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# Efficient water electrolysis with Ce-WO<sub>3</sub>@NiCo bifunctional catalysts

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The development of highly efficient non-precious metal electrocatalysts with excellent activity and durability for overall water splitting remains a critical challenge for advancing sustainable hydrogen production. In this work, Ce-WO<sub>3</sub>@NiCo nanoflower architectures are achieved on copper foam (CF) substrates through hydrothermal and electrodeposition methods. Ce doping can improve the activity of catalysts due to the adjustment of the electronic structure, thereby increasing the conductivity. In addition, an electrocatalyst with a three-dimensional (3D) heterostructure can increase the number of active sites, and this multi-component synergistic effect optimizes the electronic structure while regulating the adsorption/desorption process of electrochemical intermediates through interface engineering. The Ce-WO<sub>3</sub>@NiCo catalyst exhibits excellent electrocatalytic activity in the alkaline electrolyte, with overpotentials of 56 mV and 323 mV for the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) respectively, at a current density of 10 mA cm<sup>-2</sup>. In addition, Ce-WO<sub>3</sub>@NiCo requires only 1.624 V to achieve overall water splitting at 10 mA cm<sup>-2</sup>. Meanwhile, it maintains long-term stability for 72 h at current densities of 10, 100, and 300 mA cm<sup>-2</sup>. This research offers a viable approach for constructing highly stable and low-cost self-supported bifunctional electrocatalysts through electronic structure modulation.

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## 1. Introduction

The accelerated depletion of fossil fuel reserves and deterioration of environmental problems have made the development of sustainable and high-efficiency renewable energy urgent.<sup>1</sup> Hydrogen (H<sub>2</sub>) has emerged as a sustainable energy carrier owing to its conspicuous energy density and negligible CO<sub>2</sub> emission, and is considered as an important force for driving the clean energy transition and achieving carbon neutrality.<sup>2-5</sup> Among various production processes for hydrogen energy, electrocatalytic water decomposition, which involves the hydrogen evolution reaction and oxygen evolution reaction, has been widely recognized as a highly promising method due to its capacity to produce high purity hydrogen without carbon dioxide emissions.<sup>6-10</sup> A selection of electrolytes is crucial in the water electrolysis process, and alkaline electrolytes have received significant attention. In alkaline electrolytes, various catalysts exhibit good stability and high electrolysis efficiency, and the alkaline electrolyte causes minimal corrosion of the equipment.<sup>11,12</sup> Unfortunately, HER kinetics in alkaline electrolytes are two or three orders of magnitude slower than those in acidic conditions, originating from the sluggish water dissociation step and the competitive adsorption of OH<sup>-</sup> on catalyst active sites in alkaline systems. The relatively low

catalytic efficiency hinders the practical application of the alkaline HER in industrial-scale hydrogen production.<sup>13</sup> Therefore, the development of low-cost, highly active bifunctional HER/OER electrocatalysts in alkaline conditions holds critical importance for enabling energy-efficient pathways toward high-purity hydrogen production.<sup>14</sup>

Recently, transition metal oxides (TMOs) such as WO<sub>3</sub> have attracted increasing attention as bifunctional electrocatalysts for HER and OER, attributed to their natural abundance, cost-effectiveness, and competitive catalytic efficiencies.<sup>15,16</sup> However, the inherently low charge transfer efficiency and sparse catalytic active sites in pure WO<sub>3</sub> limit the further improvement of its water electrolytic performance. Several strategies, such as morphology controlling,<sup>17</sup> phase transition,<sup>18</sup> doping,<sup>19,20</sup> interfacial engineering<sup>21-23</sup> and the microenvironment effect,<sup>24</sup> have been demonstrated to optimize electrocatalytic activity of WO<sub>3</sub>. Notably, doping can effectively adjust the electronic structure, reduce the reaction energy barriers for both the HER and OER, thereby significantly enhancing the catalytic activity.<sup>20</sup> For instance, Liang *et al.*<sup>25</sup> prepared Ni-doped tungsten oxide nanorods (Ni-WO<sub>x</sub>) for HER in alkaline water and seawater, achieving only 40.51 and 137.04 mV at 10 and 100 mA cm<sup>-2</sup>, respectively. Furthermore, the Ni-WO<sub>x</sub> required 45.69 mV at 10 mA cm<sup>-2</sup> and had superior stability over 120 h in alkaline seawater. Liu *et al.* reported that the Co-doped WO<sub>3</sub> exhibited a HER overpotential of 117 mV at a current density of 10 mA cm<sup>-2</sup>.<sup>7</sup> Compared with 3d transition metals (TMs), the Ce

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element with 4f electrons exhibits oxidation states of +3 and +4, enabling redox interaction between  $\text{Ce}^{3+}$  and  $\text{Ce}^{4+}$ ,<sup>26</sup> thus it has a relatively prominent advantage in the field of electrocatalysis.

TMs alloys, especially Ni and Co, remain a research focus owing to their structurally adaptive coordination environments and versatile reactivity in multi-step catalytic cycles. These properties, including mechanical strength, magnetism and catalytic activity, are closely related to their tunable d-orbital configurations and chemical structures.<sup>9,27,28</sup> TM-based catalysts have been demonstrated exceptional ability in promoting HO–H bond dissociation. Coupled interfacial effects emerge when combined with other electrocatalytic materials, which significantly enhance the reaction dynamics of alkaline water splitting processes.<sup>29–31</sup> However, the intrinsically sluggish water dissociation kinetics of such composite systems fundamentally constrain their catalytic efficiency under alkaline conditions.<sup>32</sup> Heterostructures<sup>33,34</sup> are widely constructed to improve their electrocatalytic performance by enhancing interfacial electron transfer kinetics, and concurrently optimizing reactant adsorption energetics and charge transport capabilities.<sup>5,35,36</sup> Li *et al.* reported that  $\text{NiCoO}_2/\text{NiCo}@C$ , synthesized *via* a two-step method, exhibits HER and OER overpotentials of only 61.9 and 329 mV at 10 mA  $\text{cm}^{-2}$ .<sup>37</sup> Therefore, NiCo alloy can be used to construct heterostructures with other oxides to enhance the overall water splitting performance.<sup>38</sup>

In this work,  $\text{Ce-WO}_3@\text{NiCo}$  heterostructures were synthesized by hydrothermal and electrodeposition method, and served as a bifunctional electrocatalyst to evaluate overall water splitting performance. The integration of  $\text{Ce-WO}_3$  and NiCo establishes robust interfacial coupling effects that significantly accelerate charge transfer kinetics and improve conductivity, collectively contributing to the electrocatalytic performance. The  $\text{Ce-WO}_3@\text{NiCo}$  catalyst exhibited excellent electrochemical activity for both the HER and OER. The use of  $\text{Ce-WO}_3@\text{NiCo}$  as both cathode and anode enabled systematic evaluation of the full water electrolysis process.

## 2. Experimental methods

### 2.1. Materials

Copper foam ( $1 \times 1 \text{ cm}^2$ ) was served as the substrate. Reagents including absolute ethanol ( $\text{CH}_3\text{CH}_2\text{OH}$ ,  $\geq 99.7\%$ ) and potassium hydroxide (KOH,  $\geq 85\%$ ), also metal chloride precursors: nickel(II) chloride hexahydrate ( $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$ ,  $\geq 98.0\%$ ), cobalt(II) chloride hexahydrate ( $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ ,  $\geq 99.0\%$ ), cerium(III) chloride heptahydrate ( $\text{CeCl}_3 \cdot 7\text{H}_2\text{O}$ ,  $\geq 99.0\%$ ), and tungsten(VI) chloride ( $\text{WCl}_6$ ,  $\geq 99.9\%$ ) are of analytical grade and supplied by Sinopharm Chemical Reagent Co., Ltd.

### 2.2. Synthesis of $\text{Ce-WO}_3$

The  $\text{Ce-WO}_3$  nanostructures were synthesized *via* the hydrothermal method on copper foam (CF) substrates. First, several pieces of CF ( $1 \times 1 \text{ cm}^2$ ) were sequentially ultrasonic cleaned in 3 M HCl, absolute ethanol, and deionized water (resistivity: 18.2 M $\Omega$  cm). Subsequently, a precursor solution was obtained through dissolution of 250 mg  $\text{WCl}_6$  and 0.0468 g  $\text{CeCl}_3 \cdot 7\text{H}_2\text{O}$

in 40 mL absolute ethanol, and magnetic stirring over 30 minutes. The obtained solution was transferred into an 80 mL Teflon-lined stainless steel autoclave, and the cleaned CF pieces were immersed for the subsequent reaction. The autoclave underwent thermal processing at 180 °C for 12 h in a controlled oven environment, followed by spontaneous cooling to ambient temperature. The obtained  $\text{Ce-WO}_3/\text{CF}$  was sequentially rinsed with absolute ethanol and deionized water to remove inorganic and organic residues. And finally, the resultant material was vacuum-dried at 60 °C for 6 hours. For comparative analysis, pure  $\text{WO}_3$  nanostructure were fabricated *via* identical synthesis method without  $\text{CeCl}_3 \cdot 7\text{H}_2\text{O}$ .

### 2.3. Synthesis of $\text{Ce-WO}_3@\text{NiCo}$

NiCo was synthesized on  $\text{Ce-WO}_3$  *via* electrochemical deposition using a three-electrode configuration. The Pt sheet, Ag/AgCl (3.5 M  $\text{Cl}^-$ ) and *in situ* growth of  $\text{Ce-WO}_3$  on CF were used as counter, reference and working electrode, respectively. Electrodeposition was conducted at a constant potential of  $-1.0$  V (*vs.* Ag/AgCl) for 20 minutes using an electrochemical workstation in a solution composed of 0.1 M KCl, 0.05 M  $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$ , and 0.02 M  $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$  under ambient temperature conditions.<sup>39</sup>

### 2.4. Characterization

The morphology and elemental mapping were characterized by field-emission scanning electron microscopy (FE-SEM, FEI Quanta FEG 250). The crystal structure of electrocatalysts was characterized by X-ray diffraction (XRD, Rigaku Ultima IV,  $\text{Cu K}\alpha$   $\lambda = 1.5406 \text{ \AA}$ ). The surface valence states of electrocatalysts were analyzed by X-ray photoelectron spectroscopy (XPS, Thermo Scientific K-Alpha) with Al  $\text{K}\alpha$  radiation, calibrated using the adventitious carbon peak (C 1s at 284.8 eV). The lattice structure and morphology were resolved by high-resolution TEM imaging employing an FEI Talos F200X microscope.

### 2.5. Electrochemical measurements

Electrochemical measurements of as-prepared electrocatalysts were conducted using a CHI 760E workstation with a standard three-electrode system: Ag/AgCl (reference electrode), Pt foil (counter electrode), and catalyst-coated substrate (working electrode) in 1.0 M KOH electrolyte. Linear sweep voltammetry (LSV) analysis of the catalysts was conducted at scan rate of 5 mV  $\text{s}^{-1}$ . The experimental data were normalized to the reversible hydrogen electrode (RHE) through the Nernst equation:  $E$  (*vs.* RHE) =  $E$  (*vs.* Ag/AgCl) + 0.197V + 0.05916  $\times$  pH. The double layer capacitance ( $C_{dl}$ ) of different catalysts was measured using cyclic voltammetry (CV) with the voltage *vs.* RHE ranging from 0 to 0.1 V, and the CV curves were obtained at sweep rates spanning from 20 to 100 mV  $\text{s}^{-1}$  with a stepwise increment of 20 mV  $\text{s}^{-1}$ . The electrochemical impedance spectroscopy (EIS) was studied with the frequency ranging from  $1 \times 10^5$  to  $1 \times 10^{-2}$  Hz. Stability was evaluated by chronoamperometry at current densities of 10, 100, and 300 mV  $\text{s}^{-1}$ , respectively. All the electrochemical measurements were performed at room temperature.



### 3. Results and discussion

Fig. 1(a) illustrates the two-step synthetic process of the Ce-WO<sub>3</sub>@NiCo electrocatalyst. Copper foam was selected as the growth substrate for its good conductivity, three dimensional multi-porous structure and structural stability. First, the Ce-WO<sub>3</sub> nanosheets were synthesized *via* hydrothermal method with different Ce doping concentrations. Subsequently, the

NiCo layer was deposited onto the surface of Ce-WO<sub>3</sub> nanosheets *via* electrochemical deposition. Finally, the flower-like three-dimensional Ce-WO<sub>3</sub>@NiCo heterostructure was formed on CF substrate.

The morphological features of as-synthesized WO<sub>3</sub>, NiCo, Ce-WO<sub>3</sub>, and Ce-WO<sub>3</sub>@NiCo were analyzed *via* SEM. As was revealed in Fig. 1(b), the pure WO<sub>3</sub> catalyst displayed an interconnected nanosheet architecture. The pure NiCo alloy layer

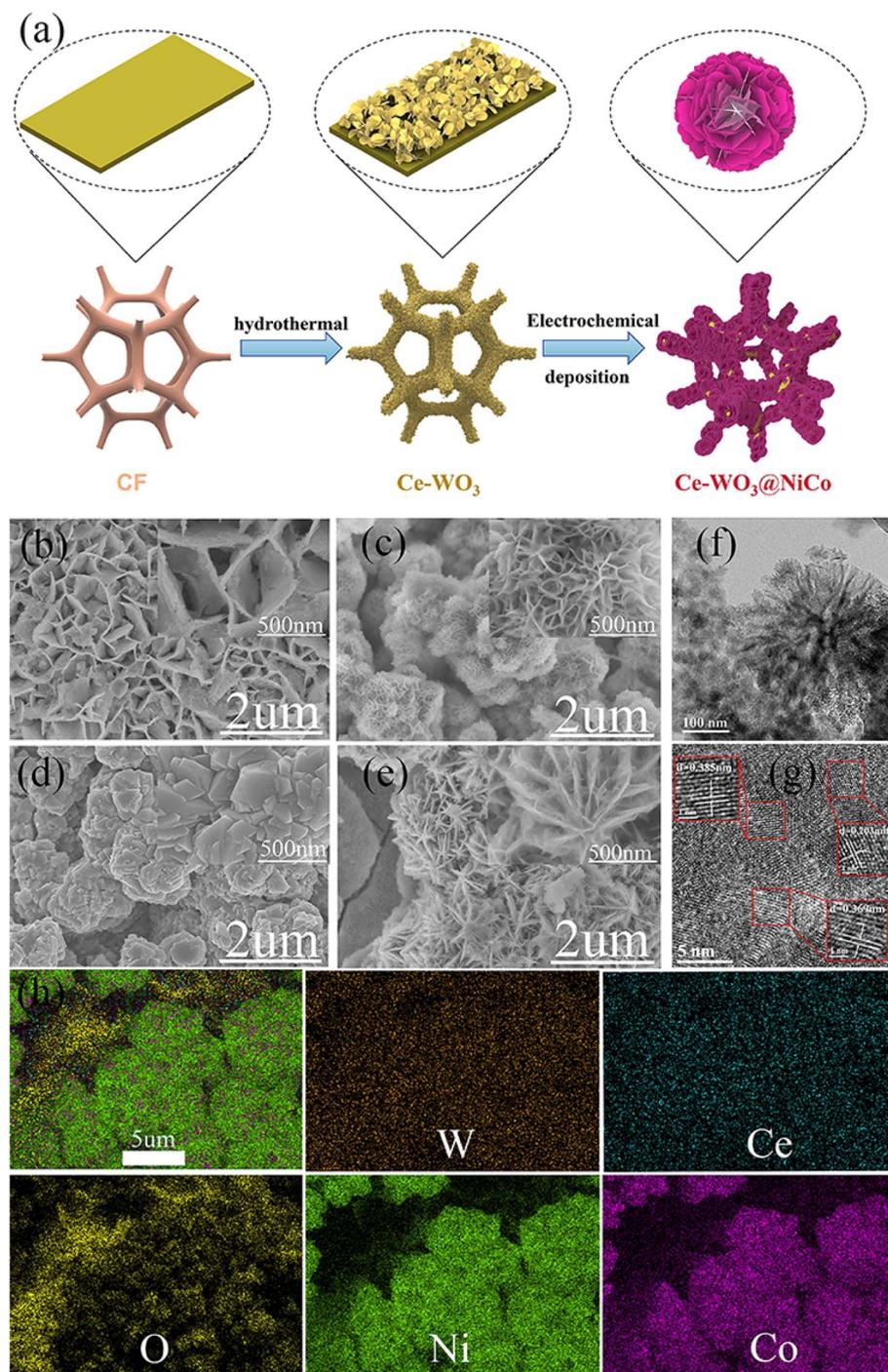


Fig. 1 (a) Synthesis process of Ce-WO<sub>3</sub>@NiCo. SEM images of (b) WO<sub>3</sub>, (c) NiCo, (d) Ce-WO<sub>3</sub>, (e) Ce-WO<sub>3</sub>@NiCo, (f) TEM, (g) HRTEM images of Ce-WO<sub>3</sub>@NiCo, (h) SEM image of Ce-WO<sub>3</sub>@NiCo, and the associated energy-dispersive spectroscopy (EDS) elemental mapping of W, Ce, O, Ni, and Co.



was sphere-like structure formed by the interweaving and stacking of nanosheets on the CF substrate, which was conducive to exposing active sites (Fig. 1(c)). For Ce-WO<sub>3</sub>, the doping of Ce significantly elevated the formation energy of oxygen vacancy (Fig. 1(d)),<sup>40</sup> thereby suppressing the dissolution of tungsten under alkaline conditions and enhancing its structural stability. Meanwhile, heteroatom doping introduces ionic mismatch, leading to lattice distortion<sup>41</sup> and internal strain.<sup>42</sup> To alleviate this strain, the nanosheets fragment and trigger recrystallization. Furthermore, the distorted lattice disrupts the original growth anisotropy, suppressing the expansion of the initially preferred planes while enhancing growth in perpendicular directions. Consequently, the doping could increase the thickness of the nanosheets and form denser particles. As illustrated in Fig. 1(e), the Ce-WO<sub>3</sub>@NiCo composite exhibited a three-dimensional nanoflower architecture composed of tightly interconnected ultrathin nanosheets. This unique structure not only provides additional active edge sites but also increases the specific surface area, reduces the charge transfer distance between active species, and thereby enhances overall conductivity. The microstructure of Ce-WO<sub>3</sub>@NiCo was further characterized by TEM and HRTEM. The TEM image confirmed the presence of the nanoflower structure, which was consistent with the SEM result. HRTEM analysis (Fig. 1(g)) revealed lattice spacings of 0.203, 0.385, and 0.369 nm, corresponding to the NiCo (111) and WO<sub>3</sub> (001)/(200) lattice planes, respectively. The EDS elemental mapping images (Fig. 1(h)) confirmed the uniform distribution of O, Co, Ni, Ce, and W elements in the Ce-WO<sub>3</sub>@NiCo catalyst. In summary, the TEM results further verified the successful fabrication of the Ce-WO<sub>3</sub>@NiCo heterostructure.

The crystallographic structure of the samples was characterized using X-ray diffraction. Fig. 2(a) displayed the XRD patterns of as-synthesized samples, and the XRD patterns of undoped WO<sub>3</sub> and Ce-doped WO<sub>3</sub> with varying Ce contents were plotted in Fig. S1. The XRD patterns were dependent on the Ce concentration. For Ce-doped WO<sub>3</sub>, the introduction of Ce could reduce the diffraction peak intensity of WO<sub>3</sub> compared to the undoped, probably resulting from lattice compression and poor crystallinity induced by Ce dopants.<sup>43</sup> Furthermore, the diffraction peak of CeO<sub>2</sub> would appear when the doping concentration of Ce reached a certain extent, which may affect the properties of the WO<sub>3</sub>.<sup>43</sup> The XRD result also showed that WO<sub>3</sub> has been synthesized by the hydrothermal method, and  $2\theta = 23.083, 23.707, 24.098, 28.775, \text{ and } 34.022$  corresponded to (002), (020), (200), (111), and (220) crystal planes, respectively, which matched the pattern of WO<sub>3</sub> (#00-020-1324). The peaks at  $2\theta = 44.428^\circ, 51.768^\circ$  and  $76.25^\circ$ , corresponding to (111), (200) and (220) crystal planes, matched the pattern of NiCo alloy (#00-003-2246). And three distinct peaks at  $2\theta = 43.407^\circ, 50.556^\circ$  and  $74.297^\circ$ , corresponding to the (111), (200), and (220) planes of CF (#00-004-0836) were observed.

The surface elemental composition and chemical states were analyzed using X-ray photoelectron spectroscopy. The W 4f spectra of different samples were shown in Fig. 2(b). Two peaks observed at 35.47 eV and 37.69 eV in the W 4f spectrum of WO<sub>3</sub> was assigned to the W 4f<sub>7/2</sub> and W 4f<sub>5/2</sub> orbitals. The binding

energy of W 4f<sub>7/2</sub> in Ce-WO<sub>3</sub> was offset by about 0.1 eV, higher than pure WO<sub>3</sub>, and the higher binding energy shift indicated electron transfer from W to Ce.<sup>44</sup> The W 4f XPS spectrum of Ce-WO<sub>3</sub>@NiCo exhibited two peaks at 35.02 eV (W 4f<sub>7/2</sub>) and 37.19 eV (W 4f<sub>5/2</sub>). The W 4f<sub>7/2</sub> binding energy of Ce-WO<sub>3</sub>@NiCo was negatively shifted by about 0.55 eV compared to the W 4f<sub>7/2</sub> binding energy of Ce-WO<sub>3</sub>.<sup>7,45</sup> These results showed that Ce-WO<sub>3</sub>@NiCo had electron transfer and interface coupling interaction between Ce-WO<sub>3</sub> and NiCo heterogeneous interfaces. In the Ni 2p XPS spectrum of Ce-WO<sub>3</sub>@NiCo (Fig. 2(c)), it could be seen that two spin-orbit coupling peaks at binding energies of 855.6 eV and 873.3 eV, corresponding to Ni<sup>2+</sup>. Furthermore, the satellite peaks located at 861.3 eV (2p<sub>3/2</sub>) and 877.6 eV (2p<sub>1/2</sub>), and the Ni<sup>0</sup> peak at 852.1 eV were observed.<sup>46,47</sup> For the Co 2p XPS curve of Ce-WO<sub>3</sub>@NiCo (Fig. 2(d)), the peaks at 781.8 eV (2p<sub>3/2</sub>) and 796.7 eV (2p<sub>1/2</sub>) indicated the presence of surface oxidation states, and the other two lower peaks at 786.2 eV and 803.4 eV were attributed to the satellite peaks. Moreover, the peak located at 777.7 eV was assigned to Co<sup>0</sup>.<sup>46,48,49</sup> The high-resolution spectrum of Ce 3d (Fig. 2(e)) revealed characteristic spin-orbit doublets, with the binding energy range of 895.5–904.9 eV corresponding to Ce 3d<sub>3/2</sub> states and 881.7–882.4 eV associated with Ce 3d<sub>5/2</sub>. The peaks of Ce<sup>3+</sup> located at 881.7 and 900.1 eV, and the five peaks of Ce<sup>4+</sup> at 882.4, 895.5, 899.5, 902.7, and 904.9 eV, clearly indicated the existence of both trivalent and tetravalent cerium ions in the sample.<sup>35,43</sup> The O 1s in Fig. 2(f) was split into three distinct peaks at 529.96 eV, 531.32 eV, and 532.53 eV, attributed to metal–oxygen bonds, hydroxyl groups, and adsorbed water molecules, respectively. Notably, the binding energy of the O 1s orbital exhibited a 0.55 eV positive shift compared with pristine WO<sub>3</sub>.<sup>44,50,51</sup> These experimental results showed that the coupling between Ce-WO<sub>3</sub> and NiCo could form a heterogeneous interface with strong electronic interaction.

Comparative HER studies were carried out in 1.0 M KOH employing a tri-electrode potentiostat setup. First, the effect of heteroatom doping on the catalytic performance of the material was investigated; and the HER performance of Ce-WO<sub>3</sub> with varying amounts of Ce was studied (Fig. S2). It can be observed that the Ce doping significantly enhanced HER performance, which could be attributed to the Ce-induced charge transfer that modified the intrinsic electronic structure of WO<sub>3</sub>. Polarization curve analysis in Fig. 3(a) revealed the superior catalytic property of Ce-WO<sub>3</sub>@NiCo under various potentials compared with CF, NiCo, WO<sub>3</sub>, and Ce-WO<sub>3</sub>. The Ce-WO<sub>3</sub>@NiCo achieved a low HER overpotential ( $\eta$ ) of 56 mV at 10 mA cm<sup>-2</sup>, showing much lower overpotentials than that of CF (262 mV), NiCo (135 mV), WO<sub>3</sub> (211 mV), and Ce-WO<sub>3</sub> (170 mV) at the same current density. Additionally, the overpotential required by Ce-WO<sub>3</sub>@NiCo was slightly higher than that of Pt/C (Fig. S3), and it exhibited superior electrocatalytic performance compared to the reported other electrocatalysts (as shown in Table S1). The Tafel slopes derived from the LSV curves of these electrocatalysts were shown in Fig. 3(b). In the range of 1.1–2.0 V, the Tafel slopes for CF, NiCo, WO<sub>3</sub>, Ce-WO<sub>3</sub>, and Ce-WO<sub>3</sub>@NiCo were 280.77, 129.71, 268.66, 146.09, and 128.21 mV dec<sup>-1</sup>, respectively. A comparison between WO<sub>3</sub> and Ce-WO<sub>3</sub>



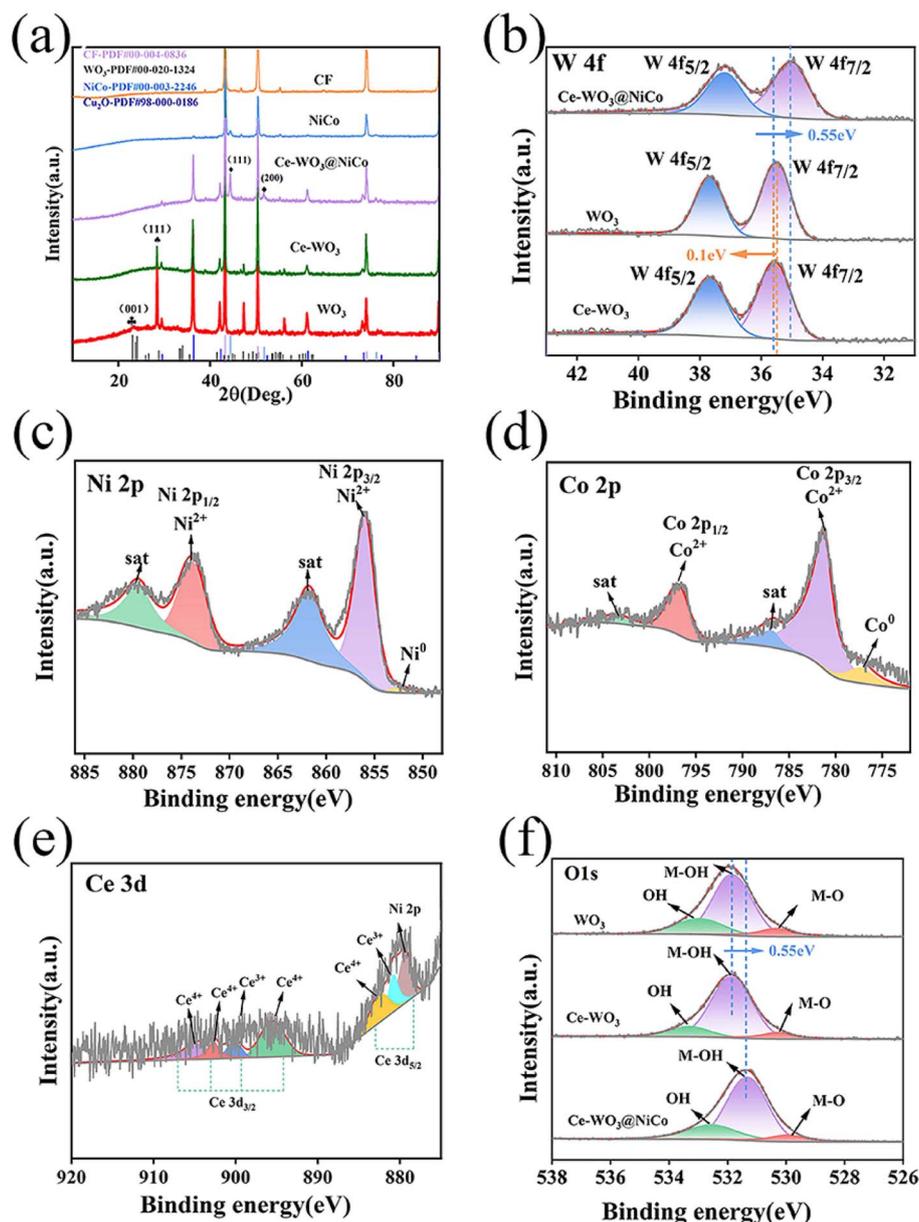


Fig. 2 (a) XRD patterns of Ce-WO<sub>3</sub>@NiCo, NiCo, Ce-WO<sub>3</sub>, WO<sub>3</sub>, and CF. XPS elemental spectra of different samples: (b) W 4f, (c) Ni 2p, (d) Co 2p, (e) Ce 3d, (f) O 1s.

demonstrated that Ce doping accelerated the HER reaction kinetics. The lower Tafel slope of Ce-WO<sub>3</sub>@NiCo indicated that the interfacial coupling between Ce-WO<sub>3</sub> and NiCo could promote water dissociation and hydrogen adsorption. To elucidate HER activity enhancement mechanisms, the electrochemically active surface area (ECSA) of electrocatalysts was systematically characterized. The CV scans in the non-Faradaic potential region were performed at different scan rates (Fig. S4), and the double-layer capacitance ( $C_{dl}$ ), which was linearly related to ECSA, was obtained by fitting the CV curves. Fig. 3(c) revealed that the  $C_{dl}$  of CF, WO<sub>3</sub>, Ce-WO<sub>3</sub>, NiCo, and Ce-WO<sub>3</sub>@NiCo were 9.46, 11.2, 15.26, 16.21, and 35.34 mF cm<sup>-2</sup>, respectively. From the experimental results, Ce-WO<sub>3</sub>@NiCo was

demonstrated a significantly enhanced electrochemically active surface area and more exposed active sites compared to other electrocatalysts. It was likely attributed to its rougher surface morphology that led to enhanced hydrophilicity, which further confirmed the structure–activity correlation.<sup>52</sup> Normalization of LSV curves was performed *via* ESCA, with the normalized data presented in Fig. S5. It could be observed from the figure that Ce-WO<sub>3</sub>@NiCo still exhibited excellent performance in the normalized LSV curves. The EIS measurements were performed to investigate interfacial charge transfer behavior across electrode/electrolyte interfaces, and the interfacial kinetics during the HER process were revealed. The Nyquist curves of CF, WO<sub>3</sub>, Ce-WO<sub>3</sub>, NiCo, and Ce-WO<sub>3</sub>@NiCo were presented in



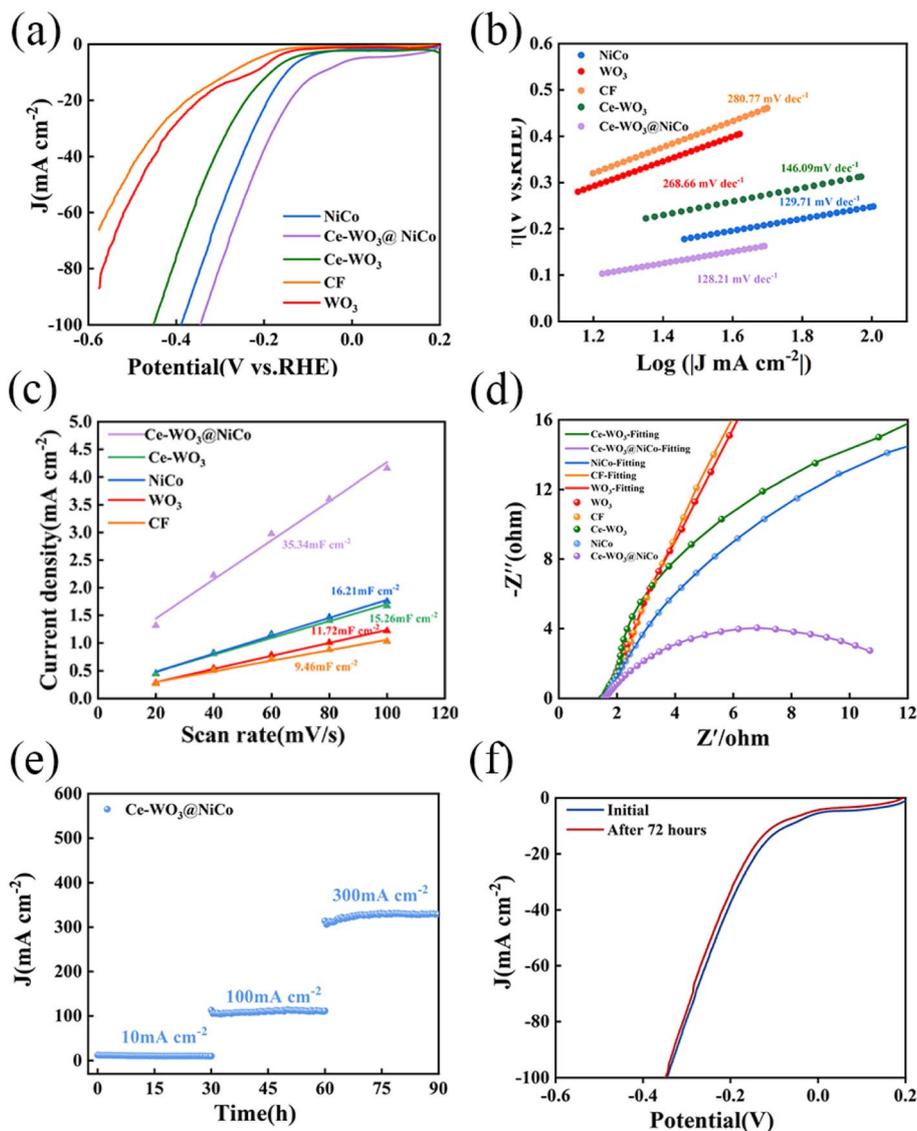


Fig. 3 HER performance of CF, NiCo/CF, WO<sub>3</sub>/CF, Ce-WO<sub>3</sub>, and Ce-WO<sub>3</sub>@NiCo samples: (a) LSV curves, (b) Tafel slopes, (c)  $C_{dl}$  measurement, (d) EIS plots, (e) stability test at different current densities, (f) LSV curves before and after Ce-WO<sub>3</sub>@NiCo stability test.

Fig. 3(d). The equivalent circuit diagram and the fitting results of experimental data are depicted in Fig. S6 and Table S2, respectively. The equivalent circuit employed for fitting was consisted of one series resistance ( $R_s$ ) and two parallel combinations of charge transfer resistance ( $R_{ct}$ ) and constant phase element (CPE). It could be observed from the Nyquist plot in Fig. 3(d) that Ce-WO<sub>3</sub>@NiCo exhibited the smallest semicircle diameter, indicating the lowest charge transfer resistance. The specific values are provided in Table S3, which demonstrated that Ce-WO<sub>3</sub>@NiCo possessed the fastest charge transfer kinetics. The Ce-WO<sub>3</sub>@NiCo heterostructure could improve charge transfer ability and increase the electron migration rate, thereby enhancing HER performance.<sup>53–56</sup> In addition to the excellent HER activity, the stability of HER was also investigated. The durability test of Ce-WO<sub>3</sub>@NiCo was shown in Fig. 3(e and f), the catalytic stability was evaluated at 10, 100, and 300 mA cm<sup>-2</sup> through  $i-t$  tests, and no significant

attenuation in the curves was observed. The LSV plots of Ce-WO<sub>3</sub>@NiCo before and after the electrochemical test were compared, and the good degree of overlap demonstrated that Ce-WO<sub>3</sub>@NiCo possessed outstanding electrochemical stability in HER. As shown in Fig. S7, the multi-step chronopotentiometry measurements were performed on WO<sub>3</sub>, Ce-WO<sub>3</sub>, NiCo, and Ce-WO<sub>3</sub>@NiCo working electrodes across a current density range of 10–210 mA cm<sup>-2</sup> with a step increment of 40 mA cm<sup>-2</sup>. Compared with electrocatalysts, Ce-WO<sub>3</sub>@NiCo exhibited the lowest initial potential, and this trend was maintained under different current densities. Specifically, Ce-WO<sub>3</sub>@NiCo exhibits excellent catalytic activity, along with satisfactory stability under stepwise increasing current densities. Generally, the outstanding HER activity and stability of Ce-WO<sub>3</sub>@NiCo could be attributed to three factors: (1) the direct growth of Ce-WO<sub>3</sub> nanosheets on CF as a scaffold for subsequent NiCo nanosheets deposition



strengthened interfacial connectivity, promoting the rapid charge transfer. (2) The three-dimensional hierarchical structure provided abundant catalytic sites and therefore facilitated gas release and prevented mechanical detachment. (3) The synergistic effect in the Ce-WO<sub>3</sub>@NiCo heterostructure enhanced its electrochemical activity.

The OER catalytic performance of as-synthesized samples was comprehensively compared under standardized testing conditions. To achieve superior electrochemical activity in Ce-WO<sub>3</sub>, the OER performance under different Ce doping levels was studied (Fig. S2). It was observed that Ce-doped WO<sub>3</sub> always exhibited much better OER performance than pure WO<sub>3</sub>. An oxidation peak was identified in the forward scanning LSV curve, which prevented the evaluation of  $\eta_{10}$ . To suppress the interference of oxidation peaks, the LSV curves in Fig. 4(a) were recorded in reverse scan mode from high to low potentials. The LSV curves of Ce-WO<sub>3</sub>@NiCo showed significantly enhanced reduction peaks, with their integrated peak area being much larger than that of the NiCo alloy. Additionally, the shift of the reduction peak indicated that the Ce-WO<sub>3</sub>@NiCo hetero-interface had a higher density of active sites and faster electron migration rates.<sup>37</sup> The enhanced redox dynamics and accelerated electron migration could improve OER reaction kinetics. At a current density of 10 mA cm<sup>-2</sup>, Ce-WO<sub>3</sub>@NiCo exhibited a substantially reduced overpotential (323 mV) compared to CF (391 mV), NiCo (355 mV), WO<sub>3</sub> (388 mV), and Ce-WO<sub>3</sub> (374 mV), and exhibited outstanding OER electrocatalytic activity than most of the catalysts reported in previous work (Table S3).

Analysis of the kinetics of the OER was conducted according to the Tafel slopes calculated from LSV polarization curves. In Fig. 4(b), the Tafel slopes of CF, NiCo, WO<sub>3</sub>, Ce-WO<sub>3</sub>, and Ce-WO<sub>3</sub>@NiCo were fitted to be 237, 190, 222, 203, and 175 mV dec<sup>-1</sup>, respectively. The lowest Tafel slope of Ce-WO<sub>3</sub>@NiCo indicated its faster OER reaction kinetics, suggesting that the catalyst possessed rapid electron migration and efficient mass transport. Additionally, the large Tafel slope in this range was likely due to the influence of oxidation peaks of Co/Ni ions during testing. The durability test of Ce-WO<sub>3</sub>@NiCo was shown in Fig. 4(c and d), the catalytic stability was evaluated at 10, 100, and 300 mA cm<sup>-2</sup> through *i-t* tests, and no obvious attenuation in the LSV curves was observed. The LSV curves of Ce-WO<sub>3</sub>@NiCo before and after the electrochemical test were compared, and the good degree of overlap demonstrated that Ce-WO<sub>3</sub>@NiCo possessed excellent electrochemical stability in OER.

Given the superior catalytic activity of the Ce-WO<sub>3</sub>@NiCo heterostructure, an integrated electrolyser was constructed and it served as cathode and anode electrodes for efficient alkaline water splitting. The polarization curves of Ce-WO<sub>3</sub>@NiCo/Ce-WO<sub>3</sub>@NiCo were investigated in 1 M KOH, as shown in Fig. 5(a). Ce-WO<sub>3</sub>@NiCo exhibited a low operating potential of 1.624 V to achieve 10 mA cm<sup>-2</sup>, showing 6.8–15.6% voltage reduction compared to control groups: CF (2.034 V), NiCo (1.702 V), Ce-WO<sub>3</sub> (1.827 V), and WO<sub>3</sub> (1.840 V). Table S4 presents the overall water splitting performance of Ce-WO<sub>3</sub>@NiCo and other reported catalysts at a current density of 10 mA cm<sup>-2</sup>. Compared with previous catalysts, the Ce-WO<sub>3</sub>@NiCo composite operates at

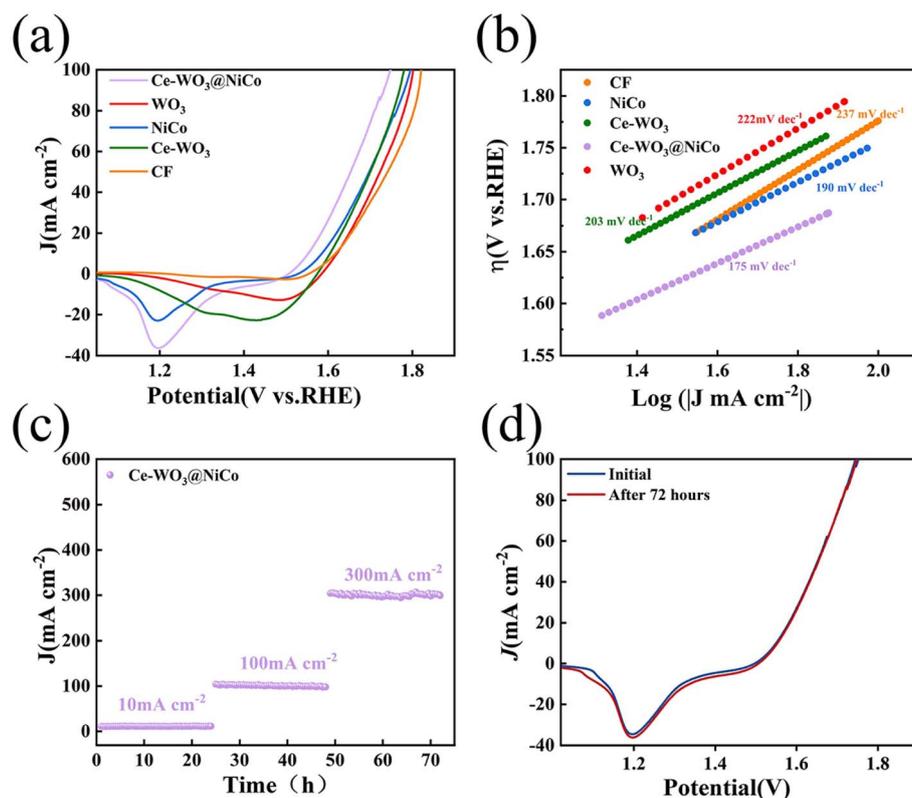


Fig. 4 OER performance of the as-grown samples: (a) LSV curves at scan rate of 5 mV s<sup>-1</sup>, (b) Tafel slopes, (c) stability test, (d) LSV curves before and after stability test of Ce-WO<sub>3</sub>@NiCo.

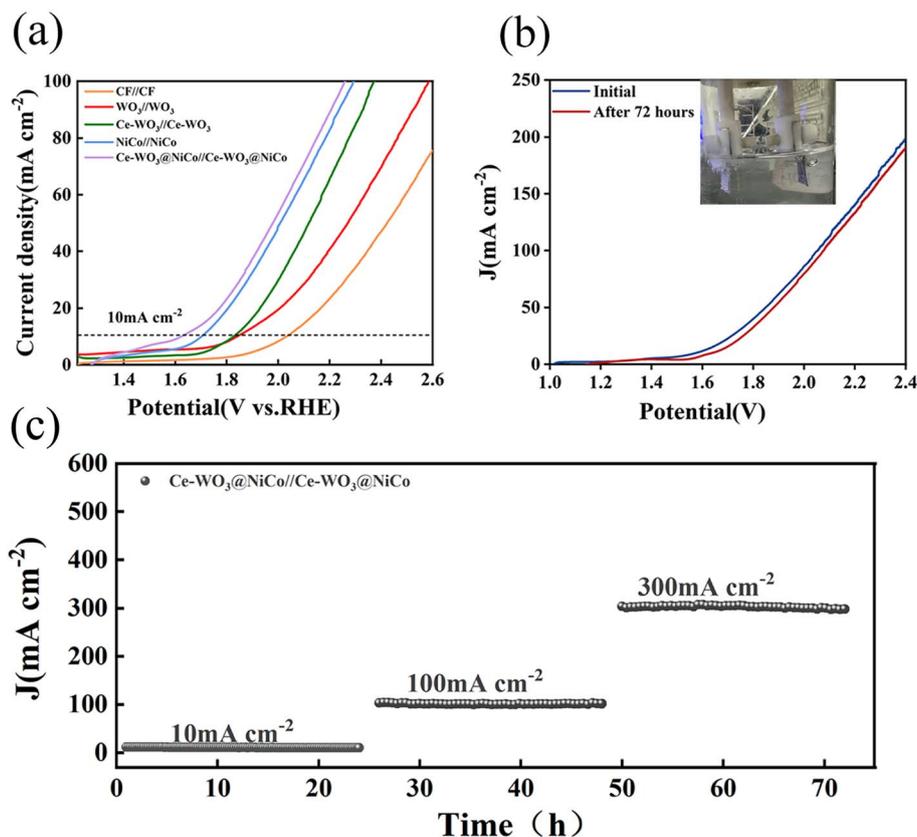


Fig. 5 (a) The overall hydrolysis performance of the sample, (b) electrochemical durability evaluation of Ce-WO<sub>3</sub>@NiCo for overall water splitting through comparative pre-/post-stability LSV measurements, (c) stability test over 72 h.

a lower overpotential, indicating excellent catalytic activity for overall water splitting in alkaline electrolytes. Fig. 5(b) revealed minimal divergence between the LSV curves before and after the stability test, confirming superior electrochemical stability. The morphology and crystal structure of the Ce-WO<sub>3</sub>@NiCo electrocatalyst after the durability test of 72 h were further studied. XRD and SEM images revealed that the structure was well preserved and that the surface shape of the Ce-WO<sub>3</sub>@NiCo material did not change significantly, as shown in Fig. S8 and S9. The Ce-WO<sub>3</sub>@NiCo nanostructure exhibited long-lasting catalytic stability (Fig. 5(c)) no noticeable degradation after 72 h undergoing sequential testing at 10, 100, and 300 mA cm<sup>-2</sup> for 24 h each at room temperature. The Ce-WO<sub>3</sub>@NiCo catalyst maintained structural integrity under large current density of 300 mA cm<sup>-2</sup>. The Ce-WO<sub>3</sub>@NiCo was demonstrated exceptional OER/HER bifunctionality and stability, establishing its potential in industrial-scale overall water splitting as an emerging catalyst.

## 4. Conclusions

In this paper, the Ce-WO<sub>3</sub>@NiCo for water splitting in an alkaline environment was prepared using hydrothermal and electrodeposition methods. The Ce-WO<sub>3</sub>@NiCo heterostructure had excellent bifunctional electrocatalytic activity and stability due to doping and synergistic effects. The catalyst achieved low overpotentials of 56 mV and 323 mV at 10 mA cm<sup>-2</sup> for HER and OER, respectively. The superior catalytic performance originated from interfacial

charge transfer dynamics and dopant-induced electronic modulation. The Ce-WO<sub>3</sub>@NiCo as bifunctional catalyst was used as cathode and anode of an electrolyzer. Notably, the potential of Ce-WO<sub>3</sub>@NiCo was only 1.624 V at 10 mA cm<sup>-2</sup>, also it had superior overall water splitting stability. In this study, a feasible design is established to fabricate an efficient and inexpensive bifunctional catalyst through interface engineering to address key challenges in industrial hydrogen production systems.

## Author contributions

Yuan Qin: investigation, writing – original draft. Meng Ding: funding acquisition, writing – review & editing. Weixiao Ji: conceptualization. Yafang Zhang: supervision. Gang Zhao: project administration.

## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.



Supplementary information (SI) is available. See DOI: <https://doi.org/10.1039/d5ra09712f>.

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