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# 1 Engineering $\beta$ -Cyclodextrin Gels with Nanoparticles: Tunable Assembly and 2 Multifunctional Applications

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10

## 11 Abstract

12 Hierarchical gels were developed through the controlled interaction of  $\beta$ -Cyclodextrin in good-  
13 poor solvent systems, incorporating small amounts of various nanoparticles and nanoclays.  
14 These new hierarchical microstructures form through the side-by-side aggregation of  $\beta$ -  
15 Cyclodextrin lamellar plates. They are stabilized by non-covalent interactions and facilitated  
16 by negatively charged nanoparticles or nanoclays. A systematic variation of nanoparticle  
17 concentration and solvent composition revealed that gelation occurs even at low concentrations  
18 of nanoparticles or nanoclays, significantly altering the typical phase behavior of  $\beta$ -  
19 Cyclodextrin in DMF-water mixtures. Interestingly, a variety of differently shaped, negatively  
20 charged nanoparticles—including nanorods, nanodisks, and nanoplatelets—supported similar  
21 hierarchical self-assembly. The smart gels exhibit responsiveness to both temperature and salt,  
22 effectively removing cationic dyes. Specifically, temperature-induced phase transitions were  
23 demonstrated using three different types of nanoparticles, highlighting their potential use as  
24 temperature sensors. By combining  $\beta$ -Cyclodextrin with nanoparticles such as cellulose  
25 nanocrystals, montmorillonite, and Laponite, we created composite gels that show improved  
26 selectivity and sensitivity for cationic dye detection.

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## 31 1. Introduction

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32 Hierarchical organization in soft matter systems provides a versatile pathway for engineering  
33 functional materials with tunable architectures and emergent properties. By integrating order  
34 across multiple length scales through the spontaneous self-assembly of amphiphilic molecules,  
35 these systems bridge molecular design with macroscopic performance, enabling applications  
36 in sensing, environmental remediation, and stimuli-responsive technologies [3–7]. These  
37 structures, classified as a specific type of soft material, can be constructed utilizing gelators  
38 such as polymers, natural polysaccharides, low-molecular-weight organic compounds, and  
39 other similar materials. [8–9]. Among the various gelators,  $\beta$ -Cyclodextrin ( $\beta$ -CD) emerges as  
40 a particularly promising candidate for gel formation.  $\beta$ -CD is a cyclic oligosaccharide  
41 comprising seven glucopyranose units, characterized by its hydrophobic cavity and hydrophilic  
42 exterior [10–12]. Due to its distinctive structure,  $\beta$ -CD is capable of forming inclusion  
43 complexes with appropriate guest molecules through host-guest interactions, thereby serving  
44 as an outstanding gelator for the creation of functional gels [13–16]. Owing to its hydrophilic  
45 outer surface and end faces,  $\beta$ -CD readily forms hydrogen-bonded channel-type stacks in the  
46 presence of water. Screening of various solvents identified N, N-dimethylformamide (DMF),  
47 dimethyl sulfoxide, and N, N-Dimethylacetamide as effective media for gel formation [17].  $\beta$ -  
48 CD dissolves well in these solvents, and subsequent addition of water at room temperature  
49 induces stable gelation via channel-type stacking into cage-like assembly. This process is  
50 characterized by water, acting as a poor solvent, which prompts the self-assembly of  $\beta$ -CD into  
51 a stable gel. This unique characteristic of  $\beta$ -CD offers promising prospects for the development  
52 of advanced materials applicable in drug delivery, environmental remediation, and stimuli-  
53 responsive systems [18–20].

54 Stimuli-responsive supramolecular assembly driven by noncovalent interactions has garnered  
55 increasing interest for fabricating and controlling hierarchical micro- and nanostructures due  
56 to its simplicity and versatility.  $\beta$ -Cyclodextrin ( $\beta$ -CD) exhibits rich morphological diversity,  
57 including lamellar plates formed via side-by-side alignment stabilized by intermolecular  
58 hydrogen bonding [21] and tubular aggregates generated through head-to-tail stacking  
59 combined with hydrophobic guest inclusion [22]. Previous studies have demonstrated the  
60 incorporation of metal ions and graphene oxide into  $\beta$ -CD assemblies to modulate their  
61 mechanical properties and stimuli-responsiveness [23–24]. However, such systems rely on  
62 specific nanomaterials and remain inherently system-dependent, limiting broader control over

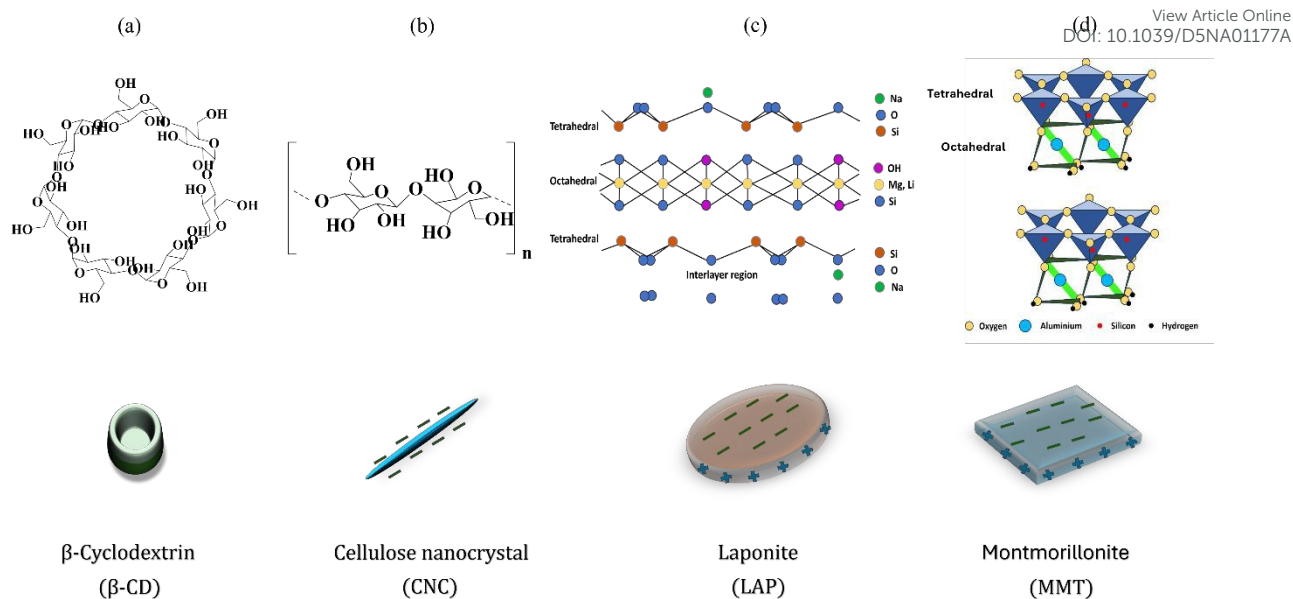


63  $\beta$ -CD self-assembly. In contrast, the present work introduces a generalizable strategy in which  
64 structurally diverse, negatively charged nanoparticles—including nanoclays and  
65 nanocellulose—direct  $\beta$ -CD assembly within a controlled good–poor solvent framework. This  
66 enables systematic tuning of gelation behavior, novel feather mimicking hierarchical  
67 architecture, and multi-stimuli responsiveness, thereby advancing beyond system-specific  
68 approaches and providing a versatile platform for designing functional supramolecular gels  
69 [25–28].

70 We selected negatively charged nanoparticles of various shapes and sizes to investigate their  
71 impact on the self-assembly of  $\beta$ -cyclodextrin ( $\beta$ -CD). The nanoparticles we examined include  
72 cellulose nanocrystals (CNC) and two types of smectite clay minerals: montmorillonite (MMT)  
73 and Laponite (LAP). Cellulose is a linear polysaccharide composed of repeating D-  
74 glucopyranose units linked by 1,4- $\beta$ -glycosidic bonds, as illustrated in Figure 1 [29, 30]. It  
75 forms rigid, rod-like particles with needle-like structures, which contain monocrystalline  
76 domains extending up to 200-400 nm in length [31]. In contrast, Laponite and montmorillonite  
77 have a sandwich-like structure consisting of an octahedral sheet flanked by two tetrahedral  
78 silica sheets [32-34]. These clays are composed of plate-shaped nanoparticles that are 1 nm  
79 thick and possess a large, permanently charged face due to isomorphic substitution, primarily  
80 occurring in the octahedral sheet (Figure 1). Typically, Laponite has a hydrodynamic diameter  
81 of 50-70 nm, while MMT has a diameter ranging from 250-350 nm (Figure S1).

82 In this study, nanoparticles such as CNC, MMT, and LAP were utilized to create smart,  
83 responsive hybrid gel systems, using  $\beta$ -cyclodextrin ( $\beta$ -CD) as the gelator. Uniform  
84 microstructures were observed to begin forming with short tips composed of  $\beta$ -CD channel-  
85 like assemblies. These nanoparticles were integrated with  $\beta$ -CD plates to create a spherulite-  
86 like hierarchical microstructure. The impact of varying DMF-water solvent ratios on the phase  
87 behavior of the system was carefully investigated. Rheological studies were conducted to  
88 evaluate the gel strength and viscoelastic properties. The gel systems containing different  
89 negatively charged nanoparticles demonstrated responsiveness to salt and temperature,  
90 highlighting their significance for sensing applications.  $\beta$ -CD/nanomaterial composites  
91 exhibited enhanced selectivity and sensitivity in detecting cationic dyes due to the synergistic  
92 combination of host-guest chemistry and the high surface area of the nanostructures. These  
93 attributes present a promising platform for the development of advanced environmental  
94 monitoring and sensing technologies.





95

96 Figure 1: Structural representations of (a)  $\beta$ -Cyclodextrin, (b) cellulose nanocrystals, (c)  
 97 Laponite, and (d) montmorillonite, highlighting their molecular frameworks and characteristic  
 98 morphologies used in hierarchical gel assembly.

## 99 2. Experimental methods

### 100 2.1 Sample preparation

101  $\beta$ -Cyclodextrin ( $\beta$ -CD,  $C_{42}H_{70}O_{35}$ ) and N, N-dimethylformamide (DMF) were purchased  
 102 from Sigma-Aldrich (India) and used without further modification. Cellulose nanocrystal  
 103 (CNC) powder was obtained from Celluforce (Canada), sodium montmorillonite ( $Na^+$ -MMT)  
 104 from Southern Clay Products (USA), and Laponite powder from BYK Additives (Germany).  
 105 All materials were used as received. CNC particles exhibited sizes of 200–400 nm with a zeta  
 106 potential of  $-31.6$  mV. MMT particles ranged from 250–350 nm with a zeta potential of  $-37.5$   
 107 mV, while LAP showed a hydrodynamic diameter of 50–70 nm and a zeta potential of  $-20$   
 108 mV. Milli-Q (MQ) water was used to prepare CNC, MMT, and LAP suspensions (0.5–1.5  
 109 wt%) under magnetic stirring to ensure uniform dispersion (Figure S1).

110 Composite samples were prepared using the good/poor solvent method as proposed by Ma M.  
 111 et al. [23]. Typically, a stock solution of 30% (w/v)  $\beta$ -CD in DMF was prepared by continuous  
 112 stirring and ultrasonic treatment, resulting in a transparent dispersion. As water is a poor  
 113 solvent for  $\beta$ -CD, when an equal volume of nanoparticles (CNC, MMT, or LAP) in MQ-water  
 114 was added to the stock solution, the sample turned into a white gel within 5 minutes for CNC  
 115 and LAP, and a pale-yellow gel for MMT.



116

## 117 2.2 Sample characterization

118 Solid-state characterization of pure  $\beta$ -CD gels, pristine nanoparticles, and their composite  
119 systems was carried out using powder X-ray diffraction (XRD). Crystal structures were  
120 analyzed using a Rigaku SmartLab diffractometer (Cu K $\beta$ -filter, 42 kV, 120 mA) over a  $2\theta$   
121 range of 5–60° with a scan rate of 2° min<sup>-1</sup>. FTIR spectra were recorded on a Bruker ALPHA  
122 spectrometer in transmission mode using KBr pellets, over the range of 4000–450 cm<sup>-1</sup> at a  
123 resolution of 2 cm<sup>-1</sup>, with 50 scans averaged. UV–vis absorption spectra for dye sensing were  
124 collected using an Agilent Cary Series spectrophotometer with a 3 mL quartz cuvette, in the  
125 wavelength range of 450–800 nm, at various time intervals.

## 126 2.3 Microstructural properties of the gels

127 Polarized optical microscopy (POM) was conducted using a polarizing microscope (ECLIPSE  
128 LV100N POL, Nikon). Scanning electron microscopy (SEM) images were obtained with a  
129 Zeiss Supra55VP, employing an acceleration voltage of 2 keV. The samples were placed on a  
130 UV-treated silicon wafer and coated with a 15-nm layer of gold, followed by a 5-nm layer of  
131 titanium to prevent charging effects. Small-angle X-ray scattering (SAXS) experiments were  
132 performed at the SWING beamline of the SOLEIL synchrotron in Orsay, France, using a fixed  
133 energy of 12 keV. The scattering patterns were recorded on a two-dimensional detector (Eiger  
134 4M, Dectris Ltd., Switzerland), which was positioned approximately 6.2 m from the sample  
135 within a vacuum tunnel. For these measurements, the dispersions were transferred into  
136 borosilicate glass capillaries (WJM Glas/Müller GmbH, Germany), which were flame-sealed  
137 and stored vertically prior to the experiments.

## 138 2.4 Rheology

139 All rheological measurements were conducted using a Modular Compact Rheometer (MCR-  
140 302). The tests utilized a cone-plate geometry (CP-25) with a diameter of 25 mm and a gap of  
141 0.104 mm, and were performed in both rotational and oscillatory modes. For the rotational  
142 tests, flow curves were analyzed, examining shear stress and viscosity as functions of shear  
143 rate. The viscoelastic properties of the samples were further explored through detailed  
144 oscillatory tests, including amplitude sweep and frequency sweep experiments. The amplitude  
145 sweep was conducted at a constant angular frequency ( $\omega$ ) of 10 rad/s, while the frequency



146 sweep was performed at a constant shear within the linear viscoelastic region (LVR) for all  
147 samples.

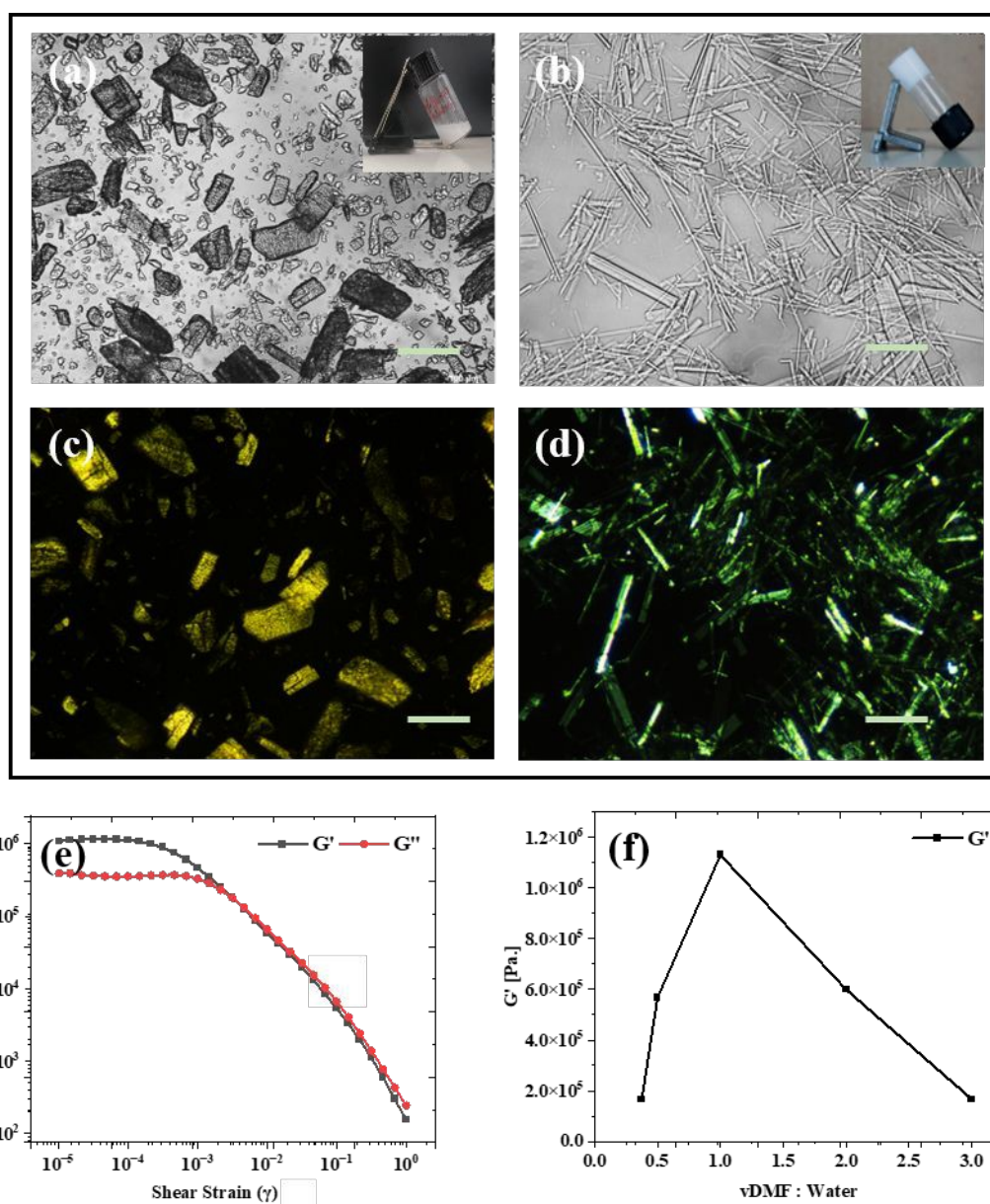
### 148 3 Results and Discussions

#### 149 3.1 Phase behavior of $\beta$ -CD in DMF and water:

150 In water,  $\beta$ -cyclodextrin ( $\beta$ -CD) forms plate-like, non-uniform crystalline assemblies that can  
151 vary in size from hundreds of nanometers to several hundred micrometers, as demonstrated by  
152 polarized optical microscopy (Figure 2a-d). The formation of these gels is favored within a  
153 specific range of  $\beta$ -CD concentrations and solvent quality. Generally, the weight ratio of the  
154 gelator within the entire gel system should not exceed 5% to 10%, as very high concentrations  
155 of the gelator could lead to insolubility in water [35-37]. In our study, we used a 30%  $\beta$ -  
156 cyclodextrin concentration, which is fully soluble in DMF (dimethylformamide), resulting in a  
157 transparent solution. Additionally, the volume ratios of DMF to water can be adjusted from 2:8  
158 to 7:3 to facilitate the formation of a stable gel [17].

159 The amplitude sweep profile (Figure 2e) of the  $\beta$ -CD gel illustrates its viscoelastic  
160 characteristics, where rheology measures the balance between elastic ( $G'$ ) and viscous ( $G''$ )  
161 responses under applied deformation. A linear viscoelastic region (LVER,  $10^{-5}$ – $10^{-2}$  strain)  
162 was observed, within which  $G'$  remains nearly constant at  $\sim 1.13 \times 10^6$  Pa and exceeds  $G''$ ,  
163 signifying a predominantly elastic, solid-like network stabilized by extensive intermolecular  
164 hydrogen bonding. Beyond the critical strain ( $\sim 10^{-3}$ ), a gradual decline in both moduli indicates  
165 microstructural softening due to partial disruption of physical cross-links. As the applied strain  
166 increased beyond  $\sim 10^{-2}$  ( $\gamma$ ), a gradual decline in  $G'$  was observed, signalling the onset of  
167 nonlinear viscoelastic behaviour.  $G''$  also showed a decrease at slightly higher strains, and at  
168 approximately  $\gamma \approx 0.3\%$ , a crossover between  $G'$  and  $G''$  was observed. This  $G'/G''$  crossover  
169 point is indicative of a yielding transition, where the material behaviour shifts from  
170 predominantly elastic to predominantly viscous. This transition suggests that the internal  
171 structure of the material begins to break down under increasing deformation, allowing it to  
172 flow. Beyond the crossover point ( $\gamma > 0.3\%$ ), both moduli decreased sharply with increasing  
173 strain, characteristic of shear-thinning or soft-solid yielding behaviour. The decline in both  $G'$   
174 and  $G''$  at high strains reflects the structural breakdown and energy dissipation that occur under  
175 large deformations. Figure 2f suggests the maximum gel strength occurred for a 1:1 volume  
176 ratio of  $\beta$ -CD (DMF) and water.





178

179 Figure 2: Polarized microscope image taken with bright field and crossed polarizers,  
 180 respectively, (a, c)  $\beta$ -CD molecules dissolved in water (crystal phase), and (b, d) for  $\beta$ -CD in a  
 181 1:1 volume ratio of DMF/ water mixture (gel phase), (scale bar-100 $\mu$ m). (e) Rheological  
 182 evidence (amplitude sweep) of gel formation in a 1:1 DMF and water volume ratio for constant  
 183 angular frequency = 10 rad/s. (f) Storage modulus ( $G'$ ) of  $\beta$ -CD for different volume ratios of  
 184 good-poor solvent.

185



### 186 3.2 Development of hierarchical microstructures with $\beta$ -CD and nanoparticles

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187 Figure 3 illustrates the progressive changes in hierarchical microstructure as the concentrations  
188 of CNC, MMT, or LAP increase from 0.5 wt% to 1.5 wt%. At a lower CNC concentration of  
189 0.5 wt%, a well-defined hierarchical network develops (Figure 3a). In contrast, at a higher CNC  
190 concentration of 1.5 wt%, stronger interactions between  $\beta$ -CD units and CNC lead to wider  
191 and more interconnected microstructural domains (Figure 3b). Increasing the CNC content to  
192 1.5 wt% significantly enhances the supramolecular assembly of  $\beta$ -CD with CNC, expanding  
193 the hierarchical framework further. A similar hierarchical structural development is evident  
194 across MMT concentrations, ranging from 0.5 wt% to 1.5 wt% (Figure 3(c-d)). Additionally,  
195 the incorporation of LAP significantly influences the self-assembly properties of the  $\beta$ -  
196 CD/LAP composite system. At lower LAP concentrations, the microstructure shows random,  
197 plate-like formations with weak birefringence observable under polarized optical microscopy,  
198 indicating minimal structural organization (Figure 3(e)). However, as LAP content increases,  
199 birefringent textures gradually appear, signalling the formation of hierarchically structured  
200 domains driven by the interaction between LAP and  $\beta$ -CD channel-type microplates (Figure  
201 3(f)) [37]. Microscopic analysis revealed the formation of distinctive bow-tie-shaped  $\beta$ -CD  
202 microstructures arising from oriented self-assembly and directional crystallization guided by  
203 nanomaterial templates (CNC, MMT, LAP), highlighting the role of surface-induced  
204 anisotropic growth in shaping supramolecular architectures (Figure S2). We also aim to explore  
205 the interactions among these components and the corresponding rheological behavior of the  
206 composites. SEM images in Figures 3(g), 3(h), and 3(i) showcase well-defined platelets formed  
207 from channel structures within the  $\beta$ -CD gel microstructure. The SEM images reveal that these  
208 platelets attach side by side, emphasizing their organized assembly.

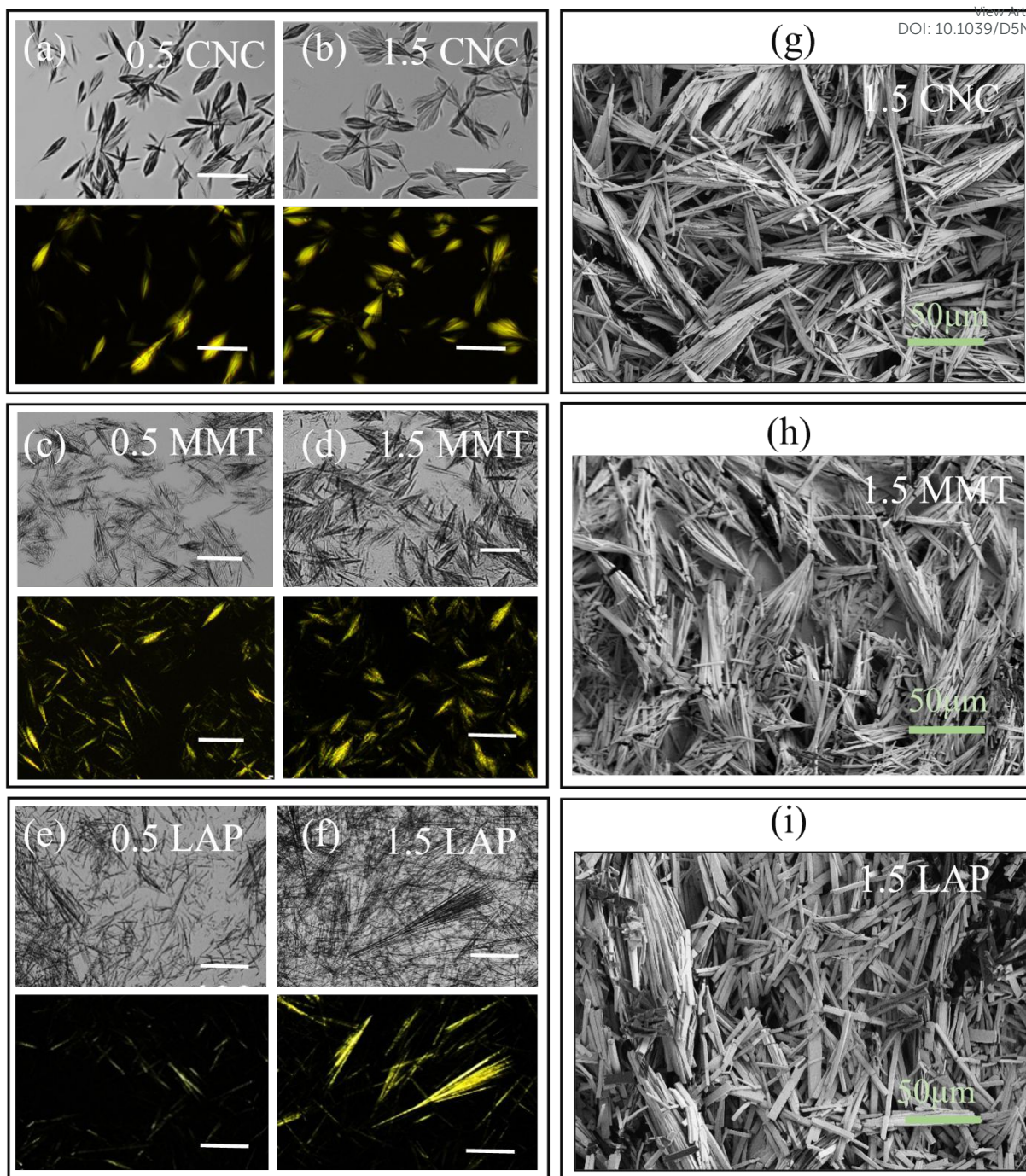
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214 Figure 3. Polarized optical microscopy images of  $\beta$ -CD/CNC (a–b) and  $\beta$ -CD/MMT (c–d), and  
 215  $\beta$ -CD/LAP (e–f) composites at varying nanoparticle concentrations (0.5 wt% and 1.5 wt%),  
 216 captured under bright-field (top) and cross-polarized light conditions (bottom). (g, h, and i)  
 217 SEM images of  $\beta$ -CD/CNC,  $\beta$ -CD/MMT, and  $\beta$ -CD/LAP composite systems, respectively. All  
 218 scale bars represent- 50  $\mu$ m.

219 **3.3 Characterization of hierarchical gels**



## 220 FTIR Analysis

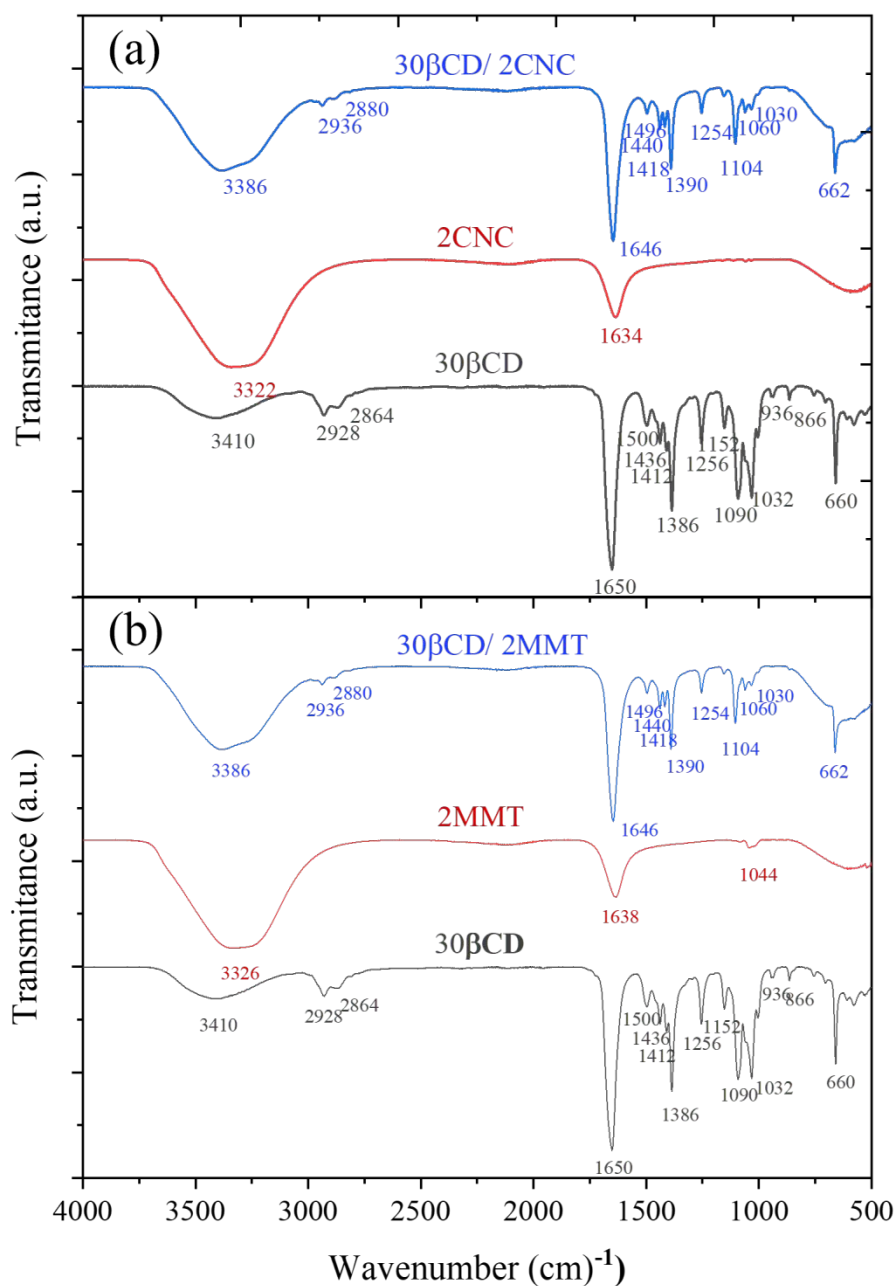
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221 Fourier-transform infrared spectroscopy was employed to analyse the functional groups and  
222 interfacial interactions within the  $\beta$ -Cyclodextrin ( $\beta$ -CD) composite systems (Figure 4). The  
223 FTIR spectrum of pure  $\beta$ -CD shows a broad O–H stretching vibration in the range of 3300–  
224 3410  $\text{cm}^{-1}$ , attributed to extensive hydrogen bonding among hydroxyl groups [38]. The band  
225 at 2920 – 2936  $\text{cm}^{-1}$  corresponds to C–H stretching, while the strong peaks between 1020–  
226 1150  $\text{cm}^{-1}$  were due to C–O–C and C–C stretching within the glucopyranose rings [39]. For  
227 pure CNC, the spectrum shows an O–H stretching band near 3322  $\text{cm}^{-1}$  and prominent C–O  
228 stretching vibrations between 1000–1160  $\text{cm}^{-1}$ , typical of cellulose. In the  $\beta$ -CD/CNC  
229 composite, the O–H band broadens and shifts slightly, while the C–O–C stretching region  
230 ( $\sim$ 1104  $\text{cm}^{-1}$ ) shows minor shifts and intensity changes, indicating stronger hydrogen bonding  
231 and possible supramolecular complexation [40]. The absence of new peaks confirms that the  
232 interaction was predominantly physical rather than covalent.

233 MMT displays a broad O–H stretching band near 3326  $\text{cm}^{-1}$  from interlayer water, an H–O–H  
234 bending vibration at 1638  $\text{cm}^{-1}$ , and a strong Si–O stretching band at 1044  $\text{cm}^{-1}$ . Bands at  
235 915  $\text{cm}^{-1}$  and 525  $\text{cm}^{-1}$  correspond to Al–Al–OH and Si–O–Al bending vibrations,  
236 respectively, typical for layered silicates [41]. In the  $\beta$ -CD/MMT composite, the O–H  
237 stretching band shifts to 3386  $\text{cm}^{-1}$  and the Si–O stretching band shifts to 1030  $\text{cm}^{-1}$ , with band  
238 broadening and reduced intensity, suggesting hydrogen bonding or electrostatic interactions  
239 between  $\beta$ -CD and MMT layers, consistent with intercalation or surface adsorption [42,43].  
240 The FTIR spectrum of Laponite shows characteristic silicate features, including a broad O–H  
241 band near 3410  $\text{cm}^{-1}$ , an –OH deformation band at  $\sim$ 1638  $\text{cm}^{-1}$ , and a strong Si–O stretching  
242 vibration at 1004  $\text{cm}^{-1}$  [44, 41] (Figure S3). In the resulting  $\beta$ -CD/LAP composite, these bands  
243 broaden and shift slightly, indicating hydrogen bonding between  $\beta$ -CD hydroxyl groups and  
244 LAP silanol sites, and possible surface adsorption or partial intercalation [45]. Thus, FTIR  
245 analysis conclusively indicates the presence of both organic and inorganic elements,  
246 highlighting hydrogen bonding as the predominant interaction mechanism responsible for the  
247 stabilisation of the composite structure.

248 Zeta potential analysis (Figure S4a-c) of the  $\beta$ -CD/nanoparticle system shows a decrease in  
249 zeta potential with increasing nanoparticle concentration, confirming the occurrence of  
250 electrostatic interactions between  $\beta$ -CD and the nanoparticles.





251

252 Figure 4: FTIR spectra of (a)  $\beta$ -CD/ CNC, and (b)  $\beta$ -CD/ MMT composites.253 **Rheological measurements**

254 In the following, we investigated the viscoelastic characteristics and hierarchical structures of  
 255  $\beta$ -CD-based supramolecular gels using amplitude sweeps, frequency sweeps, and flow  
 256 measurements (Figure 5).

257 Increasing CNC concentration from 0.5 wt% to 2 wt% influenced the gel's mechanical  
 258 robustness and internal structure (Figure 5a–b). At a lower CNC content (0.5 wt%), the gel



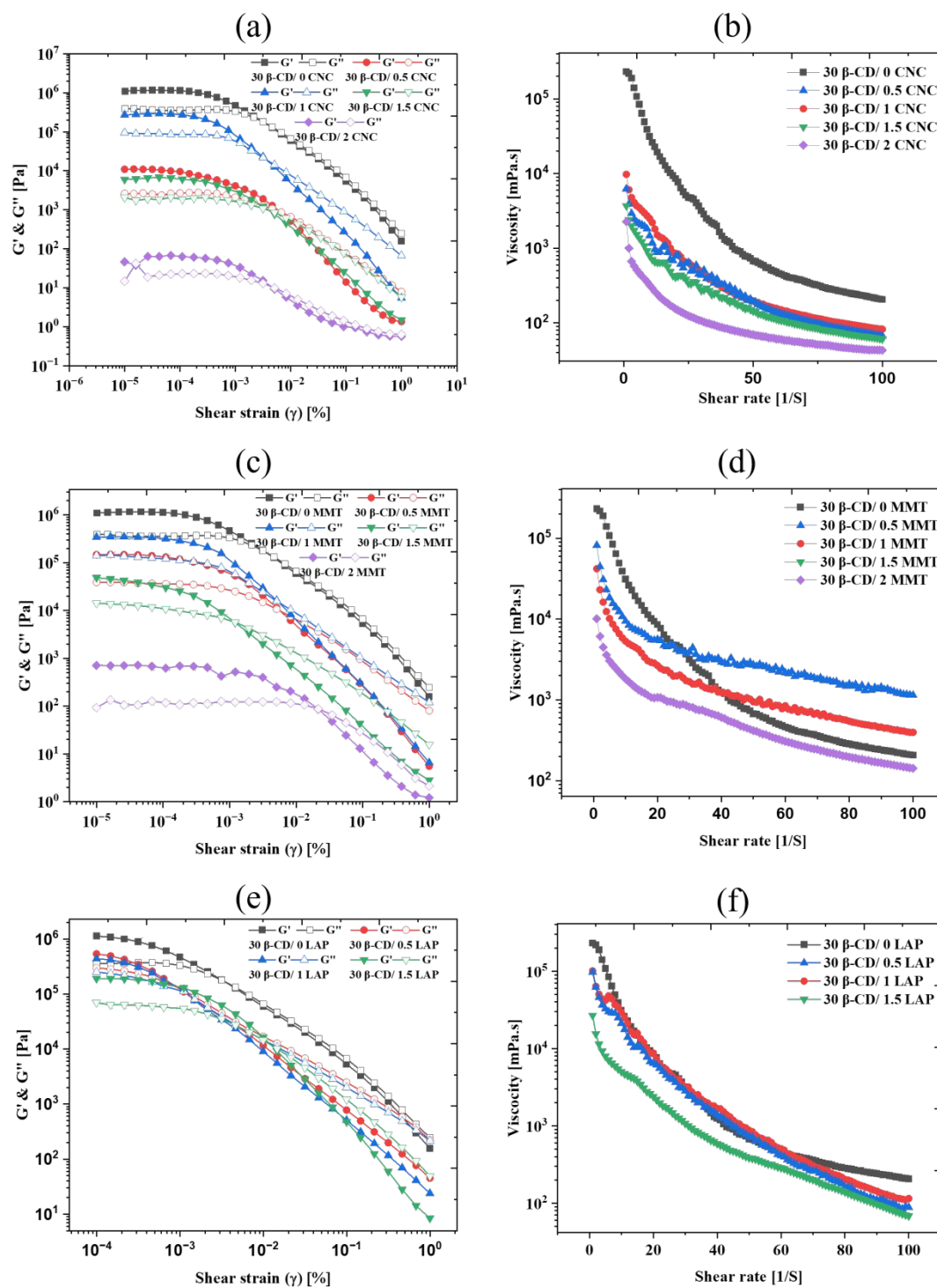
259 exhibited a well-defined linear viscoelastic region (LVR) with a  $G'$  value of approximately  $10^4$   
260 Pa, reflecting a moderately elastic network. Increasing CNC to 1.5 wt% lowered  $G'$   
261 significantly (to  $\sim 10^3$  Pa), indicating reduced network formation, by more extended  
262 hierarchical microstructures. However, at the highest CNC loading (2 wt%),  $G'$  decreased to  
263  $\sim 10^2$  Pa, suggesting that excessive CNC may interfere with optimal crosslinking, leading to a  
264 weakened network [46, 47]. The amplitude sweeps also showed that LVR expanded with CNC  
265 loading, implying greater deformation tolerance before network breakdown. Frequency sweeps  
266 showed that  $G''$  exceeded  $G'$  at higher CNC concentrations, indicating more liquid-like  
267 behaviour (Figure S5). The flow curves again confirmed pronounced shear-thinning, consistent  
268 with the progressive disruption of the self-assembled structure under shear [47-49].

269 A similar trend was observed for the  $\beta$ -CD/MMT system. Amplitude sweep data (Figure 5c)  
270 revealed a high initial  $G'$  ( $\sim 10^6$  Pa) at low MMT content (0.5 wt%), indicative of a strong,  
271 elastic gel network [49, 50]. However,  $G'$  values declined significantly with increasing MMT  
272 concentration, dropping to  $\sim 10^3$  Pa at 2 wt%. This reduction suggests that excessive MMT  
273 disrupted the  $\beta$ -CD hydrogen-bonded channels, weakening the overall supramolecular  
274 framework. Notably, all formulations exhibited a critical strain threshold ( $\sim 0.01$ ), beyond  
275 which  $G'$  declined sharply, indicating the onset of network failure and transition to a more  
276 fluid-like behavior. Flow curves (Figure 5d) again demonstrated pronounced shear-thinning  
277 behavior, typical of physically cross-linked supramolecular gels. While all MMT-containing  
278 gels displayed non-Newtonian characteristics, the steepness of the viscosity decline varied with  
279 MMT concentration, reflecting differences in structural integrity and filler-matrix interactions  
280 [60].

281 In the  $\beta$ -CD/LAP system, rheological analysis revealed consistent effects as LAP  
282 concentrations varied. Specifically, as the LAP content increased from 0.5 wt% to 1.5 wt%,  
283 there was a noticeable reduction in the storage modulus ( $G'$ ) values, decreasing from  
284 approximately  $10^6$  Pa to  $10^4$  Pa. This reduction indicates a decrease in rigidity due to the partial  
285 disintegration of the  $\beta$ -CD structured network at higher filler loadings. Frequency sweeps  
286 consistently showed a characteristic of a viscoelastic liquid, where the loss modulus ( $G''$ )  
287 exceeded the storage modulus ( $G'$ ) across all frequencies (refer to Figure S5). Additionally,  
288 flow curves (Figure 5f) displayed pronounced shear-thinning behavior, where the viscosity  
289 remained high at lower shear rates but significantly decreased as shear rates increased. This  
290 behavior is typical of physically cross-linked gels. In comparison to the CNC and MMT



291 systems, the changes in rheological parameters ( $G'$ ,  $G''$ , and viscosity,  $\eta$ ) for the  $\beta$ -CD/LAP  
 292 composite were less pronounced. Scanning electron microscopy (SEM) and polarized optical  
 293 microscopy (POM) images (Figure 3 (c, h, i)) further confirmed that the development of  
 294 microstructures was less prominent, resulting in only minor variations in viscoelasticity with  
 295 increasing LAP concentration.

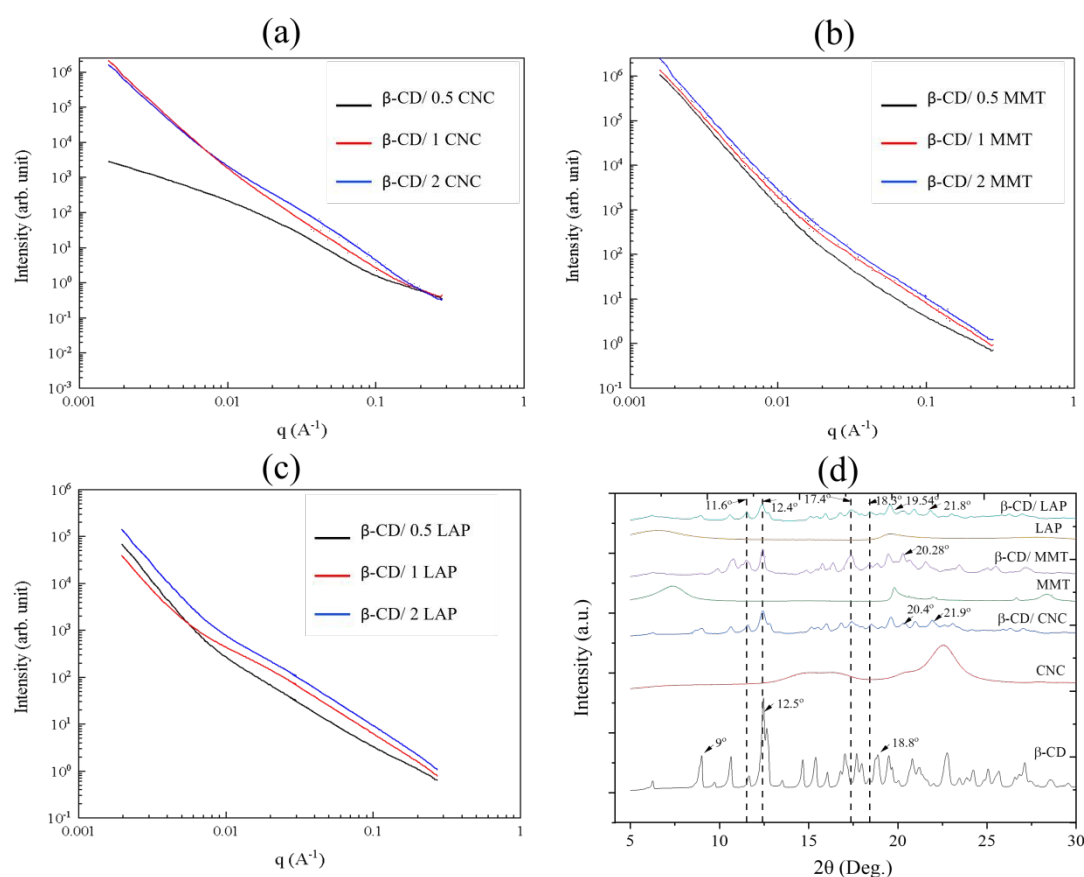


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297 Figure 5: Evolution of (*left*) amplitude sweep (constant angular frequency = 10 rad/s), and  
 298 (*right*) flow curve of  $\beta$ -CD dispersions prepared with different concentrations of (a-b) CNC.  
 299 (c-d) MMT, and (e-f) LAP respectively. Different concentrations were represented with various  
 300 colours as storage modulus  $G'$  (closed symbols) and loss modulus  $G''$  (open symbols).

### 301 SAXS and XRD Analysis



302  
 303 Figure 6:(a-c) SAXS profile of  $\beta$ -CD/ CNC,  $\beta$ -CD/ MMT, and  $\beta$ -CD/ LAP, respectively. (d)  
 304 Powder XRD patterns comparison of  $\beta$ -CD, CNC,  $\beta$ -CD/ CNC, MMT,  $\beta$ -CD/ MMT, LAP, and  
 305  $\beta$ -CD/ LAP.

306  
 307 SAXS patterns of the different  $\beta$ -CD/nanoparticles composites were displayed in Figure 6a-c.  
 308 Except for the sample prepared at 0.5 wt% CNC, all SAXS curves follow a  $Q^{-4}$  power-law  
 309 decay at low  $Q$ -values with a noticeable change in the slope for  $Q > 0.01 \text{ \AA}^{-1}$ . The increase in  
 310 the concentration of CNC or clay nanoparticles only affects the intensity of scattering curves,  
 311 as expected. However, we did not detect any characteristic correlation distances, whatever the  
 312 composition of  $\beta$ -CD/nanoparticles composites. X-ray diffraction (XRD) was extensively  
 313 utilized as a technique in the analysis of inclusion complexes to evaluate their structural



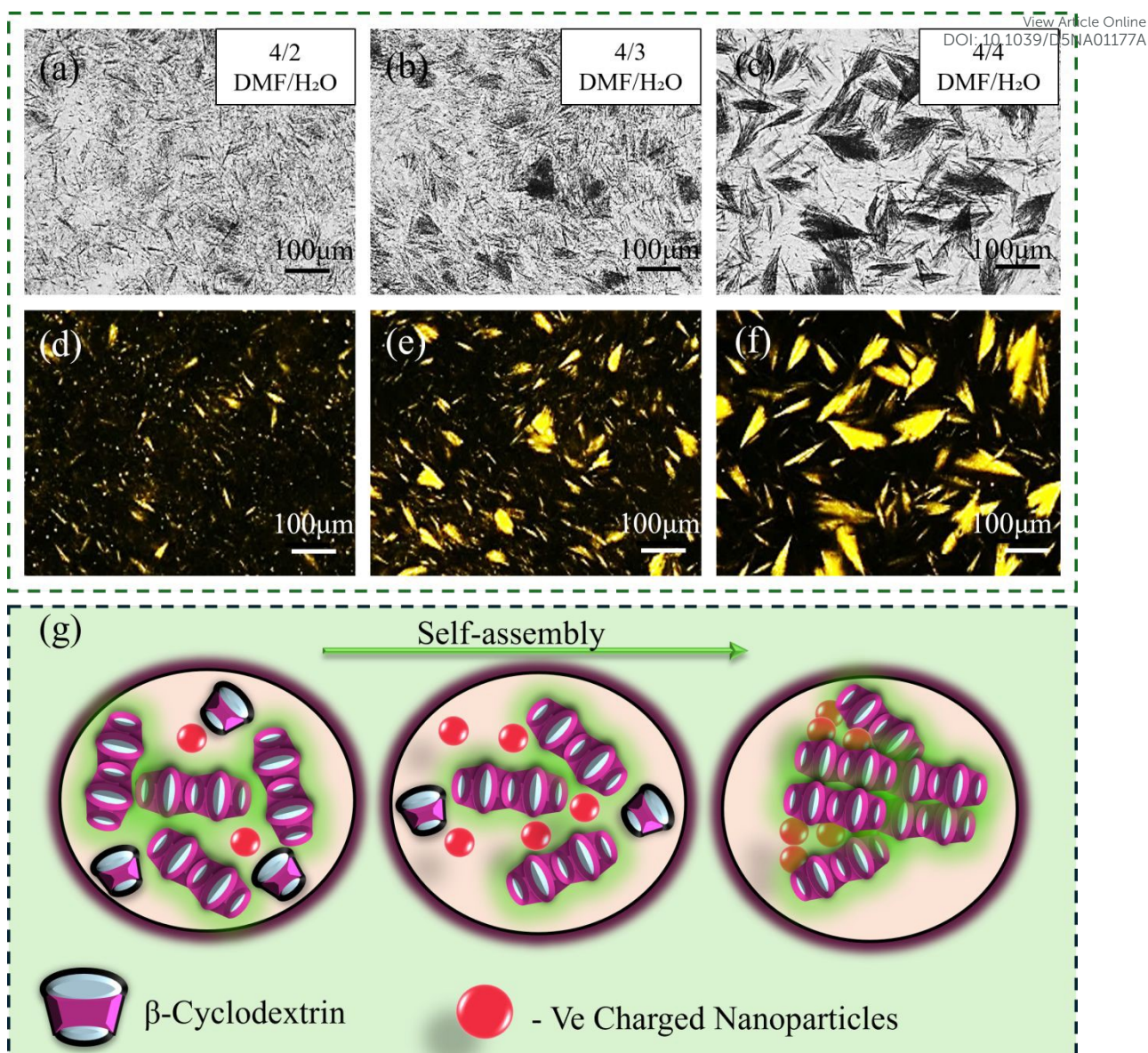
314 characteristics [52].  $\beta$ -CD mostly exhibits two characteristic patterns in crystal structures: cage-  
315 type and channel-type. Figure 6d illustrates that  $\beta$ -CD exhibits prominent peaks at  $9^\circ$ ,  $12.9^\circ$ ,  
316 and  $18.8^\circ$ , originating from its cage-like molecular structure [57]. Additionally, major peaks  
317 were observed at  $12.4^\circ$ ,  $17.2^\circ$ ,  $18.3^\circ$ , and  $11.6^\circ$ , indicating a head-to-head channel-type  
318 arrangement of  $\beta$ -CD molecules across the composite systems involving CNC, MMT, and LAP  
319 [53, 57-59]. Accordingly, Figure S6 and Table S1 show the (001) plane with an enhanced d-  
320 spacing from 13.87 to 14.11 Å, 13.57 to 14.03 Å and 13.80 to 14.12 Å for LAP, CNC, and  
321 MMT, respectively, while increasing the ratio between  $\beta$ -CD and nanoparticles as a result of  
322 the intercalation of  $\beta$ -CD into the interlayer spacing of 2D clay nanosheets [54]. New  
323 characterized peaks  $21.9^\circ$ ,  $21.55^\circ$ , and  $21^\circ$  are observed for  $\beta$ -CD/ CNC,  $\beta$ -CD/ MMT, and  $\beta$ -  
324 CD/ LAP composite systems, respectively, which can be due to the presence of nanoparticles  
325 in the system.

### 326 3.4 Phase diagrams and molecular discussion

327 Experimental observations have shown that gel formation in  $\beta$ -Cyclodextrin ( $\beta$ -CD) systems is  
328 highly dependent on the concentration of  $\beta$ -CD and the ratio of good to poor solvents. Gelation  
329 occurs exclusively within a narrow range of solvent ratios, specifically at DMF/H<sub>2</sub>O volume  
330 ratios between 1:4 and 7:3, with a critical gelation concentration (CGC) established at 1.7 wt%.  
331 To explore the influence of nanoparticles on this gelation process, small amounts of  
332 nanoparticles were dispersed in Milli-Q water and then combined with  $\beta$ -CD dissolved in  
333 DMF. The effects of both solvent composition and the concentration of nanomaterials on phase  
334 behavior—including transitions between gelation and sedimentation—were systematically  
335 studied.

336 Figure 7 a-f, demonstrates the formation of a hierarchical microstructure as the volume ratio of  
337  $\beta$ -CD (DMF)/ nanomaterials (water) increases for a constant nanomaterial concentration (2  
338 wt.% MMT). At a low ratio (4:1), gelation is predominant due to the channel-like assembly of  
339  $\beta$ -CD in response to the good-poor solvent interaction. As the ratio increases, the laminar  
340 channels formed by  $\beta$ -CD reassemble in a hierarchical manner (for 4:3 and 4:4).





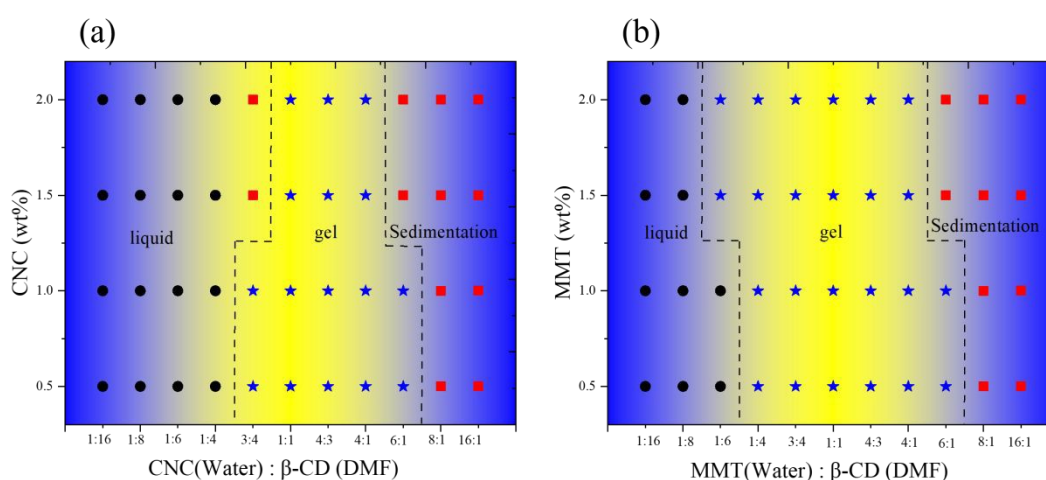
341

342 Figure 7: (a-c) Optical images and (d-f) polarized optical images between crossed polarizers of  
 343 formation of hierarchical microstructures in 30wt% β-CD/ 2wt% MMT composite gel, with  
 344 varying volume ratio of β-CD (DMF), and MMT (water) from 4:2 to 4:4. (scale bar -100  
 345 μm);(g) Schematic representation of the general gelation mechanism in the composite system,  
 346 where increasing the nanoparticle (water) content within a fixed β-CD/DMF solution promotes  
 347 supramolecular aggregation and network formation.

348 To elucidate the role of nanoparticles in the development of hierarchical microstructures, the  
 349 volume ratio of β-CD(DMF) to nanomaterial (in water) was varied from 1:16 to 16:1, and  
 350 MMT concentration was varied from 0.5 to 2 wt% (Figure 8a). At a high water content (1:16  
 351 volume ratio), the system remained a transparent liquid, indicating that neither gelation nor



352 microstructure formation occurred under these conditions. However, when the volume ratio  
 353 was adjusted to 1:6–4:1 with 30 wt%  $\beta$ -CD(DMF) and 1.5–2 wt% MMT,  $\beta$ -CD molecules were  
 354 observed to self-assemble into channel-like structures, facilitated by the presence of MMT,  
 355 leading to the formation of stable gels. At lower MMT concentrations (0.5–1 wt%), gelation  
 356 was initiated at solvent ratios of 1:4–6:1, confirming the critical role of MMT in modulating  
 357 the gelation threshold. The corresponding phase diagram (Figure 8a) clearly demonstrates the  
 358 shift in gelation behavior caused by the presence of MMT, relative to the pure  $\beta$ -CD/DMF/H<sub>2</sub>O  
 359 system.

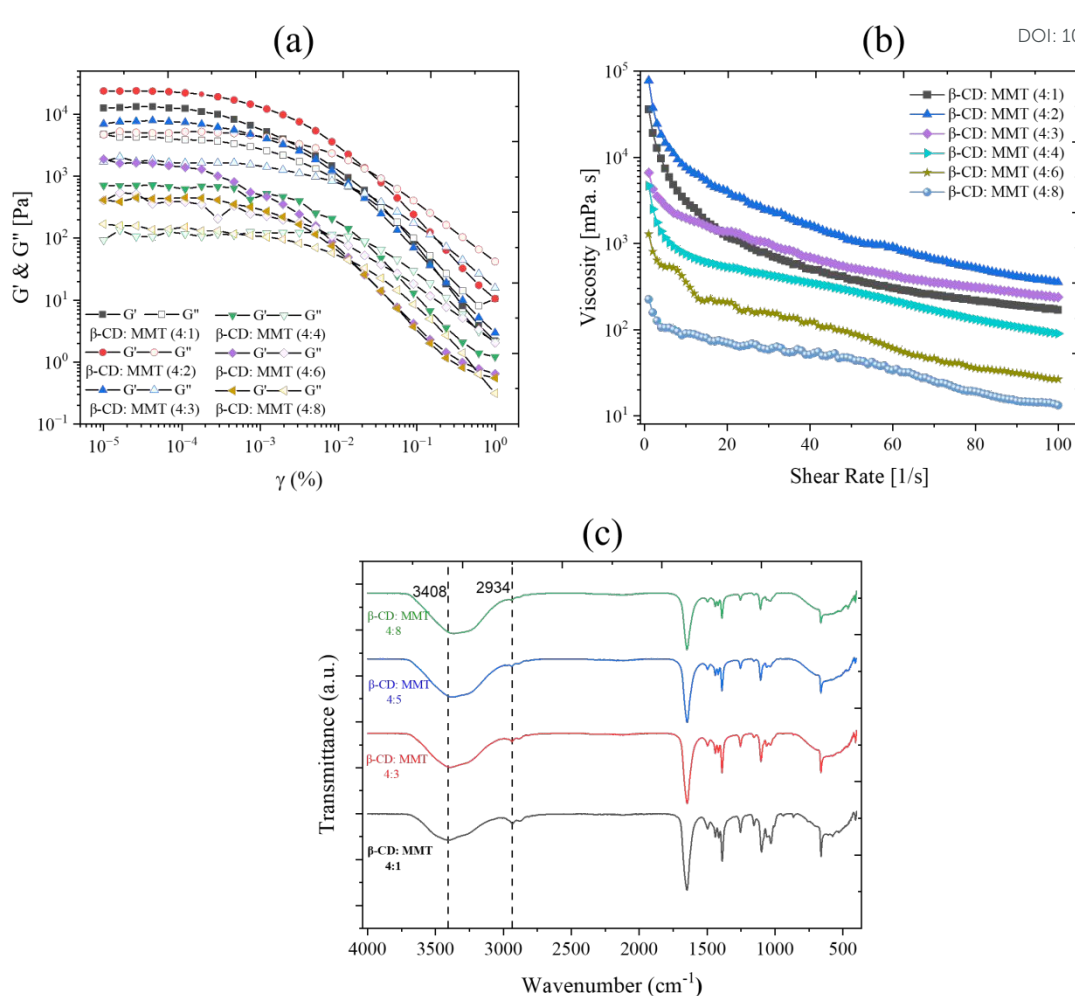


360  
 361 Figure 8: (a) and (b) represent the general phase behavior of the  $\beta$ -CD/ CNC and  $\beta$ -CD/ MMT  
 362 composite systems in varying concentrations of solvents.

363 A similar trend was observed for the other negatively charged clay mineral. In the case of  
 364 Laponite, gel formation occurred between 1:6 and 1:4 volume ratios at 0.5–1.5 wt% and shifted  
 365 slightly to 1:4–6:1 at 2 wt% (Figure S7). In contrast, with CNC, gelation occurred at a 3:4–6:1  
 366 ratio for 0.5–1.5 wt% and shifted to a 1:1–4:1 ratio for 2 wt%, highlighting that the gelation  
 367 window was dependent on both nanomaterial type and concentration (Figure 8b).

368 The results suggest that even small volume amounts of MMT significantly promote gel  
 369 formation by enabling the formation of a robust, interconnected network—something not  
 370 achieved by  $\beta$ -CD and water alone. Furthermore, stable feather-like microstructures were  
 371 observed via microscopic images (Figure 7) when the solvent system approached a 1:1  
 372 DMF/H<sub>2</sub>O ratio, indicating optimized conditions for hierarchical self-assembly in the presence  
 373 of nanoclays.



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374  
 375 Figure 9: (a) Amplitude sweep, and (b) Flow curve measurements of  $\beta$ -CD with different  
 376 concentrations of volume ratio of  $\beta$ -CD(DMF), and MMT (water) represented with various  
 377 colours as storage modulus  $G'$  (closed symbols) and loss modulus  $G''$  (open symbols). (c) FTIR  
 378 spectra of  $\beta$ -CD(DMF)/ MMT composite, with varying volume ratio of  $\beta$ -CD(DMF), and  
 379 MMT (water) from 4:1 to 4:4.

380 The oscillatory measurements of storage modulus ( $G'$ ) and loss modulus ( $G''$ ) as a function of  
 381 strain were conducted for constant angular frequency = 10 rad/s, with varying volume ratio of  
 382  $\beta$ -CD(DMF), and MMT (water) from 4:1 to 4:8 for fixed 2 wt% MMT concentration (Figure  
 383 9a). In the lower strain region,  $G'$  was observed to be greater than  $G''$  up to a strain of 0.01%.  
 384 This indicates a gel of weak structure, primarily formed by weak interactions. Notably, the  
 385 volume ratio of  $\beta$ -CD(DMF) and MMT (water) increases from 4:1 to 4:4; there was a  
 386 significant enhancement in the elastic strength, which ranged from  $10^2$  to  $10^5$  Pa. Elastic  
 387 strength increases from a 4:1 to a 4:2 volume ratio. 4:2 volume ratio of  $\beta$ -CD(DMF)/



388 MMT(Water), 4:2 volume ratio shows the maximum elastic strength. It shows a decreasing  
389 trend in the volume ratio, from 4:3 to 4:8.

390 To further investigate the role of MMT and good-poor solvent interaction in conjunction with  
391  $\beta$ -Cyclodextrin ( $\beta$ -CD), we conducted flow curve measurements (Figure 9b) by varying the  
392 shear rate at different MMT and solvent concentrations. The flow curve exhibited shear-  
393 thinning behavior at higher shear rates, indicating that the structures formed by  $\beta$ -CD in water  
394 realigned along the shear direction without disruption.

395 FTIR spectroscopy (Figure 9c) was employed to examine hydrogen bonding in  $\beta$ -CD  
396 (DMF)/MMT (water) systems with varying volume ratios. A broad O–H stretching band  
397 ( $3200\text{--}3600\text{ cm}^{-1}$ ) shifted to lower wavenumbers with increasing MMT content, indicating  
398 stronger hydrogen bonding interactions. Apart from this shift, no major spectral changes were  
399 observed, confirming that  $\beta$ -CD/MMT interactions were primarily driven by hydrogen  
400 bonding.

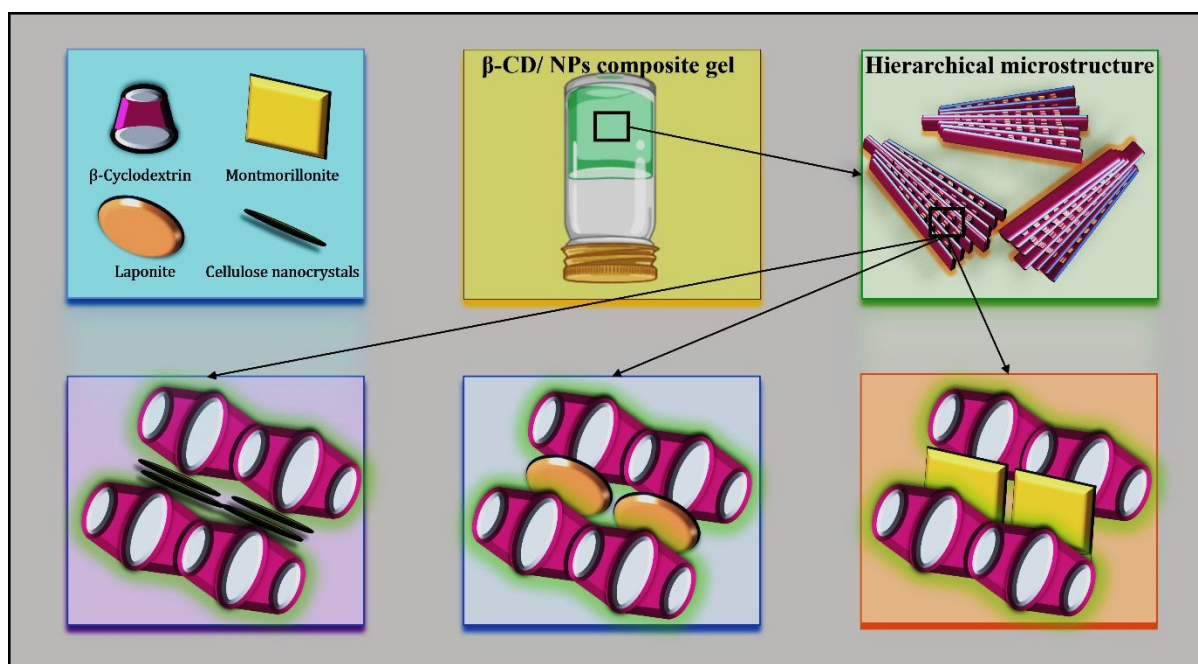
### 401 3.5 Molecular Discussion

402 The mechanism behind the formation of spherulitic microstructures in a composite system  
403 made up of  $\beta$ -Cyclodextrin ( $\beta$ -CD) and nanoparticles—such as CNC, MMT, or LAP—was  
404 systematically studied. Initially,  $\beta$ -CD nanoparticles arrange themselves into linear or channel-  
405 like structures, influenced by a complex interplay of favourable and unfavourable interactions  
406 with the solvent. These  $\beta$ -CD channels serve as essential structural components that are later  
407 assembled into larger spherulitic architectures. The introduction of nanoparticles, including  
408 CNCs, MMT, and LAP, is crucial for templating and stabilizing these structures. These  
409 nanoparticles enhance inter-channel hydrogen bonding and interfacial alignment, thus  
410 facilitating the radial expansion characteristic of spherulitic growth (Figure 10). This study's  
411 findings clearly demonstrate that the structural integrity and viscoelastic properties of  $\beta$ -  
412 Cyclodextrin-based supramolecular gels can be accurately adjusted by varying both the type  
413 and quantity of inorganic fillers. Moderate levels of fillers facilitate the creation of resilient,  
414 hierarchically structured networks with high mechanical strength, while excessive filler  
415 amounts reduce stiffness but widen the linear viscoelastic region, thereby improving flow  
416 capabilities. This presents effective strategies for engineering tunable soft materials.

417



418 Additionally, the study confirms that  $\beta$ -Cyclodextrin can form stable supramolecular gels  
 419 through hydrogen bonding and channel-type self-assembly, which can be modified using  
 420 inorganic fillers like MMT, CNC, or LAP. The findings highlight that careful manipulation of  
 421 filler content allows for precise control over the gel's mechanical properties, structural  
 422 organization, and response to shear forces. These insights provide a valuable foundation for  
 423 developing advanced hybrid gels suitable for applications in controlled release systems, soft  
 424 actuators, and functional nanocomposites.



425  
 426 Figure 10: Schematic representation of the self-assembly mechanism in  $\beta$ -CD/nanoparticle  
 427 composite gels.  $\beta$ -Cyclodextrin ( $\beta$ -CD) forms channel-type structures that act as primary  
 428 building blocks. Upon incorporation of nanoparticles such as cellulose nanocrystals,  
 429 montmorillonite, and Laponite, interfacial interactions and hydrogen bonding promote  
 430 hierarchical alignment.

#### 431 4. Sensing Application

432 Supramolecular gels are a type of material formed through non-covalent interactions or  
 433 secondary bonding, which includes hydrogen bonds, van der Waals forces, and  $\pi$ - $\pi$   
 434 interactions. What sets them apart is their ability to respond to different external stimuli, such  
 435 as changes in temperature, pH levels, exposure to light, or the presence of specific ions or  
 436 molecules. This adaptability classifies them as smart materials that can alter their properties in



437 response to environmental changes, making them highly valuable for various sensing  
438 applications.

#### 439 4.1 Salt sensing

440  $\beta$ -Cyclodextrin ( $\beta$ -CD)-based composite gels incorporating nanoparticles are formed through  
441 a self-assembly process predominantly driven by intermolecular hydrogen bonding. The  
442 resulting three-dimensional network is stabilized by hydrogen bonds among  $\beta$ -CD molecules  
443 and between  $\beta$ -CD and the embedded nanofillers, thereby imparting structural integrity to the  
444 gel. Upon the introduction of equimolar concentrations of metal salts (NaCl, KCl, MgCl<sub>2</sub>,  
445 CaCl<sub>2</sub>, AlCl<sub>3</sub>, and FeCl<sub>3</sub>), pronounced ion-dependent structural changes are observed (Figure  
446 11a). Although the gels initially remain intact, progressive phase separation, color variation,  
447 and eventual collapse over 10–30 min clearly indicate disruption of the supramolecular  
448 network.

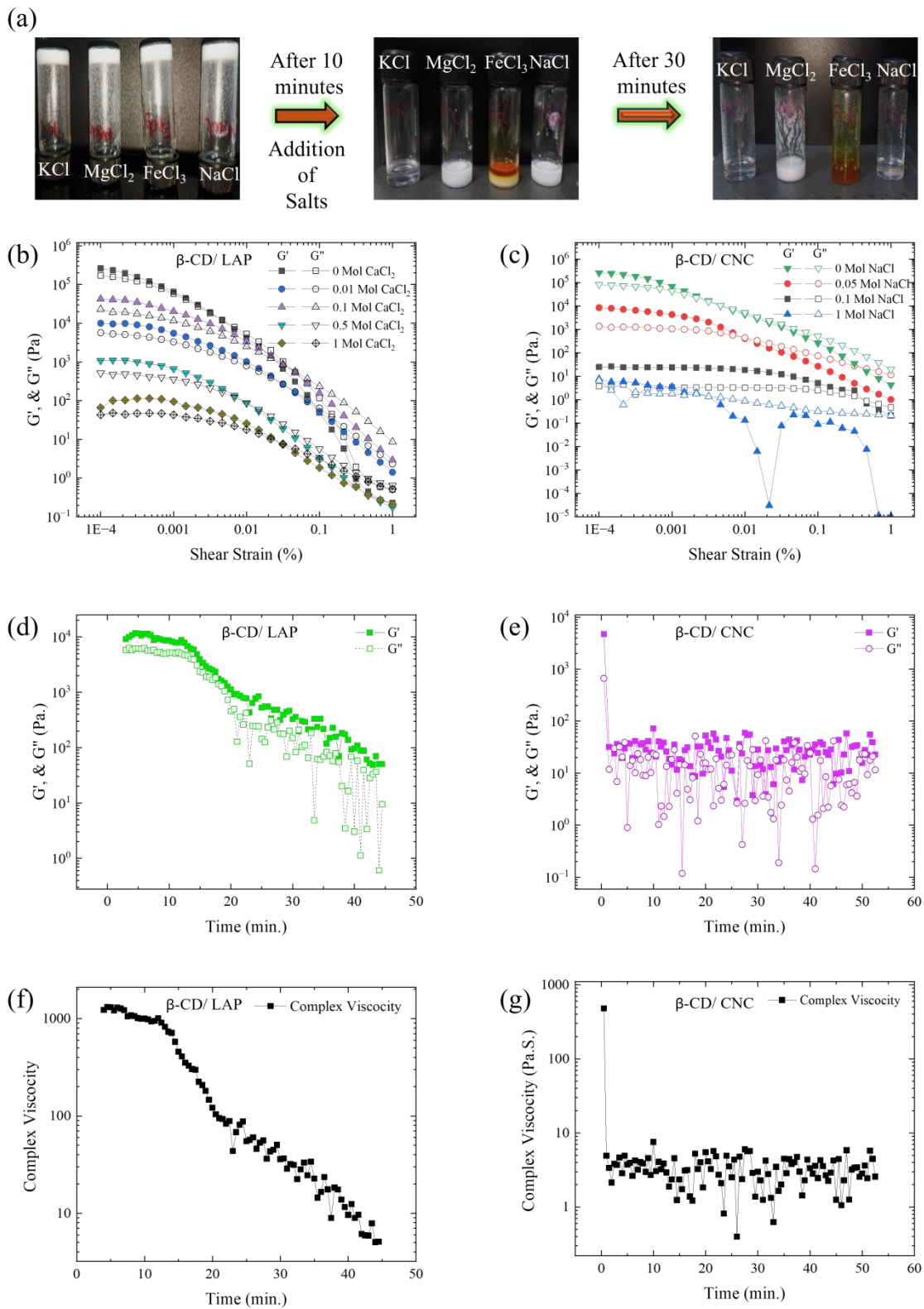
449 To substantiate these observations, quantitative rheological measurements were performed  
450 (Figure 11b–g). The  $\beta$ -CD/LAP system exhibits a systematic decrease in storage modulus ( $G'$ )  
451 with increasing CaCl<sub>2</sub> concentration (0.01–1 M), from  $\sim 10^6$  to  $10^2$  Pa, accompanied by  
452 convergence of  $G'$  and  $G''$ , indicative of rapid gel–sol transition and network breakdown  
453 (Figure 11b). In contrast, the  $\beta$ -CD/CNC system under NaCl (0.05–1 M) shows a more gradual  
454 reduction in viscoelastic strength, with  $G'$  decreasing from  $\sim 10^5$  to 10 Pa, reflecting a slower  
455 disruption of the network (Figure 11c).

456 Time-dependent analysis further reveals a pronounced decay in modulus for the  $\beta$ -CD/LAP  
457 system (Figure 11d), with  $G'$  decreasing from  $\sim 10^4$  to  $10^1$  Pa in the presence of 1 M Ca<sup>2+</sup>,  
458 confirming its higher sensitivity to divalent ions. These measurements were carried out at a  
459 constant strain of 0.001% and an angular frequency of 10 rad s<sup>-1</sup>. Consistently, the complex  
460 viscosity shows a sharp decline from  $\sim 10^3$  to 5 Pa·s for 1 M CaCl<sub>2</sub> (Figure 11f). Similarly, the  
461  $\beta$ -CD/CNC system undergoes a progressive transition from solid-like ( $G' > G''$ ) to liquid-like  
462 ( $G' \approx G''$ ) behaviour under 1 M NaCl (Figure 11g). The corresponding decrease in complex  
463 viscosity with time (Figure 11e, g) further confirms salt-induced network disassembly in both  
464 systems. Characteristic collapse times extracted from the rheological profiles provide a direct  
465 comparison of ion responsiveness, with the  $\beta$ -CD/CNC system exhibiting faster collapse than  
466 the  $\beta$ -CD/LAP system. Overall, these results establish a clear correlation between ion type,



467 supramolecular disruption, and macroscopic gel behaviour, highlighting the tuneable and ion-  
 468 sensitive nature of  $\beta$ -CD/nanomaterial composite gels for stimuli-responsive applications.

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469



470 Figure 11: (a) Time-dependent visual response of  $\beta$ -CD gels upon addition of KCl,  $MgCl_2$ ,  
 471  $FeCl_3$ , and NaCl. The gels remain intact initially, while phase separation, color variation, and  
 472 eventual collapse over 10–30 min indicate ion-specific disruption of the supramolecular  
 473 network. (b) Variation of  $G'$  and  $G''$  for the  $\beta$ -CD/LAP system with increasing  $CaCl_2$   
 474 concentration, showing a sharp decrease in  $G'$  and convergence toward  $G''$ , indicative of  
 475 network breakdown. (c) Corresponding rheological response of the  $\beta$ -CD/CNC system under  
 476 NaCl, displaying a more gradual reduction in viscoelastic strength.  
 477 (d) Time evolution of normalized modulus ( $G'/G'_0$ ), highlighting the higher sensitivity of  $\beta$ -  
 478 CD/LAP toward 1 M  $Ca^{2+}$ . (e) Time sweep of  $\beta$ -CD/CNC under 1 M NaCl, showing transition  
 479 from solid-like ( $G' > G''$ ) to liquid-like ( $G' \approx G''$ ) behavior.  
 480 (f, g) Complex viscosity vs time for  $\beta$ -CD/LAP (1 M  $CaCl_2$ ) and  $\beta$ -CD/CNC (1 M NaCl),  
 481 respectively, confirming salt-induced network disassembly.

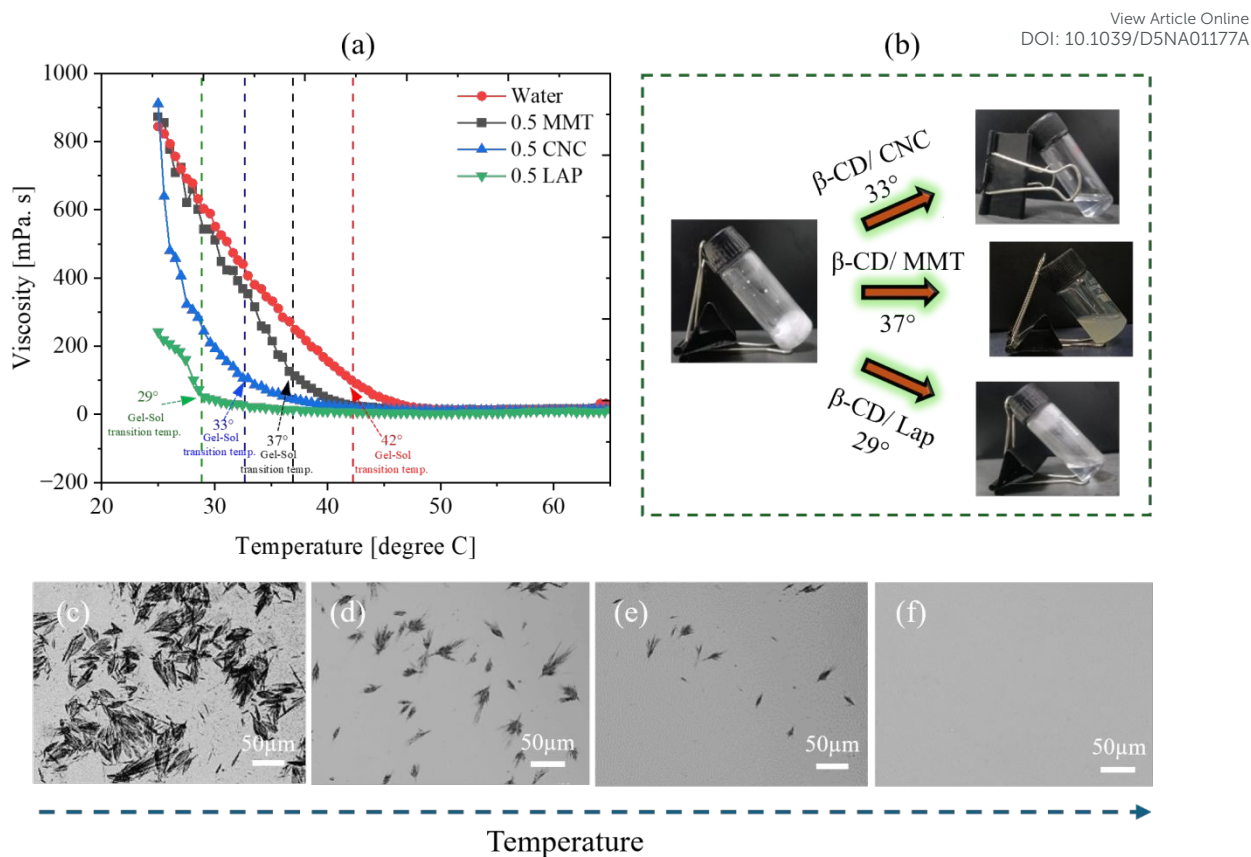
#### 482 4.2 Temperature sensing

483 Figure 12a presents the rheological response during heating, revealing the progressive  
 484 disruption of the crystalline network over the temperature range of 25–65 °C. The gel–sol  
 485 transition temperature is strongly dependent on the nature of the nanomaterial incorporated into  
 486 the  $\beta$ -CD matrix [51]. Specifically, the critical transition temperatures are 37 °C for  $\beta$ -CD/  
 487 MMT, 33 °C for  $\beta$ -CD/ CNC, and 29 °C for  $\beta$ -CD/LAP. In contrast, the pristine  $\beta$ -CD gel  
 488 remains stable up to 45.8 °C before undergoing solation, indicating superior thermal stability  
 489 in the absence of nanofillers.

490 The reduction in transition temperature upon nanoparticle incorporation can be attributed to  
 491 the influence of nanofiller geometry and surface chemistry on the supramolecular organization  
 492 of  $\beta$ -CD. Distinct morphologies—platelets (MMT), rods (CNC), and disks (LAP)—are  
 493 expected to interact differently with the  $\beta$ -CD lamellae, thereby modulating the network  
 494 architecture and its associated energy landscape. Consequently, the thermal responsiveness of  
 495 the gel is systematically altered. Notably, the process is fully reversible, as the system re-forms  
 496 a gel upon cooling to room temperature (25 °C), as shown in Figure 12.

497 Overall, these findings demonstrate that tuning the nanomaterial component enables precise  
 498 control over the thermal behaviour of the gel. The pronounced temperature sensitivity of the  
 499 phase transition underscores the potential of  $\beta$ -CD/nanomaterial composite systems as soft,  
 500 stimuli-responsive materials for temperature sensing applications.





501  
502 Figure 12: (a) Representation of Viscosity vs temperature of  $\beta$ -CD/ MMT,  $\beta$ -CD/ CNC, and  $\beta$ -  
503 CD/ LAP systems, (b) appearance of the solution phase while heating for different composites,  
504 (c)- (f) POM image for breakdown of gel network while heating from 25 degrees to 35 degrees  
505 for  $\beta$ -CD/ CNC. (scale bar 50  $\mu$ m).

### 506 4.3 Dye sensing

507 The removal of dye was quantitatively evaluated using UV- Vis absorption spectroscopy by  
508 monitoring the change in absorbance at the characteristic maximum wavelength ( $\lambda_{\max}$ ) of the  
509 dye. According to the Beer-Lambert law, the absorbance is directly proportional to the dye  
510 concentration, allowing absorbance values to be used as a direct measure of concentration  
511 during the treatment process.

512 The percentage of dye removal was calculated using the following equation:

$$513 \text{ Removal (\%)} = \frac{A_0 - A_t}{A_0} \times 100$$

514 where  $A_0$  is the initial absorbance and  $A_t$  is the absorbance at time t.



515 The adsorption performance was further evaluated in terms of adsorption capacity ( $q_t$ ), defined  
516 as the amount of dye adsorbed per unit mass of adsorbent at time  $t$ , calculated as:

$$517 \quad q_t = \frac{(C_0 - C_t) V}{m}$$

518 where  $C_0$  and  $C_t$  (mg/L) are the initial and time-dependent dye concentrations, respectively,  
519  $V$  (L) is the volume of the dye solution, and  $m$  (g) is the mass of the adsorbent. At equilibrium,  
520 the adsorption capacity is expressed as  $q_e$ , obtained by replacing  $C_t$  with the equilibrium  
521 concentration  $C_e$ .

522 Prior to analysis, all samples were centrifuged or filtered to remove suspended particles,  
523 ensuring accurate absorbance measurements. When necessary, samples were diluted to  
524 maintain absorbance within the linear range of the instrument. In cases where a calibration  
525 curve was established, concentrations were determined more precisely and used for calculating  
526 both removal efficiency and adsorption capacity.

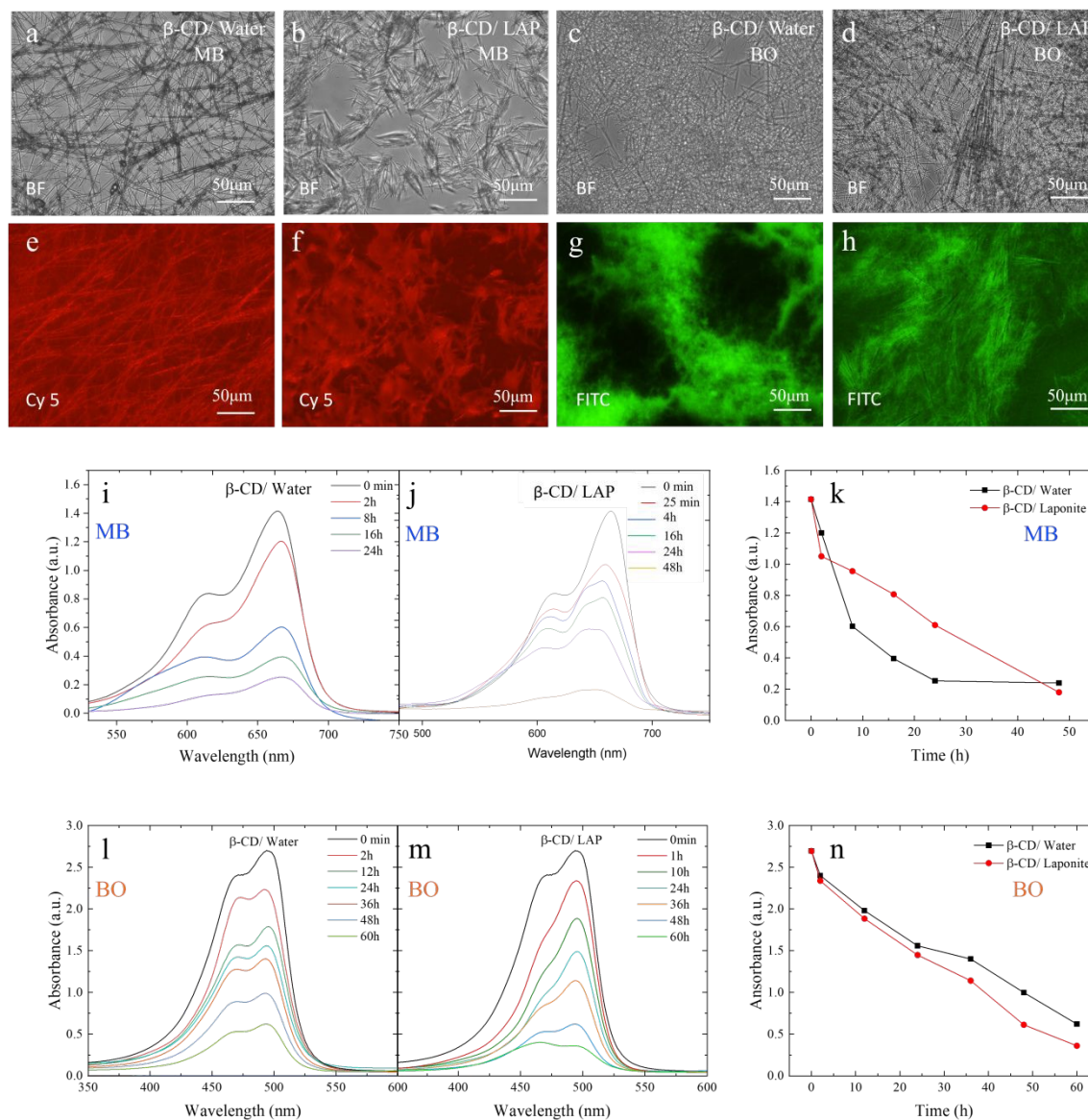
527 The microstructural features of the  $\beta$ -cyclodextrin ( $\beta$ -CD) systems were first examined by  
528 bright-field microscopy (Figure 13, a–d). The  $\beta$ -CD/water system exhibits a network of  
529 elongated plate-like structures in the presence of both methylene blue (MB) and Basic Orange  
530 14 (BO), while incorporation of Laponite leads to a denser and more compact morphology,  
531 indicative of enhanced supramolecular organization.

532 Fluorescence imaging further confirms dye incorporation and spatial distribution within the  
533 assemblies. MB-containing samples imaged using a Cy5 filter (Figure 13, e–f) display strong  
534 red emission, revealing homogeneous dye localization throughout the  $\beta$ -CD matrix, with  
535 increased intensity and uniformity in the presence of Laponite. Similarly, BO-loaded systems  
536 visualized under a FITC filter (Figure 13, g–h) exhibit bright green fluorescence, with the  $\beta$ -  
537 CD/LAP system showing a more continuous and interconnected emission pattern compared to  
538 the  $\beta$ -CD/water system. These observations suggest improved dye confinement and  
539 distribution within the hybrid system.

540 The time-dependent dye absorption behavior in both  $\beta$ -CD/water and  $\beta$ -CD/LAP systems was  
541 studied using UV–Vis spectroscopy (Figure 13, i–n). The absorbance spectra for various  
542 durations, specifically 0 minutes, 30 minutes, 1.5 hours, 4 hours, 16 hours, 24 hours, and 40  
543 hours, were recorded within the wavelength range of 500 to 750 nm for Methylene blue (MB)  
544 (Figure 13, i,j). Over time, the absorbance intensity in the  $\beta$ -CD/water mixture gradually  
545 declined, indicating continuous dye removal (Figure 13i). Notably, the prominent peak of



546 Methylene Blue, which is located between 660 and 680 nm, showed a significant reduction  
 547 over the course of 24 hours. This suggests that  $\beta$ -CD effectively interacts with the dye  
 548 molecules, likely through the formation of inclusion complexes [56].



549

550 Figure 13: Morphology, fluorescence imaging, and dye sensing behavior of  $\beta$ -CD based  
 551 systems. (a–d) Bright-field (BF) microscopy images showing the microstructure of  $\beta$ -CD in  
 552 aqueous solution and in the presence of LAP with methylene blue (MB) (b, c) and Basic Orange  
 553 (BO) (a, c). Scale bar: 50  $\mu$ m. (e, f) Fluorescence images of MB samples recorded using a Cy5  
 554 filter, displaying red emission corresponding to MB distribution in  $\beta$ -CD/water (e) and  $\beta$ -  
 555 CD/LAP (f) systems. (g, h) Fluorescence images of BO samples acquired with a FITC filter,  
 556 showing green emission in  $\beta$ -CD/water (g) and  $\beta$ -CD/LAP (h) systems. Scale bar: 50  $\mu$ m.  
 557 Time-dependent UV–Vis absorption spectra of MB and BO in  $\beta$ -CD/ Water (i, l), and  $\beta$ -



558 CD/LAP (j, m) systems at different irradiation times, (k, n) time dependent dye removal test  
559 for both  $\beta$ -CD/ Water and  $\beta$ -CD/ LAP system for MB and BO dye.

560 The dye removal process appears to consist of two stages: an initial rapid attachment followed  
561 by a slower, diffusion-dependent elimination. This is evidenced by a significant decrease in  
562 absorbance occurring within the first 1.5 hours, followed by a more gradual decline (Figure  
563 13d). In contrast, the  $\beta$ -CD/LAP composite system shows a faster and more efficient dye  
564 removal mechanism. The reduction in absorbance is more pronounced and occurs more  
565 quickly, especially within the first 25 minutes, highlighting the synergistic effect of Laponite.  
566 The layered silicate structure of Laponite likely provides additional adsorption sites, enhancing  
567 the dispersion of dye molecules. After 40 hours, the absorbance was nearly negligible,  
568 indicating that the dye had been almost completely removed. The final dye removal efficiencies  
569 were determined to be 83% and 87% for the  $\beta$ -CD/water and  $\beta$ -CD/LAP systems, respectively,  
570 based on the decrease in maximum absorbance at  $\lambda_{\max}$  as shown in Figure 13k. Under identical  
571 conditions (5 mg dye in 100 mL), the  $\beta$ -CD/LAP system exhibits a higher adsorption capacity  
572 ( $8.7 \text{ mg g}^{-1}$ ) compared to  $\beta$ -CD/water ( $8.3 \text{ mg g}^{-1}$ ), corresponding to removal efficiencies of  
573 87% and 83%, respectively, thereby confirming the enhanced performance arising from the  
574 combined effect of host-guest inclusion and Laponite-assisted adsorption.

575 To prove high absorption behaviour in Laponite based system, test was done with another dye,  
576 Basic Orange 14 (BO), characterized by an absorption maximum in the range of 480-500 nm,  
577 was used to evaluate the dye removal efficiency of  $\beta$ -CD/water (W) and  $\beta$ -CD/LAP systems.  
578 The temporal decrease in absorbance at  $\lambda_{\max}$  (Figure 13, l-n) serves as a direct indicator of dye  
579 removal from solution.

580 Both systems exhibit a progressive attenuation of the characteristic absorption band with time,  
581 confirming continuous dye uptake. However, the  $\beta$ -CD/LAP system demonstrates a  
582 significantly enhanced removal performance compared to  $\beta$ -CD/water, highlighting a clear  
583 synergistic effect.

584 Quantitative analysis reveals that the  $\beta$ -CD/water system achieves a removal efficiency of  
585 10.9% at 2h, increasing to 26.5% at 12h, 42.2% at 24h, and 48.0% at 36h. A more pronounced  
586 increase is observed at longer times, reaching 63.0% at 48 min and a maximum removal  
587 efficiency of 77.0% after 60h. In contrast, the  $\beta$ -CD/LAP system exhibits faster kinetics and  
588 higher overall efficiency, with removal values of 13.2% at 2h, 30.1% at 12h, 46.3% at 24h, and  
589 57.7% at 36h. The removal efficiency increases sharply to 77.3% at 48h and reaches a



590 maximum of 86.6% after 60h. For an initial dye loading of 9 mg in 100 mL (90 mg L<sup>-1</sup>), the  
591  $\beta$ -CD/LAP system exhibits an adsorption capacity of 15.6 mg g<sup>-1</sup> at 86.6% removal efficiency,  
592 which is significantly higher than that of the  $\beta$ -CD/water system (13.9 mg g<sup>-1</sup> at 77%),  
593 confirming enhanced dye uptake due to the synergistic effect of Laponite.

594 The findings indicate that  $\beta$ -cyclodextrin ( $\beta$ -CD) can independently enhance the removal of  
595 dye molecules through host–guest complexation. However, the addition of Laponite  
596 significantly improves both the speed and overall effectiveness of dye removal. This composite  
597 of  $\beta$ -CD and Laponite represents a promising hybrid adsorbent for the effective remediation of  
598 dyes.

## 599 Conclusions

600 We have successfully developed innovative hierarchical microstructural gels through the  
601 interaction of  $\beta$ -Cyclodextrin) with small amounts of various negatively charged nanoparticles,  
602 such as cellulose nanocrystals, montmorillonite, and Laponite, using a good–poor solvent  
603 system. The phase behavior of these gels was significantly influenced by the concentration of  
604 nanoparticles and the composition of the solvent, with gelation occurring even at low levels of  
605 nanoclay content. Rheological analysis confirmed a decrease in viscoelastic behavior with  
606 increasing nanoparticle concentration. These advanced gels exhibit responsiveness to changes  
607 in salt concentration and temperature, as well as effective dye removal capabilities,  
608 demonstrating their multifunctionality. The temperature-dependent viscosity of these  
609 composite systems revealed specific critical temperature thresholds for gel deformation,  
610 making them suitable for use as temperature sensors. Additionally, the temperature-induced  
611 phase transitions observed across different nanoparticles highlight their significant potential  
612 for applications in sensing technologies. The  $\beta$ -CD/nanomaterial composites exhibit enhanced  
613 selectivity and sensitivity for detecting cationic dyes, offering a promising foundation for  
614 advanced environmental and sensing technologies.

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## 620 Present Address



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## 624 References

625 (1) Yao, H.; Fang, H.; Wang, X. Hierarchical Micro/Nanostructures for Functional Materials.  
626 *Chem. Soc. Rev.* **2011**, 40, 3764–3787.

627 (2) Pathak, S. K.; Pujala, R. K. Soft Material–Based Photonic Crystal. In *Soft Materials for*  
628 *Functional Applications*; Springer Nature: Singapore, **2025**; pp 247–279.

629 (3) Ioele, G.; De Luca, M.; Garofalo, A.; Ragno, G. Photosensitive Drugs: A Review on Their  
630 Photoprotection by Liposomes and Cyclodextrins. *Drug Deliv.* **2017**, 24 (2), 33–44.

631 (4) Real, D. A.; Bolanos, K.; Priotti, J.; Yutronic, N.; Kogan, M. J.; Sierpe, R.; Donoso-  
632 Gonzalez, O. Cyclodextrin-Modified Nanomaterials for Drug Delivery: Classification and  
633 Advances in Controlled Release and Bioavailability. *Pharmaceutics* **2021**, 13 (12), 2131.

634 (5) Zerkoune, L.; Angelova, A.; Lesieur, S. Nano-Assemblies of Modified Cyclodextrins and  
635 Their Complexes with Guest Molecules: Incorporation in Nanostructured Membranes and  
636 Amphiphile Nanoarchitectonics Design. *Nanomaterials* **2014**, 4 (3), 741–765.

637 (6) Mazzaglia, A.; Valerio, A.; Villari, V.; Rencurosi, A.; Lay, L.; Spadaro, S.; Micali, N.  
638 Probing Specific Protein Recognition by Size-Controlled Glycosylated Cyclodextrin  
639 Nanoassemblies. *New J. Chem.* **2006**, 30 (11), 1662–1668.

640 (7) Brettner, F. E. B.; Schreiner, J.; Vogel-Kindgen, S.; Windbergs, M. Engineered Self-  
641 Assembly of Amphiphilic Cyclodextrin Conjugates for Drug Encapsulation. *ACS Biomater.*  
642 *Sci. Eng.* **2024**, 10 (1), 115–128.

643 (8) Jarak, I.; Ramos, S.; Caldeira, B.; Domingues, C.; Veiga, F.; Figueiras, A. The Many Faces  
644 of Cyclodextrins within Self-Assembling Polymer Nanovehicles: From Inclusion Complexes  
645 to Valuable Structural and Functional Elements. *Int. J. Mol. Sci.* **2024**, 25 (17), 9516.

646 (9) Smith, D. K. Supramolecular Gels—A Panorama of Low-Molecular-Weight Gelators from  
647 Ancient Origins to Next-Generation Technologies. *Soft Matter* **2024**, 20 (1), 10–70.



- 648 (10) Rumon, M. M. H.; Taufique, M. F. N.; Haque, M. A.; et al. Polysaccharide-Based  
649 Hydrogels for Advanced Biomedical Engineering Applications. *ACS Polym. Au* **2024**, *4* (6),  
650 463–486.
- 651 (11) Jain, M.; Nowak, B. P.; Ravoo, B. J. Supramolecular Hydrogels Based on Cyclodextrins:  
652 Progress and Perspectives. *ChemNanoMat* **2022**, *8* (5), e202200077.
- 653 (12) Poulson, B. G.; Alsulami, Q. A.; Sharfalddin, A.; El Agammy, E. F.; Mouffouk, F.;  
654 Emwas, A.-H.; Jaremko, Ł.; Jaremko, M. Cyclodextrins: Structural, Chemical, and Physical  
655 Properties, and Applications. *Polysaccharides* **2022**, *3* (1), 1–31.
- 656 (13) Roy, M. N.; Ekka, D.; Saha, S.; Roy, M. C. Host–Guest Inclusion Complexes of  $\alpha$ - and  $\beta$ -  
657 Cyclodextrins with  $\alpha$ -Amino Acids. *RSC Adv.* 2014, *4* (80), 42383–42390.
- 658 (14) Fermeglia, M.; Ferrone, M.; Lodi, A.; Pricl, S. Host–Guest Inclusion Complexes between  
659 Anticancer Drugs and  $\beta$ -Cyclodextrin: Computational Studies. *Carbohydr. Polym.* 2003, *53*  
660 (1), 15–44.
- 661 (15) Cruz, J. R.; Becker, B. A.; Morris, K. F.; Larive, C. K. NMR Characterization of the Host–  
662 Guest Inclusion Complex between  $\beta$ -Cyclodextrin and Doxepin. *Magn. Reson. Chem.* 2008,  
663 *46* (9), 838–845.
- 664 (16) Van De Manakker, F.; Kroon-Batenburg, L. M. J.; Vermonden, T.; Van Nostrum, C. F.;  
665 Hennink, W. E. Supramolecular Hydrogels Formed by  $\beta$ -Cyclodextrin Self-Association and  
666 Host–Guest Inclusion Complexes. *Soft Matter* 2010, *6* (1), 187–194.
- 667 (17) Xing, P.; Chu, X.; Li, S.; Hou, Y.; Ma, M.; Yang, J.; Hao, A. Self-Recovering  $\beta$ -  
668 Cyclodextrin Gel Controlled by Good/Poor Solvent Environments. *RSC Adv.* 2013, *3* (44),  
669 22087–22094.
- 670 (18) Kong, L.; Zhang, F.; Xing, P.; Chu, X.; Hao, A. A Binary Solvent Gel as Drug Delivery  
671 Carrier. *Colloids Surf., A* 2017, *522*, 577–584.
- 672 (19) Li, Z.; Hao, A.; Hao, J. Formation of Heat-Triggered Supramolecular Organogel in Which  
673  $\beta$ -Cyclodextrin Acts as Sole Gelator. *Colloids Surf., A* 2014, *441*, 8–15.
- 674 (20) Channab, B.-E.; El Idrissi, A.; Essamlali, Y.; Zahouily, M. Nanocellulose: Structure,  
675 Modification, Biodegradation, and Applications in Agriculture as Slow/Controlled Release  
676 Fertilizer, Superabsorbent, and Crop Protection: A Review. *J. Environ. Manage.* 2024, *352*,  
677 119928.



- 678 (21) Allouss, D.; Essamlali, Y.; Chakir, A.; Khadhar, S.; Zahouily, M. Effective Removal of  
679 Cu (II) from Aqueous Solution over Graphene Oxide Encapsulated Carboxymethylcellulose–  
680 Alginate Hydrogel Microspheres: Toward Real Wastewater Treatment Plants. *Environ. Sci.*  
681 *Pollut. Res.* **2020**, *27* (7), 7476–7492.
- 682 (22) Szejtli, J. Introduction and General Overview of Cyclodextrin Chemistry. *Chem. Rev.*  
683 **1998**, *98* (5), 1743–1754.
- 684 (23) Ma, M.; He, Z.; Zhou, S.; Liu, X.; Zhao, M.; Wang, X.; Liu, H.; Hao, A. A  $\beta$ -  
685 Cyclodextrin/Graphene Oxide Hybrid Gel with Smart Responsiveness. *J. Incl. Phenom.*  
686 *Macrocycl. Chem.* **2022**, *102* (1), 109–116.
- 687 (24) Sangitra, S. N.; Hotton, C.; Chazapi, I.; Paineau, E.; Pujala, R. K.  $\beta$ -Cyclodextrin and  
688 Carbonyl Iron-Based Supramolecular Gels: Microstructure and Magnetorheology. *Smart*  
689 *Mater. Struct.* **2024**, *34* (1), 015010.
- 690 (25) Park, J. S.; Jeong, S.; Chang, D. W.; Kim, J. P.; Kim, K.; Park, E.-K.; Song, K.-W.  
691 Lithium-Induced Supramolecular Hydrogel. *Chem. Commun.* **2011**, *47* (16), 4736–4738.
- 692 (26) Wang, Q.; Mynar, J. L.; Yoshida, M.; Lee, E.; Lee, M.; Okuro, K.; Kinbara, K.; Aida, T.  
693 High-Water-Content Mouldable Hydrogels by Mixing Clay and a Dendritic Molecular Binder.  
694 *Nature* **2010**, *463* (7279), 339–343.
- 695 (27) Heydari, A.; Sheibani, H. Fabrication of Poly( $\beta$ -Cyclodextrin-co-Citric Acid)/Bentonite  
696 Clay Nanocomposite Hydrogel: Thermal and Absorption Properties. *RSC Adv.* **2015**, *5* (100),  
697 82438–82449.
- 698 (28) Du, X.-Y.; Ma, K.; Cheng, R.; She, X.-J.; Zhang, Y.-W.; Wang, C.-F.; Chen, S.; Xu, C.  
699 Host–Guest Supramolecular Assembly Directing  $\beta$ -Cyclodextrin-Based Nanocrystals toward  
700 Their Robust Performances. *J. Hazard. Mater.* **2019**, *361*, 329–337.
- 701 (29) Tom, Catherine, Sagar Kumar Pathak, and Ravi Kumar Pujala. "Advances in rheological  
702 and colloidal behavior of cellulose-based materials." *Cellulose Science and Technology*.  
703 Elsevier, 2026. 113-127.
- 704 (30) Wheeler, P. A.; Wang, W.; Baker, S. P.; Giannelis, E. P. Synthesis and Characterization  
705 of Covalently Functionalized Laponite Clay. *Chem. Mater.* **2005**, *17* (11), 3012–3018.

View Article Online  
DOI: 10.1039/D3NA01177A



- 706 (31) G Frka-Petecic, Bruno, et al. "Structural color from cellulose nanocrystals or chitin  
707 nanocrystals: self-assembly, optics, and applications." *Chemical Reviews* 123.23 (2023):  
708 12595-12756. View Article Online  
DOI: 10.1039/D3NA01177A
- 709 (32) Brunchi, C.-E.; Morariu, S. Laponite®—From Dispersion to Gel—Structure, Properties,  
710 and Applications. *Molecules* **2024**, 29 (12), 2823.
- 711 (33) Kim, Bo-Hyun, et al. "Nanocomposite of polyaniline and Na<sup>+</sup>-montmorillonite clay."  
712 *Macromolecules* 35.4 (2002): 1419-1423.
- 713 (34) Madejová, J. FTIR Techniques in Clay Mineral Studies. *Vib. Spectrosc.* **2003**, 31 (1), 1–  
714 10.
- 715 (35) Lin, P.; Yan, N.; Zhou, X.; Feng, Y.; Gao, J. To Gel or Not To Gel: A Prior Prediction of  
716 Gelation in Solvent Mixtures. *Chin. Chem. Lett.* **2017**, 28 (4), 771–776.
- 717 (36) Hirst, A. R.; Coates, I. A.; Boucheteau, T. R.; Miravet, J. F.; Escuder, B.; Castelletto, V.;  
718 Hamley, I. W.; Smith, D. K. Low-Molecular-Weight Gelators: Elucidating the Principles of  
719 Gelation Based on Gelator Solubility and a Cooperative Self-Assembly Model. *J. Am. Chem.*  
720 *Soc.* **2008**, 130 (28), 9113–9121.
- 721 (37) Smith, David K. "Supramolecular gels—a panorama of low-molecular-weight gelators  
722 from ancient origins to next-generation technologies." *Soft Matter* 20.1 (2024): 10-70.
- 723 (38) Yuan, C.; Liu, B.; Liu, H. Characterization of Hydroxypropyl-β-Cyclodextrins with  
724 Different Substitution Patterns via FTIR, GC-MS, and TG-DTA. *Carbohydr. Polym.* **2015**,  
725 118, 36–40.
- 726 (39) Amani, S.; Shahin, A.; Esfandiarpour-Boroujeni, S.; Ghazimoradi, M.; Ramezani, M. The  
727 β-Cyclodextrin-Modified Nanosized ZSM-5 Zeolite as a Carrier for Curcumin. *RSC Adv.* **2019**,  
728 9 (55), 32348–32356.
- 729 (40) George, J.; Sabapathi, S. N. Cellulose Nanocrystals: Synthesis, Functional Properties, and  
730 Applications. *Nanotechnol. Sci. Appl.* **2015**, 8, 45–54.
- 731 (41) Qiao, M.; Wu, S.; Ru, Q.; Shen, J. Preparation and Properties of PU/MCMMT  
732 Nanocomposites. *Polym. Adv. Technol.* **2010**, 21 (4), 296–299.
- 733 (42) Zhang, Y.; Chen, L.; Zhao, H.; Liu, Y.; Wang, F.; Li, M.; Liu, J. Synthesis, Structure, and  
734 Photocatalytic Activity of TiO<sub>2</sub>-Montmorillonite Composites. *Catalysts* **2022**, 12 (5), 486



- 735 (43) Caccamo, M. T.; Mavilia, G.; Mavilia, L.; Lombardo, D.; Magazù, S. *Self-Assembly*  
736 *Processes in Hydrated Montmorillonite by FTIR Investigations*. **Materials** **2020**, *13* (5), 1100.
- 737 (44) Mahkam, Mehrdad, et al. "Novel methotrexate-ciprofloxacin loaded alginate-clay based  
738 nanocomposite as anticancer and antibacterial co-drug delivery system." *Advanced*  
739 *Pharmaceutical Bulletin* **11.3** (2020): 477.
- 740 (45) Wang, X.; Zhang, X.; Liu, Y.; Li, L.; Zhang, Y.; Zhao, J.; Zhang, H.; Zhang, L. Preparation  
741 and Characterization of Laponite/ $\beta$ -Cyclodextrin Nanocomposites for Controlled Drug  
742 Delivery Applications. *J. Mater. Chem. B* **2015**, *3* (6), 1234–1243.
- 743 (46) Veloso, S. R. S.; Azevedo, A. G.; Teixeira, P. F.; Fernandes, C. B. P. *Cellulose*  
744 *Nanocrystal (CNC) Gels: A Review*. *Gels* **2023**, *9* (7), 574.
- 745 (47) Morlet-Decarnin, L.; Divoux, T.; Manneville, S. Slow Dynamics and Time–Composition  
746 Superposition in Gels of Cellulose Nanocrystals. *J. Chem. Phys.* **2022**, *156* (21), 214901.
- 747 (48) Bryant, S. J.; Calabrese, V.; da Silva, M. A.; Hossain, K. M. Z.; Scott, J. L.; Edler, K. J.  
748 Rheological Modification of Partially Oxidised Cellulose Nanofibril Gels with Inorganic  
749 Clays. *PLoS ONE* **2021**, *16* (7), e0252660.
- 750 (49) Madejová, J.; Komadel, P. Baseline Studies of the Clay Minerals Society Source Clays:  
751 Infrared Methods. *Clays Clay Miner.* **2001**, *49* (5), 410–432
- 752 (50) Hsissou, R.; Bousmina, M.; El Kacemi, K.; Benzidia, B.; El Achaby, M.; Benyahia, B.;  
753 Qachaou, A.; Ezzahri, Y.; Essassi, E. M. Rheological Properties of Composite Polymers and  
754 Hybrid Nanocomposites. *Heliyon* **2020**, *6* (6), e04194.
- 755 (51) Xu, X.; Li, J.; Chen, Y.; Zhang, L.; Wang, Y.; Zhou, Z.; Liu, H. Poly(N-  
756 Isopropylacrylamide)-Based Thermoresponsive Composite Hydrogels for Biomedical  
757 Applications. *Polymers* **2020**, *12* (3), 580.
- 758 (52) Epp, J. "X-ray diffraction (XRD) techniques for materials characterization." *Materials*  
759 *characterization using nondestructive evaluation (NDE) methods*. Woodhead Publishing,  
760 2016. 81-124.
- 761 (53) Du, S.; Wang, J.; Tang, C.; Zhang, L.; Zhang, H. Pathway-Dependent Self-Assembly for  
762 Control over Helical Nanostructures and Topochemical Photopolymerization. *Angew. Chem.,*  
763 *Int. Ed.* **2024**, *63* (6), e202316863.



- 764 (54) Mandal, S.; Tom, C.; Singh, Y.; Khoder, H.; Launois, P.; Paineau, E.; Grage, S. View Article Online  
765 Schaller, M.; Alwarappan, S.; Singh, R. S.; et al. Domain Size Modulation of Clay–  
766 Nanocellulose Composites for Enhanced Na-Ion Transport. *ACS Appl. Energy Mater.* **2025**, *8*  
767 (9), 6071–6086. DOI: 10.1039/D5NA01177A
- 768 (55) Zhang, G.; Shuang, S.; Dong, C.; Pan, J. Study on the Interaction of Methylene Blue with  
769 Cyclodextrin Derivatives by Absorption and Fluorescence Spectroscopy. *Spectrochim. Acta,*  
770 *Part A* **2003**, *59* (13), 2935–2941.
- 771 (56) Łagiewka, J.; Ożga, K.; Kaczorowska, M.; Wolska, J.; Domański, A. A Novel  
772 Multifunctional  $\beta$ -Cyclodextrin Polymer as a Promising Sorbent for Rapid Removal of  
773 Methylene Blue from Aqueous Solutions. *Carbohydr. Polym.* **2023**, *307*, 120615.
- 774 (57) Muşuc, A. M.; Bibire, N.; Dinu, M.; Lupuleasa, D.; Miron, D. S.; Miron, A.; Ghiciuc, C.  
775 M. Development and Characterization of Orally Disintegrating Tablets Containing a  
776 Captopril–Cyclodextrin Complex. *Pharmaceutics* **2020**, *12* (8), 744.
- 777 (58) Okumura, H.; Kawaguchi, Y.; Harada, A. Preparation and Characterization of Inclusion  
778 Complexes of Poly(dimethylsiloxane)s with Cyclodextrins. *Macromolecules* **2001**, *34* (18),  
779 6338–6343.
- 780 (59) Kalaw, J. M.; Shigemitsu, H.; Kida, T. *2-O-Methylated  $\beta$ -Cyclodextrin as an Effective*  
781 *Building Block to Construct Supramolecular Assemblies with Various Morphologies and*  
782 *Molecular Arrangements.* *Langmuir* **2022**, *38* (27), 8407–8415
- 783 (60) Zlopasa, J.; Norder, B.; Koenders, E. A. B.; Picken, S. J. Rheological investigation of  
784 specific interactions in Na Alginate and Na MMT suspension. *Carbohydr. Polym.* **2016**, *151*,  
785 144–149.



## Data Availability Statement

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The data supporting this article have been included as part of the Supplementary Information.

