, *J. Chem. Phys.*, 1985, **82**, 3841–3846.
- 69 J. Wahl and F. Fischer, *Mol. Cryst. Liq. Cryst.*, 1973, **22**, 359–373.
- 70 J. A. Muller, R. S. Stein and H. H. Winter, *Rheol. Acta*, 1994, **33**, 473–484.
- 71 J. Kędzierski, Z. Raszewski, M. A. Kojdecki, E. Kruszelnicki-Nowinowski, P. Perkowski, W. Piecek, E. Miszczyk, J. Zieliński, P. Morawiak and K. Ogrodnik, *Opto-Electron. Rev.*, 2010, **18**, 214–218.
- 72 L. Paik and J. Selinger, *Phys. Rev. E*, 2025, **111**, L053402.
- 73 M. O. Lavrentovich, P. Kumari and O. D. Lavrentovich, *Nat. Commun.*, 2025, **16**, 6516.
- 74 P. Kumari, B. Basnet, M. O. Lavrentovich and O. D. Lavrentovich, *Science*, 2024, **383**(6689), 1364–1368.

