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Moringa oleifera mediated synthesis of SrO_2 and $\text{SrCO}_3@\text{GO}$ nanomaterials for energy storage applications

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We synthesized $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ nanoparticles (NPs) by treating strontium nitrate with aqueous and ethanolic extracts of *Moringa oleifera* leaves, respectively. This reaction was also performed in the presence of graphene oxide (GO) to produce $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ nanocomposites (NCs), respectively. The nanomaterials (NMs) were characterized by X-ray diffraction (XRD), Fourier-transform infrared spectroscopy (FTIR), ultraviolet-visible spectroscopy (UV-Vis), scanning electron microscopy (SEM), energy-dispersive X-ray spectroscopy (EDX), and thermogravimetric analysis/differential scanning calorimetry (TGA/DSC). FTIR spectroscopy confirmed the characteristic Sr–O vibrations, along with additional absorption bands arising from GO and phytochemical-derived organic moieties capping the nanoparticle surfaces. The synthesized NMs exhibited crystallite sizes ranging from 2 to 45 nm, particle sizes from 34 to 271 nm, and band gaps between 4.48 and 5.63 eV. EDX analysis verified the presence of Sr, O, and C, confirming their spatial distribution. TGA–DSC analysis confirmed distinct thermal stability profiles for all materials, with $\text{SrCO}_3@\text{GO}_{(\text{et})}$ exhibiting the highest mass loss and enthalpy. The synthesized nanoproducts were evaluated for electrochemical performance using cyclic voltammetry (CV) and galvanostatic charge–discharge (GCD), confirming their suitability as supercapacitor electrodes. The outstanding electrochemical performance of $\text{SrCO}_3@\text{GO}_{(\text{et})}$ (specific capacitance, 292.59 F g^{−1} at 1 mA) arises from the synergistic effects of *M. oleifera* ethanolic extract-derived phytochemicals, effective GO decoration, reduced crystallite and particle sizes, and the development of a porous, flower-like nanostructure. Collectively, these features position $\text{SrCO}_3@\text{GO}_{(\text{et})}$ as a highly promising electrode material for next-generation high-performance energy storage devices.

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1. Introduction

Strontium-based nanoparticles (NPs) have emerged as promising materials for battery–supercapacitor hybrid devices¹ and other advanced energy storage systems.² They find applications in electrodes, gas sensors, transistors, catalysts, semiconductors, and lithium-ion batteries, while in the biomedical field, they are widely employed in drug delivery, cancer therapy, and gas sensing.³ Owing to their natural abundance, cost-effectiveness, environmental friendliness, and biocompatibility, Sr-based materials are also considered promising candidates for environmental remediation, *i.e.*, the removal of pollutants from air or water, water purification, and corrosion-resistant coatings.⁴

Strontium carbonate (SrCO_3) has a low temperature coefficient, high dielectric constant and dispersion frequency constant⁵ and is important in the glass and electrical industries.⁶ The SrCO_3/SrO system in the presence of various stabilizing additives (MgO , SiO_2 and ZrO_2) was analyzed for thermochemical energy storage (TCES)

at a higher operational temperature of up to 1400 °C. This system was found to be efficient for TCES in a CO₂-closed loop, and its energy density was found to be higher compared to other such systems, *e.g.*, CaCO₃/CaO, which is one of the most studied TCES due to its high energy density and non-toxicity, as well as the cheaper and common availability of CaCO₃ sources (*e.g.*, limestone).⁷ SrMnO₃ in the presence of SrCO₃ as an impurity has shown higher electrochemical hydrogen storage potential compared to that of pure perovskites SrMnO₃, which is owing to the synergistic interaction between SrMnO₃ and SrCO₃, high conductivity of the final sample and the availability of more electroactive sites.⁸ A green hydrothermal route was employed to produce urchin-like SrCO₃ from urea and strontium nitrate. The controllable urchin-like morphology resulted in an increase in the specific capacitance of SrCO₃ by more than six times due to oxygen diffusion.⁹ Nanocomposites of SrCO₃ and SrNiO₃/NiO have shown significant potential for the photocatalytic removal/decomposition of organic pollutants.¹⁰ Carbon nanofibers decorated with Co/SrCO₃ nanorods have demonstrated high efficiency in catalyzing methanol oxidation in alkaline media, making them promising electrocatalysts for alkaline methanol fuel cells.¹¹ A SrCO₃-based nanocomposite containing MgO, CaO, and CaCO₃, synthesized using the Pechini sol-gel method at 500 °C and 700 °C, was found to be effective in eliminating crystal violet dye from aqueous solutions. This dye (crystal violet) is associated with numerous environmental and human health concerns, including systemic toxicity, cytotoxicity and carcinogenicity.¹²

Previous studies have shown that graphene oxide (GO) can be incorporated into metal oxides to form nanocomposites with enhanced properties. The addition of GO improves the uniform distribution of pores, interfacial bond formation and cement hydration process and increases the durability of the cement matrix.¹³ GO-based NMs have been recognized as highly effective for photoelectrochemical cell applications, supercapacitor technology, and various electrochemical energy storage and conversion systems.¹⁴

Numerous physical and chemical methods are currently available for the production of NPs, but they generally involve toxic chemicals and require high pressures and temperatures. However, green synthesis is an environmentally friendly and innovative approach that employs natural or eco-friendly materials, processes, and conditions to produce NPs,¹⁵ reducing the need for expensive reagents, toxic chemicals and specialized equipment, and it can produce metal-based NPs in a variety of shapes, sizes, compositions, and other characteristics.¹⁶ Plant-mediated synthesis is a safe and one-step protocol; hence, research on plant-based nanomaterials has been expanding rapidly.¹⁷ Plant-mediated syntheses of Sr-based NPs were previously reported with *Elodea canadensis*, *Ocimum sanctum*, *Albizia julibrissin*, *Vitis vinifera*, *Cymbopogon citratus*¹⁸ and *Pedalium murex*.¹⁹ However, to the best of our knowledge, the *Moringa oleifera*-mediated green synthesis of strontium peroxide NPs has not been previously reported. In fact, only a few studies exist on the plant-mediated synthesis of SrO nanopartides (NPs). Furthermore, reports on SrCO₃@GO hybrid nanocomposites are extremely limited in the current literature.

Additionally, previous studies on strontium-based NPs have predominantly focused on applications such as antioxidant activity,¹⁹ biomedical and antibacterial potentials, waste-water treatment, and electrocatalytic water-splitting¹⁸ of SrO NPs. However, the current study represents the first systematic investigation of plant-mediated SrO₂ and SrCO₃@GO nanostructures for electrochemical energy storage applications. It also explores the influence of aqueous and ethanolic extracts of *Moringa oleifera* leaves on the synthesis and structural, morphological, and electrochemical properties of SrO_{2(aq)} and SrO_{2(et)} NPs, as well as their corresponding graphene oxide-decorated composites, SrCO₃@GO_(aq) and SrCO₃@GO_(et). The synthesized NMs were characterized by XRD, FTIR, UV-visible spectroscopy, SEM, EDX, and TGA/DSC analyses. They were also evaluated for their electrochemical energy storage potential using cyclic voltammetry (CV) and galvanostatic charging-discharging (GCD) behavior.

2. Experimental

2.1. Materials and methods

Strontium nitrate (Analytical Grade), ethanol (AnalaR), sodium hydroxide (Sigma Aldrich), distilled water, hotplate, Pyrex origin glassware (beakers, Petri dish, conical flask and funnel), tripod stand, filter paper (Whatman No. 1), oven, dropper and crucible were used during the nano-synthetic processes. Graphene oxide (GO) was freshly prepared by applying a reported procedure.²⁰

The crystallinity of the synthesized NMs was determined using a Bruker-D8 X-ray diffractometer (XRD) with CuK α radiation at a scan rate of 5° m⁻¹. An FTIR (8400) spectrometer was utilized to identify the functional groups, and the electronic spectra were recorded using a UV-Visible spectrophotometer (BIOBASE BK-D560 VP). The morphologies of the synthesized NPs were determined using a scanning electron microscope (Zeiss Sigma 500 VP), coupled with energy-dispersive X-ray spectroscopy (EDX). TGA/DSC analyses were performed at 25–1000 °C using a Discovery 650 SDT thermal analyzer (TA Instruments, USA). Electrochemical studies were performed by employing a Galvanostat/Potentiostat of model CS-300.

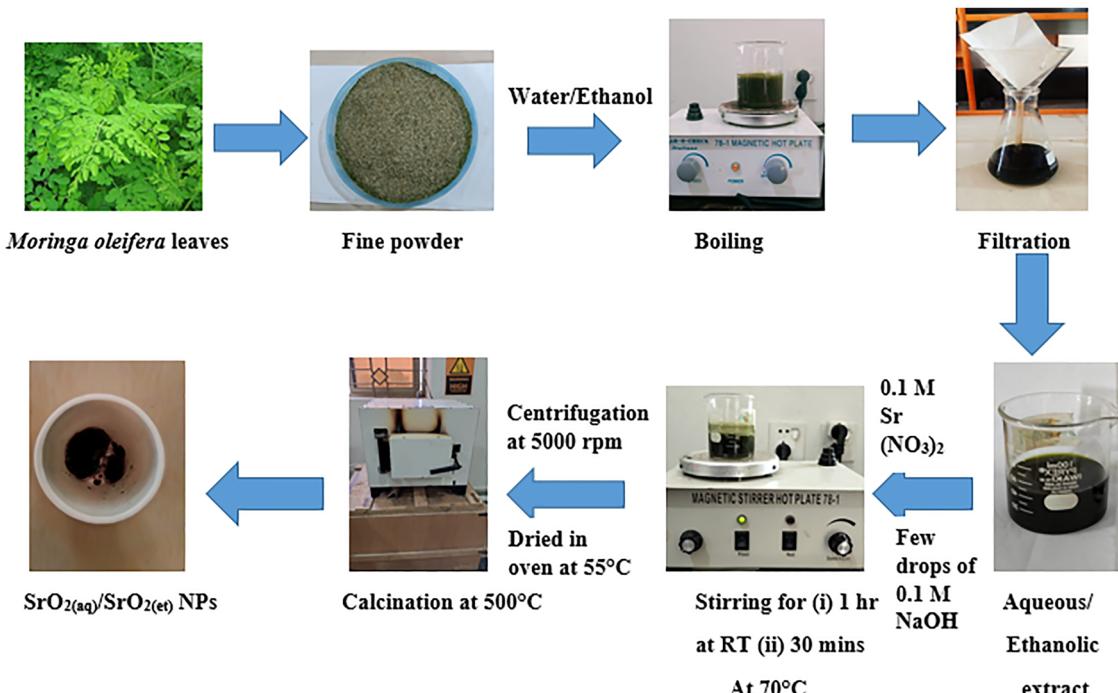
2.2. Collection of *Moringa oleifera* leaves

Moringa oleifera leaves were collected from Khanpur, Rahim Yar Khan (Punjab), Pakistan, on 28th July 2022 evening time. The leaves were washed, dried in the shade and ground into a fine powder.

2.3. Production of aqueous/ethanolic extracts of *M. oleifera* leaves

A 15 g powder of *M. oleifera* leaves was placed in 100 mL of distilled water and heated at 60 °C with stirring for 2 hours. Finally, the mixture was cooled and filtered to produce an aqueous extract (filtrate). The same procedure was adopted to produce the ethanolic extract of *M. oleifera* leaves using ethanol instead of water as a solvent (Scheme 1).





Scheme 1 *Moringa oleifera*-mediated biosynthesis of $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ NPs.

2.4. Synthesis of nanomaterials

2.4.1. *M. oleifera*-mediated synthesis of strontium peroxide ($\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ NPs). 0.26 g (0.1 M) of strontium nitrate was dissolved in deionized water (13 mL) in a 250 mL beaker, and 50 mL aqueous extract of *Moringa oleifera* leaves was added dropwise. The reaction mixture was gently stirred at room temperature for 1 hour, followed by stirring at 70 °C for 30 minutes. During heating, 5 mL of 0.1 M NaOH was added dropwise, resulting in a color change in the reaction mixture to light brown. Subsequently, the contents of the beaker were left undisturbed overnight at ambient temperature, followed by centrifugation. The obtained precipitates were washed and dried at 50 °C and finally calcined at 500 °C for 2 hours to leave behind a black solid of $\text{SrO}_{2(\text{aq})}$ NPs.

The same method was followed to produce $\text{SrO}_{2(\text{et})}$ by stirring *M. oleifera* leaf extract (50 mL) with 0.1 M strontium nitrate solution (13 mL) at room temperature for 1 hour and subsequently, at 70 °C for 30 minutes. Scheme 1 describes the flow sheet for the *M. oleifera*-mediated synthesis of $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ NPs.

2.4.2. Synthesis of graphene oxide (GO). 0.5 g of graphite powder was stirred with 250 mL of H_2SO_4 for an hour in an ice bath (0–4 °C), followed by a slow addition of 7 g of KMnO_4 and then stirring for 30 minutes. Subsequently, a total of 8 g of NaNO_3 was added in stages: 2 g during the first 24 hours, followed by an additional 5 g on the second day and 1 g on the third day. The mixture was continuously stirred throughout this period. After three days of stirring, 100 mL of H_2O was added dropwise, followed by the gradual addition of 6 mL of concentrated H_2O_2 . The reaction mixture was subsequently heated at

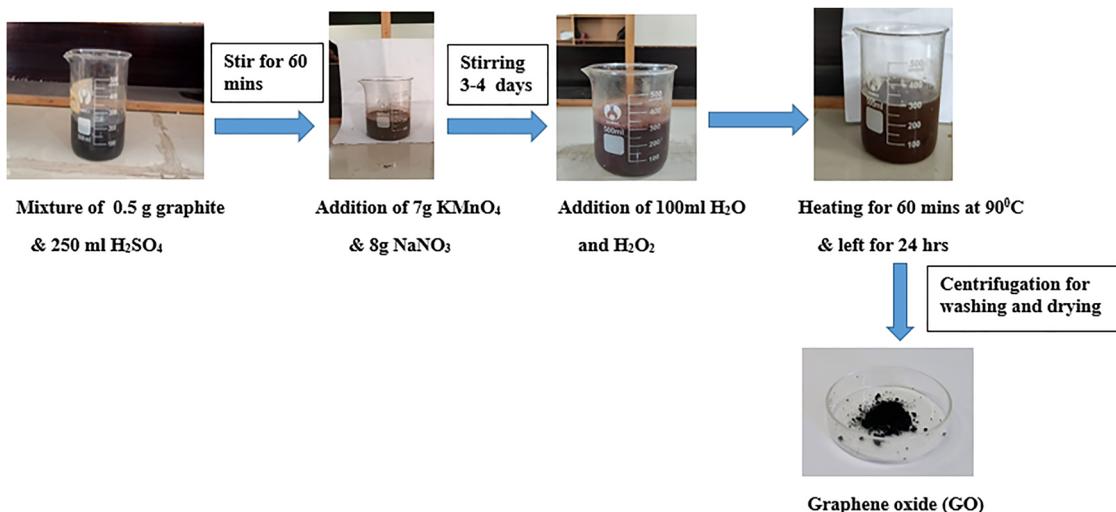
90 °C for 1.5 hours; finally, it was left for 24 hours to allow the precipitates of graphene oxide (GO) to settle down. The upper layer of liquid was poured off, leaving behind the sediment or solid material at the bottom. The remaining material was centrifuged and subsequently washed with 0.1% aqueous HCl to remove residual metal ions, followed by washing with distilled water until the pH reached neutral. The obtained product of graphene oxide (GO) was dried at 50–55 °C in an oven (Scheme 2).

2.4.3. *M. oleifera*-mediated synthesis of $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ nanocomposites. 20 mg of graphene oxide (GO) was mixed with 15 mL of distilled water, followed by the addition of 5 mL solution (0.1 M) of strontium nitrate. The reaction mixture was stirred constantly for 1 hour to ensure thorough mixing. Then, 10 mL of *Moringa oleifera* aqueous extract was added along with a few drops of 0.1 M NaOH. The mixture was stirred at 60 °C for 1.5 h during which a slight color change from light brown to dark brownish was observed. It was then cooled and centrifuged at 5000 rpm. The supernatant was decanted from the centrifuged mixture, leaving behind the precipitates of $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, which were dried in an oven at 50 °C for 1 hour.

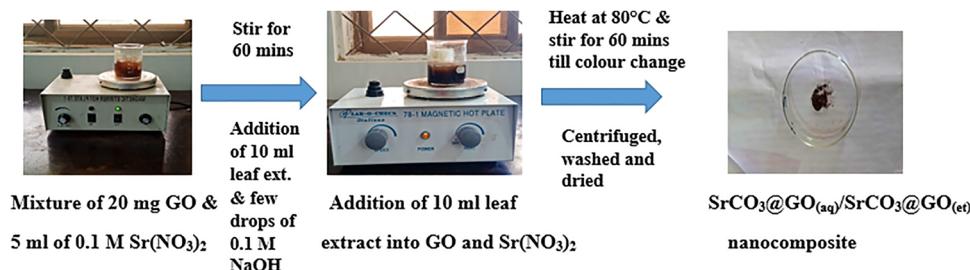
The same procedure was followed to prepare $\text{SrCO}_3@\text{GO}_{(\text{et})}$ using 10 mL ethanolic *M. oleifera* extract in place of the aqueous extract. Scheme 3 displays the *M. oleifera*-mediated synthesis of $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ nanocomposites.

2.5. Electrochemical studies

The electrochemical potential of the fabricated electrodes was investigated in terms of cyclic voltammetry (CV) and



Scheme 2 Synthetic route for graphene oxide (GO).

Scheme 3 *M. oleifera*-mediated bio-synthesis of SrCO₃@GO_(aq) and SrCO₃@GO_(et) nanocomposites.

galvanostatic charge-discharge (GCD) using a 3-electrode system and Galvanostat/Potentiostat of model CS-300. This setup included a working electrode, an Ag/AgCl (saturated KCl) reference electrode, and platinum (Pt) wire as the counter electrode with an electrolyte (6 M KOH) solution. The working electrode was prepared by forming a well-dispersed slurry containing 80 wt% of the active material (0.003 g), 15 wt% carbon black, and 5 wt% PVDF binder mixed in *N*-methyl-2-pyrrolidone (NMP). This slurry was then uniformly applied onto a nickel (Ni) foam substrate with a geometric area of 1 × 1 cm². Before deposition, the slurry was vacuum-dried at 80 °C for 24 hours to eliminate impurities. Cyclic voltammetry (CV) measurements were performed at numerous scan rates (10–50 mV s⁻¹) within a fixed potential (0–0.45 V). The specific capacitance (C_{sp} , in F g⁻¹) of each investigated sample was calculated from the CV data using eqn (1):²¹

$$C_{sp} = \left(\int I_d V \right) / (s \times m \times \Delta V), \quad (1)$$

where $\int I_d V$ denotes an area enclosed by a CV curve, s (mV s⁻¹) denotes the scan rate, m (g) represents the mass of the active material, and ΔV denotes the potential window. The specific capacitance (C_{sp}), energy density (E), and power density (P) were determined using eqn (2)–(4):²²

$$C_{sp}^d = (I_d \times T_d) / (m \times \Delta V), \quad (2)$$

$$E = (0.5 \times C_{sp}^d \times (\Delta V)^2) / (3.6), \quad (3)$$

$$P = E / \Delta t, \quad (4)$$

where I_d (A) represents the discharge current and m (g) denotes the mass of the loaded active material. The discharge time, t_d (s), is obtained from the galvanostatic charge discharge (GCD) curve, and ΔV (V) denotes the change in the potential window during the charge-discharge process.

3. Results and discussions

Moringa oleifera leaves constitute an appreciable quantity of ascorbic acid, essential amino acids, proteins, β -sitosterol, β -carotene, α -tocopherol, vitamins A and B, and polyphenols (e.g., phenolic acids and flavonoids).²³ Their aqueous and ethanolic extracts were employed as reducing, capping, and stabilizing agents for the synthesis of SrO_(aq) and SrO_(et)NPs, respectively, as well as GO-decorated SrCO₃@GO_(aq) and SrCO₃@GO_(et) NCs, respectively. The synthesis of the nanomaterials was carried out multiple times under identical conditions, and the results were found to be reproducible in terms of yield, morphology, and structural features, as confirmed by XRD and SEM analyses. Regarding yield consistency,



each batch produced comparable amounts of product (with only minor variations attributed to handling losses). For scalability, since the synthesis utilizes a simple, low-cost, and green approach (*Moringa oleifera* extracts and mild conditions), the method can potentially be scaled up without significant modifications.

We employed water/ethanol solvents during the *Moringa oleifera*-mediated synthetic pathways. Water is often regarded as the ideal solvent due to its affordability, safety, and abundant availability. Additionally, it is non-toxic, non-flammable, and cost-effective and possesses high boiling and critical points. Ethanol is a renewable source and is also called a green solvent since it shows negligible health and environmental issues and also occurs in nature. Ethanol, as a polar solvent, can effectively solubilize and extract a wide range of polar/semi-polar components. It can also act as a co-solvent with water, thus further expanding the solubility power. Due to their strong extraction efficiency, both ethanol and water are commonly used to isolate phenolic compounds from plants.¹⁷

The nanomaterials were characterized by XRD, FTIR, UV-visible spectroscopy, SEM, EDX, and TGA/DSC analyses. They were also subjected to their evaluation of electrochemical energy storage potential using cyclic voltammetry (CV) and galvanostatic charging/discharging (GCD).

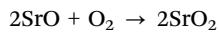
3.1. Mechanism for the synthesis of strontium peroxide (SrO₂)

The reaction mechanism involves the thermal breakdown of strontium nitrate in the presence of aqueous/ethanolic extract of *M. oleifera*, which results in the release of nitrogen dioxide (NO₂) and oxygen (O₂) gases, leaving behind strontium oxide (SrO) NPs.²⁴



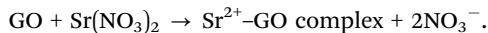
Strontium oxide (SrO) is formed by the combination of strontium ions (Sr²⁺) with oxygen (O²⁻) from nitrate ions. This type of self-decomposition is common for metal nitrates, where they break down into metal oxides and release nitrogen oxides along with oxygen gas upon heating.

The conversion of SrO to SrO₂ likely occurs when atmospheric oxygen reacts with SrO upon heating.²⁵ The driving force for this reaction is the stability of peroxide ions (O₂⁻²) in the basic reaction conditions due to the presence of 0.1 M NaOH in the reaction mixture.²⁶



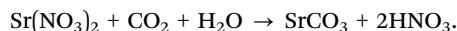
3.2. Mechanism for the synthesis of GO-decorated strontium carbonate (SrCO₃@GO)

GO comprises various oxygen-containing functional groups (−COOH, −OH, and −C=O) that can bind to Sr²⁺ ions from Sr(NO₃)₂:



Sr²⁺ ions coordinate with carboxyl (−COO[−]) and hydroxyl (−OH) groups on GO, forming an Sr-GO complex. This leads to the

chelation and functionalization of GO with Sr ions, which improves its electrical and electrochemical properties. The literature reports that graphene-based materials, including graphene oxide, are efficient in CO₂ capture and conversion.²⁷ The ability of GO to capture CO₂ is owing to its high surface area and oxygen functional groups, making it a good candidate for CO₂ adsorption and capture applications.²⁸ In the presence of CO₂ from the atmosphere in the reaction system, Sr²⁺ reacts with CO₂ and H₂O to form strontium carbonate (SrCO₃) as follows:



The SrCO₃ particles nucleate on GO, forming a SrCO₃@GO hybrid composite.

GO, with its large surface area and functional groups, can adsorb CO₂ and facilitate its diffusion in catalytic systems, potentially enhancing the efficiency of CO₂-related reactions. The thermal decomposition of strontium nitrate enhances conversion to strontium carbonate by promoting CO₂ capture, providing a large reactive surface, and aiding uniform heat distribution. The actual conversion mechanism relies on the intermediate formation of SrO, which reacts with CO₂ in the graphene oxide environment to form SrCO₃.

3.3. Structural studies by XRD analysis

XRD is primarily used to identify the crystal structures and the average sizes of the crystalline domains. The phase purity and crystalline nature of NPs can be verified by the presence of strong and sharp peaks. The obtained XRD diffractograms of our investigated materials are shown in Fig. 1(a)–(c).

In the XRD pattern of SrO_{2(aq)}, most of the reflections closely match those of SrO_{2(et)}, indicating that both synthesis routes yield comparable crystalline phases (Fig. 1(a)). SrO_{2(aq)} displayed diffraction peaks at 2θ values of 29.82, 32.92, 37.52, 42.01, 44.58, 51.33, 57.42 and 59.24°, which correspond to Miller indices of (002), (204), (110), (215), (112), (103), (131) and (202), respectively. SrO_{2(et)} showed peaks at 2θ values of 29.82, 36.45, 41.48, 44.26, 50.14 and 59.88°, which correspond to miller indices of (002), (110), (215), (112), (103) and (202), respectively. These diffraction peaks of SrO_{2(aq)} and SrO_{2(et)} show the formation of tetragonal structures (JCPDS No. 65-2652).²⁹ The peaks marked with * in Fig. 1(a) indicate the presence of residual cellulose, Sr(OH)₂ and SrO. A prominent peak at 25.53° corresponding to the (202) plane indexed the presence of cubic SrO (JCPDS No. 06-520) as a mixed phase in SrO_{2(et)} (Fig. 1(a)).³⁰ Moreover, the peak detected at 16.52° corresponded to the (020) crystal plane, indicating the presence of Sr(OH)₂ (Fig. 1(a))³¹ in the SrO_{2(et)} sample. The peaks observed at lower angles, corresponding to (1−10), (110) and (200) Miller indices, were attributed to residual cellulose originating from the plant extract used in synthesis.³² The shift of the (1−10) peak from its typical position at ~14.5° in isolated cellulose I³³ to 11.92° in both SrO_{2(aq)} and SrO_{2(et)} indicates significant lattice expansion ($d \approx 7.42 \text{ \AA}$ vs. $\sim 6.0 \text{ \AA}$). This expansion is likely caused by the interaction between the cellulose residues of *Moringa oleifera* and Sr-species during green synthesis. The obtained data thus clarify that plant-



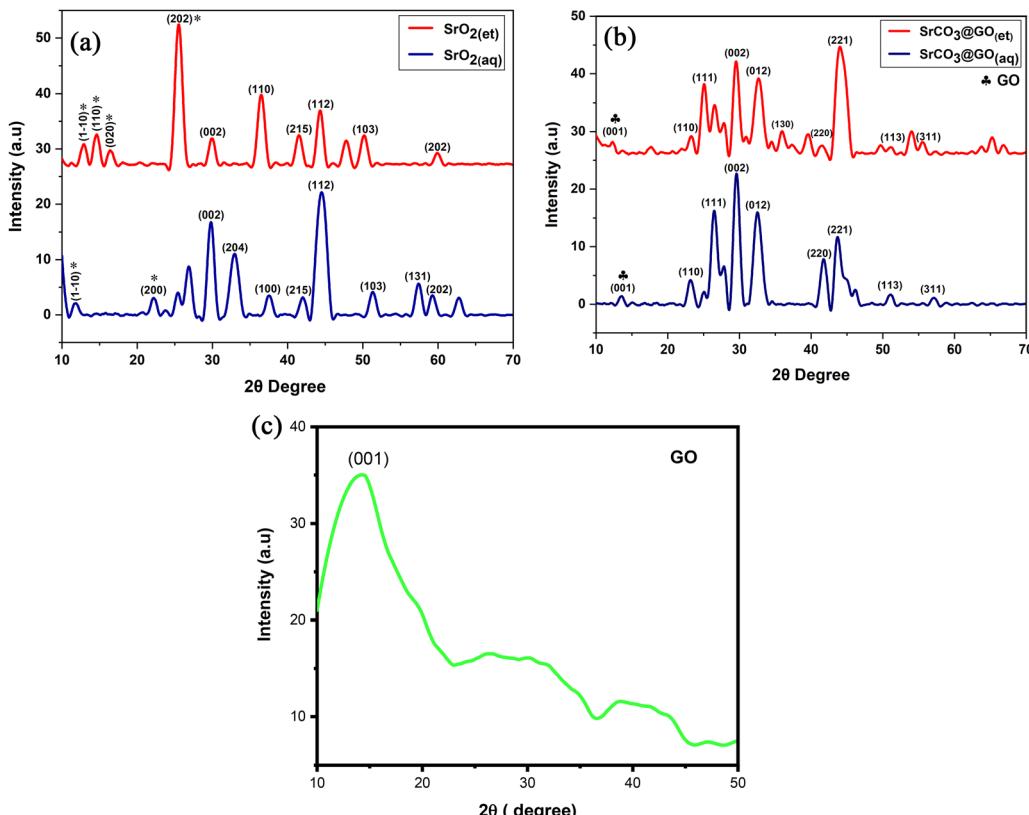


Fig. 1 (a) XRD patterns of $\text{SrO}_2(\text{aq})$ and $\text{SrO}_2(\text{et})$ NPs. (b) XRD patterns of $\text{SrCO}_3@\text{GO}(\text{aq})$ and $\text{SrCO}_3@\text{GO}(\text{et})$. (c) XRD pattern of graphene oxide.

derived biomolecules remain semi-crystalline and likely function as stabilizing/capping agents during green synthesis.

$\text{SrCO}_3@\text{GO}(\text{aq})$ (Fig. 1(b)) displayed peaks at 2 θ values of 23.16, 26.50, 29.63, 32.58, 41.70, 43.76, 51.01 and 57.19°, corresponding to the (110), (111), (002), (012), (220), (221), (113) and (311) Miller indices, respectively. $\text{SrCO}_3@\text{GO}(\text{et})$ exhibited peaks at 2 θ values of 23.16, 25.12, 29.53, 32.68, 36.01, 41.50, 44.50, 51.01, and 55.63°, corresponding to the Miller indices of (110), (111), (002), (012), (130), (220), (221), (113) and (311), respectively. The diffraction peaks of $\text{SrCO}_3@\text{GO}(\text{aq})$ and $\text{SrCO}_3@\text{GO}(\text{et})$ evidence the orthorhombic structures corresponding to JCPDS No. 05-0418 for SrCO_3 .³⁴ Graphene oxide (Fig. 1(c)) exhibited a broad diffraction peak at 2 θ = 14.70°, corresponding to the (001) plane, which indicates its layered structure and high interlayer spacing due to oxygen-containing groups.³⁵ Peaks observed at 12.27 and 13.55° in $\text{SrCO}_3@\text{GO}(\text{et})$ and $\text{SrCO}_3@\text{GO}(\text{aq})$ diffractograms, respectively, corresponded to the (001) crystal plane (marked by ♣ in Fig. 1(b)), indicating GO.³⁵ The slight shifting of these peaks (GO) toward a lower angle confirmed the successful incorporation and strong interaction of GO sheets with SrCO_3 nanoparticles.³⁶

The average crystalline sizes of NMs were determined using Scherer's eqn (5):

$$D = K\lambda/\beta \cos \theta \quad (5)$$

Average crystallite sizes of $\text{SrO}_2(\text{aq})$ (45 nm) and $\text{SrO}_2(\text{et})$ (39 nm) were lowered to 35 and 23 nm in their $\text{SrCO}_3@\text{GO}(\text{aq})$ and

$\text{SrCO}_3@\text{GO}(\text{et})$ nanocomposites, respectively. This reduction can be attributed to the strong integration of highly polar, nanosized GO sheets (~2.09 nm) within the SrO_2 crystal lattice, which hinders grain growth during composite formation. The GO sheets act as heterogeneous nucleation sites and inhibit further crystal growth, thereby reducing the crystallite sizes of $\text{SrCO}_3@\text{GO}(\text{aq})$ and $\text{SrCO}_3@\text{GO}(\text{et})$ nanocomposites. GO offers a high surface area enriched with functional groups (-OH, -COOH) that serve as anchoring sites for Sr^{2+} ion nucleation; however, the subsequent growth of large crystals is restricted, resulting in reduced crystallite sizes.

3.4. FTIR spectroscopy

The vibrational frequencies of the functional groups of the NMs were determined by performing FTIR spectroscopy in a range of 400–4000 cm^{-1} ; the obtained spectra are shown in Fig. 2 and 3.

The FAR-IR region of $\text{SrO}_2(\text{aq})$ displayed two peaks at 854 and 611 cm^{-1} corresponding to the stretching and bending vibrations, respectively, of the Sr-O bond. In $\text{SrO}_2(\text{et})$, these two peaks were observed at 858 and 698 cm^{-1} , respectively (Fig. 2). Strong intensity bands at 1072 and 1125 cm^{-1} in $\text{SrO}_2(\text{aq})$ and $\text{SrO}_2(\text{et})$, respectively, can be attributed to Sr-O-Sr stretching vibrations.³⁷ In addition to the metal oxide stretching vibrations, many additional peaks also appeared in the near-IR region, demonstrating the presence of organic moieties that originated from the aqueous and ethanolic extracts of *M. oleifera* leaves and were coated on the surfaces of $\text{SrO}_2(\text{aq})$ and $\text{SrO}_2(\text{et})$ NPs,



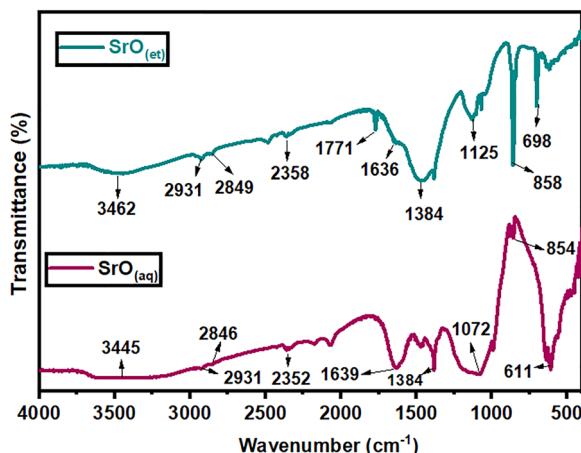


Fig. 2 FTIR spectra of *M. oleifera*-mediated $\text{SrO}_{(\text{aq})}$ and $\text{SrO}_{(\text{et})}$ NPs.

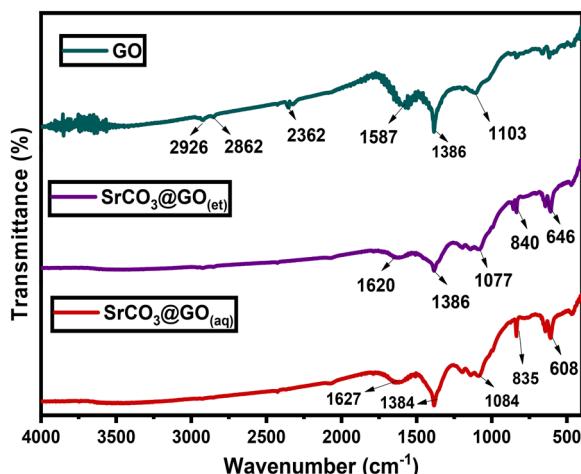


Fig. 3 FTIR spectra of GO, SrCO_3 @GO_(aq) and SrCO_3 @GO_(et).

respectively, during nano-syntheses. Such peaks were displayed at 3445, 2931, 2846, 2352, 1639 and 1384 corresponding to O-H, C-H (asymmetric), C-H (symmetric), C≡C, C=N and C-O, respectively, in $\text{SrO}_{(\text{aq})}$ and were shown at 3462, 2931, 2849, 2358, 1771, 1636 and 1384 for O-H, C-H (asymmetric), C-H (symmetric), C≡C, C=O, C=N and C-O, respectively, in $\text{SrO}_{(\text{et})}$.³⁸ The simultaneous existence of Sr-O vibrations and organic functional groups verifies that the synthesized NPs are coated with organic moieties from plant extracts. The existence of a broad band at 3445 and 3462 cm^{-1} in $\text{SrO}_{(\text{aq})}$ and $\text{SrO}_{(\text{et})}$ NPs, respectively, depicts the presence of O-H moieties from moisture or interstitial water molecules in the synthesized SrO_2 NPs.

Graphene oxide (GO) showed FTIR vibrations at 2926, 2862, 2362, 1587, 1386 and 1103 cm^{-1} (Fig. 3), which were attributed to C-H (asymmetric), C-H (symmetric), C≡O (terminal), C=C, C-O (stretching) and C-O-H, respectively.³⁵ The broad peak observed at 1587 cm^{-1} in GO is most likely due to the coupling between C=C and C=O vibrations, with the latter occurring at a slightly higher frequency than the former.

The interpretation became clearer upon comparing the FTIR spectrum of GO with those of the GO-decorated SrO nanocomposites, namely SrCO_3 @GO_(aq) and SrCO_3 @GO_(et) (Fig. 3), which exhibited distinct C=O bands at 1627 and 1620 cm^{-1} , respectively, along with separate C=C bands at 1384 and 1386 cm^{-1} , respectively. The vibration mode of C-O-H at 1103 cm^{-1} in GO was significantly shifted to 1084 and 1077 cm^{-1} in SrCO_3 @GO_(aq) and SrCO_3 @GO_(et), respectively. Additionally, the two Sr-O bands (854 and 611 cm^{-1}) of $\text{SrO}_{(\text{aq})}$ were shifted to 835 and 608 cm^{-1} in their respective GO-decorated SrCO_3 @GO_(aq) NC. There was also a significant shift of these bands from 858 and 698 cm^{-1} of $\text{SrO}_{(\text{et})}$ to 840 and 646 cm^{-1} , respectively, in their respective SrCO_3 @GO_(et) NC. The observed FTIR shifts and corresponding data clearly confirm the successful integration of SrO with GO to form SrCO_3 nanocomposites. Moreover, the FTIR spectra of SrCO_3 @GO_(aq) and SrCO_3 @GO_(et) are sharper than those of their respective $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ precursors, respectively, indicating that the SrO-decorated GO nanocomposites contain fewer organic residues.

3.5. UV-visible spectroscopy

UV-visible spectroscopy of the investigated NMs was performed in a range of 200–500 nm. Fig. 4(a) and (b) represents UV-Visible spectra and band gaps of $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$, GO, SrCO_3 @GO_(aq) and SrCO_3 @GO_(et). $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ exhibited narrow and well-defined peaks at 297 and 298 nm, respectively, indicating a uniform distribution of NPs.³⁷ GO exhibited two absorption peaks at 219 and 280 nm, which can be assigned to $\pi-\pi^*$ transitions of aromatic C=C bonds and n- π^* transitions of C=O bonds, respectively.³⁹ The peaks observed at 207 nm for SrCO_3 @GO_(aq) and 242 nm for SrCO_3 @GO_(et)⁴⁰ demonstrate the successful integration of strontium oxide nanoparticles with GO, leading to the restoration of electronic conjugation within the GO carbon framework.

From the UV-visible data, we also calculated the band gap energies to evaluate the optical and electronic properties of the synthesized nanomaterials, which are highly relevant owing to their potential application as electrode materials in supercapacitors. The band gap determines charge carrier excitation and influences electrochemical conductivity, electron transfer, and overall energy storage performance. By correlating the band gap values with the electrochemical data (CV and GCD), we were able to better understand the suitability of SrO₂ and SrCO_3 @GO nanocomposites for energy storage applications. Hence, the UV-Vis band gap analysis was performed as a complementary characterization to support electrochemical studies.⁴¹ GO had a band gap of 4.6 eV. The band gap values of $\text{SrO}_{(\text{aq})}$ and $\text{SrO}_{(\text{et})}$ NPs were found to be 5.56 and 5.36 eV, respectively, which decreased to 4.69 and 4.65 eV in the corresponding nanocomposites, SrCO_3 @GO_(aq) and SrCO_3 @GO_(et), respectively. SrCO_3 @GO_(et) exhibited the lowest band gap value (4.65 eV), indicating its superior electrochemical energy storage potential compared to the other three materials, as discussed in Sections 3.9 and 3.10. The band gap of nanomaterials typically decreases with a reduction in particle size primarily due to quantum confinement effects, changes in



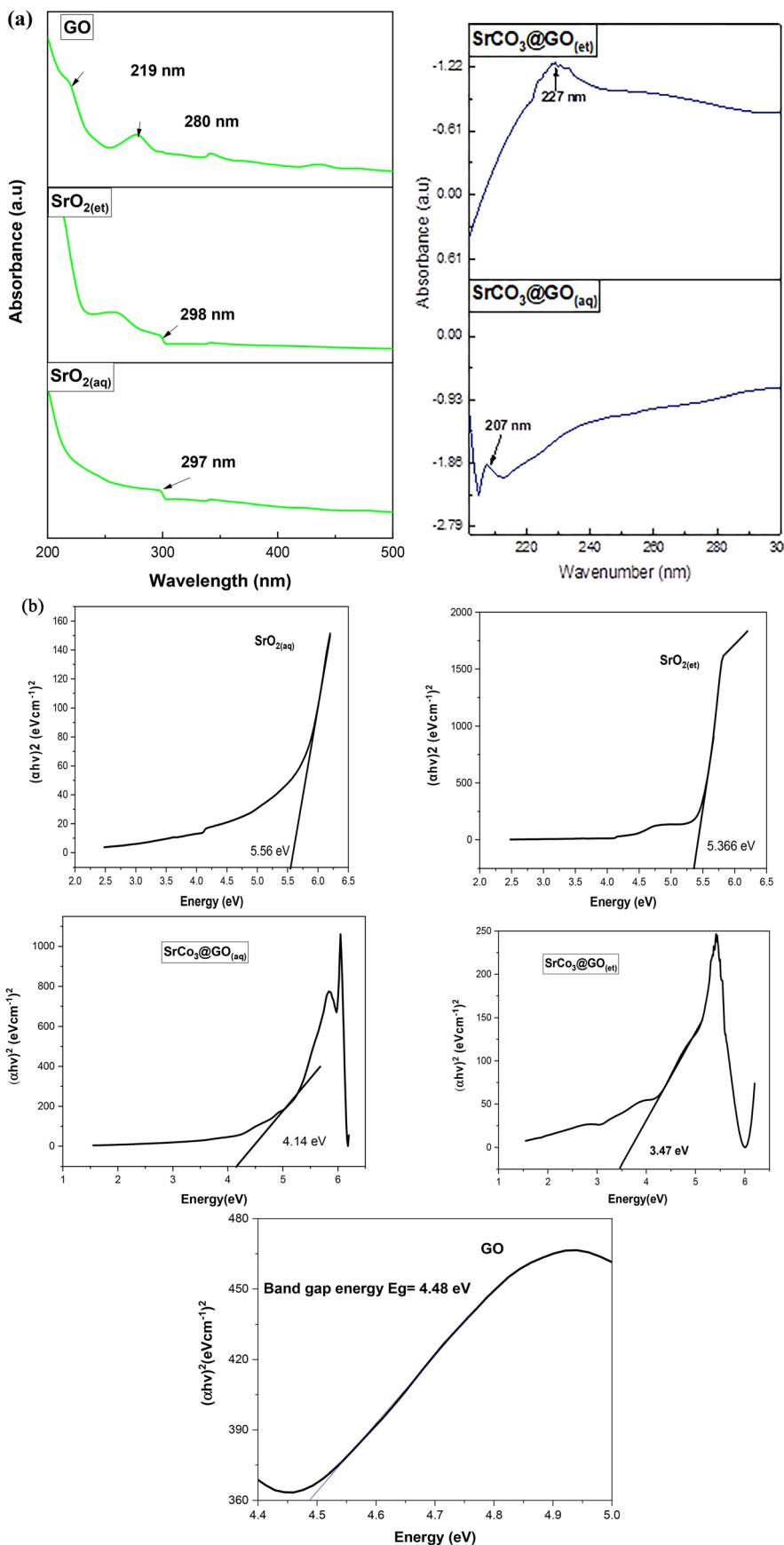


Fig. 4 (a) UV-Visible spectra of $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, $\text{SrCO}_3@\text{GO}_{(\text{et})}$ and GO. (b) Band gaps of $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, $\text{SrCO}_3@\text{GO}_{(\text{et})}$ and GO.

surface energy, and alterations in the electronic structure at the nanoscale. However, the exact trend depends on the material type and the specific size regime involved.

3.6. Scanning electron microscopy (SEM)

SEM micrographs are helpful for the calculation of sizes, determination of shapes and overall appearance of the particles. A porous and hierarchical structure facilitates ion diffusion and increases charge storage capacity. The SEM picture of $\text{SrO}_{2(\text{aq})}$ (Fig. 5(a)) revealed spherical to irregular or elongated shapes. Some particles are fused together, forming clusters and flower-like crystalline structures. The SEM image of $\text{SrO}_{2(\text{et})}$ (Fig. 5(b)) showed densely packed, polyhedral nanoparticles with significant aggregation, indicating strong particle–particle interactions

characteristic of high-surface-energy oxide systems. The nanoparticles appear well-defined with smooth surfaces although their aggregated nature suggests limited porosity at this scale. Such nanostructured morphologies can still contribute to enhanced surface area, which is beneficial for electrochemical applications, like supercapacitors, where surface-dependent ion diffusion and charge storage play a critical role. The SEM image of GO (Fig. 5(e)) shows thin, wrinkled, and randomly entangled sheets, indicating the disruption of the ordered graphite layers upon oxidation. This morphology is consistent with previously reported SEM observations of GO, which typically exhibit layered, sheet-like structures with surface wrinkles.³⁵

The morphologies of $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ underwent significant changes upon GO decoration, forming $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and

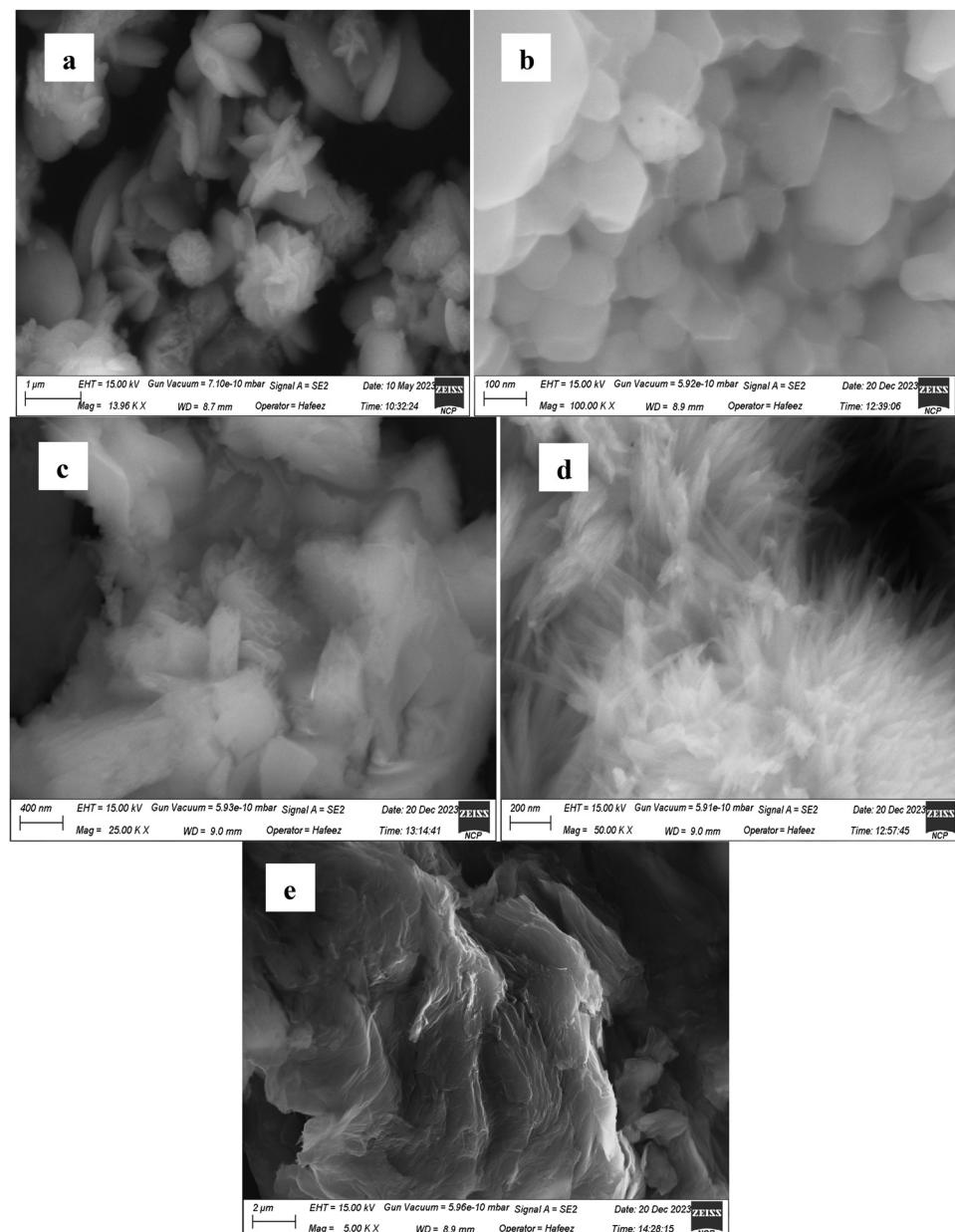


Fig. 5 SEM images of (a) $\text{SrO}_{2(\text{aq})}$ (b) $\text{SrO}_{2(\text{et})}$, (c) $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, (d) $\text{SrCO}_3@\text{GO}_{(\text{et})}$, and (e) GO.



$\text{SrCO}_3@\text{GO}_{(\text{et})}$ NCs (Fig. 5(a)–(e)). The SEM image of $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ (Fig. 5(c)) showed irregular, plate-like structures with noticeable aggregation. The broad, flat surfaces indicate the formation of micro/nanocrystals with layered arrangements, which is consistent with a crystalline nature. Such plate-like morphologies can provide an enhanced surface area, which may improve catalytic activity. The SEM micrograph of $\text{SrCO}_3@\text{GO}_{(\text{et})}$ (Fig. 5(d)) reveals a highly porous, fluffy morphology, indicating a nanostructured or flower-like growth pattern. The observed surface features consist of densely packed, sharp, needle-like structures that may correspond to nanorods, nanoflakes, or agglomerated NPs. These high-aspect-ratio features contribute significantly to an increased surface area, which is advantageous for catalytic, electrochemical, or sensing applications. The contrast between the bright and dark regions in the image (Fig. 5(d)) reflects variations in local material density and electron scattering, suggesting a heterogeneous distribution of the SrCO_3 NPs and graphene oxide (GO) sheets. The intimate interaction between the GO layers and the SrCO_3 NPs facilitates uniform dispersion and strong interfacial adhesion likely due to π - π stacking interactions and surface functional groups present on GO. Such a hierarchical nanostructure is expected to promote efficient charge carrier mobility across the composite. Specifically, the GO matrix provides a conductive pathway that supports rapid electron transport between adjacent SrCO_3 NPs. This synergistic architecture enhances the separation and migration of photo- or electro-generated charge carriers, thereby improving the overall electronic conductivity and electrochemical performance of the material.⁴² These attributes make $\text{SrCO}_3@\text{GO}_{(\text{et})}$ a promising candidate for high-performance applications in energy storage. The SEM morphology of strontium peroxide (SrO_2) and strontium carbonate (SrCO_3) synthesized in different solvent systems—namely aqueous and ethanolic plant extracts—exhibits marked variations that are attributable to differences in reaction kinetics, solvent polarity, nucleation rates, and crystal growth mechanisms. These factors critically influence the resulting particle size, degree of agglomeration, and surface architecture. In aqueous media, faster ion diffusion and higher polarity often promote rapid nucleation and irregular agglomeration, leading to more compact or amorphous structures. In contrast, ethanolic extracts, with their lower polarity and potential for selective interaction with precursor ions or intermediate species, can slow crystal growth and favor the formation of more defined anisotropic nanostructures.⁴³ Such solvent-mediated effects are clearly reflected in the distinct morphological features observed in the SEM micrographs. The average particle sizes of $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$ and GO were found to be 223, 271 and 43 nm, respectively. Upon GO decoration, the particle sizes of $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ were significantly reduced to 203 and 34 nm in the corresponding $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ nanocomposites, respectively. Notably, $\text{SrCO}_3@\text{GO}_{(\text{et})}$ exhibited a much smaller average particle size (34 nm) compared to $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ (203 nm), indicating more effective size confinement in the ethanolic extract-mediated synthesis.

The observed morphological features strongly influence the electrochemical behavior of the nanostructures. The $\text{SrO}_{2(\text{et})}$ sample, characterized by its polyhedral morphology, smooth surfaces, and uniform particle distribution, offers an increased electrochemically active surface area and reduced ion diffusion distances, thereby facilitating enhanced charge storage performance. In contrast, $\text{SrO}_{2(\text{aq})}$ exhibits larger spherical/elongated particles and partial agglomeration, which hinders efficient ion transport and reduces the electrochemically active surface. Decoration with GO further refines the morphology; the average particle size decreases from 223 nm ($\text{SrO}_{2(\text{aq})}$) to 203 nm ($\text{SrCO}_3@\text{GO}_{(\text{aq})}$) and dramatically from 271 nm ($\text{SrO}_{2(\text{et})}$) to 34 nm ($\text{SrCO}_3@\text{GO}_{(\text{et})}$) while simultaneously introducing wrinkled GO sheets that prevent agglomeration and create additional diffusion channels. The highly porous, fluffy, flower-like morphology observed in $\text{SrCO}_3@\text{GO}_{(\text{et})}$, with the smallest particle size of 34 nm, enhances ion accessibility and offers abundant redox-active sites, thereby explaining its superior specific capacitance and rate performance compared to the other materials, as discussed in Sections 3.9 and 3.10. Thus, the SEM-observed morphology, particle size reduction, and GO-induced dispersion correlate directly with the improved electrochemical storage behavior demonstrated in the CV and GCD analyses.

3.7. EDX analysis

EDX analysis (Fig. 6(a)–(e)) was conducted to confirm the presence of Sr, C, and O in the synthesized nanostructures and to compare their elemental ratios between the aqueous and ethanolic synthesis routes. For both $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ NPs, the spectra were dominated by oxygen signals, with atomic percentages of 79% and 75% accompanied by strontium signals of 20% and 24%, respectively (Fig. 6(a) and (b)). In contrast, the spectra of the $\text{SrCO}_3@\text{GO}$ nanocomposites exhibited an additional strong carbon peak, which is attributed to the graphene oxide matrix. Specifically, the $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ sample showed 65.3% carbon, 27% oxygen, and 7.7% strontium, while the $\text{SrCO}_3@\text{GO}_{(\text{et})}$ composite contained 60.8% carbon, 29.9% oxygen, and 9.3% strontium (Fig. 6(c) and (d)). The increased carbon content in these composites further validates the effective incorporation of graphene oxide into the SrCO_3 matrix, while the detectable strontium and oxygen peaks confirm the coexistence of the carbonate phase. Graphene oxide (GO) demonstrated the presence of only carbon (64.4%) and oxygen (34.6%), with negligible amounts of oxygen (0.5%) and sulfur (0.5%) (Fig. 6(e)). Overall, the EDX results corroborate the successful synthesis and compositional integrity of both the oxide nanoparticles and the graphene oxide-based nanocomposites.

Fig. 7 and 8 display the electron mapping images of $\text{SrO}_{2(\text{et})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$, respectively.

3.8. TGA/DSC analysis

The thermal properties and phase transitions of the nanomaterials (NMs) were investigated using thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC), and the obtained TGA/DSC curves are shown in Fig. 9(a)–(e).



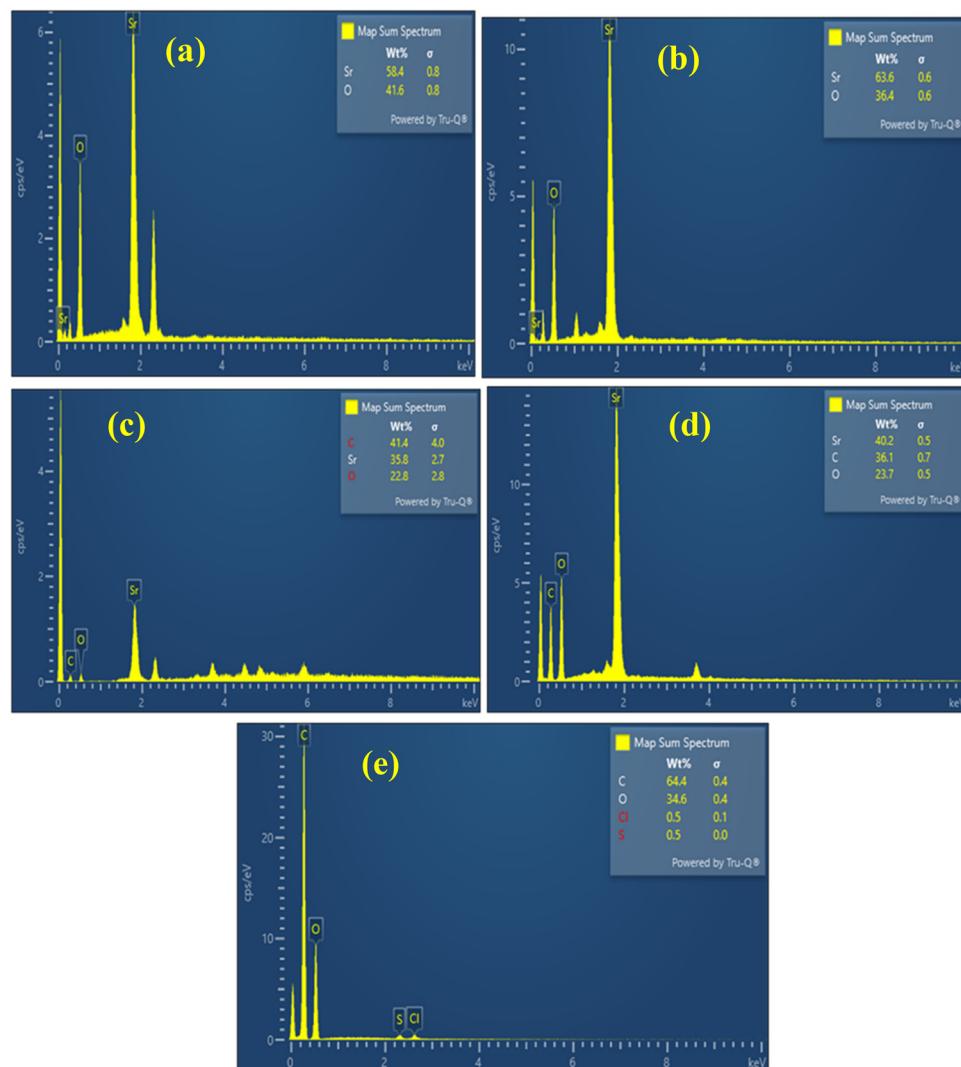


Fig. 6 SEM images of $\text{SrO}_{2(\text{aq})}$ (a), $\text{SrO}_{2(\text{et})}$ (b), $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ (c), $\text{SrCO}_3@\text{GO}_{(\text{et})}$ (d) and GO (e).

The TGA curves of the synthesized materials exhibited multi-stage decomposition, with total mass losses of 14.51, 45.26, 63.10, 61.48 and 99.64% for $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$, GO, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$, respectively, up to 1000 °C (Fig. 9). In $\text{SrO}_{2(\text{aq})}$ (Fig. 9(a)), the initial mass loss of 4.10% between 19.05 °C and 107.42 °C corresponds to an endothermic loss of adsorbed water molecules and surface OH groups. A subsequent gradual endothermic loss of 3.94% occurred between 107.42 °C and 350.26 °C, which was attributed to the removal of interstitial water and/or residual organic moieties, leaving 91.96% of the material. Beyond 350.26 °C, an exothermic decomposition phase was observed, which was associated with the degradation of plant-derived organic compounds adsorbed on nanoparticle surfaces during synthesis. A slow mass loss continued up to 744.65 °C, followed by a rapid degradation phase of up to 781.66 °C, leaving behind the final residual mass of 85.49%. From 781.66 °C to 1000 °C, the $\text{SrO}_{2(\text{aq})}$ material became almost heat resistant with negligible loss of mass, demonstrating the formation of stable metal oxide.⁴⁴

The thermal decomposition pattern of $\text{SrO}_{2(\text{et})}$ (Fig. 9(b)) closely resembles that of $\text{SrO}_{2(\text{aq})}$, exhibiting endothermic decomposition below 423.74 °C, followed by exothermic decomposition at higher temperatures. A gradual mass loss of approximately 11.01% was observed up to 661.63 °C primarily due to the removal of moisture and organic residues. Beyond this temperature, a rapid degradation occurred up to 870.01 °C, leaving a residual mass of 54.74%. Overall, $\text{SrO}_{2(\text{et})}$ exhibited a significantly higher total mass loss (45.26%) compared to $\text{SrO}_{2(\text{aq})}$ (14.51%), indicating a greater content of plant-derived organic moieties. This is attributed to the higher concentration of biomolecules in the ethanolic *M. oleifera* extract (compared to its aqueous counterpart), which likely contributes to the enhanced electrochemical properties and superior energy storage performance of $\text{SrO}_{2(\text{et})}$, as discussed in Sections 3.9 and 3.10.

The TGA/DSC curves of graphene oxide (GO) (Fig. 9(c)) exhibit a three-step decomposition pattern. The first stage (24.02–114.96 °C) involves endothermic loss of residual surface-bound water.



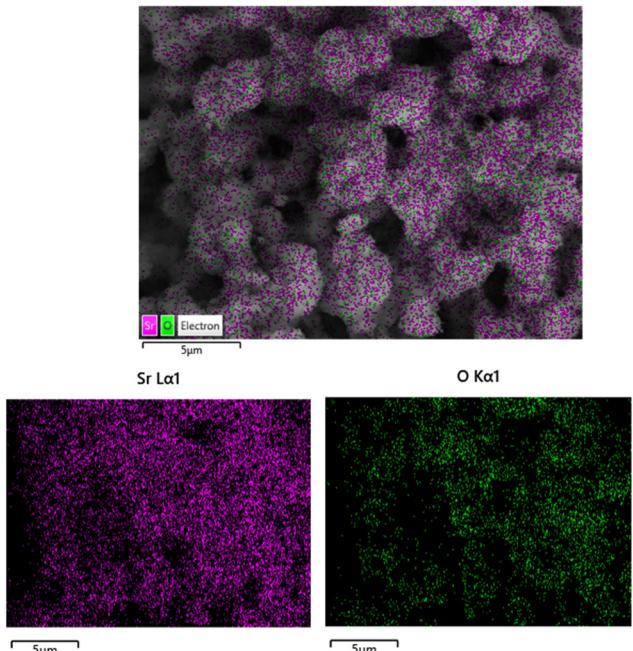


Fig. 7 Electron mapping images of $\text{SrO}_{2(\text{et})}$ NPs confirming the spatial distribution of strontium (Sr) and oxygen (O).

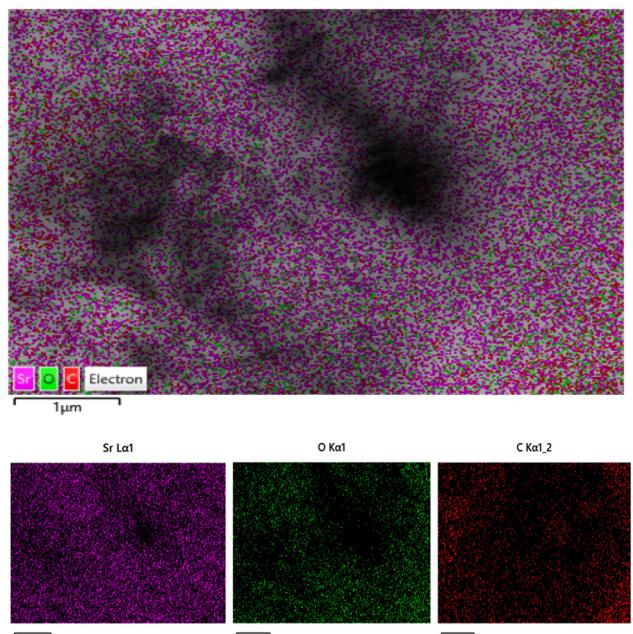


Fig. 8 Electron mapping images of $\text{SrCO}_3@\text{GO}_{(\text{et})}$ NC confirming the spatial distribution of strontium (Sr), oxygen (O) and carbon (C) atoms.

The second stage (196–246 °C) corresponds to the rapid exothermic decomposition of oxygen-containing functional groups (carboxylic, hydroxyl, and epoxide), releasing gaseous byproducts such as H_2O and CO_2 . The final stage (533.64–813.73 °C) reflects the gradual oxidative pyrolysis of the carbon skeleton,⁴⁵ leaving a 26.90% residue.

A comparison of the TGA/DSC curves of the *M. oleifera*-mediated NMs (Fig. 9(a), (b), (d) and (e)) indicates a significant increase in percentage weight loss upon heating when $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ are converted to their GO-based nanocomposites, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$. After heating to 1000 °C, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ showed markedly higher total weight losses of 61.48 and 94.66%, respectively (Fig. 9(d) and (e)), compared to their $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ precursors, which lost 14.51 and 45.26%, respectively (Fig. 9(a) and (b)). These findings suggest that GO incorporation into SrO_2 accelerates the thermal decomposition of NCs, reducing their thermal stability compared to pristine SrO_2 . In $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, a significant weight loss between 191.66 °C and 266.23 °C is attributed to the removal of oxygen-containing functional groups from the GO sheets, while the subsequent losses from 266.23 °C to 867.52 °C correspond to the thermal decomposition of SrCO_3 and the release of volatile species, including CO_2 , leaving behind 38.52% final residue. The TGA/DSC results of $\text{SrCO}_3@\text{GO}_{(\text{et})}$ are particularly striking, as the material undergoes almost complete decomposition (99.64%) up to ≈ 850 °C. It exhibited a major weight loss of 63.33% between 175.90 °C and 627.83 °C, attributable to the thermal decomposition of SrCO_3 and the removal of GO components. The TGA/DSC data thus strongly indicate the stable incorporation of plant-derived organic moieties and water within all *M. oleifera*-mediated NMs, which are gradually released as volatile components upon heating to elevated temperatures. The presence of water and organic moieties in the investigated nanomaterials aligns with the FTIR results discussed in Section 3.4, and their release during thermal degradation is further supported by previous studies.⁴⁴

The DSC-derived enthalpy data provide valuable insights into the thermal behavior of the synthesized NMs. The enthalpies (normalized) of the *M. oleifera*-mediated NPs decreased in the following order: $\text{SrCO}_3@\text{GO}_{(\text{et})}$ (31 511 J g⁻¹) > $\text{SrO}_{2(\text{aq})}$ (6443.0 J g⁻¹) > $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ (5667.4 J g⁻¹) > $\text{SrO}_{2(\text{et})}$ (3827.4 J g⁻¹) > GO (2899.7 J g⁻¹). Among all the samples, $\text{SrCO}_3@\text{GO}_{(\text{et})}$ exhibited the highest normalized enthalpy (31 511 J g⁻¹), indicating its intense thermal activity and complex decomposition behavior, likely arising from the synergistic decomposition of both SrCO_3 and GO components. In contrast, GO alone displayed the lowest enthalpy (2899.7 J g⁻¹), consistent with its relatively simple thermal degradation profile. The TGA/DSC results provide useful insight into electrochemical behavior. The initial weight losses below 150 °C correspond to moisture and occur well above the device operating window (< 80 °C), confirming operational stability. The higher organic content in $\text{SrO}_{2(\text{et})}$ explains its greater mass loss but also provides additional surface functionalities that enhance pseudocapacitance. GO decomposition around 200–250 °C reflects the removal of oxygen groups that, during cycling, can act as redox-active or degradable sites. The nearly complete decomposition of $\text{SrCO}_3@\text{GO}_{(\text{et})}$ indicates its lower ultimate thermal stability and suggests enhanced surface accessibility and electronic coupling, which is consistent with its higher capacitance and cycling stability. Thus, the thermal data



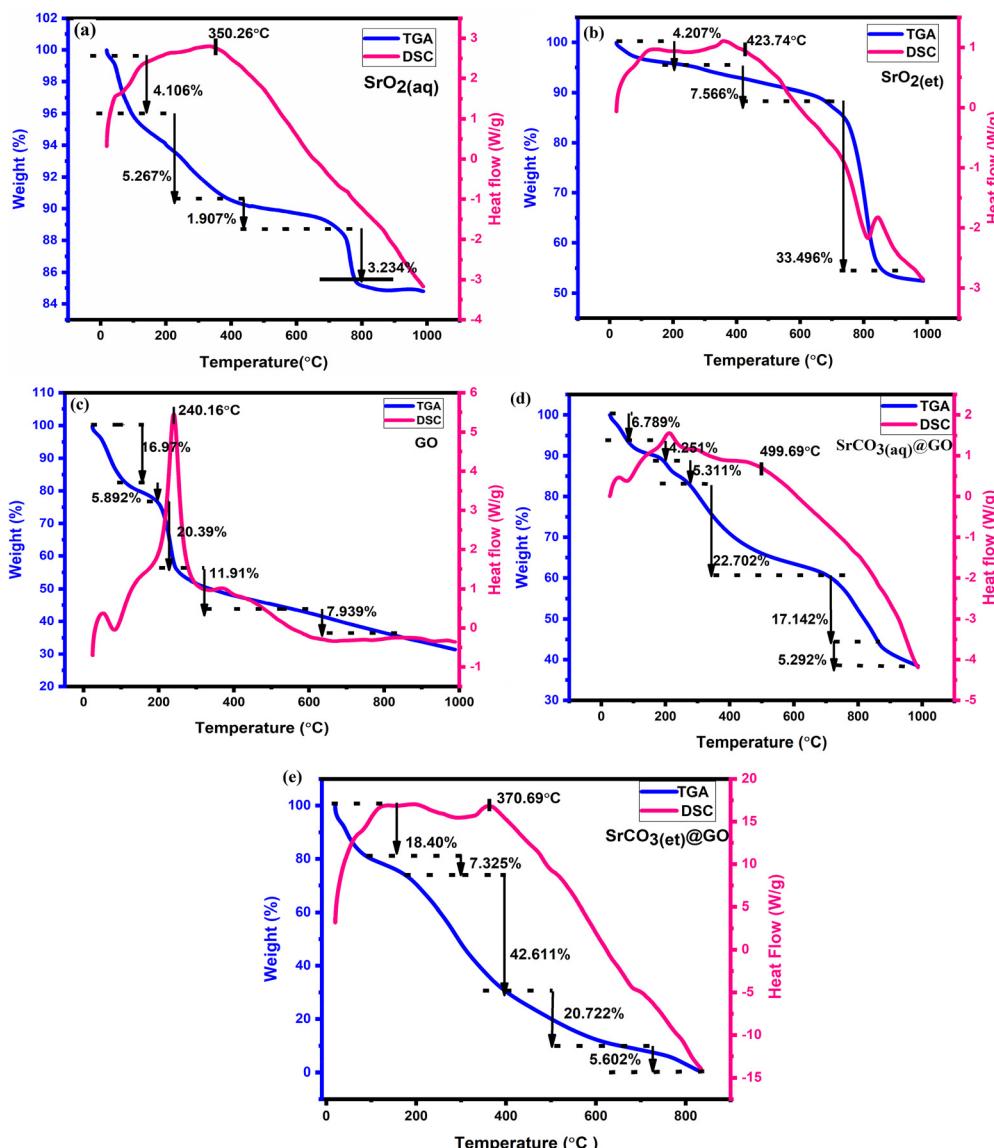


Fig. 9 (a) TGA/DSC curves of $\text{SrO}_2\text{(aq)}$. (b) TGA/DSC curves of $\text{SrO}_2\text{(et)}$. (c) TGA/DSC curves of GO. (d) TGA/DSC curves of $\text{SrCO}_3\text{@GO}_{\text{(aq)}}$. (e) TGA/DSC curves of $\text{SrCO}_3\text{@GO}_{\text{(et)}}$.

clarify how structural robustness and labile functional groups together influence charge storage and long-term electrochemical performance.

3.9. Cyclic voltammetry (CV)

Cyclic voltammetry (CV) plays a crucial role in examining the ion-electron transfer mechanisms of the prepared electrode materials. It can be used to evaluate the energy density, specific capacitance, and cycling stability of the electrode materials. Fig. 10 presents the CV loops of the prepared materials deposited on nickel foam as the working electrode, with scan rates ranging from 0 to 0.45 V (vs. Ag/AgCl). The CV profiles showed that as the scan rate increased, the enclosed current response also increased, indicating a direct proportionality between the voltammetric current and scan rate.⁴⁶

The CV curves exhibit a semi-rectangular shape, confirming the pseudocapacitive behavior of the synthesized electrode materials.

The presence of distinct redox peaks in all CV curves indicates faradaic reactions, which characterize the oxidation-reduction interactions occurring at the electrode–electrolyte interface. Furthermore, as shown in Fig. 10, the faradaic current peaks become more pronounced with increasing scan rates, while the potential difference between the cathodic and anodic peaks also widens. This phenomenon arises due to the dynamic reactions governing current dissipation. For the synthesized materials, the CV loop area increased in the following order: $\text{SrO}_2\text{(aq)} < \text{SrO}_2\text{(et)} < \text{SrCO}_3\text{@GO}_{\text{(aq)}} < \text{SrCO}_3\text{@GO}_{\text{(et)}} < \text{GO}$.

The GO-decorated nanocomposite material demonstrated a larger CV loop area compared to pristine SrO_2 , indicating enhanced reversibility, superior rate capability, and improved power delivery, making it efficient for electrochemical applications. The measured specific capacitance (C_{sp}) values for the electrode materials are displayed in Table 1.



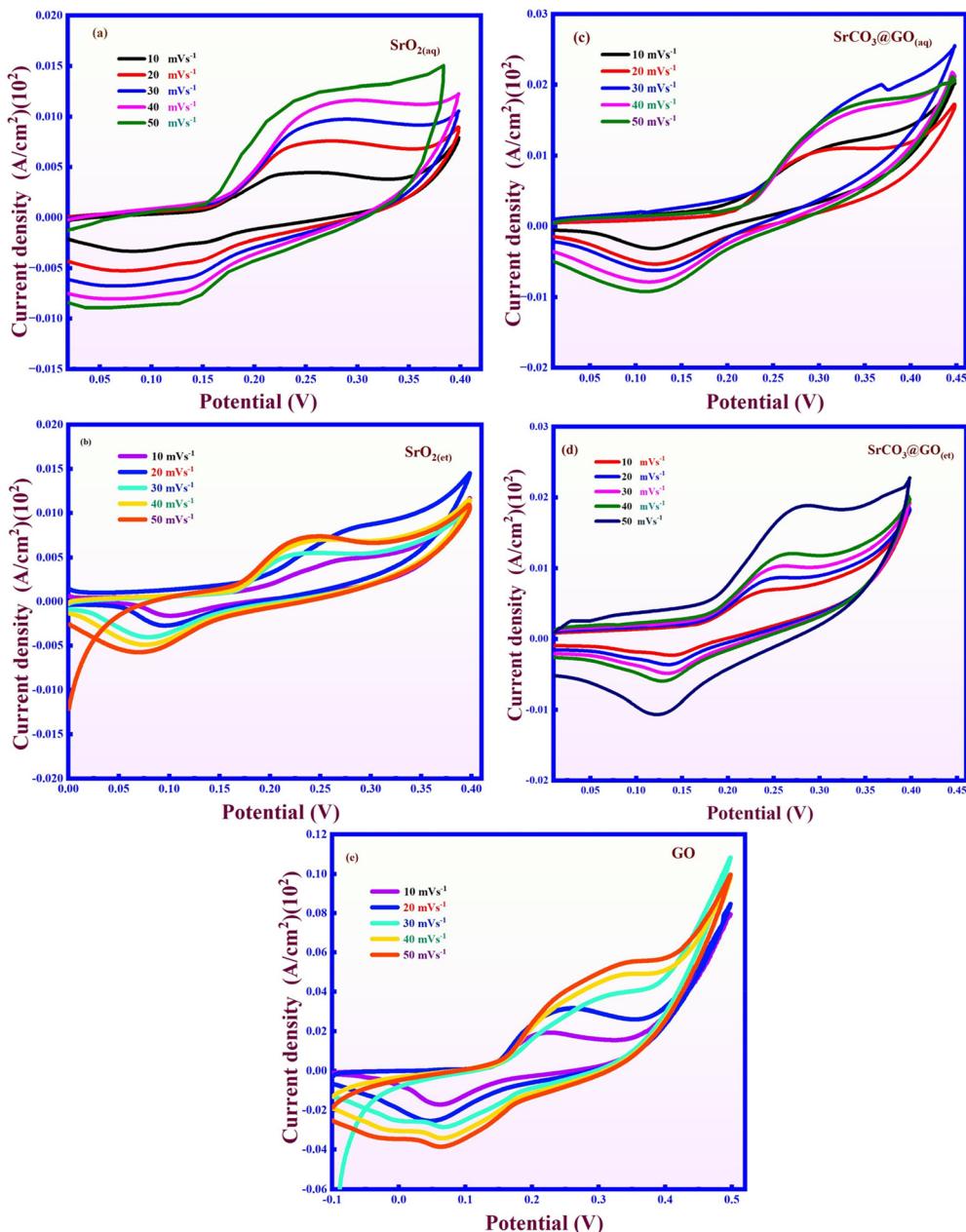


Fig. 10 Cyclic voltammetry spectra of (a) $\text{SrO}_2(\text{aq})$, (b) $\text{SrO}_2(\text{et})$, (c) $\text{SrCO}_3@\text{GO}(\text{aq})$, (d) $\text{SrCO}_3@\text{GO}(\text{et})$, and (e) GO.

Table 1 Specific capacitance of $\text{SrO}_2(\text{aq})$, $\text{SrO}_2(\text{et})$, $\text{SrCO}_3@\text{GO}(\text{aq})$, $\text{SrCO}_3@\text{GO}(\text{et})$ and GO by CV

Current density	10 mVs ⁻¹	20 mVs ⁻¹	30 mVs ⁻¹	40 mVs ⁻¹	50 mVs ⁻¹
Specific capacitance (F g^{-1})					
$\text{SrO}_2(\text{aq})$	63.64	59.46	35.26	34.2	26.08
$\text{SrO}_2(\text{et})$	106.3	95.14	81.28	72.06	65.14
$\text{SrCO}_3@\text{GO}(\text{aq})$	110.88	81.5	73.58	63.58	56.56
$\text{SrCO}_3@\text{GO}(\text{et})$	157.82	99.96	89.28	68.28	64.02
GO	307.04	3271	187.6	165.8	143.9

Table 1 illustrates the relationship between scan rate and C_{sp} , showing that the specific capacitance is enhanced at lower scan rates and decreases at higher scan rates. At low scan rates,

the C_{sp} value is maximized due to the sufficient time available for electrolyte ions to undergo complete redox reactions and diffuse into the electrode material. The increased electrochemical activity of the nanocomposite electrode can be due to the improved adsorption of charged particles on its surface and the intercalation/de-intercalation of OH^- ions within the $\text{SrCO}_3@\text{GO}(\text{et})$ composite. At a higher scan rate, the C_{sp} value decreases due to limited active sites and an insufficient amount of time for electrolyte ions to adsorb and desorb on the electrode surface, which limits the redox process. $\text{SrCO}_3@\text{GO}(\text{et})$ exhibited the highest specific capacitance, as strong van der Waals interactions within the composite significantly enhance its intrinsic electrical conductivity, facilitating rapid electron

transport during electrochemical processes. The unique morphology of $\text{SrCO}_3@\text{GO}_{(\text{et})}$ (as discussed in Section 3.6) offers a high density of active sites, facilitating both capacitive charge storage and ion diffusion. Additionally, the larger enclosed area under the CV curves confirms the dominant charge storage and diffusion-driven mechanisms.

3.10. Galvanic charge–discharge (GCD)

Galvanostatic charge–discharge (GCD) measurements were performed to evaluate the cyclic charge storage capability, energy density, and specific capacitance of the prepared electrodes. Fig. 11 illustrates the CD curves of the $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, $\text{SrCO}_3@\text{GO}_{(\text{et})}$ and GO electrodes at various current densities (1, 2, 3, 4, and 5 mA, respectively) within a voltage window of 0–0.45 V. Fig. 12 displays a graphical

comparison between the specific capacitances of all the five investigated nanomaterials. It is well established that GCD curves typically exhibit three distinct regions: (i) an initial fast voltage drop due to internal resistance, (ii) a linear potential-time relationship corresponding to a fast faradaic redox reaction, and (iii) a sloped discharge region indicating pseudocapacitive behavior, which characterizes the electrochemical interactions at the electrode–electrolyte interface. At higher current densities, shorter interaction times during redox reactions result in a decrease in specific capacitance, as reflected in the GCD measurements. GCD profiles also revealed that discharge time was extended at lower current densities, which is attributed to the improved penetration of electrolyte ions into the pores of electrode materials under reduced current conditions.⁴⁷

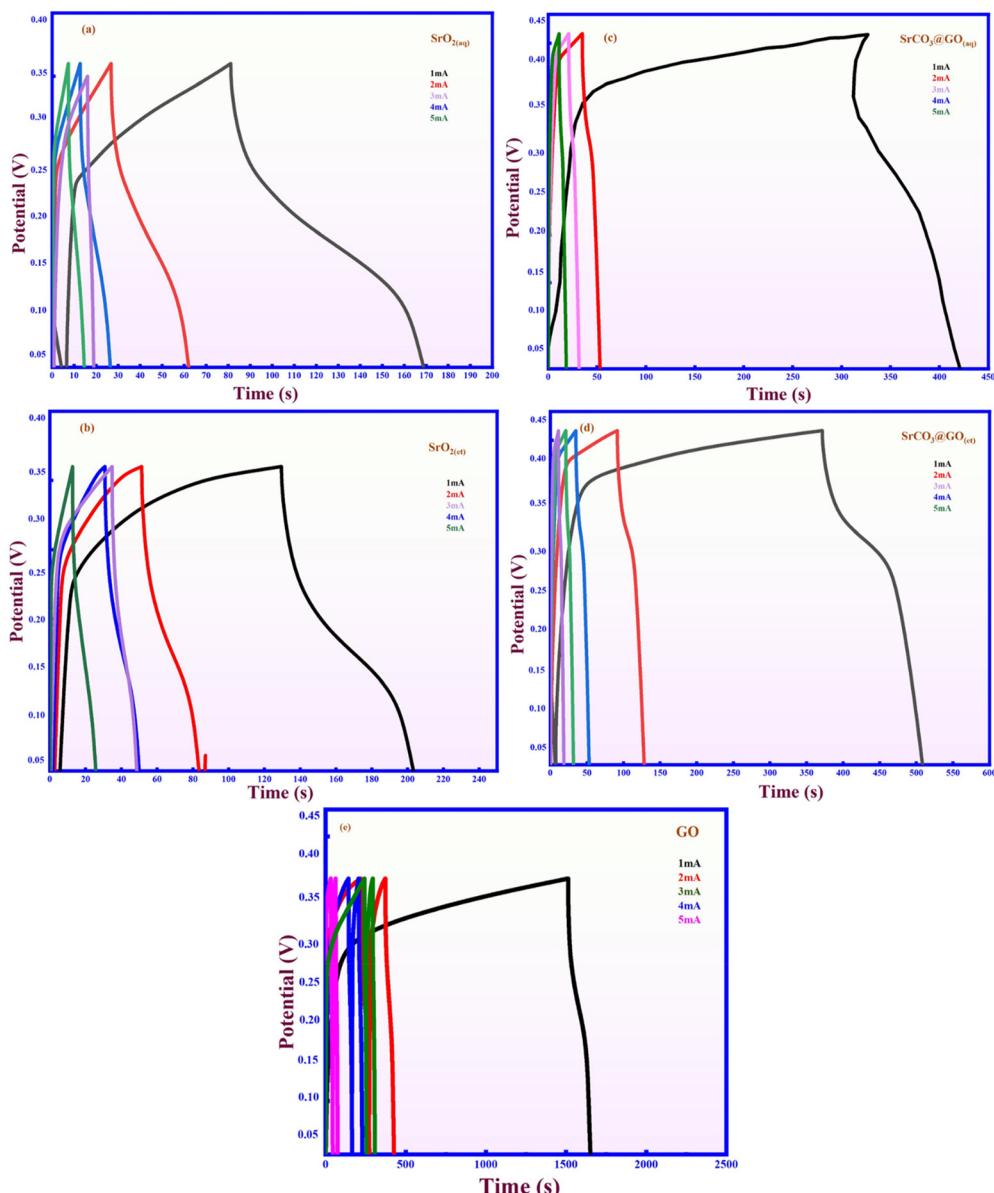


Fig. 11 Galvanostatic charging and discharging (GCD) behavior of (a) $\text{SrO}_{2(\text{aq})}$, (b) $\text{SrO}_{2(\text{et})}$, (c) $\text{SrCO}_3@\text{GO}_{(\text{aq})}$, (d) $\text{SrCO}_3@\text{GO}_{(\text{et})}$, and (e) GO.



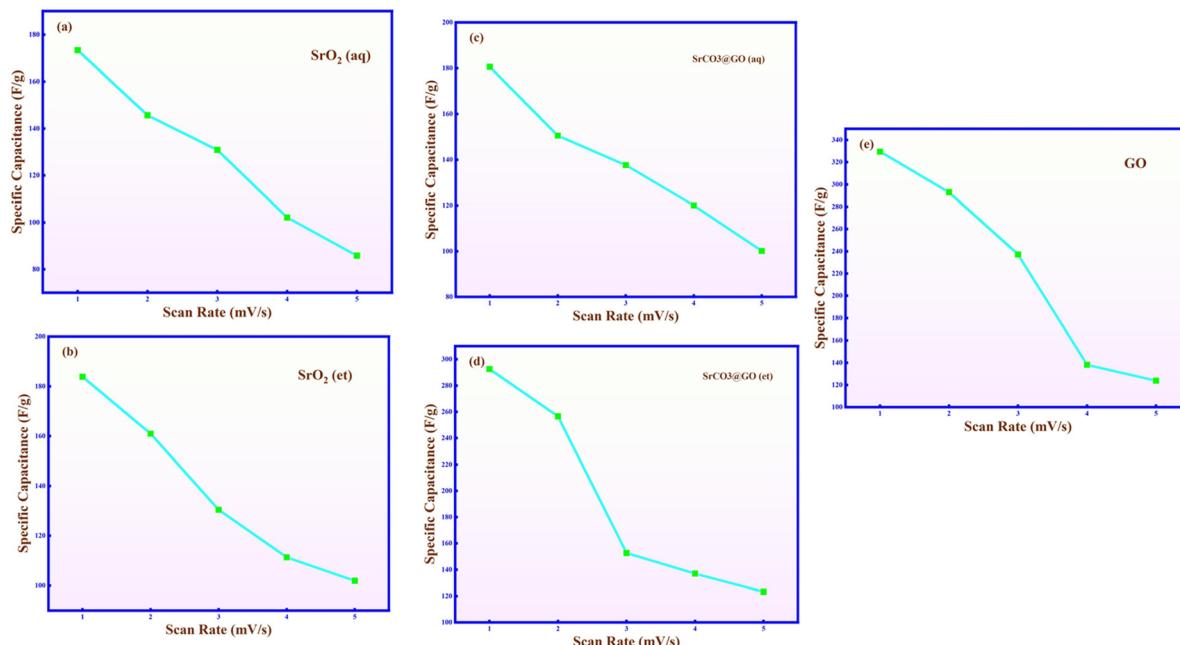


Fig. 12 Specific capacitance of (a) SrO_2 _(aq), (b) SrO_2 _(et), (c) SrCO_3 @GO_(aq), (d) SrCO_3 @GO_(et), and (e) GO by GCD.

Table 2 C_{sp} , E.D. and P.D. of SrO_2 _(aq), SrO_2 _(et), SrCO_3 @GO_(aq), SrCO_3 @GO_(et) and GO

Current density (mA)	C_{sp} (F g ⁻¹)	E.D. (Wh kg ⁻¹)	P.D. (kW kg ⁻¹)
SrO_2 _(aq)			
1 mA	173.33	3.12	11.23
2 mA	145.67	2.62	9.44
3 mA	130.86	2.36	8.48
4 mA	102.10	1.84	6.62
5 mA	85.81	1.54	5.56
SrO_2 _(et)			
1 mA	183.81	3.31	11.91
2 mA	160.95	2.90	10.43
3 mA	137.43	2.35	8.45
4 mA	111.33	2.00	7.21
5 mA	101.90	1.83	6.60
SrCO_3 @GO _(aq)			
1 mA	180.59	3.25	11.70
2 mA	150.52	2.71	9.76
3 mA	137.63	2.48	8.93
4 mA	120.04	2.16	7.78
5 mA	100.15	1.80	6.48
SrCO_3 @GO _(et)			
1 mA	292.59	5.26	18.94
2 mA	268.52	3.03	10.91
3 mA	152.63	2.75	9.90
4 mA	137.04	2.47	8.89
5 mA	123.15	2.22	7.99
GO			
1 mA	329.43	5.93	21.35
2 mA	292.95	5.27	18.98
3 mA	237.14	4.27	15.37
4 mA	138.10	2.49	8.95
5 mA	123.81	2.23	8.02

Table 2 displays the specific capacitance (C_{sp}), power density (P.D.), and energy density (E.D.) of the *M. oleifera*-mediated NMs. The observed correlation between energy density and specific capacitance suggests that the enhanced

electrochemical performance of the nanocomposites arises from rapid faradaic redox processes at the electrode–electrolyte interface. It was found that the specific capacitance (C_{sp} , F g⁻¹) of the synthesized materials, measured at current densities of 1, 2, 3, 4, and 5 mA, decreased in the following order (Table 2): SrCO_3 @GO_(et) > SrO_2 _(et) > SrCO_3 @GO_(aq) > SrO_2 _(aq). The obtained results thus confirm that the C_{sp} of both GO-decorated SrCO_3 @GO_(et) and SrCO_3 @GO_(aq) is higher than that of their respective SrO_2 _(et) and SrO_2 _(aq) counterparts. Moreover, the ethanolic extract-based SrCO_3 @GO_(et) and SrO_2 _(et) showed higher performance than their aqueous extract-based SrCO_3 @GO_(aq) and SrO_2 _(aq) counterparts, respectively. The highest specific capacitance (292.59 F g⁻¹) was displayed by the GO-based SrCO_3 @GO_(et) composite, highlighting its potential as a high-performance supercapacitor material compared with the other synthesized samples. Notably, this value ($C_{\text{sp}} = 292.59$ F g⁻¹) is significantly higher than those previously reported for related Sr- and GO-based materials ($C_{\text{sp}} = 62.4$ –276.1 F g⁻¹), as summarized in Table 3.

To further evaluate the practical energy storage capability, the energy and power densities of all electrode materials were calculated from the GCD data and represented in Ragone plots (Fig. 13). The SrCO_3 @GO_(et) nanocomposite shows the best balance, delivering higher energy density at a given power density compared to the remaining *M. oleifera*-mediated nanomaterials. In contrast, SrO_2 _(aq) and SrO_2 _(et) electrodes exhibited lower energy densities, which are consistent with their lower C_{sp} values. These results confirm that GO decoration and optimized morphology significantly enhance the energy–power characteristics of the electrodes, making them competitive with other reported supercapacitor materials.



Table 3 Comparison of specific capacitance (F g^{-1}) of the *M. oleifera*-mediated $\text{SrCO}_3@\text{GO}_{(\text{et})}$ nanocomposite with reported Sr- and GO-based materials

Electrode material	Synthetic method	Morphology	Electrolyte	C_{sp} (F g^{-1})	Ref.
SrO -polyaniline	Blending	Nanorods	1 M KOH	258	48
SrRuO_3 -rGO	Mixing	SrRuO_3 particles covered with rGO sheets	1 M KOH 1 M NaNO ₃ 1 M H ₃ PO ₄	160 62.4 101	49
SrFeO_3 perovskite	Sol-gel method	Rough and highly porous	1 M KOH	101.69	50
SrCoO_3	Sol-gel method	Porous	1 M KOH	68.639	50
$\text{Sr}_2\text{Ni}_2\text{O}_5@\text{15\%rGO}$	Solvothermal	Spherical	2 M KOH	148.09	51
rGO/SrO	Hydrothermal	Nanorods	6 M KOH	271	52
rGO@ Co_3O_4 /CoO	Solvothermal	$\text{Co}_3\text{O}_4/\text{CoO}$ NPs uniformly distributed inside rGO nanosheets	0.1 M KOH	276.1	53
La _{0.85} Sr _{0.15} MnO ₃	Sol-gel	Nano sphere	3 M KOH	248	54
SrTiO_3	Microwave	Nano cube	1 M KCl	208.47	55
$\text{SrO-RuO}_2/\text{PVA}$	Hydrothermal	Nanofibers	1 M KOH	192	56
$\text{SrCO}_3@\text{GO}_{(\text{et})}$	<i>M. oleifera</i> -mediated	Porous, fluffy, and flower-like	6 M KOH	292.59	This work

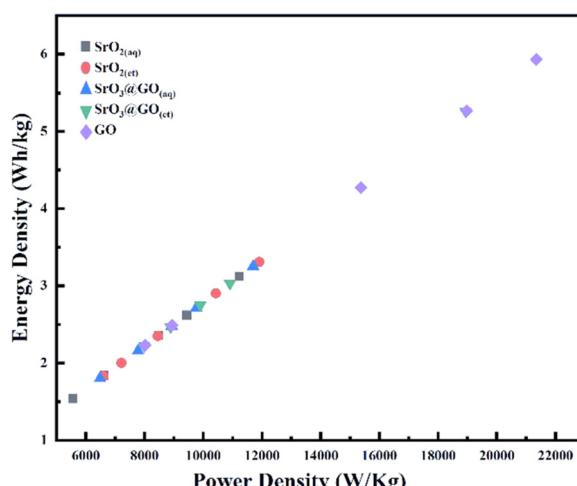


Fig. 13 Ragone plot of *M. oleifera*-mediated materials.

4. Conclusions

We successfully synthesized $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ NPs and their GO-decorated $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ NCs using aqueous and ethanolic extracts of *Moringa oleifera* leaves as reducing, capping and stabilizing agents. The XRD analysis confirmed well-defined crystalline structures, with GO incorporation reducing crystallite sizes. FTIR analysis confirmed Sr–O vibrations and the presence of GO and phytochemical-derived organic moieties capping the NPs. SEM analysis revealed that $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ exhibit distinct morphologies influenced by solvent systems, with $\text{SrO}_{2(\text{et})}$ forming well-defined polyhedral nanoparticles and $\text{SrO}_{2(\text{aq})}$ showing larger, partially aggregated particles. GO decoration significantly reduced particle sizes and introduced wrinkled, porous structures in $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$, with $\text{SrCO}_3@\text{GO}_{(\text{et})}$ displaying a highly porous, flower-like morphology. EDX analysis verified the presence of Sr, O and C atoms in the NMs. The average crystallite and particle sizes were decreased when moving from $\text{SrO}_{2(\text{aq})}$ and $\text{SrO}_{2(\text{et})}$ to their respective $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$ NCs. UV-Vis analysis confirmed the

successful integration of SrO_2 NPs with GO, with $\text{SrCO}_3@\text{GO}_{(\text{et})}$ showing the lowest band gap (4.65 eV). TGA-DSC analysis (up to 1000 °C) revealed multi-stage decomposition with total mass losses of 14.51, 45.26, 63.10, 61.48 and 99.64% for $\text{SrO}_{2(\text{aq})}$, $\text{SrO}_{2(\text{et})}$, GO, $\text{SrCO}_3@\text{GO}_{(\text{aq})}$ and $\text{SrCO}_3@\text{GO}_{(\text{et})}$, with corresponding enthalpies (normalized) of 6443.0, 3827.4, 2899.7, 5667.4, and 31 511 J g⁻¹, respectively, indicating distinct thermal stability profiles. CV analysis confirmed the pseudocapacitive behavior of all materials, with $\text{SrCO}_3@\text{GO}_{(\text{et})}$ exhibiting the largest CV loop area and highest specific capacitance, owing to its enhanced electrical conductivity and abundant active sites. GCD measurements further demonstrated that $\text{SrCO}_3@\text{GO}_{(\text{et})}$ delivered the highest specific capacitance (292.59 F g⁻¹) and exceptional energy-power characteristics among all samples. The synergistic effects of *M. oleifera* ethanolic extract-derived phytochemicals, effective GO decoration, reduced crystallite and particle sizes, and the formation of a porous, flower-like nanostructure underpin the outstanding electrochemical performance of $\text{SrCO}_3@\text{GO}_{(\text{et})}$, establishing it as a highly promising electrode material for advanced energy storage applications. Nevertheless, Sr-based phases may undergo surface reactions in alkaline electrolytes, and device-level translation requires further optimization of parameters such as mass loading, binders, and current collectors. Although full device integration was not performed, the stable CV/GCD behavior, robust TGA/DSC profiles, and GO-stabilized morphology suggest good electrochemical resilience under test conditions. Future studies should address device engineering and targeted stability studies to confirm long-term performance. This study can be extended to the evaluation of *M. oleifera*-mediated Sr-based NMs for other energy storage devices, solar cells and testing of their electrocatalytic water splitting potential.

Conflicts of interest

There are no conflicts to declare.

Data availability

All the data will be provided upon request.



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