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Polar and apolar light-induced alignment of ferroelectric nematics on photosensitive polymer substrates

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Surface alignment of a recently discovered ferroelectric nematic liquid crystal (N_F) is usually achieved using buffed polymer films, which produce a unidirectional polar alignment of the spontaneous electric polarization. We demonstrate that photosensitive polymer substrates could provide a broader variety of alignment modes. Namely, a polyvinyl cinnamate polymer film irradiated by linearly polarized ultraviolet (UV) light yields two modes of surface orientation of the N_F polarization: (1) a planar apolar mode, in which the equilibrium N_F polarization aligns perpendicularly to the polarization of normally impinging UV light; the N_F polarization adopts either of the two antiparallel states; (2) a planar polar mode, produced by an additional irradiation with obliquely impinging UV light; in this mode, there is only one stable azimuthal direction of polarization in the plane of the substrate. The two modes differ in their response to an electric field. In the planar apolar mode, the polarization can be switched back and forth between two states of equal surface energy. In the planar polar mode, the field-perturbed polarization relaxes back to the single photoinduced "easy axis" once the field is switched off. The versatility of modes and absence of mechanical contact make the photoalignment of N_F attractive for practical applications.

Introduction

Alignment of nematic (N) liquid crystals (LCs) is crucial for their applications. While historically the first alignment technique employed mechanical rubbing of glass substrates or of alignment layers such as polymers,[1–3] an exceedingly popular modern technique is photoalignment, in which a photosensitive substrate is exposed to light with a pre-designed polarization, incident angle, dose, and intensity.[4–11] Photoalignment allows one to produce complex patterns of molecular orientation with high spatial resolution,[12,13] controllable strength of surface anchoring,[14,15] dynamic alignment[16] and realignment,[17,18] and the capability to pattern molecular orientations on flexible and curved substrates.[11,19] Photoalignment avoids mechanical contact with the substrate and thus does not induce impurities, electric charges, or mechanical damage to the treated surfaces, unlike conventional rubbing. Under irradiation, a substrate coated with a photoalignment material becomes anisotropic and produces an "easy axis" of nematic alignment, the orientation of which depends on the polarization and incident angle of light, the mechanism of light-induced modifications of the substrate, and the nature of the N material.[9–11] Three types of photosensitive transformations are used to align an N: (i) photoisomerization of compounds such as azobenzene derivatives, (ii) photo-crosslinking and (iii) photodegradation of polymers.[9–11] In most cases, the photoinduced easy axis is perpendicular to the polarization of light.[9–11]

In the literature cited above, photoalignment has been successful in the alignment of the director $\hat{\mathbf{n}}$ of a conventional paraelectric N with the property $\hat{\mathbf{n}} \equiv -\hat{\mathbf{n}}$.[9–11] The prevailing types of alignment are the so-called "planar", in which $\hat{\mathbf{n}}$ makes a small "pretilt" angle ψ with the substrate, $\psi = 1^\circ$–6°, and a tilted alignment, with a higher ψ. Because of the identity $\hat{\mathbf{n}} \equiv -\hat{\mathbf{n}}$, pretilted planar alignment produced by normally incident linearly polarized light is two-fold degenerate and requires an additional step such as oblique irradiation to lift the degeneracy and produce a single "easy axis" of director orientation.[9]

An intriguing question is whether the photoalignment approach can be applied to the recently discovered ferroelectric

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nematic (N_F)[20–23] with polar orientation of molecules carrying large dipole moments, which yields spontaneous macroscopic electric polarization **P**. Spontaneous polarization imposes polar symmetry onto the material properties since the states **P** and −**P** are not equivalent. In the materials explored so far, the vector **P** is collinear with the director $\hat{\mathbf{n}} \equiv -\hat{\mathbf{n}}$ that specifies the average orientation of the long axes of the N_F molecules. The presence of electric polarization **P** makes orientations with a pretilt $\psi \neq 0$ at dielectric substrates questionable since a substrate-piercing polarization deposits a strong surface charge, as studies of mechanically rubbed substrates[24–31] or interfaces with isotropic media suggest.[32–35] A spectacular illustration of avoidance of the normal component of polarization at interfaces with dielectric materials is that in the curved channels with N_F, **P** aligns everywhere tangentially to the spatially varying walls.[36] Nevertheless, there are reports that a homeotropic alignment at the N_F–air can be achieved for an N_F material doped with an ionic polymer.[37] Furthermore, since $\mathbf{P} \neq -\mathbf{P}$, a unidirectional buffing of a polymer aligning layer produces structural polarity of the aligning layer; molecular interactions translate this surface polarity into the bulk polarization.[24,26,29,38,39] For example, the **P** of the N_F phase in the material abbreviated DIO aligns antiparallel to the buffing direction on polyimide substrates,[26,29,40] while **P** of RM734 aligns parallel[41] to this direction.

As compared to alignment by buffing, photoalignment of N_F is much less explored. So far, only photoalignment by azodyes brilliant yellow[42,43] and SD1[44,45] has been explored; in all cases, a normally incident linearly polarized light has been utilized. The mechanism of alignment produced by azodyes is well known[9–11,46] and involves a *trans–cis* photoisomerization of the azobenzene moieties. Under a linearly polarized ultraviolet (UV) irradiation, the *trans–cis* and *cis–trans* isomerization continues till the transition dipole of the molecules becomes orthogonal to the polarization of light. This approach produces two-fold degenerate planar alignment, as the states with **P** and −**P** perpendicular to the light polarization are of equal probability to form.[42–45] Besides this two-fold degeneracy, another disadvantage of the alignment by photoisomerization cited in the literature[9–11] is its instability under heat and light. In fact, our research revealed that brilliant yellow does not align well with the high-temperature N_F phase of material RM734 at temperatures 100 °C and above.

In comparison, photo-crosslinking materials undergoing irreversible photochemical reactions that limit molecular movements show a better prospect for applications.[9,11] Popular materials of this type are cinnamate-based polymers.[6,7,47] In this work, we explore one of these materials, polyvinyl-4(fluorocinnamate) (PVCN-F), introduced by Gerus' and Reznikov's group,[48–54] as a photoalignment layer for the N_F material DIO. DIO, first synthesized by Nishikawa *et al.*,[21] is formed by fluorinated rod-like molecules with strong longitudinal dipole moments. The goal of the study is to explore whether the photoalignment technique can produce unidirectional polar alignment of **P** as opposed to the two-fold degenerate planar alignment. So far, there has been no experimental evidence that such a polar orientation could be created by photoalignment, although such a possibility is expected on the grounds of symmetry.[38] We explore two geometries of irradiation: (1) normal incidence of linearly polarized light and (2) oblique incidence of linearly polarized light. Linearly polarized UV irradiation of PVCN-F induces in-plane anisotropy through a covalent molecular cross-linking and an easy axis perpendicular to the polarization of light. We demonstrate that the linearly-polarized UV-irradiated PVCN-F substrates yield apolar and polar modes of surface alignment depending on the irradiation geometry. In the apolar mode, normally incident light irradiation produces two-fold degenerate alignment, in which the N_F polarization **P** can be switched by an in-plane electric field between two states of the same surface energy. In the polar mode, additional irradiation with oblique incidence yields only one azimuthal direction of the polarization **P**. If **P** is realigned by an external electric field, this polar alignment is restored once the field is removed. The polar photoalignment mode of N_F represents a significant technological advance since this non-contact process avoids problems such as buffing-induced electric charges and impurities, to which N_F is much more sensitive than its paraelectric N counterpart.

Materials and irradiation protocols

We explore the N_F material 2,3′,4′,5′-tetrafluorobiphenyl-4yl 2,6-difluoro-4-(5-propyl-1,3-dioxan-2-yl) benzoate known as DIO,[21] Fig. 1a. On cooling from the isotropic (I) phase, the phase sequence of DIO, synthesized as described by Brown

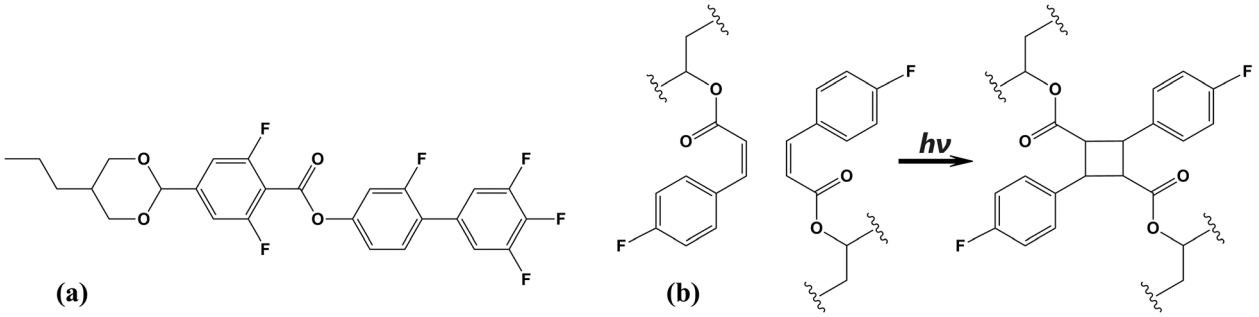

Fig. 1 (a) Chemical structure of the ferroelectric nematic liquid crystal DIO and (b) PVCN-F and its photochemical transformation.

et al.[55] is I – 174 °C – N – 82 °C – SmZ_A – 66 °C – N_F – 34 °C – crystal, where SmZ_A is an antiferroelectric smectic phase.[56]

PVCN-F of molecular weight 30 000 has been synthesized by Dr I. Gerus in the Institute of Bioorganic Chemistry and Petrochemistry, Kyiv, Ukraine.[57] Prior to UV irradiation, the fluorinated cinnamoyl groups of PVCN-F show a small out-of-plane preference in the alignment, most likely caused by the fluorine atom that imparts hydrophobic properties onto these groups.[57] Normally incident linearly polarized light realigns the cinnamoyl groups perpendicularly to the polarization direction **L** of light.[57]

Flat sandwich cells are formed by two glass plates with transparent indium-tin-oxide (ITO) electrodes and spin-coated PVCN-F layers as described by Bugayova *et al.*[52] The thickness of the ITO layer is 185 nm ± 20 nm. Each glass plate with the PVCN-F coating is photoaligned separately before the assembly, by 60 s exposure to the normally impinging linearly polarized UV light from the broadband spectrum source Panacol UV P-280. The UV intensity at the polymer surface is fixed at 10 mW cm^{-2}. Light irradiation of cinnamate-containing polymers causes photodimerization of the cinnamoyl fragments through breaking of the unsaturated C=C bonds and their replacement with the saturated ones, Fig. 1b. As a result of this photoinduced transformation, the number of cinnamoyl groups aligned along the polarization direction **L** is reduced, while their number oriented perpendicularly to **L** increases.[7,54,57,58] As noted by Ichimura,[59] the cinnamate groups experience also *trans–cis* isomerization and align perpendicularly to **L** because of it, similarly to azodyes. Exploration of the particular case of PVCN-F demonstrated that both photoeffects induce the anisotropy[54] that aligns polar N molecules such as 4′-pentyl-4-cyanobiphenyl (5CB) along the direction perpendicular to **L**.[48,51–53,57,59]

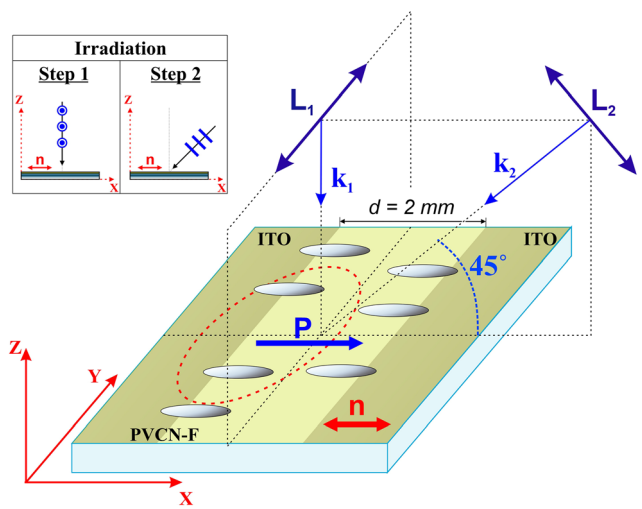

Fig. 2 Schemes of light irradiation of PVCN-F substrate. The ITO strip electrodes on a glass substrate are separated by 2 mm to produce a uniform in-plane field. In the photoalignment mode 1 the wave vector $\mathbf{k}_1$ of the linearly polarized UV light beam is normal to the substrate; the light polarization $\mathbf{L}_1$ is along the *y*-axis. In the photoalignment mode 2, the normal UV irradiation of mode 1 is followed by irradiation with an obliquely incident UV light with linear polarization $\mathbf{L}_2$ in the *xz* plane; the wave vector $\mathbf{k}_2$ is in the *xz* plane and makes an angle 45° with the substrate normal. The red dashed ellipse encloses the typical area of optical microscopy observation.

We use two UV-irradiation protocols, Fig. 2. In protocol 1, the linearly polarized UV light beam is normal to the substrate; the light polarization $\mathbf{L}_1$ is along the *y*-axis. The LC cells in this case are formed by two irradiated PVCN-F coatings facing each other with the directions $\mathbf{L}_1$ being parallel to each other. In the photoalignment protocol 2, the normal UV irradiation of step 1 of a duration 60 s is followed by 300 s irradiation with an obliquely incident UV light with polarization $\mathbf{L}_2$ in the *xz* plane; the wave vector $\mathbf{k}_2$ is in the *xz* plane and makes an angle 45° with the substrate normal, Fig. 2. The cells are assembled with two plates for which the projections of $\mathbf{k}_2$ onto the substrate are parallel to each other. The LC slabs are of a thickness (5.0–5.4) μm. Two stripes of ITO electrodes are located on one glass plate to apply an in-plane electric field, Fig. 2.

Results and discussion

Apolar planar photoalignment mode

Irradiation protocol 1 with 60 s of normally impingent UV light with linear polarization, **L** = (0, ±1, 0), yields a homogeneous alignment of the N phase, with the director $\hat{\mathbf{n}}$ = (±1, 0, 0) perpendicular to **L**. Between two crossed polarizers, the texture is dark when $\hat{\mathbf{n}}$ is along the polarizer or analyzer, Fig. 3a. Cooling down the sample into the N_F phase results in a texture with multiple stripe domains elongated along the *y*-axis, Fig. 3b. The central parts of the domains are dark but the domain walls separating the domains are not extinct. The textures observed with the analyzer rotated counterclockwise from the polarizer, Fig. 3c, and clockwise, Fig. 3d, differ little from each other, and do not produce optical contrast between the neighbouring domains, which indicates that the optical axis and the polarization **P** in the domains do not twist along the normal *z* to the cell. The light transmittance through the domains measured as a function of the angle γ between the polarizer and analyzer shows a minimum at $\gamma = 0$, Fig. 3e. All these observations suggest that the polarization **P** is along the *x*-axis, but alternates in sign, from $\mathbf{P} = P(1, 0, 0)$ in one domain to $\mathbf{P} = P(-1, 0, 0)$ in the next domain. As explained previously,[26,43,60] the formation of uniform domains with anti-parallel **P** is caused by the tendency of the material to reduce the depolarization field which would be significant in the case of a monocrystalline polar alignment of **P**. Alternating uniform stripe domains mitigate the depolarization effect when the slabs are relatively thin, a few microns.[43] An increase of the cell thickness produces twisted domains,[43,61] in which **P** twists around the normal *z* to the cell. In our case, a small (less than 5%) number of such twisted domains is also observed, as verified by their optical activity in observations with uncrossed polarizers.

The measured birefringence of DIO in homogeneous domains is $\Delta n \sim 0.2$ at the wavelength of light 532 nm, which

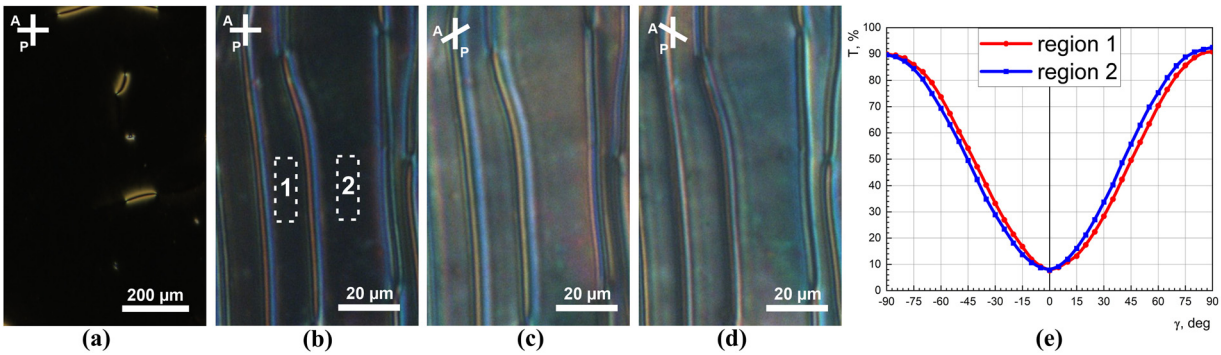

Fig. 3 Apolar planar photoalignment by normally incident linearly polarized UV light. (a) A uniform monodomain texture of the N phase of DIO between two layers of PVCN-F observed between crossed polarizers, 95 °C. (b) Polydomain N_F texture, the polarization directions in adjacent domains are antiparallel to each other, 56 °C. (c) and (d) The same, the analyzer is rotated counterclockwise and clockwise with respect to the analyzer by 40°, respectively. The textures are captured using a green interferometric filter with a centre wavelength λ = 531 nm. (e) Light transmission through a small area in two domains marked by 1 and 2 in (b) as a function of the angle γ between the analyzer and polarizer. Minimum transmittance at γ = 0 indicates that there is no twist along the normal z to the cell.

is consistent with the previously measured values[26,29] indicating that there is no detectable pretilt angle in the N_F phase and the orientation corresponds electrostatically required tangential alignment. The tangential alignment avoids a strong surface bound charge. Even a small tilt $\psi \sim 5^\circ$ of **P** from the xy plane would produce a surface charge density $P_z \sim P\psi \sim 4 \times 10^{-3}$ C m^{-2}, which is larger than the typical surface charge $(10^{-4}$–$10^{-5})$ C m^{-2} of adsorbed ions reported for nematics;[62,63] here, $P \approx 4.4 \times 10^{-2}$ C m^{-2} is the polarization of DIO.[21]

One concludes that protocol 1 of photoalignment with normally incident UV light produces uniformly aligned domains of polarization with apolar surface anchoring along both directions of the x-axis normal to the polarization of light: the polarity of **P** in neighboring domains alternates from **P** = $P(1, 0, 0)$ to **P** = $P(-1, 0, 0)$. The domain structure can be erased by a relatively weak in-plane direct current (DC) electric field **E** = $E(\pm 1, 0, 0)$, where $E \sim 100$ V m^{-1}, applied along the x-axis. The field realigns polarization direction in the domains with **P** antiparallel to **E**; the domains with **P** along **E** do not respond to the field. Upon reversing the field polarity, the response is reversed.

Homogeneous monodomain alignment of **P** can be achieved by cooling the photoaligned sample from the N phase to the N_F phase in the presence of an applied DC electric field $\mathbf{E}_{cooling}$ = 100 V m^{-1}. The uniform alignment forms in the gap between the ITO electrodes but also at the surface of the electrodes; it is preserved when the field is switched-off, Fig. 4a and Video S1. This alignment is not altered by a possible depolarization field since such a field can be reduced by ionic impurities always present in a liquid crystal and the associated torque might not be strong enough to deviate **P** from the easy axis created by light irradiation and selected by $\mathbf{E}_{cooling}$ to be oriented from left to right in Fig. 4a. Repeated application of the field along the polar direction of **P** does not change the texture much, Fig. 4b. The reversal of field polarity realigns **P** by 180° in the gap between the ITO stripes; the texture above the ITO becomes misaligned, with domains of apparent twists of opposite handedness, separated by zigzag domain walls, Fig. 4c. The field

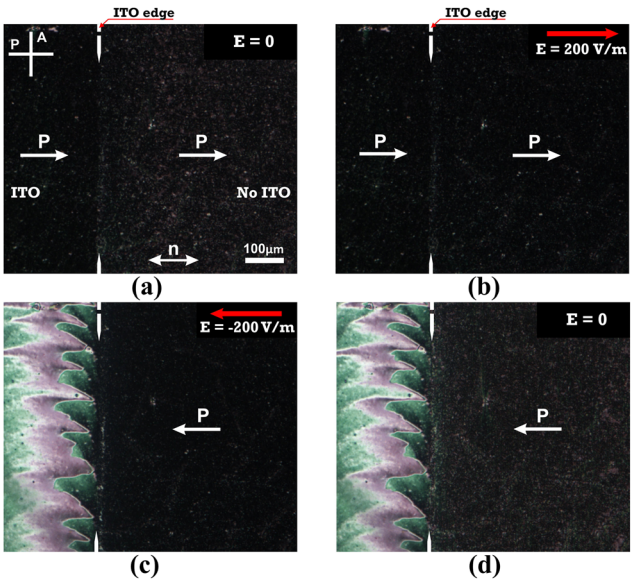

Fig. 4 Apolar photoalignment assisted by the DC electric field. (a) Initial uniform texture of N_F after cooling in the electric field $\mathbf{E}_{cooling}$. Polarization **P** (white arrows) is aligned uniformly in the gap between the ITO stripes and at the ITO surfaces. (b) The N_F texture remains uniform when the electric field **E**↑↑**P** is applied again in the same direction as during the cooling process. (c) The N_F texture after field polarity reversal. The polarization in the gap between the electrodes is uniform but realigned by 180°. At the ITO surfaces, the structure is misaligned. (d) The uniform N_F texture persists after the field is switched off.

required for reversal of **P** is $\mathbf{E}_{reversal} \approx 200$ V m^{-1}. Upon removal of the field $\mathbf{E}_{reversal}$, the alignment remains stable for 2–3 hours, Fig. 4d, after which time the uniform alignment in the gap gradually degrades, apparently because of the frustration between the memory effect of the $\mathbf{E}_{reversal}$ and $\mathbf{E}_{cooling}$ fields.

The polarization realignment from **P** = $(P, 0, 0)$ to **P** = $(-P, 0, 0)$ and back requires electric fields ~100 V m^{-1} for both directions of switching; the difference does not exceed 10 V m^{-1}. This demonstrates that the UV-irradiated PVCN-F coated surfaces produce a strong quadrupolar in-plane N_F alignment; the polar

contribution cannot be detected. This quadrupolar alignment mode is drastically different from the unipolar N_F alignment induced by a polymer substrate with a substantial polar component.[26]

Polar planar photoalignment

The symmetry of the described bistable photoalignment is broken when the UV irradiation is performed in two steps, according to the protocol 2, in which the initial normal incidence of the UV beam is followed by irradiation with an oblique incidence, Fig. 2. This second irradiation is performed with linearly polarized UV light of intensity $I = 10$ mW cm^{-2} for $t = 300$ s. The polarization direction is perpendicular to the light polarization in the first step, Fig. 2. The angle of incidence is 45°. The cells are assembled in a parallel fashion. The material is cooled from the N to the N_F phase in the absence of any electric field. The resulting alignment is uniform with the N_F director perpendicular to the polarization $\mathbf{L}_1$ of normally incident irradiation. Alignment of **P** is polar, along the direction that is opposite to the projection of wavevector $\mathbf{k}_2$ onto the substrate, *i.e.*, along the x axis in Fig. 2: $\mathbf{P} = P(1, 0, 0)$, Fig. 5a. Application of the DC electric field stabilizes the N_F texture within the domains when $\mathbf{E} = E(1, 0, 0)$ is along **P**, Fig. 5b. The opposite direction of $\mathbf{E} = E(-1, 0, 0)$ realigns **P**, Fig. 5c and Video S2. Note that the gap region in Fig. 5c appears to be darker than the gap region in Fig. 5a and b. The apparent reason is that the photocamera is overexposed by the bright texture of the deformed orientation in the electrode area (left hand side of Fig. 5c), which creates a darker appearance of the gap region.

The cell with the two-step irradiation demonstrates a monostable ground state. After removal of the electric field $\mathbf{E} = E(-1, 0, 0)$, **P** smoothly relaxes to the initial state $\mathbf{P} = (P, 0, 0)$, within (10–200) s, depending on the sample, Fig. 5d. The single equilibrium direction of **P** indicates that oblique irradiation induces a unipolar easy axis.

Qualitatively similar polar alignment is observed when the second stage of irradiation is less than 300 *s* or when the substrates are irradiated only with an obliquely incident UV light, linearly polarized or not polarized at all. However, the quality of alignment in these modes is worse than in protocol 2 with two sequential irradiation steps.

The polar in-plane anchoring in N_F cells is described by the anchoring potential

$$W(\varphi) = (1/2)W_Q \sin^2\varphi - W_P(\cos\varphi - 1) \qquad (1)$$

with the two energy minima, one global at $\varphi = 0$, and another local at $\varphi = \pm\pi$.[26,38] Here, $W_Q \geq 0$ and $W_P \geq 0$ are the apolar (quadrupolar or nematic-like) and polar anchoring coefficients, respectively, φ is the angle between **P** and the x axis. When $W_P = 0$, the anchoring is polarity-insensitive, and the minima at $\varphi = 0; \pi$ are equal. This case corresponds to protocol 1 with a single normal irradiation of the PVCN-F substrate, Fig. 3. The oblique irradiation produces a nonzero W_P, thus the alignment at $\varphi = 0$ is energetically preferable to that at $\varphi = \pi$. Since the field-assisted alignment with $\varphi = \pi$ spontaneously relaxes into $\varphi = 0$, this state does not correspond even to a local minimum of $W(\varphi)$, which means that $W_P \geq W_Q$ in the protocol 2 of the photoalignment, Fig. 5. The polar component of photoinduced alignment is very strong as compared to the case of buffing-induced anchoring at a polymer substrate in ref. 26, in which case $W_P \approx 0.1W_Q$.

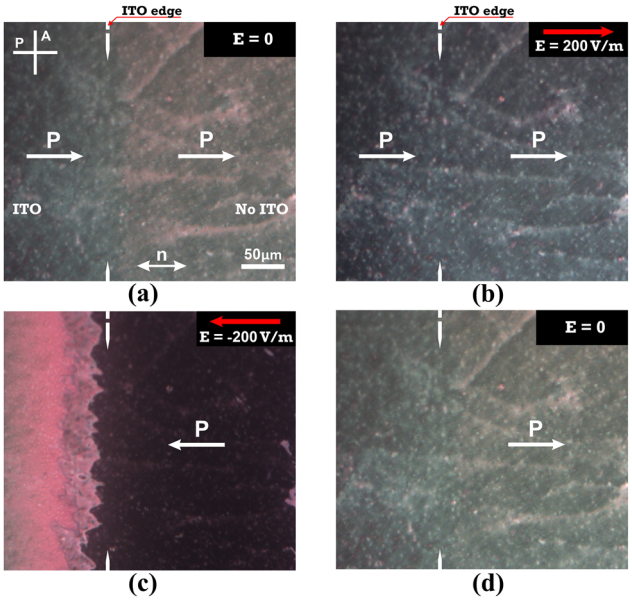

Fig. 5 Mode 2 of photoalignment. (a) A uniform texture of DIO at PVCN-F coated surfaces observed between crossed polarizers in the N_F phase; 56 °C. (b) The electric field is applied along **P**. The polarization **P** within the interelectrode region and outside it coincides. (c) The texture of the N_F phase under the opposite electric field, $\mathbf{E}\uparrow\downarrow\mathbf{P}$. **P** reorients along the applied field **E**. (d) The initial texture is restored when the electric field is removed ($E = 0$). The dashed line indicates the edge of the electrode.

Conclusions

We demonstrate two different photoalignment modes of N_F at a photosensitive polyvinyl cinnamate polymer substrate.

(1) Planar apolar mode 1, in which the normally irradiated PVCN-F surface provides a strong quadrupolar in-plane anchoring of the polarization **P**. **P** aligns perpendicularly to the polarization of normally impinging UV light. The direction of **P** can be switched back and forth between two collinear states of opposite polarity of the same surface energy.

(2) Planar polar mode 2, produced by an additional oblique irradiation with polarized UV light that results in only one equilibrium in-plane alignment of **P**. Oblique irradiation even with unpolarized beam creates a director tilt away from the substrate in photoalignment of the N phase.[9] The reason is that photosensitive substrates develop anisotropy along the direction perpendicular to the polarization of light. For unpolarized obliquely incident light there is only one such anisotropy direction, which is the tilted direction of the wavevector of light. The obliquely irradiated PVCN-F substrate thus tends to

align the nematic director $\hat{\mathbf{n}}$ in a tilted fashion, along the wavevector $\mathbf{k}_2$ of light. However, polar character of the N_F ordering requires **P** to be tangential to any interface with a dielectric medium to avoid deposition of bound charge. As a result, the combination of photoalignment-preferred tilt at the photosensitive substrate, electrostatically required tangential alignment of the N_F, and different affinity of the heads and tails of the polar N_F molecules to the substrate produces a polar in-plane alignment of **P**, either along the projection of $\mathbf{k}_2$ onto the substrate or antiparallel to it. The alignment polarity of **P** should depend on the affinity of tails and heads of the polar molecules to the substrate. Experimental data, Fig. 2, suggest that the negatively charged fluorinated ends of the DIO molecules show a better affinity to the PVCN-F than the aliphatic tails, which justifies the alignment of **P** along the direction antiparallel to the projection of $\mathbf{k}_2$ onto the substrate. The details of nanoscale surface arrangements of N_F molecules require further studies.

Note that the polar anchoring induced in protocol 2 is rather strong, with the corresponding anchoring coefficient W_P in eqn (1) being larger than the quadrupolar counterpart W_Q. For comparison, a unidirectional buffing of polyimide substrates produces noticeably lower values, $W_P \approx 0.1W_Q$.[26] The results offer an intriguing possibility to vary the ratio W_P/W_Q in a broad range.

In this work, we use a direct electric field to assist the alignment of P and to induce its reversal. Supplementing this direct addressing with the second harmonic and polarization currents measurements as described, for example, by Nishikawa *et al.*[64] would be beneficial for deepening our understanding of surface interactions at the N_F-photoaligned substrates.

Both reported schemes of photoalignment of the ferroelectric nematic liquid crystal avoid electrostatic problems often associated with mechanical treatments such as polymer buffing and offer flexibility in the design of different orientational patterns of spontaneous electric polarization with polar and quadrupolar components. The polar alignment in mode 2, besides avoiding a mechanical contact with the substrates, yields a strong polar/quadrupolar ratio of anchoring coefficients. Both modes might be of better utility when compared to the alignment by buffing.

Author contributions

Ruslan Kravchuk: investigation. Oleksandr Kurochkin: investigation and writing – review & editing. Vassili G. Nazarenko: writing – original draft, supervision, resources, project administration, and conceptualization. Volodymyr Sashuk: writing – review & editing and project administration. Mykola Kravets: investigation and chemical synthesis. Bijaya Basnet: investigation. Oleg D. Lavrentovich: writing – review & editing, project administration, and conceptualization. The manuscript was written through contributions of all authors. All authors have approved the final version of the manuscript.

Conflicts of interest

There are no conflicts to declare.

Data availability

The datasets supporting this article are available in Zenodo at **https://doi.org/10.5281/zenodo.17228356**. The dataset has been curated and deposited by the authors.

Supplementary information is available. See DOI: **https://doi.org/10.1039/d5sm00997a**.

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