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# Study of interfacial synergy in strontium-based organic framework/polyaniline/nanoporous graphene ternary composite as positive electrode for battery-supercapacitor hybrid devices

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The escalating global demand for reliable and sustainable energy sources has intensified the need for advanced supercapacitor technologies capable of bridging the gap between capacitors and conventional batteries. However, the development of novel electrode material governing competitive specific capacity, energy and power density is critical. In this work, strontium-benzene tetracarboxylic acid metal–organic framework (Sr-BTCA) incorporated with PANI and NPG was synthesized and evaluated as an electrode material. Their synergistic integration of these components exploits their individual advantages. The electrochemical measurement demonstrates that Sr-BTCA/PANI/NPG based composite delivers specific capacity of 645.5 C g<sup>-1</sup> at 1.5 A g<sup>-1</sup>. Furthermore, the assembled two electrode hybrid device exhibit maximum energy density of 74.5 Wh kg<sup>-1</sup>. Linear and quadratic models were applied on the experimental data to estimate the capacitive and diffusive contributions. The quadratic model provides a better fit on experimental as compared to the linear model which reveals that capacitive-controlled charge storage dominates at higher scan rates, confirming fast electrochemical kinetics. These results highlight the strong synergistic interaction among the MOF, PANI, and NPG components, positioning the Sr-BTCA/PANI/NPG composite as a promising electrode material for high-performance supercapacitor applications.

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## Introduction

The rapid evolution in the energy sector and the accelerating development in the use of portable electronics and electric vehicles has driven the demand for advanced energy technologies.<sup>1</sup> The conventional lithium batteries can achieve remarkable energy densities, yet challenges related to power density motivate for the exploration of alternate storage system such as supercapacitors. Supercapacitors are promising

electrochemical energy storage devices owing to their rapid charge–discharge capability, high power density and excellent operational safety.<sup>2,3</sup> Sodium, lithium, and zinc-based systems offer high energy storage performance but share common limitations such as stability issues and limited cycling durability.<sup>4,5</sup> Although conventional supercapacitors offer many benefits, their lower energy density as compared to the rechargeable batteries prompt a major limitation, motivating intensive research into novel electrode materials capable of achieving superior energy storage performance without compromising power delivery.<sup>6</sup> Supercapacitors store energy through two principal mechanisms, fast and reversible faradaic redox processes, and electric double layer capacitance (EDLC), which arises from the electrostatic accumulation of charge at the electrode–electrolyte interface.<sup>7,8</sup> Advances in electrochemical systems, including biocathodes, electrocatalysts, and porous electrodes, highlight promising opportunities for efficient energy storage. Carbon based materials such as carbon nanotubes, activated carbon, and graphene derivatives are widely employed as composite-forming components in EDLC electrode materials due to their large surface area, chemical stability but their limited pseudocapacitive contribution often

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limit the achievable specific capacitance.<sup>9–11</sup> In contrast, the transition metal compounds and conductive polymers can offer higher capacitance, but they largely suffer from poor electron transport and structural instability during long operational cycles.<sup>12,13</sup>

Several composite based studies reflect the effects of MOF integration with PANI or NPG to significantly improve the performance of the supercapacitor electrode particularly in terms of specific capacity, energy and power density. Such as the enhanced performance exhibited by the PANI/NPG based supercapacitor returned a specific capacitance of  $6.54 \text{ mF cm}^{-2}$  with  $9.11 \text{ mWh cm}^{-2}$  and  $1.56 \text{ W cm}^{-2}$  energy and power density.<sup>2</sup> At  $1 \text{ A g}^{-1}$  current density, specific capacitance of  $504 \text{ F g}^{-1}$  was achieved by Co-MOF/PANI composite.<sup>14</sup> A carbon-based porous framework PC-MOFs/PANI delivered specific capacitance of  $534.16 \text{ F g}^{-1}$  at  $0.2 \text{ A g}^{-1}$ .<sup>15</sup> A sandwiched Zn-MOF/PANI composite resulting a specific capacitance of to  $477 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$ .<sup>16</sup> An Fe-MOF incorporated with PANI (Fe-MOF@PANI) demonstrated a specific capacitance of  $234.6 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$ .<sup>17</sup> A composite electrode material Sm-MOF/rGO/PANI displayed a significantly high specific capacitance of  $1935.6 \text{ F g}^{-1}$  with an energy and power density of  $59.3 \text{ W kg}^{-1}$  and  $581 \text{ Wh kg}^{-1}$  correspondingly.<sup>18</sup> The half-cell electrochemical performance of these materials was evaluated within medium potential window, as higher applied potentials may induce water splitting reactions, leading to electrolyte decomposition and reduced electrochemical stability.<sup>19</sup> Metal-organic frameworks have gained massive attention as an electrode material for supercapacitor due to their highly ordered porous structure, high specific surface area and chemically tunable composition.<sup>20</sup> By coordinating metal nodes with organic ligands provide abundant electroactive sites and adjustable pore structure which facilitate ion diffusion and adsorption.<sup>21</sup> Recently, a variety of MOFs, MOF derived materials and composites have been explored for energy storage exhibiting excellent performance. However, their intrinsically low electrochemical features of pristine MOFs restrict their electron transport leading to adverse effects on rate capability. The strontium-based metal-organic framework (Sr-BTCA), constructed using benzene tetracarboxylic acid (BTCA) as a coordinating ligand, is anticipated to offer exceptional structural stability along with readily accessible metal active sites, underscoring its promising potential as an electrode material for supercapacitor applications. The integration of strontium (Sr) metal centers is expected to enhance both the redox activity and the overall structural stability of the framework. Nevertheless, research into Sr-based MOFs for energy storage applications remains relatively scarce, and their electrochemical properties have yet to be comprehensively and systematically investigated.

Conducting polymers, particularly polyaniline (PANI), have emerged as highly attractive pseudocapacitive materials due to their exceptional capacitance, reversible redox behavior, low cost, and relatively straightforward synthesis procedures.<sup>22</sup> Although PANI benefit from multiple oxidation states that enable efficient faradaic charge storage, but their practical applications are limited by volumetric changes while rapid cyclic often cause mechanical degradation.<sup>23</sup> Hybridization of PANI with porous matrices is an effective approach to overcome these drawbacks

while maintaining its pseudocapacitive performance. Porous, heteroatom-doped carbon materials and graphene-based materials show remarkable potential for energy storage applications.<sup>24</sup> Among these materials, nanoporous graphene (NPG) has gained particular attention owing to its well-interconnected pore structure, which is highly conducive to rapid ion diffusion, offers a high density of electrochemically active sites, and imparts enhanced mechanical and structural stability to the MOF framework.<sup>25,26</sup> In comparison with conventional 2-dimensional (2D) MXene and MOF composite offers superior electrochemical characteristics for electrode materials.<sup>27</sup> Material engineering strategies, particularly interfacial interaction and hybrid composite design, have been widely adopted to modulate the electrochemical properties. The rational integration of MOF, conductive polymer and graphene-based materials provide an effective approach to harness the complementary advantage of individual components. In such hybrid systems, MOF provide metal active sites and porosity, PANI contributes to enhanced pseudocapacitance and NPG promote rapid ion-transport and structural stabilization. Comprehensive investigations into ternary composites incorporating Sr-BTCA, PANI, and NPG remain considerably limited, and the underlying synergistic interactions that collectively dictate the electrochemical performance of such composites warrant further and more rigorous scientific investigation.

In this work, we report the synthesis and electrochemical evaluation of Sr-BTCA/PANI/NPG composite electrode material. The synergistic effects arising from the integration of these components were systematically investigated with respect to specific capacity, energy and power densities. Solvothermal method was followed for the synthesis of Sr-BTCA and Sr-BTCA/PANI/NPG composite. Structural, morphological and electrochemical characterizations were performed by employing SEM, XRD, EDX, CV, GCD and EIS. The Sr-BTCA/PANI/NPG delivered a specific capacity of  $645.5 \text{ C g}^{-1}$  at  $1.5 \text{ A g}^{-1}$ , while the two-electrode based hybrid device exhibited maximum energy and power density of  $74.5 \text{ Wh kg}^{-1}$  and  $3200 \text{ W kg}^{-1}$ . The capacitive and diffusive behaviour governed by the device was estimated through kinetic analysis by applying linear and quadratic models. This study provides valuable insight into the development of multifunctional hybrid electrode material and offers a viable strategy for developing high performance supercapacitors for energy storage.

## Experimental

### Materials

Strontium chloride dihydrate and 1,2,4,5-benzenetetracarboxylic acid were used as precursor materials. PVDF, NMP and acetylene black were utilized for the formation of slurry mixture. Ni foam was used as substrate. DI water and ethanol were used for washing steps.

### Synthesis of Sr-BTCA

Strontium chloride dihydrate ( $\text{SrCl}_2 \cdot 2\text{H}_2\text{O}$ ) was utilized as precursor material and dissolved in DI water. 1,2,4,5-



Benzenetetracarboxylic acid ( $H_4$ BTCA) was dissolved in *N,N*-dimethylformamide (DMF). Metal solution was slowly added to the ligand solution while continuously stirring for 20–30 min to achieve homogeneous mixtures. This mixture was further transferred to a sealed Teflon-lined autoclave for 16 hours at 160 °C. After cooling the solution, the product was obtained by subsequent centrifugation followed by washing, drying and finally material was obtained as powder form.

### Synthesis of Sr-BTCA/PANI/NPG composite

A well-homogenized Sr-BTCA/PANI/NPG composite was obtained by physically mixing the Sr-BTCA, PANI and NPG in (2 : 1 : 1) ratio.

### Preparation of working electrode

Nickle foam was cleaned thoroughly by ethanol, DI water and dilute HCl followed by drying in oven. A homogeneous slurry comprising 80% active material, 10% PVDF binder and 10% acetylene black in NMP was stirred for 6 hours and coated by drop casting on a  $1 \times 1 \text{ cm}^2$  nickel foam substrate and dried. The prepared electrode was subsequently subjected to electrochemical evaluation. The solvothermal preparation of Sr-BTCA and Sr-BTCA/PANI/NPG composite is depicted in Fig. 1.

## Results and discussion

### Morphological and structural analysis

X-ray diffraction (XRD) analysis was performed to determine the crystalline structure of the as prepared material of Sr-BTCA. The characteristic peaks exhibited by the diffraction pattern corresponds to the JCPDS #96-435-0449 indicating the formation of Sr-BTCA framework. The high intensity and sharpness of the

peaks verify that the synthesized material exhibit good crystallinity and highly ordered porous network as shown in Fig. 2(a). For the Sr-BTCA/PANI/NPG, the diffraction pattern of Sr-BTCA were still present which indicate the preservation of the crystalline porous framework after the incorporation of PANI and NPG. The microstructural features and surface morphology of Sr-BTCA and Sr-BTCA/PANI/NPG composite were investigated through scanning electron microscopy (SEM) as shown in Fig. 2. SEM features for Sr-BTCA are represented in Fig. 2(b). It reveals a relatively uniform microcrystalline morphology comprised of rough surface and irregular shaped particles distributed homogeneously indicate successful nucleation and growth of porous framework by solvothermal process. The SEM image for Sr-BTCA/PANI/NPG reflects a relatively distinct morphology as portrayed in Fig. 2(c). A comparatively denser and more interconnected morphology is observed in the composite structure, which strongly indicates that PANI has effectively bridged with the Sr-BTCA framework, facilitating improved interfacial contact and structural integration between the two components, While NPG provide a conductive framework. Th presence of nanoporous graphene and polymeric coatings effectively reduces particle aggregation and enhance interparticle contact. Overall, SEM reflect that the structural integrity of the pristine does not disrupt by the integration of PANI and NPG but instead forms a more compact morphology which is favourable for supercapacitor electrodes to facilitate the transfer kinetics. The elemental composition and of the synthesized Sr-BTCA and Sr-BTCA/PANI/NPG were confirmed through energy-dispersive X-ray spectroscopy (EDX). The EDX spectrum of the pristine Sr-BTCA confirms the presence of Sr, C and O which reflect the successful formation of Sr-BTCA framework without detectable impurities as manifested in Fig. 2(d). The EDX spectrum for the

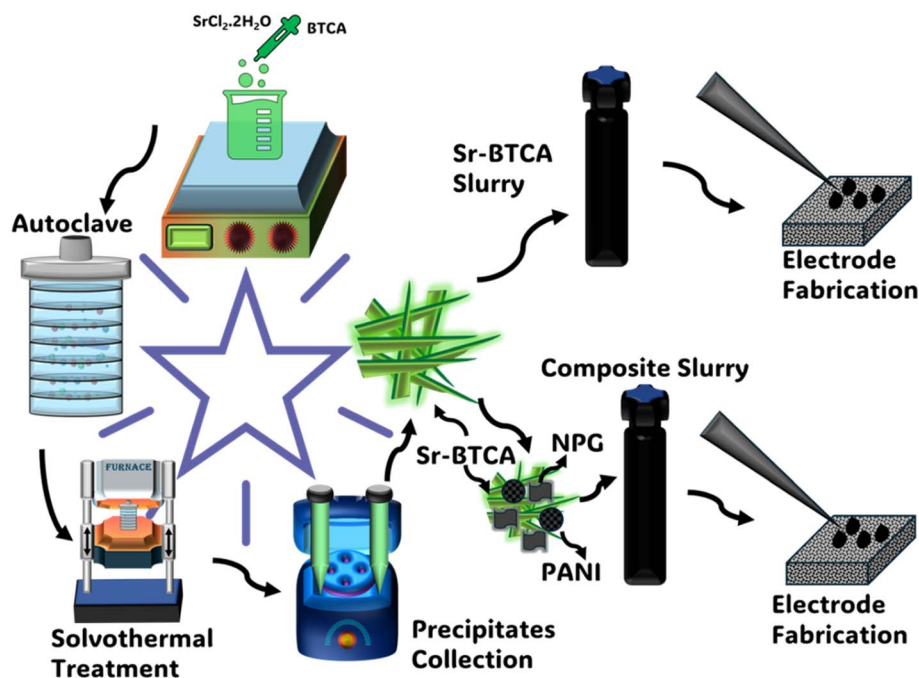


Fig. 1 Schematic illustration for the preparation of Sr-BTCA and Sr-BTCA/PANI/NPG composite.





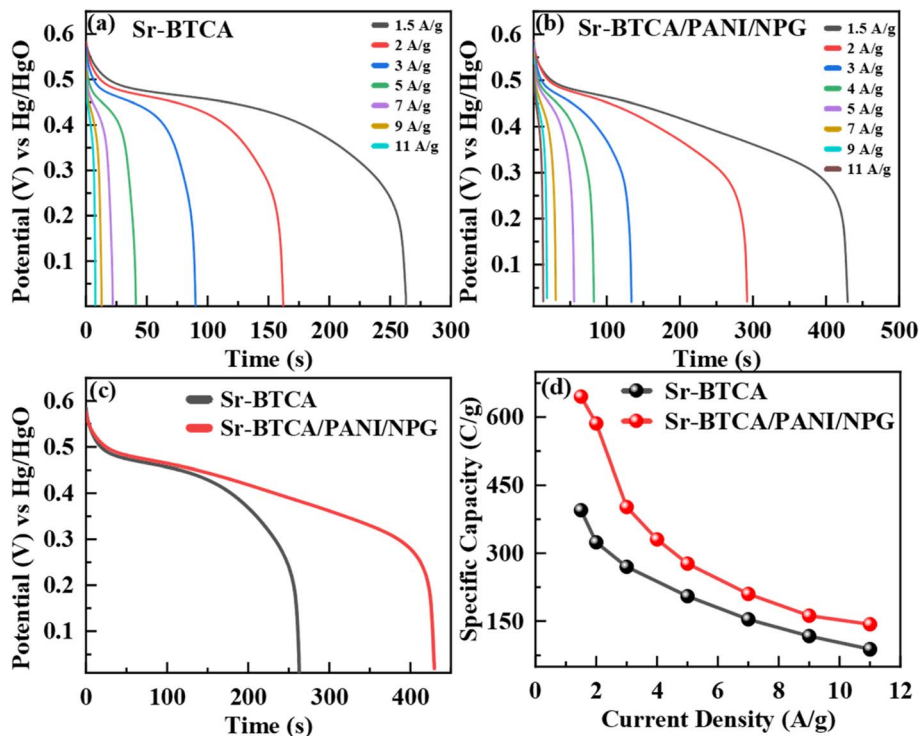


Fig. 4 GCD profiles at multiple current densities for (a) Sr-BTCA and (b) Sr-BTCA/PANI/NPG; (c) comparative GCD at  $1.5 \text{ A g}^{-1}$ , and (d) specific capacities at different  $\text{A g}^{-1}$ .

interaction between components. The following eqn (1) was employed to determine the specific capacity ( $Q_s$ ).

$$Q_s = \frac{I \times \Delta t}{m} \quad (1)$$

where,  $I$  is the applied discharge current,  $\Delta t$  is the discharge time and  $m$  is mass of the active material. The specific capacity manifested by Sr-BTCA/PANI/NPG based composite and pristine Sr-BTCA was  $645.5 \text{ C g}^{-1}$  and  $395 \text{ C g}^{-1}$ , at  $1.5 \text{ A g}^{-1}$  respectively. Electrochemical impedance spectroscopy (EIS) was carried out to investigate the charge transfer resistance of the electrodes. Fig. 5(a) and (b) shows the real and imaginary components of the impedance for Sr-BTCA and Sr-BTCA/PANI/NPG respectively. The Nyquist plots of Sr-BTCA and Sr-BTCA/PANI/NPG are illustrated in Fig. 5(c). The high frequency regions correspond to charge transfer resistance ( $R_{ct}$ ) while low frequency regions indicate Warburg impedance. The increased verticality of the composite and the reduced semicircular

diameter confirmed improved ion-transport for the composite. The value of Equivalent Series Resistance (ESR) for Sr-BTCA and Sr-BTCA/PANI/NPG was determined to be  $0.66 \Omega$  and  $0.46 \Omega$  respectively.

### Two-electrode assembly of the hybrid electrodes

A two-electrode cell was fabricated by employing Sr-BTCA/PANI/NPG composite material and activated carbon (AC) as the positive and negative electrode as displayed in Fig. 6(a). A comparative CV trend for battery grade Sr-BTCA/PANI/NPG and capacitive AC was obtained at  $3 \text{ mV s}^{-1}$  showing their distinct characteristic curve shapes as represented in Fig. 6(b). A CV analysis of the Sr-BTCA/PANI/NPG//AC device was conducted within a potential window of  $1.7 \text{ V}$  across scan rates ranging from  $3$  to  $100 \text{ mV s}^{-1}$  as illustrated in Fig. 6(c). Similarly, a comprehensive galvanostatic charge–discharge (GCD) profiles of the Sr-BTCA/PANI/NPG//AC full device were recorded across

Table 1 Comparison of specific capacity/capacitance, energy density, and power density of the fabricated device with recently reported literature values

Hybrid device	Specific capacity or capacitance	Energy density ( $\text{Wh kg}^{-1}$ )	Power density ( $\text{W kg}^{-1}$ )	Cyclic stability	Ref.
CaCo-MOF@PANI@rGO//AC	$192.8 \text{ C g}^{-1}$	$78.7 \text{ Wh kg}^{-1}$	$3086.9 \text{ W kg}^{-1}$	91.9%	28
MnLiS-MOF/PANI/rGO//AC	$260 \text{ C g}^{-1}$	$48 \text{ Wh kg}^{-1}$	$1600 \text{ W kg}^{-1}$	91%	29
PM/CNT/MOF	$650 \text{ F g}^{-1}$	$45.93 \text{ Wh kg}^{-1}$	$800 \text{ W kg}^{-1}$	72.7%	19
Sm-MOF/rGO/PANI//AC	$218 \text{ F g}^{-1}$	$59.3 \text{ Wh kg}^{-1}$	$59.3 \text{ Wh kg}^{-1}$	91%	18
Sr-BTCA/PANI/NPG	$645.5 \text{ C g}^{-1}$	$74.5 \text{ Wh kg}^{-1}$	$3200 \text{ W kg}^{-1}$	97.1%	This work



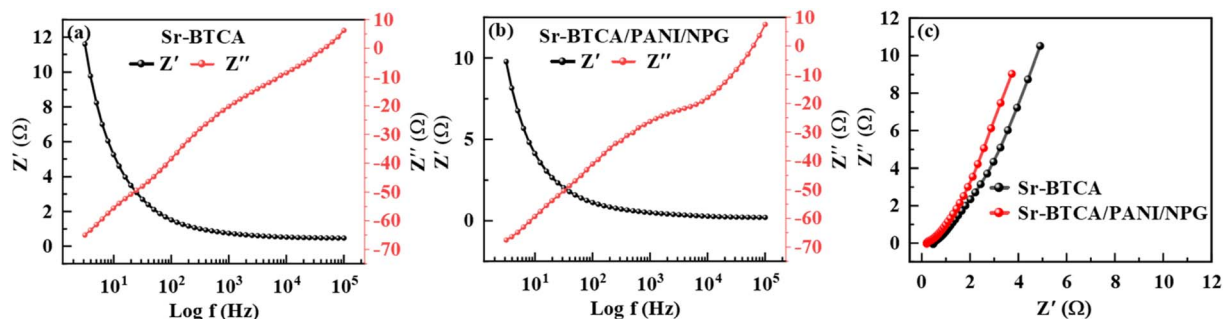


Fig. 5 Real and imaginary components of impedance as a function of frequency for (a) Sr-BTCA (b) Sr-BTCA/PANI/NPG (c) Nyquist plot.

multiple current densities ranging from  $0.6 \text{ A g}^{-1}$  to  $5 \text{ A g}^{-1}$ , within an operating potential window of 1.6 V, as illustrated in Fig. 6(d). The longer discharge time reveals the beneficial synergistic coordination of both electrodes. The specific capacity as a function of current density, derived from the GCD discharge time of the Sr-BTCA/PANI/NPG//AC device, is summarized in Fig. 7(a). The trends of specific capacity were found to be decreasing with increasing current density. The maximum specific capacity of  $335.2 \text{ C g}^{-1}$  was achieved at  $0.6 \text{ A g}^{-1}$ . Fig. 7(b) shows the Ragone plot for Sr-BTCA/PANI/NPG//AC to evaluate the relationship between the energy and power density providing insight into the electrochemical behaviour. The Sr-BTCA/PANI/NPG//AC deliver maximum energy and power density of  $74.5 \text{ Wh kg}^{-1}$  and  $3200 \text{ W kg}^{-1}$ , respectively. A comparative analysis of recently reported studies

and the current study is summarized in Table 1. Following eqn (2) and (3) were followed to calculate the energy ( $E_s$ ) and power density ( $P_s$ ).

$$E_s = \frac{Q_s \times \Delta V}{2 \times 3.6} \quad (2)$$

$$P_s = \frac{E_s \times 3600}{\Delta t} \quad (3)$$

Here,  $\Delta V$  attributes to potential window and  $\Delta t$  is the discharge time. Fig. 7(c) represent the Nyquist plot of the Sr-BTCA/PANI/NPG//AC hybrid device indicating low internal resistance and an overall capacitive behaviour. Fig. 7(d) demonstrates that the hybrid device retains 97.1% of its capacity after 5000 charge-discharge cycles.

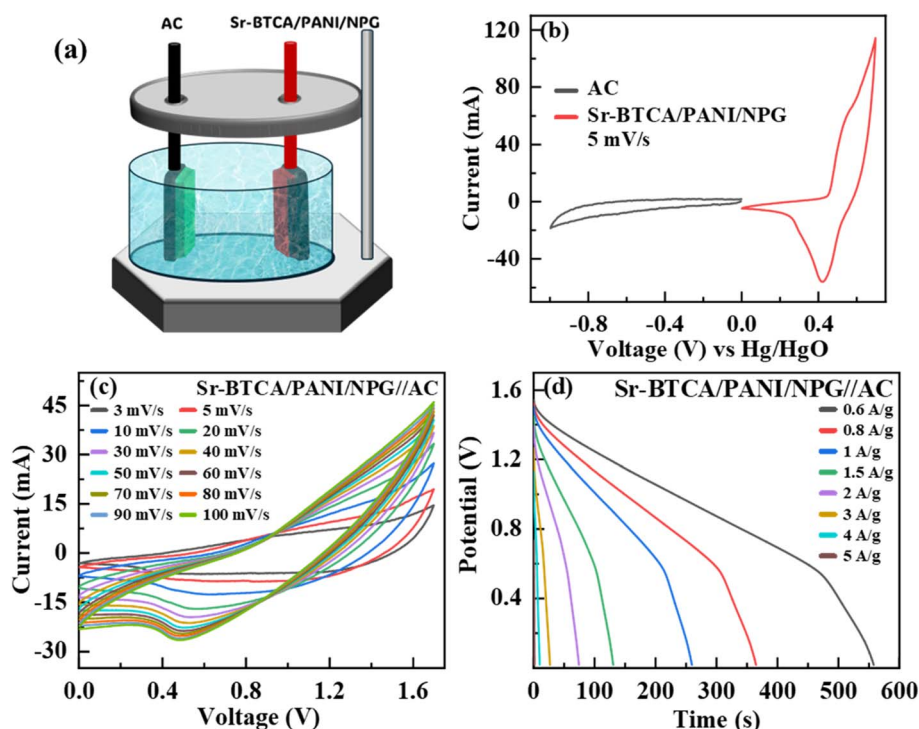


Fig. 6 (a) Sr-BTCA/PANI/NPG//AC hybrid device configuration, (b) comparative CV at  $5 \text{ mV s}^{-1}$ , (c) CV profile, and (d) GCD profile of the hybrid device.



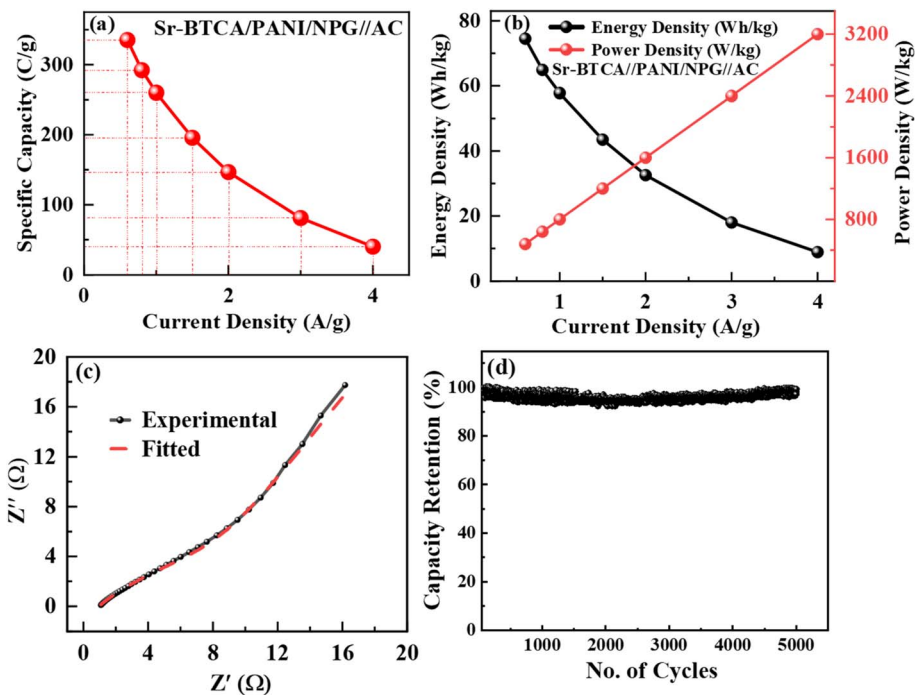


Fig. 7 Sr-BTCA/PANI/NPG//AC hybrid device, (a) specific capacity vs. current density, (b) Ragone plot, (c) impedance components, and (d) cyclic stability.

#### Estimation of the capacitive and diffusive contributions

Fig. 8(a), (b) and (c) display the analysis of capacitive and diffusive controlled contributions at  $5 \text{ mV s}^{-1}$ ,  $50 \text{ mV s}^{-1}$  and

$100 \text{ mV s}^{-1}$  by using both linear and quadratic models. At lower scan rates, the diffusive controlled process dominates while at higher scan rates the capacitive behaviour was determined as

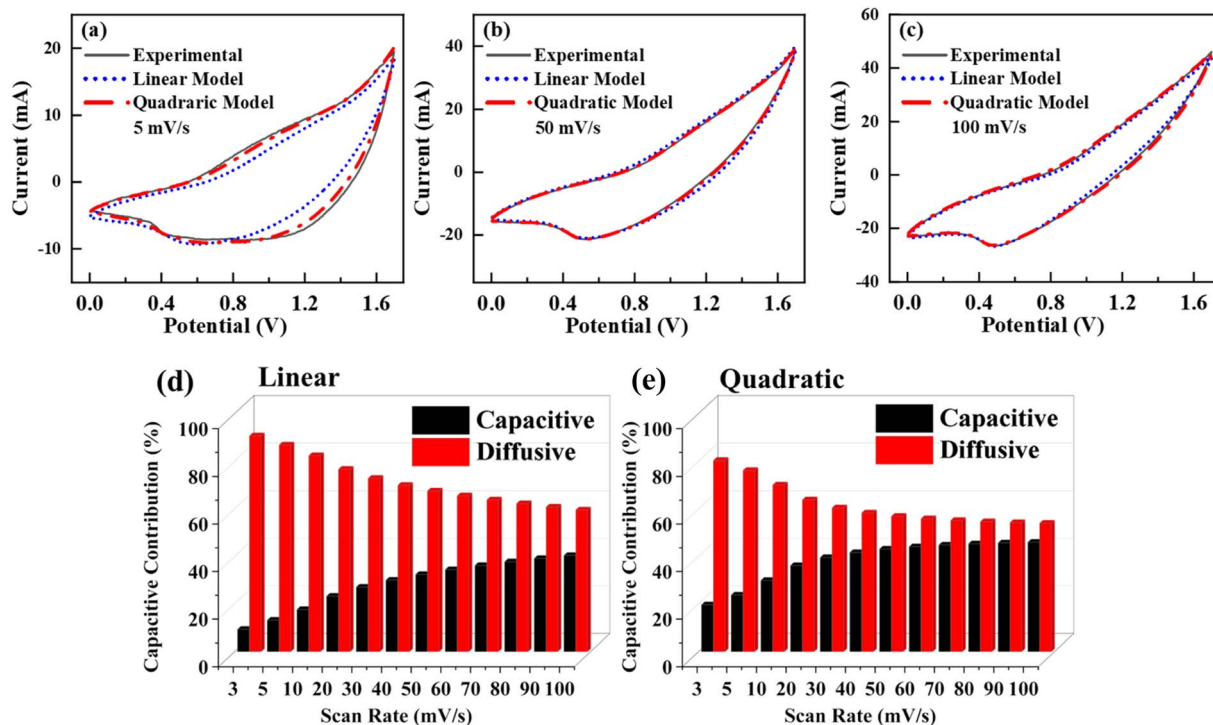


Fig. 8 Capacitive and diffusive contribution separation at (a)  $5 \text{ mV s}^{-1}$ , (b)  $50 \text{ mV s}^{-1}$ , and (c)  $100 \text{ mV s}^{-1}$ , percentage contributions estimated by (d) linear and (e) quadratic models.

dominant process which highlight effective surface utilization and rapid electrochemical kinetics of the hybrid electrode. Fig. 8(d) shows the percentage contribution of the capacitive and diffusive contributions attained through linear model across various scan rates. The diffusive contributions were decreasing with increasing scan rates while capacitive contributions were increasing progressively with increasing scan rates. At  $5 \text{ mV s}^{-1}$  the contributions were determined as highly diffusive. In contrast maximum capacitive behaviour was observed at  $100 \text{ mV s}^{-1}$ . The following eqn (4) was followed to quantify the linear model.

$$I(v) = k_1v + k_2v^{1/2} \quad (4)$$

In this,  $k_1v$  attributes to capacitive controlled contributions while  $k_2v^{1/2}$  refers to diffusive controlled contributions. Percentage capacitive and diffusive contributions were further attained through the quadratic model across various scan rates  $3 \text{ mV s}^{-1}$  to  $100 \text{ mV s}^{-1}$  as represented in Fig. 8(e). The eqn (5) was used for the estimation of quadratic model.

$$I(v) = k_1v + k_2v^{1/2} + k_3v^{3/2} \quad (5)$$

Here,  $k_1v$  defines the combined effects of battery and supercapacitors (EDLC) while  $k_2v^{1/2}$  attributes to the battery type diffusion-controlled process and  $k_3v^{3/2}$  signifies the ohmic resistance. The percentage capacitive components contributions were found to be progressively improved with increasing scan rate. While the diffusion-controlled process diminished likewise. The highest diffusive process was exhibited at  $5 \text{ mV s}^{-1}$  while maximum capacitive behaviour was obtained at  $100 \text{ mV s}^{-1}$  scan rate which confirms effective surface utilization of the hybrid electrode. Overall, the comparison of the two approaches reveals that quadratic model more accurately fit the experimental data than the linear model demonstrating its capability to capture the capacitive and diffusive controlled processes in a better way.

## Conclusion

In summary, Sr-BTCA/PANI/NPG based novel ternary composite was prepared to address the intrinsic limitations of the individual components for supercapacitor applications. The porous framework of Sr-BTCA provides abundant electrochemically active sites, incorporation of PANI improve pseudocapacitive charge storage while NPG serves as an efficient conductive scaffold to improve ion-transport. As a result of the synergistic cooperation, the composite electrode exhibits a specific capacity of  $645.5 \text{ C g}^{-1}$  while the pristine Sr-BTCA delivers  $395 \text{ C g}^{-1}$ . A hybrid two-electrode device Sr-BTCA/PANI/NPG//AC demonstrating a competitive energy density of  $74.5 \text{ Wh kg}^{-1}$  and its maximum power density of  $3200 \text{ W kg}^{-1}$  reflecting its practical applicability. Comprehensive analysis by employing the semi-empirical models reveals that capacitive contributions dominate at higher scan rates confirming efficient electrode utilization and rapid charge transfer. Overall, this study demonstrates strong potential of Sr-BTCA/PANI/NPG composite-based hybrid

electrode for next generation high performance supercapacitor devices.

## Author contributions

Gihan Abdelrahman Hassan Hammouda: project administration, conceptualization, funding acquisition. Ebraheem Abdu Musad Saleh: writing – review & editing, formal analysis. Kashif Mahmud: writing – original draft, data curation, methodology, formal analysis. Muhammad Zahir Iqbal: project administration, writing – review & editing, investigation, software, formal analysis, conceptualization. Abhinav Kumar: formal analysis. Asmaa Fathy Abd El-Aziz Kassem: writing – review & editing, formal analysis. Nusiba Mohammed Modawe Alshik: writing – review & editing, formal analysis. Ankit Dilipkumar Oza: writing – review & editing. Marwa Mostafa Moharam Haqqi Mohammed: writing – review & editing, formal analysis.

## Conflicts of interest

There are no conflicts to declare.

## Data availability

This article contains no supplementary data. All data relevant to this study are included within the manuscript.

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