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Hydrophobically Modified Poly(acrylic acid) Coatings on Silicone Hydrogel Contact Lens

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ABSTRACT

Silicone hydrogel (Si-Hy) contact lenses offer high oxygen permeability and mechanical durability but often exhibit poor surface wettability due to the intrinsic hydrophobicity of silicone components. We investigated hydrophobically modified poly(acrylic acid) (PAA)-based amphiphilic copolymers as coating materials for Si-Hy lenses to effectively cover the chemically heterogeneous surface. A series of statistical copolymers was synthesized by copolymerizing acrylic acid with alkyl acrylates bearing linear alkyl side chains with a varying number of carbon atoms from 6 to 12. By varying molecular weight, hydrophobic content, and alkyl chain length, we examined how hydrophobic interactions influence solution behavior, interfacial properties, and coating performance. Increasing hydrophobic content enhanced anchoring to silicone domains, whereas excessive hydrophobicity promoted intermolecular association of alkyl chains. Among the copolymers studied, the poly(acrylic acid-*co*-octyl acrylate) copolymer exhibited the lowest contact angle and friction coefficient, indicating improved wettability and lubrication. These results demonstrate that amphiphilic coatings with optimized hydrophobic contents and alkyl chain lengths can effectively improve the interfacial properties of Si-Hy lenses.



INTRODUCTION

Silicone-based hydrogel (Si-Hy) contact lenses offer high oxygen permeability and enhanced mechanical durability compared to hydrophilic-only soft hydrogel lenses.^{1,2,3} Incorporation of silicone materials, such as polydimethylsiloxane (PDMS), allows more oxygen to pass through the lens and can help reduce the risk of corneal hypoxia during lens wear.^{4,5,6,7,8} In commercial Si-Hy lenses, silicone polymers are typically combined with hydrophilic polymers, such as poly(2-hydroxyethyl methacrylate) and poly(*N*-vinyl pyrrolidone), within an interpenetrating polymer network (IPN).^{9,10,11} The intertwined co-network maintains dimensional stability and optical transparency by preventing phase separation between the drastically different silicone and hydrophilic polymers. However, the inherently hydrophobic nature of silicone chains reduces surface wettability and repels moisture from the lens surface, compromising wearer comfort, particularly for those with dry-eye symptoms.^{12,13}

A coating on a contact lens primarily functions as an interface-modifying layer that tailors interactions with the surrounding tear film. To be effective, such a coating must simultaneously provide strong affinity to the lens surface, excellent lubricity, long-term comfort, and non-irritating biocompatibility—all without altering the bulk optical or mechanical properties of the lens.¹⁴ Surface-active molecules have been incorporated during polymerization or applied post-polymerization as coating additives to enhance lubricity and wettability.^{15,16} Hydrophilic polymers such as poly(ethylene glycol), poly(acrylic acid), and poly(2-methacryloyloxyethyl phosphorylcholine) have been employed for this purpose.^{14,17,18} However, coatings composed of a single hydrophilic component often exhibit limited compatibility with the chemically heterogeneous Si-Hy lens surface, which contains both hydrophilic and hydrophobic domains



originating from hydrogel and silicone segments.

Recent developments in amphiphilic random copolymers highlight their unique ability to interact adaptively with complex interfaces and motivate their application to Si-Hy lens coatings. While each copolymer chain can have a drastically different sequence of hydrophobic and hydrophilic repeating units, such heteropolymers have been shown to recognize and interact with chemically heterogeneous environments, preserving protein function and enabling selective transport through lipid bilayers.^{19,20} Despite the broad sequence distribution, amphiphilic random copolymers have been reported to form micelles in aqueous media and lyotropic mesophases at higher concentrations via association between hydrophobic pendants, such as alkyl side chains.^{21,22,23,24} While the average copolymer composition primarily determines the interfacial curvature of the assemblies, similar to a diblock copolymer analog, the length scale of the assemblies follows that of the side chains, leading to much broader interface coverage per polymer.^{25,26} In addition, the higher modulus of the lyotropic mesophase with the increasing degree of polymerization suggests that the association between copolymers becomes stronger collectively by involving more side chains. Together, these studies indicate that amphiphilic copolymers bearing hydrophobic pendants can stabilize adsorption onto hydrophobic surface domains while maintaining overall aqueous compatibility—an interfacial requirement that aligns precisely with the heterogeneous architecture of Si-Hy lenses.

In this work, we investigated hydrophobically modified poly(acrylic acid)s containing statistically distributed acrylic acid and alkyl acrylate repeating units as the coating material for Si-Hy lenses (**Figure 1a**). Hydrophobically modified poly(acrylic acid)s are commercially available materials used as viscosity modifiers to thicken aqueous solutions, emulsifiers, and drug carriers.^{27,28,29,30,31} We synthesized a series of amphiphilic statistical copolymers by



copolymerizing acrylic acid (AA) with an alkyl acrylate (xA) monomer to produce P(AA-*co*- xA), in which the number of carbon atoms in the alkyl side chain (C_n) varies from 6 (hexyl) to 12 (dodecyl). We found that the viscosity and surface tension of their aqueous solutions, as well as the contact angles of the coated surfaces, were significantly affected by the copolymer's molar composition and alkyl chain length. While surface energy generally decreased with increasing alkyl chain weight fraction, excessive alkyl chain loading led to a rough surface, presumably due to alkyl domain formation. When applied to commercial Si-Hy lenses, both the contact angle and friction coefficient were minimized at intermediate alkyl chain length, suggesting that the heterogeneous lens surface can be effectively covered at an appropriately balanced hydrophobic content (**Figure 1b**). We envision that the amphiphilic copolymer can serve as a versatile platform for multifunctional lens coating materials through further side-chain engineering, incorporating various functionalities.

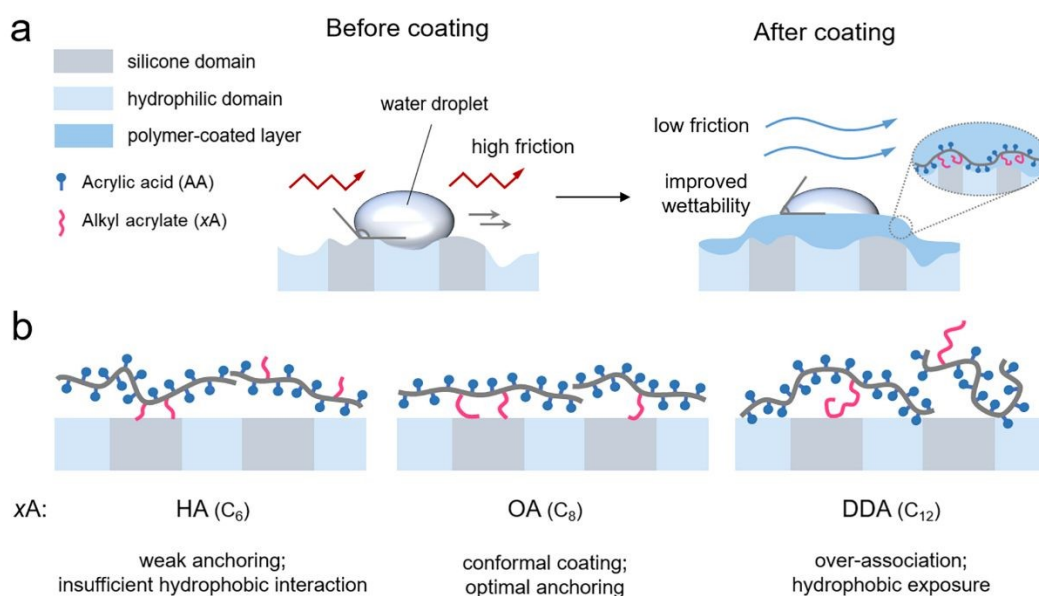


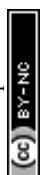
Figure 1. (a) Schematic illustration of the amphiphilic copolymer coating on chemically heterogeneous silicone hydrogel (Si-Hy) lens surface. The surface consists of silicone-rich and



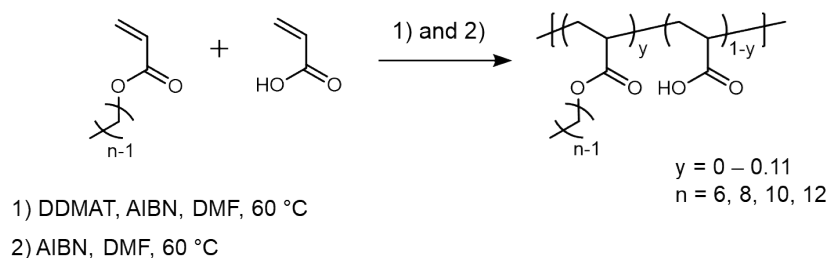
hydrophilic domains. A water droplet is placed on the surface to illustrate differences in wettability before and after coating. The darker blue region represents a hydrated polymer layer formed after coating. (b) Effect of alkyl side-chain length on interfacial anchoring and coating behavior: Hexyl acrylate (HA) exhibits weak anchoring with partially exposed chains, octyl acrylate (OA) achieves balanced and uniform anchoring, and dodecyl acrylate (DDA) shows limited anchoring with increased chain exposure due to increased hydrophobic association.

RESULTS AND DISCUSSION

Copolymer Synthesis. Hydrophobically modified PAAs were synthesized via radical copolymerization of acrylic acid (AA) with alkyl acrylates (**Scheme 1**). To systematically investigate the influence of molecular weight and hydrophobic composition, reversible addition–fragmentation chain-transfer (RAFT) copolymerization of AA and dodecyl acrylate (DDA) was conducted using 2-(dodecylthiocarbonothioylthio)-2-methylpropionic acid (DDMAT) as the chain transfer agent (CTA). This approach enabled precise control of the degree of polymerization ($N = 200–1600$) and the hydrophobic content (0–10 %) (see **Table 1** for the characterization details). RAFT copolymerization kinetics were monitored for a representative feed ratio of $[AA]:[DDA] = 90:10$ (**Figure S1**). Although the monomers exhibited different consumption rates, both showed linear pseudo–first-order kinetics, consistent with a well-controlled RAFT process. To explore the effect of alkyl side-chain length, additional hydrophobically modified PAAs were synthesized via free radical copolymerization of AA with hexyl acrylate (HA), octyl acrylate (OA), and decyl acrylate (DA), while avoiding sulfur-containing CTAs due to concerns regarding residual color and



odor in lens applications (**Table 2**). DA was synthesized following the procedure in the literature³² (**Figure S2**). Because the hydrophobic contribution varies with alkyl length, the feed compositions were selected to maintain a constant hydrophilic–lipophilic balance (HLB) calculated according to Griffin’s method³³ to ensure a constant value. All copolymerization proceeded with monomer conversions above 80%. Composition analysis using ¹H NMR showed that the copolymer composition was almost similar to the initial feed ratio, indicating that the hydrophobic content can be controlled by the feed composition (**Figure S3**). We attempted to analyze the synthesized copolymers using size exclusion chromatography but observed poor filterability and unstable elution behavior, presumably because of their partial solubility in eluents and polymer-column interactions. Detailed information, including feed composition and characterization of polymers, is provided in **Table 2**.



Scheme 1. Synthetic routes for P(AA-*co*-xA) prepared by RAFT (1) and free-radical copolymerization (2)

Table 1. Synthesis and characterization of P(AA-*co*-DDA)s produced by RAFT copolymerization

Sample name	Feed composition [AA]:[DDA]:[CTA]	Conv ^a _{AA} (%)	Conv ^a _{DDA} (%)	$M_{n,theo}^a$ (kg/mol)	$i_{DDA,theo}^b$ (mol %)	$i_{DDA,NMR}^c$ (mol %)
PAA ₁₉₀	[200]:[0]:[1]	95	-	14	0	0



P(AA ₁₈₄ - <i>co</i> -DDA ₉)	[190]:[10]:[1]	97	85	15	4	6
P(AA ₁₇₆ - <i>co</i> -DDA ₁₇)	[180]:[20]:[1]	98	87	17	9	9
PAA ₃₄₄	[400]:[0]:[1]	86	-	25	0	0
P(AA ₃₅₃ - <i>co</i> -DDA ₂₀)	[380]:[20]:[1]	93	100	30	5	5
P(AA ₃₄₆ - <i>co</i> -DDA ₄₀)	[360]:[40]:[1]	96	100	34	11	9
PAA ₅₂₈	[600]:[0]:[1]	88	-	38	0	0
P(AA ₅₃₆ - <i>co</i> -DDA ₃₀)	[570]:[30]:[1]	94	100	48	5	5
P(AA ₅₂₉ - <i>co</i> -DDA ₆₀)	[540]:[60]:[1]	98	100	52	11	11
PAA ₈₀₀	[800]:[0]:[1]	100	-	58	0	0
P(AA ₇₃₀ - <i>co</i> -DDA ₄₀)	[760]:[40]:[1]	96	100	62	6	5
P(AA ₆₄₈ - <i>co</i> -DDA ₈₀)	[720]:[80]:[1]	90	100	66	10	9

^aDetermined by ¹H NMR spectroscopy

^bDetermined by ¹H NMR spectroscopy based on conversion

^cDetermined by ¹H NMR spectroscopy based on the composition of each monomer in polymers

Table 2. Synthesis and characterization of P(AA-*co*-*x*A)s produced by free radical copolymerization

Sample name	Feed composition [AA]:[<i>x</i> A]:[AIBN]	Conv ^a _{AA} (%)	Conv ^a _{<i>x</i>A} (%)	<i>i</i> _{<i>x</i>A,theo} ^b (mol %)	<i>i</i> _{<i>x</i>A,NMR} ^c (mol %)
P(AA ₁₁₆₆ - <i>co</i> -HA ₄₈)	[2880]:[120]:[1]	81	80	4	4
P(AA ₁₃₂₄ - <i>co</i> -OA ₄₂)	[2910]:[90]:[1]	91	93	3	4
P(AA ₁₃₁₆ - <i>co</i> -DA ₃₅)	[2925]:[75]:[1]	90	93	2	2
P(AA ₁₂₂₀ - <i>co</i> -DDA ₃₀)	[2940]:[60]:[1]	83	100	2	2

^aDetermined by ¹H NMR spectroscopy

^bDetermined by ¹H NMR spectroscopy based on conversion

^cDetermined by ¹H NMR spectroscopy based on the composition of each monomer in polymers

Hydrophobic content and molecular weight effect. To identify an optimal polymer design for lens coating, we investigated how the hydrophobic content and molecular weight of P(AA-*co*-DDA) influence the solution behavior and surface characteristics. As shown in **Figure 2a and 2b**, the viscosity of P(AA-*co*-DDA) aqueous solutions tends to increase with both the theoretical molecular weight ($M_{n,theo}$) and the hydrophobic content (i_{DDA}). At both 0.01 and 0.05 wt%, the viscosity exhibited a clear dependence on molecular weight, consistent with the



molecular-weight scaling behavior described by the Mark–Houwink relationship.^{34,35} A sharp increase in viscosity was observed for the copolymer with $i_{\text{DDA}} = 9\%$ and 66 kg/mol at a 0.05 wt% concentration, indicating that substantial intermolecular association involving the alkyl side chains led to associative thickening.

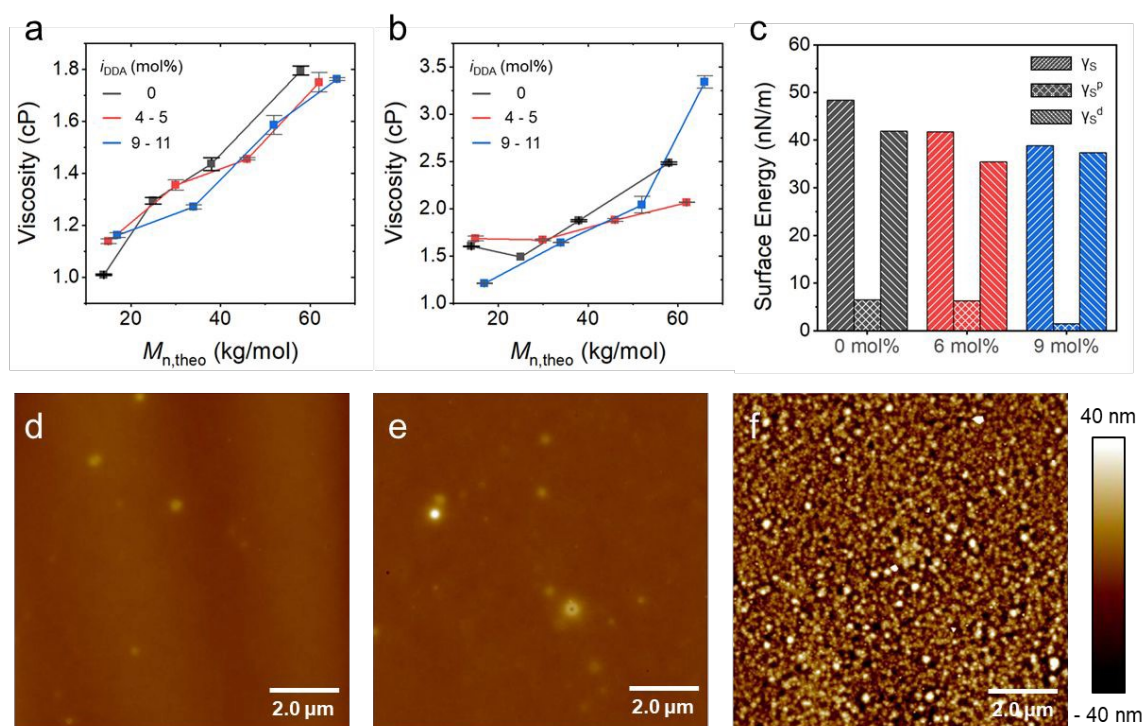
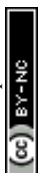


Figure 2. (a) Viscosity of P(AA-*co*-DDA) aqueous solutions at 0.01 wt% and (b) 0.05 wt%. (c) Total surface energy (γ_s) of the coated Si wafers and its polar (γ_s^p) and dispersive (γ_s^d) components calculated using the Owens–Wendt method based on the contact-angle data. (d-f) AFM image of a Si wafer coated with (d) PAA₃₄₄ ($i_{\text{DDA}} = 0\%$), (e) P(AA₃₅₃-*co*-DDA₂₀) ($i_{\text{DDA}} = 6\%$), and (f) P(AA₃₄₆-*co*-DDA₄₀) ($i_{\text{DDA}} = 9\%$).

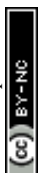
We measured the contact angles of drop-cast copolymer films on silicon wafer substrates as a function of i_{DDA} using diiodomethane (DIM) and glycerol as hydrophobic and hydrophilic probe liquids, respectively (Figure S4). Higher-molecular-weight copolymers generally



exhibited lower contact angles than the lower-molecular-weight set, suggesting improved film uniformity and enhanced substrate wetting. Significantly lower root mean-square (RMS) values for higher-molecular-weight copolymer coatings except for P(AA_{346-co}-DDA₄₀) with the highest $i_{\text{DDA}} = 9$ mol%, determined by atomic force microscopy (AFM) imaging, supported higher film quality (**Figure S5** and **Table S1**). The DIM contact angles were higher than those measured with glycerol, reflecting differences in probe-liquid polarity and solid-liquid interactions. **Figure 2c** shows the calculated surface energy (γ_s) of the higher-molecular-weight-copolymer-coated wafers based on the Owens–Wendt model^{36,37}, which decreases with increasing i_{DDA} over the investigated composition range. A decrease in both the polar (γ_s^{p}) and dispersive (γ_s^{d}) components is consistent with reduced surface polarity upon more population of dodecyl chains on the surface at the cost of acid number reduction. AFM imaging of surfaces coated with higher-molecular-weight samples showed uniform surface coverage by the PAA homopolymer ($i_{\text{DDA}} = 0\%$, **Figure 2d**) and the copolymer with moderate hydrophobic content ($i_{\text{DDA}} = 6\%$, **Figure 2e**), with relatively smooth, continuous textures and RMS roughness of 1.5 nm. A further increase in hydrophobic content to $i_{\text{DDA}} = 9\%$ resulted in a distinct morphology characterized by heterogeneous granular textures (**Figure 2f**). We posit that the formation of hydrophobic domains via alkyl–alkyl association occurs above a critical hydrophobic content. We also note that the opposing trend compared to the intermediate i_{DDA} regime (i.e., increasing γ_s^{p} and decreasing γ_s^{d}) would also be related to the segregation of hydrophobic domains from the uniformly mixed state. As lens coating requires the contact angle not to increase, we decided to investigate copolymers with low hydrophobic content further.



Effect of alkyl length. We then investigated the effect of hydrophobic side-chain length at the air-water interface by evaluating the surface tension of copolymer aqueous solutions. We synthesized a series of P(AA-*co*-*x*A) copolymers with C_n ranging from 6 to 12 by free radical copolymerization, while maintaining a constant hydrophilic–lipophilic balance (HLB) of 11–12, which corresponds to $i_{xA} = 4$ (HA) – 2 (DDA) mol%. At a 0.05 wt% concentration, the surface tension of the copolymer solutions decreased markedly with increasing alkyl length, reflecting stronger adsorption of longer side chains at the air-water interface (**Figure 3a**). Increasing alkyl length also enhanced cohesive interactions among hydrophobic segments, as evidenced by thickening of the copolymer solutions (**Figure 3b**). A non-monotonic dependence on alkyl chain length was observed in solution viscosity at both 0.01 and 0.05 wt%, with a decline at $C_n = 10$ (DA). While we are not certain about the origin of this phenomenon, we posit that the copolymer may undergo a conformational change from a relatively open state to a micelle-like form with segregated alkyl chains within it above a critical hydrophobic content. While relatively fewer alkyl chains may contribute to intermicellar interactions, further increasing alkyl length would reincrease solution viscosity, particularly at higher concentration, due to greater overlap.



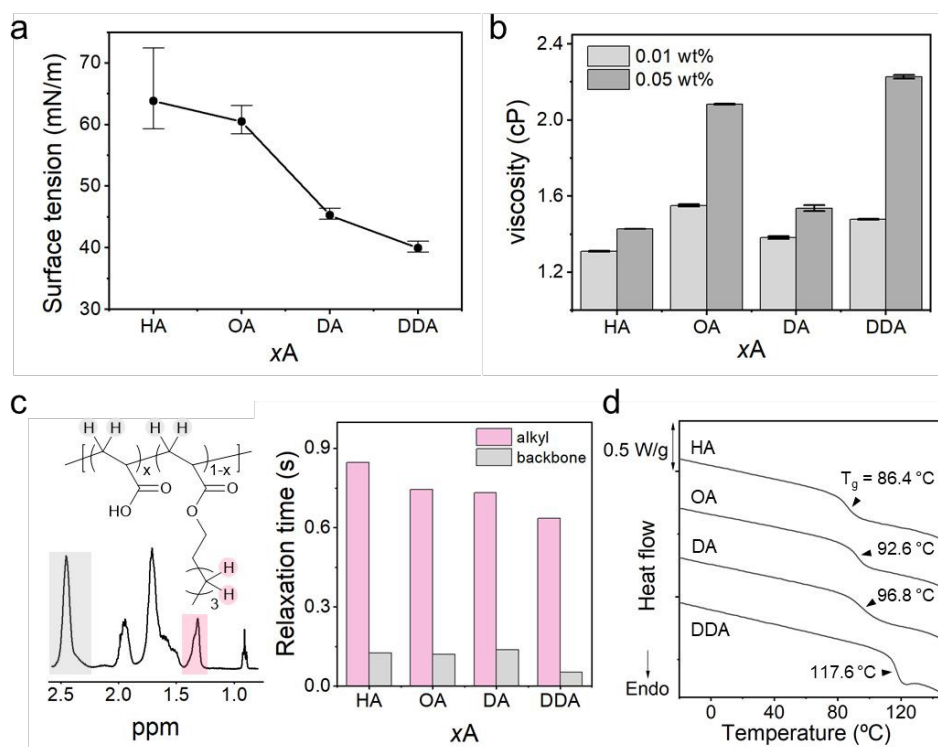


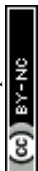
Figure 3. (a) Surface tension of 0.05 wt% P(AA-co-xA) aqueous solutions at the air-water interface. (b) Viscosity of 0.01 and 0.05 wt% P(AA-co-xA) aqueous solutions. (c) T_1 and T_2 relaxation times measured by ^1H NMR in CD_3OD (25 °C). (d) DSC thermograms of P(AA-co-xA) in the second heating cycle (10 °C/min, N_2).

The ^1H NMR transverse relaxation times (T_2) measured in CD_3OD decreased from 0.85 s (HA) to 0.63 s (DDA), confirming slower segmental dynamics of the longer alkyl side-chains because of alkyl chain segregation and intermolecular association (**Figure 3c** and **Table S2**). In contrast, the backbone T_2 values showed no clear variation with alkyl length except for the notable decrease in the DDA case, suggesting that the restricted mobility primarily originates from the alkyl side chains rather than the main chain. In addition, conventional ^1H NMR spectra acquired in D_2O showed a more pronounced reduction in alkyl peak integration



compared to CD₃OD when DA was used (**Figure S6**), suggesting the onset of alkyl chain association in aqueous environments consistent with the viscosity data. A gradual increase in the glass transition temperature (T_g) of the copolymers was observed by differential scanning calorimetry (DSC), ranging from 86 °C for HA to 118 °C for DDA (**Figure 3d**). This is also consistent with restricted chain mobility due to stronger alkyl chain association in the bulk state. Note that significantly increased T_g in P(AA-*co*-DDA), combined with T_2 reduction, suggests that strong alkyl-alkyl association can hinder backbone dynamics in bulk and also in solution. AFM images of silicon wafers coated with the P(AA-*co*-*x*A) showed continuous and uniform polymer layers with substantially lower surface roughness compared to the P(AA-*co*-DDA) series with high DDA loading, presumably because formation of large hydrophobic domains was suppressed under lower hydrophobicity (**Figure S7**). The RMS roughness showed no systematic dependence on alkyl chain length across the HA–DDA series, indicating that the coating morphology is largely preserved at comparable hydrophobic contents. These results suggest that differences in wettability and friction primarily originate from surface chemistry and interfacial interactions rather than coating topography.

Si-Hy lens coating. The analyses of P(AA-*co*-*x*A)s in aqueous solution and at the interface showed that the extent of surface hydrophobicity and alkyl chain association can be adjusted by the *x*A mol fraction in the copolymer and alkyl length. We applied the P(AA-*co*-*x*A)s produced by free radical polymerization as coating materials for commercial Si-Hy contact lenses. The Si-Hy contact lenses were subjected to solvent pretreatment, followed by polymer coating, surfactant post-treatment, and final rinsing with deionized water. Detailed experimental conditions are provided in the Supporting Information (**Figure S8**). We



determined the water contact angle and friction coefficient of the coated lens surface to evaluate their performance and summarized the results in **Figure 4a-b** and **Table S3**. Previous studies reported friction coefficients of silicone hydrogel contact lenses of $\sim 0.25 - 0.30$ under aqueous lubrication conditions³⁸ and $\sim 0.05 - 0.09$ in blink-model experiments using artificial tear solutions.³⁹ The present coatings exhibit lower friction values of $0.028 - 0.03$, indicating excellent lubrication performance.

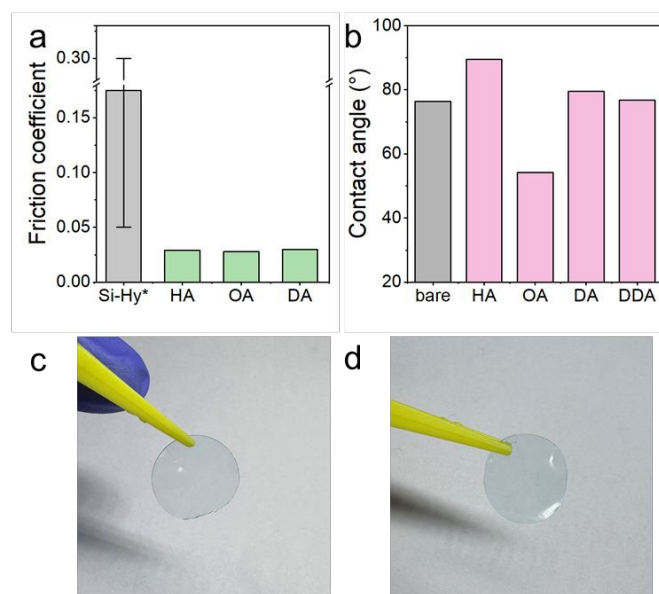


Figure 4. (a) Friction coefficient and (b) water contact angles of P(AA-*co*-*x*A)-coated Si-Hy lens. (c) Image of an uncoated Si-Hy lens and (d) a P(AA-*co*-OA)-coated Si-Hy lens. Friction coefficients of the uncoated Si-Hy lens (*) reported in the literature^{38,39} for silicone hydrogel lenses under aqueous or tear-like lubrication conditions were used. .

Among the copolymers tested, P(AA-*co*-OA) showed the most pronounced improvement, exhibiting both the lowest contact angle and the lowest friction coefficient. It is striking that P(AA-*co*-OA) outperforms the other in both properties and particularly lowers the



contact angle by 28 - 44°. Given that P(AA-*co*-HA) exhibited higher surface tension than P(AA-*co*-OA) and noticeably lower viscosity at 0.05 wt%, the result suggests that the short hexyl side-chain in HA cannot provide sufficient hydrophobic interaction with the silicone domain on the lens surface to be anchored stably. Increasing the alkyl length beyond OA also seems detrimental, as a longer alkyl chain may favor the air-polymer interface over the polymer-lens, resulting in a higher contact angle. We also note that the intermediate mobility of the OA side chain may provide more fluidic surface environments than those of the longer alkyl analogues, allowing the side chain to move and flip, thereby enhancing lubrication properties. The P(AA-*co*-OA)-coated lens maintained optical clarity without visible defects and could not be distinguished from the uncoated lens (**Figure 4b** and **c**), demonstrating that the thin polymer coating does not compromise transparency.

CONCLUSION

We explored hydrophobically modified PAAs as coating materials on Si-Hy contact lenses. Hydrophobic alkyl side chains were incorporated into PAA by copolymerizing AA with x A bearing different alkyl groups as a hydrophobic comonomer to match the hydrophobic surface characteristics of the silicone polymer, thereby improving surface performance and wearer comfort. By systematically varying the copolymer molecular weight, x A fraction, and alkyl side-chain length, we found that, while the hydrophobic interaction became stronger with increasing x A mol fraction and alkyl length, segregation of the alkyl chains, followed by intermolecular association and strong adsorption at the air-water interface, was observed above threshold. On a Si-Hy lens surface, P(AA-*co*-OA) bearing the intermediate alkyl side chain outperformed the other copolymers at the same HLB but with different alkyl lengths. Our



findings suggest that a hydrophilic-rich amphiphilic copolymer that can provide a highly fluidic surface and adequately interact with the silicone polymer surface without noticeable domain formation can preserve optical transparency while enhancing surface wettability and lubrication, making them promising candidates for next-generation contact lens coatings.

AUTHOR CONTRIBUTION

Jimin Yoo: Conceptualization, Investigation, Visualization, Writing – original draft. Writing – review & editing. Minjoong Shin: Investigation, Methodology. Si Chul Rho: Funding acquisition, Supervision. Minho Jung: Conceptualization, Supervision, Writing – review & editing. Myungeun Seo: Conceptualization, Supervision, Funding acquisition, Writing – review & editing.

NOTES

The authors declare no competing financial interest.

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DATA AVAILABILITY STATEMENT

The data supporting this article have been included as part of the Supplementary Information.

Supplementary information: materials and methods including experimental details, Tables S1

– S3 including RMS roughness of the coated surfaces, relaxation times, and lens coating

performances, Figures S1 – S7 including NMR spectra of decyl acrylate and P(AA-co-xA)

copolymers, contact angle data, and AFM images. See DOI: [URL – format <https://doi.org/DOI>]

